Effects of 2.6-GeV U ion irradiation on the superconductivity of the kagome superconductors CsV_3Sb_5 and $Cs(V_{0.93}Nb_{0.07})_3Sb_5$

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We have investigated the effect of 2.6-GeV U irradiation on the superconductivity of CsV_3Sb_5 and $Cs(V_{0.93}Nb_{0.07})$ ₃Sb₅ single crystals by using resistivity and magnetization measurements. The critical transition temperature for CsV_3Sb_5 crystal is enhanced up to 4.3 K after irradiation with a dose of $B_{\Phi} = 2.0$ T, while the critical current density is increased up to $\sim 0.8 \times 10^4$ A/cm² after irradiation with a dose of $B_{\Phi} = 0.5$ T under self-field at 2.0 K, which is about four times larger than that for the pristine crystal under the same condition. For $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ crystal, the critical transition temperature remains almost constant up to $B_{\Phi} = 2.0$ T, followed by monotonic suppression with further increase of dose. Besides the enhanced critical current density, we observe an anomalous peak effect in the magnetic hysteresis loop close to zero field at low irradiation doses, which can be interpreted by the matching effect in irradiated $Cs(V_{0.93}Nb_{0.07})$ ₃Sb₅ crystals.

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I. INTRODUCTION

Kagome materials are currently a hot topic in the field of condensed matter physics $[1-3]$. The electronic structure of kagome materials usually demonstrates interesting features such as a flat energy band, van Hove singularity, and Dirac or Weyl point. Thus, a wide variety of quantum states are expected, including bond-density wave order, charge-density waves (CDWs), spin-density waves (SDWs), and spin liquid states, in addition to superconductivity by adjusting the band filling [\[4–9\]](#page-6-0). Recently, a new family of kagome compounds, AV_3Sb_5 ($A = K$, Rb, or Cs), was reported containing the 3*d* transition metal vanadium [\[10\]](#page-6-0). Subsequently, superconductivity, CDWs with time-reversal symmetry breaking, in-plane nematic order, and the giant anomalous Hall effect have been observed in the system $[11-17]$ $[11-17]$. The coexistence and competition between superconductivity and CDWs make these materials much more interesting. For example, two superconducting domes and a completely suppressed CDW transition have been observed in hole-doped samples, such as $CsV_{3-x}Ti_{x}Sb_{5}$ and $CsV_{3}Sb_{5-x}Sn_{x}$ [\[18–20\]](#page-7-0), while the enhanced superconductivity and suppressed CDW have been found in samples with isoelectronic dopings [\[21–23\]](#page-7-0). Besides the chemical doping, high pressure is another effective method to adjust the quantum state in the system. It has been reported

that superconductivity shows two-dome-like behavior under moderate pressure and the CDW is monotonically suppressed by pressure [\[24–26\]](#page-7-0).

High-energy ion irradiation is one of the effective methods to control superconductivity, especially for superconductors with competing orders. For example, the critical transition temperature, T_c , can be enhanced up to 7.5 K in NbSe₂ after electron irradiation [\[27\]](#page-7-0). Roppongi *et al.* found that both T_c and the CDW of CsV_3Sb_5 are suppressed after 2.5-MeV electron irradiation [\[28\]](#page-7-0). Recently, we found that 3-MeV proton irradiation enhances T_c concomitant with suppression of the CDW in CsV_3Sb_5 [\[29\]](#page-7-0). In contrast to light-particle irradiations, heavy-ion irradiations have been considered to be much more effective at modifying vortex dynamics due to the larger damaged volume fraction at the same dose [\[30\]](#page-7-0). Additionally, due to the geometrical similarity between vortices and defects, the columnar defects produced by heavy-ion irradiation usually give better pinning properties than point defects introduced by light-particle irradiations [\[30,31\]](#page-7-0). In the present study, we have conducted 2.6-GeV U irradiation experiments into CsV_3Sb_5 and $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ single crystals. A systematic investigation on the effect of heavy-ion irradiation on superconductivity in these kagome superconductors is reported.

II. EXPERIMENTAL METHODS Single crystals of CsV_3Sb_5 and $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ were grown by the self-flux method, with the Cs-Sb binary

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FIG. 1. STEM images of $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ irradiated by 2.6-GeV U at $B_{\Phi} = 1$ T. (a) Plan view: complete circles indicate locations of columnar defects. (b) Cross section: the arrow indicates the *c* axis. The insets of (a) and (b) show magnifications of each image, indicating an average diameter of the columnar defect is ∼7 nm.

eutectic mixture $Cs_{0.4}Sb_{0.6}$ as the flux. The mixture was put into an alumina crucible and then the crucible was sealed in a quartz tube [\[22\]](#page-7-0). The 2.6-GeV U irradiation was performed using a ring cyclotron at the RI Beam Factory operated by RIKEN Nishina Center and CNS. The projected range of the 2.6-GeV U ion in the CsV₃Sb₅ system is \sim 71 µm, which was calculated by the SRIM program [\[32\]](#page-7-0). In the present experiment, all the samples were cleaved into thin flakes with thickness less than ∼40 µm. All irradiations were carried out at room temperature in vacuum better than 1×10^{-5} Torr. The irradiation dose is labeled by the matching field, B_{Φ} , which is related to the density of columnar defects, *n* (cm⁻²), as $B_{\Phi} =$ $n\Phi_0$, where Φ_0 (= 2.07 × 10⁻¹¹ T cm²) is the flux quantum. Namely, $B_{\Phi} = 1$ T corresponds to $n = 5 \times 10^{10}$ cm⁻². The crystal structure was determined at room temperature by using a commercial diffractometer (Smartlab, Rigaku) with Cu $K\alpha$ radiation. The measurements of resistivity and magnetization were carried out using DynaCool-PPMS-9T (Quantum Design) and a superconducting quantum interference device (SQUID) magnetometer (MPMS-XL5, Quantum Design). Critical current density (J_c) is evaluated magnetically using the extended Bean model.

III. RESULTS AND DISCUSSION

Figure 1(a) shows a scanning transmission electron microscopy (STEM) image of defects in $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ irradiated by 2.6-GeV U at $B_{\Phi} = 1$ T along the incident ion beam (plan view). The columnar defects, some of which are marked by circles, appear as circular dots indicating almost isotropic structural damage in the *ab* plane as shown in the inset of Fig. $1(a)$. Small differences in sharpness at the edge indicate that some of the ion projectiles were weakly scattered after they entered the crystal. Figure $1(b)$ shows a cross-section view, which is elongated and looks like rods. It is clear that the defects are ion trajectories or ion tracks. The contrast between the ion tracks and matrix reflects the presence of strain in the area of structural damage. Similar to those in 2.6-GeV U-irradiated K-doped BaFe₂As₂ [\[33\]](#page-7-0), these defects are continuous. These defects have similar structure with vortices, which can serve as effective pinning centers and enhance the critical current density greatly. The average density of defects is about 3.7 × 10^{10} cm⁻², which is about 75%

of the expected values. High-resolution STEM images shown in the insets of Figs. $1(a)$ and $1(b)$ indicate the diameter of columnar defects is ∼7 nm, which is slightly larger than those in 2.6-GeV U-irradiated CaKFe₄As₄ and K-doped BaFe₂As₂ [\[33–35\]](#page-7-0). The distribution of these columnar defects is random, and some of them nearly overlap each other, which increases the vortex-vortex interactions and has an important effect on their vortex pinning behavior.

Figures $2(a)$ and $2(b)$ illustrate the normalized resistance $(R(T)/R(300 \text{ K}))$ versus temperature curves for CsV_3Sb_5 and $Cs(V_{0.93}Nb_{0.07})$ ₃Sb₅, respectively, under different conditions of U irradiation. The T_c is the onset superconducting transition temperature, which is determined by a 90% criterion of the resistivity. T_{CDW} is determined by the maximum of $dR(T)/dT$. For CsV₃Sb₅, the onset T_c is about 3.2 K, while a clear kink is observed in the *R*-*T* curve at ∼87 K, which corresponds to the CDW transition. The T_c value of CsV_3Sb_5 initially increases with increasing irradiation dose and reaches the maximum value of ~4.3 K at $B_{\Phi} = 2.0$ T. However, the *T*_c value is suppressed to ~3.9 and ~2.3 K for $B_{\Phi} = 6.0$ and 8.0 T, respectively. The dependence of T_c on B_{Φ} is similar to that of 2.5 MeV electron-irradiated $NbSe₂$ single crystals, where it is initially enhanced after electron irradiation and then followed by monotonic decline with further increase of irradiation dose $[27]$. Figure $2(c)$ shows the temperature dependence of normalized magnetic susceptibility for U-irradiated $CsV₃Sb₅$, and the data confirmed the results obtained by R - T measurements as shown in the inset of Fig. $2(a)$. For $Cs(V_{0.93}Nb_{0.07})_3Sb_5$, T_c remains almost unchanged up to $B_{\Phi} = 2.0$ T, followed by suppression at higher doses as shown in Figs. $2(b)$ and $2(d)$.

In order to give a comparison, the superconducting transition temperatures determined by the *M*-*T* data marked as $T_c(M)$ are also shown in Fig. [3\(a\).](#page-2-0) It is well known that there is a delicate competition between the superconductivity and CDW order in CsV_3Sb_5 . Generally speaking, the superconductivity can be enhanced upon suppression of CDW by external perturbation [\[36\]](#page-7-0). However, in the present case of $CsV₃Sb₅$, while the T_c increases up to 4.3 K at a dose of $B_{\Phi} = 2$ T, the CDW transition temperature (T_{CDW}) is initially enhanced up to ~94 K after U irradiation with $B_{\Phi} = 0.1$ T and then keeps a nearly constant value at $B_{\Phi} < 1.0$ T until it is strongly suppressed at large enough B_{Φ} as shown in Fig. [3\(a\).](#page-2-0) Accompanied by enhanced T_c , the value of the residual resistivity ratio (RRR) almost monotonically increases from ∼15 to 77 as B_{Φ} is increased from 0 to 1.0 T and then decreases continuously with further increase of B_{Φ} , which is different from the report in Refs. [\[28,29,37\]](#page-7-0). It has been reported that $CsV₃Sb₅$ has conventional *s*-wave pairing with a nodeless superconducting gap $[28,38,39]$. Therefore, T_c is expected to be insensitive to the disorder in CsV_3Sb_5 according to Anderson's theorem [\[40\]](#page-7-0). Recently, Zhang *et al.* systematically investigated the effect of disorder on T_c and T_{CDW} of CsV_3Sb_5 , confirming the facts that both T_c and T_{CDW} are insensitive to disorder [\[41\]](#page-7-0). Thus, it is reasonable to infer that the enhancement of T_c and T_{CDW} in the present CsV_3Sb_5 crystals has also a weak dependence on disorder. The reason why the values of RRR, T_{CDW} , and T_c increase after small doses of U irradiation is still unclear. One possible reason is the structural relaxation of the crystals after U irradiation.

FIG. 2. Temperature dependence of normalized resistance $R(T)/R(300 \text{ K})$ for (a) CsV₃Sb₅ and (b) Cs(V_{0.93}Nb_{0.07})₃Sb₅ with different conditions. Insets: The corresponding magnifications of superconducting transition at temperatures close to *T*c. The temperature dependence of normalized magnetic susceptibility for (c) CsV_3Sb_5 and (d) $Cs(V_{0.93}Nb_{0.07})_3Sb_5$.

It has been reported that the Debye temperature (Θ_D) of $CsV₃Sb₅$ crystals is about 160 K, which is even smaller than that of NbSe₂ with $\Theta_{\rm D}$ ∼ 225 K [\[42,43\]](#page-7-0). Thus, the observation is expected of a much stronger dependence of lattice parameter on irradiation dose in $CsV₃Sb₅$ crystals than that in NbSe₂ due to much smaller elastic constant. Figure $4(a)$ shows the evolution of the diffraction peak of (004) versus the irradiation dose for CsV_3Sb_5 crystals. The x-ray diffraction (XRD) pattern in a broad angle range is shown in Fig. S1 of the Supplemental Material [\[44\]](#page-8-0). It can be seen that diffraction angle increases slightly with increasing irradiation dose. The corresponding values of the *c*-axis lattice parameter shrink with increasing irradiation dose as shown in Fig. [4\(b\).](#page-3-0) The

change of c -axis lattice parameter versus B_{Φ} is approximately linear up to $B_{\Phi} = 8$ T, and the corresponding strain is about -0.11% . However, the c - B_{Φ} curve strongly deviates from linearity above $B_{\Phi} = 8$ T as shown in Fig. [4\(b\).](#page-3-0) It is well known that the lattice constant generally increases with the introduction of defects [\[45\]](#page-8-0). In this sense, shrinkage of the *c*-axis lattice parameter is anomalous. Recently, Yang *et al.* have pointed out that in-plane uniaxial-strain tuning of T_c in CsV_3Sb_5 is dominated by associated *c*-axis strain [\[46\]](#page-8-0). However, the calculated strain is only about −0.006% along the *c* axis for CsV_3Sb_5 crystals after 2.6-GeV U irradiations with $B_{\Phi} \sim 2$ T. The strain is too small to explain the remarkable enhancement of T_c . Another possible explanation

FIG. 3. (a) Irradiation dose (B_{Φ}) dependence of T_c and T_{CDW} for CsV₃Sb₅ and (b) irradiation dose (B_{Φ}) dependence of T_c for $Cs(V_{0.93}Nb_{0.07})_3Sb_5.$

FIG. 4. (a) The XRD pattern of (004) peaks and (b) the curve of *c*-axis lattice parameters versus B_{Φ} for CsV_3Sb_5 single crystals with 2.6-GeV U irradiation.

could be due to the slight oxidation (hole doping) of the material triggered by absorption of oxygen on the surface. In Ref. [\[24\]](#page-7-0), systematic changes of physical properties, including the enhancement of T_c and suppression of T_{CDW} , due to surface oxidation have been demonstrated by studying $CsV₃Sb₅$ crystals with different thicknesses down to 12 nm. Usually, the effect of such hole doping is restricted only to the surface of the crystal. However, since the resistivity in the defect region is expected to increase significantly, introduced surface charge by oxidation can also modify the carrier density in the bulk of the crystal. In this case, the density of states and nesting of the Fermi surface can be changed [\[24\]](#page-7-0). The excessive reduction of the *c*-axis lattice parameter at large irradiation doses can be attributed to the electronic excitation, resulting in producing additional defects [\[47\]](#page-8-0). It has been reported that the interplay between the CDW and superconductivity in the AV_3Sb_5 ($A =$ K, Rb, Cs) system is not simply a competition for the density of states near the Fermi level, and the electronic correlations also play an important role in manipulating the CDW and its entanglement with superconductivity [\[48\]](#page-8-0). In addition, the initial increase of RRR after 2.6-GeV U irradiation can also be originated from the hole doping into the bulk.

Figures $5(a)$ and $5(b)$ show the temperature dependence of normalized resistance (*R*(*T*)/*R*(5 K)) for the U-irradiated CsV_3Sb_5 and $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ with $B_{\Phi} = 2.0$ T under different magnetic fields $(H||c)$, respectively. As we have already mentioned above, the superconductivity for $CsV₃Sb₅$ is greatly enhanced upon U irradiation, and the superconductivity is observed above 2.0 K even at a high magnetic field of 1.5 T. For $Cs(V_{0.93}Nb_{0.07})_3Sb_5$, the superconductivity is robust and no clear decay is observed at B_{Φ} less than 2.0 T. The superconducting transition is very sharp, which is similar to low-T_c conventional superconductors, and "122" and "1144" iron-based superconductors [\[49–51\]](#page-8-0), indicating small thermal fluctuations as shown in Figs. $5(a)$ and $5(b)$. As a comparison, the curves of $R(T)/R(5 K)$ versus temperature for $CsV₃Sb₅$ and $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ with $B_{\Phi} = 0.3$ T are shown by Figs. S2(a) and S2(b), respectively, of the Supplemental Material $[44]$. The upper critical fields, H_{c2} , were evaluated following the Ginzburg-Landau theory of $H_{c2} = H_{c2}(0)(1$ t^2 /(1+ t^2) by using the criterion of $R = 0.5R_n$, where R_n and t are the normal state resistance just above T_c and the reduced temperature $(t = T/T_c)$, respectively. The evaluated values of $H_{c2}(0)$ for CsV₃Sb₅ and Cs(V_{0.93}Nb_{0.07})₃Sb₅ with

 $B_{\Phi} = 2.0$ T are 1.9 and 2.2 T, respectively. The corresponding coherence lengths $\xi_{ab}(0)$ are 13 and 12 nm, respectively. These lengths are more than five times larger than that of iron-based and cuprate superconductors [\[52\]](#page-8-0).

Figures $5(c)$ and $5(d)$ show the magnetic hysteresis loops (MHLs) for CsV_3Sb_5 and $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ irradiated by 2.6-GeV U at $B_{\Phi} = 2.0$ T, respectively. MHLs for samples with other B_{Φ} are shown by Figs. S3 and S4 of the Supplemental Material [\[44\]](#page-8-0). The symmetric shape of the MHL for the irradiated $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ suggests that bulk pinning instead of surface pinning is in charge of vortex pinning in this sample. For CsV_3Sb_5 , the widths of MHLs initially become larger with increasing irradiation dose and reach the maximum value at $B_{\Phi} \sim 1.0$ T, and decline by further increase of *B*- as shown in Figs. S3 of the Supplemental Material [\[44\]](#page-8-0) and Fig. $5(c)$. For easy comparison, the MHL for pristine CsV₃Sb₅ is also shown by Fig. S5 of the Supplemental Material [\[44\]](#page-8-0). In addition, no peak effect is observed in $CsV₃Sb₅$. This is clearly different from the proton-irradiated CsV_3Sb_5 in which a broad peak was observed at intermediate fields [\[29\]](#page-7-0). The central peak near zero field for CsV_3Sb_5 is sharp even after U irradiation, unlike the case of $Ba(Fe_{0.93}Co_{0.07})$ ₂As₂ irradiated by 200-MeV Au where the magnetization peak near zero field is broad [\[53\]](#page-8-0). For $Cs(V_{0.93}Nb_{0.07})_3Sb_5$, symmetric MHLs and similar changing tendency with increasing magnetic fields are observed. In contrast to $CsV₃Sb₅$, one prominent feature is that a dip structure near zero field is observed at low B_{Φ} as shown in Fig. [7\(a\)](#page-5-0) and Fig. S4 of the Supplemental Material [\[44\]](#page-8-0), which is similar to the iron-based superconductors, cuprate superconductors, and NbSe₂ after introduction of columnar defects [\[30,](#page-7-0)[54–56\]](#page-8-0).

Figures $6(a)$ and $6(b)$ illustrate the irradiation dose dependence of J_c under the self-field at various temperatures in CsV₃Sb₅ and Cs(V_{0.93}Nb_{0.07})₃Sb₅. For CsV₃Sb₅, the J_c displays different tendencies with increasing B_{Φ} at different temperatures. For example, at 2.3 K, the J_c initially increases with increasing B_{Φ} and decreases at B_{Φ} larger than 1.0 T, while it increases monotonically with increasing B_{Φ} at temperatures above 2.6 K. One reasonable reason for such behavior is that the U irradiation induces δT_c pinning in $CsV₃Sb₅$, which is very effective at temperatures close to T_c . Here δT_c pinning is defined as flux pinning induced by spatial fluctuations of the superconducting transition temperature *T*_c. The *J*_c value is about 0.8×10^4 A/cm² for CsV₃Sb₅ at

FIG. 5. Temperature dependence of normalized resistance $R(T)/R(5 K)$ under different magnetic fields for (a) $CsV₃Sb₅$ and (b) $Cs(V_{0.93}Nb_{0.07})_3Sb_5$, at $B_{\Phi} = 2.0$ T. MHLs under different temperatures for (c) CsV_3Sb_5 and (d) $Cs(V_{0.93}Nb_{0.07})_3Sb_5$, at $B_{\Phi} = 2.0$ T.

 $B_{\Phi} = 0.5$ T, which is about four times larger than the pristine sample at the same condition. For $Cs(V_{0.93}Nb_{0.07})_3Sb_5$, the J_c initially increases with increasing B_{Φ} and takes the maximum at about $B_{\Phi} = 0.5$ T, followed by monotonic decrease above $B_{\Phi} = 0.5$ T. The B_{Φ} dependencies of J_c at various temperatures are similar to each other for $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ as shown in Fig. $6(b)$. The largest value of J_c under the self-field at 2 K for $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ after 2.6-GeV U irradiation is ∼2.3 × 10⁴ A/cm² in a sample with B_{Φ} ~ 0.5 T, which is about 3.5 times larger than the pristine sample [\[29\]](#page-7-0). It should be noted that the J_c values in 2.6-GeV U-irradiated CsV_3Sb_5 and $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ are several orders of magnitude smaller than that in cuprates ($J_c(5 \text{ K}) \sim 1.35 \times 10^7 \text{ A/cm}^2$, 580 MeV Sn, $B_{\Phi} = 3$ T) [\[57\]](#page-8-0) and iron-based superconductors $(J_c(2 \text{ K}) \sim 1.4 \times 10^7 \text{ A/cm}^2, 2.6 \text{ GeV U}, B_{\Phi} = 8 \text{ T})$ [\[33\]](#page-7-0),

indicating that pinning in kagome superconductors is weak. One possible reason is that the typical diameter of columnar defects (∼7 nm) created by heavy-ion irradiation is much smaller than the coherence length of ∼20 nm, making the flux pinning less effective [\[58\]](#page-8-0). Generally, different defect structures (pointlike, cluster, columnar, etc.) have different effects at different field and current values, and a mix of defect types may give the best pinning effect [\[59\]](#page-8-0).

Figure [7\(a\)](#page-5-0) shows MHLs for $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ irradiated by 2.6-GeV U ions at $B_{\Phi} = 0.2$ T at various temperatures. It is clear that anomalous peak structures develop close to zero field at low temperatures. Similar peak structures can be observed in a sample irradiated at B_{Φ} = 0.1 T at low temperatures as shown in Fig. S4(a) of the Supplemental Material [\[44\]](#page-8-0). The peak structure gradually

FIG. 6. The curves of J_c versus B_{Φ} under self-field and various temperatures for (a) CsV₃Sb₅ and (b) Cs(V_{0.93}Nb_{0.07})₃Sb₅.

FIG. 7. *M-H* loops at various temperatures for 2.6-GeV U-irradiated $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ with $B_{\Phi} = 0.2$ T at (a) $\theta_H = 0$ [°], (b), $\theta_H = 10$ [°], and (c) $\theta_H = 30^\circ$. (d) The *J_c* versus *H* curves at different θ_H angles at $T = 2.0$ K. The inset of (c) shows a schematic diagram explaining $θ$ ^H as the angle between the columnar defects and the applied fields *H*. The inset of (d) shows temperature dependence of $μ_0H_p$ for $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ with $B_{\Phi} = 0.1$ and 0.2 T.

fades away with increasing B_{Φ} above $B_{\Phi} = 0.2$ T as shown in Figs. S4(b)–S4(d) [\[44\]](#page-8-0). Such abnormal magnetization behavior has also been observed in heavy-ion irradiated high-*T*_c cuprates $[60]$, NbSe₂, and iron-based superconductors $[61]$. For instance, peak effects located close to $\mu_0 H = B_\Phi/3$ have been observed in $YBa₂Cu₃O₇$ irradiated by 580-MeV Sn ions and in $Ba_{1-x}K_xFe_2As_2$ irradiated by 2.6-GeV U ions [\[33\]](#page-7-0), and dips near zero field in MHLs have been observed in $Ba(Fe_{0.93}Co_{0.07})_2As_2$ irradiated with 2.6-GeV U ions [\[30\]](#page-7-0). For the present U-irradiated $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ with B_{Φ} = 0.1 and 0.2 T, the peak field ($\mu_0 H_p$) locates at ∼0.55*B*_Φ at 2 K, which decreases with increasing temperature and finally merges into the central magnetization peak. The peak field μ_0H_p is about 0.06 and 0.11 T at 2.0 K for the U-irradiated samples with $B_{\Phi} = 0.1$ and 0.2 T, respectively. These values of $\mu_0 H_p$ are much larger than the self-field of $\mu_0 H_{sf}$, \sim 0.004 and 0.012 T for $B_{\Phi} = 0.1$ and 0.2 T samples, respectively. Thus, it is unlikely that the present peak structure is caused by the self-field effect. Specific arrangements of columnar defects can be classified into three types, namely, parallel columnar defects (columnar defects parallel to the *c* axis), tilted columnar defects (columnar defects tilted from the *c* axis), and bimodal splayed columnar defects (columnar defects are tilted symmetrically from the *c* axis). In superconductors with parallel columnar defects, peak effects which have similar shapes to those observed in the present study have not been observed [\[62\]](#page-8-0). On the other hand, in superconductors with tilted columnar defects [\[57,63\]](#page-8-0) and splayed columnar defects [\[33](#page-7-0)[,56,64\]](#page-8-0), similar peak effects have been observed

at fields close to αB_{Φ} with $0.1 < \alpha < 0.3$. In the present case, $\mu_0 H_p$ is closer to B_{Φ} , suggesting that it is caused by the matching effect.

When a regular array of pinning centers is introduced into superconductors, the pinning force density reaches the maximum value when all pinning centers are occupied by vortices. On the other hand, when the arrangement of pinning centers is random as in the case of columnar defects created by heavy-ion irradiations, the optimum pinning can be realized before all pinning centers are occupied by vortices due to the competition between pinning energy and vortex-vortex interactions. However, even in superconductors with randomly distributed pinning centers, when the vortex-vortex interaction is not strong enough, the maximum pinning force density is reached when the applied magnetic field (H) is close to B_{Φ} , namely, $\mu_0 H/B_{\Phi} \sim 1$ [\[56\]](#page-8-0). In the present case of U-irradiated $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ crystals, although the exact value of the penetration depth is not known, it is expected to be shorter than that for the pristine material of 387 nm [\[28\]](#page-7-0), making vortex-vortex interactions weaker and $\mu_0 H_p/B_{\Phi} \sim 0.55$. At higher temperatures, as the penetration depth becomes longer, the optimum pinning is realized at lower applied fields as observed in the inset of Fig. 7(d) [\[58,65,66\]](#page-8-0).

Figures $7(b)$ and $7(c)$ show the MHL when the external field is applied at $\theta_H = 10^\circ$ and 30° from the *c* axis, respectively. Figure $7(d)$ shows the magnetic field dependence of J_c at 2.0 K at different θ_H . It is clear that the peak structure becomes weaker by titling the external field away from the columnar defects. At $\theta_H = 10^\circ$, the peak structure becomes less pronounced, and nearly normal field dependence of J_c with a weak hump at 0.15 T is observed at $\theta_H = 30^\circ$. Another important feature of the peak structure is that it fades away with increasing temperature. Due to the increase of the coherence length $\xi(T)$ with increasing temperature, the pinning by columnar defects becomes weaker. As a result, the peak structure fades away as shown in Fig. $7(a)$. It should be noted that similar peak structures are not observed in $CsV₃Sb₅$ after 2.6-GeV U irradiation at $T = 2.0$ K. One possible reason is that the measuring temperature of 2.0 K is too close to its T_c so that the dip structure is not developed, similar to the case of 2.6-GeV U-irradiated $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ at high temperatures.

IV. CONCLUSIONS

In summary, we have studied the effect of 2.6-GeV U irradiation on the superconductivity of $CsV₃Sb₅$ and $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ single crystals. We observe that the *c*axis lattice parameter shrinks after irradiation. Besides the enhancement of J_c , the T_c is enhanced up to ~4.3 K after U irradiation at $B_{\Phi} = 2.0$ T in CsV₃Sb₅. By contrast, the T_c value for the 2.6-GeV U-irradiated $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ initially keeps a high value of \sim 4.3 K at B_{Φ} less than

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2.0 T, followed by monotonic decrease with further increase of the irradiation dose. Upon 2.6-GeV U irradiation, the *J*^c in Cs(V_{0.93}Nb_{0.07})₃Sb₅ is enhanced up to \sim 2.3 × 10⁴ A/cm² under the self-field at $T = 2.0$ K, which is about 3.5 times larger than that in the pristine sample at the same condition. We observe peak structures in irreversible magnetization loops at low fields only in $Cs(V_{0.93}Nb_{0.07})_3Sb_5$ irradiated by 2.6-GeV U ions at $B_{\Phi} < 0.5$ T. Although the peak structure has some similarity to the self-field peak effect observed in iron-based superconductors with columnar defects along the *c* axis, the peak field is about ten times larger than the self-field, and is close to B_{Φ} . A matching effect of vortices with the density of columnar defects affected by weaker vortex-vortex interactions due to temperature-dependent penetration depth can be an alternative explanation.

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