Theoretical analysis of the concentration and distribution of carriers in YBa₂Cu₃O_{7-x}

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An ionic model and the total-energy method are used to calculate the density and distribution of carriers in $YBa_2Cu_3O_{7-x}$ systems and, in particular, to understand the relationship between the density of carriers and the density of oxygen vacancies in $YBa_2Cu_3O_{7-x}$. It is shown that the density of carriers is normally 1-x per unit cell and the carriers are holes on the oxygen ions in the CuO₂ planes. In addition, for the 1+ valence copper ions Cu⁺ present in the CuO_{\square} plane, the density of Cu⁺ is normally x per unit cell.

In order to understand the high- T_c superconductivity in $YBa_2Cu_3O_{7-x}$, a study of the concentration and spatial distribution of carriers in the lattice is very important. For x=0, there is one hole in every unit cell,¹ and it is generally accepted that the hole is present in the CuO₂ plane. It follows that in the CuO \square basal plane (\square denotes oxygen vacancy), the valence of copper ions is 2+ and that of oxygen 2-, while in the CuO₂ plane the charge distribution can be represented as $[CuO_2]^{1.5-}$. In another paper by one of the authors and co-workers,² this distribution was presented from the point of view of electrostatic screening. It has been demonstrated by many experiments³ that for $[CuO_2]^{1.5-}$ the holes occur on O^{2-} ions rather than Cu^{2+} ions. This was first proposed by Emery⁴ and has been discussed by many theorists.⁵ This has also been confirmed by the authors⁶ in the calculation of the parameters of the extended Hubbard model for La-Sr-Cu-O and Y-Ba-Cu-O. However, there remains an important question: what is the relationship between the density of carriers and the density of oxygen vacancies for $YBa_2Cu_3O_{7-x}$? In particular, experiments⁷ indicate that the YBa₂Cu₃O_{6.5} is a conductor and has a superconducting transition temperature $T_c \sim 50-60$ K. On the contrary, from the simple chemical model, all of the copper ions have a valence of 2+ and all oxygen 2- and no extra holes; therefore it should be an insulator. From the electronic spectra, in $YBa_2Cu_3O_{7-x}$ with $x \neq 0$, evidence of Cu³⁺ ions has not been observed, but the presence of Cu⁺ has evidently been confirmed. Which sites are Cu⁺ ions on? What is the relationship between Cu⁺ and hole concentration? Recently many experiments and theoretical discussions⁸ have demonstrated that oxygen vacancies are ordered at low temperature in $YBa_2Cu_3O_{7-x}$ ($x \neq 0$). As suggested by Tranquada et al.9 for each oxygen vacancy occurring, only one hole is removed from the CuO_2 plane as a Cu^{2+} ion in CuOplane is converted to Cu^+ . Thus, in YBa₂Cu₃O_{7-x} $(x \neq 0)$ the number of holes per unit cell is equal to 1-x. We feel that the assumption of Tranquada et al. is reasonable, but requires theoretical support.

As the authors emphasized in Ref. 6, because of the electron correlation, especially the on-site electron corre-

lation, a discussion held on the ionic aspect is more realistic for high-temperature copper oxide superconductors. We have used the model developed by Watson and others to treat ionic solids.¹⁰ The Madelung potential is given by

$$V_{i}(r) = \begin{cases} -V_{i}, & r < Q_{i} / - V_{i}, \\ Q_{i} / r, & r > Q_{i} / - V_{i}, \end{cases}$$
(1)

where $-V_i$ is the Madelung potential constant for the electrons on the *i*th ion site, and Q_i is the net charge of the *i*th ion (see Fig. 1).

Based on the density functional theories, the effective single-electron equation can be written as

$$(-\frac{1}{2}\nabla^2 + V_{\text{eff}})A_i = E_i A_i ,$$

$$V_{\text{eff}} = V_{\text{Coulomb}} + V_{\text{ex}} + V_{\text{corr}} .$$

Adding the Madelung contribution, the equation becomes

$$\left[-\frac{1}{2}\nabla^{2} + V_{\text{eff}} + V_{i}(r)\right]A_{i} = E_{i}A_{i}$$
 (2)



FIG. 1. The Madelung potential (solid line) and ionic potential (dashed line) (a) for Cu^{2+} ion and (b) for O^{2-} ion. $(-V_i)$ is the electrostatic interaction for the electrons at the site of the *i*th ion due to the rest of the ions in the crystals. (Q_i) is the net charge of the *i*th ion, i.e., for $Cu^{2+}Q_i = 2$ and for $O^{2-}Q_i = -2$.

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It can be solved self-consistently, and then we can calculate the total energy involving exchange energy and correlation energy (the sum of the total energy of the ions in a unit cell is the total energy of the system per unit cell). Such a calculation can be called a quasichemical method, which emphasizes the difference of energies for different ionic charge distributions.

We calculated the following situations.

(i) x=0. This situation has been presented in Ref. 6. The possible distributions are

(a)
$$-Cu^{3+}O^{2-}\Box -O^{2-}Ba^{2+} -Cu^{2+}(O^{2-})_2 -Y^{3+}\Box -(O^{2-})_2 Cu^{2+} -Ba^{2+}O^{2-} -\cdots$$
,
(b) $-Cu^{2+}O^{2-}\Box -O^{2-}Ba^{2+} -Cu^{2.5+}(O^{2-})_2 -Y^{3+}\Box -(O^{2-})_2 Cu^{2.5+} -Ba^{2+}O^{2-} -\cdots$,
(c) $-Cu^{2+}O^{2-}\Box -O^{2-}Ba^{2+} -Cu^{2+}(O^{1.75-})_2 -Y^{3+}\Box -(O^{1.75-})_2 Cu^{2+} -Ba^{2+}O^{2-} -\cdots$,

where the symbol "—" denotes that the layers are subsequently arranged along the *c* axis, and the ions between two "—" symbols denote the ions in one layer, for example, $-Cu^{3+}O^{2-}\Box$ — in the basal layer and so on. We assume that in the basal plane the oxygen vacancies are ful-





FIG. 2. The three-dimensional crystal structure of $YBa_2Cu_3O_7$ (one unit cell) and $Y_2Ba_4Cu_6O_{13}$ (double unit cells).

ly ordered into the *a* axis and there are a series of parallel Cu-O-Cu-O-··· one-dimensional chains along the b axis. In distribution (a), holes are totally situated on the one-dimensional chains. The calculation results show that the total energy of this distribution is too high. In this paper we mainly discuss the (b) and (c) distributions. For (b), holes are present on Cu^{2+} ion sites in the CuO_2 planes. There are two possible expressions: One is that half of the Cu ions remain on the 2+ valence; the other half of the Cu^{2+} ions are converted to Cu^{3+} . The other expression is that all of the Cu^{2+} ions uniformly become $Cu^{2.5+}$. In the latter situation holes are shared by the Cu^{2+} ions. In calculating the electron correlations of the two expressions there is a great difference between them, but for the quasichemical method used in this paper, there is not much difference. For (c), holes are present on O^{2-} ion sites in the CuO₂ plane. There are also two types of expressions: One is that $\frac{1}{4}$ of O(2) and O(3) become O^{1-} ; the rest retain a formal valence O^{2-} . The other is that holes are uniformly distributed among oxygen ions and every O(2) and O(3) becomes $O^{1.75-}$. In the discussion of total energy by the quasichemical method, these two expressions are not much different either.

(ii) x=0.5. We assume that the oxygen vacancies are fully ordered as illustrated in Fig. 2.¹¹ In the basal plane, two types of one-dimensional chains are possible,

$$-Cu-O-Cu-O-Cu-O-Cu-O-\cdots,$$

The former is called full chain, the latter empty chain. We analyze two different charge distributions,



FIG. 3. Charge distribution with three full-chain orderings and one empty-chain ordering in the basal plane of $YBa_2Cu_3O_{6.75}$.

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(a)
$$-Cu^{2+}(O^{2-})_{0.5}\Box_{0.5} -O^{2-}Ba^{2+} -Cu^{2+}(O^{2-})_2 -Y^{3+}\Box -(O^{2-})_2 Cu^{2+} -Ba^{2+}O^{2-} \cdots$$
,
(b) $-(Cu^{2+})_{0.5}(Cu^{+})_{0.5}(O^{2-})_{0.5}\Box_{0.5} -O^{2-}Ba^{2+} -Cu^{2+}(O^{2-})_{1.75}(O^{-})_{0.25} -Y^{3+}\Box -(O^{-})_{0.25}(O^{2-})_{1.75}Cu^{2+} -Ba^{2+}O^{2-} -\cdots$.

For the (a) distribution, all of the Cu atoms are Cu^{2+} , and all of the O atoms are O^{2-} . It is consistent with the results of the simple chemical model that there is no extra hole, i.e., no O^- . For (b), Cu atoms on the twofold-coordinated sites of the empty chains have a valence of +, and fourfold-coordinated Cu atoms on the full chains have the formal valence 2+. The extra holes O^- are presented in the CuO_2 planes with each unit cell having 0.5 hole, which is equal to the number of Cu^+ ions.

(iii) x=0.25, also assuming the oxygen-vacancy ordering.¹¹ There are several different distributions for the $\frac{3}{4}$ oxygen occupation in the basal plane.

(a) For four unit cells, there are three full chains and one empty chain in the basal plane (see Fig. 3). It is similar to the situation of x=0.5 with fourfold-coordinated Cu²⁺ on the full chains and twofold-coordinate Cu⁺ on the empty chains. The charge distribution is

$$-(Cu^{2+})_{0.75}(Cu^{+})_{0.25}(O^{-2})_{0.75}\Box_{0.25}-O^{2-}Ba^{2+}-Cu^{2+}(O^{-2})_{1.625}(O^{-})_{0.375}-Y^{3+}\Box$$

$$-(O^{-})_{0.375}(O^{2-})_{1.625}Cu^{2+}-Ba^{2+}O^{2-}-\cdots,$$

where the number of O^- is 0.75 within the CuO₂ planes in every unit cell, and thus 0.375 per CuO₂ plane, and 0.25 Cu⁺ per basal plane.

(b) For double unit cells, there are one full chain and one half-full chain in the basal plane. For the half-full chain, there are many different distributions with n oxygens locating side by side and n oxygen vacancies side by side when n varies (n = 1, 2, ...):

In addition to fourfold- and twofold-coordinated Cu atoms, there are Cu atoms in the threefold-coordinated sites of the half-full chains. The threefold-coordinated Cu sits between oxygen segment and oxygen vacancy segment. Then it can be represented as

O---Cu--O fourfold coordination Cu with a valence of +2, \Box ---Cu-- \Box twofold coordination Cu with a valence of +1, O---Cu-- \Box threefold coordination Cu with a valence of +2.

The calculations indicate that it will be unstable if threefold-coordinated Cu has the valence Cu^+ . This is because on such site the Madelung potential is too high for Cu^+ ions. It is similar to the unstable state of free O^{2^-} ions. Thus, we take threefold-coordinated Cu with 2^+ valence. The ordered half-full chains then become

 $-O^{2^{-}}-Cu^{2^{+}}-\Box-Cu^{2^{+}}-\Box-Cu^{2^{+}}-\Box-Cu^{2^{+}}-\Box$

The charge distribution is

$$-Cu^{2+}(O^{2-})_{0.75}\Box_{0.25}-O^{2-}Ba^{2+}-Cu^{2+}(O^{2-})_{1.75}(O^{-})_{0.25}-Y^{3+}\Box-(O^{-})_{0.25}(O^{2-})_{1.75}Cu^{2+}-Ba^{2+}O^{2-}-\cdots,$$

where in CuO_2 planes the number of O^{-1} is 0.5 but not 0.75, and there is no Cu^+ ion.

$$O^{2^{-}}-Cu^{2^{+}}-O^{2^{-}}-Cu^{2^{+}}-\Box-Cu^{+}-\Box-Cu^{2^{+}}-O^{2^{-}}-\cdots, n=2$$
.

The charge distribution is

$$(Cu^{2+})_{0.875}(Cu^{+})_{0.125}(O^{2-})_{0.75}\Box_{0.25}-O^{2-}Ba^{2+}-Cu^{2+}(O^{2-})_{1.6875}(O^{-})_{0.3125}-Y^{3+}\Box$$

-(O⁻)_{0.3125}(O²⁻)_{1.6875}Cu²⁺-Ba²⁺O²⁻-····,

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	Charge distribution	E _{tot} (a.u.)/unit cell
x=0	(a) $O(1234)^{2^{-}}Cu(1)^{3^{+}}Cu(2)^{2^{+}}$	- 24367 8317
	(b) $O(1,2,3,4)^2 - Cn(1)^{2+} Cn(2)^{2+} Cn(2)^{3+}$	- 24307.8317
	(a) $O(1)^{2-}O(2,2) = 1.75O(4)^{2-}O(2)^{2+}O(2)^{2+}$	24370.2837
	(c) $O(1) O(2,5)_4 = O(2)_2^- O(2)_2^- O(4)_2^- O(4)_2^- O(2)_2^+$ (d) $O(1)_2^- O(2)_2^- O(2)_2^- O(4)_2^- O(4)_2^- O(4)_2^+$	-24377.3890
	(a) $O(1)^2 O(2)_2^2 O(3) O(3)^2 O(4)_2^2 Cu(1)^2 Cu(2)_2^2$	-24377.400
	(e) $O(1) O(2,3,4)_{5}^{2} Cu(1)^{2} Cu(2)_{2}^{2}$	-24374.9274
x=0.25	(a) $O(1)_{0.75}^{2-}O(2)_{2}^{2-}O(3)_{1.25}^{2-}O(3)_{0.75}^{}O(4)_{2}^{2-}Cu(1)_{0.25}^{++}Cu(1)_{0.75}^{2+}Cu(2)_{2}^{2+}$	-24360.3477
	(b) $O(1)_{0.75}^{2-}O(2)_{2}^{2-}O(3)_{1.5}^{2-}O(3)_{0.5}^{-}O(4)_{2}^{2-}Cu(1)^{2+}Cu(2)_{2}^{2+}$	-24357.3880
	(c) $O(1)_{0.75}^{2-}O(2)_{2}^{2-}O(3)_{1.375}^{2-}O(3)_{0.625}^{2-}O(4)_{2}^{2-}Cu(1)_{0.125}^{+}Cu(1)_{0.875}^{2+}Cu(2)_{2}^{2+}$	-24358.7344
x=0.5	(a) $O(1)_{c} c^{2-} O(2,3,4) c^{2-} Cu(1)^{2+} Cu(2) c^{2+}$	- 24333 2215
	(b) $O(1)_{0.5}^{2-}O(2)_{2}^{2-}O(3)_{0.5}^{-}O(3)_{1.5}^{2-}O(4)_{2}^{2-}Cu(1)_{0.5}^{+}Cu(1)_{0.5}^{2+}Cu(2)_{2}^{2+}$	-24343.2499

TABLE I. The total energy of $YBa_2Cu_3O_{7-x}$ in the ionic model.

where in CuO₂ planes the number of O⁻ is 0.625 and that of Cu⁺ 0.125 in basal plane. Note that for distribution (a), it corresponds to $n = \infty$; when $n \neq \infty$, the number of both O⁻ and Cu⁺ ions are smaller than that in the (a) distribution.

The results of calculations are presented in Table I.

(i) For x=0, the calculations indicate that the total energy of distribution (c) is the lowest. This confirms the model previously proposed by Emery.⁴ The carriers are holes and are present in the CuO₂ planes. In this situation, in the CuO₂ planes, Cu atoms are Cu²⁺ with $\frac{1}{2}$ spin, the holes are present on the oxygen ions, and they are the conductive carriers. There is one hole per unit cell. To compare with the experimental data of the electronic spectra, we calculate the transition energy of O 1s of this model, i.e., the energy of the

$$O^{-}(1s^{2}2s^{2}2p^{5}) + hv \rightarrow O^{-}(1s^{1}2s^{2}2p^{6})$$

transition process. We obtain $hv \sim 525$ eV (see Table II), which is consistent with the experimental value 528 eV (Ref. 3). This supports the idea that holes are present on

the oxygen sites.

(ii) For x=0.5, it shows that the total energy of distribution (b) is the lowest. This confirms the model proposed by Tranquada and co-workers. In this situation, there is one oxygen vacancy in the basal plane in the "unit" $Y_2Ba_4Cu_6O_{14}$. For the two removed holes, one is removed from the basal plane leading from Cu^{2+} to Cu^+ and another is removed from the CuO₂ planes. That is, in $YBa_2Cu_3O_{7-x}$ (x=0.5) the number of carriers is 0.5 per unit cell. The charge distribution in CuO^{\Box} basal plane is illustrated in Fig. 4.

(iii) For x=0.25, it shows that the total energy of distribution (a) is the lowest. Distribution (b) has half-full chains, and its energy is somewhat higher. In the experimental samples, it might be predicted that distribution (a) retain dominant, mixing with distribution (b).

(iv) Because the Madelung potential is mainly determined by the near-neighbor charge distribution, in addition to the preceding calculations, we also calculated some localized situations and discussed the stability of different ionic valences. These calculations indicate that the following results are primarily correct: In the pyram-

$\begin{array}{ccc} O(3)^{-}(1s^{2}2s^{2}2p^{5}) & \xrightarrow{+h_{V}} & O(3)^{-}(1s^{1}2s^{2}2p^{6}) \\ & & & \\ \hline & & O^{-1}(1s^{2}) & & O^{-1}(1s^{1}) \end{array}$			P denotes the percentage on different sites of O^- .		
	$V_{ m Madelung}$	Р	$E_t(1s^2)$ (a.u.)	$E_t(1s^1)$ (a.u.)	Δ (eV)
O _{6.511}	-0.70055	50%	- 80.4159	-61.0830	525.85
	-0.50969	50%	- 79.0298	- 59.6533	527.04
O _{6.75}	-0.86273	$\frac{2}{3}$	-81.4261	-62.1241	525.01
	-0.53505	$\frac{1}{6}$	-79.2258	- 59.8549	526.89
	-0.49286	$\frac{1}{6}$	-78.8973	- 59.5173	527.14
O ₇₁₁	-0.75687	100%	- 80.7840	-61.4631	525.53

TABLE II. The calculated transition energy of O 1s in the ionic model.



FIG. 4. Charge distribution in the basal plane of $YBa_2Cu_3O_{6.5}$.

idlike or the fivefold-coordinated CuO_2 plane, the Cu ions have a valence of 2+ and extra holes occur on the O^{2-} sites. In the basal plane, the fourfold-coordinated Cu ions are also 2+ while in the oxygen-deficient basal plane the twofold Cu ions have a valence of 1+. On the halffull chain, the copper ions on threefold-coordinates sites remain 2+ valence. We believe that this conclusion should be common for the layered perovskite copperoxide superconductors besides YBa₂Cu₃O_{7-x}.

Finally, we wish to emphasize that a series of experiments¹² indicate that the major factor contributing to the superconducting transition temperature is the concentration of carriers in the CuO_2 planes. Therefore it is very important to analyze the relationships among the carrier concentration, the chemical valence of copper and oxygen ions, and the oxygen vacancy concentration. We plan to use the quasichemical model or the ionic model presented by the authors to discuss more situations.

- ¹A. J. Nazzal et al., Physica C 153-155, 1367 (1988); H. Verweij, Solid State Commun. 64, 1213 (1987); N. P. Ong et al., Phys. Rev. B 35, 8807 (1987); Z. Z. Wang et al., ibid. 36, 7222 (1987); A. T. Fiory and G. S. Grader, ibid. 38, 9198 (1988); T. Penney et al., ibid. 38, 2918 (1988); G. S. Grader, P. K. Gallagher, and A. T. Fiory, ibid. 38, 844 (1988); M. F. Hundley, A. Zettl, A. Stacy, and M. L. Cohen ibid. 35, 8800 (1987); L. Forro et al., Physica C 153-155, 1357 (1988); S. W. Tozer et al., Phys. Rev. Lett. 59, 1768 (1987).
- ²Z. Z. Gan, Yuan-dong Dai, and Han Rushan, Int. J. Mod. Phys. 1, 593 (1987).
- ³N. N. Nucker et al., Phys. Rev. B 37, 5158 (1988); J. M. Tranquada, S. M. Heald, and A. R. Moodenbaugh, *ibid.* 36, 5263 (1987); J. M. Tranquada, S. M. Heald, A. R. Moodenbaugh, and M. Suenaga, *ibid.* 35, 7187 (1987); Z. X. Shen et al., *ibid.* 36, 8414 (1987); J. A. Yarmoff et al., *ibid.* 36, 3967 (1987); A. Bianconi et al., Solid State Commun. 63, 1009 (1987); D. Sondericker, Z. Fu, D. C. Johnston, and W. Eberhardt, Phys. Rev. B 36, 3983 (1987); J. C. Fuggle et al., *ibid.* 37, 123 (1988).
- ⁴V. J. Emery, Phys. Rev. Lett. 58, 2794 (1987).
- ⁵E. B. Stechel and D. R. Jennison, Phys. Rev. B 38, 8873 (1988);
 Z. Zaanen, A. T. Paxton, O. Jepsen, and O. K. Anderson, Phys. Rev. Lett. 60, 2685 (1988); A. G. Khachaturyan and J.

W. Morris, Jr., *ibid.* 59, 2776 (1987); 61, 215 (1988); L. T.
Wille, A. Berera, and D. de Fontaine, *ibid.* 60, 1065 (1988); L.
T. Wille and D. de Fontaine, Phys. Rev. B 37, 2227 (1988); R.
S. Markiewicz, *ibid.* 38, 5010 (1988); D. de Fontaine, L. T.
Wille, and S. C. Moss, *ibid.* 36, 5709 (1987).

- ⁶Han Rushan, C. K. Chew, K. K. Phua, and Z. Z. Gan, Phys. Rev. B **39**, 9200 (1989).
- ⁷R. J. Cava *et al.*, Nature **329**, 423 (1987); Phys. Rev. B **36**, 5719 (1987).
- ⁸D. J. Werder, C. H. Chen, R. J. Cava, and B. Batlogg, Phys. Rev. B 37, 2317 (1988); 38, 5130 (1988); C. Chaillout *et al.*, *ibid.* 36, 7118 (1987); H. W. Zandbergen, G. van Tendeloo, T. Okabe, and S. Amelinckx, Phys. Status Solidi A 103, 45 (1987); R. M. Fleming *et al.*, Phys. Rev. B 37, 7920 (1988).
- ⁹J. M. Tranquada, S. M. Heald, A. R. Moodenbaugh, and Youwen Xu, Phys. Rev. B 38, 8893 (1988).
- ¹⁰R. E. Watson, Phys. Rev. **111**, 1108 (1958); R. G. Gordon and Y. S. Kim, J. Chem. Phys. **56**, 3122 (1972); L. L. Boyer *et al.*, Phys. Rev. Lett. **54**, 1940 (1985).
- ¹¹J. B. Goodenough and A. Manthiram, Int. J. Mod. Phys. B 2, 379 (1988).
- ¹²Y. Tokura, J. B. Torrance, T. C. Huang, and A. I. Nazzal, Phys. Rev. B **38**, 7156 (1988); Y. J. Uemura *et al.*, *ibid.* **38**, 909 (1988).