# Atomic-scale perspective on individual thiol-terminated molecules anchored to single S vacancies in MoS<sub>2</sub>

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Sulfur vacancies in  $MoS_2$  on Au(111) have been shown to be negatively charged as reflected by a Kondo resonance. Here, we use scanning tunneling microscopy to show that these vacancies serve as anchoring sites for thiol-based molecules (CF<sub>3</sub>-3P-SH) with two distinct reaction products, one of them showing a Kondo resonance. Based on comparisons with density-functional theory (DFT) calculations, including a random structure search and computation of energies and electronic properties at a hybrid exchange-correlation functional level, we conclude that both anchored molecules are charge neutral. We propose that one of them is an anchored intact CF<sub>3</sub>-3P-SH molecule while the other one is the result of catalytically activated dehydrogenation to CF<sub>3</sub>-3P-S with subsequent anchoring. Our investigations highlight a perspective of functionalizing defects with thiol-terminated molecules that can be equipped with additional functional groups, such as charge donor or acceptor moieties, switching units, or magnetic centers.

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## I. INTRODUCTION

Transition-metal dichalcogenides (TMDCs) are a class of materials with unwavering popularity for the last 50 years [1]. The recent interest in these materials originates from the possibilities of isolating single monolayers and stacking them in almost all imaginable ways, i. e., combining different materials, combining stacking orders, and rotations of the layers [2]. A single layer is distinct from its bulk counterpart in that the electronic band gap is changed due to confinement effects, e. g., transforming an indirect band gap to a direct one in MoS<sub>2</sub> [3,4].

The tunability of electronic properties through variable stacking allows for versatile applications in optoelectronics [5], in field effect transistors [6], in nanoelectromechanical systems [7], and in spintronics [8]. The two-dimensional (2D) nature of TMDCs does not only offer unprecedented flexibility, but also comes along with opportunities for device miniaturization [9,10].

Owing to the 2D nature, a small amount of defects already has a major effect on the electronic properties of the material.

Often they introduce localized states inside the band gap of a semiconducting monolayer, which may be detrimental for certain applications, but also beneficial for others [11–15]. Attaining control over defects opens one more channel for tuning the local and global electronic properties of TMDCs, an area usually coined as defect engineering. The tunability can be enhanced even further by the functionalization of defects with molecular adsorbates [16–18]. This functionalization can lead to further possible charge transfer processes [19] and doping [20]. The hybrid systems may act as biosensors [21,22], and offer specific reaction sites [23].

Despite these fascinating opportunities, an atomistic understanding of the molecular functionalization in terms of the precise molecular configuration, the character of their bonding to the TMDC substrate and the resulting local electronic structure is largely in its infancy. In this paper, we study a molecular model system consisting of a thiol-terminated organic molecule and investigate its bonding properties to S vacancies in  $MoS_2$ . By combining scanning tunneling microscopy and theoretical modeling we gain fundamental insights into the atomic-scale bonding and electronic properties. We surmise that the understanding of this simple model system will form a basis for the design of future hybrid systems with more specific functions.

In more detail, we create S vacancies in the terminating S layer of  $MoS_2$  on Au(111). The interaction of  $MoS_2$  with the Au substrate has been shown to create negatively charged vacancies that localize a single electron at low temperatures [24]. Nevertheless, an analysis of defect formation energies suggests that a fraction of neutral vacancies can be present at higher temperatures [25,26]. In both scenarios, these sites

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FIG. 1. Overview of STM topographies and illustrative sketches (not to scale) of the preparation steps: (a)  $MoS_2$  on Au(111), showing the characteristic moiré pattern of the  $MoS_2$  island (set point: 2 V, 200 pA). (b) Top-layer sulfur vacancies in  $MoS_2/Au(111)$ , visible as triangular shapes, created by sputtering with Ne ions (set point: 50 mV, 200 pA). (c)  $CF_3$ -3P-S(H) on  $MoS_2/Au(111)$  after evaporation and subsequent annealing. Most  $CF_3$ -3P-S(H) molecules cluster together on the Au(111) surface, with only a few remaining on the  $MoS_2$  island (set point: 2 V, 30 pA). We mark the tilting angle  $\theta$  that defines the inclination of the molecule with respect to the surface normal on the inset. (d) Molecular model of  $CF_3$ -3P-SH in gas phase (gray: carbon; white: hydrogen; blue: fluorine; yellow: sulfur).

are expected to be highly reactive and ideal candidates for anchoring molecules. We expose these defects to 4"-(trifluoromethyl)-[1,1':4', 1"-terphenyl]-4-thiol (CF<sub>3</sub>-3P-SH) molecules [Fig. 1(d)]. Thiol-based molecules are prone to covalently bind to S vacancies [20,27,28] and passivate them. In line with these expectations, we observe an anchoring of CF<sub>3</sub>-3P-SH molecules to S defects on MoS<sub>2</sub>/Au(111) and track the resulting changes in electronic structure in tunneling spectra.

We rationalize the resulting atomic and electronic structure by density-functional theory simulations. We find robust hybrid molecule-substrate localized electronic states and discuss the possibility of vacancies inducing a dehydrogenation of the molecule to  $CF_3$ -3P-S. We suggest that both the charged and neutral state of the S vacancy may play a role in the anchoring mechanism. However, regardless of the initial charge state of the vacancy, and whether dehydrogenation happens or not, a neutral anchored molecule is found to be the most stable state. This observation is compatible with two distinct experimentally observed molecular states, one of them exhibiting a Kondo resonance.

#### **II. EXPERIMENTAL AND THEORETICAL DETAILS**

The Au(111) surface was prepared for the MoS<sub>2</sub> growth by repeated cleaning cycles of sputtering and annealing under ultrahigh vacuum (UHV) conditions. A monolayer of MoS<sub>2</sub> was then grown on the atomically clean Au(111) substrate by depositing Mo atoms in an H<sub>2</sub>S-atmosphere of  $1 \times 10^{-5}$  mbar onto the sample kept at 820 K [29,30]. The as-prepared sample was inspected in the STM to assure the successful formation of monolayer-islands of MoS<sub>2</sub> of about 100 nm in diameter. To remove any further adsorbates from the sample, we annealed it to 570 K. Sulfur vacancies were then introduced by sputtering with Ne ions (~550 eV) at room temperature at a 45° angle for a few seconds. After confirming the presence of an appropriate density of S vacancies by STM (about 3–5 S vacancies in an area of  $8 \times 10 \text{ nm}^2$ ), CF<sub>3</sub>-3P-SH molecules were evaporated from a Knudsen cell at an evaporation temperature of 365 K onto the MoS<sub>2</sub>/Au(111) sample cooled to 160 K with subsequent annealing to 220 K. For details on the synthesis of the CF<sub>3</sub>-3P-SH molecules, see Supplemental Material, Sec. SI [31]. All STM measurements were performed at 4.5 K under UHV conditions and dI/dV spectra were recorded using a standard lock-in technique.

For computational analysis, we conducted an exhaustive ab initio random structure search with the GENSEC package [32,33]. For molecular placement and distortion, random values were assigned to the internal torsional angles connecting the benzene rings of the molecule and to the orientation of the molecule with respect to the MoS<sub>2</sub> substrate. We constrained the search space by requiring that at least one atom of the molecule be placed at a distance of 3 Å from the defect site. A  $6 \times 6$  unit cell containing one sulfur vacancy was assumed and, in the first step, no doping of the substrate was considered. Geometry relaxations were performed only on structures that met the criteria of no steric clashes according to a database of scaled van der Waals radii. The initial relaxations were performed with the PBE functional and MBD-NL van der Waals corrections [34], light settings in FHI aims [35], and a  $1 \times 1 \times 1$  k grid. We conducted searches with the 3P-S molecule, the CF<sub>3</sub>-3P-SH molecule and its dehydrogenated derivative, the CF<sub>3</sub>-3P-S molecule. We optimized 35 structures of 3P-S, 45 structures of CF<sub>3</sub>-3P-SH, and 75 structures of CF<sub>3</sub>-3P-S in each search.

To increase the conformational diversity, we added  $CF_3$  groups and H atoms to transform 3P-S molecules into  $CF_3$ -3P-S and  $CF_3$ -3P-SH and postrelaxed them. More details about the structure search can be found in the Supplemental Material [31], Sec. SII. We ended up with one candidate structure of  $CF_3$ -3P-SH and five candidate structures of  $CF_3$ -3P-S on  $MoS_2$  that anchored on the defect. This subset of six

structures was further optimized with tight settings and a  $4 \times 4 \times 1$  k grid, and then postoptimized with the HSE06 [36] functional with MBD-NL van der Waals corrections and intermediate settings of FHI aims [37]. The HSE06+MBD-NL functional relaxations were done on a  $2 \times 2 \times 1$  k-point grid and we only required the forces to be below  $10^{-2}$  eV/Å. Electronic structure properties and further analysis were all conducted at the HSE06 level of theory, including a post-processing spin-orbit coupling correction [38]. Some of these simulations were done with tight settings.

The same set of six conformers was reoptimized including a negative charge in the substrate, to model a negatively charged vacancy with charge -1. This was achieved by the use of a virtual-crystal approximation (VCA) scheme, as explained in Ref. [25]. We carried out geometry optimizations with the HSE06 functional with MBD-NL van der Waals corrections and intermediate settings, including the extra charge. Subsequently, electronic structure properties were analyzed at this level of theory, including the spin-orbit coupling correction.

While we did not include the Au slab explicitly in the simulations, we considered its effect implicitly in the calculation of binding energies and the estimation of charge states. Binding energies of anchored molecular structures under ultra-highvacuum conditions (partial pressure tending to zero) were calculated by

$$E_b = E_{\text{anchor}}(q) - E_{\text{MoS}_2 + \text{VS}}(q) - E_{\text{mol}} + qE_f^{\text{Au}}, \quad (1)$$

where q is the charge,  $E_{anchor}$  is the total energy of the anchored molecular system,  $E_{MoS_2+VS}$  is the total energy of the  $MoS_2$  substrate containing the S vacancy,  $E_{mol}$  is the total energy of the CF<sub>3</sub>-3P-SH molecule in the gas phase, and  $E_f^{Au}$ is the Fermi energy of Au. The total energies were obtained with the HSE06+MBD-NL functional and the Au Fermi energy at the HSE06 level of theory (-4.95 eV) was considered as an internally consistent reference. The value entering the computation was referenced to the position of the valenceband maximum (VBM) of the MoS<sub>2</sub> substrate containing the S vacancy at a consistent charge state and same DFT level of theory. With this definition, a more negative  $E_b$  means stronger binding. When the anchored molecule was dehydrogenated to CF<sub>3</sub>-3P-S, we considered an adsorbed hydrogen far away from the defect site to obtain a simple estimate of a lower bound of the binding energy. We did not correct for finite-size effects.

#### III. ANCHORING CF3-3P-SH MOLECULES TO S VACANCIES IN MoS2

The pristine  $MoS_2$  islands appear in STM topography with a characteristic moiré pattern caused by the lattice mismatch between  $MoS_2$  and the underlying Au(111) [30] [Fig. 1(a)]. After sputtering with Ne ions, the  $MoS_2$  layer is decorated with defects appearing with a triangular rim around a dark center [see Fig. 1(b)]. These defects have been attributed to S vacancies in the terminating S layer [24]. Importantly, at low temperatures these sites were shown to localize a single electron, which is exchange coupled to the metal electrons at the interface of  $MoS_2$  and Au(111) as reflected by a Kondo resonance [24]. The localized negative charge is most likely



FIG. 2. STM topographies of subsequent rotation steps of a  $CF_3$ -3P-SH molecule on  $MoS_2/Au(111)$  around its anchoring point (marked by black dot) after interaction with the STM tip [set points: (a) and (c) 1 V, 40 pA, (b) 1 V, 50 pA].

transferred from the metal substrate. The observed charge state is in agreement with theoretical predictions, favoring the stabilization of the charged defect state in proximity to a Au substrate [25,39,40].

Large organic molecules such as CF<sub>3</sub>-3P-SH have a low diffusion barrier on  $MoS_2$  [41]. To reduce diffusion and maintain a few molecules on the  $MoS_2$  islands, a deposition temperature of 160 K was chosen. However, we then observed that most molecules were physisorbed on the islands with a very large mobility under the influence of the tip. Subsequent annealing to 220 K led to major diffusion, so that most of the CF<sub>3</sub>-3P-SH molecules were found on the Au(111) substrate instead of on  $MoS_2$ . On Au(111) the molecules form clusters or are attached to the edges of the  $MoS_2$  islands [Fig. 1(c)], most likely in a physisorbed configuration [42]). Yet, we also detected a few oval-shaped protrusions indicating single molecules on the  $MoS_2$  substrate.

A close-up view on one of these molecules is shown in Fig. 2. Interestingly, the molecule is free to rotate around one of its terminations (indicated by a black dot in the series of images in Fig. 2). The rotations can be induced by approaching the STM tip. In none of the cases did the molecule move laterally. We thus assume that it is anchored to a defect site. Most probably, the defect is a S vacancy in the top layer of MoS<sub>2</sub>, as we did not observe anchored CF<sub>3</sub>-3P-SH molecules without prior creation of these sites by Ne bombardment. The robust anchoring of the molecule suggests the presence of a covalent bond to the S defect. Our preparation procedure of deposition at low temperature followed by mild annealing thus allows for some of the CF<sub>3</sub>-3P-SH to bond to the S vacancies.

To unravel the atomic structure arising from this chemisorption scenario, we performed a combination of firstprinciples simulations. We analyzed the outcome of structure searches performed on a neutral and a negatively charged S-vacancy site in a freestanding monolayer of  $MoS_2$  in contact with intact  $CF_3$ -3P-SH molecules and dehydrogenated  $CF_3$ -3P-S molecules. We considered both scenarios because in a previous publication from some of the authors [25], we have established that the formation energy difference between neutral and negatively charged vacancies only amounts to 43 meV when considering the Au substrate as the electronic reservoir. Therefore, by estimating the respective Boltzmann weights, we conclude that while at very low temperatures one would exclusively observe charged vacancy sites, at the reaction temperature of 220 K, around 10% of these vacancy sites would be neutral.

When considering a negative charge, we did not obtain any  $CF_3$ -3P-SH molecules docked to the vacancy. In contrast, we obtained stable dehydrogenated  $CF_3$ -3P-S molecular configurations anchored to the defect site. This suggests that if the molecule docks, the H atom can be dissociated from the thiol group with the dangling bond at the S termination then being the ideal candidate for healing the S vacancy site in MoS<sub>2</sub>. In the case of a neutral vacancy, both  $CF_3$ -3P-S and  $CF_3$ -3P-SH conformers are stable (see the Supplemental Material [31], Sec. SII).

Regarding the anchoring mechanism, we note that in order for CF<sub>3</sub>-3P-S to bond to the vacancy site, a reaction barrier for dehydrogenation has to be overcome. We refer to the work of Li et al., who characterized viable pathways for this reaction to happen at the defect site where the thiol group binds [28], reporting barriers of 0.2-0.3 eV. We conclude that this reaction can happen at neutral vacancies (available on the substrate at higher temperatures) and propose that these barriers can be lowered even further at negatively charged defects because these sites will be more attractive to protons. The abstracted H is likely to be highly mobile on MoS<sub>2</sub> and most probably diffuses to the metallic substrate. We surmise that a reaction, where H stays at a defect site and the unstable neutral radical diffuses and eventually finds another free defect site, is energetically highly unlikely. Indeed, negatively charged thiolates have been reported to be repelled even by neutral defects [28].

The local charge of the hybrid state formed by the molecule anchored to the defect site in MoS<sub>2</sub> is independent of whether the initial vacancy defect was neutral or negatively charged. Instead, it will be dictated by the stability of the hybrid state with respect to the electronic reservoir. Taking the electronic reservoir to be the Au substrate, we obtain the binding energy of the most favorable neutral conformer (considering  $CF_3$ -3P-SH and  $CF_3$ -3P-S) to be -0.56 eV, while the most favorable negatively charged case (considering only CF<sub>3</sub>-3P-S, as the anchored CF<sub>3</sub>-3P-SH is not stable in a negatively charged state) leads to a binding energy of only -0.08 eV, at the verge of stability within the uncertainties of our estimation. Note that the latter value does not include the energy penalty for dehydrogenation of CF<sub>3</sub>-3P-SH into CF<sub>3</sub>-3P-S. We did not explore positively charged final states because they would be even less likely in this setting. We thus propose that the final state of the anchored molecules is neutral. The structure of the negatively charged anchored molecules and the respective electronic states are presented in the Supplemental Material [31], Sec. SV, for completeness.

We now discuss the structural conformations of the anchored molecules. From the DFT simulations, the most stable CF<sub>3</sub>-3P-SH conformer presents a tilting angle  $\theta = 15^{\circ}$ with respect to the surface normal [see definition of  $\theta$  in Fig. 1(c)]. Regarding the dehydrogenated CF<sub>3</sub>-3P-S species, we found upright-standing configurations ( $\theta \approx 0^{\circ}$ ) to be energetically favored. However, other stable minima within a 300 meV energy window are also found with larger  $\theta$  (see the Supplemental Material [31], Sec. SII, Figs. S3– S6). In general, conformations with larger  $\theta$  can be stabilized when considering entropic contributions. This was confirmed by exploratory molecular-dynamics simulations at 100-300 K, presented in the Supplemental Material [31], Sec. SIII. One can easily picture this, as any tilted configuration has access to a larger configurational space of rotations around the normal axis. Of course, we also cannot exclude that considering a lower concentration of molecules in the simulations can decrease the enthalpic penalty of tilted configurations, because it would decrease spurious interactions between periodic repetitions of the molecules. All geometries we obtain from the structure search, with and without H, present a lower  $\theta$  than what experiment seems to observe, where the molecules are almost flat lying. In addition, we stress that in our simulations, several inclinations of the molecule are quite similar in energy (see the Supplemental Material [31], Sec. SII, Fig. S3), which suggests a very flat energy landscape along the inclination coordinate, such that we do not expect to be quantitative on the inclination angle based on static geometry optimizations.

In summary, we propose that, for either CF<sub>3</sub>-3P-S or CF<sub>3</sub>-3P-SH anchored to defect sites, two factors would contribute, if included in the simulations, to the stabilization of the tilted conformations with large  $\theta$  which are observed experimentally: (i) a low concentration of molecules and (ii) entropic contributions that thermally populate the softer rovibrational modes of the molecule. We cannot exclude that a lowering of zero-point-energy contributions when molecular groups are physisorbed on the surface could also contribute to this stabilization.

## IV. ELECTRONIC PROPERTIES OF ANCHORED MOLECULES

To investigate the electronic properties of the anchored molecules, we record dI/dV spectra on the anchored molecules. As we will show below, we found two types of dI/dV spectra. We investigated around 140 molecules and observed that 80% of them do not show a Kondo resonance while 20% do. We associate this electronic structure to two distinct molecules anchored to the S defects according to our theoretical simulations. One type consists of an intact CF<sub>3</sub>-3P-SH molecule, whereas the other type has undergone dehydrogenation to a CF<sub>3</sub>-3P-S molecule.

#### 1. Anchored CF<sub>3</sub>-3P-SH molecules

We start by presenting the dI/dV spectra on the more abundant species. Spectra on the anchoring site and the center of the molecule are shown in Fig. 3 (green and orange) together with a spectrum on the pristine MoS<sub>2</sub> (gray). In the latter, the conductance is essentially flat over a large bias-voltage range, in agreement with the semiconducting band gap of MoS<sub>2</sub>, with the onset of the conduction band smeared out while exhibiting a small peak at ~ 0.9 V, followed by a stronger one at ~1.4 V [41].

The anchored molecules show two main features inside the band gap of  $MoS_2$  (Fig. 3): a positive ion resonance (PIR) of high intensity at  $\sim -1$  V with satellite peaks following behind, and a broader negative ion resonance (NIR) starting at  $\sim$ 420 mV. Both states are found to be very intense at the anchored end (green spectrum in Fig. 3), and show nearly



FIG. 3. dI/dV spectra on an anchored CF<sub>3</sub>-3P-SH molecule on MoS<sub>2</sub>/Au(111), on the anchor point (green), the free end (orange), and the reference taken on bare MoS<sub>2</sub> (gray). The spectra are vertically offset by 0.125 nS from each other. The spectrum on the anchored side has been rescaled by a factor of 0.4 for better comparison with the other spectra. The gray dashed lines denote the bias values of the dI/dV maps taken in Fig. 4. [Feedback opened at 1.5 V, 100 pA, lock-in parameters: 2 mV (MoS<sub>2</sub> and anchor) and 5 mV (free end) modulation amplitude.] Inset: STM topography of the measured molecule, with colored crosses marking the positions where the spectra are recorded (set point: 2 V, 30 pA). Note that the anchoring group of the molecule is located in a moiré minimum, while the free end is on a moiré maximum. The convolution of molecular and moiré states results in a featureless molecule at this bias voltage.

no intensity on the free end of the molecule (orange spectrum in Fig. 3). To bring out the spatial distribution of these states more clearly, we present constant-height dI/dV maps recorded at the resonance energies in Fig. 4 (energies of maps marked with a gray dashed line in Fig. 3).

The map taken at the energy of the PIR [-990 mV,Fig. 4(b)] shows a strong localization at the anchoring site. Additionally, it shows some faint intensity along the molecular backbone, exhibiting two nodal planes. The NIR at ~420 mV also shows the largest contribution close to the anchoring site, but slightly offset from the molecular axis [Fig. 4(d)]. Maps at low energy [-220 meV, Fig. 4(c)] and higher energy



FIG. 5. (a) Projected electronic density of states (PDOS) of the stable conformer of  $CF_{3}$ -3P-SH anchored to the neutral  $MoS_2$ monolayer, obtained with density-functional theory and the HSE06 exchange-correlation functional. The molecular geometry is shown in the inset. Cyan and red lines represent the PDOS on the molecule and on  $MoS_2$ , respectively. The red shaded area marks occupied states and the blue shaded area marks unoccupied states. The zero of energy is set to the valence band maximum (VBM) and that state is doubly occupied. (b) Orbital densities corresponding to the highest doubly occupied state in pink and the lowest unoccupied state in green. We show an isodensity of 0.0002 e/Bohr<sup>3</sup>.

[610 meV, Fig. 4(e)] show a faint structure along the molecule with some intramolecular modulation.

To understand the origin of the observed states, we analyze the electronic structure obtained by DFT calculations of candidate molecular configurations anchored on the S vacancy. We first consider the anchoring of an intact  $CF_3$ -3P-SH molecule on a neutral vacancy. This choice is motivated by the absence of a Kondo resonance, and, hence, of a singly occupied state.

In Fig. 5 we show the corresponding density of states of the  $CF_3$ -3P-SH molecule adsorbed on the vacancy. As we neglect the Au(111) substrate in the simulations, we do not expect quantitative agreement in the energy-level alignment but aim



FIG. 4. (a) STM topography (set point: 2 V, 30 pA) and constant height dI/dV maps of an anchored CF<sub>3</sub>-3P-SH molecule (anchor point marked by green dot) taken at energies marked by gray lines in Fig. 3: (b) -990 mV, (c) -220 mV, (d) 425 mV, (e) 610 mV. [Feedback opened at (b) -990 mV, 30 pA over anchored end, (c)–(e) 610 mV, 30 pA above free end, lock-in parameters for all: 5 mV modulation amplitude.]



FIG. 6. dI/dV spectra on an anchored CF<sub>3</sub>-3P-S molecule on MoS<sub>2</sub>/Au(111), on the anchor point (green) and the reference taken on bare MoS<sub>2</sub> (gray). The gray dashed lines denote the bias values of the dI/dV maps shown in Fig. 7. (Feedback opened at 500 mV, 100 pA; lock-in parameters: 5 mV modulation amplitude.) Inset: STM topography of the measured molecule, with colored crosses marking the positions where the spectra were taken (set point: 1.3 V, 30 pA).

for a qualitative identification of the hybrid electronic states. In this case, both the highest (doubly) occupied state and the lowest unoccupied state are localized at the anchoring site. These states are separated by an energy gap of 1.3 eV. The states originate from the hybridization of the frontier orbitals of the CF<sub>3</sub>-3P-SH molecule. In experiment, the energy separation of NIR and PIR is different from the HOMO-LUMO gap because of the charging energy necessary to transiently add and remove an electron from the system. In DFT, the Kohn-Sham states employing the hybrid exchange-correlation functional are only an approximation to HOMO and LUMO states. Nevertheless, as we only aim for a qualitative identification of the states, we consider the energy gap and the state localization to agree well between experiment and theory. We recall that for CF<sub>3</sub>-3P-SH to bond to the defect site, the vacancy must be neutral. Although all pristine S vacancies exhibit a Kondo resonance at low temperatures reflecting a charged state, at the annealing temperature of 220 K the vacancy is transiently neutral with a relevant probability of 10%, as discussed above. The neutral state can thus act as an anchoring site for the intact CF<sub>3</sub>-3P-SH molecule.

#### 2. Anchored dehydrogenated CF<sub>3</sub>-3P-S molecules

In contrast to the molecular species described above, we find a few molecules with a very different spectral fingerprint. As shown in Fig. 6, these molecules exhibit a sharp resonance at zero bias that we associate to a Kondo resonance flanked by two side peaks at  $\sim -110$  mV and  $\sim 190$  mV. Differential conductance maps at these energies are presented in Fig. 7 along with the topographic image. All resonances are mainly localized at the anchoring site. Given their similar spatial distribution and close energy spacing, we suggest that the side peaks arise from tunneling through the singly occupied state at negative bias and the doubly occupied state separated by the Coulomb charging energy from the singly occupied state at positive bias. The experiments thus indicate the presence of a singly occupied state when the molecule is anchored to the S vacancy.

Following our stability analysis above, the hybrid state must be charge neutral, but a singly occupied state must be present. Therefore, we consider a dehydrogenated CF<sub>3</sub>-3P-S radical bonded to the S vacancy. We suggest that H is catalytically dissociated from the thiol group when the molecule is in the vicinity of a negatively charged vacancy. The dangling bond is then saturated upon chemisorption of the CF<sub>3</sub>-3P-S molecule to the vacancy site, and H could subsequently diffuse away from the anchored molecule. Because this incurs an energy penalty, this anchoring mode is less likely, in agreement with the less frequent observation as opposed to the first species discussed above. Overall charge neutrality results from the interaction of the Au substrate acting as a charge reservoir as discussed above. A representative molecular geometry, the projected electronic density of states and the visualization of key electronic states are shown in Fig. 8. The bonded conformations show a singly occupied state in the gap that could lead to a Kondo resonance. In agreement with experiment, this state is localized at the anchoring point. We also note that it is robust to several different molecular orientations found in the calculations. Other geometries with their corresponding electronic structure are reported in the Supplemental Material [31], Sec. SIV.

## **V. CONCLUSIONS**



Using scanning tunneling spectroscopy, we have observed two types of reaction products of CF<sub>3</sub>-3P-SH molecules

FIG. 7. (a) STM topography (set point: 1.3 V, 30 pA) and constant-height dI/dV maps of an anchored CF<sub>3</sub>-3P-S molecule (anchor point marked by green dot) taken at energies marked by gray lines in Fig. 6. (b) -110 mV, (c) 3 mV, and (d) 190 mV. [Feedback opened at 500 mV, 100 pA, and 100 pm tip approach; lock-in parameters: 5 mV modulation amplitude.]



FIG. 8. (a) Projected electronic density of states (PDOS) of the stable conformer of CF<sub>3</sub>-3P-S anchored to the neutral MoS<sub>2</sub> monolayer, obtained with density-functional theory and the HSE06 exchange-correlation functional. The molecular geometry is shown in the inset. Cyan and red lines represent the PDOS on the molecule and on MoS<sub>2</sub>, respectively. The red shaded area marks occupied states and the blue shaded area marks unoccupied states. The zero of energy is set to the valence band maximum (VBM), and that localized state is singly occupied. (b) Orbital densities corresponding to the highest singly occupied state in pink and the lowest fully unoccupied state at 0.57 eV in green. We show an isodensity of  $0.0002 \text{ e/Bohr}^3$ .

anchoring to purposefully created S vacancies in  $MoS_2$  on Au(111). Based on *ab initio* structure searches, DFT calculations, and comparison to the experimental tunneling spectra, we assign these two configurations to intact CF<sub>3</sub>-3P-SH molecules or dehydrogenated CF<sub>3</sub>-3P-S molecules and characterize their structural and electronic properties.

Due to the presence of the Au(111) substrate acting as a charge reservoir, both locally formed molecular hybrid states are charge neutral. Both species exhibit states within the band gap of MoS<sub>2</sub> and the anchored CF<sub>3</sub>-3P-SH is more abundant. Interestingly, the CF<sub>3</sub>-3P-S molecule features a singly occupied state, which explains the few anchored molecules that present a Kondo resonance in experiment. We note that the precise energy alignment and the localization of these new hybrid states depends sensitively on the adsorption configuration of the molecule on the moiré structure of MoS<sub>2</sub>/Au(111) (see the Supplemental Material [31], Sec. SVI).

The combination of well-defined experiments and theory in this model system allowed an atomic-scale understanding of the properties of defect-engineered states, where a minimally biased first-principles structure search proved essential to unravel the possible atomic and electronic structure of docked molecules. Such an understanding would not be achievable with experiment or theory alone.

Going forward, the use of molecules with functional end groups, such as charge donating or charge withdrawing groups, will enable further modulation of the energy level alignment. Attachment of molecular switches may additionally allow a reversible variation of energy levels and, thus, conductance properties, while magnetic molecular centers may merge magnetic with the optoelectronic properties of the TMDC layer for more complex device applications.

The supporting data for this article are openly available from Zenodo [45].

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