Photoemission study of $Bi_2(Sr,Ca)_3Cu_2O_y$

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The electronic structure of $Bi_2(Sr,Ca)_3Cu_2O_y$ has been studied by photoemission and electronenergy-loss spectroscopy (EELS). From analysis of the Cu 2p core-level photoemission spectra, the Cu 3d-O2p bond in the CuO₂ layer is found to be more ionic than that in La_{2-x}Sr_xCuO₄ and YBa₂Cu₃O_y, which leads to a weaker superexchange coupling between the Cu²⁺ spins within the layer. Valence-band photoemission and EELS spectra suggest a significant shift of the Bi 6scontribution from the deep-lying Bi 6s-O2p bonding states to the antibonding states in the O 2p band region as compared to the results of band-structure calculations on Bi₂Sr₂CaCu₂O₈. This is probably due to excess oxygen atoms around Bi, which may raise the Bi 6s orbital energy and provide the system with holes.

I. INTRODUCTION

Knowledge of the electronic structure is essentially important in studying the mechanism of high- T_c superconductivity in the layered perovskite oxides. So far, highenergy spectroscopic studies¹ such as photoemission spectroscopy, 2^{-4} electron-energy-loss spectroscopy (EELS), ⁵ and x-ray-absorption spectroscopy⁶ on the La-Sr-Cu-O and Y-Ba-Cu-O and related systems have revealed that Cu atoms are at most divalent and that the Cu 3d electrons are nearly localized due to strong electron correlation and that doped holes enter the O 2p band rather than the Cu 3d. Starting from this picture, various spin-fluctuation- $^{7-10}$ and charge-fluctuation-mediated¹¹ pairing mechanisms have been proposed. The recent discovery of high- T_c superconductivity in the Bi-Sr-Ca-Cu-O (Ref. 12) and Ti-Ba-Ca-Cu-O (Ref. 13) systems naturally raises the question of whether or not the same picture holds for the new systems and can explain the similarities and differences between the new and previous systems.

In this paper, we have studied $Bi_2(Sr,Ca)_3Cu_2O_y$ by photoemission spectroscopy and EELS. The results indicate that the Cu 3d - O2p bond within the CuO₂ layer is more ionic than in the La-Sr-Cu-O and Y-Ba-Cu-O systems. Further, a significant Bi 6s contribution is found in the valence-band region. We will discuss implications of these results on the character of doped holes and also on the crystal structure and oxygen content of $Bi_2(Sr, Ca)_3Cu_2O_y$ which have not been settled yet.

II. EXPERIMENT

Sintered Bi₂(Sr,Ca)₃Cu₂O_y samples were prepared by a solid-state reaction as described in Ref. 14. Singlecrystal Bi₂(Sr,Ca)₃Cu₂O_y was grown by the travelingsolvent floating-zone method¹⁵ and showed a T_c of ~90 K. Its chemical composition was determined to be Bi_{2,2}- $Sr_{1.8}Ca_{1.0}Cu_2O_y$ by electron microprobe analysis. A single crystal of $La_{1.92}Sr_{0.08}CuO_{4-\delta}$ was grown from the melt by using CuO as a flux.

The samples were introduced into the spectrometer with a base pressure of $\sim 1 \times 10^{-10}$ Torr via an airlock entry. Clean surfaces were obtained by scraping *in situ* with a diamond file. In order to minimize possible surface degradation,¹⁶ the samples were cooled to liquid-nitrogen temperature during the measurements. X-ray and ultraviolet photoemission spectroscopy (XPS and UPS) spectra were excited with a Mg x-ray source (hv=1253.6 eV) and a He resonance lamp (hv=21.2 eV and 40.8 eV), and EELS spectra with a grazing-incidence electron gun. Electrons were collected using a double-pass cylindricalmirror analyzer. The total resolution of XPS, UPS, and EELS was ~ 0.9 , ~ 0.15 , and ~ 0.7 eV, respectively. Signals due to the satellites of the x-ray source were numerically subtracted.

Spectra of the single-crystal La_{1.92}Sr_{0.08}CuO_{4- δ} were obviously of better quality than those of sintered samples as judged from the single-peaked O 1s core-level XPS spectrum and the weakness of the UPS feature of unknown origin at a binding energy $E_B \sim 9 \text{ eV}$.¹⁶ As for Bi₂(Sr,Ca)₃Cu₂O_y, spectra were virtually identical between the single-crystal and sintered samples, except for Cu deficiency in the latter as measured by core-level XPS intensities, and both were found to be clean from the O 1s XPS and valence-band UPS. All spectra presented in this paper are those of the single crystals.

III. RESULTS AND DISCUSSION

A. Cu-O bonds and superexchange interaction within the CuO₂ layer

As shown in Fig. 1, the Cu 2p XPS spectrum of Bi₂(Sr,Ca)₃Cu₂O_{ν} exhibits satellite structures charac-

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FIG. 1. Cu 2p core-level XPS spectra of $Bi_2(Sr,Ca)_3Cu_3O_y$ and $La_{1,92}Sr_{0.08}CuO_{4-\delta}$.

teristic of divalent Cu as in the case of La_{2-x}Sr_xCu- O_4 .¹⁻⁴ The figure also shows that the intensity of the satellite $(2p3d^9)$, where 2p denotes a Cu 2p core hole) relative to that of the main peak $(2p3d^{10}L)$, where L denotes an O 2p ligand hole) is higher for $Bi_2(Sr,Ca)_3Cu_2O_v$ than for $La_{2-x}Sr_{x}CuO_{4}$. The relative intensities I_{s}/I_{m} and splittings ΔE_{ms} of the main (m) and satellite (s) peaks of these compounds as well as those of $YBa_2Cu_3O_{\nu}$ (Ref. 17) are listed in Table I.¹⁸ From these values, one can estimate following the procedure of van der Laan, Westra, Haas, and Sawatzky¹⁹ the $d^9 \rightarrow d^{10}\underline{L}$ charge transfer energy, Δ , the Cu 3*d*-O 2*p* transfer integral, $T[\equiv -\sqrt{3}(pd\sigma)]$, and the on-site Coulomb energy between the Cu 2p core and 3d electron, U_{cd} , by referring to Fig. 2, where Δ and U_{cd} are plotted as functions of T for the given I_s/I_m and ΔE_{ms} values of each compound. Considering that U_{cd} should not be much different between these compounds because of its intra-atomic nature nor Tshould because of nearly the same in-plane Cu-O bond lengths, we have estimated T, Δ , and U_{cd} as given in Table I. (Note that the Δ and T thus obtained are those within the CuO₂ layer because of the $x^2 - y^2$ character of the unoccupied d orbital of Cu²⁺.) Thus, we find that a significant increase in Δ , namely, in the ionic character of the in-plane Cu-O bond on going from $La_{2-x}Sr_xCuO_4$ to $YBa_2Cu_3O_{\nu}$ to $Bi_2(Sr,Ca)_3Cu_2O_{\nu}$ (though the small



FIG. 2. $d^9 \rightarrow d^{10}\underline{L}$ charge transfer energy Δ and the on-site Coulomb energy U_{cd} between the Cu 2p core and 3d electrons as functions of the Cu 3d to 2p transfer integral T for the experimental I_s/I_m and ΔE_{ms} values of Bi₂(Sr,Ca)₃Cu₂O_y, YBa₂Cu₃O_y (Ref. 17), and La_{1.92}Sr_{0.08}CuO_{4- δ}.

difference between $La_{2-x}Sr_xCuO_4$ and $YBa_2Cu_3O_y$ may be within the experimental uncertainty).²⁰

Since the in-plane superexchange coupling J between the Cu²⁺ spins decreases as Δ increases, ^{3,21} the present results suggest a decrease in J in going from La-Sr-Cu-O to Y-Ba-Cu-O to Bi-Sr-Ca-Cu-O. Indeed, J's obtained from two-magnon Raman scattering experiments have shown this tendency,²² suggesting a correlation between J or Δ and T_c (including results on Ti-Ba-Ca-Cu-O). If this correlation is real and the superconductivity occurs within the CuO₂ layer, spin fluctuations will be favored over charge fluctuations as the origin of the high- T_c superconductivity, since spin fluctuations generally become dominant with increasing Δ while charge fluctuations become dominant with decreasing Δ . However, one cannot conclude this at this stage because layers other than the CuO_2 layer may be actively participating in the superconductivity. 23,24

TABLE I. Relative intensities I_s/I_m and splittings ΔE_{ms} of the main (m) and satellite (s) peaks in the Cu $2p_{3/2}$ core-level XPS spectra. Electronic-structure parameters, Δ , T, and U_{cd} (for definition, see text) evaluated from the I_s/I_m and ΔE_{ms} for each compound are also given. Energies are in eV.

	Is/Im	ΔE_{ms}	Δ	Т	U _{cd}
$Bi_2(Sr,Ca)_3Cu_2O_v$	0.60	7.9	2.0+0.5	2.4 ± 0.2	8.1 ± 0.2
YBa ₂ Cu ₃ O _v	0.33 ^a	8.5 ^a	0.7 ± 0.2	2.6 ± 0.2	7.3 ± 0.1
$La_{1.92}Sr_{0.08}CuO_{4-\delta}$	0.25	8.2	0.4 ± 0.3	2.7 ± 0.2	6.5 ± 0.1

^aTaken from the spectrum of Meyer et al. (Ref. 17).

B. Valence bands and possible Bi 6s contribution

In Fig. 3, the valence-band XPS and UPS spectra²⁵ of Bi₂(Sr,Ca)₃Cu₂O_y are compared with theoretical spectra calculated using band theory.²⁶ As in the case of La_{2-x}Sr_xCuO₄, ^{16,17,27} the experimental spectra are shifted to higher binding energy relative to the band-structure calculations (in this case as much as $\sim 2 \text{ eV}$). This together with the presence of a valence-band satellite at $E_B \sim 11 \text{ eV}$ demonstrates the inapplicability of one-electron band theory to this material.

Thus, we calculated valence-band spectra using the configuration-interaction cluster model, which considers an octahedral CuO_6 or a pyramidal CuO_5 cluster and takes into account Coulomb and exchange interactions on the Cu site.²⁸ First, we fitted the calculated spectrum to the XPS spectrum of $La_{1.92}Sr_{0.08}CuO_{4-\delta}$, using $\Delta = 1.0$ eV, T = 2.4 eV, [i.e., $(pd\sigma) = -2.2(pd\pi) = -1.4$ eV. Transfer integrals between the Cu 3d and off-plane O 2porbitals were taken to be about 40% of the in-plane values²⁸], and $U_{dd} = 6.5$ eV. (The values are similar but not identical to those estimated from the Cu 2p XPS because of the effective nature of the parameters.²⁹) Agreement with experiment is excellent as shown in Fig. 4. Here, the line shape and intensity of the O 2p band contribution have been derived, respectively, from the hv=21.2-eV UPS spectrum for which the O 2p cross section is by far dominant, and the theoretical O 2p to Cu 3d XPS cross-section ratio.³⁰ Following the Cu 2p XPS results, we used a slightly larger Δ [(1.2 eV) and U_{dd} (7 eV): U_{dd} is expected to increase with U_{cd} (Ref. 29)], and a slightly smaller T (2.3 eV) for $YBa_2Cu_3O_y$ and obtained again nearly perfect agreement with the XPS spectrum in Ref. 17. Accordingly, we took $\Delta = 2.0 \text{ eV}$, T = 2.2 eV, and $U_{dd} = 7.0$ eV for Bi₂(Sr,Ca)₃Cu₂O_y. However, using



FIG. 3. Valence-band XPS and UPS spectra of $Bi_2(Sr, Ca)_3Cu_2O_y$ (dots) compared with band-theoretical spectra (solid curves) calculated by Marksteiner *et al.* (Ref. 26).

these parameters or varying them somewhat around these values, we could not obtain a reasonable fit, particularly in the main-band region with binding energy $E_B < 8$ eV as shown in Fig. 4. (As for the satellite region, there may be a contribution from Bi 6s - O2p bonding states at $E_B \sim 10$ eV, as discussed below; see, e.g., the valence-band XPS spectrum of Bi₂O₃ in the inset of Fig. 5.)

The above discrepancy would be attributed to states other than Cu 3d and O 2p, probably related to Bi 6s states. The presence of Bi 6s character within several eV of the Fermi level (E_F) is supported by the following argument. In Bi oxides, Bi6s - O2p bonding states are formed at $E_B \sim 10$ eV and antibonding states within or just above the O 2p band,³¹ both being occupied in Bi³⁺ oxides. The splitting between the bonding and antibonding states is largely determined by the strong Bi6s - O2phybridization, particularly when the Bi 6s and O 2p levels are close to each other, and is relatively insensitive to the position of the Bi 6s level.³¹ In the EELS spectra shown in Fig. 5, one can see that structures A at $E_{\text{loss}} \lesssim 6 \text{ eV}$ and B at $E_{\rm loss} \sim 10 \text{ eV}$ are observed for Bi₂(Sr,Ca)₃Cu₂O_y but absent for La_{1.92}Sr_{0.08}CuO₄. We attribute these structures to transitions from the Bi6s - O2p bonding and antibonding states to Bi 6p states above E_F . This assignment is obvious for structure B of Bi₂O₃, since its XPS spectrum (inset of Fig. 5) shows a bonding feature at $E_R \sim 10$ eV; structure A is weak because the Bi 6s component is small



FIG. 4. Valence-band XPS spectra of $Bi_2(Sr,Ca)_3Cu_2O_y$ and $La_{1,92}Sr_{0.08}Cu_{4-\delta}$ compared with those calculated using the configuration-interaction cluster model. Shallow core levels have been subtracted from the raw data (dots), yielding satellites at $E_B = 10-12$ eV. The shaded area for $Bi_2(Sr,Ca)_3Cu_3O_y$ indicates discrepancy between the experiment and theory. [For $Bi_2(Sr,Ca)_3Cu_2O_y$, an inelastic background has been subtracted.]



FIG. 5. EELS spectra of $Bi_2(Ca,Sr)_3Cu_2O_y$, $La_{1.92}Sr_{0.08}$ -CuO_{4- δ}, and Bi₂O₃. *P* denotes plasmon satellites, and C transitions from O 2*p* to higher conduction bands. Structures *A* and *B* are assigned, respectively, to transitions from the Bi δs -O 2*p* bonding and antibonding states to the Bi δp states above E_F as discussed in the text. The inset shows an XPS spectrum of Bi₂O₃.

in the antibonding states. In going from Bi_2O_3 to $Bi_2(Sr,Ca)_3Cu_2O_y$, the intensity is shifted from B to A, which we interpret as due to a shift of the Bi 6s weight from the bonding to antibonding states resulting from the upward shift of the Bi 6s level.

Band-structure calculations on Bi₂Sr₂CaCu₂O₈ show no appreciable Bi 6s character in the O 2p band region.^{24,32,33} As the local-density approximation is expected to be reliable in predicting the Bi 6s- and O 2p-level positions and hybridization between them, we suspect that the crystal structure used in the calculations is not sufficiently accurate. Namely, the superstructure of the Bi₂O₂ layers has so far precluded accurate determination of the crystal structure:³⁴ Even the oxygen-coordinated number and geometry for the Bi atom nor the exact oxygen content y has not yet been determined. It is, therefore, probable that the oxygen content is greater than that assumed in the band-structure calculations (y=8) and that the excess oxygens are coordinated to Bi, resulting in the upward shift of the Bi 6s level. Indeed, $v \approx 8.25 \pm 0.1$ has been obtained from thermogravimetry.³⁵ Since charges are balanced with y=8 for valences of Bi³⁺, Sr²⁺, Ca²⁺, Cu²⁺, and O²⁻, excess oxygens are able to supply holes to the otherwise insulating system. (Alternatively, a "self-doping" mechanism has been proposed by



FIG. 6. UPS spectra of $Bi_2(Sr,Ca)_3Cu_2O_y$ and $La_{1.92}Sr_{0.08}$ - $CuO_{4-\delta}$.

Hybertsen and Mattheiss³² and Krakauer and Pickett:³³ Even with y = 8 the bottom of the Bi 6p band is populated by electrons, thus providing holes to the system. Due to the large uncertainty in the crystal structure, however, discussions on such a detail of the band structure would not be quite meaningful.) If there is significant Bi 6s mixing near E_F , the holes may have partially Bi 6s character or a fraction of Bi atoms may be converted to Bi⁴⁺ or Bi⁵⁺, but its possible role in superconductivity is not clear at present.

As shown in Fig. 6, the UPS spectrum of Bi₂(Sr,Ca)₃-Cu₂O_y is different from that of La_{2-x}Sr_xCuO₄ in that it shows a flat region within $\sim 1 \text{ eV}$ of E_F and a clear Fermi edge. [The spectra of YBa₂Cu₃O_y are rather similar to those of La_{2-x}Sr_xCuO₄, although a less clear Fermi edge has been observed for La_{2-x}Sr_xCuO₄ with higher x (Ref. 27) and for YBa₂Cu₃O_y.³⁶ As the above comparison between the XPS spectrum and the cluster calculation has suggested extra spectral intensity within several eV of E_F (Fig. 3), which we have attributed to the effect of Bi 6s, the characteristic behavior of the Bi₂(Sr,Ca)₃Cu₂O_y UPS spectra near E_F might be related to the presence of the Bi 6s character in this region. The fact that the extra intensity identified by XPS is not observed by UPS is consistent with the decrease of the Bi 6s to O 2p cross-section ratio by a factor of $\sim 10^{-3}$ in going from XPS to UPS.³⁷

C. Character of doped oxygen holes

It is now established that the doped holes in the La-Sr-Cu-O and Y-Ba-Cu-O systems are O 2p-like rather Cu 3d-like, ¹⁻⁶ but it is controversial whether the holes are created on the $p\sigma$, ⁷⁻⁹ $p\pi$, ^{10,38} or p_z (Ref. 39) orbitals of oxygen in the CuO₂ layer or on the p_z orbitals of the offplane (i.e., BaO- or LaO-layer) oxygen above the Cu atom.²³ The recent polarized Cu $L_{2,3}$ -edge x-ray absorption study has given microscopic evidence for the presence of off-plane p_z holes.²⁴ Indeed, band-structure calculations on La_{2-x}Sr_xCuO₄ and YBa₂Cu₃O_y have shown that the average energy (center of gravity) of the off-plane O 2p partial density of states (DOS) is higher than that of the in-plane O 2p.^{20,40,41} Although this is not obvious in the band-structure calculations of Bi₂Sr₂CaCu₂O₈,^{32,33,42} if oxygen content is higher than y = 8 and these oxygens are located in or near the Bi₂O₂ layer, the O 2p level in the SrO layer which is adjacent to the Bi₂O₂ layer will be raised and may be preferentially doped with holes.

D. Core levels of Bi, O, and Sr

The XPS spectra of Bi core levels exhibit a slight asymmetry. Although the core-level asymmetry is often due to excitations of conduction electrons across E_F and suggests a finite density of states (DOS) at E_F at the core-level site, 43 the Bi 4f core-level XPS spectrum (Fig. 7) could be better fitted by assuming an additional component on the high binding-energy side (solid curve) rather than by a Doniach-Sunjić-type⁴³ asymmetric Lorentzian ($\alpha \approx 0.1$, dashed line) which represents the excitations of conduction electrons in a wide band with a sufficiently large Fermi energy. The high binding-energy component is at least partly due to energy loss satellites (see the EELS spectrum in the inset of Fig. 7), and an intrinsic high bindingenergy tail, if it exists, appears to be restricted within a narrow ($\ll 1 \text{ eV}$) range from the main peak. The narrow tail is consistent with a small Fermi energy ($\ll 1 \text{ eV}$) for the doped hole carriers. As for the possibility of Bi mixed valency, the core-level shifts between Bi³⁺ and Bi⁵⁺ in oxides are unfortunately very small⁴⁴ and would not give useful information.



FIG. 7. Core-level XPS spectra of $Bi_2(Sr,Ca)_3Cu_2O_y$ and their line-shape analyses. The dashed curves for Bi 4f is a Doniach-Sunjić line-shape fit. The shaded part for the O 1s spectrum is due to contamination. The inset shows the elastic peak of an EELS spectrum (dots, $E_0 \approx 2 \text{ kV}$) decomposed into the purely elastic part (represented by the EELS spectrum of a wide-gap insulator MnO) and a low energy-loss signal ($E_{loss} \sim 1 \text{ eV}$).

The recent studies on single-crystal $La_{2-x}Sr_{x}CuO_{4}$ (Ref. 16) and well-characterized sintered $YBa_2Cu_3O_{\nu}$ samples⁴³ have revealed that the O 1s XPS spectrum consists of a single peak at $E_B \sim 529$ eV rather than two peaks at $E_B \sim 529$ eV and 531 eV as observed in earlier studies and that the 531-peak is due to surface degradation or grain-boundary contamination. The single peak, however, is considerably broader than the other core levels, suggesting the presence of overlapping multicomponents. Using approximately the same Gaussian width as for the other core levels, the O 1s spectrum could be decomposed into three components (plus one, the shaded area in Fig. 7, due to contamination as judged from its time and filing-sequence dependence) as was done for La-Sr-Cu-O and Y-Ba-Cu-O by Weaver et al.⁴⁵ We assign the lowest binding-energy component to oxygens in the SrO and " Bi_2O_2 " layers, the second one to those in the CuO_2 layers (plus an energy-loss satellite accompanying the first one), and the highest binding-energy component to the energy-loss satellite of the CuO₂-layer oxygen signal. The O 1s binding energies deduced from the band calculation on Bi₂Sr₂CaCu₂O₈ are, on the other hand, such that $Bi_2O_2 > SrO \sim CuO_2$, and cannot explain the relative intensities of the deconvoluted peaks. As band calculations on La₂CuO₄ and YBa₂Cu₃O_v have given O 1s binding energies consistent with experiment,⁴⁵ the discrepancy would be attributed to the crystal structure used in the calculation. We suspect that the excess oxygens in or near the Bi_2O_2 layer raise the O 1s orbital in the same layer most significantly and that in the adjacent SrO layer to a lesser extent, resulting in the ordering of the O 1s binding energies $CuO_2 > Bi_2O_2 \sim SrO_2$. Attempts to fit the O 1s spectrum using the Doniach-Sunjić line shape gave unsatisfactory results as in the case of Bi 4f XPS.

The high binding-energy component of the Sr 3d corelevel XPS (Fig. 7) appears too intense (~30% of the main component) to be assigned to an energy-loss satellite. This signal probably arises largely from Sr replacing Ca sites as has been suggested by a recent electron microscopy study.⁴⁶

IV. CONCLUSIONS

We have found an increase in the $d^9 \rightarrow d^{10}\underline{L}$ charge transfer energy Δ or in the Cu–O bond ionicity within the CuO₂ layer in going from La_{2-x}Sr_xCuO₄ to Bi₂(Sr, Ca)₃Cu₂O_y, which leads to a decrease in the in-plane superexchange coupling J. A significant Bi 6s contribution is suggested within several eV of E_F , although bandstructure calculations on Bi₂Sr₂CaCu₂O₈ predict most of the 6s weight to be in the Bi 6s -O 2p bonding states well below E_F . As the atomic arrangement of and around the Bi₂O₂ layer and the oxygen concentration y have not been established yet, we propose that the discrepancy is caused by excess oxygens accommodated in or near the "Bi₂O₂" layers.

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