## **Oscillatory Hall effect from magnetoelectronic coupling in flexoelectronic silicon**

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Magnetoelectronic coupling can be defined as cross-domain coupling between electronic and magnetic properties, where modulation in magnetic properties changes the electronic properties. In this Letter, explicit experimental evidence of magnetoelectronic coupling is presented, which is uncovered from the oscillatory Hall effect response in Hall measurement. The strain gradient in a MgO (1.8 nm)/*p*-Si (∼400 nm) freestanding sample leads to transfer of electrons ( $\sim 5 \times 10^{18}$  cm<sup>-3</sup>) from valence to conduction band due to flexoelectronic charge separation in the *p*-Si layer. The resulting flexoelectronic polarization gives rise to the temporal magnetic moment from dynamical multiferroicity. The external magnetic field changes the net temporal magnetic moment, which causes modulations in charge carrier concentration and oscillatory Hall effect. The period of oscillatory Hall response is 1.12 T, which is attributed to the magnitude of the temporal magnetic moment. The discovery of the oscillatory Hall effect adds another member to the family of Hall effects.

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*Introduction.* In a recent discovery, a large phonon magnetic moment  $(1.2 \mu_B / \text{atom})$  was reported in Si thin film under an applied strain gradient. The large magnetic moment was attributed to the dynamical multiferroicity  $[1-4]$ , which can be described as

$$
\mathbf{M}_t \propto \mathbf{P}_{\text{Flexoelectronic}} \times \partial_t \mathbf{P}, \tag{1}
$$

where  $M_t$ ,  $P$ , and  $P_{\text{Flexoelectronic}}$  are the temporal magnetic moment and polarization of optical phonons and the flexoelectronic polarization. The flexoelectronic polarization arises due to charge carrier transfer from metal layer to doped semiconductor layer in a metal/semiconductor heterostructure under an applied strain gradient, as demonstrated recently [\[5\]](#page-4-0). As a consequence, the flexoelectronic polarization is proportional to the gradient of the charge carrier concentration (*n***-** ) and Eq.  $(1)$  can be rewritten as

$$
M_t \propto n' \times \partial_t P. \tag{2}
$$

This equation describes inhomogeneous magnetoelectronic multiferroicity and the magnetoelectronic electromagnon (magnetoactive phonon), as demonstrated recently  $[6]$ .

Equation (2) also describes possibly the magnetoelectronic coupling in the materials. The magnetoelectronic coupling can be described as cross correlation between the magnetic properties (temporal magnetic moment) and the electronic properties (charge carrier concentration), which is the underlying cause of a recently demonstrated magnetoelectronic electromagnon [\[6\]](#page-4-0). In materials having magnetoelectronic coupling, an external magnetic field applied at an angle to the direction of the temporal magnetic moment will give rise to precession of the temporal magnetic moment. As a consequence, the net temporal magnetic moment orthogonal to the polarization of optical phonons and flexoelectronic polarization will become smaller. Hence the applied magnetic field will change the electronic charge separation (or flexoelectronic polarization) as well as the charge carrier concentration assuming polarization of phonons does not change as shown in Fig.  $1(a)$  and can be described as

$$
(\boldsymbol{M}_t \pm \mathbf{B}) \propto (\boldsymbol{n} \pm \boldsymbol{\Delta n})' \times \partial_t \boldsymbol{P}, \tag{3}
$$

where **B** and  $\Delta n$  are external magnetic field and change in charge carrier concentration, respectively. Since, the external magnetic field changes the electronic properties of the material, we call it magnetoelectronic coupling. This is analogous to magnetoelectric coupling, which allows electric control of magnetic behavior in multiferroic materials [\[7,8\]](#page-4-0). In this Letter, we present experimental evidence of the oscillatory Hall effect response due to magnetoelectronic coupling. The magnetoelectronic coupling leads to change in charge carrier concentration as a function of magnetic field.

*Experimental results.* The flexoelectronic charge carrier transfer has so far been demonstrated primarily in metal/ doped semiconductor heterostructures under applied strain gradient. However, the response from the metal layer is superimposed on the response from the semiconductor layer, which makes it difficult to segregate the responses. The potential flexoelectronic effect has also been demonstrated in MgO/*p*-Si (degenerate) bilayer structures but the mechanism was not identified since there is no charge carrier metal source in the new bilayer structures [\[9\]](#page-4-0). This led us to choose a similar bilayer configuration for this study since the transport response will only arise from the degenerately doped Si layer.

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FIG. 1. (a) Schematic showing the magnetoelectronic coupling from an externally applied magnetic field due to electronic dynamical multiferroicity. (b) Schematic showing the interfacial piezoelectric polarization in native oxide under an applied strain gradient, which acts on the degenerately doped Si layer. (c) Schematic showing the charge carrier distribution due to interfacial piezoelectriclike effect leading to nonequilibrium electrons in the conduction band giving rise to a metastable flexoelectronic polar *p*-Si layer. (d) Schematic showing the deformed charge distribution along the close packed directions  $\langle 112 \rangle$  in the [110] plane, which gives rise to the electronic polarization and temporal magnetic moment of the magnetoelectronic electromagnon due to superposition from phonons.

The charge carrier concentration is usually measured using the Hall effect. Hence the magnetoelectronic coupling and change in charge carrier concentration can also be uncovered using Hall resistance measurement. For the experimental setup, we take a degenerately doped (boron) 2 µm thick *p*-Si (0.001–0.005  $\Omega$  cm) silicon on insulator (SOI) wafer (commercially available). We then reduce the thickness of the device *p*-Si layer to ∼400 nm using successive oxidation and hydrofluoric acid (HF) etching [\[10\]](#page-4-0). The device *p*-Si layer thickness is reduced to achieve the larger strain gradient. The device layer is then patterned and etched in a Hall bar configuration using photolithography and Si deep reactive ion etching, respectively [\[9\]](#page-4-0). The sample area is made freestanding using HF vapor etch of the buried oxide layer underneath the device layer. We take two samples from the same part of the wafer in order to have similar initial doping characteristics. Sample 1 is the control sample where no further processing is carried out. For sample 2, we deposited 1.8 nm of MgO on top of the *p*-Si layer [\[9\]](#page-4-0). The thickness of the native oxide layer is expected to be  $3.7 \text{ nm}$  [\[11\]](#page-4-0).

The freestanding *p*-Si layer will have residual stress and buckling will induce a strain gradient in the bulk of the sample. In sample 1, the strain gradient may lead to small charge carrier separation due to a gradient in the band structure, but the native oxide layer is free to deform and will not influence the overall response as shown in Fig.  $1(b)$ , whereas the native oxide in sample 2 is constrained by the MgO top layer and is not allowed to freely expand or contract. As a result, the strain gradient will lead to a piezoelectriclike response [\[12\]](#page-4-0) from the native oxide layer on top of *p*-Si as shown in Fig.  $1(b)$ . The piezoelectriclike response from the native oxide layer arises, potentially, due to nonstoichiometry and dangling bonds. It is noted that deposition of any layer on top of Si with native oxide will constrain the native oxide layer and the MgO layer is not expected to contribute towards the interfacial response  $[13]$ . However, the deposition of the MgO layer is undertaken using rf sputtering, which may lead to charge accumulation at the surface. It is noted that the strain gradient is not expected to induce any phase transition in the Si [\[11,14\]](#page-4-0). In a degenerately doped *p*-Si, the impurity states give rise to a continuous impurity band and the Fermi level will be near the edge of the valence band as shown in Fig. 1(c). Now, the interfacial piezoelectriclike response can be considered as a gate bias on the Si layer. The free charge carrier distribution inside the Si layer will deform to neutralize the interfacial piezoelectriclike response and surface charge accumulation in the second case as shown in Fig.  $1(c)$ . The charge carriers are expected to jump to the conduction band as shown in Fig.  $1(c)$ , leaving behind holes in the valence band. This increase in free charge carriers will increase the conductivity as well as Hall resistance. This proposed behavior has been reported in a MgO/*n*-Si (2 µm) sample where the current-dependent response showed an increase in charge carrier concentration from  $\sim$  - 5.9×10<sup>19</sup> cm<sup>-3</sup> at 1 mA of current to  $\sim$  – 8×10<sup>19</sup> cm<sup>-3</sup> at 5 mA due to increased buckling [\[1\]](#page-4-0). Similarly, the resistivity of the sample also decreased from  $\sim$ 2.44×10<sup>-5</sup> Ωm (34.95 Ω) at 1 mA to  $\sim$ 1.5×10<sup>-5</sup> Ωm (24.73  $\Omega$ ) at 5 mA [\[1\]](#page-4-0). The nonequilibrium charge carrier transfer from the interior of the atom to the outer edge of the atom will lead to deformed charge distribution around the atom as shown in Fig.  $1(d)$  and as stated earlier. The resulting partial ionization of the *p*-Si layer from deformed charge distribution will give rise to a metastable state having dipolelike behavior as shown in Fig.  $1(d)$ . We call it flexoelectronic polarization because it is an electronic response to a strain gradient. The flexoelectronic polarization will be equal and



FIG. 2. (a) The Hall response measured in control sample 1 at 350 K. (b) The Hall response in sample 2 at 350 K. (c) The residual response from random noise in control sample 1 at 350 K. (d) The oscillatory component of the Hall response in sample 2 showing a period of 1.12 T.

opposite to the interfacial piezoelectriclike effect. Then the superposition of the flexoelectronic polarization and circularly polarized phonons will give rise to dynamical multiferroicity, essential for magnetoelectronic behavior. The nonequilibrium free charge carrier will preferentially reside along the closepacked directions in the cross section of the sample such as  $\langle 112 \rangle$  in the [110] plane giving rise to a temporal magnetic moment along the  $\langle 111 \rangle$  directions as reported earlier [\[1,6\]](#page-4-0) and as shown in Fig.  $1(d)$ .

In the first set of experiments, we measured the Hall response as a function of magnetic field from 3 to −3 T at 350 K in control sample 1 (200  $\mu$ A) and sample 2 (500  $\mu$ A) as shown in Figs.  $2(a)$  and  $2(b)$ . An alternating current (ac) bias is applied using a Keithley 6221 current source and response is measured using a Stanford Research Systems SR830 lock-in amplifier. The different biasing currents are used to account for the different resistivities of the samples. At 350 K, the Hall measurement in sample 1 shows a negative Hall resistance and charge carrier concentration of  $-3.84 \times 10^{18}$  cm<sup>-3</sup> as shown in Fig.  $2(a)$ . The negative sign of the Hall resistance is surprising at the measured charge carrier concentration and can be attributed to the Fermi level being near the valence band edge as well as from the strain induced gradient in the band structure. Zhang and Chang [\[15\]](#page-4-0) proposed that charge migration may enhance the flexoelectric response in Si. The observed sign of Hall resistance in control sample 1 may also arise from charge migration behavior. The residual response from the line fit is shown in Fig.  $2(c)$ , which exhibits the noise floor of the measurement. As compared to control sample 1, sample 2 exhibits an anomalous oscillatory Hall response as shown in Fig.  $2(b)$ . Using a line fit, we extract the linear Hall resistance and the oscillatory response as shown in Figs.  $2(b)$  and  $2(d)$ . The Hall resistance is negative and the charge carrier concentration is estimated to be  $-9.7 \times 10^{18}$  cm<sup>-3</sup>. The difference in the charge carrier

concentration between sample 1 and sample 2 is  $4.86\times10^{18}$  cm<sup>-3</sup>. This difference is attributed to the transfer of charge carriers from the interior of the atom to the conduction band in reaction to the interfacial piezoelectriclike response from constrained native oxide, as hypothesized and as shown in Fig.  $1(c)$ . This shows that in spite of charge neutrality an electronic polarization (flexoelectronic) can be achieved in an inhomogeneous system.

The oscillatory response seems to be triangular as shown in Fig.  $2(d)$ . A negative slope will add and a positive slope will act opposite to the negative Hall response. We estimate the charge carrier concentrations to be  $-15.6\times10^{18}$  and  $-7.0\times10^{18}$  cm<sup>-3</sup> for positive and negative slopes. This behavior indicates fluctuation of free charge carrier concentration due to an applied magnetic field, which we call magnetoelectronic coupling, as hypothesized earlier. The superposition of flexoelectronic polarization from the nonequilibrium free charge carrier and circularly polarized phonons gives rise to a temporal magnetic moment of the resulting magnetoelectronic electromagnon. An externally applied magnetic field acts on the temporal magnetic moment and, as a consequence, the charge carrier concentration is also modified, as hypothesized earlier. From the oscillatory response, the period of the oscillation is estimated (using sine fit) to be ∼1.12 T as shown in Fig. 2(d). We therefore propose that the period of oscillation (1.12 T) is the magnitude of the temporal magnetic moment of the magnetoelectronic electromagnon, which is larger than the value (0.7 T) reported in the case of a 2 µm thick *n*-Si sample [\[1\]](#page-4-0). The oscillatory component of the response is ∼4% of the transverse resistance value at 3 T ( $\sim$ 5 Ω). One can argue that the response is noise from measurement. However, a random noise is not expected to be a function of a magnetic field with a period of ∼1.12 T. In addition, the amplitude of the oscillatory response  $(0.2 \Omega)$  is constant for the complete range of magnetic field as shown in Fig.  $2(d)$ . Further, the oscillatory response (∼4%) is an order of magnitude larger than the noise  $(<0.1\%)$  measured in sample 1 as shown in Fig.  $2(c)$  whereas the order of the Hall response is similar in samples 1 and 2 as shown in Figs.  $2(a)$  and  $2(b)$ . Hence the observed oscillatory response is not expected to arise from random noise. We also measured the longitudinal response between 1 and −1 T at 350 K. However, the magnetoresistance is too small to uncover any effect of oscillatory response on longitudinal resistance.

We then measured the Hall response in control sample 1 at 200 K and observed a negative Hall resistance as shown in Fig.  $3(a)$ . The charge carrier concentration in control sample 1 is estimated to be  $-2.56 \times 10^{18}$  cm<sup>-[3](#page-3-0)</sup> as shown in Fig. 3 (a). We also measured the Hall response in sample 2 from 0 to 3 T at 200 K, where the largest field is larger than the expected period of oscillation. At 200 K, the oscillatory Hall response was pronounced, as shown in Fig.  $3(b)$ . The linear Hall resistance is negative and the charge carrier concentration is estimated to be  $-8.22 \times 10^{18}$  cm<sup>-3</sup>. This is larger than the charge carrier concentration in sample 1 by  $5.66 \times 10^{18}$  cm<sup>-3</sup>, which is a clear evidence of flexoelectronic polarization in Si due to the interfacial piezoelectriclike effect as hypothesized.

Similar to 350 K, the oscillatory response at 200 K has a period of  $\sim$ 1.12 T as shown in Fig. [3\(c\).](#page-3-0) However, the oscillatory response has an opposite sign based on the

<span id="page-3-0"></span>

FIG. 3. (a) The Hall response measured in control sample 1 at 200 K. (b) The Hall response in sample 2 at 200 K from 0 to 3 T. (c) The oscillatory component of the Hall response in sample 2 showing a period of 1.12 T. (d) The second Hall response in sample 2 at 200 K from 1 to −1 T.

direction of field sweep as shown in Fig. 3(c). From the Hall response in Fig.  $3(b)$ , we observe that the Hall resistance changes sign from positive to negative. A linear fit into two opposite slopes shows the charge carrier concentration fluctuates due to applied magnetic field as shown in Fig. 3(b). The estimated charge carrier concentrations are  $-0.99\times10^{18}$ and  $+2.36\times10^{18}$  cm<sup>-3</sup> for negative and positive slopes as shown in Fig.  $3(b)$ . This change shows that a large number of charge carriers move between the conduction band and the valence band during the oscillatory response. This behavior supports our primary hypothesis of flexoelectronic charge separation depicted in Fig.  $1(c)$  as well as magnetoelectronic coupling, since the magnetic field leads to change in the charge carrier concentration. The oscillatory component is absent completely in control sample 1 at both 350 and 200 K, which proves our hypothesis.

We hypothesized that if the externally applied magnetic field is smaller than the period of oscillation then the oscillatory behavior will disappear. Hence we measured the Hall response in sample 2 again for an applied magnetic field from 1 to  $-1$  T at 200 K as shown in Fig. 3(d). We do not observe any oscillatory response in this measurement, which supports our hypothesis. However, we observe a hysteretic response (possibly anomalous Hall response) at 200 K as shown in Fig.  $3(b)$ , which can also potentially arise from magnetoelectronic coupling and the temporal magnetic moment of the magnetoelectronic electromagnon. In this measurement, the charge carrier concentration is estimated to be  $-7.34 \times 10^{18}$  cm<sup>-3</sup>. It is smaller than the charge carrier concentration estimated using a previous Hall response in Fig.  $3(b)$  by  $0.89 \times 10^{18}$  cm<sup>-3</sup>. This difference showed that a larger magnetic field potentially increases the charge carrier concentration significantly, which is clear proof of hypothesized magnetoelectronic coupling.



FIG. 4. The Hall response in sample 3 from 14 to −14 T at (a) 350 K and (b) 200 K.

It is noted that the transverse resistance in the first measurement on sample 2 is higher by an order of magnitude and the underlying reason for this difference is currently unknown. Further, in the first Hall response measurement, we started the magnetic field at 0 T and then swept it to 3 T rather than starting at 3 T. In spite of that, the oscillatory response is observed in this measurement. It is noted that measurement at 350 K is followed by this measurement at 200 K without breaking the sample chamber vacuum. As a consequence, the residual magnetic moment of the magnetoelectronic electromagnon due to high magnetic field at 350 K may induce oscillation at 200 K even though we do not start the measurement at a high magnetic field. The second Hall measurement in sample 2 at 200 K was carried out separately after the sample was out of the chamber for a long duration.

The flexoelectronic effect is previously reported in metal/doped semiconductor heterostructures. The measurement in sample 1 and sample 2 demonstrated dynamical multiferroicity and magnetoelectronic coupling in Si. To further support our argument, we take a Py (25 nm)/MgO (1.8 nm)/*p*-Si (400 nm) sample (sample 3). We measure the Hall response at 350 K as a function of magnetic field from 14 to −14 T and at an applied current bias of 2 mA shown in Fig.  $4(a)$ . The measured Hall response is positive. It suggests that the dominant charge carriers are holes, which is unexpected since electrons are the dominant charge carrier in Py. Based on the previous resistance measurements, the Py resistivity at 350 K is expected to be  $5.72 \times 10^{-7}$  Ωm [\[5\]](#page-4-0) and *p*-Si resistivity is expected to be  $1.36 \times 10^{-4}$  Ωm (from sample 1). However, the resistance of the heterostructure sample is found to be 32.3  $\Omega$ . As a consequence, the *p*-Si resistivity in the heterostructure needs to be  $8.94 \times 10^{-6}$  Ωm, assuming Py resistivity remains the same. The decrease in resistivity of the *p*-Si layer is attributed to the flexoelectronic charge carrier transfer from Py to the *p*-Si layer, which leads to electron deficiency in the Py layer and is also the underlying cause of the positive Hall resistance [\[5\]](#page-4-0). In the Hall measurement, we observe a hysteretic response as shown in Fig. 4(a). Based on the measured Hall resistances, the charge carrier concentration decreases from  $4.3 \times 10^{21}$  to  $1.45 \times 10^{21}$  cm<sup>-3</sup> at the magnetic field  $-10.8$  T as shown in Fig. 4(a). In the inverse magnetic sweep, the charge carrier concentration increases from  $1.36\times10^{21}$  to  $3.65\times10^{21}$  cm<sup>-3</sup> at the magnetic field 9.2 T as shown in Fig.  $4(a)$ . The large magnetic field leads to the spin of the charge carrier aligning with the external magnetic field. As a consequence, the reduction in the charge carrier concentration is attributed to the transfer of electrons below

<span id="page-4-0"></span>the Fermi level, in turn reducing the hole population. This behavior is similar to the oscillatory behavior observed in sample 2 except there are no oscillations in the response. The absence of oscillatory response is attributed to the Py dominated Hall response, which masks the response from the Si layer. The additional effects due to interlayer coupling and the proximity effect may also diminish the oscillatory response. The Hall resistance at 200 K is also positive due to flexoelectronic charge transfer as shown in Fig. [4\(b\).](#page-3-0) The hysteresis behavior at 200 K is at a lower magnetic field and weaker in magnitude as compared to 350 K as shown in Fig.  $4(b)$  inset. This is attributed to the freezing of the magnetoelectronic electromagnon and the resulting decrease in the temporal magnetic moment. This behavior is opposite of sample 2 where the oscillatory response is larger at 200 K. The flexoelectronic effect in both samples 2 and 3 is due to different mechanisms. The Si layer is charge neutral in sample 2 whereas it is not in sample 3 since there are excess charge carriers from the Py layer in sample 3. As a consequence, the different behavior emerges from the lack of charge neutrality in sample 3. However, studies are needed to elucidate this further.

*Conclusion.* In conclusion, we presented experimental evidence of magnetoelectronic coupling in flexoelectronic Si.

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The magnetoelectronic coupling arises due to superposition of flexoelectronic polarization and circularly polarized phonons that also give rise to the temporal magnetic moment of magnetoelectronic electromagnon. The externally applied magnetic field modulates the temporal magnetic moment, which in turn changes the charge carrier concentration. This magnetoelectronic coupling leads to an oscillatory Hall effect response in the Hall measurement. The oscillatory response in the Hall resistance arises due to back and forth transfer of the charge carrier from the interior of the atom to the conduction band due to modulation of the temporal magnetic moment from the magnetic field. The period of the oscillatory Hall response is ∼1.12 T, which is expected to correspond to the magnitude of the temporal magnetic moment of the magnetoelectronic electromagnon. Though we presented evidence of magnetoelectronic coupling, a complete quantitative description and microscopic origin of the behavior will need further experimental and theoretical studies. Further, the discovery of the oscillatory Hall effect adds another member to the family of Hall effects from a different origin.

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