


Light-hole spin confined in germanium

Patrick Del Vecchio and Oussama Moutanabbir ^{*}

Department of Engineering Physics, *École Polytechnique de Montréal*,
Montréal, C.P. 6079, Succ. Centre-Ville, Montréal, Québec, Canada H3C 3A7



(Received 24 February 2024; revised 14 May 2024; accepted 17 June 2024; published 3 July 2024)

The selective confinement of light holes (LHs) in a tensile-strained germanium (Ge) quantum well is studied by mapping the electronic structure of $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructures as a function of Sn content, residual strain, and Ge well thickness. It is shown that above 12 at.% Sn and below 0.4% residual compressive strain in the barriers, the tensile strain in Ge becomes sufficiently large to yield a valence band edge with LH-like character, thus forming a quasi two-dimensional LH gas in Ge. The LH ground state has a larger in-plane effective mass than that of heavy holes (HHs) in $\text{Si}_{1-y}\text{Ge}_y/\text{Ge}/\text{Si}_{1-y}\text{Ge}_y$ quantum wells. Moreover, LHs in optimal $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructures are found to exhibit a strong g -tensor anisotropy, with the in-plane component one order of magnitude larger than that of HHs in typical planar systems. Two of three structure-inversion-asymmetry Rashba parameters, both of which are critical in electric-dipole-spin-resonance experiments, are effectively 10 times the size of the cubic Rashba parameter in HH quantum wells. In the regime of LH selective confinement, every layer of the heterostructure is of direct bandgap, which can be relevant for efficient optical photon-spin qubit interfaces. This work discusses the broad landscape of the characteristics of LH spin confined in Ge to guide the design and implementation of LH spin-based devices.

DOI: [10.1103/PhysRevB.110.045409](https://doi.org/10.1103/PhysRevB.110.045409)

I. INTRODUCTION

Hole spins in group-IV planar gated quantum dots are promising candidates for robust and scalable qubits [1–7]. Developing these qubits has been so far exclusively based on heavy-hole (HH) states, as the materials currently used are restricted to compressively strained germanium (Ge) heterostructures. Interestingly, the advent of the germanium/germanium-tin ($\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$) material system provides an additional degree of freedom to tailor the valence band character in tensile-strained Ge, thus paving the way to implement silicon-compatible platforms for light-hole (LH) spin qubits [8,9]. These LH qubits share many of the advantages benefiting the HH ones but also bring about other attractive characteristics pertaining to LHs and $\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructures. These include a strong Rashba-type spin-orbit interaction (SOI) [9] and an efficient coupling with proximity-induced superconductivity [10] in addition to the bandgap directness [11] relevant to coupling with optical photons. These characteristics can expand the functionalities of hole spin qubits by facilitating the implementation of hybrid superconductor-semiconductor devices and photon-spin interfaces.

Although $\text{Ge}_{1-x}\text{Sn}_x$ semiconductors have been the subject of extensive studies in recent years, research in this area has mainly been focused on integrated photonics and optoelectronics leveraging $\text{Ge}_{1-x}\text{Sn}_x$ strain- and composition-dependent bandgap [11] leaving their spin properties practically unexplored [9,12–14]. As a matter of fact, studies on hole spin in $\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ are still conspicuously missing in

the literature. Herein, this work addresses the dynamics of LH spin confined in $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructures and elucidates the key parameters affecting its behavior as a function of strain, well thickness, Sn content, and magnetic field orientation. First, the article describes and discusses the electronic structure of tensile-strained Ge on $\text{Ge}_{1-x}\text{Sn}_x$. Second, the parameters defining the band alignment of the $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ quantum well are introduced and the criteria for LH confinement in Ge are established. The third section outlines the Hamiltonian of the in-plane motion of LHs for out-of-plane and in-plane magnetic fields yielding LH parameters such as the effective mass, the g -tensor, and the Rashba-SOI parameters. LH-HH mixing within the LH ground state is investigated in the fourth section. It is important to note that the focus here is on LH properties in the planar system without the effects of electrostatic in-plane confinement introduced in quantum dot systems. [9]

II. LH QUANTUM WELL IN $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$

A. $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructure

Before delving into the details of the electronic structure of $\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$, it is instructive to examine the strain-related behavior of bulk Ge. Figure 1(a) outlines the effect of tensile strain on the band structure of bulk Ge calculated by 8-band (full lines) and 4-band (dashed lines) $k \cdot p$ theory. Here, the calculations assume a bi-isotropic biaxial strain in the (001)-plane without any shear deformations, which is expected for an ideal [001]-oriented epitaxial growth [8,15]. Under these conditions, the fourfold degeneracy of the VB edge at the Γ point is lifted yielding two spin-degenerate LH and HH bands. In the case of tensile strain, the VB edge is LH-like, whereas

^{*}Contact author: oussama.moutanabbir@polymtl.ca

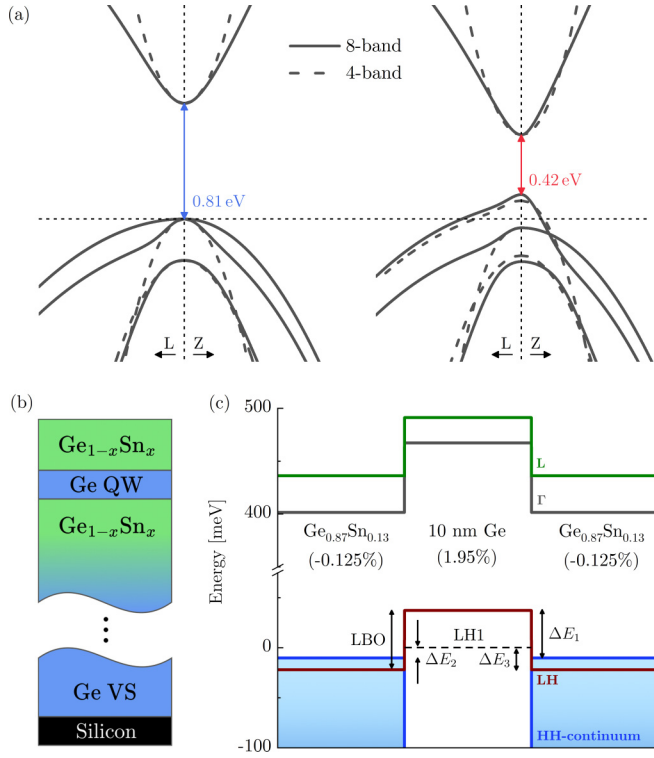


FIG. 1. (a) Bulk dispersion near the Γ point ($\|\mathbf{k}\| \leq 2.5 \text{ nm}^{-1}$ in both directions) of relaxed Ge (left) and tensile-strained Ge (right) computed by the 8-band (full lines) and the 4-band (dashed lines) $k \cdot p$ frameworks. The nonzero components of the strain tensor are $\varepsilon_{xx} = \varepsilon_{yy} = 2\%$ and $\varepsilon_{zz} = -2(c_{12}/c_{11})\varepsilon_{xx} \approx -1.3\%$. The conduction band was obtained from an effective mass approximation in the 4-band computations. (b) Schematic illustration of the proposed $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructure. (c) Band alignment profile of a $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ quantum well for selected barrier parameters. In this case, $\Delta E_{1,2,3}$ and LBO are positive. ε_{xx} in each layer is written in parentheses.

in the broadly studied compressively strained Ge, the VB edge is of the HH type. As discussed in the following, there is a threshold of tensile strain beyond which it becomes possible to control and selectively manipulate spin 1/2 LHs instead of spin 3/2 HHs.

The proposed heterostructure consists of a $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ quantum well grown on silicon [8], as illustrated in Fig. 1(b). Thick $\text{Ge}_{1-x}\text{Sn}_x$ buffer layers with an increasing Sn content from the Ge virtual substrate (VS) up to the bottom $\text{Ge}_{1-x}\text{Sn}_x$ barrier prevents the propagation of defects and dislocations near the Ge quantum well [8]. The lattice mismatch between $\text{Ge}_{1-x}\text{Sn}_x$ and Ge (with Sn contents above ~ 10 at. %) is leveraged to achieve high tensile strain in the coherently grown Ge quantum well. A top $\text{Ge}_{1-x}\text{Sn}_x$ barrier is then grown on the tensile-strained Ge layer. As shown in the following, careful engineering of the lattice strain and Sn content leads to a direct bandgap Ge LH quantum well.

B. Band alignment and LH confinement

The energy band alignment of the $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructure is computed within the assumption that the

in-plane lattice constant a_{\parallel} is the same in each layer, which is consistent with a pseudomorphic epitaxial growth. The in-plane components ε_{xx} and ε_{yy} of the strain tensor in a material i are given by $\varepsilon_{xx} = \varepsilon_{yy} = a_{\parallel}/a'_0 - 1$, where a'_0 is the lattice constant of the fully relaxed material i . The out-of-plane strain component ε_{zz} then follows immediately from the relation $\varepsilon_{zz} = -2(c_{12}/c_{11})\varepsilon_{xx}$, where c_{11} and c_{12} are the material elastic constants (see Table I) and $\varepsilon_{kl} = 0$ if $k \neq l$ [15]. Three parameters determine completely the band alignment of the $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructure: the Sn content x , the in-plane strain ε_{BR} in the barriers, and the in-plane strain in the Ge well ε_{W} . Since the latter can be counted for from the condition that a_{\parallel} is constant along the growth direction z , only the barrier composition and strain (x , ε_{BR}) are required to evaluate the band alignment.

A typical band alignment is displayed in Fig. 1(c) for $x = 0.13$ and $\varepsilon_{\text{BR}} = -0.125\%$. In this instance, HHs and electrons are pushed away from Ge and form a continuum of states in the $\text{Ge}_{0.87}\text{Sn}_{0.13}$ barriers. Meanwhile, LHs are selectively confined in Ge, thereby forming a LH quantum well. Here, the combination of large strain in Ge ($\varepsilon_{\text{W}} = 1.95\%$) with small ε_{BR} pushes the LH ground state (LH1) above the HH continuum, leading to the possibility, at very low hole density, to populate only LH1 and thus to create a pure quasi-two-dimensional LH gas in Ge. Such system is achievable only in a specific region of the parameter space (x , ε_{BR}), depending on four energy offsets [see Fig. 1(c)]:

$$\Delta E_1 = \max(\text{LH}) - \max(\text{HH}), \quad (1)$$

$$\Delta E_2 = E_{\text{LH1}} - \max(\text{HH}), \quad (2)$$

$$\Delta E_3 = E_{\text{LH1}} - \min(\text{LH}), \quad (3)$$

$$\begin{aligned} \text{LBO} &= \max(\text{LH}) - \min(\text{LH}) \\ &= \Delta E_1 - \Delta E_2 + \Delta E_3. \end{aligned} \quad (4)$$

Here, $\max(\text{LH})$ and $\min(\text{LH})$ are the energies at the bottom and top of the LH quantum well, respectively, and $\max(\text{HH})$ is the energy at the edge of the HH continuum. The zero energy point is placed on the ground LH subband (i.e., $E_{\text{LH1}} = 0$). Band offsets ΔE_1 and LBO (LH band offset) do not depend on the well thickness w .

Figure 2(a) presents a two-dimensional map of band offsets LBO and ΔE_1 with x and ε_{BR} as independent parameters. The corresponding strain in the Ge well ε_{W} is also shown (black dotted lines) only for the tensile strain regime ($\varepsilon_{\text{W}} > 0$). The Ge indirect-to-direct transition occurs at $\varepsilon_{\text{W}} = 1.68\%$ (solid black line) according to the parameterization described in Appendix A. Similarly, the $\text{Ge}_{1-x}\text{Sn}_x$ barriers exhibit bandgap directness above the dashed-dot blue line. Constant LBO are indicated by the solid red curves, where LBO = 0 corresponds to a completely flat LH profile along the growth direction. Finally, dashed red curves indicate constant ΔE_1 , where $\Delta E_1 = 0$ corresponds to the LH band edge in Ge sitting at the same energy as the HH band edge in the barriers. As discussed in the following, a large and positive ΔE_1 allows LH1 to emerge from the continuum for sufficiently thick quantum wells.

LHs are confined in Ge if and only if LBO > 0. However, depending on ΔE_1 and the well thickness w , these confined LHs could be situated within the HH continuum (negative ΔE_2). In the region $\Delta E_1 \leq 0$, LHs can never emerge from

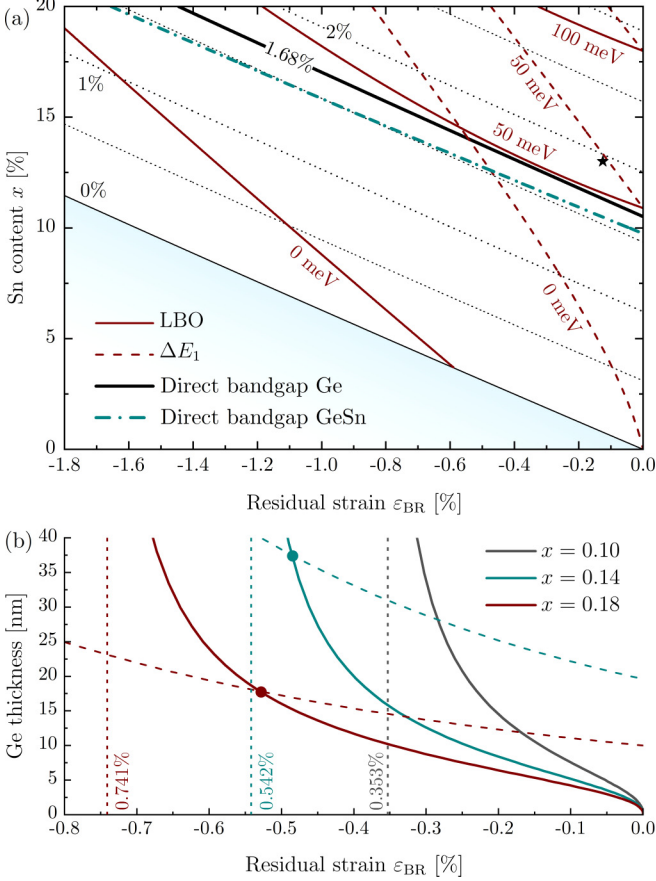


FIG. 2. (a) Two-dimensional contour map of band offsets ΔE_1 and LBO, strain in Ge, and bandgap directness as a function of x and ϵ_{BR} at $T = 300$ K. Contour lines for LBO and ΔE_1 are given at 0, 50, and 100 meV. The star indicates the system from Fig. 1(c). (b) Minimal well thickness w_0 (solid lines) required for a LH-like valence band edge as a function of ϵ_{BR} at fixed Sn content in the barriers. Dashed lines are the Ge critical growth thickness h_c estimated from the People and Bean relation. Solid circles indicate where $w_0 = h_c$ for a given x . For $x = 0.10$, $h_c \approx 46$ nm at $\epsilon_{BR} = 0$.

the continuum since the bottom of the LH well is below the HH band edge. However, if $\Delta E_1 > 0$, there is a minimal quantum well thickness w_0 for which LH1 emerges from the continuum. This lower bound depends on both Sn content and lattice strain (i.e., x and ϵ_{BR}) in the barriers and is plotted in Fig. 2(b) (solid lines). At $w = w_0$, the energy required for LH1 to escape the well is exactly $LBO - \Delta E_1$ (i.e., the strain-induced HH-LH splitting in the barriers). $w_0 \rightarrow \infty$ on the $\Delta E_1 = 0$ curve, whereas $w_0 = 0$ if the barriers are fully relaxed. In addition to the lower bound $w_0 < w$, the maximum strain energy that the Ge quantum well can accommodate also introduces an upper bound on w . This upper bound is given by the Ge critical growth thickness h_c , beyond which misfit dislocations start to appear at the interfaces and tensile strain in Ge is strongly suppressed. The dashed lines in Fig. 2(b) show an estimation of h_c based on the People and Bean formula [16,17]. Here, this model is applied for the critical thickness of a Ge layer with equilibrium lattice constant $a_0^{\text{Ge}} \equiv a_0(0)$ pseudomorphically grown on $\text{Ge}_{1-x}\text{Sn}_x$ with lattice constant $a_0(x)(1 + \epsilon_{BR})$.

III. LH SPIN PROPERTIES

A. Effective masses and spin parameters

In this section, the LH subband effective mass, the out-of-plane and in-plane g factors, and the Rashba parameters are computed as a function of Sn content in the $\text{Ge}_{1-x}\text{Sn}_x$ barriers (x) and well thickness w . These parameters give important information on how LHs move in the plane and how they respond to magnetic fields. Despite being intrinsic to a given subband (here, the focus is given to the lowest subband LH1), there is generally a significant influence from neighboring levels through inter-subband couplings. Moreover, the wavefunction spread across interfaces can also influence subband parameters. In heterostructures such as $\text{Ge}/\text{Si}_{1-y}\text{Ge}_y$ quantum wells, where both types of holes are confined into the same layer and the band offsets are large, intersubband coupling is significant to an energy scale comprising only a few tens of subbands due to quantization effects. For instance, it is often a reasonable approximation to include the coupling to only 1 or 2 LH subbands for a HH ground state [18], or around 50 LH subbands when band offsets are taken into account [19]. In $\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ quantum wells, intersubband couplings must include $\sim 10^2$ subbands due to the neighboring continuum. Moreover, small LBOs leading to a sizable spread of LH1 into the barriers require an accurate description of the subband envelopes. To address these effects, a numerical approach is employed instead of a variational method for the envelope problem [9]. Moreover, due to the nearby continuum, the coupling from the 200 closest subbands to LH1 are taken into account. Our implementation of $k \cdot p$ theory and how strain is incorporated into the model is described in Appendix B.

From the point of view of 8-band $k \cdot p$ theory, a subband such as LH1 always consists of the superposition of spin 1/2 CB electron, LH, and split-off (SO) hole envelopes. At $\mathbf{k}_{\parallel} = 0$, this can be written as

$$|\eta, \sigma\rangle = \left| \frac{1}{2}, \frac{\sigma}{2} \right\rangle_c |c\rangle + \left| \frac{3}{2}, \frac{\sigma}{2} \right\rangle_{\ell} |\ell\rangle + \sigma \left| \frac{1}{2}, \frac{\sigma}{2} \right\rangle_s |s\rangle, \quad (5)$$

where $\sigma = \pm 1$ is a pseudo-spin quantum number and $\langle z | c, \ell, s \rangle = \psi_{c,\ell,s}(z)$ are the CB, LH, and SO envelope functions, respectively. To avoid any confusion, ‘‘LH’’ subbands (e.g., LH1) are designated as η subbands to distinguish them from their LH envelope component. The kets $|\frac{3}{2}, \frac{\sigma}{2}\rangle \equiv |j, m\rangle$ and so on are the bulk Bloch states. The additional contribution from HHs away from $\mathbf{k}_{\parallel} = 0$ is investigated in the next section. An η subband is normalized according to

$$1 = \sum_{\tau=\{c,\ell,s\}} \langle \tau | \tau \rangle, \quad (6)$$

with $\langle \ell | \ell \rangle > \langle c | c \rangle$ and $\langle \ell | \ell \rangle > \langle s | s \rangle$ for a level such as LH1. For instance, the SO contribution $\langle s | s \rangle$ in LH1 is typically smaller than 10% for the range of barrier and well parameters considered here, but it plays an important role in the effective mass calculations, as discussed in the following. The CB contribution $\langle c | c \rangle$ in LH1 is around 5%, where the envelopes $|c\rangle$ are antisymmetric with respect to the center of the well and have their maximal amplitude near the interfaces.

For an out-of-plane magnetic field $\mathbf{B} = B\mathbf{e}_z$, the in-plane motion of $|\eta, \sigma\rangle$ is described by an effective two-dimensional

Hamiltonian (see Appendix C)

$$H_{\text{eff}}^{\perp} = \alpha_0 \gamma K_{\parallel}^2 + \frac{\alpha_0 g_{\perp}}{\lambda^2} \frac{\sigma_z}{2} + i\beta_1(K_- \sigma_+ - \text{h.c.}) - i\beta_2(K_+^3 \sigma_+ - \text{h.c.}) + i\beta_3(K_- K_+ K_- \sigma_+ - \text{h.c.}), \quad (7)$$

where $\alpha_0 = \hbar^2/(2m_0)$ with m_0 the free electron mass, $\mathbf{K} = \mathbf{k} + e\mathbf{A}/\hbar$ is the mechanical wavevector, $\mathbf{A} = B/2(-y\mathbf{e}_x + x\mathbf{e}_y)$ is the vector potential such that $\mathbf{B} = \nabla \times \mathbf{A}$, and $\mathbf{k} \rightarrow -i\nabla$ is the canonical wavevector. Notably, $K_{\parallel}^2 = \{K_-, K_+\}/2 = K_x^2 + K_y^2$, $1/\lambda^2 = [K_-, K_+]/2 = eB/\hbar$, and $K_{\pm} = K_x \pm iK_y$. The effective parameters in (7) are the following: $\gamma = m_0/m^*$ is the in-plane inverse effective mass, g_{\perp} is the out-of-plane g factor, and $\beta_{1,2,3}$ represent the three types of Rashba splittings. The first is linear in K , whereas the last two are cubic in K . Rashba parameters arise from space inversion asymmetry. When a small DC electric field $\mathbf{E} = E_z \mathbf{e}_z$ is applied to an otherwise symmetric well, all β parameters behave linearly with E_z and involve only odd powers of the field:

$$\beta_i = \alpha_i E_z + O(E_z^3), \quad (8)$$

where $i = 1, 2, 3$. For an in-plane magnetic field $\mathbf{B} = B(\mathbf{e}_x \cos \phi + \mathbf{e}_y \sin \phi)$ with vector potential $\mathbf{A} = B(\mathbf{e}_x \sin \phi - \mathbf{e}_y \cos \phi)z$, the in-plane motion is instead described by

$$H_{\text{eff}}^{\parallel} = \alpha_0 \gamma k_{\parallel}^2 + \frac{\alpha_0 g_{\parallel}}{\lambda^2} \frac{\sigma_z}{2} (e^{-i\phi} \sigma_+ + \text{h.c.}) + i\beta_1(k_- \sigma_+ - \text{h.c.}) - i\beta_2(k_+^3 \sigma_+ - \text{h.c.}) + i\beta_3(k_- k_+ k_- \sigma_+ - \text{h.c.}). \quad (9)$$

The in-plane g factor is given by

$$\frac{g_{\parallel}}{2} = \Im\{\langle c | z g k_z | c \rangle - 2\langle + | u'_+ | - \rangle\} - \sqrt{2}\Re\left\{\frac{1}{\sqrt{3}\alpha_0} \langle c | z P | + \rangle + \langle - | s \rangle\right\}, \quad (10)$$

where

$$|\pm\rangle = |\ell\rangle \pm 2^{\pm 1/2} |s\rangle, \quad (11)$$

$$u'_{\pm} = \{z\gamma_3, k_z\} \pm \{z\kappa, k_z\}, \quad (12)$$

with γ_3 a Luttinger parameter, κ the bulk hole g -factor, and P the so-called Kane momentum matrix element (see Appendix A). In the context of heterostructures, material parameters such as γ_3 , κ , or P are operators that act on envelope functions. Importantly, they do not commute with k_z and are diagonal in position basis. For example, $\gamma_3 |z\rangle = \gamma_3(z) |z\rangle$, where the function $\gamma_3(z)$ gives the value of γ_3 at coordinate z . In (10), the $z = 0$ coordinate is chosen such that

$$\langle z \rangle = \sum_{\tau=\{c,\ell,s\}} \langle \tau | z | \tau \rangle = 0. \quad (13)$$

This ensures, when setting $k_x = k_y = 0$, that g_{\parallel} is gauge independent and corresponds to that at the subband edge. This is because the quantum numbers k_{\pm} are generally gauge dependent, and if $\langle z \rangle = 0$, taking the expectation value on both sides of $K_{\pm} = k_{\pm} \mp iz e^{\pm i\phi}/\lambda^2$ gives $\langle K_{\pm} \rangle = k_{\pm}$, and thus

associates to k_{\pm} the gauge independent quantity $\langle K_{\pm} \rangle$. Equation (10) reduces to the well-known $|g_{\parallel}| = 4\kappa$ in the special case of 4-band Luttinger Hamiltonian with $LBO \rightarrow \infty$. The in-plane effective mass γ , the g -factor components, and the three Rashba parameters α_i are plotted as a function of x and w in Fig. 3 panels (a)–(c), and (d)–(f), respectively. A negative g means that the spin-down level ($\sigma = -1$) is closer to the bandgap than the spin-up level.

B. LH-HH mixing

In the vicinity of $\mathbf{k}_{\parallel} = 0$, η subbands acquire a small HH component in addition to the three terms in (5), resulting in a η -HH mixed state $|\psi, \mathbf{k}_{\parallel}\rangle$. For LH1, this can be written as (up to a normalization constant and to first order in k):

$$|\psi, \mathbf{k}_{\parallel}\rangle = |\eta, \sigma\rangle - i\sigma\alpha_0 k_{-\sigma} \left\{ \frac{3}{2}, \frac{3\sigma}{2} \right\} \sum_l \frac{T_l^x |h_l\rangle}{E_{\text{LH1}} - E_l^{\text{H}}} + \dots, \quad (14)$$

where $|\frac{3}{2}, \frac{3\sigma}{2}\rangle$ is the HH bulk Bloch state, $|h_l\rangle$ is the l -th HH envelope with energy E_l^{H} at $\mathbf{k}_{\parallel} = 0$, and the coefficients T_l^x are

$$T_l^x = \langle h_l | \left(\frac{P}{\sqrt{2}\alpha_0} |c\rangle - \sqrt{3}iu_+ |-\rangle \right), \quad (15)$$

$$u_{\pm} = \{\gamma_3, k_z\} \pm \{\kappa, k_z\}. \quad (16)$$

To first order in k , mixing is stronger between η and HH subbands with opposite parity (from k_z terms in T_l^x) and with the same spin component sign (i.e., $|\eta, \sigma\rangle$ couples with $|\frac{3}{2}, \frac{3\sigma}{2}\rangle$). The HH contribution ρ in the mixed subband $|\psi, \mathbf{k}_{\parallel}\rangle$ is given by the sum of the absolute square of each coefficient in front of HH terms. By symmetry, only even powers of k must appear in ρ :

$$\rho = ak_{\parallel}^2 + O(k^4), \quad (17)$$

where a can be found from (14):

$$a = \alpha_0^2 \sum_l \frac{|T_l^x|^2}{(E_{\text{LH1}} - E_l^{\text{H}})^2}. \quad (18)$$

The expression $\rho \approx ak_{\parallel}^2$ is valid for small \mathbf{k}_{\parallel} such that $\rho \ll 1$. In general, ρ lies in the interval $[0, 1]$ with $\rho = 0$ ($\rho = 1$) corresponding to a pure η (HH) subband.

ρ as a function of k_x is displayed in Fig. 4(a) for $w = 6$ nm and $w = 10$ nm at $x = 0.13$. When k_x is small, the parabolic term in (17) fits well the numerically computed ρ . Mixing decreases with increasing energy splitting between LH1 and the HH continuum, as indicated by smaller ρ at the larger well thickness $w = 10$ nm. This remains true for different Sn compositions, as illustrated in Fig. 4(b) where a is plotted as a function of w and x for $\varepsilon_{\text{BR}} = 0$.

IV. DISCUSSION

The preceding results demonstrate that the lattice mismatch between $\text{Ge}_{1-x}\text{Sn}_x$ alloys and Ge provides an additional degree of freedom to engineer the tensile strain required to confine LHs in Ge. According to Fig. 2(a), the region of interest, as defined by the parameters $(x, \varepsilon_{\text{BR}})$, lies in the range where x is above 0.12 and the compressive strain in

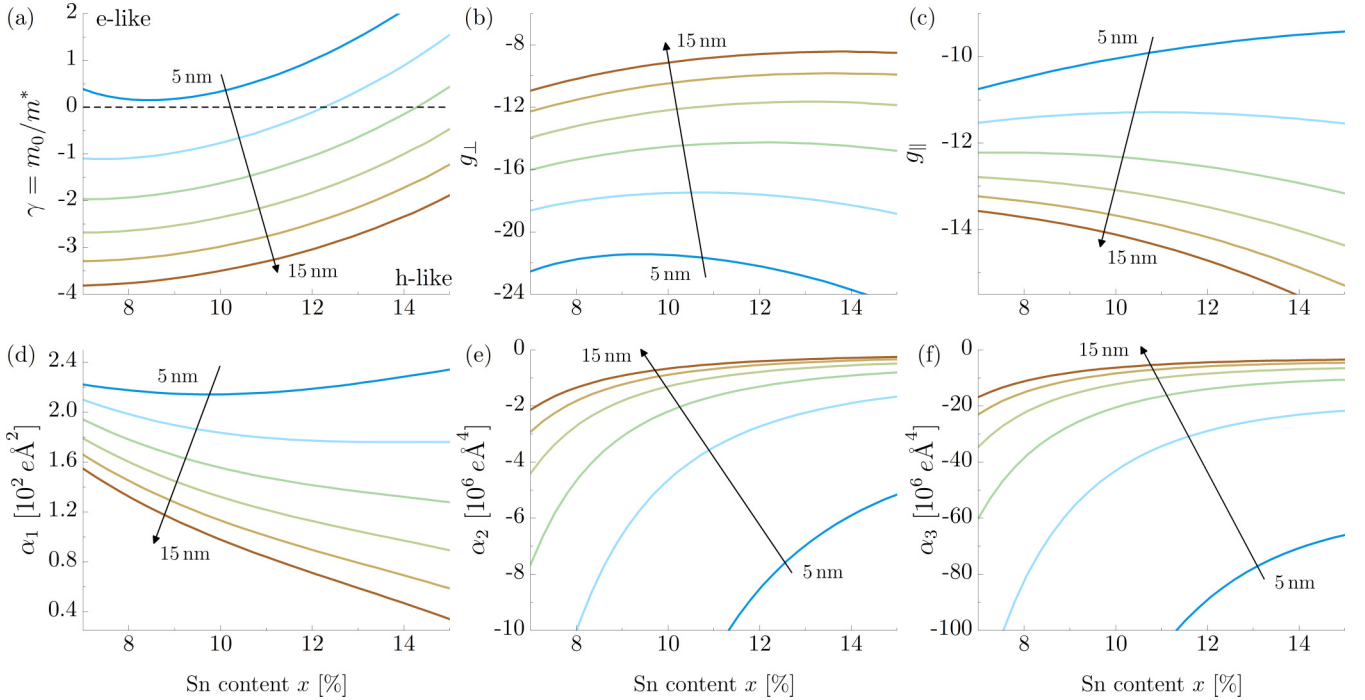


FIG. 3. LHI subband parameters as a function of the well thickness and x for fully relaxed $\text{Ge}_{1-x}\text{Sn}_x$ barriers ($\varepsilon_{\text{BR}} = 0$). (a) Inverse effective mass γ . (b) Out-of-plane g factor. (c) In-plane g factor. (d), (e), and (f) are α_1 , α_2 , and α_3 Rashba parameters, respectively. The calculations were carried out for a well thickness in the 5–15 nm range. The data displayed here are for thicknesses separated by a 2 nm step.

the barriers ε_{BR} is below -0.4% (i.e., $|\varepsilon_{\text{BR}}| \lesssim 0.4\%$). In this range, all band offsets ΔE_i are positive and the Ge layer is of direct bandgap. $\text{Ge}_{1-x}\text{Sn}_x$ layers at Sn content in the proposed range have already been demonstrated experimentally [11]. However, the addition of a highly tensile strained Ge layer on top of strain-relaxed $\text{Ge}_{1-x}\text{Sn}_x$ is still under development. For instance, the authors in Ref. [8] reported a 1.65% tensile-strained Ge quantum well on partially relaxed $\text{Ge}_{0.854}\text{Sn}_{0.146}$ barriers with a residual strain to $\varepsilon_{\text{BR}} \approx -0.54\%$. This system would be located near the crossing between the 0 meV ΔE_1 line and the 50 meV LBO line in Fig. 2(a), very close to the optimal region of interest mentioned earlier. Strain relaxation in the barriers is necessary to enhance confinement in Ge, while relaxing the criterion of minimal well thickness w_0 required for a LH-like valence band edge ($\Delta E_2 > 0$). The ideal amount of strain relaxation for a given barrier Sn content can be estimated from Fig. 2(b). For example, a barrier with $x = 0.14$ does not allow a LH-like valence band edge for $|\varepsilon_{\text{BR}}| > 0.542\%$ compressive strain. The additional requirement $w_0 < h_c$, where h_c is the critical thickness of Ge, further reduces the range of $|\varepsilon_{\text{BR}}|$ to around $<0.5\%$ compressive strain. Reducing the amount of Sn in the barriers relaxes the limit imposed by the critical thickness h_c , but at the cost of a smaller LBO. In contrast, increasing x to 0.18 for instance slightly increases the range for $|\varepsilon_{\text{BR}}|$ to around $<0.525\%$, and increases significantly the LBO and ΔE_1 [Fig. 2(a)] but at the expense of a smaller h_c and a narrower window $w_0 < w < h_c$, but still in the range typically achievable in epitaxial growth experiments.

The in-plane effective mass γ [Fig. 3(a)] shows a strong dependence on both x and w , with small γ expected from the general rule that LHs are heavier in the plane than HHs. There

is also an interesting feature where the dispersion changes from a hole-like ($\gamma < 0$) to an electron-like ($\gamma > 0$) curvature at $\mathbf{k}_{\parallel} = 0$. In the hole-like regime, the valence band edge is formed by a single valley located at $\mathbf{k}_{\parallel} = 0$. In contrast, in the electron-like regime, the valence band edge consists of four valleys, each located a distance k_0^* from $\mathbf{k}_{\parallel} = 0$ along the four equivalent $\langle 110 \rangle$ crystallographic directions in the quantum well plane. For instance, for a 5 nm well at $x = 0.13$ and $\varepsilon_{\text{BR}} = 0$, the valley minima are located at $k_0^* \approx 0.0813 \text{ nm}^{-1}$ away from $\mathbf{k}_{\parallel} = 0$. For larger wavevectors, the dispersion goes away from the bandgap as required, owing to k^4 -terms or higher that are not taken in account by the effective Hamiltonians (7) and (9). According to Fig. 3(a), electron-like subbands occur for small w and large x . This effect takes place for two reasons. The first is when the LH subband anticrosses a neighboring HH subband such that the curvature is inverted at $\mathbf{k}_{\parallel} = 0$. This is typical in systems where the ground state is HH-like and the first LH subband is allowed to mix strongly with the first excited HH subband [20], or when the LH subband is close to a HH continuum (e.g., when w is small). When a LH is far from neighboring HH levels (e.g., when tensile strain is large), mixing decreases, as illustrated in Fig. 4(b), and becomes too weak to invert the curvature. The second reason for a curvature change is related to the anticrossing between the LH and the SO bands in the bulk dispersion of Ge for $k_z > 0$ [Fig. 1(a)]. This anticrossing results in a curvature sign change of the LH band at some point k_z^* such that the bulk energy dispersion $E(k_x, k_y, k_z)$ has a hole-like (electron-like) curvature when k_z is fixed to a value smaller (larger) than k_z^* and $k_{x,y}$ are close to zero. For a quantum well along the z direction, the reciprocal-space envelope functions $\tilde{\psi}(k_z)$ become wider for thinner wells and thus get a larger contribution from

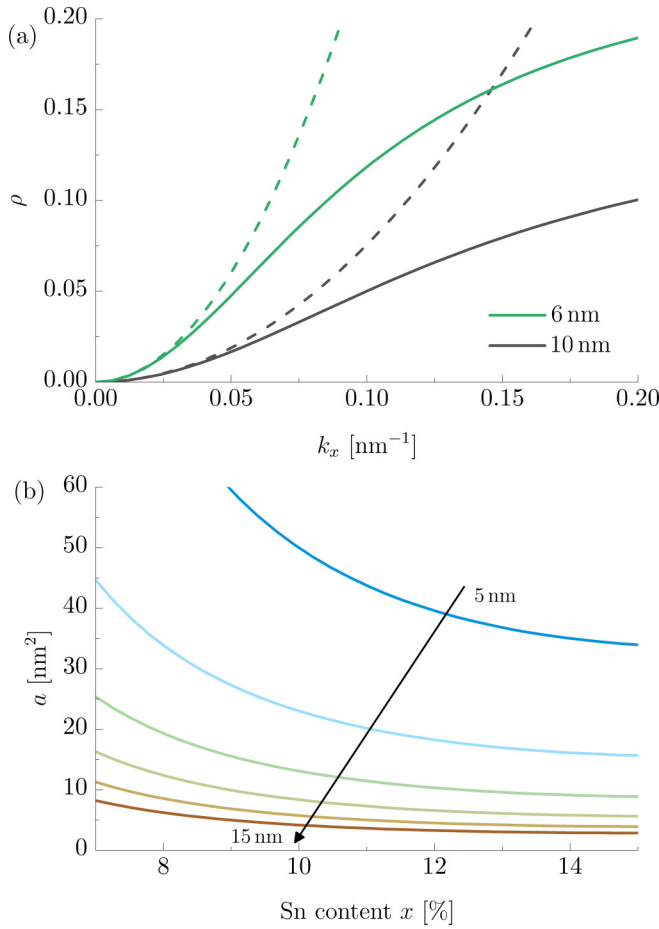


FIG. 4. (a) HH contribution ρ as a function of the wavevector k_x for a quantum well thickness $w = 6$ nm and $w = 10$ nm. Dashed lines correspond to the quadratic approximation $\rho \approx ak_{\parallel}^2$. (b) Coefficient a as a function of w and x . The data displayed here are for thicknesses separated by a 2 nm step.

the electron-like regions in k -space, resulting in a dispersion with inverted curvature at $\mathbf{k}_{\parallel} = 0$.

A major difference with η subbands is the large g_{\parallel} compared to HH systems [6,7,21–23]. A comparison between g_{\parallel} and g_{\perp} reveals an anisotropy for well thicknesses away from ~ 10 nm. Both components have a stronger dependence on w than on x but have opposite behavior with w due to how they couple with neighboring subbands. For the out-of-plane component, $g_{\perp} \sim 2\kappa$ for large w because the coupling with neighboring levels becomes weaker as LH1 gets further from the continuum. In contrast, the in-plane g -factor does not depend on couplings with neighboring HH levels [c.f. (10)] and is instead more influenced by the spatial distribution of the envelopes across the layers. Thus, for large (small) w , $g_{\parallel} \sim 4\kappa$ with κ being that of Ge ($\text{Ge}_{1-x}\text{Sn}_x$).

Another peculiar feature associated with η subbands is the absence of direct connection between γ and g_{\perp} in contrast with HHs (i.e., see Eq. (5) in Ref. [24]). Perturbation theory gives the following for γ and g_{\perp} :

$$\gamma = \Gamma^{\eta} + C + D, \quad (19a)$$

$$\frac{g_{\perp}}{2} = \frac{G^{\eta}}{2} - C + D, \quad (19b)$$

where Γ^{η} , G^{η} are described in Appendix B, whereas C and D are described in Appendix C. For HHs, a similar expansion would give

$$\gamma^{\text{H}} = \Gamma^{\text{H}} + C', \quad (20a)$$

$$\frac{g_{\perp}^{\text{H}}}{2} = \frac{G^{\text{H}}}{2} + C'. \quad (20b)$$

In the latter case, one can combine the equations for γ^{H} and g_{\perp}^{H} to eliminate the C' term, resulting in an expression involving only the mass and the g -factor [24,25]:

$$\frac{g_{\perp}^{\text{H}}}{2} = \frac{G^{\text{H}}}{2} - \Gamma^{\text{H}} + \gamma^{\text{H}}. \quad (21)$$

However, the result is different for η subbands due to the additional D term in (19). The latter is also related to the nonzero β_1 coefficient of η subbands [9].

Rashba parameters follow the general behavior $\alpha_i \rightarrow 0$ as w increases. This is caused by a reduced sensitivity of the wavefunction to electric fields when the level does not spread as extensively into the barriers. Although the quantum well is characterized by relatively small LBOs ($\lesssim 100$ meV) and small out-of-plane LH effective masses, the device operation can comfortably sustain realistic DC electric fields along the growth direction regardless of the well thickness and LBO without inducing any envelope leak into the barriers. In devices where space inversion symmetry needs to be broken, such as in electric dipole spin resonance (EDSR) experiments, this should not be an issue as the relevant Rashba parameter α_3 for EDSR, (which is proportional to $\gamma_2 + \gamma_3$) is one order of magnitude larger than the α_2 Rashba parameter (proportional to $\gamma_2 - \gamma_3$), thus requiring smaller out-of-plane fields [9].

V. CONCLUSIONS

This work demonstrates how $\text{Ge}_{1-x}\text{Sn}_x/\text{Ge}/\text{Ge}_{1-x}\text{Sn}_x$ heterostructures can be tailored to achieve a selective confinement of LHs in Ge while pushing HHs into the $\text{Ge}_{1-x}\text{Sn}_x$ barriers. For a sufficiently large Sn content ($x > 0.12$), small residual compressive strain in the barriers ($|\varepsilon_{\text{BR}}| < 0.4\%$), and a well thickness $w > w_0$, the LH ground state emerges from within the HH continuum, thus yielding a pure LH-like valence band edge ($\Delta E_2 > 0$). This regime also corresponds to a direct bandgap in both the well and its barriers, owing to the large tensile strain in the Ge well and the high Sn content in the barriers. Satisfying the condition of $\Delta E_2 > 0$ imposes a threshold for residual strain in the barriers beyond which a LH-like VB edge becomes virtually impossible.

The in-plane effective mass, the out-of-plane and in-plane g -factor, and the Rashba parameters $\alpha_{1,2,3}$ were computed by explicitly taking in account the spread of the LH envelopes into the barriers and the coupling with the neighboring HH continuum. Small inverse effective masses γ are obtained. A peculiar sign change in γ appearing for small well thicknesses ($w \lesssim 7$ nm) is observed and attributed to the proximity of the LH to the HH continuum (larger LH-HH mixing) and the contribution of the SO band in the LH spinor. An increasingly strong anisotropy in the g -factor components is also observed for well thicknesses away from ~ 10 nm. Most notably, the in-plane component of the g -tensor is significantly larger than what is expected in HH systems. A nonzero linear Rashba

parameter α_1 was obtained, as anticipated for LH systems, with an α_3 coefficient one order of magnitude larger than α_2 .

ACKNOWLEDGMENTS

O.M. acknowledges support from NSERC Canada (Discovery Grant and Alliance Quantum CQS2Q Consortium), Canada Research Chairs, Canada Foundation for Innovation, PRIMA Québec, Defence Canada (Innovation for Defence Excellence and Security, IDEaS), the European Union's Horizon Europe research and innovation program under Grant Agreement No. 101070700 (MIRAQLS), the U.S. Army Research Office Grant No. W911NF-22-1-0277, and the Air Force Office of Scientific and Research Grant No. FA9550-23-1-0763.

APPENDIX A: PARAMETERIZATION OF $\text{Ge}_{1-x}\text{Sn}_x$

The material parameters for the $\text{Ge}_{1-x}\text{Sn}_x$ alloy were calculated in the full composition range by interpolating the parameters from pure Ge and pure Sn:

$$A(x) = (1-x)A^{\text{Ge}} + xA^{\text{Sn}} - x(1-x)b. \quad (\text{A1})$$

Here, $0 \leq x \leq 1$ is the alloy fraction and b is a bowing constant for the parameter A if necessary. For temperature-dependent quantities such as the lattice constant or the bandgaps, we apply (A1) on the temperature-dependent parameters of the alloy's constituents. For instance, the bandgap of $\text{Ge}_{1-x}\text{Sn}_x$ at the Γ point, $E_{g\Gamma}(x, T)$, is given by

$$E_{g\Gamma}(x, T) = (1-x)E_{g\Gamma}^{\text{Ge}}(T) + xE_{g\Gamma}^{\text{Sn}}(T) - x(1-x)b_{\Gamma}, \quad (\text{A2})$$

where

$$E_{g\Gamma}^{\text{Ge}}(T) = E_{g\Gamma}^{0,\text{Ge}} - \frac{\alpha_{\Gamma}^{\text{Ge}} T^2}{\beta_{\Gamma}^{\text{Ge}} + T} \quad (\text{A3})$$

and similar equations for Sn and the L valley. The average VB energy $E_{v,\text{avg}}$ is given by $E_{v,\text{avg}} = E_v - \Delta/3$, where E_v is the VB edge energy and Δ is the spin-orbit splitting [15].

The electron effective mass $m_{c\Gamma}^{\text{L}}$ and the electron g^{L} factor were calculated following the approach in Ref. [31] to make them consistent with the bandgaps and spin-orbit couplings listed in Table I. The Luttinger parameters $\gamma_{1,2,3}^{\text{L}}$ were interpolated between pure Ge and $\text{Ge}_{0.80}\text{Sn}_{0.20}$ using the data from Ref. [40], giving

$$\gamma_i^{\text{L}} = \gamma_i^{\text{L,Ge}} \left(1 - \frac{x}{0.2}\right) + \gamma_i^{\text{L,GeSn}} \left(\frac{x}{0.2}\right) - b_i \frac{x}{0.2} \left(1 - \frac{x}{0.2}\right), \quad (\text{A4})$$

with $\gamma_i^{\text{L,Ge}}$ listed in Table I, $\gamma_1^{\text{L,GeSn}} = 29.2108$, $\gamma_2^{\text{L,GeSn}} = 12.2413$, $\gamma_3^{\text{L,GeSn}} = 13.7387$, and $b_1 = 20.3391$, $b_2 = 9.6609$, $b_3 = 9.8187$.

The Kane momentum matrix element P , which couples the two conduction bands to the six valence bands, is known to sometimes cause spurious solutions, often appearing as levels within the bandgap or with violently oscillating envelopes [41–43]. The approach employed here to eliminate spurious

TABLE I. Input parameters with bowings for the 8-band $k \cdot p$ model.

Parameter	Germanium	Tin	Bowing
Lattice constant			
a_0 (Å, 300 K)	5.657956 ^a	6.489417 ^b	−0.083 ^{c*}
Energy gaps			
$E_{g\Gamma}^0$ (eV)	0.8981 ^b	−0.413 ^b	2.46 ^{d*}
E_{gL}^0 (eV)	0.740 ^o	0.100 ^c	1.23 ^{k*}
α_{Γ} (10^{-4} eV/K)	6.842 ^e	−7.94 ^d	
α_{L} (10^{-4} eV/K)	4.561 ^e		
β_{Γ} (K)	398 ^e	11 ^d	
β_{L} (K)	210 ^e		
Δ (eV)	0.290 ^b	0.770 ^f	−0.100 ^c
$E_{v,\text{avg}}$ (eV)	0	0.69 ^g	
Elastic constants			
c_{11} (GPa)	124 ^b	69.0 ^b	
c_{12} (GPa)	41.3 ^b	29.3 ^b	
c_{44} (GPa)	68.3 ^b	36.2 ^b	
Deformation potentials [†]			
$a_{c\Gamma}$ (eV)	−8.24 ^h	−6.00 ^l	
$a_{c\text{L}}$ (eV)	−1.54 ^h	−2.14 ^m	
a_v (eV)	1.24 ^h	1.58 ^m	
b (eV)	−2.86 ⁱ	−2.7 ⁿ	
Effective mass and spin parameters			
$m_{c\Gamma}^{\text{L}}$ (m_0)	0.0386	−0.057	
γ_1^{L}	13.38 ^j		
γ_2^{L}	4.24 ^j	See Eq. (A4)	
γ_3^{L}	5.69 ^j		
κ^{L}	3.41 ^f	−11.84 ^f	
g^{L}	−2.77	86.6	

[†]The convention $a = a_c - a_v$ is used.

*See Eq. (A2).

^aReference [26].

^bReference [27].

^cReference [28].

^dReference [29].

^eReference [30].

^fReference [31].

^gReference [32].

^hReference [15].

ⁱReference [33].

^jReference [34].

^kReference [35].

^lReference [36].

^mReference [37].

ⁿReference [38].

^oReference [39].

solutions is described in Ref. [41] and consists of rescaling P :

$$P^2 = \frac{3\hbar^2}{2m_{c\Gamma}^{\text{L}}} \left(\frac{2}{E_{g\Gamma}} + \frac{1}{E_{g\Gamma} + \Delta} \right)^{-1}. \quad (\text{A5})$$

The remote band contributions in the Luttinger parameters are then re-adjusted [34] according to $P \equiv \sqrt{\alpha_0 E_p}$ given

in (A5):

$$\gamma_1 = \gamma_1^L - \frac{E_p}{3E_g} \quad (\text{A6})$$

$$\gamma_{2,3} = \gamma_{2,3}^L - \frac{E_p}{6E_g} \quad (\text{A7})$$

$$\kappa = \kappa^L - \frac{E_p}{6E_g} \quad (\text{A8})$$

$$g = g^L + \frac{2E_p}{3E_g} \frac{\Delta}{E_g + \Delta} \quad (\text{A9})$$

$$\frac{m_0}{m_{c\Gamma}} = \frac{m_0}{m_{c\Gamma}^L} - \frac{2E_p}{3E_g} \frac{3E_g/2 + \Delta}{E_g + \Delta} = 0. \quad (\text{A10})$$

APPENDIX B: $k \cdot p$ FRAMEWORK

Our implementation of $k \cdot p$ theory is based on the model presented in Ref. [44], which is an extension of standard $k \cdot p$ frameworks [34,45] for heterostructures with finite energy band offsets at the interfaces (i.e., when a proper ordering between material parameter operators and k_z is critical).

The 8-band $k \cdot p$ Hamiltonian can be written as a sum of different contributions:

$$H = H_k + H_{\text{SO}} + H_\varepsilon + V. \quad (\text{B1})$$

The first term, H_k , depends on the mechanical wavevector $\mathbf{K} = \mathbf{k} + e\mathbf{A}/\hbar$. It automatically includes the Zeeman Hamiltonian through the relation $\mathbf{K} \times \mathbf{K} = e\mathbf{B}/(i\hbar)$. In the Cartesian basis

$$\mathcal{B}_X = \{|S+\rangle, |S-\rangle, |X+\rangle, |Y+\rangle, |Z+\rangle, |X-\rangle, |Y-\rangle, |Z-\rangle\}, \quad (\text{B2})$$

H_k is given by

$$H_k = \begin{bmatrix} H_{cc}^k & 1_{2 \times 2} \otimes H_{cv}^k \\ 1_{2 \times 2} \otimes H_{cv}^{k\dagger} & 1_{2 \times 2} \otimes H_{vv}^k \end{bmatrix} + \begin{bmatrix} H_B & 0 \\ 0 & H_B \otimes 1_{3 \times 3} \end{bmatrix}, \quad (\text{B3})$$

with

$$H_{cc}^k = E_g + \sum_{\alpha} K_{\alpha} A K_{\alpha} + \frac{i\alpha_0}{2} \sum_{\alpha\beta\gamma} \epsilon_{\alpha\beta\gamma} K_{\alpha} (g - g_0) K_{\beta} \sigma_{\gamma}, \quad (\text{B4})$$

$$H_{cv}^k = [iPK_x \quad iPK_y \quad iPK_z], \quad (\text{B5})$$

$$(H_{vv}^k)_{i,j} = \begin{cases} \sum_{\alpha} K_{\alpha} M K_{\alpha} + K_i (L - M) K_i & i = j \\ K_i N_+ K_j + K_j N_- K_i & i \neq j \end{cases}, \quad (\text{B6})$$

$$H_B = \frac{i\alpha_0 g_0}{2} \sum_{\alpha\beta\gamma} \epsilon_{\alpha\beta\gamma} K_{\alpha} K_{\beta} \sigma_{\gamma}, \quad (\text{B7})$$

where $\epsilon_{\alpha\beta\gamma}$ is the Levi-Civita tensor, $A = \alpha_0 m_0 / m_{c\Gamma} = 0$ [see (A10)], and $g_0 \approx 2$ is the free electron g -factor. The term H_B arises from the interaction of the free electron spin with the magnetic field [44].

The second term in (B1), H_{SO} , includes the effects of the SO band and is proportional to the spin-orbit gap Δ :

$$H_{\text{SO}} = \frac{\Delta}{3} \begin{bmatrix} 1 & 0 & 0 & 0 & 0 & 0 & 0 & 0 \\ 0 & 1 & 0 & 0 & 0 & 0 & 0 & 0 \\ 0 & 0 & 0 & -i & 0 & 0 & 0 & 1 \\ 0 & 0 & i & 0 & 0 & 0 & 0 & -i \\ 0 & 0 & 0 & 0 & 0 & -1 & i & 0 \\ 0 & 0 & 0 & 0 & -1 & 0 & i & 0 \\ 0 & 0 & 0 & 0 & -i & -i & 0 & 0 \\ 0 & 0 & 1 & i & 0 & 0 & 0 & 0 \end{bmatrix}. \quad (\text{B8})$$

Strain is incorporated by means of the Bir-Pikus formalism [46], resulting in the term

$$H_\varepsilon = \begin{bmatrix} H_{cc}^\varepsilon & 0 \\ 0 & 1_{2 \times 2} \otimes H_{vv}^\varepsilon \end{bmatrix}, \quad (\text{B9})$$

with

$$H_{cc}^\varepsilon = 1_{2 \times 2} a_c \text{Tr}\{\varepsilon\}, \quad (\text{B10})$$

$$(H_{vv}^\varepsilon)_{i,j} = \begin{cases} m \text{Tr}\{\varepsilon\} + (l - m) \varepsilon_{ii} & i = j \\ n \varepsilon_{ij} & i \neq j \end{cases}. \quad (\text{B11})$$

Finally, the potential energy $V = E_{v,\text{avg}} + eE_z z$. The parameters L , M , N_{\pm} , l , m , and n are related to the usual Luttinger parameters and deformation potentials by

$$\begin{bmatrix} L \\ M \\ N_+ + \alpha_0 \\ N_- - \alpha_0 \end{bmatrix} = -\alpha_0 \begin{bmatrix} 1 & 4 & 0 & 0 \\ 1 & -2 & 0 & 0 \\ 0 & 0 & 3 & 3 \\ 0 & 0 & 3 & -3 \end{bmatrix} \begin{bmatrix} \gamma_1 \\ \gamma_2 \\ \gamma_3 \\ \kappa \end{bmatrix}, \quad (\text{B12})$$

$$\begin{bmatrix} l \\ m \\ n \end{bmatrix} = \begin{bmatrix} 1 & 2 & 0 \\ 1 & -1 & 0 \\ 0 & 0 & \sqrt{3} \end{bmatrix} \begin{bmatrix} a_v \\ b \\ d \end{bmatrix}. \quad (\text{B13})$$

A change of basis from \mathcal{B}_X to the so-called angular momentum basis \mathcal{B}_J brings the Hamiltonian H in a 2×2 block diagonal matrix (each block being 4×4) when evaluated with $B = 0$ and $k_x = k_y = 0$:

$$H_0 \equiv U_{X \leftarrow J}^\dagger H(k_x = k_y = 0, B = 0) U_{X \leftarrow J} = \begin{bmatrix} H_+ & 0 \\ 0 & H_- \end{bmatrix},$$

where

$$\mathcal{B}_J = \left\{ \left| \frac{1}{2}, \frac{1}{2} \right\rangle_c, \left| \frac{3}{2}, \frac{1}{2} \right\rangle_c, \left| \frac{1}{2}, \frac{1}{2} \right\rangle_c, \left| \frac{3}{2}, \frac{3}{2} \right\rangle_c, \right. \\ \left. \left| \frac{1}{2}, -\frac{1}{2} \right\rangle_c, \left| \frac{3}{2}, -\frac{1}{2} \right\rangle_c, \left| \frac{1}{2}, -\frac{1}{2} \right\rangle_c, \left| \frac{3}{2}, -\frac{3}{2} \right\rangle_c \right\}, \quad (\text{B14})$$

$$U_{X \leftarrow J} = \begin{bmatrix} 1 & 0 & 0 & 0 & 0 & 0 & 0 & 0 \\ 0 & 0 & 0 & 0 & 1 & 0 & 0 & 0 \\ 0 & 0 & 0 & -s_2 & 0 & s_6 & -s_3 & 0 \\ 0 & 0 & 0 & -is_2 & 0 & -is_6 & is_3 & 0 \\ 0 & s_{23} & -s_3 & 0 & 0 & 0 & 0 & 0 \\ 0 & -s_6 & -s_3 & 0 & 0 & 0 & 0 & s_2 \\ 0 & -is_6 & -is_3 & 0 & 0 & 0 & 0 & -is_2 \\ 0 & 0 & 0 & 0 & 0 & s_{23} & s_3 & 0 \end{bmatrix}, \quad (\text{B15})$$

$$\begin{aligned}
 H_\sigma = V + \frac{\Delta}{3} + & \begin{bmatrix} E_g \Gamma & 0 & 0 & 0 \\ \dagger & 0 & 0 & 0 \\ & & -\Delta & 0 \\ & & & 0 \end{bmatrix} \\
 + & \begin{bmatrix} k_z A k_z & i\sqrt{2/3} P k_z & -i\sigma P k_z / \sqrt{3} & 0 \\ \dagger & -\alpha_0 k_z \gamma_+ k_z & \sqrt{8\sigma} \alpha_0 k_z \gamma_2 k_z & 0 \\ & & -\alpha_0 k_z \gamma_1 k_z & 0 \\ & & & -\alpha_0 k_z \gamma_- k_z \end{bmatrix} \\
 + \text{Tr}\{\varepsilon\} & \begin{bmatrix} a_c & 0 & 0 & 0 \\ \dagger & a_v & 0 & 0 \\ & & a_v & 0 \\ & & & a_v \end{bmatrix} \\
 + b\delta\varepsilon & \begin{bmatrix} 0 & 0 & 0 & 0 \\ \dagger & -1 & \sqrt{2}\sigma & 0 \\ & & 0 & 0 \\ & & & 1 \end{bmatrix}, \quad (\text{B16})
 \end{aligned}$$

with $s_2 = 1/\sqrt{2}$, $s_3 = 1/\sqrt{3}$, $s_6 = 1/\sqrt{6}$, $s_{23} = \sqrt{2/3}$, $\sigma = \pm 1$ the pseudo-spin index, $\gamma_\pm = \gamma_1 \pm 2\gamma_2$, and $\delta\varepsilon = \varepsilon_{xx} - \varepsilon_{zz}$ (all shear components in the strain tensor vanish). Since H_+ and H_- differ only by a minus sign in the LH-SO and the CB-SO coupling elements, both share the same energy spectrum (Kramers' degeneracy) and the eigenstates of H_- are the time-reversed eigenstates of H_+ . Additionally, H_σ is itself block diagonal: one 3×3 block representing a CB-LH-SO superposition (or a η level) and one 1×1 block representing a pure HH level (or H for short). Eigenstates of H_σ are thus either of type η or H (with pseudo-spin σ):

$$|\eta, \sigma\rangle = \left| \frac{1}{2}, \frac{\sigma}{2} \right\rangle_c |c\rangle + \left| \frac{3}{2}, \frac{\sigma}{2} \right\rangle_\ell |\ell\rangle + \sigma \left| \frac{1}{2}, \frac{\sigma}{2} \right\rangle_s |s\rangle, \quad (\text{B17a})$$

$$|\text{H}, \sigma\rangle = \left| \frac{3}{2}, \frac{3\sigma}{2} \right\rangle |h\rangle. \quad (\text{B17b})$$

The energies and eigenstates of H_+ are computed for each set of quantum well parameters (x , ε_{BR} and w) with the substitution $k_z \rightarrow -id_z$, without any assumptions on the shape of the envelopes. We choose a grid spacing of 0.01 nm for the finite differences and keep the 200 subbands that are the closest to LH1 ($N = N_\eta + N_{\text{H}} = 200$). We found that $N = 200$ is large enough for the effective parameters in (7) and (9) to converge.

We diagonalize H away from $k_x = k_y = 0$ and $B = 0$ by first projecting H onto the orthonormal basis (B17). This brings H to a 4×4 block-matrix form (with total dimension $2N \times 2N$), where each block consists of all the subbands of one aforementioned type (H or η , spin up/down) and couplings thereof. Taking the basis ordering $\mathcal{B}_0 = \{|H+\rangle, |\eta+\rangle, |\eta-\rangle, |H-\rangle\}$ (and bold characters to emphasize that we are in basis \mathcal{B}_0), the Hamiltonian H when \mathbf{B} is perpendicular to the plane is given by

$$\begin{aligned}
 \mathbf{H} = \mathbf{E}_0 + \alpha_0 & \left[\mathbf{M}_\gamma K_\parallel^2 + \frac{1}{2\lambda^2} \mathbf{M}_g \right. \\
 & \left. + (i\mathbf{M}_1 K_- + \mathbf{M}_2 K_-^2 + \text{h.c.}) \right], \quad (\text{B18})
 \end{aligned}$$

where $\mathbf{E}_0 = \text{diag}\{\mathbf{E}^{\text{H}}, \mathbf{E}^\eta, \mathbf{E}^\eta, \mathbf{E}^{\text{H}}\}$ are the energies associated to H_0 ($\mathbf{E}^{\text{H},\eta}$ are also diagonal) and

$$\mathbf{M}_\gamma = \text{diag}\{\mathbf{\Gamma}^{\text{H}}, \mathbf{\Gamma}^\eta, \mathbf{\Gamma}^\eta, \mathbf{\Gamma}^{\text{H}}\}, \quad (\text{B19a})$$

$$\mathbf{M}_g = \text{diag}\{\mathbf{G}^{\text{H}}, \mathbf{G}^\eta, -\mathbf{G}^\eta, -\mathbf{G}^{\text{H}}\}, \quad (\text{B19b})$$

$$\mathbf{M}_1 = \begin{bmatrix} 0 & \mathbf{T}^x & 0 & 0 \\ 0 & 0 & \mathbf{T}^\eta & 0 \\ 0 & 0 & 0 & \mathbf{T}^{x\dagger} \\ 0 & 0 & 0 & 0 \end{bmatrix}, \quad (\text{B19c})$$

$$\mathbf{M}_2 = \begin{bmatrix} 0 & 0 & \mu & 0 \\ 0 & 0 & 0 & \mu^\dagger \\ \delta^\dagger & 0 & 0 & 0 \\ 0 & \delta & 0 & 0 \end{bmatrix}, \quad (\text{B19d})$$

with (assuming $A = 0$ and $g_0 = 2$):

$$\Gamma_{l,l'}^{\text{H}} = -\langle h_l | \gamma_1 + \gamma_2 | h_{l'} \rangle, \quad (\text{B20a})$$

$$G_{l,l'}^{\text{H}} = -\langle h_l | 6\kappa | h_{l'} \rangle, \quad (\text{B20b})$$

$$\Gamma_{j,j'}^\eta = -\frac{1}{3} \langle +j | \gamma_1 + \gamma_2 | +j' \rangle - \frac{2}{3} \langle -j | \gamma_1 - 2\gamma_2 | -j' \rangle, \quad (\text{B20c})$$

$$\begin{aligned}
 G_{j,j'}^\eta &= \langle c_j | g | c_{j'} \rangle - 2 \langle +j | \kappa | +j' \rangle \\
 &\quad - \frac{4}{3} (\langle +j | +j' \rangle - \langle -j | -j' \rangle), \quad (\text{B20d})
 \end{aligned}$$

$$\begin{aligned}
 T_{j,j'}^\eta &= \frac{1}{\sqrt{6}\alpha_0} (\langle c_j | P | +j' \rangle + \langle +j | P | c_{j'} \rangle) \\
 &\quad + i \langle c_j | [g/2, k_z] | c_{j'} \rangle \\
 &\quad - i (\langle +j | u_+ | -j' \rangle - \langle -j | u_- | +j' \rangle), \quad (\text{B20e})
 \end{aligned}$$

$$T_{l,j}^x = \langle h_l | \left(\frac{P}{\sqrt{2}\alpha_0} |c_j\rangle - \sqrt{3} i u_+ | -j \rangle \right), \quad (\text{B20f})$$

$$\mu_{l,j} = \frac{\sqrt{3}}{2} \langle h_l | \gamma_2 + \gamma_3 | +j \rangle, \quad (\text{B20g})$$

$$\delta_{l,j} = \frac{\sqrt{3}}{2} \langle h_l | \gamma_2 - \gamma_3 | +j \rangle. \quad (\text{B20h})$$

Here, $T_{l,j}^x$ really is the same as in (15) but with the explicit dependence on the η subband index j . When \mathbf{B} is in-plane, the Hamiltonian H is given by

$$\begin{aligned}
 \mathbf{H} = \mathbf{E}_0 + \alpha_0 & \left\{ \mathbf{M}_\gamma k_\parallel^2 + \frac{1}{\lambda^4} \mathbf{M}'_\gamma \right. \\
 & \left. + \left[i \left(\mathbf{M}_1 + \frac{2}{\lambda^2} \mathbf{M}'_2 e^{-i\phi} - \frac{1}{\lambda^2} \mathbf{M}'_\gamma e^{i\phi} \right) k_- \right. \right. \\
 & \left. \left. + \mathbf{M}_2 k_-^2 - \frac{1}{\lambda^2} \mathbf{M}'_1 e^{-i\phi} - \frac{e^{-2i\phi}}{\lambda^4} \mathbf{M}'_2 + \text{h.c.} \right] \right\}, \quad (\text{B21})
 \end{aligned}$$

where the \mathbf{M}_i with primes are defined similarly to those without primes [c.f. (B19)] but with

$$\Gamma_{l,l'}^{p\text{H}} = -\langle h_l | z^p (\gamma_1 + \gamma_2) | h_{l'} \rangle, \quad (\text{B22a})$$

$$\begin{aligned}
 \Gamma_{j,j'}^{p\eta} &= -\frac{1}{3} \langle +j | z^p (\gamma_1 + \gamma_2) | +j' \rangle \\
 &\quad - \frac{2}{3} \langle -j | z^p (\gamma_1 - 2\gamma_2) | -j' \rangle, \quad (\text{B22b})
 \end{aligned}$$

$$\begin{aligned}
T_{j,j}^{\eta} &= \frac{1}{\sqrt{6}\alpha_0} (\langle c_j | z^P | +_j \rangle + \langle +_j | z^P | c_j \rangle) \\
&+ i \langle c_j | [zg/2, k_z] | c_j \rangle \\
&- i (\langle +_j | u'_+ | -_j \rangle - \langle -_j | u'_- | +_j \rangle) \\
&+ \frac{1}{\sqrt{2}} [\langle s_j | -_j \rangle + \langle -_j | s_j \rangle], \tag{B22c}
\end{aligned}$$

$$T_{l,j}^x = \langle h_l | \left[\frac{z^P}{\sqrt{2}\alpha_0} | c_j \rangle - \sqrt{3} \left(i u'_+ | -_j \rangle + \frac{1}{\sqrt{2}} | s_j \rangle \right) \right], \tag{B22d}$$

$$\mu_{l,j}^p = \frac{\sqrt{3}}{2} \langle h_l | z^p (\gamma_2 + \gamma_3) | +_j \rangle, \tag{B22e}$$

$$\delta_{l,j}^p = \frac{\sqrt{3}}{2} \langle h_l | z^p (\gamma_2 - \gamma_3) | +_j \rangle, \tag{B22f}$$

where $p = 1$ corresponds to one prime and $p = 2$ corresponds to two primes. We point out that in basis \mathcal{B}_0 , the strain components and the SO energy Δ do not appear explicitly in \mathbf{H} , since they are already taken in account by the energies $\mathbf{E}^{\text{H},\eta}$ and the envelope functions associated with H_0 . This is because H does not contain any terms such as $\varepsilon_{ij}k_j$ or $\Delta \cdot k_j$.

APPENDIX C: PERTURBATIVE EXPANSION OF γ AND g_{\perp}

A perturbative expansion of the $k \cdot p$ Hamiltonian for small \mathbf{K}_{\parallel} yields explicit formulas for the effective parameters

appearing in (7) and (9). This is obtained by means of a Schrieffer-Wolff transformation (SWT) [34]. To this end, the basis \mathcal{B}_0 is convenient since \mathbf{H} is exactly diagonal when \mathbf{K}_{\parallel} and B are zero (see Appendix B for the notation). The zero-th order terms in the SWT are directly the energies \mathbf{E}_0 . The first-order terms are given by the diagonal elements in the nonzero blocks of the matrices \mathbf{M}_i . For instance, the first-order contribution to the effective mass of η subbands are the diagonal entries of $\mathbf{\Gamma}^{\eta}$. Similarly, the Rashba parameter $\beta_1 = \alpha_0 T_{j,j}^{\eta}$. The in-plane g factor stems from the \mathbf{M}'_1 term in (B21), which has a nonvanishing block in the $(\eta+, \eta-)$ subspace, thus yielding $g_{\parallel} = -2T_{j,j}^{\eta}$ [c.f. (10)]. Besides β_1 and g_{\parallel} , the effective parameters in (7) and (9) require a second- or third-order SWT to be described exactly. In particular, γ and g_{\perp} are exactly described by second-order perturbation only. For the j -th η subband, the terms C and D in (19) correspond to the second-order corrections, and are given by

$$C = \alpha_0 \sum_l \frac{|T_{l,j}^x|^2}{E_j^{\eta} - E_l^{\text{H}}}, \tag{C1}$$

$$D = \alpha_0 \sum_{j' \neq j} \frac{|T_{j,j'}^{\eta}|^2}{E_j^{\eta} - E_{j'}^{\eta}}. \tag{C2}$$

For the l -th HH subband, the C' term in (20) is

$$C' = \alpha_0 \sum_j \frac{|T_{l,j}^x|^2}{E_l^{\text{H}} - E_j^{\eta}}. \tag{C3}$$

-
- [1] Y. Fang, P. Philippopoulos, D. Culcer, W. A. Coish, and S. Chesi, *Mater. Quantum Technol.* **3**, 012003 (2023).
- [2] W. I. L. Lawrie, M. Rimbach-Russ, F. van Riggelen, N. W. Hendrickx, S. L. de Snoo, A. Sammak, G. Scappucci, J. Helsen, and M. Veldhorst, *Nat. Commun.* **14**, 3617 (2023).
- [3] D. Jirovec, A. Hofmann, A. Ballabio, P. M. Mutter, G. Tavani, M. Botifoll, A. Crippa, J. Kukučka, O. Sagi, F. Martins, J. Saez-Mollejo, I. Prieto, M. Borovkov, J. Arbiol, D. Chrastina, G. Isella, and G. Katsaros, *Nat. Mater.* **20**, 1106 (2021).
- [4] N. W. Hendrickx, W. I. L. Lawrie, M. Russ, F. van Riggelen, S. L. de Snoo, R. N. Schouten, A. Sammak, G. Scappucci, and M. Veldhorst, *Nature (London)* **591**, 580 (2021).
- [5] N. W. Hendrickx, W. I. L. Lawrie, L. Petit, A. Sammak, G. Scappucci, and M. Veldhorst, *Nat. Commun.* **11**, 3478 (2020).
- [6] N. W. Hendrickx, D. P. Franke, A. Sammak, G. Scappucci, and M. Veldhorst, *Nature (London)* **577**, 487 (2020).
- [7] W. I. L. Lawrie, N. W. Hendrickx, F. van Riggelen, M. Russ, L. Petit, A. Sammak, G. Scappucci, and M. Veldhorst, *Nano Lett.* **20**, 7237 (2020).
- [8] S. Assali, A. Attiaoui, P. Del Vecchio, S. Mukherjee, J. Nicolas, and O. Moutanabbir, *Adv. Mater.* **34**, 2201192 (2022).
- [9] P. Del Vecchio and O. Moutanabbir, *Phys. Rev. B* **107**, L161406 (2023).
- [10] A. G. Moghaddam, T. Kernreiter, M. Governale, and U. Zülicke, *Phys. Rev. B* **89**, 184507 (2014).
- [11] O. Moutanabbir, S. Assali, X. Gong, E. O'Reilly, C. A. Broderick, B. Marzban, J. Witzens, W. Du, S.-Q. Yu, A. Chelnokov, D. Buca, and D. Nam, *Appl. Phys. Lett.* **118**, 110502 (2021).
- [12] C.-T. Tai, P.-Y. Chiu, C.-Y. Liu, H.-S. Kao, C. T. Harris, T.-M. Lu, C.-T. Hsieh, S.-W. Chang, and J.-Y. Li, *Adv. Mater.* **33**, 2007862 (2021).
- [13] B. M. Ferrari, F. Marcantonio, F. Murphy-Armando, M. Virgilio, and F. Pezzoli, *Phys. Rev. Res.* **5**, L022035 (2023).
- [14] G. Fettu, J. E. Sipe, and O. Moutanabbir, *Phys. Rev. B* **107**, 165202 (2023).
- [15] C. G. Van de Walle, *Phys. Rev. B* **39**, 1871 (1989).
- [16] R. People and J. C. Bean, *Appl. Phys. Lett.* **47**, 322 (1985).
- [17] R. People and J. C. Bean, *Appl. Phys. Lett.* **49**, 229 (1986).
- [18] Z. Wang, E. Marcellina, A. R. Hamilton, J. H. Cullen, S. Rogge, J. Salfi, and D. Culcer, *npj Quantum Inf.* **7**, 54 (2021).
- [19] C.-A. Wang, G. Scappucci, M. Veldhorst, and M. Russ, *arXiv:2208.04795* [cond-mat.mes-hall].
- [20] R. K. Hayden, D. K. Maude, L. Eaves, E. C. Valadares, M. Henini, F. W. Sheard, O. H. Hughes, J. C. Portal, and L. Cury, *Phys. Rev. Lett.* **66**, 1749 (1991).
- [21] D. Jirovec, P. M. Mutter, A. Hofmann, A. Crippa, M. Rychetsky, D. L. Craig, J. Kukučka, F. Martins, A. Ballabio, N. Ares, D. Chrastina, G. Isella, G. Burkard, and G. Katsaros, *Phys. Rev. Lett.* **128**, 126803 (2022).
- [22] H. Watzinger, J. Kukučka, L. Vukušić, F. Gao, T. Wang, F. Schäffler, J.-J. Zhang, and G. Katsaros, *Nat. Commun.* **9**, 3902 (2018).

- [23] N. W. Hendrickx, D. P. Franke, A. Sammak, M. Kouwenhoven, D. Sabbagh, L. Yeoh, R. Li, M. L. V. Tagliaferri, M. Virgilio, G. Capellini, G. Scappucci, and M. Veldhorst, *Nat. Commun.* **9**, 2835 (2018).
- [24] I. L. Drichko, A. A. Dmitriev, V. A. Malysh, I. Y. Smirnov, H. von Känel, M. Kummer, D. Chrastina, and G. Isella, *J. Appl. Phys.* **123**, 165703 (2018).
- [25] T. Wimbauer, K. Oettinger, A. L. Efros, B. K. Meyer, and H. Brugger, *Phys. Rev. B* **50**, 8889 (1994).
- [26] R. R. Reeber and K. Wang, *Mater. Chem. Phys.* **46**, 259 (1996).
- [27] *Semiconductors, Group IV Elements and III-V Compounds*, edited by O. Madelung (Springer-Verlag, Berlin, 1991).
- [28] M. P. Polak, P. Scharoch, and R. Kudrawiec, *J. Phys. D* **50**, 195103 (2017).
- [29] M. Bertrand, Q.-M. Thai, J. Chrétien, N. Pauc, J. Aubin, L. Milord, A. Gassenq, J.-M. Hartmann, A. Chelnokov, V. Calvo, and V. Reboud, *Ann. Phys.* **531**, 1800396 (2019).
- [30] Y. Varshni, *Physica* **34**, 149 (1967).
- [31] P. Lawaetz, *Phys. Rev. B* **4**, 3460 (1971).
- [32] J. Menéndez and J. Kouvetakis, *Appl. Phys. Lett.* **85**, 1175 (2004).
- [33] C. G. Van de Walle and R. M. Martin, *Phys. Rev. B* **34**, 5621 (1986).
- [34] R. Winkler, *Spin-Orbit Coupling Effects in Two-dimensional Electron and Hole Systems*, Springer Tracts in Modern Physics (Springer, Berlin, 2003).
- [35] V. R. D'Costa, C. S. Cook, A. G. Birdwell, C. L. Littler, M. Canonico, S. Zollner, J. Kouvetakis, and J. Menéndez, *Phys. Rev. B* **73**, 125207 (2006).
- [36] Y.-H. Li, X. G. Gong, and S.-H. Wei, *Phys. Rev. B* **73**, 245206 (2006).
- [37] T. Brudevoll, D. S. Citrin, M. Cardona, and N. E. Christensen, *Phys. Rev. B* **48**, 8629 (1993).
- [38] M. Willatzen, L. C. Lew Yan Voon, P. V. Santos, M. Cardona, D. Munzar, and N. E. Christensen, *Phys. Rev. B* **52**, 5070 (1995).
- [39] J. Weber and M. I. Alonso, *Phys. Rev. B* **40**, 5683 (1989).
- [40] K. Lu Low, Y. Yang, G. Han, W. Fan, and Y.-C. Yeo, *J. Appl. Phys.* **112**, 103715 (2012).
- [41] B. A. Foreman, *Phys. Rev. B* **56**, R12748 (1997).
- [42] R. G. Veprek, S. Steiger, and B. Witzigmann, *Phys. Rev. B* **76**, 165320 (2007).
- [43] T. Eissfeller and P. Vogl, *Phys. Rev. B* **84**, 195122 (2011).
- [44] T. Eissfeller, Ph.D. thesis, Technische Universität München, 2001.
- [45] L. Voon and M. Willatzen, *The $k \cdot p$ Method: Electronic Properties of Semiconductors* (Springer, Berlin, 2009).
- [46] G. Bir and G. Pikus, *Symmetry and Strain-induced Effects in Semiconductors* (Wiley, New York, 1974).