# Study of some magnetic properties of a mixed phase $(Mn_3O_4)$ in MnO crystals

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Magnetic susceptibility and magnetization measurements have been carried out on several samples of MnO in the temperature range of 4.2 to 300 K and in magnetic fields up to 10 kOe using the Faraday technique. The measurements were done on four commercial single crystals, a powder MnO specimen (pure MnO), and an annealed MnO single crystal. All four commercial crystals were found to contain significant amounts (2 to 8%) of Mn<sub>3</sub>O<sub>4</sub>, a ferrimagnet with  $T_c \approx 43$  K. Hence, the Mn<sub>3</sub>O<sub>4</sub> impurity in these samples completely dominates the magnetic properties at lower temperatures. The impurity concentrations are estimated from observed magnitudes of the spontaneous moment, and confirmatory evidence is obtained from x-ray analysis of the samples. A displaced hysteresis loop has been observed at 4.2 K, suggesting the presence of exchange anisotropy at the ferrimagnetic-antiferromagnetic (Mn<sub>3</sub>O<sub>4</sub>-MnO) interfaces. Annealing of a commercial crystal in Mn atmosphere (using an evacuated sealed quartz tube) at 1293 K for about 200 h is shown to convert the Mn<sub>3</sub>O<sub>4</sub> impurity to MnO, thus yielding a pure single crystal of MnO.

## I. INTRODUCTION

The first-row transition-metal mono-oxides remain of fundamental interest since their electronic properties have defied simple explanations.<sup>1</sup> Various properties of manganese mono-oxide, MnO, have been under investigation for over fifty years now. From magnetic-susceptibility, specific-heat, and strain data it became known quite early<sup>2-4</sup> that MnO undergoes a transition to an antiferromagnetic state below the Néel temperature  $T_N \simeq 120$  K. From neutron diffraction measurements, the magnetic structure of MnO has been found to consist of ferromagnetic (111)-type sheets, with the neighboring (111) planes stacked antiferromagnetically along a [111] direction.<sup>5</sup> The antiferromagnetic ordering is accompanied by a rhombohedral distortion of the high-temperature fcc structure.<sup>6</sup> Recently it has been proposed that the transition to the antiferromagnetic states in MnO and NiO should be of first order.<sup>7,8</sup> Although a number of recent investigations<sup>9-11</sup> have shown the transition in MnO to be of first order, the same has not been found to be true in NiO, where the transition clearly appears to be of second order.<sup>12</sup>

Magnetic-susceptibility  $(\chi)$  measurements on powdered MnO specimens (usually obtained by the decomposition of MnCO<sub>3</sub>) have been performed by several groups, near or below  $T_N^{3,11,13}$  and above room temperature.<sup>14</sup> The only reported  $\chi$  measurements on a single crystal have been those of McGuire *et al.*<sup>15</sup> In these studies,  $\chi$  was found to depend on applied magnetic field *H* below  $T_N$  and in the case of single-crystal studies,<sup>15</sup> even above  $T_N$ . Perhaps the most careful analysis of the data on a powder specimen has been that of Bloch *et al.*<sup>13</sup> in the temperature range of 4.2 to 200 K and in H up to 50 kOe. In their study X was found to be independent of H above  $T_N$  and to increase with increasing H below  $T_N$ . From the latter, these authors estimated  $\chi_1(4.2 \text{ K}) = 79 \times 10^{-6} \text{ cm}^3/\text{g}$ , whereas  $\chi_{\text{max}}(T \sim T_N) = 83 \times 10^{-6} \text{ cm}^3/\text{g}$ .

In the single-crystal studies by McGuire *et al.*, <sup>15</sup>  $\chi$ at 4.2 K was found to be irreproducible, and it was suggested that this might be due to the presence of a magnetic impurity, perhaps Mn<sub>3</sub>O<sub>4</sub>. However, no further work to prove the presence of the  $Mn_3O_4$ phase in single crystals of MnO (usually grown by the flame fusion method) has since been carried out, although these crystals in recent years have been used in a number of other investigations.<sup>9,10,16,17</sup> Furthermore, no direct measurements of the principal susceptibilities of MnO have yet been carried out. The present work was undertaken to fill this gap. Our initial measurements on a single crystal of MnO at 4.2 K clearly indicated the presence of a ferromagnetic-type impurity, not too different from the observations of McGuire et al.<sup>15</sup> Following this, we obtained single crystals of MnO from four different sources and found similar observations in all of them. Since these crystals have been used around the world for many other investigations for the properties of pure MnO, it was thought to be extremely important to establish the nature and magnitude of this magnetic impurity in MnO single crystals and determine its effect on the magnetic properties of pure MnO. In this work, we provide evidence that the impurity is indeed Mn<sub>3</sub>O<sub>4</sub> and its concentrations are on the order of few atomic percent. However, since Mn<sub>3</sub>O<sub>4</sub> is ferrimagnetic with  $T_c \simeq 43$  K, <sup>18, 19</sup> even such a small concentration completely dom-

<u>21</u>

2897

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inates the magnetic properties at the lower temperatures and even has noticeable effects near and above  $T_N$ . The exchange anisotropy, usually present due to ferromagnetic-antiferromagnetic interfaces,<sup>20</sup> has also been observed. The magnetic susceptibility of a relatively pure MnO powder has also been studied for comparison purposes. Finally, a successful experiment in which, an impure MnO single crystal with Mn<sub>3</sub>O<sub>4</sub> impurity, has been converted to a pure MnO crystal, is described. The results of these studies are presented and discussed in this paper.

#### **II. EXPERIMENTAL DETAILS**

The measurements of magnetizations and magnetic susceptibilities presented in this paper were carried out by the Faraday technique, using a Cahn RG electrobalance, Lewis field-gradient coils, and a Janis cryostat. The details of the experimental apparatus and procedures are described elsewhere.<sup>21</sup> The samples used in these studies are listed in Table I, along with their sources. The samples, initially in the form of single crystals (except sample E, which was obtained in the powder form) were powdered with a mortar and pestle to about 100- $\mu$ m size, were mounted in a platinum boat and covered with Duco cement before being suspended in the cryostat for the measurenents. The corrections for the platinum boat and the Duco cement are less than 1% and have been disussed in detail elsewhere.<sup>21</sup> We chose to carry out ur initial measurements on powder samples in order ) avoid complications due to anisotropies and econdly, the most important point, to facilitate comparison with the results of a relatively pure MnO powder (sample E). Most of the temperaturedependent measurements were carried out at 1 kOe, although fields up to 10 kOe were available. The data on each sample was usually taken in the temperature range of 4.2 to 300 K.

#### **III. RESULTS AND ANALYSES**

The variation of X with temperature T, for samples A through E of Table I, is shown in Fig. 1. There are noticeable differences in the absolute magnitudes of x for some of the samples above  $T_N$  (~120 K). Below about 70 K,  $\chi$  for samples A through D increases whereas for sample E, X continues to decrease. Our data on sample E are quite similar to the observation of Bloch et al., <sup>13</sup> who had carried out measurements on the same sample. The small increase in X observed below about 30 K may be due to the presence of a minute quantity of a paramagnetic impurity. Such a rise is not evident from the observation of Ref. 13 at low fields; however, they had only two data points below 50 K. A somewhat similar rise, observed in CoO samples by Seehra and Silinsky,<sup>22</sup> has been attributed to nonstoichiometry (cation deficiencies). It is quite unlikely that this rise is due to the presence of Mn<sub>3</sub>O<sub>4</sub> since that leads to the presence of remanent magnetization. This was not observed in sample E, although an increase of  $\chi$  with increasing H is observed, similar to the observation of Bloch et al.<sup>13</sup> This has been discussed in detail in Ref. 13 and this field dependence was attributed to domain-wall rotation. Above  $T_N$ ,  $\chi$  was observed to

Sample	Source	<i>T<sub>N</sub></i> (K)	X at 130 K (10 <sup>-6</sup> cm <sup>3</sup> /g)	M <sub>s</sub> at 4.2 K (cm <sup>3</sup> Oe/g)	at. % of Mn <sub>3</sub> O <sub>4</sub> estimated	Identification of Mn <sub>3</sub> O <sub>4</sub> by x-ray diffraction
A	Atomergic	T-13 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1 - 1				
	Chemical Co.	119.6	83.90	1.485	7.9	yes
В	Alfa-Ventron					
	Corporation U. S. A.	117.0	80.00	0.950	5.1	yes
С	Nakazumi					
	Crystals Japan	117.6	80.75	0.625	3.3	yes
D	Cristal-Tec Grenoble, France	118.6	82.00	0.380	2.0	yes
Ε	General Electric U. S. A.	119.0	79.85	nil	nil	no
F	Annealed Sample D	118.0	82.36	nil	nil	no

TABLE I. Summary of some results for various samples of MnO.



FIG. 1. The temperature variation of magnetic susceptibility for MnO powder samples obtained from five different sources (Table I). The increasing susceptibility below 80 K is shown by the lines drawn through the data points for clarity.

be independent of H (up to 10 kOe), within an accuracy of better than 0.4%, for all the samples.

Next we consider the determination of the  $T_N$ 's for various samples from the data shown in Fig. 1. It is known that in antiferromagnets, the singularity in  $d(\chi T)/dT$  is similar to that in the specific heat, and in MnF<sub>2</sub> this yielded  $T_N$  quite accurately.<sup>23</sup> Therefore, we used a similar procedure in MnO.  $d(\chi T)/dT$  was obtained by computer fitting the data in the temperature range of 105 and 125 K to a second-order polynomial, choosing a four-point interval at a time, and evaluating  $d(\chi T)/dT$  at median temperature. As an example, the results for sample E are shown in Fig. 2, and  $T_N$ 's so determined are listed in Table I. The Néel temperatures so determined<sup>24</sup> are about 0.5 K lower than the temperatures at which a near discontinuity in  $\chi$  occurs. We will return to the discussion of  $T_N$ 's and their variations with samples later.

For samples A through D, obtained by powdering the four commercial single crystals, remanent magnetization was observed below about 50 K, which increased in magnitude as the temperature was lowered. To get at the spontaneous magnetization  $M_s$  at different temperatures for the various samples, the M vs H plots were extrapolated to H = 0. The results so obtained are plotted in Fig. 3 for samples A to D and the  $M_s$  values at 4.2 K are listed in Table I. Since it is known that Mn<sub>3</sub>O<sub>4</sub> is a ferrimagnet with  $T_c = 43 \text{ K}$ ,<sup>18, 19</sup> the results of Fig. 3 are consistent with Mn<sub>3</sub>O<sub>4</sub> being an impurity in the MnO crystals A through D. Using a saturation moment<sup>19</sup> of 18.8  $cm^{3}Oe/g$  for  $Mn_{3}O_{4}$ , the estimated concentrations of Mn<sub>3</sub>O<sub>4</sub> in various samples of MnO are listed in Table I. If we use the value of 35.0 cm<sup>3</sup>Oe/g for the saturation moment as determined by Jacobs<sup>18</sup> from the high-field studies, the estimated concentrations would be reduced by a factor of nearly 2. In any case, the important point is that Mn<sub>3</sub>O<sub>4</sub> has been found to be present at the level of several percent in all commercially obtained single crystals.

In most cases, the samples were cooled from 77 to 4.2 K in a field of 1 kOe. During the first cycle (field increased to +8 kOe and then brought back to 1 kOe), a hysteresis of about 10% for samples A through D was noticed. If the samples are cooled in a field of 8 kOe, no similar hysteresis was observed. Subsequently, we cooled one sample (D) in nearly zero magnetic field (actually the residual field in the electromagnet was about 50 Oe) and studied its mag-



FIG. 2. The magnetic-susceptibility variation with temperature for a pure MnO powder sample (same as sample E in Fig. 1, but in the complete temperature range). The inset shows the method adopted for obtaining  $T_N$ .

netization behavior and results are shown in Fig. 4. The magnetization process has several distinct features. Significant hysteresis is observed during the first cycle. However, in subsequent cycles, the hysteresis loop is quite narrow and displaced (Fig. 5). This observation may be due to exchange anisotropy<sup>20</sup> induced at the ferrimagnetic-antiferromagnetic ( $Mn_3O_4$ -MnO) interfaces. We will return to these results later.



+24

MnO

SAMPLE D

FIG. 3. The temperature variation of the zero-field magnetization for four MnO samples (Table I).

FIG. 4. Magnetic hysteresis obtained on cooling the MnO sample D to 4.2 K in nearly zero magnetic field. Triangles ( $\Delta$ ) refer to the data obtained after reaching -8 kOe. The starting point is denoted by *a*.

8



FIG. 5. Expanded view of the magnetic hysteresis (that shown in Fig. 4) at very low fields. Note the shifted hysteresis loop, away from the origin.

The magnetic susceptibility below 60 K has been evaluated by the following technique. An examination of M vs H curve of Fig. 4 suggests that the magnetization M approximately varies as  $M = M_s + \chi_p H$ , where  $\chi_p$  is the susceptibility (in this case,  $\chi_p \sim$  the sum of the susceptibilities of Mn<sub>3</sub>O<sub>4</sub> and MnO, assuming negligible interaction between the two species). It follows that  $(M/H) = (M_s/H) + \chi_{p_s}$ Therefore, a plot of (M/H) vs (1/H) should yield a straight line with the slope  $M_s$  and  $\chi_p$  as the intercept. It was seen that this plot is linear below 20 K and above 50 K (up to H = 8 kOe used in these measurements). The  $M_s$  values agreed perfectly with those shown in Fig. 3. For the 20-50-K range, where nonlinearity is present in these plots, the lowfield side of the curve was chosen, where the slope yielded  $M_s$  values in agreement with those shown in Fig. 3. The important new information is in the variation of  $X_p$  with temperature, shown in Fig. 6. The peak in  $X_p$  occurs near 43 K, the  $T_c$  for Mn<sub>3</sub>O<sub>4</sub>. This is understandable since the variation of X for pure MnO is relatively small (  $\sim 20\%$ ) in the temperature range of 4.2-50 K. We believe that this is further



FIG. 6. The temperature variation of the magnetic susceptibility of  $Mn_3O_4$  phase in MnO samples, derived by the method described in the text. The peaks occur near 43 K, the  $T_c$  for  $Mn_3O_4$ . The lines through the points are drawn for visual aid and clarity.

confirmation of our suggestion that the impurity in MnO is Mn<sub>3</sub>O<sub>4</sub>. Powder x-ray analysis of all the samples was also carried out to detect Mn<sub>3</sub>O<sub>4</sub>. Several major lines characteristic of Mn<sub>3</sub>O<sub>4</sub> (with sample E as a reference for pure MnO) were observed in samples A, B, C, and D using long scans (at least 16 h), with Cr  $K_{\alpha}$  radiation.

In order to convert Mn<sub>3</sub>O<sub>4</sub> to MnO present in our "MnO" samples, we attempted several experiments. In an initial experiment, an MnO crystal was placed in a quartz tube, evacuated to  $10^{-5}$  Torr and sealed. The sample was then annealed for two days at 1353 K and quenched to room temperature. The magnetic properties were again measured. It was found that a reduction in the concentration of Mn<sub>3</sub>O<sub>4</sub> did occur. Annealing another sample for four days produced further reduction. These results are consistent with the Mn-O phase diagram at high temperatures.<sup>25</sup> Following the work of Price and Wagner<sup>26</sup> who studied the diffusion of Mn in crystalline MnO, we then tried a different experiment to achieve a pure MnO crystal. In this experiment, a small amount of Mn was placed at one end of the quartz tube, whereas a



FIG. 7. The magnetic-susceptibility variation with temperature for an annealed MnO single crystal oriented randomly (sample F).

MnO single crystal (sample D) was placed at the other end and the tube was sealed under a vacuum of  $10^{-5}$  Torr. The tube was then heated for about 200 h at 1293 K and quenched to room temperature. The magnetic susceptibility of the sample in the singlecrystal form (sample F) was again measured after etching in dilute HCl and washing with H<sub>2</sub>O and acetone, and the results for a random orientation are shown in Fig. 7. There was no hint of the remanent magnetization at 4.2 K or at higher temperatures. The M vs H data for this sample and sample E are shown in Fig. 8. From this and the comparison of the data of Figs. 2 and 7, it is clear that the results for samples E and F are quite similar, suggesting the complete absence of Mn<sub>3</sub>O<sub>4</sub> after the annealing in Mn atmosphere. Although gravimetric<sup>27</sup> and electrical properties<sup>28</sup> of MnO for different levels of non-



FIG. 8. The magnetic-field dependence of the magnetization for pure MnO samples (E and F). The straight lines (full for sample E and dotted for sample F) demonstrate the deviation at higher fields from the low-field behavior.

stoichiometry have been reported in the literature, to our knowledge this is the first time that a pure MnO single crystal has been obtained by the technique described above.

#### **IV. DISCUSSION**

From the data in Fig. 1, it is noted that below  $T_N$ , the minimum in the susceptibility for samples A through D occurs at different temperatures. There appears to be a direct correlation between the magnitude of Mn<sub>3</sub>O<sub>4</sub> present in a sample and the temperature at which the minimum occurs. The approximate temperatures for the minimum are, respectively, 80, 75, 70, and 70 K for samples A, B, C, and D (Fig. 1). Now the magnetic susceptibility of  $Mn_3O_4$  is expected to increase very rapidly as the temperature is lowered towards  $T_c$  (43 K), whereas  $\chi$  for MnO follows the variation shown in Fig. 2. Therefore the above behavior of the shift in the minimum temperature with Mn<sub>3</sub>O<sub>4</sub> concentration is easily understood if we make the reasonable assumption that the measured susceptibility is some combination of the relative susceptibilities of MnO and Mn<sub>3</sub>O<sub>4</sub>.

Next we consider the possible effects of the Mn<sub>3</sub>O<sub>4</sub> impurity on the magnetic properties of MnO near and above  $T_N$ . From the data presented in Table I and Fig. 1, it follows that there is no systematic variation of the susceptibility or the Néel temperature with the concentration of  $Mn_3O_4$ . However, a more definite result of the effect of  $Mn_3O_4$  impurity on  $\chi$  and  $T_N$  is obtained by comparing the values for these parameters for sample F (with Mn<sub>3</sub>O<sub>4</sub> impurity removed) with those for its parent sample D (with 2 at. %  $Mn_3O_4$ ). This comparison (Table I) shows that the absolute values of X and  $T_N$ , within the experimental uncertainties of  $\pm 1\%$  and  $\pm 0.5$  K, respectively, in the absolute values,<sup>24</sup> are unaffected by about 2 at. % Mn<sub>3</sub>O<sub>4</sub> impurity. To provide an explanation for this behavior, we follow similar line of reasoning as used in the preceding paragraph. Near room temperature, X/g for Mn<sub>3</sub>O<sub>4</sub> is smaller<sup>19</sup> than that for MnO. A simple calculation, assuming negligible interaction between the two phases, shows that a concentration of about 3 at. % Mn<sub>3</sub>O<sub>4</sub> in MnO would actually lower the total susceptibility by about 1% (equal to the experimental uncertainty). Therefore, the impurity levels of few atomic percent are not detectable by the magnetic-susceptibility studies above  $T_N$ . The value of  $T_N$  should also be unaffected by a few atomic percent of  $Mn_3O_4$  since x for  $Mn_3O_4$  is varying very slowly relative to that for MnO near the Néel temperature of MnO. The irregular variation of  $\chi$  and  $T_N$  with the level of Mn<sub>3</sub>O<sub>4</sub> impurity in the unannealed samples A through D (Table I) cannot, however, be understood on the basis of the above arguments. These irregularities perhaps reflect the way in

which the  $Mn_3O_4$  impurity is distributed in the MnO matrix in various samples.

At 4.2 K, in the ordered state of Mn<sub>3</sub>O<sub>4</sub> and MnO, the observation of a displaced hysteresis loop (Fig. 5) suggests the presence of an interaction between the two phases. Such a displaced loop is usually attributed to the exchange anisotropy induced at ferromagnetic-antiferromagnetic interfaces when a sample is cooled in a magnetic field and the magnitude of the exchange anisotropy is given by the displaced area.<sup>20</sup> Such a displaced loop has also been observed in wustite (Fe<sub>x</sub>O, x = 0.945)<sup>29</sup> and Co-CoO system.<sup>30</sup> There is, however, an important difference in the present observations in that the displaced loop is observed whether the sample is cooled in a magnetic field or in a nearly zero magnetic field. In the latter case, after the initial hysteresis in the first cycle, the displaced loop is observed in subsequent cycles (Fig. 4). This may in some way be related to the fact that  $T_c$  for the ferrimagnetic phase is less than  $T_N$  for the antiferromagnetic phase in the  $Mn_3O_4$ -MnO system, in contrast to the situation in the Co-CoO system. Additional studies are necessary to understand the observations presented here.

### V. CONCLUDING REMARKS

Definite evidence for the presence of  $Mn_3O_4$  in commercially available single crystals of MnO has been presented in this paper. We have also described a procedure for the conversion of the  $Mn_3O_4$  impurity found in these crystals into MnO. The effect of even a few atomic percent of  $Mn_3O_4$  impurity on the magnetic properties of MnO is very significant at low temperatures. A displaced hysteresis loop, observed at 4.2 K, is attributed to the exchange anisotropy induced at the ferrimagnetic-antiferromagnetic (Mn<sub>3</sub>O<sub>4</sub>-MnO) interfaces although additional studies are needed to explain all the observations. The effects of a few percent of Mn<sub>3</sub>O<sub>4</sub> usually present in commercial crystals of MnO on other properties (such as optical<sup>16,17</sup> and elastic<sup>6</sup>) remain to be investigated. However, it is clear that care must be taken to characterize these samples before studies on such samples are undertaken.

In the near future, we hope to complete our studies on the principal susceptibilities of MnO, now that a pure single crystal of MnO is available. Then one can hope to improve upon the calculations of the exchange and anisotropy parameters for MnO based on the powder susceptibilities.<sup>13</sup> Plans are also underway to study the details of the kinetics of  $Mn_3O_4$ -MnO conversions under different atmospheres, temperatures, and durations.

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