## Symmetry-induced local magnetic moments in icosahedral Al-Mn alloys

M. E. McHenry,\* D. D. Vvedensky,<sup>†</sup> M. E. Eberhart,<sup>‡</sup> and R. C. O'Handley\*

Materials Science and Technology Division, Los Alamos National Laboratory, Los Alamos, New Mexico 87545

(Received 19 January 1988)

Electronic-structure calculations of icosahedral Al-Mn clusters suggest well-defined quasiatomic Mn moments. The driving force for moment formation is a large Mn d density of states at the Fermi energy in a non-spin-polarized calculation. The onset of Mn magnetism results directly from the local coordination in the icosahedral environment, which does not split the Mn 3d levels and inhibits Al-Mn *sp-d* hybridization.

The experiments of Shechtman et al.<sup>1</sup> demonstrating icosahedral symmetry in rapidly solidified Al<sub>0.86</sub>Mn<sub>0.14</sub> have produced a torrent of studies aimed at elucidating the structure and properties of quasiperiodic materials. Many theoretical approaches to characterizing structure-property relationships in quasicrystals have been based on the Penrose lattice,<sup>2,3</sup> a nonperiodic tiling of the two-dimensional plane, and the one-dimensional analogue, the Fibonacci lattice.<sup>4</sup> The methods employed have been by necessity schematic, e.g., tight-binding Hamiltonians for noninteracting electrons,<sup>5</sup> the philosophy being to identify topologically induced features in the electronic structure which are expected to persist qualitatively (even in higher spatial dimensions) when more realistic computational techniques are used. One of the most striking features which has emerged from the electronic structure of a two-dimensional Penrose lattice<sup>2</sup> is the occurrence in the density of states of a peak of zero width associated with localized states which appear to be induced only by local lattice structure (i.e., not by the quasiperiodicity of the lattice).

While a comprehensive characterization of the electronic structure of a three-dimensional quasicrystal comparable to state-of-the-art band-structure calculations is not yet feasible, inroads can nevertheless be made by considering features in the electronic structure resulting directly from local icosahedral symmetry. A prior calculation by McHenry et al.<sup>6</sup> compared the electronic structure of Al and Al-Mn clusters chosen to represent local environments of fcc (cubo-octahedral,  $O_h$  symmetry) and quasicrystalline (icosahedral,  $I_h$  symmetry) phases, respectively. The authors inferred that the  $I_h$  Al clusters were more stable than their  $O_h$  counterparts and found that Mn atoms placed at the center of the  $I_h$  cluster produced a sharp predominantly Mn d peak in the density of states near the Fermi energy  $E_F$ . The large density at  $E_F$ pointed to the inherent instability of this configuration, though it has been noted that moment formation could enhance the stability of such a configuration.<sup>7</sup>

This Brief Report goes beyond earlier results of McHenry *et al.*<sup>6</sup> in three respects. First, additional calculations are done in the spin-unrestricted mode so that local Mn moments can be evaluated. Second, we focus

on the consequences of  $I_h$  symmetry for magnetism and use our results to interpret recent experimental data. Finally, we find important new aspects of chemical bonding in icosahedral Al-Mn that address the role of chemistry in icosahedral stability, a feature lacking in other recent calculations.<sup>2-5</sup>

This work is motivated in part by recent magnetic measurements in quasicrystals which reveal the existence of a large local moment on the manganese site  $^{8-10}$  as well as spin-glass behavior at low temperature,<sup>8</sup> neither of which is evident in the corresponding crystalline phases of Al-Mn. Further, heat capacity, electron transport, and optical properties have been reported to be strongly influenced by scattering by resonant Mn d states in the conduction band, which have been interpreted as arising from the existence of Mn virtual bound states at the Fermi level in the quasicrystals.<sup>11,12</sup> These results and earlier heat-capacity results<sup>9</sup> suggest that only a small fraction of the manganese atoms contribute to magnetism and to dresonance states in these materials. This behavior suggests that in quasicrystals, even with Mn concentrations of 14–22 at. %, a certain fraction of the Mn sites behaves like dilute impurities in an Al host.

In this Brief Report one possible explanation of this behavior is discussed. In lower-symmetry sites Mn atoms are shown to effectively form directional p-d hybrid bonds which removed Mn d character from the Fermi level, thus reducing the moment-forming capabilities of the Mn sites, an idea consistent with expectations based on crystal-field effects alone. In Mn-centered icosahedral Al the absence of crystal-field effects inhibits the breakdown of intra-atomic correlations (i.e., the d orbitals transform according to the fully fivefold degenerate  $H_g$ irreducible representation of the  $I_h$  group). Further, markedly diminished sp-d matrix elements result from the orientational incommensurability between the 3d orbitals and the fivefold rotational axes of the icosahedron. It is possible that the above-mentioned anomalies in magnetic and resistive properties may also be adequately explained by clustering of Mn atoms in the quasicrystals. However, the influence of symmetry upon the moment formation of Mn in aluminum provides an interesting and provocative perspective on a classic problem in solidstate physics.

Our calculations were carried out with the multiplescattering  $X\alpha$  (MS  $X\alpha$ ) cluster technique to examine the electronic structure of an Al<sub>18</sub>Mn cluster representing a Mn impurity coordinated by first- and second-nearestneighbor Al atoms in  $O_h$  symmetry, Al<sub>32</sub>Mn, which represents Mn surrounded by two coordination spheres in icosahedral symmetry, and MnAl<sub>9</sub>, which represents Mn in a low-symmetry  $(D_{3h})$  environment, possibly representative of an intericosahedral region. Figures 1(a) and 1(b) show the MS-X $\alpha$  energy levels for the non-spinpolarized  $O_h$  Al<sub>18</sub>Mn cluster as compared with those of the  $I_h$  Al<sub>32</sub>Mn cluster. We note first that the groundstate configuration of the Al<sub>18</sub>Mn is nonmagnetic. A cluster moment of 1  $\mu_B$  is observed, but this can be explained simply as the result of the fact that Mn has an odd number of electrons and successive Al shells always have an even number of electrons in  $O_h$  symmetry. Thus the cluster moment may be an artifact of an unpaired electron on the manganese site, which disappears if one allows fractional occupation. Recent spin-polarized calculations on an  $O_h$  MnAl<sub>18</sub> cluster using a Gaussian orbital basis set<sup>13</sup> also revealed a cluster moment of  $1\mu_B$  but with a  $1.5\mu_B$  local moment on Mn which was partially compensated by an antiferromagnetic coupling to the Al conduction band. This small local moment is inconsistent with experimental observations of nonmagnetic Mn impurities in fcc Al and may well also be an artifact of the odd number of electrons in a finite cluster as was noted in Ref. 13. For systems near the magnetic instability as is Mn in fcc Al (it is observed to be a spinfluctuation system),<sup>14</sup> the exclusion of host band-structure



effects in the cluster representation due to the small spatial extent of the cluster may well prove to be a less than adequate representation in these systems. However, features in the bonding do help to explain the lack of a strong local moment in Mn as well as the small or nonexistent exchange splitting in these systems. The energy levels for the  $D_{3h}$  cluster (not shown) are characterized by eigenfunctions showing strong *d-sp* (Mn-Al) mixing throughout the manifold. This cluster could not be made to sustain a moment.

In the  $O_h$  Al<sub>18</sub>Mn cluster the hybridization between Al-Mn states is observed predominantly in the upper bands. The low-lying states are of Als - Mnd bonding character, whereas the  $3e_g$  and  $2t_{2g}$  levels just below the Fermi level are Al p – Mn d bonding in character. The octahedral crystal field of the neighboring Al atoms is such that the Mn d states are split into threefold degenerate  $t_{2g}$  states and twofold degenerate  $e_g$  states (sixfold and fourfold degenerate counting spin degeneracy). These representations are uniquely qualified for directed nearest-neighbor  $\sigma$  bond formation between Al p and Mn d states in the case of the  $2t_{2g}$  level and between Mn d and second nearest neighbor Al p states for the  $3e_g$  level as illustrated in Fig. 2. The effect of this bonding on moment formation is twofold: (1) The p-d hybrid bonding results in lower-energy states, which effectively removes



FIG. 1. The MS  $X\alpha$  energy levels for (a) MnAl<sub>12</sub>, (b) MnAl<sub>18</sub>, and (c) MnAl<sub>12</sub> (spin-polarized) labeled according to the irreducible representations of the  $I_h$  [(a) and (c)] on  $O_h$  [(b)] point groups. Occupied and unoccupied orbitals are indicated by solid and dashed lines, respectively.

FIG. 2. The (a)  $2t_{2g}$  orbitals of Al<sub>18</sub>Mn plotted in the xy plane and (b) the  $4H_g$  orbital of Al<sub>32</sub>Mn plotted in the xy plane. Positive and negative values of the wave function are indicated by solid and dashed lines, respectively, and the contour values are  $\pm 0.081, \pm 0.027, \pm 0.009$ , and  $\pm 0.003$ .

Mn d states from the Fermi level; (2) the  $t_{2g}$ - $e_g$  splitting results from the disparity in bonding between the Mn with first and second nearest neighbors, respectively and thus acts to broaden the manifold associated with Mn d character as compared with a more atomiclike configuration. Because of these bonding states the total Mn-Al manifold is slightly more stable for the  $O_h$  cluster than for the  $I_h$  cluster.

In Fig. 1 are shown the MS  $X\alpha$  energy levels for the  $Al_{32}Mn$  cluster in  $I_h$  symmetry as well as for the spinpolarized counterpart. Two pertinent features of the non-spin-polarized states should be emphasized. First a contraction of and movement to slightly lower energies of the Al manifold in the  $I_h$  clusters leaves atomiclike Mn d states at the Fermi level. Second this Mn d manifold is degenerate in the  $I_h$  crystal field and forms no significant bonding with the aluminum p orbitals as shown in Fig. 2. This configuration thus yields a sharply peaked narrow Mn d band at  $E_F$  (Fig. 1) which is an ideal situation for local moment formation. As shown in Fig. 1(c) this  $H_g$ state drives the Al<sub>32</sub>Mn complex to develop a sizable local moment of  $3.6\mu_B$  on the Mn site, which is partly compensated by an antiferromagnetic coupling with the Al conduction electrons, so that a total cluster moment of  $3.0\mu_B$  is observed. The overall exchange splitting induced on the Al subband is small but the exchange splitting associated with the Mn d states is as much as  $\sim 8 \text{ eV}$ , and is therefore nearly atomiclike.

Although Mn shows well-defined magnetic behavior in both an  $I_h$  aluminum environment and an  $O_h$  copper environment with calculated magnetic moments of  $3.5\mu_B$ (Ref. 15) and  $3.6\mu_B$ , respectively, there are two notable differences: (1) The Mn exchange splitting is considerably larger for  $I_h$  Al<sub>32</sub>Mn (~8 eV) than for  $O_h$  Cu<sub>18</sub>Mn (~2 eV); (2) the 3d charge density as shown in Fig. 2 for the highest occupied  $H_g$  orbital in  $I_h$  Al<sub>32</sub>Mn is considerably more localized around Mn than the corresponding  $T_{2g}$  and  $E_g$  orbitals in Cu<sub>18</sub>Mn.<sup>15</sup>

Whether one adopts the Friedel-Anderson virtual bound-state model<sup>16,17</sup> or the Schrieffer-Hirst<sup>18,19</sup> configuration-based approach, the electronic structure of dilute 3d impurities in metallic hosts results from the competition between intra-atomic correlations of the impurity and from covalent (*sp-d*) mixing between the 3d electrons and the conduction electrons of the host. This delicate balance between competing interactions is illustrated by the contrasting electronic structures of the 3d transition metals when dissolved in copper or in aluminum. The

well-defined moments associated with several transition metals in a copper environment have been adequately reproduced in earlier calculations.<sup>15</sup> The magnetism of these impurities in crystalline Al is nearly completely absent, while local moments are stable in our calculations for Al-Mn in  $I_h$  symmetry. The stability of the magnetic moment in either the Cu and  $I_h$  aluminum environments arises from the dominance of the Mn intra-atomic correlation energy over the impurity-host bonding energy. As discussed earlier the icosahedral environment promotes magnetism of Mn in Al by inhibiting both the breakdown of intra-atomic correlations and the formation of impurity-host bonding. The effect is clearly promoted by symmetry rather than chemistry, as the associated cubooctahedral complex exhibits much weaker magnetism. The precursor to magnetism in the icosahedral case can be understood by the character of the non-spin-polarized density of states, which reveals a high density of Mn d character at  $E_F$ .<sup>6</sup> The Mn character in  $O_h$  Al is below  $E_F$  by virtue of its bonding character and is broader due to the  $t_{2g}$ - $e_g$  splitting.

We conclude by saying that the contrast in the electronic states of Mn in  $I_h$  and  $O_h$  environments provides a new perspective to the problem of characterizing local magnetic moments in metals by highlighting the role of local symmetry as a barrier or aid to sp-d mixing. We believe this offers a plausible explanation for observed local moments and the impuritylike scattering for certain Mn sites in quasicrystalline Al-Mn. Admittedly this is not the only explanation. It has been suggested that the increased Mn moment could be due to Mn-Mn clustering. Such clustering is manifest in Mn-Al phases ( $\beta$ -Al-Mn-Si) but with only modest increases in the moment.<sup>20</sup> Further, the moment in quasicrystalline Mn-Al is significantly larger than in its crystalline counterparts. As only a fraction of the Mn atoms present in Al-Mn quasicrystals is required to account for the observed moment formation, it can be inferred, from the calculation on the low-symmetry  $D_{3h}$  cluster, that the remaining Mn atoms occupy low-symmetry, not moment-forming sites, where through bond formation the bond strain inherent to the nonperiodic structure is relieved.

Three of us (M.E.M., D.D.V., and R.C.O.) are grateful for the hospitality of the Center for Materials Science at the Los Alamos National Laboratory, where this work was carried out.

- \*Permanent address: Department of Materials Science and Engineering, Massachusetts Institute of Technology, Cambridge, MA 02139.
- <sup>†</sup>Permanent address: The Blackett Laboratory, Imperial College, London SW7 2BZ, United Kingdom.
- <sup>‡</sup>Also at Department of Materials Science and Engineering, Massachusetts Institute of Technology, Cambridge, MA 02139.
- <sup>1</sup>D. S. Shechtman, L. Blech, D. Gratias, and J. W. Cahn, Phys.

Rev. Lett. 53, 1951 (1984).

- <sup>2</sup>T. C. Choy, Phys. Rev. Lett. 55, 2915 (1985).
- <sup>3</sup>M. Kohmoto and B. Sutherland, Phys. Rev. Lett. 56, 2740 (1986).
- <sup>4</sup>M. Kohmoto and J. R. Banavar, Phys. Rev. B 34, 563 (1986).
- <sup>5</sup>D. R. Nelson and M. Widom, Nucl. Phys. B 240, 113 (1984).
- <sup>6</sup>M. E. McHenry, M. E. Eberhart, R. C. O'Handley, and K. H. Johnson, Phys. Rev. Lett. **56**, 81 (1986).
- <sup>7</sup>M. E. McHenry and R. C. O'Handley, in Science and Technolo-

gy of Rapidly Quenched Alloys, Vol. 80 of MRS Symposium Proceedings, edited by M. Tenhover, W. L. Johnson, and L. E. Tanner (Materials Research Society, Pittsburgh, 1987), p. 363.

- <sup>8</sup>J. J. Hauser, H. S. Chen, and J. Y. Wasczak, Phys. Rev. B 33, 3577 (1984).
- <sup>9</sup>F. L. A. Machado, W. G. Clark, L. J. Azevedo, D. P. Yang, W. A. Hines, J. L. Budnick, and M. X. Quan, Solid State Commun. 61, 145 (1987).
- <sup>10</sup>S. E. Youngquist, P. F. Miceli, D. G. Wiezler, H. Zabel, and H. L. Fraser, Phys. Rev. B 34, 2960 (1986).
- <sup>11</sup>F. Cyrot-Lackmann, P. Gonmi, and A. Pasturel, Solid State Commun. **59**, 11 (1986).
- <sup>12</sup>C. Berger, J. C. Lasjaunias, J. L. Tholence, D. Pavuna, and P.

Germi (unpublished).

- <sup>13</sup>D. Bagayoko, N. Brewer, D. Kanhere, and J. Callahan (unpublished).
- <sup>14</sup>G. Gruner, Adv. Phys. 23, 941 (1974).
- <sup>15</sup>D. D. Vvedensky, M. E. Eberhart, and M. E. McHenry, Phys. Rev. B 35, 2061 (1987).
- <sup>16</sup>J. Friedel, Nuovo Cimento, Suppl. 7, 287 (1958).
- <sup>17</sup>P. W. Anderson, Phys. Rev. **124**, 41 (1961).
- <sup>18</sup>J. R. Schrieffer, J. Appl. Phys. 38, 1143 (1967).
- <sup>19</sup>L. L. Hirst, Phys. Kondens. Mater. **11**, 250 (1970); Z. Phys. **224**, 230 (1971); **245**, 378 (1971).
- <sup>20</sup>M. E. McHenry, A. Chow, R. Chatterjee, and R. C. O'Handley (unpublished).