# Weak effects of electron-phonon interactions on the lattice thermal conductivity of wurtzite GaN with high electron concentrations

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Wurtzite gallium nitride (GaN) has great potential for high-frequency and high-power applications due to its excellent electrical and thermal transport properties. However, enhancing the performance of GaN-based power electronics relies on heavy doping. Previous studies showed that electron-phonon interactions have strong effects on the lattice thermal conductivity of GaN due to the Fröhlich interaction. Surprisingly, our investigation reveals weak effects of electron-phonon interactions on the lattice thermal conductivity of *n*-type GaN at ultrahigh electron concentrations and the impact of the Fröhlich interaction can be ignored. The small phonon-electron scattering rate is attributed to the limited scattering channels, quantified by the Fermi surface nesting function. In contrast, there is a significant reduction in the lattice thermal conductivity of *p*-type GaN at high hole concentrations due to the relatively larger Fermi surface nesting function. Meanwhile, as *p*-type GaN has relatively smaller electron-phonon matrix elements, the reduction in lattice thermal conductivity is still weaker than that observed in *p*-type silicon. Our work provides a deep understanding of thermal transport in doped GaN and the conclusions can be further extended to other wide-band-gap semiconductors, including

 $\beta$  – Ga<sub>2</sub>O<sub>3</sub>, AlN, and ZnO.

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## I. INTRODUCTION

GaN has a high Baliga figure of merit and large lattice thermal conductivity, which makes it promising for power electronics [1] and optoelectronics [2]. Generally, high charge carrier concentrations are required to obtain large electrical conductivity in GaN. For instance, when the hole concentration is increased to  $2 \times 10^{18}$  cm<sup>-3</sup> doping with magnesium and oxygen atoms, the resistivity is reduced to  $0.2 \ \Omega \, \text{cm}$ [3]. Similarly, for *n*-type GaN, a high electrical conductivity exceeding  $4 \times 10^3 \ \Omega^{-1} \text{ cm}^{-1}$  is achieved with an electron concentration of  $3.7 \times 10^{20}$  cm<sup>-3</sup> through germanium doping [4]. However, high-concentration doping can have a negative influence on the lattice thermal transport. An extra phonon scattering term, namely phonon-electron scattering, is introduced when there are plenty of charge carriers. This may lead to a pronounced reduction in lattice thermal conductivity, which is harmful to the performance of power semiconductor devices [5].

Thanks to the giant advancement in computational capability, it is possible to rigorously consider the electron-phonon interactions on the lattice thermal conductivity from firstprinciples calculations [6–8]. Liao *et al.* found that there is a significant reduction of  $\sim 37\%$  in lattice thermal conductivity for *n*-type silicon (Si) with an electron concentration of  $10^{21}$  cm<sup>-3</sup> [9]. Their predictions were later verified by the three-pulse femtosecond photoacoustic technique [10]. Similar trends were also observed in two-dimensional semiconductor materials, such as  $MoS_2$  and PtSSe [11] and phosphorene and silicene [12]. Notably, GaN holds a much larger lattice thermal conductivity than Si because there is a large acoustic-optical phonon frequency gap which greatly limits available phonon-phonon scattering channels [13]. Given the weak phonon-phonon scattering, the phononelectron scattering is expected to play a crucial role in GaN at high carrier concentrations, as observed in another wideband-gap semiconductor 3C-SiC [14]. Previous studies by Yang et al. and Tang et al. reported that the lattice thermal conductivity of GaN is severely limited by the strong Fröhlich interaction between electrons and long-wavelength longitudinal optical phonons [15,16]. However, recent experiments showed that the thermal conductivity of *n*-type GaN is almost a constant within the concentration range of  $10^{17}$  to  $10^{19}$  cm<sup>-3</sup> [17,18]. On the theoretical side, the lattice thermal conductivity of GaN matches experimental results better when accounting for four-phonon scattering [19]. It should be noted that electron-phonon interactions are not included in Ref. [19]. Therefore, the effect of electron-phonon interactions on the lattice thermal conductivity of GaN is still under debate.

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In this work, the lattice thermal conductivities of *n*-type and *p*-type wurtzite GaN (space group  $P6_3mc$ , No. 186) under different carrier concentrations are investigated through mode-level first-principles calculations. It is found that the electron-phonon interactions have weak effects on the lattice thermal conductivity of *n*-type GaN even at an ultrahigh electron concentration. The impact of the Fröhlich interaction on lattice thermal conductivity can be ignored, which contradicts the previous understanding on the thermal transport in doped GaN, whereas there is a significant reduction in the lattice thermal conductivity of *p*-type GaN at high hole concentrations. We provide a comprehensive analysis of the contribution terms ascribed to electron-phonon interactions, including the electron density of states (DOS), Fermi surface nesting function, and electron-phonon matrix elements.

#### **II. THEORY AND METHODS**

Combining the linearized phonon BTE and Fourier's law, the lattice thermal conductivity ( $\kappa_{lat}$ ) can be expressed as [20]

$$\kappa_{\text{lat},\alpha\beta} = \sum_{\lambda} c_{\nu,\lambda} v_{\lambda,\alpha} v_{\lambda,\beta} \tau_{\lambda} = \frac{1}{\Omega} \sum_{\lambda} \hbar \omega_{\lambda} \frac{\partial n_{\lambda}}{\partial T} v_{\lambda,\alpha} v_{\lambda,\beta} \tau_{\lambda},$$
(1)

where  $\alpha$  and  $\beta$  are the Cartesian coordinates,  $\lambda \equiv (\mathbf{q}, \nu)$ denotes the phonon mode with wave vector  $\mathbf{q}$  and phonon polarization  $\nu$ ,  $c_{\nu,\lambda}$  is the phonon specific heat capacity,  $v_{\lambda}$ is the phonon group velocity, and  $\tau_{\lambda}$  is the phonon relaxation time.  $\Omega$  is the volume of the primitive cell,  $\hbar$  is the reduced Planck's constant,  $\omega_{\lambda}$  is the phonon frequency, and  $n_{\lambda}$  is the Bose-Einstein distribution at temperature *T*.

The essential step is to obtain  $\tau_{\lambda}$ , which is associated with several scattering processes. The effective phonon scattering rates can be obtained using Matthiessen's rule:  $1/\tau_{\lambda} = 1/\tau_{\lambda}^{\text{ph-ph}} + 1/\tau_{\lambda}^{\text{ph-el}} + 1/\tau_{\lambda}^{\text{ph-el}}$ , where  $1/\tau_{\lambda}^{\text{ph-ph}}$  is the phonon-phonon scattering rates,  $1/\tau_{\lambda}^{\text{ph-iso}}$  is the phonon-isotope scattering rates, and  $1/\tau_{\lambda}^{\text{ph-el}}$  is the phonon-electron scattering rates.

According to Fermi's golden rule, the phonon-phonon scattering rates can be expressed as [21]

$$\begin{aligned} \frac{1}{\tau_{\lambda}^{\text{ph-ph}}} &= 2\pi \sum_{\lambda_1 \lambda_2} |V_{\lambda \lambda_1 \lambda_2}|^2 \\ &\times \left[ \frac{1}{2} \left( 1 + n_{\lambda_1}^0 + n_{\lambda_2}^0 \right) \delta(\omega_{\lambda} - \omega_{\lambda_1} - \omega_{\lambda_2}) \right. \\ &+ \left( n_{\lambda_1}^0 - n_{\lambda_2}^0 \right) \delta(\omega_{\lambda} + \omega_{\lambda_1} - \omega_{\lambda_2}) \right], \end{aligned}$$

where  $V_{\lambda\lambda_1\lambda_2}$  denote the three-phonon scattering matrix element.  $\delta$  is the Dirac delta function which ensures the conservation of energy during the scattering processes. Recent studies have demonstrated that the four-phonon scattering has strong effects on lattice thermal conductivity at room temperature [22–26]. To evaluate the four-phonon scattering effects on  $\kappa_{\text{lat}}$  of GaN, we calculated the weighted phase space of three-phonon and four-phonon scattering. As shown in Fig. S1 of the Supplemental Material [27] (see also Refs. [28–33] therein), the weighted phase space of four-phonon scattering is significantly smaller than that for three-phonon scattering.

Therefore, the four-phonon scattering has negligible effects on the lattice thermal conductivity of GaN at room temperature. The phonon-isotope scattering rates are evaluated by the Tamura theory and the details can be found in Ref. [34]. The phonon-electron scattering rates are related to the imaginary part of the phonon self-energy, which can be expressed as [35]

$$\frac{1}{\tau_{\lambda}^{\text{ph-el}}} = -\frac{2\pi}{\hbar} \sum_{mn,\mathbf{k}} \left| g_{mn}^{v}(\mathbf{k},\mathbf{q}) \right|^{2} (f_{n\mathbf{k}} - f_{m\mathbf{k}+\mathbf{q}}) \\ \times \delta(\varepsilon_{m\mathbf{k}+\mathbf{q}} - \varepsilon_{n\mathbf{q}} - \hbar\omega_{\lambda}), \tag{3}$$

where  $g_{mn,v}(\mathbf{k}, \mathbf{q})$  is the electron-phonon matrix element, which quantifies probability amplitude for scattering between the electronic state  $n\mathbf{k}$  and  $m\mathbf{k} + \mathbf{q}$ , f is the Fermi-Dirac distribution function,  $\varepsilon$  is the electron energy, and  $\varepsilon_F$  is the Fermi energy

The first-principles calculations are performed using the QUANTUM ESPRESSO package [36]. The electron exchangecorrelation functional is treated by the generalized gradient approximation of Perdew, Burke, and Ernzerhof (PBE) [37] and optimized fully relativistic norm-conserving pseudopotentials [38] from PSEUDODOJO [39]. The kinetic energy cutoff for plane waves is set to be 100 Ry, and the convergence of electron energy is  $10^{-10}$  Ry. It was shown that the spin-orbit coupling (SOC) has significant effects on the valance band structure in Ref. [40]. Therefore, the SOC is also included in our electronic band structure calculations. The harmonic and cubic force constants are calculated from density-functional perturbation theory. The electron-phonon interactions are first calculated under coarse  $\mathbf{k}/\mathbf{q}$  meshes and then interpolated to dense meshes with the Wannier interpolation technique [41]. The convergence of phonon-electron scattering rates with respect to the k-point mesh is verified in Fig. S2 of the Supplemental Material [27]. The in-house modified D3Q package [42] is employed to calculate the  $\kappa_{lat}$ , incorporating the phonon-electron scattering using the iterative calculation scheme [43-45]. The rigid shift of the Fermi energy is utilized to imitate the change of carrier concentration [46]. More details about the first-principles calculations are provided in the Supplemental Material [27].

#### **III. RESULTS AND DISCUSSION**

Figure 1(a) shows the electron band structure of GaN that includes the SOC effect. The Fermi energy corresponding to different carrier concentrations is represented by horizontal dashed lines. There is only one conduction band in the vicinity of the conduction band minimum (CBM), while there are multiple valance bands near the valence band maximum (VBM). Note that the heavy-hole and light-hole bands are split along the  $\Gamma$ -M high-symmetry path of the first Brillouin zone due to the SOC effects, as shown in Fig. S3(b) of the Supplemental Material [27]. This subtle variation, compared to the electronic band structure without SOC, has non-negligible effects on the electron-phonon interactions and the value of in-plane lattice thermal conductivity ( $\kappa_{1at,a}$ ) of *p*-type GaN, as shown in Table I. Our calculation shows that GaN is an indirect semiconductor with a band gap of 1.86 eV, strongly underestimating the measured value of 3.5 eV [47,48]. This is attributed to the well-known drawback in DFT [49].



FIG. 1. (a) Band structures of GaN along the high-symmetry paths. The horizontal lines are Fermi energy related to the carrier concentrations of  $10^{17}$  (green),  $10^{18}$  (orange),  $10^{19}$  (yellow),  $10^{20}$  (light purple), and  $10^{21}$  (light pink) cm<sup>-3</sup> at room temperature. The electron energy is normalized to the VBM. (b)  $\kappa_{1at,a}$  as a function of carrier concentration at room temperature for undoped, *n*-type, and *p*-type GaN. The scatters are experimental results, reported by Beechem *et al.* [17] and Simon *et al.* [18], respectively.

However, this discrepancy has no influence on our conclusions on  $\kappa_{lat}$  since the profile of our DFT band structure matches well with the experimental results [50] and only the electron modes in the vicinity of the band edges have contributions to electron-phonon interactions. As for the phonon dispersion, our theoretical results agree well with the experimental data (Fig. S4 of the Supplemental Material [27]), which further confirms the reliability of our first-principles calculations.

The  $\kappa_{1at,a}$  as a function of carrier concentrations  $(10^{17}-10^{21} \text{ cm}^{-3})$  for undoped, *n*-type, and *p*-type GaN are shown in Fig. 1(b). The  $\kappa_{1at,a}$  at room temperature for undoped GaN, namely without phonon-electron scattering, is 264 W/mK, which is in good agreement with experimental [51,52] and former theoretical results [13]. With phononelectron scattering included, the  $\kappa_{1at,a}$  falls within the range of 250-258 W/mK for n-type GaN, which is quite close to the value of the undoped case. On the contrary,  $\kappa_{1at,a}$  is dramatically reduced to 228 W/mK and 148 W/mK at the hole concentrations of  $10^{19}$  cm<sup>-3</sup> and  $10^{21}$  cm<sup>-3</sup>, respectively. A similar variation trend is also observed in time-domain thermoreflectance (TDTR) measurement [17,18], where the  $\kappa_{lat}$ of *n*-type GaN is almost a constant within the doping concentration range of  $10^{17}$  to  $10^{19}$  cm<sup>-3</sup>. However, the  $\kappa_{lat}$  of p-type GaN exhibits a noticeable decrease. The quantitative differences between our work and Refs. [17,18] may be attributed to experimental sample sizes, impurities, or defects, which are not considered in our calculations. Recently, Pang et al. found that phonon-defect scattering can also have significant effects on the lattice thermal conductivity of 3C-SiC with B doping [53]. Here we mainly focus on the impact of electron-phonon interactions on lattice thermal conductivity. The phonon-defect scattering effects on the lattice thermal

TABLE I.  $\kappa_{1at,a}$  of *n*-type and *p*-type GaN with and without SOC/polar effects. The carrier concentrations are  $10^{21}$  cm<sup>-3</sup> in all the cases.

$\kappa_{1at,a}$ (300 K)	With SOC	Without SOC	With Polar	Without Polar
<i>n</i> -type	250.34	249.59	250.34	246.51
<i>p</i> -type	147.85	125.51	147.85	146.20



FIG. 2.  $1/\tau_{\lambda}^{\text{ph-ph}}$ ,  $1/\tau_{\lambda}^{\text{ph-iso}}$ , (a)  $1/\tau_{\lambda}^{\text{ph-el}}$ , and (b)  $1/\tau_{\lambda}^{\text{ph-hole}}$  at room temperature with carrier concentration of  $10^{21} \text{ cm}^{-3}$ .

conductivity of GaN are out of the perspective of the present work and will be further discussed in future work.

The effects of electron-phonon interactions on  $\kappa_{lat}$  are significantly different from the observations in Si [9] and 3C-SiC [14]. Both electron and hole have strong interactions with phonons, resulting in a substantial decrease in  $\kappa_{lat}$  [9]. Since the  $1/\tau_1^{ph-ph}$  for phonon frequencies of GaN below 10 THz is smaller than that in Si (Fig. S5 of the Supplemental Material [27]), the nearly flat  $\kappa_{lat}$  with different concentrations of *n*-type GaN is anomalous. To reveal the underlying mechanisms, the mode-level phonon-electron and phononhole scattering rates at the carrier concentration of  $10^{21}$  cm<sup>-3</sup> are calculated, as shown in Figs. 2(a) and 2(b), respectively. The scattering rates at other carrier concentrations are also provided in Fig. S6 of the Supplemental Material [27]. Remarkably,  $1/\tau_{\lambda}^{\text{ph-el}}$  is significantly lower than  $1/\tau_{\lambda}^{\text{ph-ph}}$  for low-frequency phonons (below 10 THz), which have the main contribution to  $\kappa_{lat}$  [54,55]. In contrast,  $1/\tau_{\lambda}^{ph-hole}$  is significantly larger than  $1/\tau_{\lambda}^{\text{ph-ph}}$  within the same frequency range. Note that  $1/\tau_{\lambda}^{\text{ph-el}}$  and  $1/\tau_{\lambda}^{\text{ph-hole}}$  for high-frequency optical phonons (above 15 THz) become even larger due to the Fröhlich interaction in polar materials [56]. This phenomenon is also observed in Ref. [15]. However, optical phonon modes have an ignorable contribution to the  $\kappa_{lat}$  of GaN [54]. Therefore, the strong Fröhlich interaction does not lead to a large reduction in  $\kappa_{lat}$  of GaN. This hypothesis is further verified by the calculated  $\kappa_{1at,a}$  with/without considering the Fröhlich

interaction, as shown in Table I. The distinctions in  $1/\tau_{\lambda}^{\text{ph-el}}$  and  $1/\tau_{\lambda}^{\text{ph-hole}}$  at the same concentration are primarily attributed to the distinctions in electron DOS within the Fermi window regarding *n*-type and *p*-type GaN. GaN becomes a degenerate semiconductor when the carrier concentration reaches  $10^{21}$  cm<sup>-3</sup> for both *n*-type and *p*-type. Therefore, the Fermi energy moves into the conduction band or the valence band. As shown in Fig. 3(a), the DOS for holes is notably larger than that for electrons at  $10^{21}$  cm<sup>-3</sup>. It should be noted that the DOS exhibits a pronounced asymmetric profile. As shown in Fig. S7 of the Supplemental Material [27], the DOS in the vicinity of the VBM is primarily contributed by p orbitals, which exhibit a triple degeneracy and minimal band dispersion. Conversely, the vicinity of the CBM is primarily contributed by s orbitals, resulting in nondegenerate and dispersive electron bands [57-59]. According to Eq. (3) and its variant based on the double-delta approximation [44,60,61], the DOS determines the Fermi surface nesting function  $(\zeta_q)$  to a large extent, which



FIG. 3. (a) Electron DOS near the valence- and conduction-band edges. The purple dotted curve represents the Fermi window. The electron energy is normalized to the VBM. The position of the Fermi energy for electron and hole concentrations of  $10^{21}$  cm<sup>-3</sup> is indicated with black dashed lines. Room-temperature phonon linewidth  $\Gamma_{pe}$  of TA1, TA2, and LA along the high-symmetry path due to (b) phonon-electron and (c) phonon-hole scattering with carrier concentration of  $10^{21}$  cm<sup>-3</sup>. The Fermi surface nesting function is inserted to the top left in (b) and (c).

is expressed as

$$\zeta_{\mathbf{q}} = \sum \delta(\varepsilon_{n\mathbf{k}} - \varepsilon_F) \delta(\varepsilon_{m\mathbf{k}+\mathbf{q}} - \varepsilon_F).$$
(4)

 $\zeta_{\mathbf{q}}$  quantifies the phonon-electron scattering channels [62]. The phonon linewidth with respect to the phonon-electron scattering ( $\Gamma_{pe} = 1/2\tau_{\lambda}^{ph-el}$ ) and  $\zeta_{\mathbf{q}}$  of the two transverse acoustic phonon branches (TA1 and TA2), as well as the longitudinal acoustic phonon branch (LA), are shown in Figs. 3(b) and 3(c). The  $\zeta_{\mathbf{q}}$  in the vicinity of the  $\Gamma$  point is quite large due to the collinearity of electron group velocities  $\mathbf{v}_{\mathbf{k}}$  and  $\mathbf{v}_{\mathbf{k}} + \mathbf{q}$  [63]. The  $\zeta_{\mathbf{q}}$  of *n*-type GaN is much lower than that of *p*-type GaN, which results in its relatively smaller  $\Gamma_{pe}$ . Therefore, the electron-phonon interactions have weak effects on that of *p*-type GaN. This conclusion can be further extended to other wide-band-gap semiconductors, like  $\beta - \text{Ga}_2\text{O}_3$ , AlN, and ZnO [64], whose electron DOS is relatively larger than the hole DOS.

We further compare the  $\kappa_{lat}$  of Si and GaN for both *n*-type and *p*-type at the carrier concentration of  $10^{21}$  cm<sup>-3</sup>. The  $\kappa_{lat}$ of *n*-type Si is reduced by ~37% [9] due to phonon-electron scattering, while the *n*-type GaN is reduced only by 2.85% in our calculation. As shown in Fig. 4(a),  $1/\tau_1^{\text{ph-el}}$  is larger than  $1/\tau_{\lambda}^{\rm ph-ph}$  for plenty of low-frequency phonons in Si. As a contrast, fewer phonon modes have larger  $1/\tau_{\lambda}^{\rm ph-el}$  than  $1/\tau_{\lambda}^{\rm ph-ph}$ in GaN. This discrepancy can be attributed to differences in electron DOS. As shown in Fig. 4(b), the electron DOS of Si is much larger than that of GaN at the electron concentration of  $10^{21}$  cm<sup>-3</sup>. Note that the hole DOS of GaN is close to that of Si at the hole concentration of  $10^{21}$  cm<sup>-3</sup>. Nevertheless, the  $\kappa_{lat}$  of Si is decreased by 45%, while that of GaN is decreased by a smaller value of 29%. As shown in Fig. S8 of the Supplemental Material [27], the  $\zeta_{\mathbf{q}}$  of *p*-type GaN is much larger than that of *p*-type Si, which cannot interpret the discrepancy in their  $\kappa_{lat}$ . According to Eq. (3), the  $1/\tau_{\lambda}^{\rm ph-el}$ is also related to electron-phonon matrix elements *g*, which quantifies the coupling strength between phonon modes and electron states,

$$g_{mn}^{\nu}(\mathbf{k},\mathbf{q}) = \sqrt{\frac{\hbar}{2\omega_{\lambda}}} \langle \psi_{m\mathbf{k}+\mathbf{q}} | \partial_{\lambda} V | \psi_{n\mathbf{k}} \rangle, \qquad (5)$$

with  $\psi$  the ground-state Bloch wave function and  $\partial_{\lambda}V$  the first-order derivative of the Kohn-Sham potential with respect to the atomic displacement. As shown in Fig. 4(c), the magnitude of |g| for Si is larger than that for GaN, resulting in the smaller  $\kappa_{\text{lat}}$  for Si. Recent studies have demonstrated that



FIG. 4. (a) The ratio of  $1/\tau_{\lambda}^{ph-el}$  to  $1/\tau_{\lambda}^{ph-ph}$  with  $10^{21}$  cm<sup>-3</sup> electron concentration of GaN and Si, which is used to identify the dominant scattering term in phonon thermal transport. The horizon green line marks the equal importance of  $1/\tau_{\lambda}^{ph-el}$  and  $1/\tau_{\lambda}^{ph-ph}$ . (b) The DOS for GaN and Si. The positions of the Fermi energy for the electron and hole concentrations of  $10^{21}$  cm<sup>-3</sup> are indicated with black dashed lines for GaN and Si. The purple and red dotted curves represent Fermi windows. (c) Absolute value of electron-phonon matrix elements |g| of GaN and Si.

a 2% biaxial tensile strain can increase the hole mobility of GaN by 230% [40,65], which indicates great promise for the application of *p*-type GaN in power electronics. However, our findings reveal that electron-phonon interactions can significantly limit  $\kappa_{\text{lat}}$  of *p*-type GaN. Therefore, it is not only urgent to synthesize high-quality *p*-type GaN samples but also important to overcome the challenges in the thermal management of power electronics based on *p*-type GaN in the future.

## **IV. CONCLUSIONS**

In summary, the lattice thermal conductivities of *n*-type and *p*-type GaN are investigated through mode-level first-principles calculations. The effects of electron-phonon interactions on the lattice thermal conductivity are shown to be weak in *n*-type GaN, even at an ultrahigh electron concentration of  $10^{21}$  cm<sup>-3</sup>. Intriguingly, our findings indicate that the Fröhlich interaction has an ignorable influence on the lattice thermal conductivity. The weak phonon-electron scattering is attributed to the limited scattering channels, which

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is reflected by the Fermi surface nesting function. In addition, our study reveals that the electron-phonon interactions significantly limit the lattice thermal conductivities of p-type GaN and Si. Importantly, it is the electron-phonon matrix elements, rather than the Fermi surface nesting function, that are ascribed to the relatively larger reduction in the thermal conductivity of p-type Si compared to p-type GaN.

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