Optical properties of GeO₂

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Interpretations for the energies corresponding to the peaks in the reflectivity and the imaginary part (ϵ_2) of the complex dielectric constant ($\epsilon = \epsilon_1 + i\epsilon_2$) have been sought successfully using Penn-like model calculations. The valence-band-*d*-band coupling present in Ge may contribute to the effective number of electrons in GeO₂ at the observed peaks at high energies.

I. INTRODUCTION

Unlike SiO₂, which has a tremendous importance in the semiconductor device industry as well as in glass manufacture, GeO₂ has had limited applications, primarily as an experimental optical-fiber material. Applications have been limited by the hygroscopic character of glassy GeO₂. Because of the fact that the temperature of melting of GeO₂ ($T_m = 1150$ °C) is much less than that of SiO₂ ($T_m = 1750$ °C), preparation of GeO₂ in the glassy form over a range of temperatures and oxygen partial pressures is much less difficult than that of SiO₂. The kinetics of crystallization of GeO₂ and the effect of stoichiometry on the kinetics seem to be well understood.¹ Recently reported studies² of the x-ray absorption have shown that the Ge-O and Ge-Ge distances in the GeO₂ glass are close to those of trigonal GeO₂ crystals.

The types and concentrations of intrinsic paramagnetic states in GeO₂ glasses equilibrated at temperatures T_{ϕ} between T_m , the melting temperature, and T_b , the boiling temperature, and cooled at various rates have been reported.³⁻⁵ The intensity of an optical-absorption band at 245 nm (first observed by Garino-Cannina⁶) has been shown to have an Arrhenius dependence on the glass equilibration temperature (T_{ϕ}) with an activation energy of 2.3 eV. This band has been attributed to an oxygen deficiency of GeO₂.⁴ However, some very basic problems of GeO₂ remain unresolved. One such problem concerns an analysis of the optical spectra of GeO₂. Here, we report an analysis of the reflectivity spectra of GeO₂.

Pajasova⁷ has reported detailed experimental measurements of the reflectivity spectra of crystalline (c) and glassy (g) GeO₂ in the photon-energy (E) range of 0.5-25 eV. Utilizing the Kramers-Kronig relation, the optical constants including the ϵ_2 -E spectra have been derived. However, no interpretation for the observed peaks and the corresponding energy has been offered. In Figs. 1 and 2, the reflectivity spectra and the ϵ_2 -E spectra, reported earlier by Pajasova, respectively, for c- and g-GeO₂ are presented. As is expected because of the crystalline nature, c-GeO₂ exhibits peaks which are sharper than those of g-GeO₂. However, the second peak in the ϵ_2 -E spectra of c-GeO₂ is spread out in energy. The maxima in intensities and the corresponding energy in Figs. 1 and 2 are listed in Table I. The reflectivity spectra of c-GeO₂ exhibit several kinks, at 5.94 eV (R = 11.5%), 7.25 eV (R = 10.4%), and 12.2 eV (R = 13.5%). It is interesting to note that the first two kinks are absent in the reflectivity spectra of g-GeO₂ while the third kink is shifted in energy; 11.59 eV (R = 11.05%). This may be attributed to the absence of translational symmetry in glasses.

While the maximum reflectance R_{max} is higher for c-GeO₂ than that of g-GeO₂, the energies corresponding to the peaks are the same (Table I). In Fig. 2, ϵ_2 remains practically zero until 4.85 eV, and increases abruptly to a value of 0.9 in the energy range of 4.85-6.6 eV for c-GeO₂. In a narrow energy range of 6.6-7.0 eV, ϵ_2 increases from 0.9 to 2.34. The kinks observed in the reflectivity spectra of c-GeO₂ are all evidenced in the ϵ_2 -E spectra, 6.6 eV (ϵ_2 =0.9), 7.95 eV (ϵ_2 =1.6), and 14 eV (ϵ_2 =2.34), except that the kinks are shifted in energy. None of the kinks are resolved in the ϵ_2 -E spectra of g-GeO₂. While the $\epsilon_{2,\text{max}}$ in c-GeO₂ has a higher value than in g-GeO₂, the energy corresponding to the first



FIG. 1. Reflectivity spectra of crystalline and glassy GeO_2 (from Pajasova, Ref. 7).

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FIG. 2. ϵ_2 as a function of photon energy for crystalline and glassy GeO₂ [see Pajasova (Ref. 7) for further details].

peak (\overline{E}_1) is the same for both phases. The second peak in the ϵ_2 -*E* spectra of *c*-GeO₂ is spread out in energy and occurs at an energy higher than that of *g*-GeO₂ (Table I).

II. APPLICATION OF PENN MODEL

It is well known that the reflectivity spectra or the ϵ_2 - E spectra of a given material are determined by its band structure. By definition, an intensity maximum in ϵ_2 (or R) in the ϵ_2 -E spectra (or R-E spectra) represents a maximum number in the optically induced electronic transitions in the material. The energy corresponding to the peak should therefore correspond to a band-to-band energy difference or a band gap. Since this is a macroscopic gap,⁸ it should be related to the high-frequency dielectric constant ϵ_{∞} ($\simeq n^2$), where n is the refractive index. Several models have been proposed⁸⁻¹⁰ to interpret

Several models have been proposed⁸⁻¹⁰ to interpret the frequency and wave-vector dependence of the dielectric function. All these models have, however, been proposed for elemental semiconductors. Extrapolation of the applicability of these models to amorphous semiconductors,¹¹ and narrow- and wide-gap materials including alkali halides¹²⁻¹⁴ has been carried out with reasonable success. Here, we demonstrate the applicability of one such model to *c*- and *g*-GeO₂.

For a model semiconductor, the high-frequency dielectric constant is given by 8

TABLE I. Reflectivity and ϵ_2 data of crystalline and glassy GeO₂. See Figs. 1 and 2 for definitions of E_1 , E_2 , \overline{E}_1 , and \overline{E}_2 . $E_R = (E_1 + E_2)/2$; $\overline{E} = (\overline{E}_1 + \overline{E}_2)/2$.

$\frac{-\pi}{2}$	- C-O		
Property	c-GeO ₂	g-GeO ₂	
$R_{\rm max}$ (%)	15.44	12.50	
E_1 (eV)	6.52	6.52	
$R_{\rm max}$ (%)	14.70	11.62	
E_2 (eV)	10.72	10.72	
$\epsilon_{2, \max}$	2.34	2.22	
\overline{E}_1 (eV)	7.03	7.03	
$\epsilon_{2,\max}$	2.53	2.38	
\overline{E}_2 (eV)	11.72-12.81	11.09	

$$\epsilon_{\infty} = 1 + (\hbar\omega_P / E_P)^2 [1 - (E_P / 4E_F) + \frac{1}{3} (E_P / 4E_F)^2]$$
(1)

or

$$1/E_P = 1/8E_F + [(\epsilon_{\infty} - 1)/(\hbar\omega_p)^2 - \frac{1}{3}(1/8E_F)^2]^{1/2},$$
(2)

where the valence-electron plasmon energy $\hbar \omega_p$ is given by¹⁵

$$\hbar\omega_n = 28.8 (N_{\rm eff} \rho / M)^{1/2}$$
,

 E_P is the Penn gap⁸ or the macroscopic gap accounting for all the possible optically induced electronic transitions in the material, E_F is the Fermi energy evaluated using the relation¹³

$$E_{\rm F} = 0.2947 (\hbar \omega_{\rm p})^{4/3}$$
,

 $N_{\rm eff}$ is the number of valence electrons per atom determined by

$$N_{\text{eff}} = Ma + N(8-b)$$

for a compound of the $A_M B_N$ type, where *a* is the number of valence electrons per atom of type *A*, and *b* is the corresponding number for type *B*, and *M* and *N* are the respective atomic fractions of the elements *A* and *B*, and ρ is the density.

For GeO₂, $N_{\text{eff}} = 1 \times 4 + 2(8 - 6) = 8$ electrons per atom. In Table II, the results of the calculations are presented. GeO₂ is known to have two crystalline structures apart from its glassy form with significant differences in densities (Table II). The most studied crystalline form of GeO₂ is its hexagonal structure, which is easily soluble in water. The quoted value of the density and the reflective index (or ϵ_{∞}) for g-GeO₂ concerns a glass for which $T_{\phi} = 1650 \,^{\circ}\text{C}$. As can be seen in Table II, the calculated value of E_P is very close to E_R , the average of all the energies corresponding to the peaks in the reflectivity spectra, for c-GeO₂. E_P of c- GeO_2 is also in good accord with E, the average of the energies corresponding to the peaks in the ϵ_2 -E spectra. However, for g-GeO₂, E_P differs considerably from \overline{E} (or E_R). This may be due to the fact that ρ and ϵ_{∞} quoted in Table II for g-GeO₂ do not pertain to the same g-GeO₂ on which the optical measurements were made. In fact, these values in Table II concern nonstoichiometric g-GeO₂. It is worthwhile to note here that such a procedure comparing the calculated E_P with the average of the energies corresponding to the peak in ϵ_2 (or **R**) was proposed by Phillips.¹⁶

Although, in principle, germanium has four electrons per atom in its outermost shell, it has a d band very close to the top of its valence band.¹² The effect of this d band is to increase $N_{\rm eff}$. Van Vechten¹² has considered, in detail, the effect of d electrons on the dielectric properties of semiconductors. In Table II, we also evaluate the Penn gap incorporating the d-electron contribution to $N_{\rm eff}$ (in parentheses in Table II). A comparison between Tables I and II indicates that $\epsilon_{2,\max}$ has a maximum corresponding to \overline{E}_2 for both forms of GeO₂. It is notable that the evaluated E_P incorporating the delectron contribution is in very good agreement only

TABLE II. Properties of crystalline and glassy GeO₂. N_{eff} is the effective number of valence electrons per atom. $\hbar\omega_{\rho}$ is the valence-electron plasmon energy. E_F is the Fermi energy. ϵ_{∞} is the high-frequency dielectric constant. E_P is the Penn gap. \overline{E} is the average of all the energies corresponding to the peaks in ϵ_2 . E_R is the average of all the energies corresponding to the peaks in ϵ_2 . E_R is the average of all the energies corresponding to the peaks in reflectivity. E_G is the energy gap. The numbers in parentheses correspond to the derived optical properties considering the contribution due to the *d* electron.

Structure	Crystalline-tetr.	Crystalline-hex.	Glassy
Solubility (in H ₂ O)	insoluble	soluble	soluble
Density ρ (g/cm ³)	6.239ª	4.228ª	3.677 ^b
Mol wt. M	104.59	104.59	104.59
$N_{ m eff}$	8 (9)	8 (9)	8 (9)
$\hbar\omega_p$ (eV)	19.90 (21.10)	16.38 (17.37)	15.27 (16.20)
$E_F(eV)$	15.89 (17.18)	12.26 (13.26)	11.17 (12.08)
<i>e</i> _∞		2.72	2.59°
E_P (eV)		8.68 (11.80)	10.69 (11.37)
\overline{E} (eV)		9.64	9.06
E_R (eV)		8.62	8.62
E_G (eV)		5.56	5.63

^aReference 18.

^bReference 19.

^cReference 20.

with \overline{E}_2 , for both g and c forms of GeO₂. This rather interesting result seems to suggest the tendency of the valence-band-d-band coupling present in germanium to contribute to the effective number of electrons in GeO₂ only at the observed peaks at high energy in the optical spectra, and, to the maximum of $\epsilon_{2,max}$.

The successful extrapolation of the application of Penn-like models⁸⁻¹⁰ to crystalline and glassy GeO₂ leads to interesting conclusions. Although Fritzsche¹⁷ finds it useful to distinguish between glasses and amorphous materials in terms of their structure and the average covalent coordination, disorder or lack of order in these systems does not seem to have any influence on the variation of the optical properties such as reflectance (or ϵ_2) on photon energy. An isotropic, nearly-free-electron model, such as the Penn model, seems to be valid for almost all materials, irrespective of their structure and the energy gap, in explaining the energies corresponding to

the peaks in reflectivity spectra (or the ϵ_2 -E spectra) and correlating these energies to the band structure of the material.

III. CONCLUSION

Penn-like model calculations have been utilized with reasonable success to explain the optical spectra of the crystalline and glassy GeO₂. The increased effective number of electrons in GeO₂ due to the valence *d*-band coupling in elemental Ge is seen to favor the increased ϵ_2 at high photon energies.

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- ¹P. J. Vergano and D. R. Uhlmann, Phys. Chem. Glasses 11, 30 (1970).
- ²M. Okuno, C. D. Yin, H. Murikawa, F. Marumo, and H. Oyanagi, J. Non-Cryst. Solids 87, 312 (1986).
- ³R. A. Weeks and T. Purcell, J. Chem. Phys. 43, 483 (1965).
- ⁴J. M. Jackson, M. E. Wells, G. Kordas, D. L. Kinser, R. A. Weeks, and R. H. Magruder, J. Appl. Phys. 58, 2308 (1985).
- ⁵G. Kordas, R. A. Weeks, and D. L. Kinser, J. Appl. Phys. **54**, 5394 (1983).
- ⁶V. Garino-Cannina, C.R. Acad. Sci. 247, 593 (1958).
- ⁷L. Pajasova, Czech. J. Phys. B 19, 1265 (1969).
- ⁸D. R. Penn, Phys. Rev. **128**, 2093 (1962).
- ⁹R. A. Breckenridge, R. W. Shaw, Jr., and A. Sher, Phys. Rev. B 10, 2483 (1974).
- ¹⁰R. D. Grimes and E. R. Cowley, Can. J. Phys. **53**, 2549 (1975).

- ¹¹N. M. Ravindar and J. Narayan, J. Appl. Phys. **60**, 1139 (1986).
- ¹²J. A. Van Vechten, Phys. Rev. 182, 891 (1969).
- ¹³N. M. Ravindra, R. P. Bhardwaj, K. Sunil Kumar, and V. K. Srivastava, Infrared Phys. 21, 369 (1981).
- ¹⁴A. Kumar and N. M. Ravindra, Phys. Rev. B 25, 2889 (1982).
- ¹⁵J. D. Jackson, *Classical Electrodynamics* (Wiley, New York, 1978).
- ¹⁶J. C. Phillips, Bonds and Bands in Semiconductors (Academic, New York, 1973).
- ¹⁷H. Fritzsche, Phys. Today 37 (10) (1984).
- ¹⁸CRC Handbook of Chemistry and Physics, 49th ed., edited by R. C. Weast (CRC Press, Ohio, 1974-1975).
- ¹⁹P. J. Vergano, D.Sc. thesis, Massachusetts Institute of Technology, 1970.
- ²⁰R. H. Magruder, S. Morgan, D. L. Kinser, and R. A. Weeks, J. Non-Cryst. Solids (to be published).