

## Raman and infrared studies of $\text{Sr}_2\text{TiO}_4$ : A material isomorphic to $(\text{La,Sr})_2\text{CuO}_4$ superconductors

Gerald Burns and F. H. Dacol

*IBM Thomas J. Watson Research Center, P.O. Box 218, Yorktown Heights, New York 10598*

G. Kliche and W. König

*Max-Planck-Institut für Festkörperforschung, Heisenbergstrasse 1,  
D-7000 Stuttgart 80, Federal Republic of Germany*

M. W. Shafer

*IBM Thomas J. Watson Research Center, P.O. Box 218, Yorktown Heights, New York 10598*

(Received 13 October 1987)

The high-temperature superconductor  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$  has the  $\text{K}_2\text{NiF}_4$  structure, as does  $\text{Sr}_2\text{TiO}_4$ . In order to understand better the phonon spectra obtained from ceramic  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$ , we have performed Raman and infrared measurements on ceramic, insulating  $\text{Sr}_2\text{TiO}_4$ . All four Raman-allowed phonons are observed and found to be quite sharp, because in this centrosymmetric structure the Raman-mode frequencies are isotropic. The infrared reflectivity spectra are more difficult to interpret because reststrahlen bands are measured, which extend from the transverse-optic (TO) frequency to the longitudinal-optic (LO) frequency. Symmetry forbids overlap of the  $3A_{2u}$  bands as well as the  $4E_u$  bands. The interpretation, nevertheless, is difficult because the  $A_{2u}$  and  $E_u$  bands can overlap one another. By comparison with single-crystal results in fluorides and chlorides having the same structure we have been able to identify many of the infrared modes in our spectra. This knowledge permits a determination of most of the infrared active modes in  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$ . In the superconductor, two other effects aid our understanding of the spectra. (a) The effective charges are smaller than in  $\text{Sr}_2\text{TiO}_4$ , reducing the TO-LO splitting, and thus the overlap of the reflectivity bands. (b) The electron plasma is confined to the  $xy$  plane, and consequently it interacts strongly only with the  $E_u$  phonons, and leaves the  $A_{2u}$  phonons largely unaffected.

### I. INTRODUCTION

Since the discovery of high-temperature superconductivity in  $(\text{La, Ba, Sr})_2\text{CuO}_4$ -type materials<sup>1</sup> there has been a huge research effort on these and the related  $\text{YBa}_2\text{Cu}_3\text{O}_7$  superconductors.<sup>2,3</sup> Of course, electron-phonon coupling is a central issue in the theory of superconductivity.<sup>4</sup> Thus, it is natural to study the phonons in these systems with Raman, infrared, and neutron techniques, although it is possible that high-temperature superconductivity in these oxides is not phonon mediated.<sup>2,3</sup>

These high-temperature superconducting oxides have been generally available only in the form of polycrystalline ceramics. Thus, neutron techniques yield only the density of phonon states.<sup>5</sup> However, Raman and infrared measurements, which measure the zone center ( $k \approx 0$ ) phonons, can be used to determine these phonon frequencies. For Raman-active phonons in a centrosymmetric material, the frequencies do not depend on the angles between the phonon propagation direction and the crystallographic directions. Thus, sharp phonon lines can be measured in ceramic samples, but the selection rules cannot be used to sort out the irreducible representations of the various phonons. In noncentrosymmetric crystals, where the Raman phonon frequencies do depend on the various angles, the angular dependence can be exploited to yield the

phonon frequencies.<sup>6</sup> In noncubic crystals, the frequencies of infrared-active modes depend on the various angles of incidence.<sup>7</sup> In analogy with the Raman technique, we demonstrate that data from ceramics can be used to obtain the infrared-active modes. By performing both Raman and infrared measurements on materials that are insulators, but with the same crystal structures as the high-temperature superconductors, we are able to interpret the experimental results,<sup>8-10</sup> and to determine the phonon frequencies in  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$ .

### II. EXPERIMENTAL PROCEDURE

Following previous work,<sup>11,12</sup> the samples were prepared by reacting titanium dioxide with the alkaline-earth carbonates. They were first reacted at  $950^\circ\text{C}$  for 12 h, then reground and heated to  $1275^\circ\text{C}$  for 18 h. The poorly sintered powders were then ground thoroughly, pressed into pellets, fired in oxygen at  $1600^\circ\text{C}$  for 20 h, and then cooled to room temperatures at  $\approx 50^\circ\text{C/h}$ . Some samples were annealed at  $1250^\circ\text{C}$  in oxygen for several days, but these gave similar experimental results. As determined by x-ray diffraction, the materials were single phase with the  $\text{K}_2\text{NiF}_4$  structure, containing less than 5% of the  $\text{Sr}_3\text{Ti}_2\text{O}_7$  phase.

The Raman results were measured on a standard double monochromator using an argon-ion laser operating at 5145 and 4880 Å. The infrared measurements were carried out on a Bruker Fourier-Spectrometer IFS 113v with a resolution of  $2 \text{ cm}^{-1}$ . An aluminum mirror was used as a reference.

### III. STRUCTURE AND SYMMETRY CONSIDERATIONS

Both  $\text{Sr}_2\text{TiO}_4$  and the  $(\text{La,Sr})_2\text{CuO}_4$ -type superconductors crystallize in the tetragonal  $\text{K}_2\text{NiF}_4$  structure,<sup>13</sup> whose generic formula can be written as  $A_2\text{BO}_4$ . The space group is  $I4/mmm(D_{4h}^2)$ , with  $z=2$ , and the primitive unit cell contains one formula unit.<sup>13-15</sup> The structure is shown in Fig. 1, and the positions of the atoms are given in Table I, part (a). Group theory predicts 7 infrared- and 4 Raman-active modes for this structure type [Table I, parts (b) and (c)]. From the symmetry-adapted vectors [Table I, part (b)], obtained using projection operator techniques,<sup>14</sup> we see that in the Raman-active  $E_g$  mode, the two  $\text{O}_z$  atoms (as well as the  $A$  atoms) move in opposite directions along the  $x$  axis (i.e.,  $x_4 - x_3$ ). Likewise the motion can be along the  $y$  axis. For the silent mode the  $\text{O}_x$  and the  $\text{O}_y$  atoms move parallel to the  $z$  axis in the opposite directions (i.e.,  $z_2 - z_3$ ). The normal modes are linear combinations of the symmetry-adapted vectors that transform as the same irreducible representations. Of course, using a model for the short- and long-range interatomic potentials, the normal modes themselves can be calculated, and they will be discussed later.

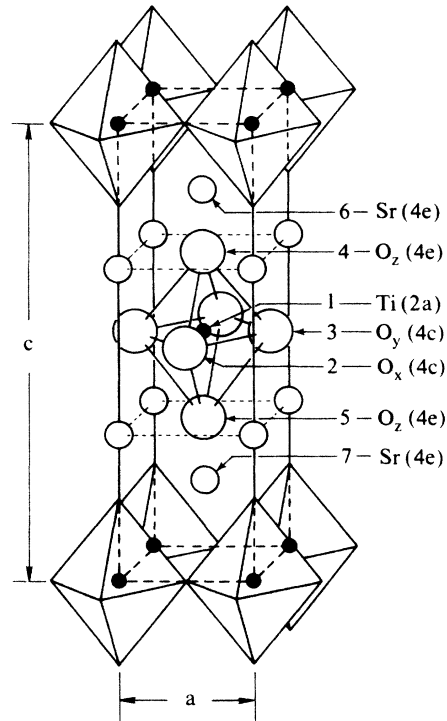


FIG. 1. The multiprimitive unit cell for  $\text{Sr}_2\text{TiO}_4$  which has the  $\text{K}_2\text{NiF}_4$  structure. The positions of the seven atoms in the primitive unit cell are shown along with the Wyckoff notation for their sites. The seven atoms are labeled 1 to 7, in an arbitrary manner for future use.

TABLE I. Various symmetry results for  $A_2\text{BO}_4$  with the  $\text{K}_2\text{NiF}_4$  structure, space group  $I4/mmm(D_{4h}^2)$ .

(a) Atomic positions		
$B$ atoms	$2a$	$(000), (\frac{1}{2}, \frac{1}{2}, \frac{1}{2})$
$\text{O}_x$ and $\text{O}_y$ atoms	$4c$	$(0, \frac{1}{2}, 0), (\frac{1}{2}, 0, 0), (\frac{1}{2}, 0, \frac{1}{2}), (0, \frac{1}{2}, \frac{1}{2})$
$A$ atoms	$4e$	$(0, 0, z), (0, 0, \bar{z}), (\frac{1}{2}, \frac{1}{2}, \frac{1}{2} + z), (\frac{1}{2}, \frac{1}{2}, \frac{1}{2} - z)$ $z \approx 0.35$
$\text{O}_z$ atoms	$4e$	Same, with $z \approx 0.15$
(b) Classification of the normal modes		
$B$ atoms	$2a$	$A_{2u} \quad z_1$ $E_u \quad x_1, y_1$
$\text{O}_x$ and $\text{O}_y$ atoms	$4c$	$A_{2u} \quad z_2 + z_3$ $B_{2u} \quad z_2 - z_3$ $2E_u \quad x_2, y_2 \quad x_3, y_3$
$A$ atoms	$4e$	$A_{1g} \quad z_6 - z_7$ $E_g \quad x_6 - x_7, y_6 - y_7$ $A_{2u} \quad z_6 + z_7$ $E_u \quad x_6 + y_7, y_6 + y_7$
$\text{O}_z$ atoms	$4e$	$A_{1g} + E_g + A_{2u} + E_u$ Same symmetry coordinates as $A$ atoms but with $z_4 - z_5$ , etc.
(c) The sum of the modes grouped according to type of activity		
	$A_{2u} + E_u$	Acoustic
	$3A_{2u} + 4E_u$	ir
	$2A_{1g} + 2E_g$	Raman
	$B_{2u}$	Silent

## IV. RESULTS

## A. Raman

Figure 2 shows the low-temperature Raman spectra for Sr<sub>2</sub>TiO<sub>4</sub>, as well as the two closely related materials, as labeled. The four first-order Raman lines expected from the group theory (Table I) are labeled 1 to 4. Let us first discuss the features labeled *i*. From x-ray measurements we find about 5% of Sr<sub>2</sub>TiO<sub>4</sub> in Sr<sub>3</sub>Ti<sub>2</sub>O<sub>7</sub> and vice versa. The Raman modes from these impurities can be seen at room temperature<sup>8</sup> but become sharper and more apparent at low temperature. These impurity lines are labeled *i* and the frequencies correspond exactly to those found in the related materials.

A surprising feature of the Raman-active modes in Sr<sub>2</sub>TiO<sub>4</sub> is that some of them are very sharp at low temperatures, even in these ceramic samples. The linewidths and positions found in Sr<sub>2</sub>TiO<sub>4</sub> are given in Table II, part (a). For crystals with a structure that has a center of symmetry, the exclusion principle<sup>14</sup> requires that the Raman modes may not be simultaneously infrared active. Thus, the Raman-mode frequencies do not vary with the angle between their propagation direction (*k* direction) and crystallographic axis,<sup>6</sup> giving a natural explanation for the observed narrow linewidths.

## B. Infrared frequencies

The positions of the infrared modes, in contrast, do vary with the angle between the infrared electric field vector,

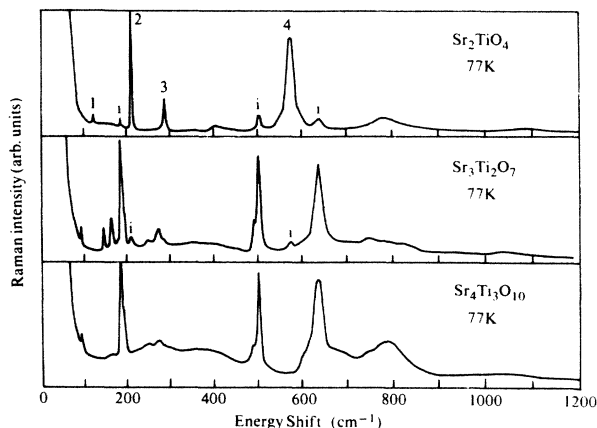


FIG. 2. The Raman spectra, at 77 K, of Sr<sub>2</sub>TiO<sub>4</sub> as well as two closely related compounds, as indicated (Ref. 8). The four Raman-allowed modes are labeled 1 to 4. See the text for the meaning of the modes labeled *i*.

*E*, and the crystallographic *c* axis.<sup>6,7</sup> Because of this angular dependence, it is worthwhile to discuss what should be observed in a ceramic material in reflectivity measurements. Such measurements are shown in Fig. 3(a) for Sr<sub>2</sub>TiO<sub>4</sub>. Let  $\theta$  be the angle between the *E* vector of the infrared radiation and the crystallographic *c* axis. The angular dependence is very similar to that of Raman-active modes that are simultaneously infrared active.<sup>6</sup> Figure 4 shows a schematic diagram of the  $\theta$  dependence

TABLE II. (a) The positions and linewidths of the phonon modes in Sr<sub>2</sub>TiO<sub>4</sub> at room temperature and 77 K. For the mode assignments, see the text. (b) The frequencies and mode assignments for the various infrared-active modes. See the text for a discussion of how these modes are assigned. (c) The modes obtained for (La<sub>2-x</sub>Sr<sub>x</sub>)CuO<sub>4</sub>, as discussed in the text.

(a) Phonon modes in Sr <sub>2</sub> TiO <sub>4</sub>				
	Position		Linewidth	
	RT	77 K	RT	77 K
1	123.9	123.2	4.6	1.3
2	205.4	207.2	3.9	1.7
3	286.2	285.9	15.6	3.5
4	578.3	573.2	40.1	21.5

(b) Frequency and mode assignments			
Freq.	Type	Freq.	Type
151	<i>E<sub>u</sub></i> (TO)	182	<i>E<sub>u</sub></i> (LO)
197	<i>E<sub>u</sub></i> (TO)	239	<i>E<sub>u</sub></i> (LO)
≈ 242	<i>A<sub>2u</sub></i> (TO) or <i>E<sub>u</sub></i> (TO)		
≈ 259	<i>E<sub>u</sub></i> (TO) or <i>A<sub>2u</sub></i> (TO)	≈ 467	<i>A<sub>2u</sub></i> (LO) and <i>E<sub>u</sub></i> (LO)
545	<i>A<sub>2u</sub></i> (TO)	≈ 685	<i>A<sub>2u</sub></i> (LO)
		727	<i>E<sub>u</sub></i> (LO)

(c) Modes for (La <sub>2-x</sub> Sr <sub>x</sub> )CuO <sub>4</sub>			
Freq.	Type	Freq.	Type
117	<i>E<sub>u</sub></i> (TO)	≈ 140	<i>E<sub>u</sub></i> (LO)
232	<i>A<sub>2u</sub></i> (TO)	(340)	<i>A<sub>2u</sub></i> (LO)
350	<i>A<sub>2u</sub></i> (TO)	460	<i>A<sub>2u</sub></i> (LO)
510	<i>A<sub>2u</sub></i> (TO)	544	<i>A<sub>2u</sub></i> (LO)
672	<i>E<sub>u</sub></i> (TO)	691	<i>E<sub>u</sub></i> (LO)

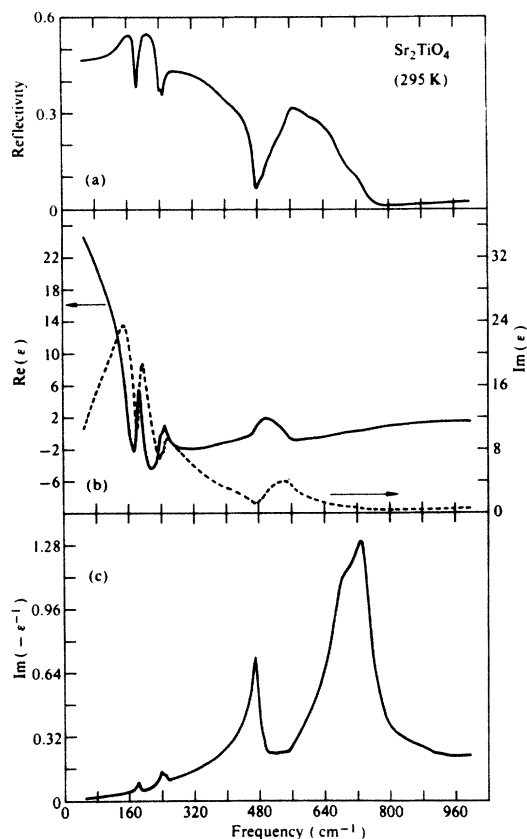


FIG. 3. (a) The frequency dependence of the reflectivity of  $\text{Sr}_2\text{TiO}_4$ . (b) The real and imaginary part of the dielectric constant obtained by a Kramers-Kronig analysis of the reflectivity data. (c) A plot of  $\text{Im}(-\epsilon^{-1})$  vs frequency.

of the infrared-active modes.<sup>7</sup> In single crystals in which reflectivity has been measured as a function of  $\theta$ , good agreement with experiment has been found.<sup>7</sup>

Ceramics consist of small crystals of all orientations. Thus, when a response function, such as reflectivity is measured, the angular dependence of the mode frequencies (Fig. 4) must be taken into account. For any value of  $\theta$ , the crystal has normal modes, but it is only at the extreme ( $\theta=0$  and  $\pi/2$ ) that the modes are called "the" modes with the labels  $E_u(\text{TO})$ ,  $A_{2u}(\text{TO})$ ,  $A_{2u}(\text{LO})$ , and

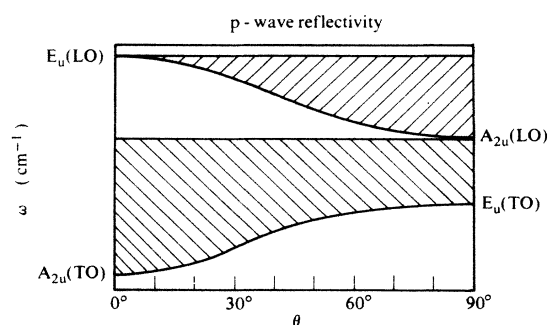


FIG. 4. A schematic of the angular dependence of a reststrahlen band for  $p$ -wave polarization (Ref. 7).

$E_u(\text{LO})$ , as indicated in Fig. 4. At other angles the modes are called quasimodes. Furthermore, Fig. 4 only applies to  $p$  polarization, where the  $E$  vector is in the plane containing the incoming and reflected light, and the  $c$  axis. For  $s$  polarization ( $E$  perpendicular to this plane) only  $E_u(\text{TO})$  and  $E_u(\text{LO})$  modes are measured, independent of  $\theta$ . As in the Raman case,<sup>6</sup> small crystallites near the extreme ( $\theta=0$  and  $\pi/2$ ) give the predominant contribution to observed signals because the smallest angular dependence of the frequency occurs for such crystals. In other words, the number of crystallites with responses in a certain energy range is largest whenever the energy versus  $\theta$  has extremes.

For the infrared response from ceramic samples, we expect to observe the modes in a similar way as has been observed in Raman spectra.<sup>6</sup> However, in noncubic materials the reststrahlen bands that transform as different irreducible representation can overlap each other in some frequency regions. In Fig. 5, the work by Geick and others<sup>16-21</sup> on the fluorides and chlorides of materials with the  $\text{K}_2\text{NiF}_4$  structure are summarized. We have connected the TO to LO mode as in a bar graph to represent a reststrahlen reflectivity band. For a series of modes transforming as the same irreducible representations the

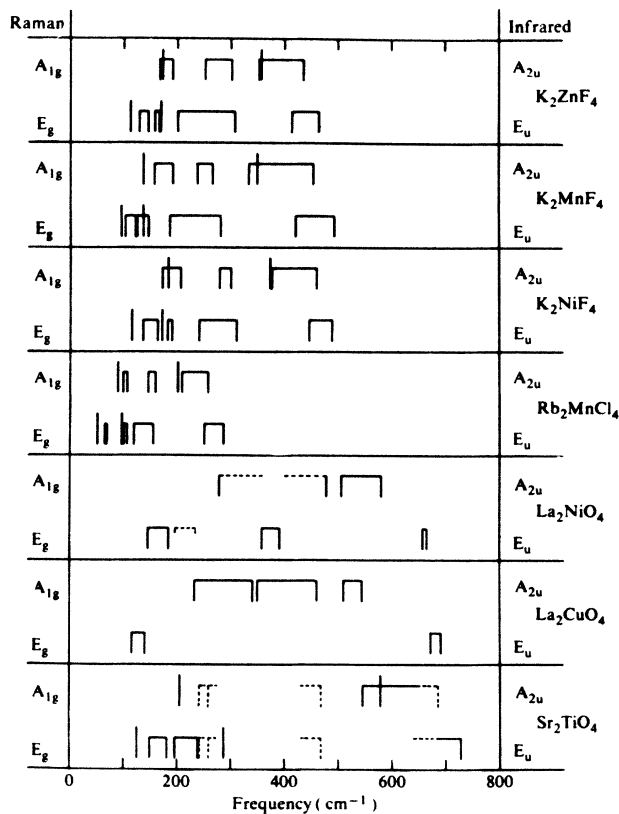


FIG. 5. The frequencies for various bands and modes for the materials as listed. The simple vertical lines represent Raman modes either of  $A_{1g}$  or  $E_g$  type, as indicated on the left. The bars represent infrared reflectivity bands of  $A_{2u}$  or  $E_u$  type, as indicated on the right. The lower-energy side of the reflectivity band corresponds to the TO mode and the higher energy to the LO mode. Dashed lines represent uncertainties.

reststrahlen bands cannot overlap in frequency. For example, this can be seen for the  $E_u$  bands in Fig. 5, and also separately for the  $A_{2u}$  bands. However, there are frequency regions where the  $A_{2u}$  bands overlap the  $E_u$  bands, and this gives rise to the problem of interpreting the spectra measured in ceramics.

Using the single-crystal fluoride and chloride results shown in Fig. 5, we can predict what we expect to see in a ceramic material. In the high-frequency region, in the fluorides and chlorides the highest frequency  $A_{2u}$  and  $E_u$  modes always have the same characteristic behavior. The  $A_{2u}$ (TO) mode is always lower in energy than the corresponding  $E_u$ (TO), and the reststrahlen bands always overlap because the  $A_u$ (LO) frequency is higher than the  $E_u$ (TO) frequency. For a ceramic, the infrared response should be a broad reststrahlen band composed of both the  $A_{2u}$  and  $E_u$  bands. Figure 4 shows the single-crystal  $p$  polarization of this effect, but the  $s$  polarization must be superimposed. The  $s$  polarization effects gives a band that goes from  $E_u$ (TO) to  $E_u$ (LO). Thus, for ceramics, a band that starts at  $A_{2u}$ (TO) and ends at  $E_u$ (LO) is expected. The occurrence of dips in this broad reststrahlen band at  $A_{2u}$ (LO) or  $E_u$ (TO) will depend on many things, certainly on the sharpness (the damping) of these features.

We believe that the broad highest-frequency reststrahlen band in Sr<sub>2</sub>TiO<sub>4</sub> that occurs between 500 and 750 cm<sup>-1</sup> [Fig. 3(a)] has the characteristics described above. In fact, based on the analogue with the fluoride results shown in Fig. 5, Fig. 4 has been drawn to represent this case. Hence, from the Kramers-Kronig analysis [Figs. 3(b) and 3(c)] of the reflectivity data several modes can be determined. The peak of  $\text{Re}(\epsilon)$  gives  $A_{2u}$ (TO) mode at 500 cm<sup>-1</sup> and from the peaks in  $\text{Im}(-\epsilon^{-1})$ , the  $A_{2u}$ (LO) and  $E_u$ (LO) are at  $\approx 680$  cm<sup>-1</sup> and at 720 cm<sup>-1</sup>, respectively. We see no signs of the  $E_u$ (TO) mode even in low-temperature measurements.

Now we consider the lowest-frequency modes. For some of the fluorides and chlorides (Fig. 5) the two lowest  $E_u$  modes are lower in frequency than the lowest  $A_{2u}$  mode.<sup>17-20</sup> Also, the second highest  $E_u$  mode is a weak infrared mode; it arises from the silent,  $T_{2u}$  mode from the oxygen octahedra in the perovskites,<sup>14</sup> which in this structure correlates to an  $E_u$  mode and the silent  $B_{2u}$  mode [Table I, part (b)]. In the reflectivity results, this causes a particularly sharp, almost resonantlike, dip between the lowest  $E_u$ (LO) feature and this second  $E_u$ (TO) feature, because they are so close in energy.<sup>18,19</sup> We believe the rather sharp feature in our reflectivity results [Fig. 3(a)] is just this dip. Hence, from the Kramers-Kronig analysis, the two lowest  $E_u$ (TO) frequencies are obtained as well as the  $E_u$ (LO) mode between them. These frequencies and assignments are listed in Table II, part (b).

We now are left with two  $A_{2u}$ (TO) and one  $E_u$ (TO) mode to be assigned, as well as their corresponding LO components. These modes are located between the lowest and highest frequency grouping discussed above, and are more difficult to assign because of the overlap effects discussed above. Hence, these assignments are much more tentative. From the reflectivity results in Fig. 3(a), there

appear to be two closely spaced TO modes in the frequency region  $\approx 250$  cm<sup>-1</sup>, which become even sharper at low temperatures. We assume that the lower one at  $\approx 242$  cm<sup>-1</sup> is an  $A_{2u}$ (TO) and the higher one at  $\approx 259$  cm<sup>-1</sup> is an  $E_u$ (TO) mode. This is based on the idea that the  $E_u$  spectra will be more dominant than that from the  $A_{2u}$  modes because of  $s$ -polarization-type effects. By analogy with the fluoride results (Fig. 5), the strong LO feature at  $\approx 440$  cm<sup>-1</sup> is probably due to an LO feature from both  $A_{2u}$ (LO) and  $E_u$ (LO). Hence, these are so listed in Table II, part (b).

### C. Normal modes

Extensive calculations have been carried out by Rauh and Geick<sup>17</sup> of the normal modes in K<sub>2</sub>ZnF<sub>4</sub>. Since the distances within the octahedra are approximately equal, they use a rigid octahedra model. Of course, this is only an approximation, but is a good starting point for both the fluorides and for Sr<sub>2</sub>TiO<sub>4</sub>. However, it is not a valid assumption when considering (La<sub>2-x</sub>Sr<sub>x</sub>)CuO<sub>4</sub> as we discuss later.

Remembering that only O<sub>z</sub> and Sr motion takes place in Raman-active modes [Table I, part (b)], and using these calculations<sup>17</sup> as a basis, we discuss the Raman-active modes first. The highest-frequency Raman  $A_{1g}$  mode principally involves the light O<sub>z</sub> atoms moving along the  $z$  axis, with a smaller amount of Sr motion along this axis, out of phase with the O<sub>z</sub> motion. The eigenvector of the lower-frequency  $A_{1g}$  mode is orthogonal to the motion in the higher-frequency mode so it principally involves motion of the heavier Sr atom along the  $z$  axis. While the  $A_{1g}$  modes can be thought of as bond-stretching modes, the  $E_g$  modes are similar to bond-bending modes, and thus are lower in frequency. Correspondingly, the highest-frequency  $E_g$  mode mostly involves O<sub>z</sub> motion in the  $xy$  plane, while the lower-frequency mode involves mostly Sr motion in the  $xy$  plane. Note that all of these motions are similar to the symmetry-adapted vectors listed in Table I, part (b). The lower-frequency  $A_{1g}$  mode and higher-frequency  $E_g$  mode tend to have similar frequencies [Fig. 2 and Table II, part (a)] and some direct method to distinguish between them is desirable.

The silent  $B_{2u}$ -mode motion is completely determined by symmetry. The motion just involves O<sub>x</sub> moving in the plus  $z$  direction against O<sub>y</sub> moving in the minus  $z$  direction [Table I, part (b)]. In cubic perovskite materials this mode is threefold degenerate, allowing for the same type of motion, by pairs of oxygen atoms, in the  $x$  and  $y$  directions. However, the lower symmetry in Sr<sub>2</sub>TiO<sub>4</sub> makes this type of oxygen motion in the  $x$  and  $y$  directions infrared active. This is the second lowest-frequency  $E_u$  mode, already discussed because of its sharp resonantlike reflectivity.<sup>17</sup>

Since all of the atoms can move in the modes that are infrared active, the motions are generally more complicated. We will focus on the lowest- and highest-frequency infrared-active modes since they tend to be the simplest to understand in any model. As would be expected, the lowest-frequency mode involves motion primarily of the heavy Sr atoms against an approximately rigid TiO<sub>6</sub> oc-

tahedron.<sup>17</sup> For the  $A_{2u}$  and  $E_u$  modes this motion is in the  $z$  direction and  $xy$  planes, respectively. The highest-frequency  $A_{2u}$  motion involves Ti against an approximately rigid octahedron, with the motion necessarily being in the  $z$  direction. The highest-frequency  $E_u$  mode involves motion in the  $xy$  plane of Ti against  $O_x$  and  $O_y$ .

As has been pointed out,<sup>9</sup> there is an interesting difference between the fluorides and insulating oxides with the  $K_2NiF_4$  structure, and the oxides that are conductors<sup>22</sup> such as  $La_2NiO_4$  and  $(La,Sr)_2CuO_4$ . The former materials have rigid octahedra in the sense that the Ti- $O_z$  distance is approximately equal to the Ti- $O_x$  distance. In the latter materials the Ni- $O_z$  distance is considerably greater than the Ni- $O_x$  distance,<sup>13</sup> and in  $(La,Sr)_2CuO_4$  this distance is further increased.<sup>23</sup> These different distances effects certain force constants, and hence some of the normal modes.

### V. COMPARISON WITH $(La_{2-x}Sr_x)CuO_4$

In Fig. 6, our reflectivity measurements for  $Sr_2TiO_4$  and  $(La_{2-x}Sr_x)CuO_4$  for two different values of  $x$ , are shown. Our results for  $(La_{2-x}Sr_x)CuO_4$  appear similar to other published work<sup>24-27</sup> except that we seem to resolve a bit more detail.  $(La_{2-x}Sr_x)CuO_4$  is orthorhombic for  $x \lesssim 0.05$  at room temperatures,<sup>28</sup> but the effect on the modes appears to be small since the reported spectra<sup>24-27</sup> for both tetragonal and orthorhombic material are similar to our results shown in Fig. 6, as are our measurements for samples with larger  $x$ . From these measurements certain aspects of the  $(La_{2-x}Sr_x)CuO_4$  results become clearer. We discuss these points and assign some of the modes.

First, consider the high-frequency region, above  $\approx 450$   $cm^{-1}$ . For  $Sr_2TiO_4$ , as discussed above, this reflectivity response consists of a lower-frequency  $A_{2u}$  mode and a

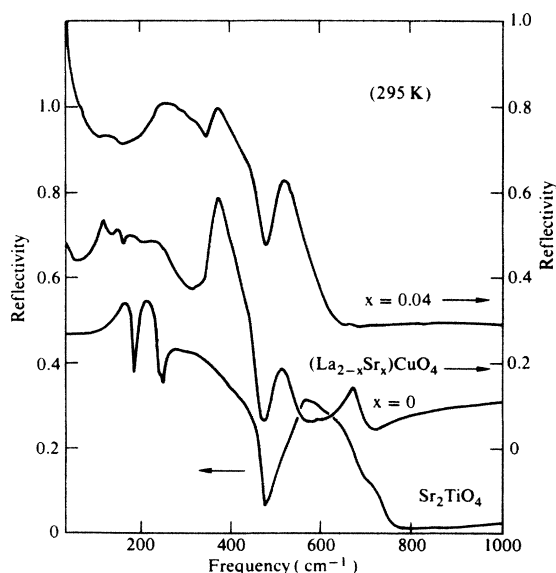


FIG. 6. The frequency dependence of the reflectivity of  $Sr_2TiO_4$  compared to that of  $(La_{2-x}Sr_x)CuO_4$  for two different values of  $x$ . The reflectivity scale for the latter two materials is shifted for clarity; it is to the right.

higher-frequency  $E_u$  mode.  $(La_{2-x}Sr_x)CuO_4$  might be expected to be less ionic with a smaller effective charge, which will reduce the TO-LO splittings making it easier to identify some of the modes since overlap between the  $A_{2u}$  and  $E_u$  reststrahlen bands will be decreased. Indeed, this appears to be the case for  $x=0$ . As can be seen in Fig. 6, the high-frequency composite response in  $Sr_2TiO_4$ , when observed in  $(La_{2-x}Sr_x)CuO_4$ , clearly divides into two, relatively sharp reststrahlen bands. This immediately suggests that the lower one ( $\approx 510$   $cm^{-1}$ ) is due to an  $A_{2u}$  phonon and that the higher one ( $\approx 670$   $cm^{-1}$ ) is due to an  $E_u$  phonon. This assignment is confirmed by the results for the  $x=0.04$  sample which contains many free electrons. Since this electron plasma is free to move in the  $xy$  plane and not in the  $z$  direction,<sup>21,22</sup> it couples much more strongly to  $E_u$  phonons than to  $A_{2u}$  phonons.<sup>21</sup> This happens because for  $E_u$  phonons, the atoms vibrate in the  $xy$  plane while  $A_{2u}$  phonons vibrate along the  $z$  direction. This point has been also discussed previously.<sup>24</sup> Thus, the assignment and frequencies (from a Kramers-Kronig analysis) can be obtained, and the mode frequencies from the  $x=0$  sample, for the two highest-frequency modes are listed in Table II, part (c).

Other features for  $x=0$  now become clearer. Although this material is actually orthorhombic, we are treating it as if it is tetragonal. The differences between the orthorhombic and tetragonal structure are small,<sup>23</sup> resulting from a soft Brillouin-zone edge phonon.<sup>29</sup> There is no hint of a splitting in the  $\approx 650$   $cm^{-1}$  reststrahlen feature; yet being a doubly degenerate  $E_u$  mode in the tetragonal phase, for an orthorhombic crystal it must split into two singly degenerate modes (transforming as the  $B_{2u}$  and  $B_{3u}$  irreducible representations of the  $mmm-D_{2h}$  point group). Thus, the lower symmetry in  $(La_{2-x}Sr_x)CuO_4$  has only a small effect on splitting of the modes from those in the tetragonal phase. The same effect has been noted<sup>10</sup> for orthorhombic  $YBa_2Cu_3O_7$ , which can be made into tetragonal  $YBa_2Cu_3O_6$ .

The lowest-frequency feature ( $\approx 120$   $cm^{-1}$ ) in the  $x=0$  material is probably an  $E_u$  mode as in  $Sr_2TiO_4$ , the fluorides, and the chlorides (Fig. 5). Again, further evidence for this assignment is obtained from  $x=0.04$  sample where the free electrons couple to the  $E_u$  modes. For  $x=0.04$  there is a small remnant of this feature still observable at the same frequency. Thus, in Table II, part (c) we assign this lowest-frequency mode at  $117$   $cm^{-1}$  to the  $E_u(LO)$  mode obtained from the Kramers-Kronig analysis of either the  $x=0$  or  $0.04$  samples, and the corresponding  $E_u(LO)$  mode at  $\approx 140$   $cm^{-1}$  from the latter sample.

For the  $x=0.04$  sample, except for this small feature at  $\approx 117$   $cm^{-1}$  discussed above, there are three prominent reflectivity bands. We have already assigned the highest-frequency band, starting at  $510$   $cm^{-1}$  [Table II, part (c)], to an  $A_{2u}$  band. We assign the other two features to the allowed  $A_{2u}$  phonons. The two highest-frequency  $A_{2u}$  bands are clearly visible in the  $x=0$  sample, and the lowest  $A_{2u}$  band is discernible even though it overlaps with  $E_u$  bands, as expected. Frequency values, from a Kramers-Kronig analysis, for these lower-frequency  $A_{2u}$  bands, from the  $x=0.04$  sample, are listed in Table II,

part (c). However, we are not certain of the lowest-frequency  $A_{2u}(\text{LO})$  mode because of the apparent large shift between the  $x=0$  and  $x=0.04$  sample. Hence, in Table II, part (c) we list this value as (340).

#### A. Normal modes

There are large relative changes in the  $B-\text{O}_z$  compared to  $B-\text{O}_x$  distances in  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$  when compared to the insulating materials. These changes must affect certain eigenvectors. For example, based on the lattice dynamics of the fluorides,<sup>17</sup> in  $\text{Sr}_2\text{TiO}_4$  the highest-frequency Raman-active  $A_{1g}$  eigenvector is principally  $\text{O}_z$  motion along the  $z$  direction with very little Sr motion. The lowest-frequency  $A_{1g}$  motion, however, is orthogonal, involving mostly Sr motion. For  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$  this eigenvector must change because of the substantial decrease in the  $\text{Cu}-\text{O}_z$  force constant relative to the  $\text{O}_z-\text{La}$  force constant. Consequently, the highest-frequency  $A_{1g}$  motion involves out-of-phase motion of  $\text{O}_z$  against La along the  $z$  direction, while in-phase motion of these two atoms takes place in the lowest-frequency  $A_{1g}$  mode. For the highest-frequency  $E_g$  modes, the La and  $\text{O}_z$  will move in the  $xy$  plane out of phase with each other. While for the lowest-frequency mode, these two atoms will move in phase. Of course, since the La atom is so much heavier than the  $\text{O}_z$  atom, it will move much less in these motions.

For the infrared-active  $3A_{2u}$  and  $4E_u$  modes it is more difficult to generalize because the motion of all the atoms is involved. However, the highest-frequency  $E_u$  mode in  $\text{Sr}_2\text{TiO}_4$  and  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$  should be essentially the same, with the Cu moving against  $\text{O}_x$  and  $\text{O}_y$ . The highest-frequency  $A_{2u}$  motion in  $\text{Sr}_2\text{TiO}_4$  is primarily Ti against a rigid oxygen octahedron. However, in  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$  the eigenvector must be rather different because of the relatively stronger forces between  $\text{La}-\text{O}_z$  than between  $\text{Cu}-\text{O}_z$ . Some of the lattice dynamical calculations<sup>30-33</sup> come to similar conclusions.

## VI. CONCLUSIONS

We have measured the Raman modes and infrared reflectivity in ceramic  $\text{Sr}_2\text{TiO}_4$ , a material that is isostruc-

tural with the superconductor  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$ . Since the material has a center of symmetry, we expect, and observe, sharp Raman-active modes, and determine all of the four allowed modes. Interpretation of the infrared reflectivity is more difficult since reststrahlen bands extend from the TO to the LO frequency for all  $3A_{2u}$  and all  $4E_u$  bands. Although all of the  $3A_{2u}$  bands must occur in different frequency ranges, as must all  $4E_u$  bands, the difficulty in analyzing the data occurs because  $A_{2u}$  and  $E_u$  bands can overlap each other.

With the help of the single-crystal infrared results in the isostructural fluorides and chlorides, for  $\text{Sr}_2\text{TiO}_4$  we have been able to assign two  $E_u$  modes and one  $A_{2u}$  mode, and with less certainty one more of each type. Thus, five out of the seven allowed infrared modes can be assigned using the ceramic samples.

The assignment of modes from the infrared reflectivity measurements in  $(\text{La}_{2-x}\text{Sr}_x)\text{CuO}_4$  is easier for two reasons. First, the smaller effective charges reduces the TO-LO splittings. This makes the reststrahlen bands narrower, causing less interference between those of the  $A_{2u}$ - and  $E_u$ -type bands. Second, the structure has a free-electron-like plasma response only on the  $xy$  plane, and not perpendicular to it. Thus, the free electrons couple much more strongly to the  $E_u$  phonons, because their motion is also in the  $xy$  plane. The free carrier concentration can be increased by increasing  $x$  from 0, thereby increasing the plasma frequency, which permits easier observation of all of the  $3A_{2u}$  modes (Fig. 6). From the combined spectra for different  $x$ , the highest- and lowest-frequency  $E_u$  modes also may be identified. Thus, we determine five of the seven infrared-active modes in the superconducting material.

## ACKNOWLEDGMENTS

One of us (G.B.) gratefully acknowledges financial support from the Alexander von Humboldt Foundation, and the hospitality and stimulation of colleagues at the Max Planck Institute for Solid State Research where some of this work was carried out. Discussions with Professor E. Burstein, T. Brun, R. E. Cohen, and W. Weber are gratefully acknowledged.

<sup>1</sup>J. G. Bednorz and K. A. Müller, *Z. Phys. B* **64**, 189 (1986).

<sup>2</sup>For a review of the early work in this field, see A. Khurana, *Phys. Today* **40** (No. 4), 17 (1987).

<sup>3</sup>Also see A. Khurana, *Phys. Today* **40** (No. 6), 17 (1987).

<sup>4</sup>See most solid state books for a discussion of the Bardeen-Cooper-Schrieffer theory.

<sup>5</sup>For example, see B. P. Schweiss, B. Renker, E. Schneider, and W. Reichardt, in *Superconductivity in d- and f-Band Metals* edited by D. H. Douglass (Plenum, New York, 1976), p. 189.

<sup>6</sup>G. Burns and B. A. Scott, *Phys. Rev. Lett.* **25**, 167 (1970); G. Burns, *Mater. Res. Bull.* **6**, 923 (1971).

<sup>7</sup>I. G. Lang and U. S. Pashabekova, *Fiz. Tverd. Tela* **6**, 3640 (1964) [*Sov. Phys. Solid State* **6**, 2913 (1965)]; K. R. Allakhverdiev, S. S. Babaev, E. Yu, and M. M. Tagyev, *Phys. Status Solidi (b)* **96**, 177 (1979).

<sup>8</sup>G. Burns, F. H. Dacol, and M. W. Shafer, *Solid State Commun.* **62**, 687 (1987).

<sup>9</sup>U. Venkateswaran, K. Strössner, K. Syassen, G. Burns, and M. W. Shafer, *Solid State Commun.* **64**, 1273 (1987).

<sup>10</sup>G. Burns, F. H. Dacol, P. Freitas, T. S. Plaskett, and W. König, *Solid State Commun.* **64**, 471 (1987).

<sup>11</sup>S. N. Ruddlesden and P. Popper, *Acta Crystallogr.* **10**, 538 (1957).

<sup>12</sup>G. J. McCarthy, W. B. White, and R. Roy, *J. Am. Ceram. Soc.* **52**, 463 (1969).

<sup>13</sup>R. W. G. Wyckoff, *Crystal Structures* (Wiley, New York, 1975), Vol. 3, p. 63; A. F. Wells, *Structural Inorganic Chemistry* (Oxford Univ. Press, New York, 1975); F. S. Glasco, *Structure and Properties of Inorganic Solids* (Pergamon, New York, 1970).

- <sup>14</sup>G. Burns and A. M. Glazer, *Space Groups for Solid State Scientists* (Academic, New York, 1978); G. Burns, *Introduction to Group Theory with Applications* (Academic, New York, 1977).
- <sup>15</sup>*International Tables for X-ray Crystallography*, edited by N. F. M. Hendry and K. Lonsdale (Kynoch, London, 1952, 1965, 1968), Vol. I.
- <sup>16</sup>D. J. Tomes, W. O'Sullivan, and H. J. Guggenheim, *Solid State Commun.* **14**, 715 (1974); M. Kaneko, G. Kuwabara, and A. Misu, *ibid.* **18**, 1085 (1976).
- <sup>17</sup>H. Rauh and R. Geick, *Phys. Status Solidi (b)* **127**, 55 (1985).
- <sup>18</sup>H. Büger, K. Strobel, R. Geick, and W. Müller-Lierheim, *J. Phys. C* **9**, 4213 (1976).
- <sup>19</sup>K. Strobel and R. Geick, *J. Phys. C* **15**, 2105 (1982).
- <sup>20</sup>H. Rauh, R. Geick, N. Lehner, J. Bouillot, and G. Eckold, *Phys. Status Solidi (b)* **115**, 463 (1983).
- <sup>21</sup>J. M. Bassat, P. Odier, and F. Garvais, *Phys. Rev. B* **35**, 7126 (1987).
- <sup>22</sup>C. N. R. Rao, D. J. Buttrey, N. Otsuka, P. Ganguly, H. R. Harrison, C. J. Sandberg, and J. M. Honig, *J. Solid State Chem.* **52**, 266 (1984).
- <sup>23</sup>R. J. Cava, A. Santoro, D. W. Johnson, Jr., and W. W. Rhodes, *Phys. Rev. B* **35**, 6716 (1987); P. Day, M. Rosseinsky, K. Prassides, W. I. F. David, O. Moze, and A. Soper, *J. Phys. C* **20**, L429 (1987).
- <sup>24</sup>Z. Schlesinger, R. T. Collins, and M. W. Shafer, *Phys. Rev. B* **35**, 7232 (1987).
- <sup>25</sup>D. A. Bonn, J. E. Greedan, C. V. Stager, T. Timusk, M. G. Doss, S. L. Herr, K. Kamaras, C. D. Porter, D. B. Tanner, J. M. Tarascon, W. R. McKinnon, and L. H. Greene, *Phys. Rev. B* **35**, 8843 (1987).
- <sup>26</sup>H. P. Geserich, G. Scheiber, and B. Renker, *Solid State Commun.* **63**, 657 (1987).
- <sup>27</sup>S. Sugai, S. Uchida, H. Takagi, K. Kitazawa, and S. Tanaka, *Jpn. J. Appl. Phys.* **26**, L879 (1987).
- <sup>28</sup>R. M. Fleming, B. Batlogg, R. J. Cava, and E. A. Rietman, *Phys. Rev. B* **35**, 13 (1987); K. Kang, G. Collin, M. Ribault, J. Friedel, J. Jerome, J. M. Bassat, J. P. Coutoures, and Ph. Odier, *J. Phys. (Paris)* **48**, 1181 (1987).
- <sup>29</sup>R. J. Birgeneau, C. Y. Chen, D. R. Gabbe, H. P. Janssen, M. A. Kastner, C. J. Peters, P. J. Picone, Tineke Thio, T. R. Thurston, H. L. Tuller, J. D. Axe, P. Boni, and G. Shirane, *Phys. Rev. Lett.* **59**, 1329 (1987).
- <sup>30</sup>T. Brun, M. Grimsditch, K. E. Gray, R. Bhadra, and V. Maroni, *Phys. Rev. B* **35**, 16 (1987).
- <sup>31</sup>W. Weber, *Phys. Rev. Lett.* **58**, 1371 (1987).
- <sup>32</sup>R. E. Cohn, W. E. Pickett, H. Krakauer, L. L. Boyer, *Physica B* (to be published).
- <sup>33</sup>J. Prade, A. D. Kulkarni, F. W. de Wette, W. Kress, M. Cardona, R. Reiger, and U. Schröder, *Solid State Commun.* (to be published).