Comparison of interband related optical transitions and excitons in ZnGeN2 and GaN

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The optical dielectric function of ZnGeN₂ is calculated from the interband transitions using the energy bands calculated in the quasiparticle self-consistent (QS) $G\hat{W}$ method using two different levels of approximation: the independent particle approximation and the Bethe-Salpeter equation (BSE) approach. The first allows us to relate peaks in $\varepsilon_2(\omega)$ to specific bands and **k** points but does not include electron-hole interaction effects. The second includes electron-hole interaction or excitonic effects. The corresponding changes in the shape of $\varepsilon_2(\omega)$ are found to be similar to those in GaN. The screened Coulomb interaction \hat{W} is here calculated including electron-hole interactions in the polarization function and gives a band structure already going beyond the random phase approximation. The static dielectric constants including only electronic screening, commonly referred to as ε^{∞} , were calculated separately by extrapolating the wave-vector-dependent macroscopic $\varepsilon_M(\mathbf{q},\omega=0)$ for $\mathbf{q}\to 0$. Below the quasiparticle gap, we find three bound excitons optically active for different polarization. The convergence of these bound excitons with respect to the density of the k mesh used in the BSE is studied and found to require a fine mesh. It is also found that these bound excitons originate from only the lowest conduction band and the top three valence bands. To incorporate the lattice screening, we include a scaling factor $(\varepsilon^{\infty}/\varepsilon^{0})^{2}$, which allows us to obtain exciton binding energies of the correct order of magnitude similar to those in GaN. The excitons are related to each of the threefold split valence bands and the splittings of the latter are also studied as a function of strain. Finally, a relation between the anisotropic effective masses and the valence-band splitting is pointed out and explained.

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I. INTRODUCTION

ZnGeN₂ is a heterovalent ternary semiconductor closely related to wurtzite GaN. The Pna21 structure, which is the ground-state structure, corresponds to a specific ordering of the divalent Zn and tetravalent Ge ions on the cation sublattice of wurtzite, in which each N is surrounded by two Zn and two Ge, thus locally obeying the octet rule. The resulting orthorhombic symmetry lowering leads to some distortion of the parent hexagonal lattice vectors away from the perfect $a = \sqrt{3}a_w$, $b = 2a_w$, and $c = c_w$, with a_w and c_w the corresponding wurtzite lattice constants. (Unlike in some of our previous papers on these materials, we here use the standard setting of the space group of the International Tables of X-ray Crystallography.) ZnGeN₂ and other heterovalent II-IV-N₂ ternaries have recently received increasing attention as being complementary to the well-studied III-N semiconductors, providing several opportunities for increased flexibility in tuning their properties. Several recent reviews are available on the current state of knowledge on crystal growth, electronic band structure, phonon, elastic, piezoelectric, and defect properties of these materials [1-3].

Specifically for the Zn-based Zn-IV- N_2 , the electronic band structure was previously calculated [4] using the quasiparticle self-consistent GW method, which is one of the most accurate and predictive methods available. It uses manybody perturbation theory in the GW approximation due to Hedin [5,6], where G is the one-electron Green's function and W is the screened Coulomb interaction. The quasiparticle self-consistent version [7,8] of this method, named

QSGW, optimizes the noninteracting H^0 , from which the dynamic self-energy effects are calculated as $\Sigma = iG^0W^0$, by extracting a static (i.e., energy-independent) and Hermitian (Re) but nonlocal exchange-correlation potential $\tilde{\Sigma}_{ij}$ = $\frac{1}{2}\text{Re}\{\Sigma_{ij}(\epsilon_i) + \Sigma_{ij}(\epsilon_j)\}$. This self-energy $\tilde{\Sigma}_{ij}$ is obtained in the basis of eigenstates of H^0 , and by iterating $H^0 + \tilde{\Sigma} - v_{xc}$, with $v_{\rm rc}$ the exchange-correlation potential in the initial H^0 , to convergence. It thereby becomes independent of the starting H^0 which is usually taken as the density functional theory (DFT) Kohn-Sham Hamiltonian in either the local density approximation (LDA) or the generalized gradient approximation (GGA). The quasiparticle energies then become identical to the Kohn-Sham eigenvalues at convergence, hence the name "quasiparticle" self-consistent. However, the H^0 Hamiltonian is then no longer a Kohn-Sham Hamiltonian in the DFT sense, since it is not the functional derivative $\delta E^{\rm DFT}/\delta n({\bf r})$ of a corresponding DFT total energy anymore.

Here we revisit the QSGW calculations of ZnGeN₂ for various reasons. First, there have been recent improvements in the QSGW method including electron-hole effects in the screening of W. Second, the QUESTAAL codes [9,10] used for these calculations have been made more efficient, which allows us to check the convergence with stricter criteria. Third, the structural parameters used in previous calculations [4] were, in retrospect, not fully optimized. The main difference is that the larger a/b ratio leads to a different ordering of the top two valence bands as will be discussed in Sec. III E. The structural parameters were already corrected in Ref. [11]. To extract the best possible band structure and optical properties,

we here use the experimental crystal structure parameters (a=5.45 Å, b=6.44 Å, and c=5.19 Å). The first focus of the paper is on calculating the optical dielectric function and analyzing it in terms of band-to-band transitions. To do this we calculate the optical dielectric function in the continuum region above the quasiparticle gap in the independent particle approximation (IPA) and including electron-hole interaction effects using the Bethe-Salpeter equation (BSE) approach. We also perform similar calculations for the parent compound GaN to highlight the similarities in optical properties and to facilitate the comparison by considering the band-structure folding effects from the GaN wurtzite Brillouin zone to the $Pna2_1$ Brillouin zone.

Our second aim is to understand the valence-band maximum splitting and the resulting exciton fine structure below the band-to-band continuum. We study the convergence of the exciton gaps with increasingly finer \mathbf{k} mesh near Γ and show how their polarization is related to the symmetry of the valence bands involved and study their dependence on strain. Finally, we point out a relation of the symmetry of the top three valence bands to the anisotropy of the effective mass tensor of each of these bands.

II. COMPUTATIONAL METHOD

As already mentioned, the calculations in this study are carried out using the QUESTAAL code suite [10], which implements DFT using a full-potential linearized muffin-tin-orbital (FP-LMTO) basis set and extends beyond DFT to many-body perturbation theory by incorporating the GW and BSE methods. For details about the QSGW method, see Ref. [8] and for a full description of the FP-LMTO method, see Ref. [10]. While the QSGW method is a great advance compared to DFT calculations in terms of the accuracy of band gaps and other band-structure features, it tends to somewhat overestimate the band gaps because it underestimates the screening by not incorporating the electron-hole interactions in the screening of the screened Coulomb interaction $W = (1 - Pv)^{-1}v$, where v is the bare Coulomb interaction and P is the twopoint polarization propagator. This is commonly known as the random phase approximation (RPA). Calculations using an exchange-correlation kernel f_{xc} , extracted from BSE calculations or using a bootstrap kernel approximation [12,13] within time-dependent DFT (TDDFT), i.e., W = [1 - P(v + v)] $[f_{xc}]^{-1}v$, have shown that this tends to reduce the self-energy by a factor of ~ 0.8 and this has led to the commonly used approach, namely, the 0.8Σ approximation [14–16]. That approach was also used in the previous calculations on $ZnGeN_2$ [4,11].

Recently, a new approach, calculating directly the four-point polarization function via a BSE at all $\bf q$ points and then recontracting it to a two-point function was introduced by Cunningham *et al.* [17–19]. This approach is equivalent to adding a vertex correction to the polarization function in the Hedin set of equations with the vertex based on $\delta \Sigma^{GW}(12)/\delta G(34)$. However, it was argued that, within the spirit of the QSGW approach, no corresponding vertex corrections are needed in the self-energy because of the cancellation of the Z factor between the coherent part of the Green's function and the vertex. The Z factor $Z = (1 - d\Sigma/d\omega)^{-1}$

measures the reduction of the coherent quasiparticle part of the dynamic Green's function G compared to the independent particle Green's function G^0 . Since only G^0 is made self-consistent in the QSGW method, not G itself, the Z factor is omitted. In any case, this new approach, named QSG \hat{W} , removes the arbitrariness of an $ad\ hoc$ correction factor.

While in the previous calculations of $ZnGeN_2$ the 0.8Σ approach was used, we here use the $QSG\hat{W}$ approach. Second, other factors still limit the accuracy of our previous calculations of $ZnGeN_2$. The calculations of Ref. [4] used, in retrospect, a somewhat imperfectly relaxed crystal structure. This was later corrected in Ref. [11], where we used a GGA relaxed structure within the Perdew-Burke-Ernzerhof (PBE) parametrization. Here, we prefer to use the actual experimental structure because the typical overestimate of the lattice constants, in particular the volume per unit cell, leads to an underestimation of the gap.

The present calculations also use a larger basis set than in the previous work [4,11]. LMTO envelope functions are included with two smoothed Hankel function basis sets including up to $l \leq 3$ and $l \leq 2$, respectively, and standard choices of the smoothing radii and Hankel function kinetic energy κ^2 suitable for GW calculations. In addition, the Zn 3d orbitals are treated as augmented orbitals while the 4d are included as local orbitals to better represent the high-lying conduction-band states. For Ge, the 3d semicore orbitals are included as local orbitals while the 4d are augmented orbitals. Augmentation inside the spheres is done up to $l \leq 4$. By augmentation we mean that the expansion in spherical harmonics of a basis envelope function centered on one site about another site is replaced by a combination of the solutions of the radial Schrödinger equation $\phi_l(\epsilon_v, r)$ at a linearization energy ϵ_{ν} and its energy derivative ϕ_l that matches the envelope in value and slope. This expansion is carried out by means of structure constants. All calculations are including scalar relativistic effects in the potential. In the GW selfenergy calculations, we include all the bands generated by this basis set. This includes bands up to about 10 Ry. Two point quantities are expanded in an auxiliary basis set of products of partial waves inside the spheres and products of interstitial plane waves. This basis set describes screening as embodied in the dielectric response function $\varepsilon_{II}^{-1}(\mathbf{q},\omega)$ more efficiently than a plane-wave expansion and thereby bypasses the need for higher conduction bands in the generation of the dielectric function. Here the subscripts I, J label the auxiliary basis set. The short-wavelength screening is taken care of by the products of partial waves inside the spheres. Other convergence parameters include the k mesh on which the selfenergy is calculated for which we went up to $4 \times 4 \times 4$ and found negligible difference from the $3 \times 3 \times 3$ mesh. Using the atom-centered LMTO basis set the self-energy can then be expanded in real space and transformed back by Fourier transformation to a finer k mesh used for the self-consistent charge density or spectral functions, and for the ${\bf k}$ points along symmetry lines. In other words, the LMTO atom-centered basis set serves as a Wannier-type expansion for interpolations with the difference that it is not an orthogonal basis set and thereby more localized. The self-energy matrix in the basis set of the Kohn-Sham eigenstates is fully included instead of only carrying out a perturbative calculation of the Kohn-Sham

to GW eigenvalue shifts. To facilitate the above-mentioned interpolation of the self-energy, which requires sufficiently well-localized basis functions, it approximates the full self-energy matrix by a diagonal average value above a certain energy cutoff chosen to be 2.71 Ry here. This average value of the self-energy is evaluated over a range of 0.5 Ry above this cutoff. The rationale behind these approximations and other details about the method are explained in Ref. [8].

The main focus of the present paper is the optical interband transition properties. We calculate the imaginary part of the dielectric function $\varepsilon_2(\omega)$ at various levels of approximation. First, in the IPA one uses

$$\varepsilon_{2}(\omega) = \frac{8\pi^{2}e^{2}}{\Omega\omega^{2}} \sum_{v} \sum_{c} \sum_{\mathbf{k} \in BZ} f_{v\mathbf{k}} (1 - f_{c\mathbf{k}})$$
$$\times |\langle \psi_{v\mathbf{k}} | [H, \mathbf{r}] | \psi_{c\mathbf{k}} \rangle|^{2} \delta(\omega - \epsilon_{c\mathbf{k}} + \epsilon_{v\mathbf{k}}). \tag{1}$$

Here, the dielectric function is calculated as a sum over band-to-band transitions weighted by the matrix elements of the velocity operator. In this calculation we used a fine and well-converged $16 \times 16 \times 16$ **k** mesh. In case of the bands being calculated at the QSGW or QSGŴ level, the nonlocal exchange-correlation potential leads to additional terms beyond the momentum matrix elements, which involve $d\tilde{\Sigma}/dk$. The advantage of this approach is that, in principle, it takes the $\mathbf{q} \to 0$ limit exactly. However, technical difficulties in evaluating $d\tilde{\Sigma}/dk$ lead to some overestimate of the optical matrix elements. Alternatively, one can evaluate directly $\varepsilon_2(\mathbf{q},\omega)$, which would involve matrix elements of $e^{i\mathbf{q}\cdot\mathbf{r}}$ rather than the commutator, at small but finite \mathbf{q} , and take the limit numerically by extrapolation.

The main advantage of the sum-over-transitions expression of Eq. (1) is that we can take the optical function apart into its separate band-to-band contributions. Comparing these with the interband differences plotted as functions of **k** along symmetry lines, we can identify how peaks in the $\varepsilon_2(\omega)$ contributions are related to singularities (corresponding to critical points) or flat regions of the interband differences (corresponding to parallel bands) and therefore high joint density of states (JDOS) contributions.

On the other hand, this approach does not include electronhole interaction or local field effects which modify the shape of the optical dielectric function. These can be obtained within the BSE approach to many-body perturbation theory. Here, the key step is to calculate the four-point polarization

$$\bar{P}(1234) = P^{0}(1234) + \int d(5678)P^{0}(1256)K(5678)\bar{P}(7834),$$
(2)

with the kernel

$$K(1234) = \delta(12)(34)\bar{v} - \delta(13)\delta(24)W(12), \tag{3}$$

where $\bar{v}_{\mathbf{G}}(\mathbf{q}) = 4\pi/|\mathbf{q} + \mathbf{G}|^2$ if $\mathbf{G} \neq 0$, and zero otherwise, is the microscopic part of the bare Coulomb interaction and we have expressed it here in terms of a plane-wave basis set. The macroscopic dielectric function is then given by

$$\varepsilon_{M}(\omega) = 1 - \lim_{\mathbf{q} \to 0} v_{\mathbf{G}=0}(\mathbf{q}) \bar{P}_{\mathbf{G}=\mathbf{G}'=0}(\mathbf{q}, \omega). \tag{4}$$

The contribution from \bar{v} adds local field effects while the contribution from W, or in our case \hat{W} , leads to the explicit

excitonic effects on the dielectric function. Here, the four-point polarization depends on the coordinates, spin, and time of four particles. The numbers 1,2,3,4 are a shorthand for position, spin, and time of each particle; in other words, $1 \equiv (\mathbf{r}_1, \sigma_1, t_1)$. In practice, this integral equation is solved by introducing a resonant transition state basis set $\psi_{v\mathbf{k}}(\mathbf{r}_h)^*\psi_{c\mathbf{k}}(\mathbf{r}_e)$, which are products of one-particle eigenstates. Specifically,

$$P(\mathbf{r}, \bar{\mathbf{r}}, \mathbf{r}', \bar{\mathbf{r}'}) = \sum_{n_1, n_2, n_3, n_4} P_{n_1 n_2 n_3 n_4} \times \psi_{n_1}(\mathbf{r})^* \psi_{n_2}(\bar{\mathbf{r}}) \psi_{n_3}(\mathbf{r}') \psi_{n_4}(\bar{\mathbf{r}}')^*.$$
(5)

The polarization is then given by

$$P_{n_1 n_2 n_3 n_4}(\omega) = (f_{n_4} - f_{n_3})(H^{2p} - \omega)^{-1}, \tag{6}$$

with the two-particle Hamiltonian

$$H_{n_1 n_2 n_3 n_4}^{2p} = (\epsilon_{n_1} - \epsilon_{n_2}) \delta_{n_1 n_3} \delta_{n_2 n_4} - K_{n_1 n_2 n_3 n_4}(\omega), \tag{7}$$

where K uses a similar expansion of K(1, 2, 3, 4) in the basis functions of the one-particle Hamiltonian. In the present work a static approximation $\omega = 0$ for the W, hence for the kernel K, and the Tamm-Dancoff approximation of only keeping resonant and not antiresonant transitions are used. In the above, the basis state indices n_1 and n_3 can then be identified with valence bands $v\mathbf{k}$ and $v'\mathbf{k}'$, respectively, and n_2 and n_4 with conduction-band states $c\mathbf{k}$ and $c'\mathbf{k}'$. In the first term only direct vertical transitions $\epsilon_{c\mathbf{k}}$ - $\epsilon_{v\mathbf{k}}$ occur, while the kernel term mixes states at different \mathbf{k} and \mathbf{k}' , as well as between different valence- and conduction-band pairs. In other words, the two-particle Hamiltonian of the BSE $H_{vc\mathbf{k},v'c'\mathbf{k}'}$ has dimensions $N_v N_c N_k$ with N_v the number of valence bands, N_c the number of conduction bands, and $N_{\mathbf{k}}$ the number of \mathbf{k} points included. The k points are taken on a regular mesh in the Brillouin zone. The equations used here and approximations follow the review paper of Onida et al. [20]. Diagonalizing this two-particle Hamiltonian yields the eigenvalues and eigenvectors of the excitons in terms of the one-particle eigenstates, from which the dielectric function can be obtained. The same cautions apply about the accuracy of the optical dipole matrix elements as mentioned above. The eigenstates of the two-particle Hamiltonian, however, are mixtures of different vertical transitions and hence the decomposition in individual band-to-band transitions, strictly speaking, no longer makes sense. We should also mention that we here only calculate the spin singlet excitons. As explained in Ref. [21], the spin structure, if no spin polarization is present and we ignore spinorbit coupling, can readily be taken into account. The singlet excitons then involve $2\bar{v} - W$ in the kernel in Eq. (3). For the calculations of the full energy range of $\varepsilon_2(\omega)$ we choose $N_v =$ 24, $N_c = 16$, which encompasses all N 2*p*-derived valence bands and conduction bands up to a similar energy above the conduction-band minimum. To examine the excitons in more detail we use a smaller set of $N_v = 3$ and $N_c = 1$, which were found to be the main contributing bands to these excitons but with a larger k mesh, which is required to converge the exciton binding energies.

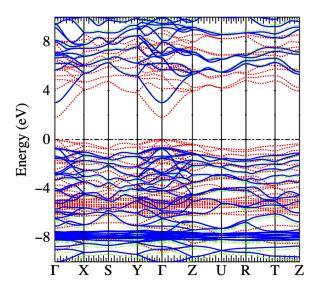


FIG. 1. Band structure of $Pna2_1$ ZnGeN₂ at experimental lattice parameters in GGA (red dotted), QSGW (green dashed), and QSGŴ (blue solid). The zero is placed at the VBM of GGA.

III. RESULTS

A. Band structure

The band structure of ZnGeN₂ is shown in Fig. 1 in three different approaches: GGA, QSGW, and QSG \hat{W} . The zero of energy is placed at the valence-band maximum (VBM) of the GGA. One can thus see that in QSGW, the VBM shifts down by 0.77 eV while the conduction-band minimum (CBM) shifts up by 1.30 eV. Adding the ladder diagrams reduces the self-energy shifts of both band edges by a small amount. The band gaps are summarized in Table I and for comparison we include the corresponding results for GaN. For both QSGW and QSGW calculations we used a $4 \times 4 \times 4$ k mesh for ZnGe N_2 . To be consistent, an $8 \times 8 \times 4$ k mesh was used for GaN. One can see that the self-energy corrections to the gap, $\Delta \Sigma_{OSGW} = E_g(QSGW)$ – $E_g(GGA)$ and $\Delta \Sigma_{QSG\hat{W}} = E_g(QSG\hat{W}) - E_g(GGA)$, have a ratio of $\Delta \Sigma_{QSG\hat{W}}/\Delta \Sigma_{QSGW} = 0.89$ in ZnGeN₂, and the same ratio is 0.88 for GaN. Thus, the inclusion of ladder diagrams reduces the self-energy gap correction by slightly larger than 10%, which is somewhat smaller than the often used 20% reduction in the 0.8Σ approach. We caution that this ratio is

TABLE I. Band gaps in eV for ZnGeN₂, and GaN in different approximations. The GaN band gap is reported for two different lattice constants and calculations where we include different numbers of conduction bands in the QSG \hat{W} approach.

		GGA	QSGW	QSGŴ
ZnGeN ₂ GaN	a = 3.189 Å a = 3.215 Å	1.801 1.884 1.697	3.874 3.868 3.638	3.641 3.624 ^a 3.487 ^b 3.392 ^a 3.261 ^b

 $^{^{}a}\hat{W}$ was calculated using $N_{c}=4$.

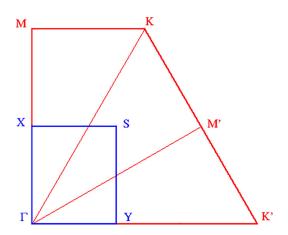


FIG. 2. Relation of hexagonal to Pna21 Brillouin zone.

sensitive to the numbers of bands N_v and N_c in the QSGW calculations. We here used $N_v = 24$ valence and $N_c = 16$ conduction bands for ZnGeN2. To obtain the same accuracy, we use $N_v = 6$ and $N_c = 4$ for GaN. As seen in Table I, using a higher $N_c = 10$ reduces the gap slightly more, so that the QSGW to QSG \hat{W} gap reduction factor becomes 0.81 instead of 0.88. However, this would then also lead to a smaller \hat{W} in the BSE calculations, which would correspondingly reduce the exciton binding energies and shift the peaks between IPA and BSE, but the actual exciton gaps would stay similar. Furthermore, the gaps depend strongly on the lattice constants. For ZnGeN2, we had used experimental lattice constants but for GaN, the value of a = 3.215 Å corresponds to the calculated PBE-GGA value and is slightly overestimated. Using the room-temperature experimental lattice constant of 3.189 Å [22,23], we obtain a higher gap by 0.23 eV. We should note that these gaps do not include zero-point-motion (ZPM) corrections or finite temperature corrections. The ZPM band-gap renormalization is expected to be of order -0.15 eV in ZnGeN₂ by comparison with GaN [24].

It is instructive to compare the band structure of $ZnGeN_2$ with that of its parent compound GaN. The band structure of GaN in the wurtzite structure is well known and is shown in the Supplemental Material [25] in the usual hexagonal Brillouin zone as obtained with the present computational method. Here we discuss how it relates to the $ZnGeN_2$ band structure by band folding.

The $Pna2_1$ structure can be viewed as a 2×2 supercell of the wurtzite, with $\mathbf{a}^o = 2\mathbf{a}_2^w + \mathbf{a}_1^w$, $\mathbf{b}^o = 2\mathbf{a}_1^w$, and $\mathbf{c}^o = \mathbf{c}^w$. The relation of the $Pna2_1$ Brillouin zone to that of wurtzite is illustrated in Fig. 2. One can see that the M-K line will be folded about the X-S line onto the bottom Γ -K' line and subsequently a second folding takes place about the Y-S line. Thus, the eigenvalues of the M point will now occur at Γ but also the M' point is folded onto X, so the same M eigenvalues will also occur at X. In Fig. 3 we show how the bands along Γ -M of wurtzite are found back in folded fashion along Γ -X by a single folding about the point X. Additional details of the band folding of K are discussed in the Supplemental Material [25].

These help to understand the band structure of the GaN 16-atom supercell directly plotted in the *Pna2*₁ Brillouin zone,

 $^{{}^{}b}\hat{W}$ was calculated using $N_c = 10$.

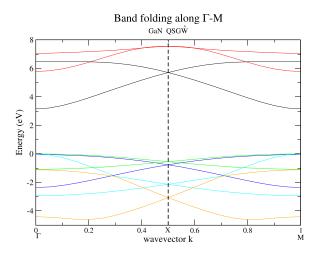


FIG. 3. Folding of GaN bands along wurtzite Γ -M onto the Γ -X line of $Pna2_1$.

which can then in turn be compared with that of $ZnGeN_2$ as shown in Fig. 4. The correspondence is seen to be quite close, especially in the conduction-band region. The perturbation due to the Zn-Ge difference in potential is thus rather small. Of course, in the deeper valence-band region near -8 eV we recognize the close set of bands from the Zn 3d bands while the Ga 3d lie much deeper and are not included in the presently shown energy range. The results for GaN in the

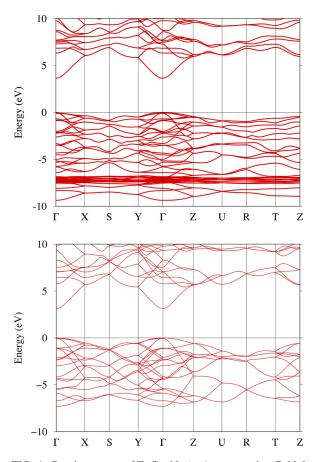


FIG. 4. Band structure of $ZnGenN_2$ (top) compared to GaN (bottom) both in $Pna2_1$ structure.

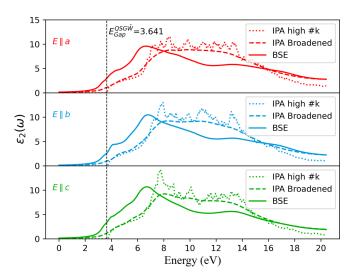


FIG. 5. Imaginary part of the dielectric function of ZnGeN $_2$ for three polarizations and comparing IPA with BSE approximations. Imaginary part of energy is set to 0.02 Ry. The IPA is once calculated via Eq. (1) with a $16 \times 16 \times 16$ k mesh and the tetrahedron integration method, and also with an imaginary part of ω and a $6 \times 6 \times 6$ mesh. The latter k mesh was also used in BSE calculation.

supercell were obtained from those in the primitive wurtzite cell by converting the self-energy matrix to real space using the procedure of Ref. [26] followed by a simple band-structure step without having to reconstruct the self-consistent potential. Just as M-L is folded onto Γ -X the points above it along the z direction L will fold onto Z of $Pna2_1$. The whole M-L line is thus folded onto Γ -Z. This will be seen to be relevant in the next section.

B. Dielectric function in BSE

In Fig. 5 we compare the dielectric function in the IPA with the BSE calculation for $ZnGeN_2$. We show the IPA result once with a converged number of $\bf k$ points as obtained from Eq. (1) and once with the same broadening and $\bf k$ -point mesh as used in the BSE, where we can only afford a less dense $\bf k$ mesh. The broadening is chosen to match the results of the IPA as closely as possible between the two approaches and amounts to an imaginary part of the energies of \sim 0.3 eV. For a smaller broadening, unphysical oscillations appear in the spectrum. These oscillations result from the coarse sampling which misses the eigenvalues at points in between the sampled $\bf k$ points and make it difficult to discern actual peak structure from artifacts of the $\bf k$ -point sampling.

The overall spectrum of $\varepsilon_2(\omega)$ in Fig. 5 can be described as follows. First, there is a region of slowly increasing $\varepsilon_2(\omega)$ ($\propto \sqrt{E-E_{\text{onset}}}$) between 3.6 eV and about 6 eV in both BSE and IPA. In the BSE at 6 eV a new feature sets in, peaking at about 6.5 eV, and decreasing to about 10 eV followed by another broad peak between about 12 and 14 eV after which the intensity decreases. In the IPA, the first peak reaches its maximum at a somewhat higher energy, around 7.5 eV, but the spectrum does not show a minimum near 10 eV and instead shows a broad series of peaks of about the same value till 14 eV. One can see that the BSE overall shifts oscillator

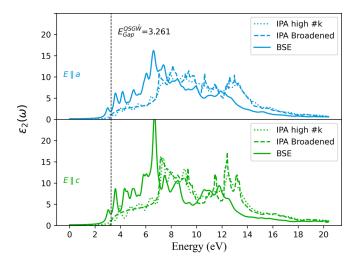


FIG. 6. Imaginary part of the dielectric function in BSE and IPA for GaN for polarizations perpendicular and parallel to the $\bf c$ axis. The imaginary part of the energy was set to 0.01 Ry. The IPA is once calculated via Eq. (1) with a $32 \times 32 \times 16$ k mesh and the tetrahedron integration method, and also with an imaginary part of ω and a $12 \times 12 \times 6$ mesh. The latter $\bf k$ mesh was also used in the BSE calculation.

strength to lower energies and results in a clearer two-broadpeaks structure: one between 6 and 10 eV and one between 10 and 16 eV. The first peak in BSE appears to occur at the onset of the steep rise in $\varepsilon_2(\omega)$ of the IPA spectrum. The quasiparticle gap is indicated by the vertical dashed line. With a smaller imaginary part, the BSE also shows peaks below the quasiparticle gap, which correspond to bound excitons, but with the broadening in Fig. 5 these features are smeared out and appear only as a shoulder below the quasiparticle gap. We will discuss the excitons separately in a later section.

A similar overall behavior of the dielectric function occurs in GaN, as seen in Fig. 6. We thus expect that there is a similar underlying physics going on in the band-to-band contributions and the BSE effects. We now discuss the origin of these observations. Because of the smaller number of bands, the analysis in terms of separate band-to-band transitions and where they mostly occur from in \mathbf{k} space is somewhat easier to analyze in GaN. We will then use the band folding of the wurtzite to $Pna2_1$ Brillouin zone to translate these findings to $ZnGeN_2$.

In Fig. 7 we show the contribution to the dielectric function in the IPA from the top valence band (VB₁) to conduction bands 1-3 (CB₁₋₃) along with the vertical difference in the bands at each k point along the symmetry lines. We can see that the onset comes only from $VB_1 \rightarrow CB_1$ transitions and is mainly polarized $\mathbf{E} \perp \mathbf{c}$. This is consistent with a strongly dispersing lowest conduction band near Γ and a gradual $\sqrt{\omega}$ increase of $\varepsilon_2(\omega)$ in the IPA. The IPA $\varepsilon_2(\omega)$ is essentially a matrix-element-modulated JDOS which is proportional to $\sum_{v \in \mathbf{k}} \delta(\omega - (\epsilon_{c\mathbf{k}} - \epsilon_{v\mathbf{k}}))$. One expects the JDOS contributions from different band pairs vc to show peaks when this energy band difference is nearly parallel over some region of **k** space. However, the matrix elements play a significant role in decreasing the intensity as energy increases and the $\varepsilon_2(\omega)$ shows a clearer peak structure than the JDOS. We thus focus directly on the IPA $\varepsilon_2(\omega)$ rather than the JDOS. The JDOS

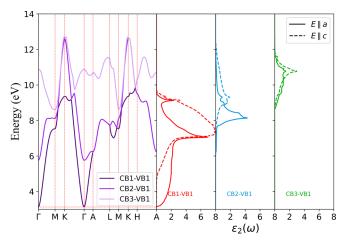


FIG. 7. Imaginary part of the dielectric function $\varepsilon_2(\omega)$ of GaN in the IPA associated with the transitions from the VBM and the lowest three conduction bands (right), and the differences between the band pairs along symmetry lines of the Brillouin zone. Higher ε_2 values are expected where the band differences (solid purple lines in the left panel) are flat.

is shown and further discussed in the Supplemental Material [25].

Returning to Fig. 7, the peak starting at \sim 7.0 eV is seen to be related to a relatively flat energy band difference along the M-L line and the onset of that peak coincides with the minimum of the $CB_1 - VB_1$ energy difference along M-L, which is a saddle point in the interband energy difference surface. We can thus conclude that the electron-hole interaction effects in BSE shift the oscillator strength toward the van Hove singularity saddle point. This provides an a posteriori justification for the common practice of analyzing the optical dielectric function in terms of a critical point analysis, emphasizing the behavior near high-symmetry points where van Hove–type singularities occur [27–29]. This was previously discussed in Ref. [30] and is here confirmed. It was noted in Ref. [30] that the electron and hole have exactly the same velocity at these points, which could heuristically explain why they might plausibly interact more strongly and make a large contribution when electron-hole pair interactions are included. A full analytical explanation of this observation would require an analysis of the matrix elements of W as a function of k near these singularities, which is a challenging task and has not yet been done. However, one does expect that W shifts the transitions down to lower energy, which explains why the peak in IPA is redshifted in the BSE. Various interband transitions close in energy to each other in this region of high IPA contribute to a "continuum" excitonlike peak in the BSE just below or at the saddle point. In other words, the two-particle Hamiltonian or "exciton" Hamiltonian has eigenvalues more concentrated in energy and right at the singularity that derive from a wider region of k space individual band-to-band transitions.

The CB₂ is close to CB₁ along M-L and also contributes to the peak within the range 7–9 eV. There is also a rather flat interband difference of CB₂ – VB₁ along Γ -M near M. This appears at about 8 eV and contributes to the second peak in the IPA dielectric function. Its relation to the BSE peaks is not

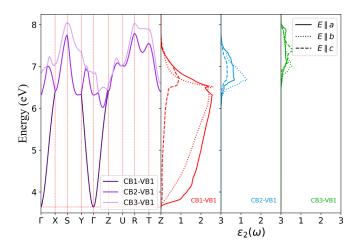


FIG. 8. Imaginary part of the dielectric function $\varepsilon_2(\omega)$ of ZnGeN₂ in the IPA associated with the transitions from the VBM and the lowest three conduction bands (right) and the differences between the band pairs along symmetry lines of the Brillouin zone. Higher ε_2 values are expected where the band differences (solid purple lines in the left panel) are flat.

entirely clear because in the BSE formalism one can no longer associate peaks to particular band-to-band transitions. On the other hand, CB_3 is related to flat band regions along Γ -A-L and contributes to the peaks near 11 eV.

A similar behavior is seen in ZnGeN₂. Figure 8 shows various contributions $VB_1 - CB_n$ to ε_2 for three polarizations. Additional figures are shown for the next top valence bands in the Supplemental Material [25]. We can again first see a relatively slow onset from the strongly dispersing conduction band near Γ . It is primarily polarized along x = a near its onset, consistent with the symmetry of VB_1 (which is b_1 - or x-like) and the s-like (a_1 symmetry of CB₁ at Γ). A peak starts emerging at about 6.5 eV and has contributions from CB₂ and CB₃. The onset of the peak near 6 eV is similar to what we just discussed for GaN but now is related to van Hove singularities and nearly parallel bands along Γ -Z which is the folded version of M-L in the wurtzite. One may also note the rather flat energy band difference of $CB_1 - VB_1$ along S-Y near Y. This could be related to the discontinuity in slope for the corresponding contribution to the IPA $\varepsilon_2(\omega)$ for **E** \parallel **a** (red solid line). However, it has only minor effects on the BSE dielectric function.

In summary of this section, the major peak structure of the $\varepsilon_2(\omega)$ in ZnGeN₂ has similar origins to that in GaN and can be explained in terms of the band-folding effects. The peaks in the IPA correspond to regions of high JDOS (modified by matrix elements which decrease with increasing energy) where various bands are close to being parallel and allowed dipole matrix elements. The electron-hole effects further shift the oscillator strength towards the van Hove singularities, where the bands become exactly parallel, which facilitates the stronger electron-hole coupling.

C. Static dielectric constant $\varepsilon_1(\omega=0)$

We also calculate the real part of the dielectric function $\varepsilon_1(q \to 0, \omega = 0)$. In principle, we can obtain it directly from

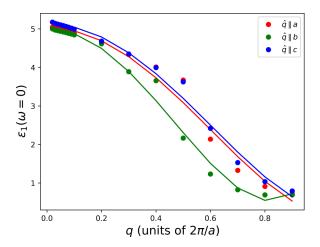


FIG. 9. $\varepsilon_1(\omega=0)$ of ZnGeN₂ is plotted with respect to q points around q=0.

a Kramers-Kronig transformation of the results of Fig. 5. However, this quantity is quite sensitive to the accuracy of the matrix elements of the velocity operator because through the Kramers-Kronig relation,

$$\varepsilon_1(0) = \mathcal{P} \int_0^\infty \frac{2\omega d\omega}{\pi} \frac{\varepsilon_2(\omega)}{\omega^2},\tag{8}$$

it depends on an integral over $\varepsilon_2(\omega)$ for all $\omega > 0$ and is thus sensitive to the values of ε_2 not just where its peaks occur. Because of the difficulties in incorporating the self-energy contribution to the velocity matrix elements, it is more accurate to directly examine $\varepsilon_1(q,0)$ for finite q values and then extrapolate numerically to zero. Furthermore, we obtain additional information on the \mathbf{q} dependence by looking at q throughout the Brillouin zone.

We use a model dielectric function, which was introduced by Cappellini et al. [31], to fit to the directly calculated results from the BSE at finite q shown as solid circles in Fig. 9. Extrapolation of this fit to the q = 0 limit gives the $\varepsilon_1(q =$ $0, \omega = 0$) values of 5.04 along the x, 5.01 along the y, and 5.13 along the z direction. These values of course correspond to electronic screening only and, therefore, to what is commonly referred to as ε^{∞} , rather than the true static value which would include the phonon contributions. In other words, it applies to the frequency range well below the band gap but high above the phonon frequencies and is related to the index of refraction measured in this region by $n = \sqrt{\varepsilon_1}$. The indices of refraction are thus 2.24, 2.24, and 2.26 for x, y, and z directions. Our values of $\varepsilon_1(\mathbf{q} \to 0, \omega = 0)$ obtained here are close to those obtained from a Berry phase calculation of the polarization in the LDA [32] which are 5.232, 5.166, and 5.725 for a, b, and c directions, respectively, or corresponding to indices of refraction of 2.29, 2.26, and 2.39. On the other hand, if we had used the $\varepsilon_1(\mathbf{q}=0,\omega=0)$ obtained directly from the Kramers-Kronig transformation of the BSE results in Fig. 5, we would find ε^{∞} values of 9.79, 9.02, and 9.30 for the a, b, and c directions. These are significantly overestimated by almost a factor of 2 and this results from the incorrect velocity matrix elements when taking the limit $\mathbf{q} \to 0$ analytically. Furthermore, the values obtained by the numerical extrapolation are close to those for GaN which is about 5.35 and

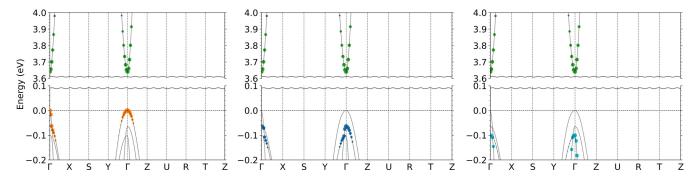


FIG. 10. Exciton band weights for the lowest three excitons, associated each with a single valence band. Note that $N_v = 24$ and $N_c = 16$ bands were included but only the bands shown contribute significantly to these excitons.

with only small anisotropy for the high-frequency dielectric constant. They are slightly smaller than the values obtained by calculating the adiabatic response to a static electric field using Berry phase calculations of the polarization as reported in Ref. [32], which used the local density approximation.

D. Valence-band splittings and excitons

In this section, we turn our attention to the energy region below the band-to-band continuum. Using a smaller broadening or examining the exciton eigenvalues below the quasiparticle gap directly, we find that there are three bright excitons, each polarized along a different direction, from lower to higher energy x, y, and z. The excitons for each polarization clearly correspond to transitions from each of the top three valence bands to the CBM and are consistent with the symmetry analysis of these bands given below. The exciton energies, however, are found to be sensitive to the k mesh used in the BSE equation. For the BSE dielectric function calculation over the full ω range as shown in Fig. 5 we used an $N_k \times N_k \times N_k$ mesh with $N_k = 6$ and with $N_v = 24$ and $N_c = 16$. Subsequently, we used different meshes with N_k varying from 3 to 6 to study the exciton convergence while maintaining the same N_v and N_c . Separately, because we found (see Fig. 10) that these excitons arise primarily from the top three valence bands and the lowest conduction band, we solved the BSE for only $N_v = 3$ and $N_c = 1$, but now going up to an $N_k = 10$ mesh. The extrapolation to $N_k \to \infty$ can be done in various ways, either fitting to an exponential saturation curve or by plotting versus $1/N_k$ and extrapolating the points with smallest $1/N_k$ to zero with a polynomial. The details of our extrapolation are described in the Supplemental Material [25].

We obtain a lowest exciton gap of 3.51 ± 0.01 eV. Comparing with the quasiparticle gap of 3.64 eV, this gives an exciton binding energy of 0.13 ± 0.01 eV. To the precision of the calculations, the same exciton binding energy is obtained for the excitons polarized in the other directions. These exciton binding energies are overestimated because they only include electronic screening. Within a hydrogenic model of the Wannier-Mott exciton, the binding energy would be given by an effective Rydberg divided by the dielectric constant squared and multiplied by a reduced effective mass. We can thus correct for our use of only electronic screening by multiplying by a scaling factor $\varepsilon_\infty^2/\varepsilon_0^2$. Table II shows the ε_∞ and

 ε_0 values from the literature and our scaled binding energies. The high-frequency dielectric tensor of GaN was calculated by various authors using density functional perturbation theory [33,34]. The phonon calculations in these papers provide ω_{LO}/ω_{TO} for $\mathbf{E} \parallel \mathbf{c}$ from the A_1 modes and for $\mathbf{E} \perp \mathbf{c}$ from the E_1 modes and through the Lyddane-Sachs-Teller relation, $\varepsilon_0/\varepsilon_\infty = (\omega_{LO}/\omega_{TO})^2$, provide us the scaling factor needed. In Table II we used the values from Karch et al. [33], but to the precision we here require, there is good agreement also with the results extracted from Bungaro et al. [34] and also with the experimental results $\varepsilon_{\infty} \approx 5.35$ for both parallel and perpendicular to **c** directions, and $\varepsilon_0^{\perp} = 9.5$ and $\varepsilon_0^{\parallel} = 10.4$ given by Barker and Ilegems [35]. We can see from Table II that the $\varepsilon_{\infty}/\varepsilon_0$ is about 0.55 ± 0.01 and $[\varepsilon_{\infty}/\varepsilon_0]^2 \approx 0.3$ and hence we obtain a binding energy of about 0.04 ± 0.02 eV. Similar results are obtained for GaN and the details of these calculations are given in the Supplemental Material [25]. The resulting estimates for the exciton binding energy are of the right order of magnitude and consistent with the known values for GaN of 22-26 meV [36]. However, the accuracy with which we can obtain these exciton binding energies is still limited by the number of k points we can afford, by the number of bands we can include concurrently, and by the limitations of the present BSE implementation which does not include lattice dynamical effects.

To further prove that these excitons are associated with each of the valence bands, the contribution of the valence and conduction bands in a band plot to each exciton are shown in

TABLE II. Literature values of static dielectric constant ε_0 , high-frequency dielectric constant ε_∞ , and the scaled binding energies. The estimated uncertainty results primarily from the uncertainties in the extrapolation.

		ε_0	$arepsilon_{\infty}$	$(\varepsilon_{\infty}^2/\varepsilon_0^2)$	BE (eV)
ZnGeN ₂	$\mathbf{E} \parallel \mathbf{a}^{\mathbf{a}}$	9.28	5.24	0.32	0.044 ± 0.003
	$\mathbf{E} \parallel \mathbf{b^a}$	9.22	5.17	0.31	0.042 ± 0.003
	$\mathbf{E} \parallel \mathbf{c^a}$	10.61	5.73	0.29	0.040 ± 0.003
GaN	$\mathbf{E} \parallel \mathbf{c}^{b}$	10.3	5.41	0.28	0.042 ± 0.001
	$\mathbf{E}\perp\mathbf{c}^{b}$	9.22	5.21	0.32	0.049 ± 0.001

^aReference [32].

^bReference [33].

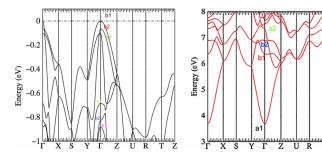


FIG. 11. Zoom in on the valence-band maximum region (left) and conduction-band range (right) of ZnGeN₂ as obtained in QSG \hat{W} . The bands at Γ are labeled by their irreducible representations.

Fig. 10. Again, this is obviously consistent with their polarization and the symmetry of the valence bands at Γ .

We now return to the valence-band splitting itself. In Fig. 11 we show a zoom in on the valence-band maximum energy range as well as the conduction-band region. The bands are labeled at Γ by their irreducible representations in the point group C_{2v} . The VBM has b_1 symmetry, which corresponds to x, and transitions from this band to the CBM, which has s-like a_1 (Γ_4 in Koster et al. notation [37]) symmetry, and is thus allowed for $E \parallel a$. The next band has symmetry b_2 (Γ_2) corresponding to y or **b** and the third band has a_1 (Γ_1) symmetry corresponding to z or **c**. Note that although this ordering appears to be the same as in Ref. [4], it is in fact different because a different setting of the space group, namely, $Pbn2_1$, was used in that paper, which means that the a and b directions are reversed from the present paper and hence also x and y or b_1 and b_2 . This difference is due to a difference in the lattice constant ratio a/b and already indicates that this splitting is sensitive to strain. We will discuss this strain dependence in the following section. The next few valence bands are also labeled in Fig. 11. The splittings between the top VBM and the next two bands are given in Table III. The next higher conduction bands have b_1 , b_2 , and a_2 symmetry and are situated at 2.68, 3.20, and 3.66 eV higher in energy, respectively.

In terms of the valence bands, one may notice that the top valence band with b_1 or x-like symmetry has the smallest hole mass in the x direction, while the next one with b_2 or y-like symmetry has the lowest mass in the y direction and the third valence band (counted from the top down) has the lowest mass in the z direction. Thus, the symmetry of the band corresponds to the direction in which the hole mass is the smallest. This was also noted in a recent paper on MgSiN₂-GaN alloys [26], although the explanation there is not quite correct. It can be explained in terms of the well-known $\mathbf{k} \cdot \mathbf{p}$ expression for the

TABLE III. Valence-band splittings in meV in ZnGeN₂.

	GGA	QSGW	QSGŴ
b_1-b_2	72	64	65
b_1 - a_1	94	102	101

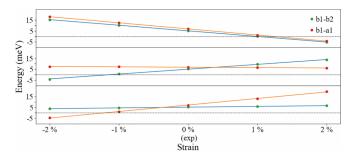


FIG. 12. Splitting between the top valence bands due to strain in x, y, and z directions (from top to bottom) are observed. The percentages on the x axis of the figure indicate the amount of strain applied to one major axis in comparison to the experimental lattice constants, while the volume of the unit cell is kept fixed.

effective mass,

$$M_{\alpha\beta}^{-1} = \delta_{\alpha\beta} \frac{1}{m_e} + \frac{1}{m_e^2} \sum_{n' \neq n} \frac{\langle n\mathbf{k} | p_{\alpha} | n'\mathbf{k} \rangle \langle n'\mathbf{k} | p_{\beta} | n\mathbf{k} \rangle + \text{c.c.}}{E_{n\mathbf{k}} - E_{n'\mathbf{k}}},$$

and the matrix elements between states that are nonzero can be found by symmetry. For example, for the VBM of b_1 symmetry and focusing first on the interaction between VBM and CBM, momentum matrix elements are only allowed in the x direction, so these bands are pushed away from each other along Γ -X, which reduces the hole mass in this direction. In the y direction, only interactions with the higher a_2 conduction band, or in the z direction, only interactions with the higher b_1 conduction band, would come in. These will be smaller because of the larger energy denominator; hence the hole mass is expected to be smallest in the x direction due to the strongest interaction with the conduction band. On the other hand, the interactions between nearby valence bands below the b_1 state would have the opposite effect of increasing the VBM hole mass. While the VBM splittings are much smaller, the momentum matrix elements must be smaller so these have less of an effect. This is because both valence bands are N_{2n} -like and intra-atomic matrix elements of the momentum operator are forbidden for the same angular momentum ℓ .

Likewise, for the second valence band the interaction with the lowest conduction band is only allowed for the y direction, and for the third valence band of a_1 symmetry this interaction is only allowed for the z direction. On the other hand, from the CBM point of view, there are matrix elements with either the valence bands of symmetries b_1 in the x direction, with b_2 along the y direction and with a_1 along the z direction. Hence, the CBM has a nearly isotropic effective mass tensor. These considerations apply quite generally to the II-IV-N₂ semiconductors.

E. Strain effects on valence-band splittings

In Fig. 12, we show the strain effect on the band splittings for the top three valence bands. We analyze the changes in the splitting due to a uniaxial strain within $\pm 2\%$, applied on one major axis at a time, where negative strain means that the experimental lattice constants are compressed, and positive strain means they are stretched. The unit cell volume is kept constant at the experimental value. In other words,

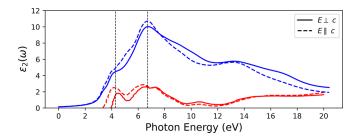


FIG. 13. Comparison between calculated results (blue curves) and digitalized data (red curves) from Misaki *et al.* [38]. We used the average of $\varepsilon_2(\omega)$ in *a* and *b* directions shown in Fig. 5.

we consider pure shear strains. As shown in Fig. 11, the top three valence bands have the symmetry of b_1 , b_2 , and a_1 . Figure 12 shows the splittings b_1 - b_2 and b_1 - a_1 , following the same symmetry labeling. Based on symmetry grounds, we expect that with a strain along x, which has b_1 symmetry, only the b_1 eigenvalue will shift. Hence, both band splittings will shift parallel to each other. We can see that compressive strain along x leads to a larger b_1 - a_1 and b_1 - b_2 splitting. For strain along b or y, only b_2 eigenvalues should be affected, so the b_1 - a_1 splitting stays constant but the b_1 - b_2 splitting behaves linearly with strain. Again, compression raises the b_2 level and hence reduces the b_1 - b_2 splitting. At some point it crosses through zero at which point the VBM becomes the b_2 band. Likewise, for strain along c, the a_1 level shifts and the b_1 - b_2 splitting stays constant but the b_1 - a_1 splitting shifts linearly and crosses through zero for a critical strain.

F. Comparison to experimental results

We are aware of only one experimental report of the optical dielectric function in the visible-to-UV region. Misaki *et al.* [38] reported reflectivity spectra of ZnGeN₂ parallel and perpendicular to the $\bf c$ axis, then subsequently extracted $\varepsilon_1(\omega)$ and $\varepsilon_2(\omega)$ by using the Kramers-Kronig analysis. Unfortunately, the data do not distinguish the a and b directions in the c plane. It is also possible that cation disorder led to effectively a wurtzite-type structure. We therefore compare an average of a and b polarization with their results in Fig. 13. The location of the main peaks is in reasonable agreement but the magnitude of ε_2 differs markedly between theory and experiment. They show a first peak near 4 eV, a main peak near 7 eV, a dip

and small peak near 10 eV, and then a plateau between 15 and 20 eV. These results are roughly consistent with our calculated optical properties. The experimental data show only a small anisotropy. The cutoff at lower energies is likely affected by the presence of finite film-thickness-related interference effects. Our calculated values of ε_2 are notably higher than the experimental values. On the theory side, this could be, in part, due to an overestimate of the velocity matrix elements. This is known to lead to an overestimate of $\varepsilon_1(\omega = 0)$ as well as we discussed earlier. It can be remedied by performing calculations at finite \mathbf{q} close to $\mathbf{q} = 0$ and extrapolating instead of relying on an analytical derivation of the limit for $\mathbf{q} \to 0$. On the other hand, the experimental values, extracted from reflectivity, may also be underestimated because of surface roughness, which can reduce the fraction of the signal that is specularly reflected [39].

IV. CONCLUSION

In this paper we revisited the QSGW band-structure calculations of ZnGeN2 using the experimental lattice constants and an improved calculation of the screened Coulomb interaction which includes ladder diagrams or electron-hole effects in the polarization. We also calculated the optical dielectric function corresponding to interband transitions at both the IPA and BSE levels. The differences between the two were compared with those in GaN and indicate a shift of oscillator strength of the first peak toward the critical points near its onset. These were analyzed in terms of individual band-to-band transitions, taking band-folding effects into account between GaN and ZnGeN₂ Brillouin zones and analysis of the IPA $\varepsilon_2(\omega)$ in terms of band-to-band differences in k space. Well-defined excitons below the gap are found but require very fine k meshes to determine with sufficient accuracy. Strain effects on the band splittings and hence exciton splittings resulting from the orthorhombic crystal field splitting were presented.

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