# Electronic structure of metal hydrides. IV. TiH<sub>x</sub>, ZrH<sub>x</sub>, HfH<sub>x</sub>, and the fcc-fct lattice distortion

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The electronic structures of  $\text{TiH}_x$ ,  $\text{ZrH}_x$ , and  $\text{HfH}_x$  have been studied using photoelectron spectroscopy and synchrotron radiation. Structures in the metal *d*-derived band within  $\sim 3 \text{ eV}$  of the Fermi level  $E_F$  and in the bonding band ( $\sim 3-10 \text{ eV}$  below  $E_F$ ) are compared with theory. In each dihydride, the bonding band center falls at -5.5 eV, at approximately the same energy as previously observed for the dihydrides of Sc and Y. Changes in the emission features near  $E_F$  and at -7 eV have been observed in samples bridging the fcc $\rightarrow$ fct distortion in  $\text{ZrH}_x$ ,  $1.63 \leq x \leq 1.94$ . The changes at  $E_F$  demonstrate the Jahn-Teller effect for the electronic states of  $\text{ZrH}_x$ . The binding energies of the Ti 3p, Zr 4p, Hf 5p, and Hf 4f cores are observed to be greater than in the elemental metals, consistent with charge transfer to the hydrogen site.

### INTRODUCTION

Many, but not all, transition metals react with hydrogen to form stable metal dihydrides.<sup>1</sup> These interstitial alloys exhibit properties which are quite different from those of the elemental metals, including crystal structures (generally hcp+fcc) and mechanical properties (ductile - brittle). The electronic properties of dihydrides are strikingly different from those of the parent metal, and in a series of recent papers<sup>2-8</sup> we have examined many of the hydrogen-induced band-structure changes. We have shown, for example, that dihydrides exhibit wide metal-hydrogen bonding bands centered about 5.5 eV below the Fermi level  $E_F$ , which have no counterpart in the metal band structure, and that the states near  $E_F$  are primarily metal d-derived.

The group IIIB metals (Sc, Y, and most lanthanides) and the IVB metals (Ti, Zr, Hf) form stable metallic dihydrides with a CaF<sub>2</sub> crystal structure over a wide range of composition x, where x = H/M. In general, the hydrogen atoms occupy the tetrahedral lattice sites. It has recently been observed,<sup>2,9-12</sup> however, that the group IIIB dihydrides can have partial octahedral site occupation for x - 2. In contrast to this, the group IVB dihydrides exhibit a crystal-structure distortion<sup>1,13,14</sup> from fcc to fct for x - 2. These deviations from the ideal CaF<sub>2</sub> structure with completely occupied tetrahedral sites suggest that the ideal dihydride is energetically unstable.

We have conducted a series of photoemission experiments to examine the electronic structure of the dihydrides  $\text{TiH}_x$ ,  $\text{ZrH}_x$ , and  $\text{HfH}_x$ . Two aspects were particularly interesting. First, we wished to compare the bonding-band energies of the group IV*B* dihydrides with those of the group III*B* dihydrides to see if the hydrogen-induced bands remained ~5.5 eV below  $E_F$  as  $E_F$  moved higher in the *d* bands (they did). Second, we wished to observe changes which might accompany the fcc-fct distortion of  $\text{ZrH}_x$  for  $1.63 \le x \le 1.94$ . We have observed significant changes in the states near  $E_F$  and in the deeper-lying states near -7 eV.

This study of the group IVB dihydrides reflects our ongoing interest in the fundamental properties of metal-hydrogen systems. In the first three papers of this series we have discussed the electronic structure of the group IIIB dihydrides ScH<sub>r</sub>,  $YH_{x}$ , and  $LuH_{x}$  as studied by optical spectroscopy<sup>2</sup> (paper I), self-consistent band calculations<sup>3</sup> (paper II), and photoelectron spectroscopy<sup>4</sup> (paper III). In a fifth paper, optical and photoemission techniques were used to examine the x-dependent properties of LaH<sub>x</sub>,  $1.9 \le x \le 2.89$ , including the metal-tosemiconductor transition.<sup>8</sup> Our work has been complemented by recent band-structure calculations of Misemer and Harmon,<sup>15</sup> Switendick,<sup>16-18</sup> Gupta and Burger,<sup>19-21</sup> and Kulikov and co-workers.<sup>22</sup>

## EXPERIMENTAL CONSIDERATIONS

Photoelectron spectroscopy is a versatile tool with which to probe the band-structure properties of a system. Measurements with polycrystalline samples such as hydrides (single crystals of dihy-

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drides are not generally available) provide valuable information about features of the density of (occupied) states and, through studies of the  $h\nu$ dependence of observed emission features, the angular momentum character of these states can be inferred.

The present measurements were performed using synchrotron radiation emitted by 240-MeV electrons stored in the Tantalus I electron storage ring. The radiation was dispersed with an ultrahigh vacuum 3m toroidal grating monochromator or a UHV 1m Seya-Namioka monochromator. The photon energy range was  $7 \le h\nu \le 130$  eV; in this paper the low photon energy spectra are emphasized because the spectra change relatively little with  $h\nu$  for  $h\nu \gtrsim 40$  eV. All measurements were performed at 300 K in a UHV experimental chamber as discussed elsewhere<sup>4,23</sup> with operating pressure during measurements of  $6-8 \times 10^{-11}$  Torr. Photoelectron energy analysis was done with a doublepass cylindrical mirror analyzer. The total resolution (electrons plus photons) ranged from 0.4 to 0.6 eV.

The samples used in this study were prepared in a separate hydrogen-charging system. In preparation for charging, posts of polycrystalline metal  $\sim 2 \times 2 \times 10$  mm<sup>3</sup> were cut from crystal bar stock.<sup>24</sup> These posts were chemically polished (HF-HNO<sub>3</sub> solution), wrapped in Ta foil, then degassed in situ at 1000°C before high-purity hydrogen was admitted to the charging vessel (pressure  $\sim 1$  atm). The samples were then cooled and held at the temperature required to synthesize the desired hydride composition (575 °C for TiH<sub>1.5</sub>, 600 °C for  $HfH_{1,56}$ , 600°C for  $ZrH_{1,87}$ , 700°C for  $ZrH_{1,77}$ , and 800°C for ZrH<sub>1.63</sub>). ZrH<sub>1.94</sub> was prepared by first charging at low pressure and subsequently exposing it to 150 atm of hydrogen at 600°C. The hydrogen composition of sections taken from the samples studied was determined by hot vacuum extraction at 950°C with an estimated accuracy of better than 1%.

Optical metallography showed the samples to be single phase. For  $ZrH_x$ , large columnar grains grew inward from the outer edges. X-ray analyses gave the following for  $ZrH_x$ : for x = 1.63,  $a_0 = 4.781$ Å (fcc); for x = 1.77,  $a_0 = 4.943$  Å and  $c_0 = 4.493$  Å (fct); for x = 1.86,  $a_0 = 4.965$  Å and  $c_0 = 4.464$  Å (fct); for x = 1.94,  $a_0 = 4.976$  Å and  $c_0 = 4.454$  Å (fct). The volume of the ZrH<sub>x</sub> unit cell increased from 109.3 Å<sup>3</sup> (x = 1.63) to 110.3 Å<sup>3</sup> (x = 1.94), an increase of 0.91%. In the fct phase,  $a_0$  increased by 0.67% and  $c_0$  decreased by 0.88% with c/a = 0.909 for x = 1.77and c/a = 0.895 for x = 1.94. The large-grained ZrH<sub>x</sub> samples were relatively rugged, the smallgrained TiH<sub>x</sub> samples were more brittle, and highcomposition samples of HfH<sub>x</sub> were extremely fragile.

The hydride samples were attached to copper holders and mounted on a sample bank (capacity  $\approx 20$ ) for photoemission measurements. Following a routine bake ( $\leq 24$  h,  $\leq 150$  °C), the operating vacuum of the measurement chamber was  $\sim 6 \times 10^{-11}$ Torr. Immediately before study, the desired sample was retrieved from the bank with an  $x-y-z-\theta-\phi$ -tilt manipulator, fractured, and translated to the mutual focus of the monochromatic radiation beam and the electron energy analyzer. Data collection began within 2-3 min of sample fracture.

#### **RESULTS AND DISCUSSION**

#### fcc dihydrides

The photoelectron energy distribution curves (EDC's) for  $TiH_{1,50}$ ,  $ZrH_{1,63}$ , and  $HfH_{1,56}$  shown in Fig. 1 for  $40 \le h\nu \le 80$  eV offer an overview of the electronic structure of these metal dihydrides. Two spectra for  $TiH_{1,50}$  at  $h\nu = 45$  and 80 eV show the metal *d*-derived emission band within  $\sim 3 \text{ eV}$  of  $E_{F}$  and the covalent bonding band centered at about -5.5 eV extending from -3 to -10 eV. (All energies are referenced to the Fermi level  $E_F$ .) Energy distribution curves for ZrH<sub>1,63</sub> and HfH<sub>1,56</sub> reveal analogous band-structure features within ~10 eV of  $E_{F}$ . These band-structure features reflect the chemical bonding of the metal-hydrogen system. Also shown in Fig. 1 are the Ti 3p core levels (binding energy 33.3 eV), the Zr 4p's (28.1 and 29.5 eV), the Hf 5p's (30.3 and 38.3 eV), and the Hf 4f's (14.7 and 16.4 eV). Comparison of these binding energies with available literature values<sup>25,26</sup> for elemental Ti, Zr, and Hf shows the latter to be smaller by  $\sim 1 \text{ eV}$ , consistent with charge redistribution away from the metal upon hydriding.<sup>25</sup> For TiH<sub>1.5</sub>, the Auger structures centered at -16 eV (full width  $\sim 12 \text{ eV}$ ) result from the Auger decay of the Ti 3p core hole. For TiH, with  $33.3 < h\nu \leq 43$  eV, Auger and band-structure emission features overlap and the latter are obscured. Analogous Auger emission is shown for  $\operatorname{ZrH}_{1,63}$  at -17 eV for  $h\nu = 40$  eV. For  $\operatorname{HfH}_{1,5}$  at  $h\nu$ = 50 eV, the Auger emission falls just below the  $4f_{5/2}$  core level.

The experimental spectra of Fig. 1 indicate that the states contributing to the bonding bands of these dihydrides (as with others studied) are strongly hybridized, containing d as well as spcharacter. Were it not for this d character, the bonding band-emission features would diminish in strength with  $h\nu$  as the photoionization cross section decreases. Furthermore, the double-structured Auger emission indicates two valence bands, each having d character, contributing to the decay of the np core hole.<sup>4</sup>



FIG. 1. Representative photoelectron energy distribution curves for TiH<sub>1,50</sub>, ZrH<sub>1,63</sub>, and HfH<sub>1,56</sub> showing the metal *d*-derived bands within  $\sim 3 \text{ eV}$  of  $E_F$ , the hydrogen-induced bonding bands  $\sim 3-10 \text{ eV}$  below  $E_F$ , and the core level emission for the Ti 3p, Zr 4p, Hf 5p, and Hf 4f cores. The features labeled Auger arise from the Auger decay of the *p*-core hole; the double structural Auger emission reveals the *d* character of the bonding bands.

There have been three non-self-consistent calculations<sup>16,19,22</sup> of the band structure of TiH<sub>2</sub>. In Fig. 2 the calculated bands of Gupta<sup>19</sup> for TiH<sub>2</sub> are reproduced. As shown, the Fermi level lies just above the  $L_3$  and  $\Gamma_{25}$ , symmetry points. The bands centered at -6 eV constitute what we have termed collectively the "bonding band" (derived from hy-



FIG. 2. Energy bands of TiH<sub>2</sub> calculated by Gupta (Ref. 19) showing the Fermi-level cutting band three between  $\Gamma$  and L. The tetragonal distortion lifts the degeneracy of  $\Gamma_{25} \cdot -\Lambda_3 - L_3$  with one portion falling below  $E_F$  and the other above  $E_F$  (Jahn-Teller splitting). The bands within ~2 eV of  $E_F$  are largely *d*-like; the two bands centered at -6 eV give rise to the experimental features which we have termed the bonding or hydrogen-induced band.

drogen bonding and antibonding states). The bands near  $E_F$  are largely *d* derived, as can be seen, for example, by examining the calculations by Switendick<sup>17</sup> for the YH<sub>0</sub>-YH<sub>1</sub>-YH<sub>2</sub>-YH<sub>3</sub> series.

The total density of states for  $\text{TiH}_2$  is shown in the lower portion of Fig. 3 (dashed curve labeled  $\text{TiH}_2$  DOS). From these calculations, a sharp peak is predicted at -6 eV which is related to the flat bands near and along L-Q-W-K-X in the Brillouin zone.<sup>3,4</sup> The shoulder in the DOS at -7.3 eV reflects the  $L_1$  critical point.<sup>3,4</sup> Near  $E_F$  the calculations indicate a very high DOS peaking within a fraction of an electron volt of  $E_F$  originating from the flat band along  $\Gamma-\Lambda-L$ . The flat portion of the band extending from  $W_{2'}$  to  $K_1$  gives rise to the shoulder in the DOS at about -2 eV.

The energy bands shown in Fig. 2 and the DOS of Fig. 3 are typical of the various calculations for dihydrides. These calculations predict a double-featured valence band made up of *d*-derived states near  $E_F$  and hybridized *spd* states ~6 eV below  $E_F$ .

Comparison of theory with the experimental results can be made through Fig. 3, where energy distribution curves for TiH<sub>1.5</sub>, ZrH<sub>1.63</sub>, ZrH<sub>1.94</sub>, and HfH<sub>1.56</sub> taken at  $h\nu = 21$  eV are reproduced. A smoothly varying background arising from scattered secondary electrons has been subtracted. The similarities for Ti, Zr, and Hf dihydrides are evident: All have structured emission within 3–3.3 eV of  $E_F$  and a deeper, structured, bonding band extending from -3 to -10 eV peaking at -5.1 to -5.5 eV with a shoulder near -7.3 eV. For HfH<sub>1.56</sub> a



FIG. 3. Comparison of the calculated density of states for TiH<sub>2</sub> with our experimental photoelectron energy distribution curve for  $h\nu = 21$  eV where the secondary electron background has been subtracted. Experiment shows the calculated bonding band to be too far from  $E_F$  with too small a full width at half maximum. The spectra for ZrH<sub>x</sub>, x=1.63 and x=1.94, reveal the change in the band structure associated with the fcc --fct distortion, notably near  $E_F$  and near -7 eV. The experimental results for HfH<sub>1.56</sub> are compared to a rigidband-shifted density of states of ErH<sub>2</sub> showing the theoretical basis for the 4-eV structure (greater dispersion in the bonding band compared to TiH<sub>2</sub>). The experimental resolution was ~0.4 eV.

second shoulder appears at approximately -4.4 eV. For each, the dominant central peak is associated with the predicted DOS structure and the deepest shoulder reflects the  $L_1$  critical point.

Gupta's non-self-consistent calculations<sup>19</sup> for TiH<sub>2</sub> place the bonding-band center too far from  $E_F$  and its width is underestimated. Our own nonself-consistent calculations for YH<sub>2</sub> (Refs. 3 and 5) showed better agreement with the experimental bonding-band features. Our self-consistent prescription (including an improved method for handling exchange) led to bonding bands which were shifted ~1 eV toward  $E_F$ , so that iterating to selfconsistency sacrificed agreement in those bands while improving agreement within the *d* bands near  $E_F$ .

While the  $\text{TiH}_2$  calculations appear to underestimate the width of the *d* bands near  $E_F$ , they offer an explanation for the two features observed near -1 and -1.5 eV for  $\text{TiH}_2$ . We associate these features with the flat portion of the band along W-K and with the states near X.

For ZrH, and HfH, there have been no published band calculations. However, we can infer from Gupta's results (including those shown in Fig. 3), our own, those of Switendick, and those of Kulikov et al. that the dihydride bands and DOS will resemble those shown for TiH<sub>2</sub>. To estimate differences, we reproduce in Fig. 3 the DOS calculated by Gupta<sup>20</sup> for  $ErH_2$ , where we have rigidly shifted  $E_F$ to account for the added electron in HfH, relative to ErH<sub>2</sub> (knowing this to be only a rough approximation). We schematically indicate that for ErH,  $(or HfH_2)$  the greater dispersion in the bonding bands (compared to TiH<sub>2</sub>) results in more structure in the DOS. Experimentally we see that the photoemission density-of-states features for all three of these dihydrides are very similar. For  $HfH_{1,56}$ , an additional structure near - 4.4 eV is observed, consistent with the predictions for ErH<sub>2</sub>. In Fig. 4, spectra at  $h\nu = 15$  eV are shown



FIG. 4. Photoelectron emission spectra for fcc TiH<sub>1.50</sub>, ZrH<sub>1.63</sub>, and HfH<sub>1.56</sub> for  $h\nu$ =15 eV emphasizing the *d*-band region near  $E_F$ . For TiH<sub>x</sub>  $\rightarrow$  ZrH<sub>x</sub>  $\rightarrow$  HfH<sub>x</sub> the width of the *d* band increases and the band along  $W_2$ -K<sub>1</sub> falls away from  $E_F$ .

for each of the cubic dihydrides for the *d*-band region. TiH<sub>1.5</sub> exhibits shoulders at -0.9 eV and -1.5 eV and a peak near  $E_F$ ; ZrH<sub>1.63</sub> has a shoulder near -1.8 eV, a peak near -1.0 eV, and diminished Fermi-level emission. These features near -1 and -2 eV must be associated with the states along *W*-*K* and near *X*.

Comparison of experimental and calculational trends relating to the *d* bands indicates good qualitative agreement. The calculated width of the occupied *d* bands (i.e.,  $X_1 - E_F$ ) is smallest for ScH<sub>2</sub> (~1.9 eV) and increases to 2.4 eV for TiH<sub>2</sub> and 2.3 eV for YH<sub>2</sub>. Experimentally, we have observed that the width of the occupied *d* bands increases in going across the Periodic Table (group IIIB to group IVB) or down the Periodic Table (3*d* to 4*d* transition-metal dihydrides).

#### fcc-fct distortion

Switendick,<sup>16</sup> Gupta,<sup>19</sup> and Kulikov *et al.*<sup>22</sup> have performed calculations which show that the doubly degenerate  $L_1$  and the triply degenerate  $\Gamma_{25}$ , states of TiH<sub>2</sub> lie just below the Fermi level and that the band along  $\Gamma_{25'}$ - $\Lambda$ - $L_3$  crosses  $E_F$  twice. Earlier workers had observed an fcc-fct distortion of TiH<sub>x</sub> and related it to a Jahn-Teller effect,<sup>13,14</sup> and the band-structure calculations supported that interpretation. According to the model proposed by Switendick, the Fermi level of fct TiH<sub>x</sub> falls as hydrogen is removed until the occupied  $L_3$  state is reached. When both of the  $L_3$  states are empty, no total energy lowering can take place by splitting (distorting) and fct TiH<sub>x</sub> is no longer energetically favored over fcc TiH<sub>x</sub>.

This fcc-fct distortion of the group IVB dihydrides stands in contrast to the occupation of octahedral sites in the IIIB dihydrides prior to complete filling of all tetrahedral sites.<sup>1,2,8-12,18</sup> Apparently, in both cases the purely CaF<sub>2</sub> fcc energy states are energetically unfavorable for dihydrides with  $x \simeq 2$ . For group IIIB dihydrides, the total energy is lowered by occupation of the octahedral sites (the tendency to do so varies with the lattice constant) and in group IVB dihydrides the lattice distorts.

To reveal the x dependences of the electronic properties associated with the fcc-fct distortion, we have examined  $\text{ZrH}_x$  for  $1.63 \le x \le 1.94$ . The photoemission results are shown in Figs. 5 and 6 and in Fig. 3.

In Fig. 5 we show a series of EDC's for  $ZrH_{1.63}$ (fcc) and  $ZrH_{1.94}$  (fct) measured at room temperature. The vertical scales have been normalized for convenience to keep constant the height of the *d*-band emission maximum. The emission features for  $ZrH_{1.94}$  are shown on the left; on the right we

reproduce only the d bands for  $ZrH_{1,63}$ . Clearly, emission of states close to  $E_F$  is considerably more pronounced in  $ZrH_{1,94}$  than in  $ZrH_{1,63}$ . The series of spectra for  $h\nu = 14$  eV shown in Fig. 6 reveal the x dependence of the bands within 1.5 eV of  $E_F$ . For x = 1.63 (fcc), features are observed at -1.0 and -1.8 eV and the DOS appears to be diminishing as  $E_F$  is approached from below  $E_F$ . For x = 1.75 (fct) the Fermi-level emission is slightly enhanced (our instrumental resolution is  $\simeq 0.4$  eV) and the lower two features appear unchanged. For x = 1.88, two distinct features appear at 1.25 and 1.9 eV and the Fermi-level emission is much more prominent. For x = 1.94, the Fermi-level emission is much greater than that of the features near 1.25 and 1.9 eV and the latter appear to have shifted or broadened almost beyond recognition. We associate these changes with the predicted splitting of the degenerate  $L_3 - \Lambda_3 - \Gamma'_{25}$ states, a splitting which increases as the distortion increases (i.e., with increasing hydrogen content). As shown, a change in hydrogen concentration of ~10% results in a major band-structure modification near  $E_F$ . (Note that in Switendick's calculation for fct TiH<sub>r</sub> with c/a = 0.972 the calculated  $L_3$  splitting was ~0.2 eV; for ZrH, with c/a $\simeq 0.9$  we see experimentally that the distortion is much greater.)

In the emission spectra for  $\operatorname{ZrH}_{1.94}(14 \le h\nu \le 26 \text{ eV})$ shown in Fig. 5, two features appear in the bonding band, namely, the peak at -5.3 eV, which is nearly invariant with  $h\nu$ , and the deeper shoulder near -7 eV which disperses with  $h\nu$ . The  $h\nu$  dependence of the -7 eV peak energy is shown in the inset of Fig. 5; it moves downward  $\sim 1 \text{ eV}$  for  $h\nu$ increasing from 15 to 26 eV. At higher photon energy it is obscured by overlapping Auger emission and ultimately it is lost because of its low angular momentum character. This feature reflects the  $L_1$  band, as discussed in detail in III and supported by the calculations and l projections of II.

In the inset of Fig. 5, results for samples of three different compositions are shown, namely, x = 1.63, 1.75, and 1.94. With increasing x, the shoulder moves to higher binding energy compared to the same feature observed at the same photon energy for a lower-x dihydride. This indicates a lowering in energy of the  $L_1$  critical point caused by the tetragonal distortion. That shift amounts to  $\sim 0.65 \text{ eV}$  for  $\text{ZrH}_{1,63}$  (cubic)  $\rightarrow \text{ZrH}_{1,94}$  (fct with c/a = 0.909).

In Fig. 3, spectra for  $ZrH_{1.63}$  and  $ZrH_{1.94}$  are shown for  $h\nu = 21$  eV free of the secondary background. There, the change in the -7 eV feature is clearly evident: It is a shoulder at -7.2 eV for the cubic phase and it is a well defined maximum





FIG. 5. Photoelectron energy distribution curves for fcc  $ZrH_{1.63}$  and fct  $ZrH_{1.94}$ . For x=1.63, features are observed at -1 and -1.3 eV and are larger at all hv than the emission just below the Fermi-level cutoff. For x=1.94, the Fermi-level emission is dominant and the deeper features in the *d* bands are less pronounced. Within the bonding band, the shoulder at  $\sim 7$  eV shows the same hv dependence (inset) for different values of *x* but falls farther from  $E_F$  with increasing *x*. No change in the feature at  $\sim 5.3$  eV is observed as a function of *x*. No such change has been observed as a function of *x* for YH<sub>x</sub>, which does not exhibit the fet distortion.

at -7.8 eV for the tetragonal phase. At the same time the central peak at -5.3 eV is nearly independent of x.

## CONCLUSIONS

The group IVB dihydrides display *d*-derived emission features within ~3 eV of  $E_F$  which can be associated with the band along  $W_{2'}-K_1$ . The bonding bands extend from -3 to -10 eV and are centered at approximately -5.5 eV. Experimentally, the width of the *d* bands has been shown to follow the calculational trends by increasing in going from group IIIB to group IVB dihydrides. The center of the bonding band is observed at approximately the same energy for all dihydrides, leading us to conclude that stable dihydrides favor a bonding band centered near -5.5 eV. Photoemission results bridging the fcc-fct distortion in ZrH, are consistent with a Jahn-Teller splitting

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FIG. 6. Photoemission spectra showing the x dependence of the *d*-band spectrum. With increasing x, the fct distortion increases and the Fermi-level emission becomes successively more dominant as the bands along  $\Gamma$ - $\Lambda$ -L split apart.

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of the bands near  $E_F$ , showing a change in Fermilevel emission with increasing x. In contrast to this, x dependences of photoemission spectra for cubic YH<sub>x</sub>, 1.7 < x < 1.98 are very slight and reflect octahedral site occupancy.<sup>2,4</sup> Also associated with the tetragonal distortion is a shift to greater binding energy of the critical point at  $L_1$ .

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