Superconducting transition temperatures of pure vanadium and vanadium-titanium alloys in the presence of dynamical electronic correlations

D. Jones , 1,2 A. Östlin, A. Weh, F. Beiuşeanu, U. Eckern, L. Vitos, and L. Chioncel, and L. Chioncel ¹Theoretische Physik III, Center for Electronic Correlations and Magnetism, Institute of Physics, University of Augsburg, 86135 Augsburg, Germany ²Augsburg Center for Innovative Technologies (ACIT), University of Augsburg, 86135 Augsburg, Germany ³Faculty of Science, University of Oradea, 410087 Oradea, Romania ⁴Theoretische Physik II, Institute of Physics, University of Augsburg, 86135 Augsburg, Germany ⁵Applied Materials Physics, Department of Materials Science and Engineering, Royal Institute of Technology, Brinellvägen 23, 10044, Stockholm, Sweden

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Ordinary superconductors are widely assumed insensitive to small concentrations of random nonmagnetic impurities, whereas strong disorder suppresses superconductivity, ultimately leading to a superconductor-insulator transition. In between these limiting cases, a most fascinating regime may emerge where disorder enhances superconductivity. This effect is discussed here for the β phase of vanadium-titanium alloys. Disorder is modeled using the coherent potential approximation while local electronic interactions are treated using dynamical mean-field theory. The McMillan formula is employed to estimate the superconducting transition temperature, showing a maximum at a Ti concentration of around 0.33 for a local Coulomb interaction U in the range of 2 eV to 3 eV. Our calculations quantitatively agree with the experimentally observed concentration-dependent increase of T_c , and its maximal value of about 20%.

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I. INTRODUCTION

The superconducting critical temperature T_c is a key quantity connecting materials' chemistry with the crystal and electronic structure of superconducting materials. At the same time T_c emerges from the underlying mechanism of superconductivity. It is generally understood that superconductivity of many elemental metals at ambient pressures can be described by the Bardeen-Cooper-Schrieffer (BCS) theory [1,2] based on the model of an effective instantaneous attractive interaction between electrons mediated by phonons (weak-coupling limit). The BCS model was extended by considering the retarded nature of the electron-phonon interaction: in particular, the (strong) Coulomb repulsion between the paired electrons is reduced because of the large difference in electron and phonon velocities [3]. The proper retardation effects, beyond BCS, lead to a complex and frequency-dependent order parameter, and is captured within the Eliashberg theory [4].

The estimation of the critical temperature within Eliashberg theory involves moments of the Eliashberg function $\alpha^2 F(\nu)$, which describes the phononic modes through which electrons effectively interact, and their coupling to electrons. The Eliashberg theory also accounts for the renormalization of the direct Coulomb interaction between two electrons (μ^*) which is known as the pseudopotential effect [3]. The McMillan formula allows the estimation of the critical temperatures from basic experimental data [5], including the electron mass enhancement factor due to the electron-phonon interaction, $m^*/m \approx 1 + \lambda$.

The alloys of transition metals constitute a wide class of superconducting materials. The theory of disordered

superconductors was formulated within the seminal work of Abrikosov and Gorkov [6]. At the same time, Anderson [7] has shown that superconductivity with s-wave pairing symmetry is insensitive to weak nonmagnetic disorder. Subsequent experimental studies, however, demonstrated that superconductivity is suppressed in strongly disordered samples. Very strong disorder can lead to a metal-insulator transition in the normal state, to the appearance of a pseudogap in the spectrum, to larger spatial fluctuations of superconductive pairing, and to an increased ratio of zero-temperature gap versus critical temperature Δ/T_c [8–14] in agreement with the experimental results.

Thus, introducing random impurities and defects in ordered materials can be regarded as a tool for controlling the superconducting characteristics of materials. The main problem, however, is to identify the nature of the impurity distribution, or more exactly the shape of the probability distribution function for the disorder realizations. In many materials simulations, the coherent potential approximation (CPA) [15–17] is used. As the CPA is a local theory, it captures only the average presence of different atomic species, and therefore cannot account for more subtle aspects like possible short-range order. These effects can be partly addressed using the dynamical cluster approximation [18], a cluster extension of dynamical mean-field theory (DMFT) [19-21] including electronic interactions, and the typical medium theory (TMT) [22,23]. Some of these advances have been used for models [24,25] or in conjunction with material-specific computations [26,27] in the framework of density functional theory (DFT). In this context, we have recently compared different modeling approaches to disorder and estimated the possible superconducting temperature of the high-entropy HfNbTaTiZr alloy [28].

In the present paper we calculate the superconducting transition temperatures of pure vanadium (V) and vanadiumtitanium (V-Ti) alloys, respectively, using the McMillan formula [5]. We start by probing its capability in the case of the pure element V for which the description of dynamic electronic correlations is necessary to describe the Fermi surface in its normal state [29]. Further we continue in addressing the problem of the enhancement of the critical temperature in some of the V-Ti alloys. In Sec. II we provide a brief description of the computational methods: the LDA + DMFT framework using a muffin-tin based approach, followed by a brief presentation of the essentials of the McMillan formula to estimate the critical temperatures. Then, Sec. III, an analysis of the electronic structure of vanadium and the vanadiumtitanium alloys in the normal state subject to dynamical electronic correlations and the corresponding estimates for the critical temperatures is presented. We summarize our results in Sec. IV.

II. METHODS

Density functional theory methods incorporating dynamical electronic correlations have proven essential in modeling physical properties of materials containing narrowband electronic states [20,21,30]. An additional difficulty is brought about by the existence of structural disorder. Green's function—based DFT methods are able to address the combined dynamic electronic correlation and disorder effects efficiently, since they compute directly the one-particle Green's function that can be averaged according to the various disorder realizations [26,27,31,32].

We start with a short description of the LDA + DMFT method used in the present computations, and then describe the methodology for computing the Hopfield parameters [33] as formulated by Gaspary-Gyorffy in the language of the multiple-scattering formalism [34]. This parameter contains the electron-phonon coupling matrix elements which are essential for the estimation of the superconducting critical temperatures employing the McMillan formula [5], and can be directly used in combination with the interacting LDA + DMFT Green's function as implemented in standard packages of electronic structure calculation.

A. The combined disorder and dynamical electronic correlations

To discuss dynamic correlation effects within the framework of LDA + DMFT [21,30] the standard methodology is to consider the following multiorbital on-site interaction term:

$$H_U = \frac{1}{2} \sum_{im,\sigma} U_{mm'm''m'''} c^{\dagger}_{im\sigma} c^{\dagger}_{im'\sigma'} c_{im''\sigma'} c_{im''\sigma}. \tag{1}$$

Here $c_{im\sigma}$ ($c_{im\sigma}^{\dagger}$) destroys (creates) an electron with spin σ on orbital m at the site i. The Coulomb matrix elements $U_{mm'm''m'''}$ are parametrized in terms of the average local Coulomb U and exchange parameter J [30]. In principle, the dynamical electron-electron interaction matrix elements can be com-

puted [35]; however, substantial variations exists depending on the used local orbitals [36]. We consider U in the range from 0 eV to 5 eV and J=0.6 eV. We have checked that the results essentially do not change when increasing J to, say, 0.9 eV.

The LDA + DMFT implementation used here is a charge and self-energy self-consistent scheme extending the exact muffin-tin orbitals method [37–39] using the local-density approximation (LDA). The full d manifold of states is considered in the many-body computations, while the charge self-consistency—through hybridization—ensures that dynamic correlation effects are felt by all orbitals. The impurity solver produces the many-body self-energy $\Sigma_{\sigma}(E)$ by the method of spin-polarized T-matrix fluctuation exchange [40-42], and corrects the starting LDA Green's function as discussed in previous publications [31,32,43]. To eliminate double counting of the interactions already included in the LDA exchange-correlation functional, the self-energy $\Sigma_{\sigma}(E)$ is replaced by $\Sigma_{\sigma}(E) - \Sigma_{\sigma}(0)$ in all equations of the LDA + DMFT scheme. Throughout this paper finite temperatures are only considered for the electronic subsystem, where the temperature enters in the Matsubara frequencies $\omega_n = (2n +$ $1)\pi T$. We perform total energy calculations within the charge self-consistent LDA + DMFT framework which follows the prescription in the literature [30]. The total energy is obtained from the LDA + DMFT functional [30] using the converged charge density and local Green's function. In addition to the standard LDA total energy, the Kohn-Sham band energy correction due to the DMFT is added together with the trace of the matrix product between the DMFT self-energy and the local Green's function (ΣG), the so-called Galitskii-Migdal contribution. For every pair of (U, J) values, the LDA + DMFT total energy was minimized as a function of the lattice parameter, and the superconducting critical temperature was computed.

Disorder was modeled through the CPA [15,16] which allows computation for any fractional concentration of disorder realizations in multicomponent alloys. Charge and self-energy self-consistency is achieved through the scattering path operator. This is the central quantity in the multiple-scattering formulations of the electronic structure and allows the simultaneous computation of the real space charge densities through charge self-consistency and a proper normalization the Green's function (DMFT self-consistency). Technically the CPA implements an algebraic average of the scattering path operator. Details of the implementation and a comparison between CPA/TMT+DMFT methods have been published previously [27,44].

B. McMillan critical temperatures from electronic structure computations

An estimate of the critical temperature in conventional superconductors can be obtained once the strength of the electron-phonon couplings is known. There are significant advances in the first-principle calculations of T_c using BCS-Eliashberg types of theories [45–47]. However, since electron-phonon calculations in disordered alloys from first principles remain computationally expensive, simplified

approaches like the Gaspari-Gyorffy theory [34] are used to provide the input for these estimates.

Within the theory of strongly coupled superconductors [5] the electron-phonon coupling constant λ can be expressed as

$$\lambda = N(E_F) \langle I^2 \rangle / M \langle \omega^2 \rangle, \tag{2}$$

where $N(E_F)$ is the density of states (DOS) at the Fermi level, $\langle I^2 \rangle$ is the squared electron-phonon matrix element, averaged over the Fermi surface, M is the atomic mass, and $\langle \omega^2 \rangle$ is the average squared phonon frequency. The numerator, $\eta = N(E_F)\langle I^2 \rangle$, is known as the Hopfield parameter. The observation that the electron-ion interaction in transition metals has a d resonance located in the vicinity of the Fermi surface allowed the formulation of a simple prescription to compute electron-phonon matrix elements, and finally λ , within the multiple-scattering formalism [34]. Early studies attempted to estimate λ for elemental metals from the knowledge of the experimentally measured T_c and the phonon spectra obtained from neutron scattering experiments [5]. It was observed that although the individual values of the DOS at the Fermi level and the averaged electron-phonon matrix elements change across the 3d series, the product of these two quantities remains approximately constant for all bcc transition metals [33,34]. In a so-called local-phonon representation, the electron-phonon interactions mainly consist of scatterings that change the electronic angular momentum l [33].

Using these ingredients, Gaspari and Gyorffy [34] proposed an approximate way to compute $\langle I^2 \rangle$ using the multiple-scattering Green's function formalism and adopting the rigid muffin-tin approximation. In these approximations, the Hopfield parameter is computed from a combination of electronic scattering phase shifts and the electronic densities of states. The average squared phonon frequency $\langle \omega^2 \rangle$ can be approximated by using the Debye temperature θ_D via $\langle \omega^2 \rangle \approx \frac{1}{2} \theta_D^2$. The same formalism was extended for disordered systems by including the proper disorder averaging of the physical quantities. Thus, the electron-phonon coupling constant for disordered alloys can be computed as $\lambda = \langle \eta \rangle_d / \frac{1}{2} \langle M \rangle_d \langle \theta_D \rangle_d^2$, where $\langle \dots \rangle_d$ represents the disorderaveraged quantities. As these parameters are accessible from a multiple-scattering based electronic structure calculation [48], the McMillan [5] formula

$$T_c = \frac{\theta_D}{1.45} \exp\left[-\frac{1.04(1+\lambda)}{\lambda - \mu^*(1+0.62\lambda)}\right]$$
(3)

can then be directly applied. The factor $(1+\lambda)$ plays the role of an electron mass enhancement and the parameter μ^* is an effective Coulomb repulsion reflecting the retardation effect of electron-phonon coupling with respect to the instantaneous Coulomb repulsion [5]. In our computations μ^* will be further considered as a variable parameter. All other ingredients of the formula besides μ^* are λ , computed using the exact muffin-tin orbitals method [38,39], and the Debye temperature collected from experimental data.

III. VANADIUM AND VANADIUM-TITANIUM ALLOYS

The physical properties of V have been intensively studied [49]. In fact, V is a type-II superconductor with a transition temperature of $T_c = 5.38$. Its normal state

thermodynamic properties were explained invoking phonons contribution [50].

Electronic structure computations were also performed by various methods including standard LDA and extensions such as LDA + *U* [51], LDA + DMFT [29,52], or including *GW*-like corrections [53]. An alternative LDA+FLEX computation for V [54] showed a sizable, remarkably local (**k**-independent) self-energy, which demonstrates the importance of local dynamic correlation effects in computing the ground state properties. The LDA + DMFT treatment revealed significant quantum fluctuations, provided a better agreement with the experimental Fermi surface [29], and allowed the formulation of a microscopic theory for the transition from the Pauli paramagnetism to Curie-Weiss behavior [52]. Computed estimates for the critical temperature of V [55] under pressure, using the McMillan formula, were also reported to be in agreement with experiment.

For the Ti_xV_{1-x} alloys, resistivity studies showed a relatively wide transition from the normal to the superconducting state which cannot be simply explained by the existence of a secondary superconductive phase [56]. The broad transition was observed for a relatively large concentration range. Thus, proposals were formulated for an additional scattering mechanism to the regular electron-phonon coupling based on spin-fluctuation theories [57,58]. In addition, experiments showed that Ti doping in bcc V leads to an enhanced T_c which exceeds the individual T_c values of the alloy components. As dynamical electronic correlations seem to provide an improved picture of physical properties of pure V [29,52,53] it is natural to expect that these effects play also an important role for the disordered alloy. The aim, thus, is to explain the increase in the critical temperature, T_c , as a consequence of the combined effect of disorder and dynamical electronic correlations in the dilute concentration limit.

A. Dynamic electronic correlation in the normal state of vanadium

In the following, we present the results of the electronic structure of pure vanadium computed within LDA + DMFT. Vanadium possesses a body-centered-cubic (bcc) structure within the so-called β phase. The space group corresponding to the β phase is $Im\overline{3}m$ with V occupying the Wyckoff 2a position, and the experimental lattice parameter is $a_{\text{exp}} = 3.02 \text{ Å}$. To determine the equilibrium lattice parameter a_{eq} for the pure vanadium crystal, we computed the LDA + DMFT total energy of the system for different values of the interaction strength U and a range of lattice parameters a around the expected energy minimum using the formalism presented in Ref. [31]. The convergence was checked on various k-mesh sizes up to $57 \times 57 \times 57$ k points in the irreducible part of the first Brillouin zone, although the saturation of the results [no significant change in $N(E_F)$, or in the Hopfield parameters] was noticed for smaller numbers of k points.

The electronic structure calculations provide information about the electronic bands and the density of states. The computed results can be compared to experimental measurements with different spectroscopic techniques such as the x-ray photoemission spectra (XPS) [59] for the valence band, and bremsstrahlung isochromat spectra (BIS) [60] for the

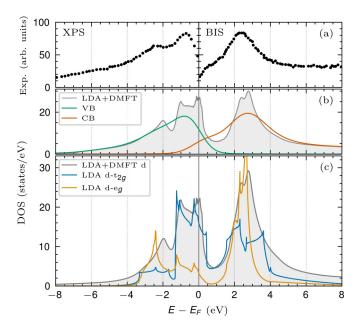


FIG. 1. Vanadium spectral functions. (a) Experimental XPS [59] and BIS [60] spectra in arbitrary units. (b) Valence band (VB) and conduction band (CB) DOS computed as convolution of the total density of states with distributions corresponding to the experimental resolution. (c) DOS of the d-orbital of vanadium obtained using LDA and LDA + DMFT with U = 2 eV and J = 0.6 eV at 300 K.

conduction band DOS or the one-particle spectral function. The XPS spectra of the valence band are seen on the left side in panel (a) of Fig. 1, while panel (b) show results of the empty electronic states up to 8 eV above E_F using BIS [60]. Within the valence band (VB) the two main features are the broad peaks at energies at about -2.4 eV and -0.7 eV. In the conduction band (CB) a hump at about 0.7 eV and a larger peak at 2.3 eV can be seen. The computed total DOS obtained using LDA + DMFT can be seen in Fig. 1(b). In order to compare the experimental and theoretical results for the VB the computed DOS is multiplied by the Fermi-Dirac distribution $f_{FD}(E,T)$ and for the CB the multiplication is performed with the factor $[1 - f_{FD}(E, T)]$. In addition, the spectra of VB and CB are convoluted with a constant Lorentzian broadening of 0.1 eV followed by a Gaussian broadening of 0.55 eV (0.7 eV) corresponding to the experimental resolution [53]. After the application of broadening and convolution functions to the full DOS the theoretical spectra are significantly smoothed out. Nevertheless, they capture the formation of the pseudogap above the Fermi energy E_F and the position of the main peaks in the spectral function. Note also the presence of tails in the LDA + DMFT DOS at high binding energies ($E - E_F \in$ [-8, -4] eV) which cannot be captured in plain LDA as these are characteristic features of quantum fluctuations seen in many transition metal elements. A particular example is the formation of the so-called -6 eV satellite in Ni [61–63].

Figure 1(c) shows a direct comparison of LDA and LDA + DMFT densities of states. The gray area corresponds to the LDA + DMFT total d-orbital DOS while the blue and orange lines are the LDA V-d t_{2g}/e_g -orbital DOS. Note that within LDA the d-electron bandwidths of the t_{2g} and e_g orbitals have similar energy extension. At the Fermi level a dominant d- t_{2g}

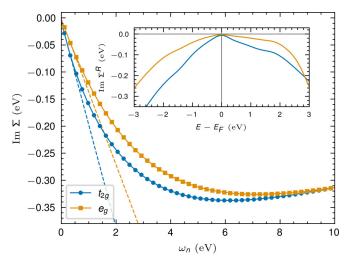


FIG. 2. Imaginary part of the self-energies as a function of Matsubara frequencies of the vanadium t_{2g} and e_g orbitals obtained from LDA + DMFT with U = 2 eV and J = 0.6 eV at 400 K. The inset shows the parabolic behavior of the imaginary part of the retarded self-energy around E_F , indicating Fermi liquid behavior.

contribution can be seen. The negative slope of the real part of the self-energy in the LDA + DMFT computation leads to the slight compression of spectra toward the Fermi level, which is a characteristic Fermi liquid behavior. This effect is easily visible as the position of the d- e_g -orbital peak in the vicinity of -2.5 eV in the valence band is shifted closer to about -2 eV in the LDA + DMFT results, and at the same time a more substantial DOS is obtained at the Fermi level. The unoccupied part dominated by the e_g -DOS peak at about 2.7 eV is slightly shifted in this case away from the Fermi level, which indicates that a mean-field Hartree-Focklike contribution dominates dynamic local correlations in this energy range. Further discrepancies between the measurement and computed results in both occupied and unoccupied parts of the spectra have been previously addressed [53]. It was concluded that alternative computations such as LDA + U fail to describe the weight of computed spectra situated between the main peaks. It was also emphasized that electron-electron interactions described by variants of GW and LDA + DMFT bring the computed spectra in fairly good agreement with the experiment [53].

In the following, we discuss the effects of dynamical correlations based on the results of the electronic self-energies. In Fig. 2 we show the imaginary part of the orbitally resolved self-energies. The inset shows the analytically continued retarded self-energies on the real axis which follows a quadratic energy dependence in the vicinity of the Fermi level: $\text{Im}\Sigma^R(E) \propto -(E-E_F)^2$. Both the t_{2g}/e_g orbitals show a Fermi liquid behavior. However, as expected the effective masses are different. The analytically continued self-energies are obtained by using Padé-approximate methods [64,65]. To avoid the possible inaccuracies of the analytical continuation, we interpret the computational results using the self-energy on the Matsubara axis. The Fermi liquid state is characterized by a linear dependence of the imaginary part of the self-energy on imaginary frequencies

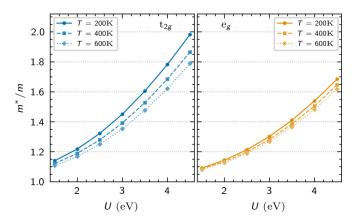


FIG. 3. Interaction strength dependence of the effective mass renormalization of the t_{2g} and e_g orbitals of vanadium for different temperatures.

 ω_n : Im $\Sigma(\omega_n) = -\Gamma(\omega_n) - (Z^{-1} - 1)\omega_n$, where Z is the quasiparticle spectral (inverse of the effective mass ratio m/m^*) weight and $\Gamma(\omega_n)$ the quasiparticle damping. In the region of low Matsubara frequencies Im $\Sigma_{t_{2g}/e_g}(\omega_n)$ approaches zero linearly (dashed lines Fig. 2) for both orbitals. Thus, the t_{2g}/e_g electrons are long-lived quasiparticles; i.e., their scattering rate vanishes as the Fermi surface is approached. In the entire temperature range $100 \text{ K} \leqslant T \leqslant 600 \text{ K}$ the linearity of the imaginary part of the self-energy is preserved; thus the results provide a clear indication of the Fermi liquid character of the electronic system.

Within DMFT the quasiparticle mass enhancement m^*/m , relative to the band mass m, becomes local and can be approximately computed from the Matsubara frequencies in the zero-temperature limit as $m^*/m = 1 - \text{Im}\Sigma(\omega_n)/\omega_n|_{\omega_n \to 0}$. It is often computed numerically in quantum Monte Carlo (QMC) calculations as the slope of the imaginary part of the self-energy for a few low Matsubara frequencies [19]. In Fig. 2 dashed lines show the slopes for the t_{2g} and e_g . The results are presented in Fig. 3 for various Hubbard parameters up to U = 5 eV. In computations with U above 2 eV we used a constant J = 0.6 eV, while for U < 2 eV we kept a constant $U/J = 2/0.6 \approx 3.33$ ratio to ensure the positiveness of the effective Coulomb repulsion U - 3J. Increasing the value of the Hund's coupling up to J = 0.9 eV no significant change in the spectral function occurs. For all temperatures the effective mass of both orbitals is increasing with increasing U values. At higher temperatures we estimate a somewhat smaller value. As seen from Fig. 3 the effective mass enhancements m^*/m are in the range $1 \dots 2$, with an increasing tendency for larger U values. Enhancements in this range are characteristic for medium-correlated electronic systems, in contrast to heavyfermion or strongly correlated systems for which m^*/m can be of the order of hundreds or thousands [66,67]. Thus our results indicate the presence of moderate dynamical correlation effects for both orbitals, but for the e_g electrons the effect is less substantial.

Deviations from the Fermi liquid behavior and appearance of nonanalytic frequency dependence of the self-energy for the t_{2g} states have been recently discussed in Ref. [52]. Unlike our computation, which is based on a perturbative

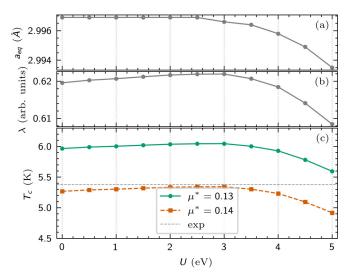


FIG. 4. (a) Computed equilibrium lattice parameter $a_{\rm eq}$, (b) electron-phonon coupling strength λ , and (c) the superconducting critical temperature T_c with respect to the interaction strength U of vanadium. The dashed line indicates the experimental T_c . The experimental lattice parameter is $a_{\rm exp} = 3.02$ Å; it is not shown as it falls outside the shown scale.

impurity solver, the strong dynamical correlations are more precisely captured by the continuous-time QMC solvers [68] used in Ref. [52]. In our computations we did not address the high-temperature and high-U limit. Nevertheless, for the relevant temperatures and U range (between 2 and 3 eV) which correctly capture Fermi surface features [29] our results agree well with the QMC results [52], as well as with the LDA+FLEX calculations [54] reported earlier.

B. McMillan estimates for the superconducting transition temperature of pure V

Ab initio calculations of superconducting temperatures have been attempted using the Eliashberg equation and produced prominent results [45–47]. These computations did not include the presence of dynamic electronic correlations in disordered alloys. However, the combined dynamical electronic correlation and disorder problem may be considered directly in the McMillan theory, which provides reliable estimates in such cases.

Figure 4 summarizes the results of the LDA + DMFT computations: the equilibrium lattice parameter is seen in subplot (a), the mass-enhancement factor λ is shown in panel (b), and finally panel (c) shows the variation of the superconducting temperature. Computations were performed for U values up to 5 eV and for a constant J=0.6 eV. The values for U=0 eV correspond to the LDA results. All quantities presented in Fig. 4 have a slight variation with the strength of the Coulomb interaction U. The equilibrium lattice constant does not change significantly up to 3 eV and for larger U values has a descending trend. For the value U=5 eV a reduction of about 2% is obtained. It is interesting to note that the mass enhancement factor λ reaches a maximum value of 0.622 for the realistic U values in the range from 2 eV to 3 eV. For higher U values λ decreases as well.

With the first-principles estimates $N(E_F)$, λ and the experimental data of Debye temperatures (θ_D) the values of T_c are obtained choosing a suitable value for μ^* . McMillan suggested taking μ^* in the vicinity of 0.13 for transition metal elements [5]; however, one finds different μ^* values in the current literature. References [56,69] observed a range of μ^* between 0.11 to 0.14 computed via various forms of local field correction functions [69], or have chosen values of μ^* between 0.12 and 0.175 to match their results to experimental data [56]. Figure 4(c) shows our estimates for T_c : choosing a value of $\mu^* = 0.13$, our results for pristine vanadium slightly overestimate the transition temperature, whereas for a value of $\mu^* = 0.14$ a better match with experiment is found. Although the transition temperature T_c is shifted for different choices of the effective Coulomb repulsion parameter μ^* , the interaction strength dependence of T_c shows the same trend: a mild increase with U, a maximum value in the range of U in which λ is maximum, and a decreasing trend for higher U values. Our overall results of LDA + DMFT computations with the Hubbard parameter values U in the range of 2 eV to 3 eV and $\mu^* = 0.14$ provide T_c next to the experimental value $T_c = 5.73$ K, the dashed line of Fig. 4(c). Note also that the same Hubbard U values proved to provide an excellent description of the Fermi surface of vanadium [29].

C. Binary Ti-V alloys

 $Ti_{r}V_{1-r}$ alloys exist in different metallurgical phases. For concentrations $0 \le x \le 0.68$ the β phase is observed with a bcc lattice structure. The α phase is formed for high concentrations of Ti, $0.86 \le x \le 1$, in which the system has a hexagonal close-packed (hcp) structure. For concentrations in the range of 0.67 < x < 0.86 an additional hexagonal ω phase (precipitation) can occur apart from the α and β phases [56]. The ω phase is a primitive hexagonal crystal structure that has been found to be a common metastable phase in body-centered-cubic metals and alloys. In general, the ω phase precipitates and has a coherent interfacial structure with its bcc matrix phase, with the parameters $a_{\omega} = \sqrt{2}a_{\rm bcc}$ and $c_{\omega} = \sqrt{3/2}a_{\rm bcc}$. In such a mixed phase an anomalous increase in electrical resistivity with decreasing temperatures was reported [70]. This behavior was addressed invoking various models including weak localization [70], Kondo s-d interactions, and localized spin fluctuation [71,72]. In addition, for concentrations $x \le 50\%$ [56] an increase in the superconducting T_c has been observed. As previously noted [56] this effect cannot be associated with the change in the Debye temperature as Ti and V atoms are situated beside each other in the periodic table. It is our aim to investigate whether dynamic electronic correlation effects contribute to the enhancement of the T_c .

We consider in the following the bcc β phase in which Ti atoms occupy randomly the V 2a site with concentrations up to 65%. We have computed the total energy of the system for different U and x values and determined the equilibrium lattice parameter $a_{\rm eq}$. After calculating multiple $a_{\rm eq}$, a linear concentration dependence of the equilibrium lattice parameter could be observed, which is also known as Vegard's law [73]: $a_{\rm eq} = x \cdot a_{\rm eq}^{\rm Ti} + (1-x) \cdot a_{\rm eq}^{\rm V}$, where $a_{\rm eq}^{\rm V}$ is the equilibrium lattice constant of pure vanadium and $a_{\rm eq}^{\rm Ti}$ is the (extrapolated)

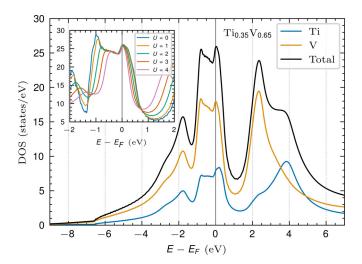


FIG. 5. Total and sublattice DOS of $Ti_{0.35}V_{0.65}$ obtained using LDA + DMFT with U = 2 eV and J = 0.6 eV at 400 K. The inset shows the total DOS for different interaction strengths U around E_F .

equilibrium lattice constant of pure titanium in the bcc phase. The extrapolated value of $a_{\rm eq}$ for a pure titanium crystal in the bcc β phase is $a_{\rm eq}^{\rm Ti}=3.231$ Å.

Figure 5 shows the electronic DOS of the $Ti_x V_{1-x}$ alloy for x = 0.35 concentration at which the superconducting critical temperature is the highest among all $Ti_x V_{1-x}$ alloys. Independent of the impurity concentration x, the electronic structure is dominated by broad d-electron bands with a specific pseudogap in the DOS as a consequence of the bcc structure. As can be seen in Fig. 5 most of the contribution at the Fermi level is provided by the V-d states. The inset shows a comparison of LDA + DMFT results for DOS for various U values. Similarly to the computations for vanadium, we used a constant J = 0.6 eV for U above 2 eV, while for U < 2 eV we kept the ratio $U/J = 2/0.6 \approx 3.33 \text{ constant}$. We applied DMFT to both Ti and V components using the same Hubbard U value. The computation with U=0 corresponds the LDA-CPA calculation. One can see a clear tendency: a tiny reduction of DOS at the Fermi level and a narrowing of the d bands in the vicinity of the Fermi level.

In Fig. 6 we present the imaginary part of the self-energies of V and Ti in the alloy $\text{Ti}_x \text{V}_{1-x}$ with x=0.35 in the low-energy Matsubara region. A linear behavior can be seen, which demonstrates that the Fermi liquid character of d electrons is preserved also in the presence of site disorder. We estimated the quasiparticle weights for the different concentrations for both alloy components.

A quantitative analysis of the concentration dependence of the effective masses at temperatures 200, 400, and 600 K is presented in Fig. 7. The values of the effective mass renormalization $Z^{-1} = m^*/m$ for e_g orbitals of both V and Ti show a very weak concentration dependence and are distributed around a value $Z_{\text{Ti}-e_g}^{-1} \approx 1.07$ and $Z_{\text{V}-e_g}^{-1} \approx 1.16$. By contrast, the effective mass of the t_{2g} electrons increases slowly upon doping with Ti. As the temperature is raised a reduction in the magnitude of the effective masses is also visible. Note that a reduction of m^*/m is seen also in pristine V; however, the effect is less significant.

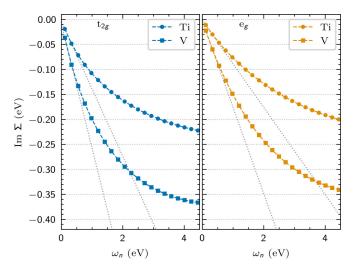


FIG. 6. Imaginary part of the self-energies as a function of Matsubara frequencies of the t_{2g} and e_g orbitals of $\text{Ti}_{0.35}\text{V}_{0.65}$ for U=2~eV, J=0.6~eV at 400 K.

A characteristic phenomenon of disordered and strongly correlated electron systems is the decoherence of quasiparticles. In the presence of disorder a resistivity change is observed above some decoherence temperature which is connected with the quasiparticle weight renormalization [74]. Calculations performed within the framework of DMFT support this model [75,76]. Thus, we interpret our LDA + DMFT results for Ti_xV_{1-x} alloys using the same picture: in the presence of a low concentration of Ti impurities at V sites dynamical correlations, through the real part of the self-energy (see Fig. 8), renormalize the on-site energies (the random on-site potentials): $\epsilon_{t_{2g}/e_g} + \text{Re}\Sigma_{t_{2g}/e_g}^R(E)$. For a given energy E in the vicinity of E_F the screening effect is more important for V-d orbitals in comparison with those of Ti-d orbitals. In a similar analysis for the orbitally resolved selfenergies one observes that in the occupied part of the spectra

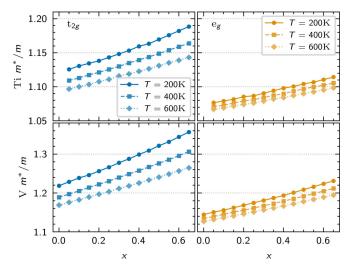


FIG. 7. Concentration dependence of the effective mass renormalization of the t_{2g} and e_g orbitals for the alloy components Ti and V of Ti_xV_{1-x} with U = 2 eV and J = 0.6 eV at different temperatures.

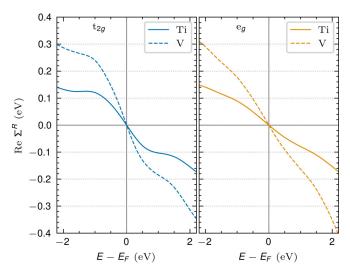


FIG. 8. Real part of the retarded self-energies as a function of energy of the t_{2g} and e_g orbitals of ${\rm Ti}_{0.35}{\rm V}_{0.65}$ for $U=2~{\rm eV}, J=0.6~{\rm eV}$ at 400 K.

 $(E \leqslant E_F) \; \mathrm{Re} \, \Sigma^R_{t_{2g}}(E) > \mathrm{Re} \, \Sigma^R_{e_g}(E),$ while in the unoccupied part the spectra the opposite is true: $\mathrm{Re} \, \Sigma^R_{t_{2g}}(E) < \mathrm{Re} \, \Sigma^R_{e_g}(E).$ Consequently the screening effect is more/less efficient for the t_{2g} orbitals below/above the Fermi level. We did not notice significant change in the self-energies. Our computations show that as temperature is increasing the screening of the random potential does not change significantly. Thus, with increasing temperature the t_{2g} quasiparticle scattering becomes less coherent and the effective masses slightly increase in comparison with the e_g electron quasiparticles.

D. Enhanced critical temperatures in the pure β and α phases

To apply the McMillan formula Eq. (3) the parameters such as the atomic masses and Debye temperatures were taken from Ref. [77]. For the latter the concentration dependence has been estimated as the algebraic average of the corresponding Debye temperatures by the relation $\theta_D(x) = x \cdot \theta_D^{Ti} + (1-x) \cdot \theta_D^V$. The dependence of the lattice parameter with concentration follows Vegard's law.

Figure 9 shows the concentration dependence of the critical temperature for the $\text{Ti}_x \text{V}_{1-x}$ alloys. The critical temperature reaches a maximal value (independent of the interaction strength) at a concentration of $x \approx 0.33$. The inset of Fig. 9 shows the critical temperature T_c in dependence on the interaction strength U for the concentration around the maximum of T_c and for two concentrations below and above. Similarly to the behavior of pristine vanadium in Fig. 4, the critical temperature rises slightly with increasing U before it drops for strong interactions of about U > 3 eV.

LDA + DMFT computations were also performed for the α (hexagonal) phase of $\text{Ti}_x \text{V}_{1-x}$. Similarly to vanadium, we have optimized the lattice parameter a of the closed-packed hexagonal (hcp) lattice structure, while maintaining a constant ratio of c/a. The McMillan approximation reproduces the increasing trend of the critical temperature T_C for small concentrations 1-x of vanadium. Computations within the mixed $\beta + \omega$ and $\alpha + \beta + \omega$ phase require a modeling of

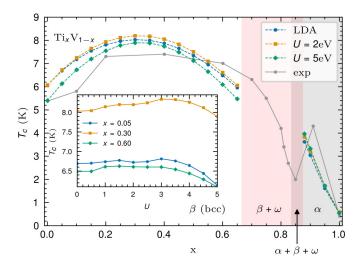


FIG. 9. Concentration dependence of T_c in the β (bcc) and α (hcp) phases of $\mathrm{Ti}_x \mathrm{V}_{1-x}$. The maximal value for T_c can be observed at $x \approx 0.33$ and is independent of U. The inset shows the interaction strength dependence of T_c .

the crystal structure that cannot be addressed in our current methodology.

The McMillan formula of T_c , Eq. (3), emerges from the strong-coupling theory with the additional rigid-ion and oneelectron approximations. The key physical quantity accessible also through the *ab initio* computation is the mass enhancement factor λ which contains the electron-phonon matrix element averaged over the Fermi surface (see the discussion in Sec. II B).

LDA + DMFT computations show that dynamic electronic correlation in vanadium-titanium alloys leads to narrower DOS around the Fermi level (Fig. 5) and an increased effective mass (see Fig. 7). For this reason quasiparticles with energy close to the d resonance of the electron-ion interaction spend longer times in the vicinity of the scatterer (the so-called Wigner delay time). Therefore, it is reasonable to suppose that the scattering of electrons is dominated by the screened local potential, namely, the potential change due to the displacement of an ion and the additional real part of the local self-energy. Thus, the local dynamic correlations lead to a modified electron-phonon coupling which may finally lead to an increase in T_c .

Figure 10 shows a surface plot of $\lambda(U,x)$ at temperature T=400 K. For a given U value the λ increases, reaching a maximum at a concentration about $x\approx 0.33$ then following a descending trend. The behavior of λ provides a coherent picture since it yields a good estimate for the concentration dependence of the critical temperature seen in Fig. 9.

IV. SUMMARY

We computed the superconducting critical temperature of pure V and Ti_xV_{1-x} alloys within the $\beta(bcc)$ and α phases, on the basis of BCS-Eliashberg theory. This theory describes superconductivity caused by an effective electron-electron attraction resulting from the electron-phonon interaction that overcomes the repulsive Coulomb interaction. The magnitude of the critical temperature is limited by the fact that the

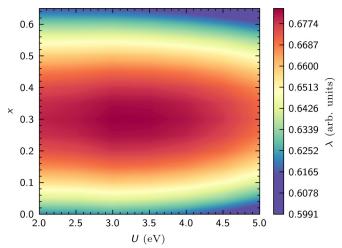


FIG. 10. Surface plot of $\lambda(U,x)$ for $\mathrm{Ti}_x\mathrm{V}_{1-x}$ alloys computed with LDA + DMFT at 400 K.

phonon energy scale is much lower than the electronic energy scales (which, in fact, validates the theory). In particular, we estimated the critical temperatures T_c employing McMillan's formula in which the experimental Debye temperature, the electron-phonon coupling constant λ (obtained through *ab initio* computation), and the pseudopotential parameter μ^* are used. We did not attempt to use more advanced formulas for T_c (for example, the Allen-Dynes prescription [78]) involving the phonon band structure and the moments of the Eliashberg function as our aim was to model the enhancement of T_c in Ti_xV_{1-x} alloys for which *ab initio* phonon computations would have involved very expensive supercells. Instead, we modeled the dynamical electronic correlations in the studied alloys using the LDA + DMFT method.

As shown previously [29] a correct description of the Fermi surface properties of V can be achieved on the basis of LDA + DMFT methods which indicated that vanadium is a conventional correlated Fermi liquid. Obviously this does not exclude the possibility that in the regime of very high T and large Hubbard U departures from the Fermi liquid behavior may occur [52]. A quantitative comparison with computations involving nonlocal dynamic correlation effects, LDA+FLEX [54] or GW, showed an excellent agreement with LDA + DMFT results; in particular, the mass enhancements practically coincide.

Theoretical investigations of disorder in materials including superconductors are usually performed considering the existence of uncorrelated disorder. In real systems, however, the disorder is almost never completely random. Inhomogeneities may lead to the appearance of short- or long-range spatial correlations, and the modeling of such effects is far more complicated and requires approaches beyond the CPA that are not discussed here. In our work we considered the regular CPA supplemented by the modeling of dynamical electronic correlations in the normal state of the superconducting $\text{Ti}_x \text{V}_{1-x}$ alloys. In particular, we computed the spectral functions (the *local* DOS) of the alloy component self-energies and the mass enhancements.

Our critical temperature estimates for the $\text{Ti}_x \text{V}_{1-x}$ alloys with concentrations x in the range of single β or α phases are

in very good agreement with the experimental measurements. In the rich V region (small x) a maximum in the concentration dependence is obtained for $x \approx 0.33$. Although CPA results also indicate such a T_c increase, we show that dynamical electronic correlations modeled with DMFT and a Hubbard U parameter are expected to improve the accuracy of the description.

Early experimental studies interpreted the anomalous behavior of resistivity measurements of certain V-Ti alloys invoking weak localization effects [70]. The recently implemented TMT into the DFT framework allows one to search for the existence of precursors of localization [27] in alloys with substitutional disorder. Within the DFT-TMT framework the *typical* DOS is obtained as a geometrical average of the local DOS of the alloy components. A vanishing value of the typical DOS is considered to signal the presence of weak localization effects [24,27,44]. We have obtained a nonzero typical DOS; thus, no signature of localization was found for concentrations in which alloys are formed with single phases (α and β). Our study does not cover the concentration range $0.65 \le x \le 0.9$ in which a chemically disordered composition with mixed crystal structure phases, $\beta + \omega$ and $\alpha + \beta + \omega$,

exists as discussed in the beginning of Sec. III C. In this concentration range a direct LDA + DMFT computation is a major challenge because of limitations in computational time and simulation size. We expect that more advanced multiscale techniques and possible artificial intelligence methodologies such as machine learning may contribute significantly to data-driven estimations of T_c for disordered superconducting alloys.

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