# Low-temperature heat-capacity study of the  $U_6X$  ( $X \equiv Mn$ , Fe, Co, Ni) compounds

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Measurements of the superconducting- and normal-state heat capacity of  $U_6X$  ( $X \equiv Mn$ , Fe, Co, Ni) compounds have been performed over a temperature range 1 K  $\lesssim T \lesssim 35$  K. The U<sub>6</sub>X compounds have strong renormalizations of the free-carrier effective mass  $m^*$  in the range  $10m_e \lesssim m^* \lesssim 50m_e$  ( $m_e$  =free-electron mass). The unusual magnitude and temperature dependence of the  $U_6X$  heat capacities suggest the presence of high densities of low-energy excitations of undetermined nature. The results are analyzed in terms of models appropriate to heavy-fermion liquids, and anisotropic or strong-coupled superconductors. The  $U_6X$  compounds form a link between relatively low- $m^*$ , high-transition-temperature  $A15$  compounds and the more extreme examples of heavy-fermion superconductors such as UBe<sub>13</sub>, UPt<sub>3</sub>, and CeCuSi<sub>2</sub> for which  $m^* \sim 10^2 m_e$ .

## I. INTRODUCTION

The  $U_6 X$  ( $X \equiv Mn$ , Fe, Co, Ni) phases are among the earliest known U superconductors and include the first examples of superconducting Fe or Mn compounds.<sup>1</sup> In spite of their content of ferromagnetic 3d elements and strong paramagnetism, $2^{-5}$  the highest-known upper critical magnetic field  $H_{c2}$  and superconducting transition temperature  $T_c$  for U compounds are observed within this class of materials.<sup>6</sup> Indeed, these compounds have long been cited as intriguing examples of the puzzling interplay between magnetism and superconductivit Recently,  $U_6$ Fe was observed<sup>11-13</sup> to exhibit anomalou low-temperature properties that were attributed to "heavy fermions"<sup>14</sup> resulting from the hybridization between localized  $5f$  and itinerant conduction electron states.<sup>15</sup>

Measurement of the low-temperature heat capacity has proven to be a very useful method of characterizing the strength and character of the many-body correlations between heavy-fermion quasiparticles.<sup>16</sup> The magnitude and temperature dependence of the low-temperature electronic contribution of the normal-state heat capacity reflect the magnitude of the fermion effective mass  $m^*$ and low-energy structure in the quasiparticle density of states. Measurements at and below  $T_c$  are crucial in proving the bulk nature of the superconducting state and provide useful constraints on parameters characterizing the heavy quasiparticles. $^{\rm l}$ 

We have performed measurements of the lowtemperature heat capacity of the  $U_6X$  ( $X \equiv Mn$ , Fe, Co, Ni) compounds. Analyses of the resulting data indicate that these materials have strong renormalizations of the

fermion mass in the range  $10\lesssim m^*/m_e \lesssim 50$  ( $m_e \equiv$ freeelectron mass). The temperature dependences of the  $U_6X$ heat capacities generally reflect the presence of a high density of low-energy excitations of undetermined nature. The data are consistent with a picture in which the  $U_6X$ compounds are exchange-enhanced paramagnets that form a link between narrow-band transition metals particularly high- $T_c$   $A15$  compounds —and heavyfermion superconductors such as  $UBe_{13}$  (Ref. 17) and  $UPt_3.$ <sup>18</sup>

### II. EXPERIMENTAL DETAILS

Samples were prepared from high-purity starting materials by arc melting in a pure argon atmosphere. A Zr getter was melted several times, followed by several cycles of melting, turning, and remelting of the  $U-X$  charge on a water-cooled copper hearth. A slight excess of U was used in the sample melt in order to suppress the amount of  $UX_2$  phase that is introduced in the peritectic formation of all of the  $U_6X$  compounds.<sup>19</sup> After removal from the arc furnace, the samples were wrapped in Ta foil and sealed in evacuated quartz ampoules and annealed at temperatures just below the peritectics. Details of the sample preparation are summarized in Table I.

Routine x-ray diffraction scans of the  $U_6Mn$ ,  $U_6Co$ , and  $U_6$ Ni materials studied herein were conducted on unannealed powder samples. The diffraction peaks were broad and poorly resolved, but could be completely indexed in terms of the majority  $U_6Mn$ -type structure, <sup>19</sup> although a few weak peaks were explained as due to U oxides or C15-type,  $\bar{UMn}_2$  (in the case of the U<sub>6</sub>Mn sample, only) phase.<sup>19</sup> The short penetration depth of x rays in U

Compound stoichiometry	Sample designation	<b>Starting</b> materials	$V_m^{\;a}$ (cm <sup>3</sup> )	$T_c^{\ b}$ (K)	$\Delta T_c^{\text{c}}$ (K)	Anneal
$U_{6,15}$ Mn	$II-46$	U: Atomergic, 3-9's, No. U022Cl752 Mn: Johnson-Matthey, "puratronic" grade	84.167	2.19	0.55	11d $500 - 700$ °C
$U_{6,10}Fe$	$II-33^d$	U: Alfa Products, 3-9's, No. 100677 Fe: Alfa Products, 3-9's No. 062278	83.703	3.764	0.136	$10 d$ , $660 - 770$ °C
$U_{6.05}Co$	$II-39B$	U: Alfa Products, 3-9's, No. 100677 Co: Jarrell-Ash, "high purity"	83.319	2.08	1.47	14 d, $600 - 760$ °C
$U_{6.05}$ Ni	$II-49B$	U: Alfa Products, 3-9's, No. 100677 Ni: Alfa Products 3-9's	83.838	0.50	0.22	7 d, $500 - 730$ °C

TABLE I. Selected experimental properties of  $U_6X$  samples used in the present study.

<sup>a</sup>After Ref. 20.

<sup>b</sup>Inductively determined. Unannealed powder was prepared from  $C_P$  samples for U<sub>6</sub>Mn, U<sub>6</sub>Co, U<sub>6</sub>Ni; U<sub>6</sub>Fe data was obtained from a piece cut from the  $C_P$  sample.

 $\Omega$  Difference in the temperatures at which the inductive transition signal reached 10% and 90% completion.

<sup>d</sup>Same sample studied in  $C_p$  and  $H_{c2}$  measurements reported in Refs. 11–13.

compounds and the strain broadening expected in powder samples of the  $U_6X$  phases<sup>5</sup> can account for the poor quality of the x-ray data. Broadening of the inductively measured superconducting transitions of unannealed powder samples of  $U_6Mn$ ,  $U_6C$ o, and  $U_6Ni$  (see Table I) has also been observed.

Our  $U_6$ Fe sample No. II-33 is the same one studied in the heat capacity,  $H_{c2}$ , and electrical resistivity measurements reported in Refs. 11—13. X-ray diffraction and metallography studies were conducted on this sample with contributions by Roof and Pereyra of Los Alamos National Laboratory. Sample II-33 was found to be at least 95% single phase from x-ray measurements, although more sensitive metallography work detected small amounts of U oxide inclusions and excess U lodged at grain boundaries.<sup>12,13</sup> Engelhardt<sup>20</sup> and White *et al.*<sup>21</sup> have done extensive work on the effects of stoichiometry and annealing on the  $T_c$ 's of  $U_bX$  compounds. A detailed discussion of impurity phases present in  $U_6X$  compounds is included in a report<sup>5</sup> of magnetization measurements on sample II-33 and other  $U_6X$  materials, and will not be discussed further here.

Heat-capacity measurements were performed using a semiadiabatic heat-pulse calorimeter with a mechanical heat switch. The calorimeter was attached to a  ${}^{3}$ He evaporation refrigerator providing an experimental temperature range 0.85 K  $\leq T \leq 35$  K in applied magnetic fields of zero and 2 kOe. Details of the methods used in analyzing the calorimetric data are given in Sec. III.

Magnetization measurements were performed on a section of the  $U_6Mn$  sample II-46, only. An SHE (Biotechnologies, Inc.) magnetometer was used to cover the temperature range 2 K  $\leq T \leq 300$  K and applied magnetic fields  $0 \le H \le 2.0$  T. Demagnetization and anisotropy effects were neglected in these measurements and the associated analysis.

## III. METHODS OF DATA ANALYSIS

The "strong form" of the third law of thermodynamics implies that the entropy  $S$  can be related to the constantvolume heat capacity  $C_V$  via

$$
S(T) = \int_0^T \frac{C_V}{t} dt
$$
, (1)  
where  $S(0) \equiv 0$ . Most experimental heat-capacity mea-  
surements, on the other hand, yield data for the

constant-pressure heat capacity  $C_p$ . These results can, in principle, be used to calculate the more fundamental quantity  $C_V$  through the relation

$$
C_P - C_V = VT \frac{\alpha_P^2}{\kappa_T} \t\t(2)
$$

where  $\alpha_p$  is the isobaric thermal expansion coefficient and  $\kappa<sub>T</sub>$  is the isothermal compressibility. However, the difference between  $C_p$  and  $C_V$  is usually less than 1% for most metals below room temperature,<sup>22</sup> and is generally neglected in the analysis of experimental results. We will therefore assume in this paper that our  $C_p$  data accurately reflect the behavior of  $C_V$ .

There are several methods of analysis which can be employed to extract important physical parameters from experimental heat capacity data for metals. We have chosen to apply two quite different methods that are reasonably representative of the more familiar models of the thermal behavior of superconductors. We will describe the strengths and weaknesses of each of these methods, and summarize and compare the numerical results that they yield.

#### A. Polynomial method

The results of low-temperature normal-state heatcapacity measurements are frequently analyzed by fitting data to a low-power polynomial of the form

$$
C_{V_n} = \gamma^* T + \beta T^3 + \alpha T^5 \tag{3}
$$

 $\gamma^*$  is the electronic coefficient given by

$$
\gamma^* = \frac{2\pi^2}{3} k_B^2 N^*(E_F) \tag{4}
$$

where  $N^*(E_F)$  is the renormalized electronic density of states at the Fermi energy, and the coefficient  $\beta$  of the low-temperature phonon term is given by

$$
\beta = \frac{12\pi^4 N_A k_B n}{5\theta_B^3} \tag{5}
$$

where  $N_A$  is Avogadro's number,  $\theta_D$  is the Debye temperature and  $n =$  number of atoms per formula unit. The  $\alpha$  coefficient is best considered as a fitting parameter that approximates finite-temperature corrections to the low-temperature behavior of the electron and lattice systems. Some care must be exercised in applying the model of Eqs. (3)—(5) to particular metals. Strictly speaking, Eq. (3) should not be applied at temperatures  $T \gg \theta_p / 50$ (Ref. 23), or in situations where  $\gamma^*$  is effectively "temperature dependent" as is the case in UPt<sub>3</sub>, UBe<sub>13</sub>, CeCu<sub>2</sub>Si<sub>2</sub>, and certain other heavy-fermion materials.<sup>16</sup>

Analyses performed on superconducting-state data must deal with the finite discontinuity  $\Delta C$  in heat capacity at  $T_c$ . This discontinuity is always broadened in laboratory situations due to the presence of coexisting normal and superconducting regions in imperfect sample materials within a temperature interval  $\Delta T_c$  about the ideal transition point. Therefore, we have chosen to initially integrate the experimental data according to Eq. (1) in order to obtain the superconducting-state entropy  $S_{\rm s}(T)$ . This procedure has several advantages. First, since the normal-state entropy  $S_n$  satisfies the relation  $S_n(T_c)$  $=S_{s}(T_{c})$ , the errors introduced by fitting the *integrated*  $C_V$  data near  $T_c$  are minimal. Second, the entropy relation at  $T_c$  provides a useful constraint on the parameters derived from the fits of  $S_n(T)$  conducted over finitetemperature intervals. Third, the entropy difference  $\Delta S = S_n - S_s$  is also required in calculating the behavior of the thermodynamic critical field  $H_c$ . Of course, a realistic extrapolation of  $S<sub>s</sub>(T)$  as  $T\rightarrow 0$  is required, similar to the normal-state case.

The superconducting-state heat capacity is assumed to have the form [see the discussion of Eqs. (27) and (28)]

$$
C_{V_S} = Ae^{-B/T} + \beta T^3 + \alpha T^5 \tag{6}
$$

We note that this form is not strictly valid (especially for strong-coupled materials), but should provide a reasonably accurate extrapolation of  $C_{V_s}$  in the absence of anomalous low-temperature behavior [such as observed for  $(U_{1-x}Th_x)Be_{13}$  alloys (Ref. 24)].

An iterative method was employed to obtain selfconsistent fits for both the normal- and superconductingstate data. The procedure can be broken down into several steps.

(1) "Zeroth-order" approximations  $\alpha = \alpha_0$  and  $\beta = \beta_0$ are obtained from fits of the data to Eq. (3) over limited

temperature intervals extending above  $T_c$  where  $C_{V_n}T^{-1}$ versus  $T^2$  plots are nearly linear.

2) The  $\beta_0$  and  $\alpha_0$  parameters of step (1) are used to cal-<br>ate the electronic contributions  $C_{es}$  to  $C_{V_s}$  where<br> $C_{es} \equiv A_0 e^{-B_0/T} = C_{V_s} - \alpha_0 T^5 - \beta_0 T^3$ . (7) culate the electronic contributions  $C_{es}$  to  $C_{Vs}$  where

$$
C_{es} \equiv A_0 e^{-B_0/T} = C_{V_s} - \alpha_0 T^5 - \beta_0 T^3 \tag{7}
$$

A linear, least-squares fit of the processed data of  $\ln C_{es}$ versus  $T^{-1}$  yields the parameters  $A_0$  and  $B_0$  for "zeroth order."

(3) The parameters  $\alpha_c$ ,  $\beta_0$ ,  $A_0$ , and  $B_0$  are then used to calculate the entropy for  $T = T_{min}$ , the lowest measuring temperature:

$$
S_{\min} = A_0 \int_0^{T_{\min}} \frac{e^{-B_0/t}}{t} dt + \frac{1}{3} \beta_0 T_{\min}^3 + \frac{1}{5} \alpha_0 T_{\min}^5 . \tag{8}
$$

(4) The entropy of the sample is then calculated for  $T > T_{\text{min}}$  using experimental data and  $S_{\text{min}}$ :

$$
S(T) = S_{\min} + \sum_{i=0}^{N-1} \frac{1}{2} \left[ \frac{C_i}{T_i} + \frac{C_{i+1}}{T_{i+1}} \right] (T_{i+1} - T_i) \ . \tag{9}
$$

(5) The  $S(T)$  obtained from step (4) is then used to recalculate the parameters  $\alpha$  and  $\beta$  using<br>  $C_{V_n}(T) - S_n(T) = \frac{2}{3}\beta T^3 + \frac{4}{5}\alpha T^5$  (10)

$$
C_{V_n}(T) - S_n(T) = \frac{2}{3}\beta T^3 + \frac{4}{5}\alpha T^5 \tag{10}
$$

at each temperature experimentally investigated. A inear, least-squares fit of  $(C_{V_n} - S_n)T^{-3}$  versus  $T^2$  yields new values of  $\alpha$  and  $\beta$  as input for step (2) above.

(6) The steps  $(2)$ – $(5)$  are repeated, yielding new values of  $A$ ,  $B$ , and  $S<sub>min</sub>$ , until all parameters converge to within  $0.1\%$ .

(7) The value of  $\gamma^*$  is determined from the final results of step (6) using

$$
\gamma^* T = S_n(T) - \frac{1}{3} \beta T^3 - \frac{1}{5} \alpha T^5 \tag{11}
$$

The thermodynamic critical field can be calculated from

$$
\frac{1}{8\pi}[H_C^2(T) - H_C^2(0)] = -V_m^{-1} \int_0^T \Delta S(t) dt , \qquad (12)
$$

where  $V_m$  is the molar volume,

$$
\frac{H_c^2(0)}{8\pi} = V_m^{-1} \int_0^{T_c} \Delta S(t) dt \t{,} \t(13)
$$

$$
\Delta S(T) = \gamma^* T + \frac{1}{3} \beta T^3 + \frac{1}{5} \alpha T^5 - S(T) , \qquad (14)
$$

and  $S(T)$  is taken from the final iteration's results in Eq. (9).

Several elementary models of superconductivity predict the following approximate form for  $H_c(T)$  at  $T \ll T_c$ (Ref. 25):

$$
H_c(T) = H_c(0)[1 - (T/T_c)^2].
$$
 (15)

The results of Eq. (12) were fitted to Eq. (15) using the method of least squares. The best-fit value of  $H_c(0)$  was then compared to that obtained from the experimental data using the low-temperature extrapolation of Eq. (7) with Eq. (13).

#### B. Phonon moment method

Analytical techniques for extracting important information from the lattice specific-heat data for metals have appeared in recent years.  $26,27$  These are particularly useful in the case of superconductors since the traditional electron-phonon interaction mechanism for superconductivity implies that important physical parameters such as  $T_c$ ,  $\Delta$ , and  $\lambda$  will depend upon the phonon and electronic densities of states. These techniques can be invaluable in situations where inelastic electron tunneling or neutron scattering measurements are difficult or impossible to make. Further, recent model calculations for strongcoupled superconductors have demonstrated that  $T_c$ 's can be quantitatively estimated from a knowledge of various frequency moments  $\langle \omega^n \rangle$  of the phonon spectrum. $26 - 28$ 

The  $\langle \omega^n \rangle$  are defined in terms of the phonon density of states  $F(\omega)$  by

$$
\langle \omega^n \rangle = \int \omega^n F(\omega) d\omega \;, \tag{16}
$$

which is an expression that is difficult to compare with experiment. However, it can be shown that the  $\langle \omega^n \rangle$  can be directly obtained from lattice heat-capacity data without resort to detailed measurements of phonon dispersion curves.<sup>29</sup> Some of the superconducting properties can then be calculated with the aid of the Allen-Dynes formula:

$$
T_c = \frac{f_1 f_2}{1.20} \omega_{\log} \exp \left[ \frac{1.04(1+\lambda)}{\mu^* + 0.62\lambda \mu^* - \lambda} \right].
$$
 (17)

Here, the parameter  $\omega_{\text{log}}$  is a characteristic phonon scaling frequency given by

$$
\omega_{\text{log}} = \exp\left[\frac{2}{\lambda} \int \alpha^2 F(\omega) \ln \omega \, d \ln \omega \right],\tag{18}
$$

and the electron-phonon coupling  $\lambda$  is calculated from

$$
\lambda = 2 \int \alpha^2 F(\omega) \, d \ln \omega \tag{19}
$$

 $\omega_{\text{log}}$  cannot be directly calculated from heat-capacity data without making additional assumptions concerning the frequency dependence of the electron-phonon spectral function  $\alpha^2 F(\omega)$ . Fortunately, previous experiments and analyses<sup>28</sup> have shown that  $\omega_{\text{log}} \approx \overline{\omega}_{\text{log}}$ , where

$$
\overline{\omega}_{\log} \equiv \exp \left( \frac{\int F(\omega) \ln \omega \, d \ln \omega}{\int F(\omega) \, d \ln \omega} \right). \tag{20}
$$

A schematic form for  $F(\omega)$  can be obtained by fitting the frequency moments obtained from heat-capacity data to the following expression:

$$
\langle \omega^n \rangle = \frac{\int F(\omega) \omega^n d\omega}{\int F(\omega) d\omega} \ . \tag{21}
$$

Note that a customary normalization sets  $\int F(\omega) d\omega = 1$ . Using the facts that the functions  $f_1$  and  $f_2$  are near unity for  $\lambda \lesssim 1.5$ , and the Coulomb pseudopotential  $\mu^* \approx 0.1$ for transition metals, one obtains a reasonably quantitative relation between  $T_c$ ,  $\lambda$ , and the heat-capacity data:

$$
T_c \frac{\overline{\omega}_{\log}}{1.2} \exp\left[\frac{1.04(1+\lambda)}{\mu^* + 0.62\lambda\mu^* - \lambda}\right].
$$
 (22)

Although there are no particular assumptions which must be made concerning the detailed behavior of  $F(\omega)$ , in practice it is sufficient to assume a four- to sixparameter model spectrum of the form

$$
F(\omega) = \sum D_i F_D(\omega/\theta_i) , \qquad (23)
$$

where  $F_D$  is an elementary Debye spectrum characterized by a "partial Debye temperature"  $\theta_i$  (Ref. 27):

$$
F_D(\omega/\theta_i) = \begin{cases} 3\omega^2 \theta_i^{-3}, & \omega \le \theta_i \\ 0, & \omega > \theta_i \end{cases}
$$
 (24)

The fitting procedure is constrained by the normalization condition  $\sum D_i = 1$  (equivalent to the fact that the hightemperature lattice heat capacity  $\rightarrow$  3R ), and the entropy constraint on the electronic heat capacity at  $T_c$ . This procedure has been found to yield values of  $\langle \omega^n \rangle$  and  $\bar{\omega}_{\rm log}$  that compare favorably with the results of tunneling and neutron scattering experiments for  $A15$  compounds.<sup>30,31</sup> The procedure is initially aided by the fact hat  $[C(T) - \gamma^*T]T^{-3}$  is an approximate image of the spectral weight  $D(\omega) - \omega^{-2}F(\omega)$  for  $\hbar\omega \approx 5k_B T$  (Refs. 27, 29, and 30). The rms deviations of the data from the best-fit model heat capacities derived from Eq. (23) are  $\sim 1\%$ .

The phonon moment method we have used is subject to a number of important assumptions. All of the contributions to  $C_{Pn}(T)$  except the electronic term  $\gamma^*T$  are assumed to be phononic in origin; otherwise, our procedure may be viewed as a convenient way to analyze the normal-state data over a wide temperature range [where Eq. (3) fails] using only a small number of adjustable parameters. The assumption of a composite Debye spectrum for  $F(\omega)$  ignores anharmonicity. However, considerably more detail and additional assumptions must be ncorporated into this type of analysis in order to approxmate anharmonic effects,  $26.32$  and we have neglected them in the fitting procedure.

TABLE II. Parameters derived from a polynomial analyses of the normal- and superconducting-state heat capacities of  $U_6X$  compounds,

Compound	Temperature interval of fit	(mJ/mol K <sup>2</sup> )	(mJ/mol K <sup>4</sup> )	$\theta_D$ (K)	α (mJ/mol K <sup>6</sup> )	(mJ/mol K)	(K)
$U_6Mn$	$3 K \leq T \leq 5.75 K$	98.81	1.819	195.6	$8.911 \times 10^{-2}$	2252	3.452
$U_6$ Fe	4.2 K $\leq T \leq 10.2$ K	151.7	9.104	114.3	$-3.045 \times 10^{-2}$	4396	5.947
$U_6$ Co	2.6 K $\leq T \leq 5.5$ K	129.7	9.809	111.5	$-8.672 \times 10^{-2}$	3310	3.732
$U_6$ Ni	0.85 K $\leq T \leq 5.0$ K	91.36	2.375	178.9	$\gtrsim$ 0		

## IV. EXPERIMENTAL RESULTS AND ANALYSIS

The experimental data for  $C_P T^{-1}$  are plotted versus  $T^2$  for each of the U<sub>6</sub>X compounds in Figs. 1 and 2. There is marked nonlinear character in these plots at  $T \geq 5$  K, as shown in Fig. 3. This behavior implies that a high density of low-energy excitations (e.g., "soft" phonons or paramagnons) dominates the heat capacity of these materials. The strong negative curvature at modest temperatures in  $C_pT^{-1}$  versus  $T^2$  is very similar to the data for high- $T_c$ , high- $H_{c2}$ , A 15, and Chevrel phase materials.  $26,30,33$  However, the overall magnitudes of the heat capacities of the  $U_{6}X$  compounds are several times larger than many high- $T_c$  materials, and are more than an order of magnitude larger than those of simple metals such as  $Cu.<sup>5,22,23</sup>$ 

#### A. Polynomial analysis

Extreme care must be exercised in the use of the polynomial ("POLY") method to extract physical parameters relevant to the  $U_6X$  materials. Specifically, the POLY analysis must be restricted to a narrow temperature interval above  $T_c$  where the curvature of  $C_{p_n}T^{-1}$  versus  $T^2$  is relatively insignificant. The best-fit parameters from the POLY analysis of the normal- and superconducting-state data are summarized in Table II.

Comparisons of the experimental data and the best-fit predictions of Eqs. (3) and (6) are given in Fig. 3. We note that the transition anomaly at  $T_c$  is relatively broad in the case of  $U_6Mn$ . This is probably due to the poorer quality of U used in preparing this sample and the relatively high vapor pressure of Mn in the melt. Peculiarities in the metallurgy and lattice parameters of  $U_6Mn$ have been noted earlier.<sup>8,20</sup> Unfortunately, a significant transition width introduces uncertainties in the estimation of the jump  $\Delta C_P(T_c)$  in heat capacity at  $T_c$  and the intrinsic behavior of  $H_c(T)$ , not to mention  $T_c$  itself.

We have defined a calorimetric width  $\delta T_c$  of the transition by noting the temperatures of the peak in  $C_{Ps}$  and



FIG. 1. Heat capacity  $C_P$  divided by temperature T vs  $T^2$  for  $U_6$ Fe and  $U_6$ Co.



FIG. 2. Heat capacity  $C_P$  divided by temperature T vs  $T^2$  for  $U_6Mn$  and  $U_6Ni$ .

the intersection of the extrapolations of  $C_{p_n}$  (given by the parameters of Table II) and the quasilinear- $T$  region of the transition anomaly. A minimum value  $\Delta C_p^{\text{min}}$  of  $\Delta C_P(T_c)$  can be estimated from the difference in  $C_{Ps}$  at the peak of the anomaly and  $C_{p_n}$  near the onset of the transition anomaly. This number can be compared to an "ideal" value,  $\Delta C_P^{\text{ideal}}$  determined by extrapolating 'ideal'' value,  $\Delta C_p^{\text{local}}$  determined by extrapolating  $C_{P_5}(T < T_c)$  and  $C_{P_n}(T > T_c)$  into the transition region. An estimate  $T_c$ (cal) for  $T_c$  is obtained from a graphical analysis of a  $C_pT^{-1}$  versus T plot.  $T_c$ (cal) is adjusted until the entropy difference  $\Delta S(T)$  [see Eq. (14)] estimated from the actual data near  $T_c$ (cal) and an idealized, sharp transition anomaly at  $T_c$  (cal), are equal.

Additional estimates of  $T_c$  can be extracted from fits of  $H_c(T)$  to the "parabolic law" of Eq. (15) (yielding " $T_c^{pb}$ "), and by extrapolating the calculated values of  $H<sub>c</sub>(T)$  and



FIG. 3. Heat capacity  $C_P$  divided by temperature T vs  $T^2$  for  $U_6X$  (X=Mn, Fe, Co, Ni) compounds. Experimental data are represented by the points, and fits of the data by the polynomial method are represented by the lines. Details of the fit parameters are given in Table II. The polynomial method provides an excellent fit for the normal-state  $U_6$ Fe data for  $T \lesssim 10$  K, whereas the method is successful only for  $T \lesssim 5.5$  K in the case of the other compounds.

Compound	$T_c$ (cal)	$\delta T_c$	$\Delta C_p^{\rm min}$	$\Delta C_P^{\rm ideal}$	$S_{\rm s}$ $(T_{\rm min})$	$r_{\scriptscriptstyle{\min}}$	$\Delta C_P(T_c)^a$
	(K)	(K)	(mJ/mol K)	(mJ/mol K)	(mJ/mol K)	$(\mathbf{K})$	$\nu^*T$
$U_6Mn$	$2.21 \pm 0.01$	0.275	223.7	344.8	22.255	0.925	1.6
$U_6$ Fe	$3.695 \pm 0.01$	0.250	854.5	1197	7.407	1.110	2.1
$U_6$ Co	$2.29 \pm 0.01$	0.165	368.9	513.0	14.54	0.925	

TABLE III. Additional superconducting-state parameters derived from polynomial analyses of the heat capacities of  $U_{6}X$  compounds.

<sup>a</sup>Calculated using  $\Delta C_P^{\text{ideal}}$  and  $T_c$  (cal), and  $\gamma^*$  (from Table II).

 $\Delta S(T)$  [from Eqs. (12) and (14)] to  $H_c$  or  $\Delta S = 0$  at " $T_c^{th}$ ." Additional parameters, chiefly relevant to the superconducting state, are collected in Table III. The data for  $\Delta C_P(T_c)/\gamma^*T_c$  show evidence for strong coupling (compared to a BCS value of 1.43), particularly in the case of  $U_6$ Fe.

Measurements of the thermodynamic critical field  $H_c(T)$  can also provide additional information concerning the coupling strength and the quality of the fits. Plots of  $H_c(T)$  versus T, as calculated from Eq. (12), are shown in Fig. 4. Various parameters concerning  $H_c(T)$  are summarized in Table IV. The tabulated values of the ratio  $H_c^2(0)/\gamma^*T_c^2$  are to be compared to the BCS ratio, 5.94. These ratios corroborate the evidence for strong-coupling renormalizations of  $\Delta C_p(T_c)/\gamma^*T_c$  values listed in Table III:  $U_6Mn$  appears to be BCS-like, while  $U_6C$  is moderately strong coupled and  $U_6$ Fe is substantially strong coupled. The close agreement between various estimates of  $T_c$  and  $H_c(0)$  for each compound indicates that the fitting procedure is satisfactory. However, we note that the relatively poor quality of the  $U_6Mn$  sample is clearly reflected in the strong positive curvature of  $H_c(T)$  near a poorly-defined  $T_c$ .

Another check of the fitting procedure can be made by comparing the low-temperature behavior of the



FIG. 4. Thermodynamic critical field  $H<sub>c</sub>$  vs temperature T for  $U_6Mn$ ,  $U_6Fe$  and  $U_6Co$ . Points were calculated from the experimental data using Eq. (12), as described in the text. The lines represent fits of the  $H_c(T)$  points to the parabolic form of Eq. (15). Relevant parameters are collected in Table IV.

superconducting-state data  $C_{Ps}$  to the extrapolation formula of Eq. (6). We define the electronic contribution to the superconducting-state heat capacity by

$$
C_{es} \equiv C_{Ps} - \beta T^3 - \alpha T^5 \ . \tag{25}
$$

Plots of  $\ln [C_{es}/\gamma^*T_c]$  versus  $T_c/T$  for  $U_6Mn$ ,  $U_6Fe$ , and U6Co are shown in Fig. 5. These data can be compared to BCS model predictions where

$$
\frac{C_{es}}{\gamma^* T_c} \approx 8.5 \exp(-1.44 T_c / T)
$$
\n(26)

for 2.5  $\lesssim T_c/T \lesssim 6$  (Ref. 25). Comparisons of the data of



FIG. 5. Electronic contribution  $C_{es}$  to the superconductingstate heat capacity divided by the normal-state electronic heat capacity  $\gamma^*T_c$  at the transition temperature  $T_c$  vs  $T_c$  divided by temperature T for  $U_6Mn$ ,  $U_6Fe$ , and  $U_6Co$ . The points were derived from experimental data using Eq. (25), as described in the text. The solid lines represent fits of the points to Eq. (26) over the temperature range 0.9—1.<sup>5</sup> K.

	$T^{th}$ a	$H_c(0)^b$	$T^{pb}$	$H_c(0)^c$	
Compound	(K)	(Oe)	(K)	(Oe)	$V_m H_c^2(0)^d$ $\gamma^*T_c^2$
$U_6Mn$	$2.25 \pm 0.05$	598.1	2.156	595.8	5.97
$U_6Fe$	$3.705 \pm 0.015$	1312	3.746	1317	6.97
$U_6$ Co	$2.30 \pm 0.01$	721.2	2.225	723.3	6.37

TABLE IV. Thermodynamic critical field parameters derived from polynomial analyses of the heat capacities of  $U_{6}X$  compounds.

<sup>a</sup>Estimated from linear extrapolations of  $H_c(T)$  and  $\Delta S(T)$  data and  $T_c$ (cal) from Table III. <sup>b</sup>Calculated from Eq. (12).

<sup>c</sup>Derived from fits of the  $H_c(T)$  data to Eq. (15), as described in the text.

<sup>d</sup>Calculated in Gaussian units using  $T_c^{th}$ , and  $H_c(0)$  from Eq. (12).

Fig. 5 to fits using Eq. (26) are also shown in Fig. 5.

The precise form of the temperature dependence of the superconducting energy gap  $\Delta(T)$  may cause deviations from Eq. (26) near  $T_c$ , and gap anisotropy or strongcoupling effects may cause additional deviations at very low temperatures.<sup>25,34,35</sup> The data for  $U_6Mn$  and  $U_6Co$ are well fitted by Eq. (26) to rather high temperatures [outside the expected range of validity of Eq. (26)], and  $U_6$ Fe exhibits a BCS-like behavior over the expected temperature range. The quality and self-consistency of the fits using Eq. (6) are strongly supported by the results shown in Fig. 5. Lower-temperature measurements of  $C_{P_s}$  for  $U_6 X$  compounds would be desirable in view of recent claims<sup>36,37</sup> that the non-BCS behavior of  $C_{es}$  in  $UBe_{13}$  provides clear evidence for triplet pairing.

#### B. Phonon moment analysis

Results of the fits of the heat-capacity data of  $U_6X$ compounds over a wide temperature range using the phonon moment ("MOM") analysis are summarized in Table V. Plots of  $C_p T^{-1}$  versus T are shown in Fig. 6, along with the behavior calculated for selected "best" fits. The parameters relevant to the model curves shown in Fig. 6 are collected in Table VI. The agreement between the values of  $\gamma^*$  and  $\Theta_D$  from the POLY analysis (see Table II) and the average values of  $\gamma^*$  and  $\Theta(T\rightarrow 0)$  from the MOM analysis (see Table V) is excellent.

Our analyses indicate that the small, systematic differences between the corresponding parameters given in Tables II and V are primarily due to differing estimates of  $S_{\text{min}}$  [see Eq. (8)] obtained in the two methods used to fit the data.  $S_{\text{min}}$  is estimated in the MOM method by fitting the first 5-10 superconducting-state data at fitting the first 5–10 superconducting-state data at  $T > T_{\text{min}}$  to a form  $C_s = \zeta T^3 + \eta T^5$ , yielding  $S_{\text{min}} = \frac{1}{3} \zeta T_{\text{min}}^3 + \eta T_{\text{min}}^5$ . This procedure will generally  $\zeta T_{\min}^3 + \eta T_{\min}^5$ . This procedure will generally overestimate  $S_{\text{min}}$ , leading to systematically higher  $\gamma^*$ values compared to the POLY method. Note also that the POLY method gives a much stronger weight to lower-temperature data due to the reduced size of the fitting interval and strict enforcement of the entropy constraint at  $T_c$ . The MOM method treats all of the data with equal weight. Additional calculations of  $T_c$  and  $H<sub>c</sub>(T)$  within the MOM analysis are in excellent agreement with the POLY results quoted earlier and will not be detailed here.

The qualitative nature (e.g., the relative weight of lowfrequency modes) of the  $D(\omega) \sim \omega^{-2} F(\omega)$  required to fit the heat-capacity data for the  $U_6X$  samples can be inferred from our analyses; whereas it is important to note that several quantitatively different model  $F(\omega)$ 's have been shown previously to provide equally satisfactory fits of the data for A 15 materials.<sup>30</sup> Plots of  $\omega^{-2}F(\omega)$  are shown in Fig. 7, corresponding to the "best" fit models given in Fig. 6 and Table VI. Note that the higher- $T_c$  $U_6$ Fe compound has significantly more weight in the low-frequency end of  $F(\omega)$  than the lower-T<sub>c</sub> U<sub>6</sub>Ni ma-terial. It is interesting that experimental data for  $\gamma^*$  and  $\Theta$  (T $\rightarrow$ 0) roughly correlate and anticorrelate, respectivey, with  $T_c$  in a number of transition metal materials.  $27, 30, 38, 39$  Such a trend presumably results from the increased renormalization of long-wavelength acoustic phonons that accompanies an increased electronic density of

TABLE V. Parameters derived from phonon moment analyses of the heat capacities of  $U_{6}X$  compounds. The stated ranges of parameters are representative of the variations incurred when two or three Debye spectra are used, entropy matching at  $T_c$ , or proper normalization of the phonon spectrum are enforced or not, etc. These limits should not be considered as true uncertainties in the quoted numbers; rather they provide some estimate of the degree of arbitrariness of the fitting procedure.

		. .	ັ້			
Compound	$\sim$ * (mJ/mol K <sup>2</sup> )	$\Theta$ (T $\rightarrow$ 0) (K)	$\overline{\omega}_{\log}$ (K)	$\langle \omega^{-2} \rangle^{-1/2}$ (K)	$\langle \omega^{-1} \rangle$ <sup>-1</sup> (K)	
$U_6Mn$	$97 - 104$	$184 - 215$	$92 - 93$	$89 - 90$	105	
$U_6$ Fe	157	$117 - 118$	83	$76 - 78$	$90 - 97$	
$U_6$ Co	$133 - 136$	$115 - 124$	$84 - 91$	$78 - 84$	$97 - 119$	
$U_6$ Ni	92	$182 - 187$	$99 - 100$	$95 - 97$	$107 - 113$	

Parameter	$U_6Mn$	$U_6$ Fe	$U_6$ Co	$U_6$ Ni
$\gamma^*$ (mJ/mol K <sup>2</sup> )	103	157	133	92
$\Theta_1$ $(K)$	22.2	47.6	22.6	38.3
$\Theta_{2}(K)$	68.2	160.5	74.2	132.1
$\Theta_{3}(K)$	178.6		250 <sup>a</sup>	250 <sup>a</sup>
$D_1$	$-0.0040$	0.042	0.0026	$-0.0060$
$D_{2}$	0.098	0.958	0.156	0.567
$D$ ,	0.906		0.841	0.439
$\Theta$ (T $\rightarrow$ 0) (K)	215.2	117.4	114.6	181.6
$\langle \omega^{-2} \rangle^{-1/2}$ (K)	89.8	77.5	84.2	97.0
$\langle \omega^{-1} \rangle^{-1}$ (K)	105.3	97.4	$\sim$ 119 <sup>b</sup>	113.2
$\exp(\ln\omega)$ (K)	117.4	109.4	$\sim$ 147 <sup>b</sup>	126.2
$\langle \omega \rangle$ (K)	126.3	116.9	$~166^{\circ}$	138.3
$\overline{\omega}_{2}$ (K)	115.3	106.7	$\sim$ 141 <sup>b</sup>	125.1
$\overline{\omega}_{\log}$ (K)	92.4	83.3	91.2	100.1
rms deviation	$0.99\%$	0.82%	0.67%	$1.5\%$

TABLE VI. A selection of model parameters derived from phonon moment analyses of the heat capacities of  $U_{6}X$  compounds. Entries are taken from a particular "best" fit of the experimental data. All fits conserve entropy at  $T_c$ .

'Arbitrarily fixed in the fit. The exact value of  $\Theta_3$  will not strongly influence the lower-order frequency moments.

"Values are not reliable due to a lack of higher-temperature  $C_P$  data.

states.<sup>26</sup> Nevertheless, it is still surprising that such an eftect could result in the fourfold to fivefold increase in the  $\beta$  coefficient of the  $T^3$  term of the lattice heat capacity noted in comparing the close analogues  $U_6Mn$  and  $U_6$ Fe, or  $U_6$ Ni and  $U_6$ Co. This situation suggests that additional terms (e.g.,  $T^3 \ln T$  terms) must be included in





FIG. 6. Heat capacity  $C_P$  divided by temperature T vs T for  $U_6Mn$ ,  $U_6Fe$ ,  $U_6Co$ , and  $U_6Ni$ . The symbols represent experimental data at  $T > T_c$ , and the lines represent a fit of the data using the MOM analysis described in the text. Relevant parameters are collected in Tables V and VI.

FIG. 7. Approximations to the phonon density of states  $F(\omega)$ divided by phonon frequency squared  $\omega^2$  vs  $\omega$  for  $U_{\phi}X$  ( $X \equiv Mn$ , Fe, Co, Ni) compounds. Note that  $\omega^{-2}F(\omega)$  is the weighting of the various branches of Debye spectra used in obtaining the MOM fits of  $C_p(T)$  detailed in Tables V and VI, and Fig. 6. The Debye cutoffs  $\theta_i$  for each branch, the relevant values of the effective Debye temperature  $\Theta(T\rightarrow 0)$ , and the approximate Allen-Dynes moment (Ref. 28)  $\bar{\omega}_{\text{log}}$  [see Eq. (22)] are shown for each compound.

model fits of the electronic heat capacity. Further experiments, such as inelastic neutron scattering, would be useful to justify such an analysis.

Although the MOM fits to the  $U_6Mn$  and  $U_6Ni$  data are excellent over a wide range of temperature, there are significant discrepancies between the fits and the data at  $T \gtrsim 25$  K. This is particularly noticeable for U<sub>6</sub>Ni (see Fig. 6), suggesting that it may be necessary to add Einstein modes to  $F(\omega)$  in order to reproduce the extreme negative curvature observed in  $C_p T^{-1}$  versus  $T^2$  at higher temperatures. However, the upper cutoffs  $(\theta_3)$  in Table VI) are not uniquely defined when they are the rable VI) are not uniquely defined when they are the<br>same order as  $\sim 8T_{\text{max}}$ , where  $T_{\text{max}}$  is the maximum temperature achieved in the measurements. Highertemperature data, especially for  $U_6$ Co and  $U_6$ Fe, are required to fully reveal the differences in behavior of the  $U_{\alpha} X$  compounds.

The  $C_p T^{-1}$  versus T plots of Fig. 3 emphasize the remarkable differences in low-temperature curvature of  $C_p$ between the closely analogous pairs of compounds,  $U_6$ Fe and  $U_6Mn$ , and  $U_6Co$  and  $U_6Ni$ . Such differences among analogous compounds have also been observed in Chevrel phases<sup>33,39</sup> Mo<sub>6</sub>Se<sub>6</sub> ( $T_c = 6.3$  K) and Mo<sub>6</sub>S<sub>8</sub> ( $T_c = 1.8$  K). Although the magnitude of  $C_p(T)$  for the  $U_6X$  materials is <sup>1</sup> order of magnitude greater than that of the Chevrel phases, the absolute differences in  $C_pT^{-1}$  between higher- and lower- $T_c$  analogues are quite comparable for the two types of materials.

The large  $\gamma^*$  coefficients typically observed in heavyfermion materials automatically lead to very high entropies at modest temperatures, as demanded by Eq. (I). It is reasonable to associate these large entropies with the presence of "quasilocalized" excitations arising from the presence of quasificant<br>diversals at the interant conduction states.<sup>40,41</sup> It has been pointed out<sup>41</sup> that so far the electronic entropy  $S_e$  has only been extracted from calorimetric data for two heavy-ferrnion compounds (UBe<sub>13</sub> and U<sub>2</sub>Zn<sub>17</sub>) due to the limited availability of suitable analogue materials that are useful in approximating the lattice contribution  $C_l$ . However, it is still very difficult to clearly separate electronic and lattice contributions to the heat capacity, even when suitable analogue compounds (which display a Debye behavior over a significant temperature interval) are available.<sup>42</sup> Recent inelastic neutron scattering measurements<sup>43,44</sup> on  $UBe_{13}$ and  $UPt<sub>3</sub>$  have provided more direct means for deducing  $C<sub>l</sub>$ , and surprisingly, these data show no evidence for a "strong coupling" (e.g., phonon softening at low  $T$ ) between the electron and lattice systems in these two materials.

The four  $U_6 X$  compounds constitute a unique system of analogues exhibiting reasonably large effective masses and an  $\sim$  50% variation in  $\gamma^*$ . It is, therefore, of particular interest to compare the variations of the lowtemperature  $S_e$  of these compounds. A plot of the total entropies  $S(T)$  for  $U_6Mn$  and  $U_6F$ e are shown in Fig. 8. Note that by  $T \approx 15$  K an entropy difference of  $\sim 2$ J/mol K  $\sim$  0.35R ln2 has been established between these compounds. A similar plot, including  $U_6Mn$ ,  $U_6Co$ , and  $U_6$ Ni, is shown in Fig. 9. An entropy difference of



FIG. 8. Total entropy S vs temperature T for  $U_6$ Fe and U6Mn. The inset illustrates the lower-temperature behavior of  $S(T)$  below  $T_c$ , where  $S(T)$  suffers an abrupt change in slope. The curves were calculated using the POLY method, as described in the text.

 $\sim$ 0.45R ln2 is established between U<sub>6</sub>Co and U<sub>6</sub>Ni by  $T\approx$ 13 K. Note also that these entropy differences appear to saturate with increasing temperature by  $\sim$  15 K, although this conclusion is based on a somewhat limited data set. These entropy differences are remarkable, even if they are normalized to the U concentration. Unfortunately, there is no Th-based, non- $5f$  (and therefore low- $\gamma^*$ ) analogue available for such comparisons. Higher-temperature heat capacity and low-temperature inelastic neutron scattering data would be desirable in order to better quantify these results and confirm the saturation of the entropy differences between various  $U_6X$ compounds.

The low-temperature behavior of  $C_P(T)$  for the  $U_6X$ compounds can be discussed from yet another interesting point of view. DeLong et  $al.^{12}$  reported that the



FIG. 9. Total entropy S vs temperature T for  $U_6Mn$ ,  $U_6Ni$ , and  $U<sub>6</sub>Co$ . The inset illustrates the lower-temperature behavior of  $S(T)$ , which suffers an abrupt change of slope below  $T_c$  in the cases of  $U_6Mn$  and  $U_6Co$ . The curves were calculated using the POLY method, as described in the text.

normal-state heat capacity of  $U_6$ Fe exhibited an unusual  $T^2$  behavior over the temperature range  $4 \lesssim T \lesssim 20$  K, as shown in Fig. 10. The solid line in Fig. 10 corresponds to a temperature dependence  $T^n$  with  $n = 2.06$ .

A  $T^2$  dependence of the *lattice* heat capacity  $C_l$  has previously been observed in high- $T_c$  A 15 compound such as  $Nb_3Sn$  and  $Nb_3Al$  by Webb *et al.*,<sup>45</sup> and was correlated with a strong  $T^2$  dependence of the electrical resistivity over a similar temperature interval. Webb et al. proposed a model phonon density of states  $F(\omega)$ that simultaneously reproduced the behavior of both  $C_l(T)$  and the  $T<sup>2</sup>$  term of the resistivity, suggesting that it arose from electron-phonon scattering in the presence of a high density of low-energy phonon modes.  $T^2$  terms in the low-temperature ( $T < 5$  K) resistivities of U<sub>6</sub>Fe and  $U_6$ Co have been recently observed.<sup>46</sup>

We have intensified our interest in the  $T^2$  behavior of  $U_6$ Fe in view of recent observations<sup>47</sup> of a very lowtemperature  $T^2$  dependence of  $C_p(T)$  for the heavyfermion material, CeAl<sub>3</sub>. We anticipated that  $U_6$ Co might also display a  $T^2$  character to  $C_P(T)$  due to this compound's high  $\gamma^*$  and strong low-frequency weight in  $F(\omega)$  (see Fig. 7). Indeed, we have found  $T^2$  behavior, but within two separate temperature intervals,  $8 \text{ K}$  $\leq T \leq 13$  K and 2 K  $\leq T \leq 5$  K; We also note that  $C_p(T)$ roughly varies as  $T<sup>n</sup>$  with  $n \approx 2.14$  in the interval 5 K  $T \lesssim 13$  K, demonstrating that the approximate variation of  $C_P(T)$  is close to  $T^2$  over the entire normal-stat range investigated, 2.3 K  $\leq T \leq 13$  K.

Plots of  $C_p$  versus  $T^2$  are given in Fig. 11 for U<sub>6</sub>Fe,  $U_6$ Co, and  $U_6$ Mn.  $C_P$  of  $U_6$ Mn only roughly follows a  $T^2$ dependence for 10 K  $\leq T \leq 20$  K. If we demand that the dependence for 10 K  $\approx$  *I*  $\approx$  20 K. If we demand that the total  $C_p(T)$  be described by a "simple  $T^{2n}$ " law that intersects the origin, we find that  $U_6$ Fe obeys such a law at all



FIG. 10. Logarithm of the normal-state heat capacity  $C_P$  vs the logarithm of temperature T for  $U_6$ Fe. The straight line is a guide to the eye and represents a  $T<sup>n</sup>$  dependence of  $C<sub>p</sub>$  with  $n = 2.06$ .



FIG. 11. Heat capacity  $C_p$  vs the square of temperature,  $T^2$ for  $U_6Mn$ ,  $U_6Fe$ , and  $U_6Co$ .

experimental temperatures  $T \geq T_c$ ,  $U_6$ Co satisfies the law less well for  $T \geq T_c$ , and U<sub>6</sub>Mn follows such a dependence only for  $T \gtrsim 14$  K. Alternatively, we observe that the region of  $T^2$  behavior may shift to higher temperatures as  $y^*$  decreases. We note that Webb *et al.*<sup>45</sup> observed nonzero intercepts for extrapolations of the  $T^2$  regions of  $C_l(T)$  for A 15 compounds.

We wish to emphasize that the difficulty in separating the lattice and electronic contributions to  $C_p(T)$  makes it dangerous to attribute low-energy structure in the experimentally deduced  $F(\omega)$  to phonon excitations alone, although this certainly cannot be ruled out. It is possible that a  $T^2$  behavior of  $C_p(T)$  may be an artifact of high- $\gamma^*$  materials for which a clear separation of the lattice contribution is most suspect. For example, we have found that the  $Mo<sub>6</sub>Se<sub>8</sub>$  and  $Mo<sub>6</sub>Se<sub>8</sub>$  data of Ref. 33 do not exhibit a  $T^2$  phenomenon. This may be associated with the fact that  $\gamma^* \lesssim 2.6 \times 10^3$  erg/cm<sup>3</sup> K<sup>2</sup> for these two Chevrel phases,  $33,39$  1 order of magnitude smaller than



FIG. 12. Normal-state heat capacity  $C_p$  divided by temperature T vs T for the pseudobinary A 15 material  $Ti_{0.75}Ir_{0.20}Pt_{0.05}$ . The straight line is a guide to the eye. Data were taken from Ref. 48.



FIG. 13. Heat capacity  $C_P$  vs temperature T for  $U_6$ Fe in applied magnetic fields  $H = 0$  and 2 kOe.

the  $\gamma^*$  of the U<sub>6</sub>X compounds. On the other hand, we find that the data of Junod *et al*.<sup>48</sup> for the pseudobinar A 15  $Ti_{75}Ir_{20}Pt_5$  do follow a  $T^2$  dependence (as shown in Fig. 12); and this material has a  $\gamma^* \approx 9 \times 10^3$  erg/cm<sup>3</sup> K<sup>2</sup>, more comparable to the  $U_6X$  and high- $T_c$  A 15 compounds.

Measurements of  $C_p(T)$  were also carried out in applied magnetic-field strengths of 2 kOe in the case of  $U_6$ Fe and  $U_6$ Co, and the results are shown in Figs. 13 and 14. There is no apparent field-induced change in the data, save the small reductions in  $T_c$  and  $\Delta C$ , for a 2 kOe field. Calorimetric determinations of the initial slope of the upper critical field  $H'_{c2}(T)$  were made taking care to idealize the transition anomaly in such a manner as to comply with the entropy constraint at  $T_c$ . The results are  $H'_{c2}(T) = -3.64$  T/K and  $-3.33$  T/K for U<sub>6</sub>Fe and  $U_6$ Co, respectively. Resistive measurements<sup>12,13</sup> on the same U<sub>6</sub>Fe sample II-33 have yielded  $H'_{c2}(T_c) = -3.42$ T/K. Menovsky et  $al.^{49}$  have measured the heat capaci-



FIG. 14. Heat capacity  $C_P$  vs temperature T for  $U_6$ Co in applied magnetic fields  $H = 0$  and 2 kOe.



FIG. 15. Magnetization  $\sigma$  divided by applied magnetic field H vs temperature T for  $U_6Mn$ . Data were taken in a constant  $H=20$  kOe. Traces of ferromagnetism are visible for  $T \lesssim 130$ K.

ty of  $U_6$ Co in applied fields of 0 to 5 T (in 1 T steps), and found  $H'_{c2}(T_c) = -3.7T/K$ ,  $T_c(H=0) = 2.3$  K and  $\gamma^*(H=0) = 126 \text{ mJ/mol K}^2$ , all in very good agreement with our results. DeLong et al.<sup>50</sup> have resistively measured two different  $U_6$ Co samples and found  $H'_{c2}(T_c) = -3.3$  T/K and  $-3.9$  T/K.

The relatively large transition width observed for the  $U_6$ Mn sample II-46 was attributed to impurities in the U starting materials, and it was therefore useful to measure the magnetic susceptibility of this sample in order to check for temperature-dependent paramagnetism or ferromagnetic impurity phases. Data for the magnetization  $\sigma$  were taken at several temperatures. The susceptibility  $\chi^*$ , defined by the slope of  $\sigma$  versus H, was found to be temperature independent for 6 K  $\leq T \leq 240$  K, but a small remanence developed at  $T \lesssim 120$  K. This is confirmed in the plot of  $\sigma/H$  versus T in a fixed field  $H = 2.0$  T, as shown in Fig. 15. The value of  $\sigma/H = 3.1 \times 10^{-3}$  cm<sup>3</sup>/mol at  $\widetilde{T} = 6$  K is in excellent agreement with the results of DeLong et  $al.^5$  Sample II-46 exhibits about <sup>1</sup> order of magnitude less remanence than their data for  $U_6$ Fe sample II-33, suggesting that there were not abnormally large amounts of ferromagnetic phases present in our  $U_6Mn$  sample.

## V. DISCUSSION OF RESULTS

We now turn to a discussion of possible interpretations of our heat-capacity data. The unusually high values of  $\gamma^*$  and the evidence for a high density of low-energy excitations in the  $U_6X$  compounds imply that simple models and traditional interpretations for these data should be regarded with caution. In view of the probable strong 5f contributions to the electronic density of states of these materials, several phenomenological models that recently have been applied to heavy-fermion systems are discussed below.

## A. Fermi-liquid approach

Several arguments $^{40,51,52}$  have been given that imply the superconductivity in heavy-fermion metals should be of an odd-parity type mediated by the exchange of paramagnons. Given that liquid  ${}^{3}$ He is an exchangeenhanced, triplet-paired superfluid that can also be described as "nearly localized",  $53-55$  Fermi-liquid theory has frequently been used to compare heavy-fermion superconductors to  ${}^{3}$ He. Many of these approaches should be carefully distinguished from the microscopic "paramagnon model"<sup>56,57</sup> for exchange-enhanced metals and <sup>3</sup>He. Our heat-capacity results have already been used<sup>5</sup> to analyze magnetic susceptibility data for the  $U_6X$ compounds in terms of the paramagnon model, and this approach will not be discussed further. We will instead compare our results to recent applications of Landau Fermi-liquid theory to the heavy-fermion problem. The reader is referred to Refs. 41 and 58 for reviews of these techniques.

We are specifically interested in comparing our heatcapacity data with model predictions of the superconducting transition temperature and pairing symmetry. We use the weak-coupling calculation of  $T_c$  by Patton and Zaringhalam<sup>59</sup> as outlined by Pethick and Pines.<sup>58</sup> The normal-state properties are described by the symmetric  $(s)$  and antisymmetric  $(a)$  Landau parameters

$$
A_{l}^{s,a} = \frac{F_{l}^{s,a}}{1 + F_{l}^{s,a}/(2l+1)} \tag{27}
$$

We assume the forward-scattering sum rule is satisfied in the "sp approximation:"60

$$
\sum_{l=0}^{1} (A_l^s + A_l^a) = 0 \tag{28}
$$

The transition temperature is given in terms of a weak-coupling formula for pairing in a state of orbital angular momentum  $L = [l(l+1)]^{1/2} \hbar$ :

$$
\mu T_c = 1.13 \Theta_c \exp(1/\lambda_\mu) , \qquad (29)
$$

where the pairing interaction strengths in the singlet  $(\mu \equiv 0)$  and triplet  $(\mu \equiv 1)$  channels are given by

$$
\lambda_0 = \sum_{l} \frac{(-1)^l}{4} (A_l^s - 3 A_l^a) ,
$$
  

$$
\lambda_1 = \sum_{l} \frac{(-1)^l}{12} (A_l^s + A_l^a) .
$$
 (30)

 $\Theta_c$  is a cutoff that must satisfy  ${}^{\mu}T_c \ll \Theta_c \ll T_F^*$  in the weak-coupling limit.<sup>41</sup> Note that in the  $sp$  approximation the sums in Eq. (30) are taken only over  $l = 0, 1$ , and  $\lambda_u$ must be less than zero for an attractive interaction and nonzero  $T_c$  to occur.

We will discuss our attempts to apply Eqs.  $(27)$ – $(30)$  to our data using four different types of approximations. The first method was used by Valls and Tesanovic<sup>61</sup> in a direct application of the "almost localized" model of  ${}^{3}$ He to heavy-fermion superconductors. The assumption of a large  $m^*/m_e$  and incipient localization imply  $A_1^s \sim 3$ ,  $A_0^s$  ~ 1, and  $A_0^a$  ~ -3, and the sum rule [Eq. (30)] is used

to obtain  $A_1^a = -1$ . Equations (30) then yield  $\lambda_0 = 1$  and , and thereby predict triplet superconductivity It is customary<sup>58</sup> to reexpress  $\Theta_c$  in terms of either the Fermi or spin-fluctuation temperatures:

$$
1.13\Theta_c = \delta T_F^* = \beta T_{\text{sf}} \tag{31}
$$

 $\delta$  and  $\beta$  are unknown scaling factors, and it is assumed that  $T_{\text{sf}} = (1 + F_0^a)T_F^{*}$ .<sup>58</sup> A summary of experimental parameters for the four  $U_6X$  compounds is given in Table VII, and a list of parameters derived from the "almost localized" model is given in Table VIII.

The method used by Valls and Tesanovic suffers from several serious defects. It assumes Galilean invariance within the Fermi liquid and sets  $m^*/m_e = 1 + F_1^s/3$ , implying  $A_1^s \sim 3$  for heavy fermions. However, the Bloch symmetry of crystals invalidates the relation between ' $m^*/m_e$  and  $F_1^s$ . <sup>41,58</sup> Further, the  $m^*$  values of the  $U_6X$ compounds are roughly <sup>1</sup> order of magnitude less than those of UPt<sub>3</sub> and UBe<sub>13</sub>, and the assumption of  $A_1^s \sim 3$  is probably not justified in our case.

A second method, which we label the "induced interacion" approach,  $58,62,63$  relies on a general treatment of the Landau interaction function in the case of a short-range potential between quasiparticles.<sup>64,65</sup> We assume  $A_0^s \sim 1$ for a charged Fermi liquid,<sup>58</sup> and the effective mass is given by

$$
m^*/m_b = 1 + F_1^s / 3 \tag{32}
$$

where  $m_b$  is the quasiparticle mass in the absence of 'backflow, " $62,63$  and contains renormalizations due to band structure and the electron-phonon interaction. The quasiparticle mass  $m_b$  can be measured in the superconducting state via observation of the London penetration depth. $\overline{63}$  The essential point is that this approach does not rely on Galilean invariance of the Fermi liquid. Unfortunately, the practical application of the model is best achieved by fixing  $F_0^a$  and  $A_0^a$  with the  $T^3 \ln T$  contribution to  $C_p$ , such as is possible in the case of <sup>3</sup>He, UPt<sub>3</sub>, and  $UAl_2$ .<sup>58</sup> Since very few materials unambiguously exhibit the  $T^3 \ln T$  term (the U<sub>6</sub>X compounds evidently do not), one must adopt another means of fixing  $F_0^a$ .

We have chosen to relate  $F_0^a$  to the "R ratio" for a spin-<del>!,</del> Fermi liquid:

$$
R = \frac{1}{3} \left[ \frac{\pi k_B}{\mu_B} \right]^2 \frac{\chi^*(T \to 0)}{\gamma^*} \sim (1 + F_0^a)^{-1} \ . \tag{33}
$$

The validity of this relation has been discussed<sup>5</sup> in ap-

TABLE VII. Experimental parameters for Fermi-liquid models of the  $U_6X$  compounds.

	T,		$k_F^a$		$T_F^*$ <sup>a</sup>
Compound	(K)	R	$(10^8 \text{ cm}^{-1})$	$m^*/m_e^{\alpha}$	(K)
$U_6Mn$	2.21	2.2	1.56	15	7200
$U_6$ Fe	3.70	1.4	1.57	23	4690
$U_6$ Co	2.29	1.5	1.57	20	5540
$U_6$ Ni	0.35	2.4	1.565	13.5	8020

<sup>a</sup>Based on an estimate of  $Z = 3$  fermions per U atom.

TABLE VIII. Parameters derived from an "almost localized" Fermi-liquid model.

Compound	Θ, $(\mathbf{K})$		
$U_6Mn$	39	$6.2 \times 10^{-3}$	
$U_6$ Fe	66	$1.6 \times 10^{-2}$	
$U_6$ Co	41	$8.3 \times 10^{-3}$	
$U_6$ Ni	6.2	$8.8 \times 10^{-4}$	

plication to heavy-fermion systems. The major risks in using Eq. (33) stem from its omission of band-structure and spin-orbit effects on the magnetic susceptibility  $\chi^*$ . However, these approximations are certainly consistent with those used in the majority of Fermi-liquid models. $41$ 

The "induced interaction" approach yields, in the case of U<sub>6</sub>Fe,  $\lambda_1=0.1$ , thereby precluding a triplet pairing state, and  $\lambda_0 = 0.1 - A_1^s$ .  $A_1^s$  cannot be determined from Eq. (32) without knowledge of  $m_b$ . Assuming  $1 \lesssim m_b/m_e \lesssim 14$ , we deduce  $1.2 \lesssim A_1^s \lesssim 2.9$  and  $-2.8 \lesssim \lambda_0 \lesssim -1.1$ . However, these parameters imply 4.7 K  $\leq \Theta_c \leq 8.1$  K, indicating a breakdown of the weak-coupling assumption due to the proximity of  $\Theta_c$  to  $T_c$ . On the other hand,  $A_1^s > 0.1$  is required for a finite  $T_c$ . We can only conclude that  $1 \lt m_b / m_e \leq 22$  with  $10^{-3} < \delta$ , and that  $1 \lesssim m^*/m_b < 2$  in order for the weakcoupling approximation to be valid.

A third approach was recently suggested by Varma,<sup>40</sup> and postulates that the momentum dependence of the quasiparticle self-energy is negligible compared to its energy dependence.  $m^*$  is then dominated by a strong wave-function renormalization (SWR) that yields no effect on a number of transport properties, similar to the electron-phonon interaction in metals.<sup>66</sup> The SWR approach results in an explicit relationship between  $m^*$  and  $F_0^s$ :

$$
\frac{m^*}{m_e} = 1 + F_0^s \tag{34}
$$

We have applied the SWR method to our  $U_6$ Fe data and deduced  $A_0^a = -0.40$ ,  $A_0^s = 0.96$ ,  $\lambda_1 = 0.09$ , and  $\lambda_0 = 0.12 - A_1^s$ , in good agreement with the induced interaction approach parameters calculated above. Therefore, depending on the proper value of  $A_1^s$ , the weak-coupling approximation may again be violated within this method. It is noteworthy that recent experiments<sup>46</sup> on the upper critical field and magnetoresistance of UPt<sub>3</sub>, U<sub>6</sub>Fe, and U<sub>6</sub>Co have provided independent evidence for anomalously low values of a cutoff energy comparable to  $T_c$ , and a failure of the weak-coupling approximation for heavy-fermion superconductors.

Finally, we have elaborated on a conjecture of Pethick and Pines<sup>58</sup> that a "law of corresponding states" (LCS) may apply to the Fermi-liquid properties of  ${}^{3}$ He, UPt<sub>3</sub>, and other heavy-fermion materials. We assume  $A_0^s \sim 1$ and that the scaling parameter  $\beta$  of Eq. (31) is 0.05 for all Fermi liquids of interest. Again using Eq. (33), we deduce the parameters summarized in Table IX. Note that singlet pairing is predicted for all four  $U_{\phi}X$  com-



pounds (triplet  $T_c$ 's are extremely low, whenever  $\lambda_1 < 0$ ), although these materials appear to be near a crossover to the triplet state. Note also that the values of  $\Theta_c$  for the  $U_{\alpha}X$  compounds justify the weak-coupling formula, and that  $\Theta_c \ll T_{\text{sf}}$ , suggesting that paramagnon pairing is "cut off"<sup>51</sup> by a lower-energy excitation such as phonons

Several caveats to the above Fermi-liquid analyses should be mentioned. Band-structure effects arising from a breakdown of Galilean invariance lead to a "bare band mass" and orbital contributions to  $\chi^*$  that either have been neglected or only roughly approximated in our analyses. Spin-orbit and anisotropy effects are also omitted, and the weak coupling Eq. (29) should not be trusted for calculation of accurate values of relevant parameters.<sup>41</sup> The parameters  $k_F$ ,  $m^*$ , and  $T_F^*$  are also uncertain since they depend upon a proper choice of the fermion density n, which is not easily determined from independent measurements. Pethick and  $Pines^{57}$  and  $DeLong^{67}$  have given phenomenological arguments for assuming that each U atom contributes three "heavy fermions" (other contributions to  $n$  are neglected). We have adopted this assumption given the lack of direct experimental measurements of <sup>n</sup> and our desire to express our results within a common convention used by other authors.

In spite of the many approximations that one is forced to make in the application of Fermi-liquid theory, the various models discussed above generally yield quite reasonable values for the Landau parameters and associated quantities. At present there is no accepted microscopic model for heavy-fermion metals, and it is therefore useful to compare the Fermi-liquid properties of the  $U_{6}X$ compounds to those of related materials. We believe that the relative trends and semiquantitative results given in Table IX have physical significance.

#### B. Strong-coupling analysis

The results of the Fermi-liquid analyses suggest that even-parity superconductivity due to the usual electronphonon interaction is probably appropriate to the  $U_6X$ compounds. The large size of the specific-heat jump at  $T<sub>c</sub>$  and the zero-temperature thermodynamic critical field for  $U_6$ Co and  $U_6$ Fe (see Tables III and IV) imply that our results should be analyzed with a theoretical model that takes into account strong-coupling effects. We have accordingly compared our data to the Allen-Dynes model<sup>28</sup> in order to extract values of the electron-phonon coupling  $\lambda$  [see Eq. (19)] and the band-structure density of states  $N_b(E_F)$  given by

$$
N_b(E_F) = \frac{N^*(E_F)}{1+\lambda} = \frac{3\gamma^*}{2\pi^2 k_B^2 (1+\lambda)} \ . \tag{35}
$$

 $\lambda$  must be deduced from the  $T_c$  formula [Eq. (17)] using the approximation  $\omega_{\text{log}} \approx \overline{\omega}_{\text{log}}$  [see Eqs. (18) and (20)] and the functions  $f_1$  and  $f_2$  defined by

$$
f_1 = [1 + (\lambda / \Lambda_1)^{3/2}]^{1/3}
$$
 (36)

and

$$
f_2 = 1 + \frac{\lambda^2 (\overline{\omega}_2 / \omega_{\text{log}} - 1)}{\lambda^2 + \Lambda_2^2}, \qquad (37)
$$

TABLE X. Parameters for the strong-coupling analysis of the  $U_6X$  compounds. All parameters calculated assuming  $\mu^* = 0.13$ .

Compound		$\sqrt{ }$ ,		$N_b(E_F)$ $(eV-atom-spin)^{-1}$
$U_6Mn$	1.03	1.01	0.668	1.85
$U_6$ Fe	1.04	1.01	0.841	2.58
$U_6$ Co	1.03	1.01	0.678	2.40
$U_6$ Ni	1.01	1.00	0.437	1.94

where

$$
\Lambda_1 = 2.46(1 + 3.8\mu^*)
$$
\n(38)

and

$$
\Lambda_2 = 1.82(1 + 6.3\mu^*)\overline{\omega}_2/\omega_{\log} . \tag{39}
$$

We assume a typical value  $\mu^* = 0.13$  and initially set  $f_1 \approx f_2 \approx 1$  in order to arrive at an initial estimate or  $\lambda$ using the phonon moment values in Table VI and  $T_c$ 's from Table III. The estimate of  $\lambda$  is then used to recalculate  $f_1$ ,  $f_2$ , and  $\lambda$  in an iterative procedure until they converge to self-consistent values. The results of the iteration are given in Table X.

Values of  $\lambda$  for the U<sub>6</sub>X compounds are large but reasonable. On the other hand, the values of  $N_b(E_F)$  for the  $U_{\beta}X$  compounds are approximately 1 order of magnitude larger than those of typical transition metal compounds, and are several times larger than for high- $T_c$  $A$  15 materials.<sup>68</sup> If we use the relation

$$
n^* = (1 + \lambda)m_b , \qquad (40)
$$

where  $m_b$  is the average "bare band-structure" mass, we conclude that  $m_b \gtrsim 10m_e$  for the U<sub>6</sub>X compounds. These are remarkably large masses considering that they are obained in the approximation of a spherically averaged  $N^*(E_F)$ . However, Leggett,<sup>69</sup> has pointed out that Eq. (40) may not be valid for heavy-fermion materials. Moreover, the large  $N^*(E_F)$  found in the  $U_{\phi}X$  materials would normally imply a very high  $T_c$ , indeed. It is therefore reasonable to postulate that some "repulsive" mechanism prevents  $T_c$  from exceeding  $\sim$  4 K in U<sub>6</sub>Fe.

### VI. CONCLUDING REMARKS

We have already mentioned several similarities between the high- $T_c$  Chevrel, A 15, and  $U_6X$  compounds. However, the behavior of narrow $-d$ -band materials is far from understood,<sup>70,71</sup> and a strict analogy between  $A15$ and 5f-band or heavy-fermion compounds cannot presently be constructed. It is nevertheless important to point out some of the paradoxes or difhculties that remain in attaining a unified picture of materials whose bandwidth is intermediate between that of  $UBe_{13}$  and conventional BCS superconductors such as Al or Sn.

Here, it is instructive to compare  $U_6$ Fe with the A 15 compound V<sub>3</sub>Ga, which has a  $T_c = 15$  K, a value that is comparable to, or less than, that of the highest- $T_c$  materials of this class. However, the  $\gamma^*$  value of V<sub>3</sub>Ga is approximately double that of the higher- $T_c$ , A 15's (Refs. 30) and 68) and is comparable to that of  $U_6$ Fe.<sup>12</sup> The evidence for a strong Pauli limiting of  $H_{c2}$  for  $V_3Ga$  (Ref. 68) suggests that paramagnons may be the appropriate mechanism responsible for the suppression of the superconducting properties of this compound. Indeed, phenomenological arguments<sup>67,71</sup> imply that magnetic interactions strongly suppress superconductivity for  $N^*(E_F)$  values slightly larger than those observed for the  $U_{6}X$  and  $V_{3}Ga$  compounds.

However, applying the same line of reasoning to the  $U_{\phi}X$  materials does not yield a straightforward explanation of their anomalous properties. In particular, no clear evidence has yet been found that the  $H_{c2}$  of  $U_6$ Fe is at all Pauli limited; indeed, the temperature dependences of  $H_{c2}$  for  $U_6$ Fe and  $U_6$ Co exhibit unusual positive curvature and low-temperature values that are significantly greater than expected.  $46,50$  Further, one must also simultaneously explain why lower  $T_c$ 's and *larger* values of the susceptibility enhancement ratio  $R$  are correlated with smaller  $N^*(E_F)$  in these materials<sup>71</sup> (see Tables VII and X). Note that considerable difficulties have been encountered in explaining similar effects in  $\lambda$  15 materials.<sup>68</sup>

We have neglected exotic or complex contributions to

the heat capacity in our analyses. It is possible that anharmonic corrections (e.g., "phonon softening") can account for the unusual behavior of the  $U_6 X$  materials, as has been suggested for  $A$  15 materials.<sup>26,72</sup> Subtle lowemperature phase transitions<sup>6,73,74</sup> and polaronic effects are also possible mechanisms for anomalous behavior in these materials.

A complete analysis of the heat capacity of the  $U_6X$ compounds, including a well-defined separation of the electronic and lattice contributions, may prove particularly difficult in view of the breakdown of adiabaticity inferred from the heat-capacity behavior of a number of erred from the heat-capacity behavior of a number of<br>narrow-band materials.<sup>5,67,70,71,79</sup> The properties of the  $U_6X$  compounds and their incomplete parallels with A 15 and Chevrel compounds remain enigmatic and deserve further study.

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