Prediction of high-*Tc* **superconductivity in ternary actinium beryllium hydrides at low pressure**

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Hydrogen-rich superconductors are promising candidates to achieve room-temperature superconductivity. However, the extreme pressures needed to stabilize these structures significantly limit their practical applications. An effective strategy to reduce the external pressure is to add a light element M that binds with H to form MH*^x* units, acting as a chemical precompressor. We exemplify this idea by performing *ab initio* calculations of the Ac– Be–H phase diagram, proving that the metallization pressure of Ac–H binaries, for which critical temperatures as high as 200 K were predicted at 200 GPa, can be significantly reduced via beryllium incorporation. We identify three thermodynamically stable $(AcBe₂H₁₀, ACBeH₈,$ and $AcBe₂H₁₄)$ and four metastable compounds (fcc AcBeH₈, AcBeH₁₀, AcBeH₁₂ and AcBe₂H₁₆). All of them are superconductors. In particular, fcc AcBeH₈ remains dynamically stable down to 10 GPa, where it exhibits a superconducting-transition temperature T_c of 181 K. The Be–H bonds are responsible for the exceptional properties of these ternary compounds and allow them to remain dynamically stable close to ambient pressure. Our results suggest that high-*T*^c superconductivity in hydrides is achievable at low pressure and may stimulate experimental synthesis of ternary hydrides.

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I. INTRODUCTION

The renaissance of high- T_c hydride superconductors began in 2004 [\[1,2\]](#page-6-0) and has recently attracted growing attention, after a series of spectacular experimental confirmations, such as superconductivity at 200 K in SH_3 when subjected to an external pressure of 155 GPa $[1-3]$, at 260 K and 170– 180 GPa in LaH10 [\[4–8\]](#page-6-0), at 215 K and 160 GPa or 172 GPa in CaH₆ [\[9–11\]](#page-6-0), at 220 K and 166 GPa or 237 GPa in YH_6 [\[12–14\]](#page-6-0), and at 243–262 K and 182–201 GPa in YH₉ [\[13,15\]](#page-6-0). Compared to pure hydrogen, the pressures required to realize metallization in binary hydrides are reduced considerably, but they are still so large to represent a serious limitation to any practical application. The search for high- T_c superconductors at lower or even ambient pressure remains an open challenge.

Thorough theoretical investigation of binary hydrides has revealed that most systems that display high- T_c (>150 K) superconductivity are stable only above 150 GPa [\[7,8,13,14\]](#page-6-0), while those that could be stabilized at lower pressure exhibit poor superconductivity [\[16–19\]](#page-6-0). These relatively disappointing conclusions have however cleared the way for testing

the largely unexplored family of ternary hydrides. In fact the additional chemical degree of freedom allows to enlarge enormously the search space, leading to exciting predictions with T_c close to the room temperature, such as Li_2MgH_{16} $(T_c = 473 \text{ K at } 250 \text{ GPa})$ [\[20\]](#page-6-0), YCeH₂₀ $(T_c = 246 \text{ K at }$ 350 GPa), LaCeH₂₀ ($T_c = 233$ K at 250 GPa) [\[21\]](#page-6-0), (La,Y)H₁₀ $(T_c = 253 \text{ K}$ at 183 GPa) [\[22\]](#page-6-0), and CaBeH₈ $(T_c = 254 \text{ K})$ at 210 GPa) $[23]$. Encouragingly, several high- T_c ternary hydrides have already been experimentally synthesized, including LaBeH₈ ($T_c = 110$ K, at 80 GPa) [\[24\]](#page-6-0), (La,Ce)H₉ ($T_c =$ 48–172 K at 92–172 GPa) [\[25,26\]](#page-6-0), (La,Y)H₁₀ ($T_c = 253$ K at 183 GPa) $[22]$, and $(La, Nd)H_{10}$ $(T_c = 148$ K at 180 GPa) $[27]$.

More specifically, we want to consider here the effect of adding a light element M that can bind with the H atoms to form small MH*^x* units. Such units can serve as precompression factor and act on the lattice of the parent binary to potentially reduce further the metallization pressure. For example, it was shown that introducing C atoms in a S-H system to obtain CH4 molecules yields dynamically stable ternary compounds with good superconducting properties, comparable with those of SH_3 [\[28,29\]](#page-6-0). In particular, the work of Cataldo *et al.* on LaBH₈ ($T_c = 126$ K at 50 GPa) opened a new era of ternary superconducting hydrides with higher T_c in low-pressure environments [\[30\]](#page-6-0). Moreover, the incorporation of B or Be in lanthanum hydrides, forming Be/BH_8 units, makes this system dynamically stable down to below 50 GPa

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with a T_c above 100 K [\[23,30,31\]](#page-6-0). The formation of SiH₈ molecules with the same symmetry ensures the dynamical stability of $BaSiH_8$ and $SrSiH_8$ down to 3 GPa and 27 GPa, respectively $[32]$. Moreover, $BH₄$ molecules intercalating fcc lattices of alkaline metals X ($X = K$, Rb, Cs) lead to energetically stable, superconducting ternary hydrides XB_2H_8 with T_c above 100 K at 10 GPa [\[33,34\]](#page-7-0).

Particularly interesting are the results obtained by incorporating Be atoms in binary hydrides: in fact, Be can act as electron donor to break the H_2 molecule and improve the superconducting properties, in the same way as Mg in Li_2MgH_{16} [20]. Moreover, it can lead to the formation of BeH₈ units that have the potential to reduce the metallization pressure of the parent hydride, as observed in LaBeH₈ (T_c = 183 K at 20 GPa) and YBeH₈ ($T_c = 249$ K at 100 GPa) [\[23\]](#page-6-0). The light Be atoms bind with H, replacing some H-H bonds in the hydrides, potentially leading to a reduction of the stabilization pressure. Additionally, doping with light elements can increase the average phonon frequencies, and if these vibrational modes are properly coupled with electrons at the Fermi energy (E_F) , this can also contribute to an increase of *T*c. Therefore, we consider very promising to insert Be atoms in high-*T*^c superconducting binary hydrides that are stable only at high pressures.

Actinium hydrides have been thoroughly studied at high pressures and several thermodynamically stable compounds have been proposed (e.g., AcH_2 , AcH_3 , Ac_3H_{10} , AcH_8 , $AcH₁₀$, and $AcH₁₆$). Among these, $AcH₁₆$ is predicted to be a good superconductor with T_c of 241 K at 150 GPa [\[35\]](#page-7-0). Here, we study the phase diagram of Ac–Be–H to find if the formation of BeH*^x* units can act as chemical precompressor of the fcc framework composed of the large Ac atoms, leading to ternary compounds that can be stable at low pressure. We perform therefore *ab initio* structural prediction calculations of AcBe*x*H*^y* (with *x* = 1−2, *y* = 1−10, 12, 14, 16, 18, 20) in search of thermodynamically stable and metastable structures that are metallic and potentially superconducting. Successive electron-phonon coupling (EPC) calculations are employed to evaluate the transition temperature for phonon-mediated superconductivity at different pressures.

We discover three stable phases $P1$ $AcBe₂H₁₀$, Pmmn AcBeH₈, *Cmcm* AcBe₂H₁₄ and four metastable phases $Fm3m$ AcBeH₈, $Fm3m$ AcBeH₁₀, $C2/m$ AcBeH₁₂, and $P4/mbm$ AcBe₂H₁₆. These structures are all dynamically stable, metallic, and superconducting. Our calculations predict that the $P4/mbm$ AcBe₂H₁₆ and $Fm\overline{3}m$ AcBeH₁₀ structures are superconductors with T_c of 150 K at 200 GPa and 165 K at 300 GPa, respectively. Particularly interesting is a metastable fcc structure of $ACBeH₈$ that remains dynamically stable down to 10 GPa with a high T_c of 181 K. Such a pressure is easy to achieve in an experiment, and is substantially lower than the required pressure to stabilize binary actinium hydrides.

II. COMPUTATIONAL DETAILS

The search for crystalline structures was performed using a particle-swarm optimization algorithm, as implemented in the CALYPSO code [\[37–40\]](#page-7-0). This method has been extremely successful in predicting stable and metastable superconducting hydrides [\[41\]](#page-7-0), some of which have already been confirmed by experiments [\[28](#page-6-0)[,42–45\]](#page-7-0). The calculations to predict the lowest enthalpy crystal structures of Ac*x*Be*^y* binary system (with $x = 1-2$, $y = 1-6$) and of AcBe_xH_y (with *x* = 1−2, *y* = 1−10, 12, 14, 16, 18, 20), considering up to four formula units, were done at 200 GPa. More than 2000 structures were sampled for each prediction run, for every composition, and each generation of structures was evolved by selecting the 60% lowest-enthalpy structures in the last step and randomly producing the remaining 40%. The structure searches were considered converged when $~\sim$ 1000 successive structures were generated without finding a new lowest-enthalpy structure. Structural relaxations and electronic band-structure calculations were performed using projector augmented-wave (PAW) method as implemented in the Vienna *Ab initio* Simulation Package (VASP) [\[46\]](#page-7-0). The exchange-correlation functional of density-functional theory was approximated by the generalized gradient approximation of Perdew, Burke, and Ernzerhof [\[47\]](#page-7-0). The ion-electron interaction was described by PAW potentials with $6s^26p^66d^17s^2$, $1s²2s²$, and $1s¹$ electrons configurations as valence states for Ac, Be, and H atoms, respectively. In order to further test the reliability of the adopted PAW pseudopotentials under pressure, we compared the Birch-Murnaghan equation of state obtained with PAW potentials with the full-potential linearized augmented plane-wave method (LAPW), using local orbitals (as implemented in WIEN2k [\[48\]](#page-7-0)). The results, in Fig. S1 of the Supplemental Material (SM) [\[49\]](#page-7-0), fully validate our approach. The cutoff energy for the expansion of the wave functions in the plane wave basis was set to 1000 eV. Monkhorst-Pack *k*-point meshes [\[50\]](#page-7-0) with a grid density of 0.20 Å^{-1} were chosen to ensure a total energy convergence better than 1 meV per atom. The phonon spectrum and EPC were calculated within linear-response theory with the QUAN-TUM ESPRESSO code [\[51\]](#page-7-0). We define the EPC parameter

$$
\lambda = 2 \int_0^{\omega} \frac{\alpha^2 F(\omega)}{\omega} d\omega, \tag{1}
$$

calculated from the $\alpha^2 F(\omega)$ is the Eliashberg spectral function and

$$
\omega_{\text{log}} = 2 \int_0^{\omega} \frac{\alpha^2 F(\omega)}{\omega} \ln(\omega) d\omega. \tag{2}
$$

Ultrasoft pseudopotentials for Ac, Be, and H were used in EPC calculations [\[52\]](#page-7-0). The detailed encut, *k* meshes, and *q* points for these seven compounds can be found in Table S1 of the SM [\[49\]](#page-7-0) for a Gaussian smearing width of 0.05 Ry. The convergence of the EPC parameter λ was tested using a set of Gaussian broadenings in steps of 0.001 Ry from 0.01 to 0.05 Ry with various *k* grids, as shown in Fig. S2 of the SM $[49]$. The superconducting critical temperature T_c is evaluated based on the Allen-Dynes-modified McMillan (ADM) equation [\[53\]](#page-7-0) when the λ is smaller than 1.5:

$$
T_{\rm c} = \frac{\omega_{\rm log}}{1.2} \exp\bigg[-\frac{1.04(1+\lambda)}{\lambda - \mu^*(1+0.62\lambda)}\bigg],\tag{3}
$$

and the corrected ADM by including two correction factors f_1 and f_2 to account for strong-coupling ($\lambda > 1.5$) and

FIG. 1. Thermodynamical stability. Ternary convex hull of Ac– Be–H at 200 GPa. The stable and metastable phases are depicted as hexagons (green) and circles (pink and gray), respectively. Boundary binary phases of Ac–H [\[35\]](#page-7-0) and Be–H [\[36\]](#page-7-0) systems are chosen from the previous works and there is no stable structure in Ac–Be system, as shown in Fig. S3 [\[49\]](#page-7-0). We indicate in light red metastable structures that lie less than 10 meV/atom above the convex hull.

shape-correction multipliers, respectively:

$$
T_{\rm c} = f_1 f_2 \frac{\omega_{\rm log}}{1.2} \exp\bigg[\frac{-1.04(1+\lambda)}{\lambda - \mu^*(1 + 0.62\lambda)}\bigg].\tag{4}
$$

III. RESULTS AND DISCUSSION

We calculated the phase diagram of AcBe*x*H*^y* at 200 GPa (shown in Fig. 1) and found out that three ternary phases with stoichiometries $AcBe₂H₁₀$, $AcBeH₈$, and $AcBe₂H₁₄$ are stable against decomposition into elemental or binary solids. Additionally, we uncovered several metastable structures with distances to the convex hull smaller than around 10 meV/atom, namely, AcBeH₆ (9.3 meV/atom), AcBe₂H₈ (2.8 meV/atom), $AcBe₂H₁₂$ (2.4 meV/atom), $AcBe₂H₁₆$ (5.0 meV/atom), and $AcBeH_{12}$ (10.9 meV/atom). In view of the fact that hydrogen-rich phases with high symmetry have an increased potential to exhibit high- T_c superconductivity [\[6](#page-6-0)[,54–58\]](#page-7-0), we also consider two metastable phases with fcc symmetry $AcBeH_8$ (4 meV/atom) and $AcBeH_{10}$ (35 meV/atom) . In particular, for the fcc AcBeH₈ phase, we compared relative enthalpies with respect to competing structures and possible decomposition products from 0 to 500 GPa, as shown in Figs. $S4(a)$ and $S4(b)$ [\[49\]](#page-7-0). In comparison to the work of Wan *et al.* [\[59\]](#page-7-0), we performed prediction runs for several compositions of the Ac–Be–H phase diagram and not only for $XBeH_8$. Fig. $S4(a)$ [\[49\]](#page-7-0) indicates that the formation enthalpy of the fcc phase of $AcBeH_8$ is higher than the one of the thermodynamically stable *Pmmn* phase and of other metastable structures, as shown in Fig. S4(b) [\[49\]](#page-7-0), but all the decomposition products exhibit positive formation energies relative to fcc $AcBeH_8$ above 30 GPa. The structure of AcBeH₈ with symmetry *Fm3m* is not thermodynamically stable in the range of pressures 0–30 GPa but our calculations suggest that there is a good possibility to synthesize this structure starting from $Ac_3H_{10} + BeH_2 + H_2$ below 200 GPa or at even lower pressure (30–50 GPa), through the synthesis paths $Ac_3H_{10} + Be + H_2$. As 10 GPa is an extremely low pressure, it is worth performing exploratory experiments. Other low-enthalpy structures may be accessible experimentally thanks to their small enthalpy distance to the convex hull at 200 GPa [\[5–8,22](#page-6-0)[,60,61\]](#page-7-0).

The crystal structures of thermodynamically stable and metastable $AcBe_xH_y$ compounds of 200 GPa are shown in Figs. [2](#page-3-0) and S5 $[49]$. AcBe₂H₁₀ adopts a low-symmetry *P*1 structure [Fig. $2(a)$] that contains BeH₈, BeH₇ units, and H₂ molecules, characterized by Be–H and H-H distances of 1.29– 1.59 Å and 0.87 Å, respectively. The orthorhombic phase of AcBeH8 [Fig. [2\(b\)\]](#page-3-0) with space group *Pmmn* is composed of BeH₈ octahedra with a Be–H distance of 1.30–1.43 Å, as well as Ac atoms occupying the $2a$ Wyckoff positions. AcBe₂H₁₄ is also orthorhombic with *Cmcm* symmetry [Fig. [2\(c\)\]](#page-3-0). In this structure, Ac atoms occupying the 4*b* Wyckoff positions and three pairs of H_2 molecules, with bond lengths of around 0.85 Å, are positioned between pairs of Ac atoms. Additionally, two inequivalent Be atoms occupy the 4*a* and 4*c* Wyckoff positions. These Be atoms are surrounded by eight H atoms forming BeH₈ octahedra with Be–H distances of 1.36–1.56 Å at 200 GPa. In the metastable structure of $AcBe₂H₁₆$ with *P*4/*mbm* symmetry [Fig. [2\(d\)\]](#page-3-0), the Ac atoms occupy the 4*g* position: each Be is bonded with ten H atoms to form a hexadecahedal BeH10 unit. Adjacent BeH10 units along the *y* axis are connected with each other by sharing one hydrogen atom, and neighboring BeH_{10} units along the *c* axis are connected by dimers and trimers with H-H distances of 1.08 and 0.92 Å, respectively. The covalent nature of these latter bonds can be confirmed by analysis of the electron localization function (ELF), as shown in Fig. $S6(a)$ [\[49\]](#page-7-0). The fcc phase of AcBeH₈ [Fig. $2(e)$], isostructural to $Fm\bar{3}m$ LaB(Be)H₈ [\[23,24,30,31\]](#page-6-0), consists of BeH_8 hexadra that occupy the octahedral sites of the fcc lattice formed by Ac atoms. The $AcBeH_{10}$ phase with *Fm3m* symmetry [Fig. [2\(f\)\]](#page-3-0) is 15 meV/atom energetically higher than the lowest enthalpy phase $P2₁$. Compared with fcc AcBeH $_8$, the four additional H atoms of AcBeH $_{10}$ occupy tetrahedral sites that are connected by four BeH_8 octahedra to form H_5 regular tetrahedra with H-H distances of 1.00 Å. Other metastable structures (e.g., $P1$ AcBeH₆, $P\overline{1}$ AcBe₂H₈, *Cmmm* AcBe₂H₁₂, and *C*2/*m* AcBeH₁₂) are shown in Fig. S5 [\[49\]](#page-7-0). In these structures, the Be and Ac atoms donate electrons to hydrogen, forming typical ionic hydrides (see Table SII [\[49\]](#page-7-0)). Apart from the Be–H bonds, these metastable structures present also H-kagome lattices, together with dimers and trimers (see Fig. S5 [\[49\]](#page-7-0)).

In order to further explore the fascinating properties of Ac–Be–H compounds, we calculated their electronic structure and phonon properties. Almost all predicted structures are also dynamically stable, as confirmed by inspection of the phonon bands, with the exception of $AcBe₂H₁₂$ and $AcBeH₁₀$ that display imaginary phonons (see Figs. [3](#page-3-0) and S7 [\[49\]](#page-7-0)). Even if $AcBe₂H₁₂$ and $AcBeH₁₀$ (Figs. S7(a) and S7(d) [\[49\]](#page-7-0)) are not dynamically stable at 200 GPa, $AcBeH_{10}$ becomes stable at higher pressure [Fig. $3(f)$]. The case of $Fm\overline{3}m$ AcBeH₈ is particularly interesting as it remains dynamically stable down to pressures of 10 GPa, as shown in Figs. $4(d)$ and S8 [\[49\]](#page-7-0). All considered structures are metallic with several bands crossing

FIG. 2. Structural configuration. The stable (a)–(c) and metastable structures (d)–(f) of Ac–Be–H compounds. (a) $P1$ AcBe₂H₁₀, (b) *Pmmn* AcBeH₈, (c) *Cmcm* AcBe₂H₁₄, (d) *P4/mbm* AcBe₂H₁₆, (e) *Fm*3 m AcBeH₈, and (f) *Fm*3 m AcBeH₁₀ at 200 GPa. Ac atoms are depicted in blue, Be in gray, and H in pink.

the Fermi energy E_F , and have therefore promising electronic bands for high-temperature superconductivity (see Figs. S9– S11 [\[49\]](#page-7-0)). The significant overlap of the partial electronic density of states (DOS) of the different atoms indicates a strong hybridization of Ac–H and Be–H under pressure. The results clearly indicate that hydrogen atoms make a substantial contribution to the total DOS of E_F , e.g., for *Cmcm* AcBe₂H₁₄ (49%), *P*4/*mbm* AcBe2H16 (55%), *C*2/*m* AcBeH12 (49%), and $Fm3m$ AcBeH₈ (44%) at 200 GPa (Table SIII [\[49\]](#page-7-0)).

The contribution of hydrogen to the DOS at E_F (DOS_H) is large for the three stable structures $P1$ AcBe₂H₁₀, *Pmmn* AcBeH₈, and *Cmcm* AcBe₂H₁₄, with values $8.9 \times$ 10^{-4} , 9.2×10^{-4} , and 6.8×10^{-4} states/(eV Å³) at 200 GPa, respectively **(**Figs. S9(a)−S9(c)[\[49\]](#page-7-0)**)**. For the highly symmetric metastable structures, this contribution is even larger, especially in the case of $Fm\overline{3}m$ AcBeH₈, where we can observe an enhancement from 44% (200 GPa) to 65% (10 GPa). The corresponding hydrogen contributions to the DOS are 6.28×10^{-3} and 4.31×10^{-3} states/(eV Å³), respectively. This enhances the coupling of the electrons to the H phonons, facilitating the formation of more Cooper pairs at E_F , and increasing the possibility of obtaining a higher superconducting transition temperature T_c (Figs. [4\(a\)](#page-4-0) and S10(a) [\[49\]](#page-7-0)). There are three bands of fcc $AcBeH_8$ at 10 GPa that cross E_F , yielding the Fermi surface shown in Figs. $4(a)$ and $4(b)$. Bands 1 and 2 are degenerate along the high-symmetry line

FIG. 3. Phonon band structures, phonon density of states (PHDOS), EPC coefficient $\lambda(\omega)$, and Eliashberg spectral function $\alpha^2 F(\omega)$ of (a) *P*1 AcBe2H10, (b) *Pmmn* AcBeH8, (c) *Cmcm* AcBe2H14, (d) *P*4/*mbm* AcBe2H16, (e) *Fm*3¯*m* AcBeH8, all at 200 GPa, and (f) $Fm\overline{3}m$ AcBeH₁₀ at 300 GPa.

FIG. 4. Electronic and superconductivity properties of AcBeH₈ at 10 GPa. Calculated (a) band structures and projected density of states (PDOS), (b) Fermi surfaces corresponding to the three bands crossing the Fermi energy level, colored with respect to the Fermi velocity $\langle v \rangle$ (10^5 m/s) , (c) nesting function $\xi(Q)$ along special *Q* trajectories, (d) phonon band structures, PHDOS, EPC coefficient $\lambda(\omega)$ and Eliashberg spectral function $\alpha^2 F(\omega)$, and (e) isotropic superconducting gap of $Fm\overline{3}m$ AcBeH₈ at 10 GPa.

from *L* to *X* in the first Brillouin zone and their Fermi surface exhibits pockets at *X* and complex semiclosed shapes with a hexagonal opening at *L*. In addition to this, band 3 forms a closed electronlike surface at the Γ point, which leads to larger DOS at E_F and to a further enhancement of the EPC [\[62\]](#page-7-0). Despite the higher hydrogen content, the contribution of the H atoms to the DOS of $Fm\overline{3}m$ AcBeH₁₀ at E_F is less than that of AcBeH₈ (only 4.28×10^{-3} states/(eV Å³) at 300 GPa, see Fig. S10(b) [\[49\]](#page-7-0)**)**. This is due to shift of about 2 eV of the bands around Fermi level, comparing with the bands of $AcBeH_{10}$ and $AcBeH_8$, we observe a shift of about 2 eV of a group of bands with strong H character close to the Fermi energy. This leads to a reduced H contribution to the DOS of E_F . More specifically, the H atoms in the same crystal graphic positions as in $AcBeH_8$ contribute now only 24% to the total DOS. The additional H atoms add only "a 15%" contribution. For other metastable structures with less symmetry (e.g., P1 AcBeH₆, P1 AcBe₂H₈), the partial DOS of H at E_F amounts to less than $\sim 6.5 \times 10^{-4}$ states/(eV Å³), which is not characteristic of superconductors with high critical temperature.

To investigate superconductivity and its mechanism, we calculated phonon dispersion curves, partial phonon density of states (PHDOS), the Eliashberg spectral function $\alpha^2 F(\omega)$ and the EPC integrated $\lambda(\omega)$ of Ac–Be–H hydrides and we plotted these quantities in Figs. [3](#page-3-0) and S8 [\[49\]](#page-7-0). The PH-DOS of all structures has common features: the heaviest Ac atoms contribute to the lowest-frequency vibrational modes, the middle-frequency modes are mainly dominated by Be and

H vibrations, and the high-frequency modes are exclusively due to H atoms.

The superconducting properties of the Ac–Be–H system can be estimated using the McMillan formula (as modified by Allen-Dynes [\[53\]](#page-7-0)) or by solving numerically the Eliashberg equations [\[63\]](#page-7-0). We used the typical value of the Coulomb pseudopotential $\mu^* = 0.10$. A summary of the results can be found in Table I. Among the thermodynamically stable compounds, $AcBe₂H₁₀$ possesses the strongest EPC parameter $\lambda = 2.59$, mainly owing to the soft modes along the Z- Γ high-symmetry line [Fig. [3\(a\)\]](#page-3-0), yielding a T_c of 115 K at

TABLE I. Superconducting-transition temperature. Logarithmic average phonon frequency (ω_{log}), EPC parameter λ , and T_c $(\mu^* = 0.10)$ for *P*1 AcBe₂H₁₀, *Pmmn* AcBeH₈, *Cmcm* AcBe₂H₁₄, $P4/mbm$ AcBe₂H₁₆, $Fm\overline{3}m$ AcBeH₈, $C2/m$ AcBe₂H₁₂ at 200 GPa, and $Fm\overline{3}m$ AcBeH₁₀ at 300 GPa.

Compounds	λ	ω_{log} (K)	T_c (K)	
			ADM	Eliashberg
$P1$ AcBe ₂ H ₁₀	2.59	337	78	115
$Pmmn$ AcBeH _s	0.84	1262	66	75
Cmcm AcBe ₂ H ₁₄	0.67	1240	39	42
$P4/mbm$ AcBe ₂ H ₁₆	1.19	1381	122.	150
$Fm\overline{3}m$ AcBeH _s	1.32	724	73	118
$C2/m$ AcBe ₂ H ₁₂	0.9	1130	66	78
$Fm\bar{3}m$ AcBeH ₁₀	1.76	909	138	165

200 GPa. This is the highest T_c among the thermodynamically stable compounds. The low frequency region $(0-245 \text{ cm}^{-1})$, dominated by the Ac atoms, contributes a significant amount (∼53%) to the total λ, but also the range of intermediate frequencies (245–1063 cm⁻¹), related to the Be–H bonds and the high-frequency region (1081–2358 cm−1, due to H atoms only) contribute 30% and 17%, respectively, to λ , as shown in Table SIV [\[49\]](#page-7-0). The two other stable compounds *Pmmn* AcBeH₈ and *Cmcm* AcBe₂H₁₄ have values of λ of 0.67 and 0.84, respectively, at 200 GPa, and in both cases H atoms give the largest contribution to λ with 74% and 54%, respectively **(**Figs. [3\(b\)–3\(c\)](#page-3-0) and Table SIV [\[49\]](#page-7-0)**)**. The metastable structures $AcBe₂H₁₆$ and $AcBe₂H₁₂$ have values of λ of 1.19 and 0.9 at 200 GPa, yielding a *T*^c of 150 K and 78 K, respectively (see Table [I\)](#page-4-0).

In what concerns fcc $AcBeH_{10}$, we remark that it is not stable at 200 GPa **(**Fig. S7(a) [\[49\]](#page-7-0)**)**, but it is stabilized at the higher pressure of 300 GPa [Fig. $3(f)$]. EPC calculations show that $Fm\overline{3}m$ AcBeH₁₀ phases are promising conventional superconductors with $\lambda = 1.76$ and $\omega_{\text{log}} = 909$ K at 300 GPa. This leads to an estimated T_c of 165 K. The phonon modes at frequencies between 427 and 2858 cm−1, corresponding to vibrations of hydrogen atoms, contribute a large amount (72%) to the total λ (Tables [I](#page-4-0) and SIV [\[49\]](#page-7-0)).

The other fcc structure, $AcBeH_8$, is dynamically stable from 200 to 10 GPa, as shown by the phonon dispersion depicted in Figs. [3\(e\),](#page-3-0) S8 [\[49\]](#page-7-0), and [4\(d\).](#page-4-0) The calculated λ and phonon frequency logarithmic average ω_{log} are 1.32 and 724 K at 200 GPa, respectively, leading to a T_c of 118 K for $\mu^* = 0.10$. Note that the vibrations related to H atoms at high frequencies give a significant contribution to the total λ (64–78%) for pressures in the range of 30–200 GPa, and the contribution of Be and H at intermediate frequencies only accounts for less than 6%. As the pressure decreases, λ increases proportionally to DOS_H , and the critical temperature of AcBeH₈ is gradually enhanced. λ increases from 1.32 to 2.3 from 200 to 100 GPa, and increases further from 2.3 to 4.5 from 100 GPa to 10 GPa, with T_c proportional to λ , as shown in Fig. 5. We conclude that $AcBeH_8$ is a typical superconductor dominated by λ . Strikingly, λ increases to 4.5 with a T_c of 181 K at 10 GPa (see Figs. [4\(d\)–4\(e\),](#page-4-0) 5, and Table SIV [\[49\]](#page-7-0)**)**. When the pressure decreases progressively to 10 GPa, the interaction between Be and H becomes stronger, and the contribution to λ due to soft phonon modes (110–540 cm⁻¹) increases to 60% (λ_{Be} % = 38%), especially thanks to contributions from the *W* point with symmetry C_{2v} , originating from A_1 modes that are mainly related to vibrations of hydrogen atoms **(**Figs. [4\(d\)](#page-4-0) and S12 [\[49\]](#page-7-0)**)**. There appears also a significant increase in the nesting function at 10 GPa along *W* -*L* and *X*-*W* -*K*, which is perfectly consistent with the phonon softening at W , as shown in Fig. $4(c)$. Our calculations indicate that under a pressure larger than 10 GPa, the main contribution to superconductivity comes from the H phonon modes, but down to very low pressure, the interaction between Be and H plays a key role in enhancing superconductivity while ensuring structural stability. This observation is surely relevant for future research of atmospheric-pressure high-temperature superconductors.

FIG. 5. Trend of calculated EPC parameters (λ) , logarithmic average phonon frequency (ω_{log}), and the estimated T_c for selected structures using the Allen-Dynes modified McMillan (ADM) equation [\[53\]](#page-7-0), and numerically solving the Eliashberg equations [\[63\]](#page-7-0) with $\mu^* = 0.10$ for *Fm*3 *AcBeH₈ at high pressures.*

IV. CONCLUSIONS

In summary, we have investigated the crystal structures and superconductivity of stable and metastable ternary Ac–Be– H compounds, combining crystal structure prediction under pressure and first-principle calculations. At 200 GPa we uncover three thermodynamically stable compounds with stoichiometries $AcBe₂H₁₀$, $AcBe₁H₈$, $AcBe₂H₁₄$, as well as four metastable superconducting compounds, close to the convex hull: $AcBe₂H₁₆$, fcc $AcBeH₁₀$, fcc $AcBeH₈$ and $AcBe₂H₁₂$. All these structures exhibit metallic nature. Electron-phonon coupling calculations show that $AcBe₂H₁₆$ and fcc $AcBeH₁₀$ are good phonon-mediated superconductors with T_c of 150 K at 200 GPa and 165 K at 300 GPa, respectively. Very interestingly, fcc $AcBeH_8$ remains dynamically stable down to 10 GPa where it exhibits a T_c of 181 K. The soft phonon vibration modes originating from Be–H interactions in BeH8 units contribute to the enhancement of superconductivity with decreasing pressure. We expect that our results will stimulate more research on ternary superconducting hydrides in the hope to uncover systems with high critical temperature and stability close to ambient pressure.

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