Optimization strategies developed on NiO for Heisenberg exchange coupling calculations using projector augmented wave based first-principles DFT+U+J

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High-performance batteries, heterogeneous catalysts, and next-generation photovoltaics often centrally involve transition metal oxides (TMOs) that undergo charge or spin-state changes. Demand for accurate DFT modeling of TMOs has increased in recent years, driving improved quantification and correction schemes for approximate DFT's characteristic errors, notably those pertaining to self-interaction and static correlation. Of considerable interest, meanwhile, is the use of DFT-accessible quantities to compute parameters of coarsegrained models such as for magnetism. To understand the interference of error corrections and model mappings, we probe the prototypical Mott-Hubbard insulator NiO, calculating its electronic structure in its antiferromagnetic I/II and ferromagnetic states. We examine the pronounced sensitivity of the first-principles calculated Hubbard parameters U and J to choices concerning projector augmented wave (PAW) based population analysis, we reevaluate spin quantification conventions for the Heisenberg model, and we seek to develop best practices for calculating Hubbard parameters specific to energetically metastable magnetic orderings of TMOs. Within this framework, we assess several corrective functionals using in situ calculated U and J parameters, e.g., DFT+U and DFT+U+J. We find that while using a straightforward workflow with minimal empiricism, the NiO Heisenberg parameter RMS error with respect to experiment was reduced to 13%, an advance upon the state-of-the-art. Methodologically, we used a linear-response implementation for calculating the Hubbard U available in the open-source plane-wave DFT code ABINIT. We have extended its utility to calculate the intra-atomic exchange coupling J; however, our findings are anticipated to be applicable to any DFT+U implementation.

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I. INTRODUCTION

Antiferromagnetic transition metal oxides (TMOs) are increasingly vital constituents of many technologies, from high-performance battery cathodes [1–5], to water-splitting catalysts [6,7], to efficient photovoltaic devices for sustainable energy storage [8–10]. Understanding the macroscopic magnetic properties of this class of material requires a sophisticated description of the competing spin-related energy scales relevant at microscopic scales.

A variety of simulation techniques exist to calculate the material properties of TMOs, such as DMFT [11–14], quantum Monte Carlo [15–18], and coupled cluster methods [19–21]. Each method comes with its own strengths and weaknesses, regimes of applicability, and computational cost and scaling. For example, among the more computationally inexpensive of such techniques is density functional theory (DFT), a popular electronic structure method that predicts material properties from first principles. However, it is well documented that approximate DFT struggles to accurately describe the strongly correlated electron systems for which the TMO class is prototypica [22–28]. By strongly correlated, in this context, we refer to the poor description of the electronic structure associated with localized subspaces within the local (LDA) and semilocal (GGA) mean-field approximations to

the XC energy functional [29]. This results from the challenge of describing exchange and correlation effects with only a local or semilocal functional dependence, and based on a single-determinant reference system [25,26,30–32].

Chief among the characteristic errors in the available computationally tractable local or semilocal approximate functionals [22–24] is the self-interaction error (SIE), that is the spurious exposure of an electron to its own Coulombic potential, and more broadly, the many-body SIE or delocalization error, a tendency to artificially diffuse the electron density [25–28,33]. Less discussed, but no less pervasive, is the static-correlation error (SCE) that particularly arises in degenerate and multivalence systems and, more generally, in systems of significant multireference character.

Computationally inexpensive techniques that may be interpreted as rectifying SIE, specifically, include the Hubbard-like corrective functionals. Originally inspired by the Hubbard model, DFT+U and related functionals [e.g., DFT+U, DFT+(U-J_z)] append to the approximate DFT Hamiltonian certain compensating energy terms that penalize fractional electron charges within defined subspaces. Moreover, a steadily increasing body of literature finds associations between SCE and the parameter J_z (simply denoted J in the literature; renamed here only to avoid confusion with the Heisenberg J) [28,34–36]. The efficacy of Hubbard corrective techniques depends on the underlying exchange-correlation functional, as well as the appropriate prescription of the accompanying constants, the Hubbard U and intra-atomic

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FIG. 1. The magnetic orderings of monoxide antiferromagnetic TMOs crystallizing in the rock salt structure, oxygen atoms (orange) and TMs (blue). Isomagnetic planes represented by color gradients: spin up (green) and spin down (navy).

exchange coupling J_z (collectively referred to herein as the Hubbard parameters) [34,37–40].

Many readily-available DFT programs have an implementation of DFT+U in different guises; however, relatively few among them have utilities for calculating the Hubbard parameters from first principles. In preparation for the present paper, we analyzed in detail the numerical performance of a linear-response implementation for calculating the Hubbard U that is embedded in the projector augmented wave (PAW) [41] functionality of the open-source plane-wave DFT package ABINIT [42,43]. Furthermore, we extended its capability to also calculate the parameter J_z from first principles. Our technical study here has led to the development of what is now called theABINIT LRUJ utility [44].

Within the context of Hubbard-corrected DFT, much scientific attention is directed towards understanding the effect of the corrective functionals and their parameters upon bond distances, band gap predictions [45-47], total energy differences [48-54], and other parameterized spectroscopic properties. Meanwhile, comparatively few studies have probed how DFT+U(+J) and related functionals impact metrics of magnetism, such as the magnetic exchange parameters in the Heisenberg model [54,55]. When calculated using total energy differences, these parameters become valid ground-state DFT quantities, and they thereby become concise indicators of the reliability of DFT+U(+J) total energies. Magnetic exchange parameters thus constitute expedient indicators of the quality of first-principle DFT+U(+J) total energy differences, which are often supposed to be within the method's regime of reliable applicability, albeit without much concrete literature evidence of that to date. It is essential, and even more so due to its widespread adoption, to work to establish the regime of reliability of first principles for DFT+U(+J) total-energy differences, and to explore and delineate its remaining weaknesses and ambiguities.

For these reasons, and given this context, we undertook a multi-pronged investigation designed to systematize the complexities of the magnetic Heisenberg model, firstprinciples Hubbard corrective functionals, intra-atomic spin parametrization schema, and the PAW method for the prototypical antiferromagnetic TMO, nickel oxide (NiO).

Following a working description of our methodology in Sec. II, we examine, in Sec. III A, the sensitivity of the *in situ* calculated Hubbard parameters U and J_z with respect to the available charge population analyses of the PAW method.

With Hubbard parameters defined, we map selected points in the energy-magnetization landscape of bulk NiO onto the classical Heisenberg model by calculating its electronic structure in its three main stable, metastable, or enforceable magnetic configurations (AF_I—antiferromagnetic in the [001] direction, AF_{II}—antiferromagnetic in the [111] direction, and ferromagnetic, FM; see Fig. 1). We extract Heisenberg (interatomic) exchange coupling parameters in terms of strict density-functional accessible total energy differences between these magnetic configurations.

In Secs. III B and III C, we explore the potential of incorporating not only total energies but also local moments in the mapping onto the Heisenberg model, and we assess several different ways to perform this. In doing so, we effectively reevaluate Heisenberg exchange coupling calculation conventions and scrutinize a variety of moment estimation techniques, both classical and semiclassical. Adopting this framework enables a direct and rigorous assessment of the efficacy of 6+ corrective Hubbard functionals via the quality of the Heisenberg exchange coupling parameters as compared to both experiment (i.e., magnon frequencies or inelastic neutron scattering) and ostensibly more comprehensive theory (e.g., hybrid functionals). We relay this comparative analysis in Secs. III D–III F, finishing with a closer look at the effect of Hubbard treatment on the O-2*p* sites in Sec. III G.

A. NiO

As a first-row transition metal harboring several accessible oxidation states, nickel is a relevant candidate for many sustainable technologies. Its monoxide, NiO, is promising as a *p*-type semiconductor with multiple technological applications, such as in photocathodes [56–58], lithium-ion battery cathodes for vehicles [59], OH- carrier battery cathodes for portable electronics [60], efficiency enhancers in solar cells [61], among others. It is relatively nontoxic, has a wide and tunable band gap (around 3.5-4.0 eV) [61–64], and is relative stable.

NiO crystallizes above the Néel temperature (523 K) in the rocksalt structure (space group Fm3m) with an experimental lattice parameter consistently estimated at around 4.17 Å (7.88 a_0) across a variety of diffraction studies [65–73]. Other experiments indicate that Ni sites in bulk NiO take on magnetic moments between 1.96 μ_B [70] and 2.2 μ_B [73–76]. While under certain conditions the TMO can be observed in many magnetic orderings, its ground-state magnetic ordering (MO) is AF_{II}, the isomagnetic planes of which alternate antiferromagnetically in the [111] direction (see schematic in Fig. 1) [65–67]. NiO AF_I and FM magnetic states yield experimental lattice parameters of 4.168 and 4.171 Å, respectively [77].

Bulk NiO is a well-studied system with a wealth of literature attesting to its total energy difference parameters. Its status as the prototypical Mott-Hubbard insulator renders NiO a good benchmark system through which to observe the effect of SIE corrective techniques on total energies. To date, relatively little is known of the systemic effects of the Hubbard parameters on magnetic exchange interactions in NiO, specifically of the Heisenberg model.

II. METHODOLOGY

A. The formalism of DFT+U

Inspired by the Hubbard model [78], DFT+U [45–47,79,80] offers effective treatment of SIE in well-localized electronic states while minimizing additional computational expense [81]. These corrective functionals are applied to predefined localized subspaces that are supposed to be poorly described by the underlying XC functional [29]. The total energy in DFT+U is defined as

$$E_{\text{DFT+U}}[\hat{\rho}] = E_{\text{DFT}}[\hat{\rho}] + E_{\text{U}}[\{n_{mm'}^{i\sigma}\}], \qquad (1)$$

where $\hat{\rho}$ is the Kohn-Sham density matrix, $E_{\text{DFT}}[\hat{\rho}]$ is the approximate DFT total energy, E_{U} is the corrective term, and $\{n_{mm'}^{i\sigma}\}$ refers to the set of projected Kohn-Sham density matrices of the local atomic subspaces that we wish to treat for SIE on atom *i* with spin σ (e.g., those spanned by atomic-like 3*d* orbitals on Ni atoms). Although all electronic subspaces are afflicted with SIE to some extent, not all are treated with Hubbard corrections, only those with some occupancy matrix eigenvalues significantly deviating from integer values and, correspondingly, spectral weight adjacent to the band-gap edges. Strictly speaking, many-body SIE is uniquely defined at present only for open quantum systems in their totality, and partitioning by subspaces implies that pragmatic choices are being made.

The original derivations of DFT+U considered the energy E_{Hub} associated with the Hubbard model and a singledeterminant assumption for the wavefunction, together with a double counting term E_{dc} for such a component of the latter that may be already included in the underlying Hartree and XC energy. The double-counting correction is usually considered to be dependent solely on $N^{i\sigma}$, the total electron occupation of each subspaces and spin component. The exact form of E_{dc} is not precisely known, a consequence of the fact that the XC constituent of the Kohn-Sham Hamiltonian is mutually unknown. One rotationally invariant formulation of DFT+U is known as the Dudarev functional [79,82],

$$E_{\mathrm{U}}[\{n_{mm'}^{i\sigma}\}] = E_{\mathrm{Hub}}[\{n_{mm'}^{i\sigma}\}] - E_{\mathrm{dc}}[\{N^{i\sigma}\}]$$
$$= \sum_{i,\sigma} \frac{U_{\mathrm{eff}}^{i}}{2} \mathrm{Tr}[\mathbf{n}^{i\sigma}(1-\mathbf{n}^{i\sigma})].$$
(2)

This formulation of DFT+U makes it obvious that the corrective terms preceded by the Hubbard parameters penalize fractional occupation of atomic orbitals, for positive U_{eff}^{i} values. When an eigenvalue of $\mathbf{n}^{i\sigma}$ assumes an integer value, the corresponding contribution to E_{U} is zero. By contrast, when that eigenvalue is maximally fractionalized at 0.5, the contribution to E_{U} is also maximized.

In Eq. (2), the effective parameter is $U_{eff} = U - J_z$, where U is the Hubbard parameter and J_z is the intra-atomic exchange parameter, more often denoted J but renamed here merely to avoid confusion with interatomic Heisenberg parameters. The degree of corrective power of the functional, then, is determined by the magnitudes of U and J_z . In the literature, U_{eff} and U are often referenced interchangeably depending on the consideration given to J_z [83]. An alternative and more elaborate functional that we reference here as DFT+U+J_z, in which J_z is not simply a mitigator of U, is described in Liechtenstein *et al.* [84]. For atomic *s* orbitals, the Liechtenstein functional reduces to the Dudarev functional of Eq. (2) with U^{*i*} = U_{eff}.

It is worth noting that the J_z parameter computed and employed here pertains, as defined, only to the spin population present in collinear spin DFT. It is elsewhere sometimes termed the Hund's coupling strength because it is affiliated with Hund's coupling-related exchange splitting that manifests in collinear-spin DFT. Strictly speaking, however, to measure that mechanism's strength, one would look instead at the energy prefactor of the effective interaction of the form $\mathbf{S} \cdot \mathbf{S}$. This for vector spin \mathbf{S} , either on a per-spatial-orbital or per-atomic-subspace basis, depending on the preferred definition as opposed to its z axis projection S_z . It is further worth noting that, based on recent investigations into the suitability of DFT+U-like functionals in correcting SIE and SCE, the two functionals used here, when applied to s orbitals including terms proportional to a positive-valued first-principles J_z, do not succeed in correcting for SCE and may even increase the magnitude of SCE [36].

Which Hubbard functional, and which subspaces require numerical attention in excess of the underlying XC functional, are both topical inquiries that this article does not directly address for NiO. For the purposes of brevity, we restrict our assessment to DFT+U, DFT+U_{eff} and DFT+U+J_z, built

upon the Perdew-Burke-Ernzerhof (PBE) GGA [85] XC functional. We first test the effect of these functionals when *in situ* treatment is administered to the Ni 3*d* orbitals alone (requiring calculation and application of U^{*d*} and J^{*d*}_z). Then, we test the same of functionals when *in situ* treatment is administered to both the Ni 3*d* and O 2*p* orbitals together (requiring calculation and application of U^{*d*}, J^{*d*}_z, U^{*p*}, and J^{*p*}_z; collectively referred to in each functional as U^{*d*, *p*} and J^{*d*, *p*}_z). More information is provided in Sec. II E.

B. Calculation of the Hubbard parameters via linear response

The Hubbard parameters U and J_z are intrinsic, measurable ground-state properties of any multi-atomic system treated with a given approximate XC functional and thus objects that admit first-principles calculation [80,86]. They are related to the interaction contribution to spurious curvature of the total energy with respect to subspace occupation and magnetization, respectively. When properly calculated, the Hubbard parameters are expected to appropriately set the strength of the corrective functionals and precisely cancel the systematic errors characteristic of approximate DFT [80,86,87].

Many methodologies for the *in situ* calculation of the Hubbard U parameter exist, such as minimum tracking linear response [88,89], the constrained random phase approximation (cRPA) [90,91], and the more recent methodology based on density functional perturbation theory (DFPT) [92], to name a few. We focus here on the method from Cococcioni and de Gironcoli [37], who, following from the work of Pickett *et al.* [87], developed a practical protocol for calculating U from first-principles called the self-consistent field (SCF) linear response method. This method eliminates reliance on external empirical data and paves the road for more accurate simulations of obscure or yet-to-be discovered materials. We refer the reader to these references for a more comprehensive picture of the theory and the mathematical formalism behind the SCF linear response [37,87,93–95].

The SCF linear response approach begins with the application of a small, uniform perturbation to the external potential of the subspace for which U is under assessment. The change in electronic occupation induced by that perturbation is then monitored. The occupancy response is, for small perturbations, ordinarily expected to be a linear function of the perturbation's magnitude.

Formulated practicably, the SCF linear response procedure for calculating U is as follows [35,89]. To begin, we converge a calculation to the ground state. We then individually (in parallel restarts) apply several weak perturbations of strength { α } in equal magnitude to the up and down spin channels of the external potential. In the basis of the orbitals that define the subspace *i*, this contribution to the external potential $V_{mm'}^{i,\sigma}$ is defined in terms of the subspace projection operator $P_{mm'}^{i}$ by

$$V_{mm'}^{i,\sigma} = \alpha P_{mm'}^i \delta_{m'm}.$$
 (3)

Here, $P_{mm'}^i$ is replaced by $\tilde{P}_{mm'}^i$ via Eq. (11) in the PAW case, which may be approximated in different ways as we later discuss. Once the potential perturbation is applied, but before the density is calculated and the Hamiltonian updated to start the new iteration, the spin-up and spin-down charge occupations on the perturbed atom change in response. Next, we must

distinguish between the interacting response and the response arising from a conjectured system in which electrons do not interact [i.e., one with Kohn-Sham (KS) energy] [80,93,96]. Within this system, we are interested only in the change in total noninteracting occupation $(n_0^{\uparrow} + n_0^{\downarrow})$ of subspace S with respect to α , which we call the noninteracting response function χ_0 , defined as

$$\chi_0 = \frac{d(n_0^{\uparrow} + n_0^{\downarrow})}{d\alpha}.$$
 (4)

We harvest n_0^{\uparrow} and n_0^{\downarrow} , the noninteracting spin-up and spindown occupations, respectively, after the first self-consistent iteration. The system then attempts to screen the perturbation, effectively compensating for the disturbance by reorganizing its charge once again until, after the last self-consistent iteration, it reaches an equilibrium state. The derivative of this equilibrium occupation with respect to the perturbation magnitude is the interacting response function,

$$\chi = \frac{d(n^{\uparrow} + n^{\downarrow})}{d\alpha},\tag{5}$$

where n^{\uparrow} and n^{\downarrow} are the ground-state occupations of S harvested after the last self-consistent iteration. Finally, we invert the response functions and take their difference to acquire the Hubbard U via the Dyson equation

$$U = \chi_0^{-1} - \chi^{-1}.$$
 (6)

The linearity condition manifests primarily in the nature of χ and χ_0 as functions. However, the response behavior is not always linear. Figure 2 demonstrates this for an exceptionally ill-behaved system. With some exceptions, linearity is expected in the limit of small perturbations, but this region does not always exist, particularly when the response is excessively hard (e.g., close to full filling). If the perturbations are too large, of course one can generally expect some nonlinear behavior. Nonlinearity, and sometimes asymmetry across the zero-perturbation axis, is also expected if the system has a particularly shallow energy landscape, (that is, when the response is excessively soft, e.g., close to a phase transition). To compensate for such anomalies, we may fit a regression function (typically polynomial) to the data and differentiate it at $\alpha = \beta = 0.0$ eV to find the response functions.

Ideally, U is calculated for a cell of infinite size such that the perturbed subspace is isolated from its periodic images. Since this is unfeasible computationally, U must be converged with respect to an increasing number of atoms, organized into a roughly cubic supercell to isotropically distribute the effect of the perturbation.

The intra-atomic exchange parameter J_z can be calculated similarly [35,89], following the proposal of Ref. [98]. However, instead of monitoring the change in total subspace occupancy as a function of the applied perturbation α , the J_z relates to changes in magnetization, or difference between the up and down spin occupancies (i.e., $n^{\uparrow} - n^{\downarrow}$), in response to perturbations { β } applied in positive magnitude to the spin-up potential and in negative magnitude to the spin-down potential, specifically (adding PAW augmentation as required)

$$V_{mm'}^{i,\uparrow} = \beta P_{mm'}^{i} \delta_{m'm}, \quad V_{mm'}^{i,\downarrow} = -\beta P_{mm'}^{i} \delta_{m'm}.$$
(7)



FIG. 2. (Top) Example of the linear response (LR) procedure with first-order polynomial regression only, conducted to find the Hubbard *U* for O 2*p* orbitals on AF_{II} NiO. (Bottom) J_z parameter LR for Ni 3*d* orbitals on AF_I NiO. Line color indicates first-, secondor third-order polynomial fit on the data. Axes are aligned with ground-state (unperturbed) data. Note that χ_0 , χ_{M_0} data points and fits are transformed to account for technical considerations [97].

Noting the minus-sign convention, within SCF linear response J_z is given by [35]

$$J_{z} = \frac{d\beta}{d(n^{\uparrow} - n^{\downarrow})} - \frac{d\beta}{d(n_{0}^{\uparrow} - n_{0}^{\downarrow})} = \chi_{M}^{-1} - \chi_{M_{0}}^{-1}.$$
(8)

In the Appendix we show that χ_0 and χ_{M_0} are equivalent, a particularly convenient result for verifying a correct implementation. We note in passing that Eqs. (6) and (8) have been derived for use within approximate collinear-spin Kohn-Sham DFT, where only the *z* projection of spin magnetic moments is considered in the treatment of exchange-correlation effects. The extension of such error quantities to approximate noncollinear spin DFT is a worthy topic outside the scope of this investigation.

We emphasize that we denote the intra-atomic exchange parameter J_z as such only to avoid confusion with the interatomic Heisenberg exchange parameters J, not because it is thought to differ significantly from the familiar J in the context of DFT+U type functionals. We note that, defined as they are here, the U and J_z can be thought of as error descriptors for SIE and SCE, respectively, in the approximate exchange-correlation functional within the subspace-bath decoupling picture of DFT+U rather than as absolute interaction strengths. Furthermore, on the topic of interpretation and terminology, we note that J_z is an *exchange* parameter only in the magnetic sense of that word, and it may comprise a significant or even dominant contribution from *correlation* (as distinct from exchange, in the sense of accessibility at the Hartree-Fock level of theory). Indeed, as calculated here it will, by construction, contain a "spin-flip exchange" contribution, which is a contribution to correlation as it does not appear in a Hartree-Fock treatment [99].

Based on these definitions, notwithstanding, one may deduce that changes in the atomic magnetic moments quantities that are informed by the MO of the system—will induce noticeable differences in the Hubbard parameters. That is, the U and J_z for a particular subspace in FM NiO is expected to be different than those calculated for AF_I or AF_{II} NiO. However, it is not clear whether the total energies resulting from the application of these MO-specific parameters to their corresponding system remain comparable to each other. We test and discuss this point in Sec. III D.

C. PAW-based population analysis

The special ingredient in both the linear response determination of the Hubbard parameters as well as their host corrective functionals is the subspace occupation matrix $\mathbf{n}^{i\sigma}$ and its elements $\{n_{mm'}^{i\sigma}\}$. The DFT+U($\pm J_z$) occupation matrix elements are defined as

$$n_{mm'}^{i\sigma} = \sum_{k,v} f_{kv}^{\sigma} \langle \psi_{kv}^{\sigma} | \phi_{m'}^{i} \rangle \langle \phi_{m}^{i} | \psi_{kv}^{\sigma} \rangle, \tag{9}$$

where ψ_{kv}^{σ} is the Kohn-Sham (KS) orbital corresponding to k-point k and band index v, ϕ_m^i is the localized basis function assigned to magnetic quantum number m on atom i, and f_{kv}^{σ} is the Fermi-Dirac distribution function applied to the KS orbitals.

The machinery that isolates charged subspaces for population analysis are the projector functions. The occupation matrix in Eq. (9) is a generalized projection of the KS density matrix onto a localized basis set of choice, where that choice is typically limited by the options made available by the developers of the DFT+U(\pm J_z) implementation at hand. Reliance on projector functions that are unsuitable reflections of the correct localization, either by excessive localization, unphysical discontinuities, or by excessive overlap [100], will result in erroneous population analysis and, by consequence, inefficient or otherwise defective corrective protocol [37].

In ABINIT, the DFT+U(\pm J_z) formalism is built into the program's projector augmented wave (PAW) [41] functionality, so the population analysis relies heavily on the PAW projectors. While an understanding of PAW and its pseudopotential generation schemata is not imperative for the following sections, we encourage the reader to consult, for a more global understanding of the PAW functionality, Refs. [[41,43,101–104]]. It suffices to know that the namesake PAW projectors [41] are one of three predefined basis sets fed into DFT algorithms via the pseudopotential apparatus. These projectors are designed to isolate a spherically symmetric, element-specific region—an augmentation sphere—about each atomic site, using a relatively small cutoff radius r_c .

Within PAW, the DFT+U($\pm J_z$) occupation matrix elements are obtained via the pseudo (PS) wavefunctions $\widetilde{\Psi}_{kv}^{\sigma}$ by computing

$$n_{mm'}^{i\sigma} = \sum_{k,v} f_{kv}^{\sigma} \langle \widetilde{\Psi}_{kv}^{\sigma} | \widetilde{P}_{mm'}^{i} | \widetilde{\Psi}_{kv}^{\sigma} \rangle.$$
(10)

Assuming that the charge of the subspace under scrutiny is localized about the atom, we may use the definition of a local PAW PS operator (Eq. 11 in Ref. [41]) to expand the pseudized projection operator $\tilde{P}_{mm'}^{i}$ in terms of the KS projection operator $P_{mm'}^{i}$ and the PAW basis functions: the all-electron (AE) basis ϕ_{i} , the PS basis $\tilde{\phi}_{i}$, and the projectors \tilde{p}_{i} . That is

$$\widetilde{P}_{mm'}^{i} = P_{mm'}^{i} + \sum_{IJ} |\widetilde{p}_{I}\rangle\langle\phi_{I}|P_{mm'}^{i}|\phi_{J}\rangle\langle\widetilde{p}_{J}| - \sum_{IJ} |\widetilde{p}_{I}\rangle\langle\widetilde{\phi}_{I}|P_{mm'}^{i}|\widetilde{\phi}_{J}\rangle\langle\widetilde{p}_{J}|, \qquad (11)$$

where *I* is a shorthand index assigned to sets of $(i, \ell_i, m_{\ell_i}, n_i)$, which respectively refer to the site index, angular quantum number, magnetic quantum number, and PAW partial-wave index. Similarly, an analogous set of numbers $(j, \ell_j, m_{\ell_j}, n_j)$ pertains to a different site *J*. When Eq. (11) is inserted into Eq. (10), three terms emerge, two of which cancel because a requisite in the formation of the PAW basis functions assumes that $\sum_i |\widetilde{p}_i\rangle\langle \widetilde{\phi}_i| = 1$ is maintained inside the augmentation region. Acknowledging that $\phi_i = \phi_{l_i,m_i,n_i} = \phi_{n_i}Y_{l_i,m_i}$, we are left with

$$n_{mm'}^{i\sigma} = \sum_{IJ} \rho_{IJ}^{\sigma} \langle \phi_{n_i} | P_{mm'}^i | \phi_{n_j} \rangle, \qquad (12)$$

where ϕ_n are the radial parts of the PAW AE basis functions, and the PAW-sphere density matrix is

$$\rho_{IJ}^{\sigma} = \sum_{k,v} f_{kv}^{\sigma} \langle \widetilde{\Psi}_{kv}^{\sigma} | \widetilde{p}_{I} \rangle \langle \widetilde{p}_{J} | \widetilde{\Psi}_{kv}^{\sigma} \rangle.$$
(13)

The distribution of charge density about an atom is, in practice, oblivious to the sharp truncation of these PAW projectors at the cutoff radius; even the charge of supposedly localized orbitals often escapes beyond the confines of the augmentation sphere. Care must be taken in selecting an appropriate projector, as the selection can have far-reaching numerical implications [101–106].

Four PAW occupation matrix projector options are available in ABINIT. For simplicity and ease of reference, we refer to each option by the integer value of its ABINIT input file variable, dmatpuopt (Density MATrix for PAW + U OPTion), which may take on values one through four [107]. When dmatpuopt = 1, occupations are projections on atomic orbitals [43], per the definition

$$n_{m,m'}^{i,\sigma} = \sum_{IJ} \rho_{IJ}^{\sigma} \langle \phi_{n_i} | \phi_0 \rangle \langle \phi_0 | \phi_{n_j} \rangle, \qquad (14)$$

where ϕ_0 are normalized, bound state atomic eigenfunctions pertaining to PAW partial-wave index $n_i = 1$, drawn from the list of AE wavefunctions pertaining to the subspace in question [108] PAW datasets generated via the *atompaw* code [103] will always feature a normalized, bound atomic eigenfunction as the first atomic wavefunction of a PAW dataset. Step 4 on page 2 of Ref. [109] states clearly that *atompaw* mandates the use of "atomic eigenfunctions related to valence electrons (bound states)" as the partial waves included in the PAW basis. Therefore, all PAW datasets generated by *atompaw*, including the JTH sets on PseudoDojo [110], have atomic eigenfunctions as the first atomic wavefunctions of the correlated subspace. Their normalization, however, depends on the pseudo partial-wave generation scheme. *atompaw* provides two options for this scheme: the Vanderbilt or the Blöchl. We are able to reasonably infer that the JTH table of PAW datasets [110] available on the PseudoDojo website, match all criteria as a suitable dataset with which one may use dmatpuopt = 1. Importantly, the integrals $\langle \phi_{n_i} | \phi_0 \rangle$ and $\langle \phi_0 | \phi_{n_j} \rangle$ are computed within the PAW augmentation sphere.

By contrast, when dmatpuopt = 2, which is the default setting, occupations are integrated in PAW augmentation spheres of charge densities decomposed by angular momenta, per

$$n_{m,m'}^{i,\sigma} = \sum_{IJ} \rho_{IJ}^{\sigma} \langle \phi_{n_i} | \phi_{n_j} \rangle.$$
 (15)

Again, $\langle \phi_{n_i} | \phi_{n_j} \rangle$ is an integral defined only in the PAW augmentation region [43]. Equation (15) may be derived from Eq. (10) by setting [106,111]

$$P^{i}_{mm'}(\mathbf{r},\mathbf{r}') = 1_{\circ}(\mathbf{r})\delta(|\mathbf{r}'-\mathbf{R}_{i}| - |\mathbf{r}-\mathbf{R}_{i}|)$$
$$\times Y_{\ell m}(\hat{\mathbf{r}})Y^{*}_{\ell m'}(\hat{\mathbf{r}}'), \qquad (16)$$

where δ is the Dirac-Delta function that effectively counts spatial overlap, $1_{\circ}(\mathbf{r})$ is a step function equal to unity when \mathbf{r} is inside the augmentation region and zero elsewhere, and $Y_{\ell m}$ are the spherical harmonics.

When dmatpuopt ≥ 3 , occupations take the same form as Eq. (14) scaled by normalization constants. When dmatpuopt = 3 or dmatpuopt = 4,

$$n_{m,m'}^{i,\sigma} = \frac{1}{\mathcal{N}_0^{(\text{dmatpuopt}-2)}} \sum_{IJ} \rho_{IJ}^{\sigma} \langle \phi_{n_i} | \phi_0 \rangle \langle \phi_0 | \phi_{n_j} \rangle, \qquad (17)$$

where \mathcal{N}_0 is a normalization constant representing the overlap between atomic wavefunctions of the bounded AE basis function inside the PAW augmentation sphere, delimited by cutoff radius r_c , per

$$\mathcal{N}_0 = \int_0^{r_c} \phi_0^2(r) \, dr. \tag{18}$$

The use of dmatpuopt = 3 is equivalent to Scheme o_B in Eq. (C2) in Geneste *et al.* [105], wherein the PAW-truncated atomic orbitals ϕ_0 are effectively renormalized according to the percentage of the total orbital charge inhabiting the augmentation region. When dmatpuopt = 4, both the ϕ_0 and the part of the KS orbital falling inside the augmentation sphere are supposedly renormalized (only if the KS orbital matches the atomic orbital in shape), so that \mathcal{N}_0 is squared in the denominator.

Based on the description in Timrov *et al.* [112], the planewave DFT code QUANTUM ESPRESSO [113,114] uses none of these options [see Eqs. (9), (10) and (A4) in Timrov *et al.*]. Instead, this schema takes the full PS atomic orbitals and

TABLE I. Exchange coupling parameters J_{ij}^* across various conventions of the Heisenberg Hamiltonian in terms of the exchange coupling parameters J_{ij} used here, where J_{ij}^* are the parameters and γ the real constant pertaining to that particular convention. The Heisenberg convention used here (orange) assumes that there is no interaction between one magnetic site and itself. Sometimes this assumption is made explicit by replacing the sum over ij with that over $i \neq j$. The use of angle brackets denotes a set of all pairs $\{i, j\}$ excluding $\{j, i\}$. A similar table may be found in He *et al.* [115].

| Convention | J_{ij}^* |
|--|---------------------------|
| $H = -\sum_{ij} J_{ij}^* \mathbf{S}_i \cdot \mathbf{S}_j$ | J_{ij} |
| $H = \gamma \sum_{ij} J_{ij}^* \mathbf{S}_i \cdot \mathbf{S}_j$ | $\frac{-1}{\gamma}J_{ij}$ |
| $H = \gamma \sum_{\langle ij \rangle} J_{ij}^* \mathbf{S}_i \cdot \mathbf{S}_j$ | $\frac{-2}{\gamma}J_{ij}$ |
| $H = \gamma \sum_{i>j} J_{ij}^* \ \mathbf{S}_i \cdot \mathbf{S}_j$ | $\frac{-2}{\gamma}J_{ij}$ |

augments them with an overlap operator defined in terms of the overlap of the PAW basis functions. In this way, the population analysis avoids relying on orbitals truncated at the PAW cutoff radius and is conducted without significant approximation. For this reason, we calculate the Hubbard U via linear response first in QUANTUM ESPRESSO for use as a benchmark to determine which PAW-based population analysis involving truncated orbitals is most effective. The results of this investigation are itemized in Table III.

D. The Heisenberg model

The prevalence of DFT calculations in predicting properties of quantum materials has impelled the development of several effective spin Hamiltonian methods that are compatible with electronic structure theory. Many of these classical methods rely on the justifiable assumption that electron spin, and thus electron magnetic moments, can be treated as discrete vectors, which maintain rigidity in their magnitudes despite rotation.

The simplest effective spin Hamiltonian is the classical Heisenberg model, which long predates DFT and assumes that the magnetic moments of electrons are localized, thus defining the Hamiltonian

$$H = -\sum_{ij} J_{ij} \mathbf{S}_i \cdot \mathbf{S}_j, \tag{19}$$

where *i* and *j* represent all magnetic centers of the system under consideration, and S_i and S_j are their corresponding total atomic spin vectors. In the literature there is no unique formulation of Eq. (19), and variation exists based on the definition of the exchange coupling parameters J_{ij} and the constants (i.e.,-1, 2, 1/2) that they absorb. Table I outlines these variations in convention and how they relate to that used here.

We dissect Eq. (19) in the following sections to further deduce how it may best operate in a DFT context.

1. Exchange coupling parameters

The Heisenberg model quantifies the phenomenon of interatomic exchange, an effective long-range manifestation of the Pauli exclusion principle, which forbids doubly occupied quantum states. Exchange in the intra-atomic short range, by contrast, leads to Hund's rules and is several orders of magnitude stronger than its interatomic counterpart [116]. The exchange coupling parameters between magnetic atom centers J_{ij} can be considered spin-based measures of repulsion between electrons on neighboring atoms. When $J_{ij} > 0$, a ferromagnetic interaction is preferred, meaning the atoms will adopt the same spin orientation, while $J_{ij} < 0$ indicates the atomic spins would rather oppose each other in an antiferromagnetic ordering.

We consider two exchange parameters (refer to the inset in Fig. 3 for visualization) following Refs. [117–119]. The first is exchange between nearest-neighbor (NN) Ni atoms (J_1) , wherein the overlap of the t_{2g} states of the 3*d* orbitals on NN Ni atoms gives rise to a strongly AF direct exchange. This antiferromagnetic disposition is mitigated, however, by a primarily FM 90°-oriented indirect exchange, also known as superexchange. This round-the-corner exchange mechanism is strongly mediated by the intra-atomic Coulomb exchange between orthogonal 2*p* orbitals on the pivotal oxygen atom. The schematic in Fig. 3 illustrates why we expect this superexchange to favor ferromagnetic alignment. Overall, we expect the FM superexchange to overpower the AF direct exchange, meaning that J_1 will adopt a positive value overall.

The schematic in Fig. 3 also illustrates why we expect J_2 , a quantifier of next-nearest-neighbor (NNN) exchange interactions separated by 180° , to adopt a profoundly negative value, indicating strong preference for AF alignment.

With these phenomena in mind, we map the electronic Hamiltonian onto the Heisenberg model and look at J_1 and J_2 of NiO. We opt to compute the exchange coupling parameters using total energy differences per Ni-O pair resulting from relaxing different magnetic configurations of the same NiO solid in the rocksalt structure. The parameters may be computed as

$$J_1 = \frac{1}{16} (E_{AF_1} - E_{FM})$$
, and (20)

$$J_2 = \frac{1}{48} \left(4E_{\rm AF_{II}} - E_{\rm FM} - 3E_{\rm AF_{I}} \right), \tag{21}$$

where the $E_{\rm MO}$ are the total energies per Ni-O pair achieved by relaxing NiO in each of its three main stable or metastable magnetic configurations. These orderings are depicted in Fig. 1. A similar figure can be found in Ref. [120].

2. Spin parametrization and magnetic moment

An electron with quantum numbers (n, ℓ, m_ℓ, m_s) will have spin magnetic moment $\mu_s = -g_e \mu_B s$, where $g_e \approx 2.002318$ is the g factor for a free electron and $\mu_B = \frac{e\hbar}{2m} = \frac{1}{2}$ [a.u.] is the Bohr magneton [121,122]. The total spin magnetic moment is the sum of spin magnetic moments of all electrons on that atom $\mu_S = -g_e \mu_B S$ (where **S** is the summation over electrons) [122,123]. Referencing atomic magnetic moments in units of μ_B , then, **S** $[\mu_B] = -\mu_S/g_e$. In atomic units, **S** [a.u.] = $-2\mu_S/g_e$.



FIG. 3. Schematic of *xy* plane (z = 0) of an AF_{II} NiO in the rocksalt structure, demonstrating how the crystal field splitting influences the preferred orientation of the Heisenberg interatomic exchange interactions J₁ (black) and J₂ (teal) corresponding, respectively, to the interactions between nearest neighbor transition metal sites and next-nearest-neighbor TM sites. More information is provided in Sec. II D.

Conventionally, the \mathbf{S}_i of Eq. (19) are considered (a) projections $(\mathbf{S}_z)_i$ of the quantum spin vector \mathbf{S}_i on the quantization axis, and (b) of unit magnitude in accordance with nickel's expected oxidation state of +2, corresponding to an atomic magnetic moment of +2 μ_B (= 1 in atomic units, hence $\mu_S^2 \approx 1$). We seek to reevaluate these conventions given the multitude of DFT-accessible population analyses that can influence, to a large degree, the predicted magnetic moment on Ni atoms.

For example, in Sec. II C, we found that the PAW population analysis extends naturally to a definition of the magnetic moment that is both atom centered and restricted to the augmentation sphere. That is, for magnetic centers, we take

$$\left(\mathbf{S}_{z}^{\mathrm{AS}}\right)_{i} = \int_{|\mathbf{r}|=0}^{r_{c}} \left(\rho^{\uparrow}(\mathbf{R}_{i}+\mathbf{r}) - \rho^{\downarrow}(\mathbf{R}_{i}+\mathbf{r})\right) d\mathbf{r}.$$
 (22)

However, because the PAW augmentation spheres do not envelop all the charge in the system, leaving some interstitial charge uncounted, this option is expected to produce moments smaller than expected. We may address this by normalizing the subspace occupations via the projector, as in Sec. II C, and calculating

$$\left(\mathbf{S}_{z}^{\mathrm{PAW}}\right)_{i} = \sum_{m} n_{m,m}^{i,\uparrow} - n_{m,m}^{i,\downarrow}, \qquad (23)$$

which, by consequence, encodes the choice of PAW projectors described in Sec. II C. Alternatively, we may obtain magnetic moments via the extraction of S_z from the entire spin density (SD), by using

$$\left(\mathbf{S}_{z}^{\mathrm{SD}}\right)_{i} = \frac{1}{N_{\mathrm{Ni}}} \int |\rho^{\uparrow}(\mathbf{r}) - \rho^{\downarrow}(\mathbf{r})| \ d\mathbf{r}, \qquad (24)$$

where N_{Ni} is the number of Ni atoms in the system. While it produces larger magnetic moments, this method is accompanied by the particular limitation that the atom-specific magnetic moments are no longer resolved, and any moments formed near the oxygen atoms are absorbed into those of the magnetic centers. It is conventional in the relevant literature to consider oxygen atoms as mere extensions or mediators of the magnetic character of the TM magnetic centers, thereby restricting assessment of exchange interactions to those between Ni atoms. A similar study conducted by Logemann *et al.* [118] investigates the appropriateness of this convention. In using Eq. (24), we follow this study.

It should be noted, however, that the Heisenberg Hamiltonian in Eq. (19) calls for the use of the quantum spin vectors \mathbf{S}_i , as opposed to their scalar and DFT-accessible counterparts, the expectation values of the *z* axis projection of the spin angular momentum \mathbf{S}_z . Suppose we instead consider an approximate quantum model in which $|\mathbf{S}| = \sqrt{\langle \mathbf{S}^2 \rangle}$. Coupling

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the spins on each magnetic site such that s = 1 for the two unpaired spins on Ni together, the expectation value of the spin angular momentum of an atomic system is

$$|\mathbf{S}| = \sqrt{\langle \mathbf{S}^2 \rangle} = \hbar \sqrt{s(s+1)} \mathbf{S}_z$$
$$= \sqrt{2} \mathbf{S}_z \text{ [a.u.].}$$
(25)

That is, in a sort of crude quantum approximation, we scale the DFT-accessible magnetic moments, which may be thought of as time averages of precessing, canted spins, by a factor of $\sqrt{2}$. To our knowledge, this approximation has been made to magnetic properties in other contexts—such as in the works of Harrison [124], Kotani *et al.* [125], and Wan *et al.* [126]—but not tested systematically against other DFT-accessible spin objects for NiO.

Once we settle on an appropriate treatment of the spin vector, we can examine how it factors into the numerical calculation of the exchange coupling parameters. Equations (20) and (21) can be considered simplifications of a more generalized set of total energy difference equations under the Heisenberg model, derived by considering all NN and NNN of one Ni atom (or Ni-O pair) in the cell. They are solutions to a system of linear equations mapping the total energy of each MO onto the Heisenberg model, solutions that simplify when we impose assumptions that the spin magnetic moments of all atoms are (a) integers and (b) the same regardless of the cell's magnetic environment. That is, the magnitudes of the atomic magnetic moments are $|S_{FM}| = |S_{AF_T}| = |S_{AF_T}| = 1$ in

$$E_{\rm FM} = E_0 - 12J_1 S_{\rm FM}^2 - 6J_2 S_{\rm FM}^2,$$

$$E_{\rm AF_I} = E_0 + 4J_1 S_{\rm AF_I}^2 - 6J_2 S_{\rm AF_I}^2,$$

$$E_{\rm AF_{II}} = E_0 + 6J_2 S_{\rm AF_{II}}^2.$$
(26)

Suppose we assume instead that $|S_{FM}| \neq |S_{AF_1}| \neq |S_{AF_1}| \neq |S_{AF_1}| \neq 1$. Since DFT uses the noninteger spin degree of freedom to deepen the total energy well of each MO, and since that relaxed magnetization is accessible, we can perhaps explicitly use the final magnetization to better weigh the energetic contributions of each MO to the exchange coefficients. Then the system of linear equations in Eqs. (26), solved for J_1 and J_2 , becomes [127]

$$J_{1} = \frac{E_{AF_{II}}(S_{AF_{I}}^{2} - S_{FM}^{2}) + E_{AF_{I}}(S_{AF_{II}}^{2} + S_{FM}^{2}) - E_{FM}(S_{AF_{I}}^{2} + S_{AF_{II}}^{2})}{4[3S_{AF_{II}}^{2}S_{FM}^{2} + S_{AF_{I}}^{2}(S_{AF_{II}}^{2} + 4S_{FM}^{2})]} \quad \text{and}$$
(27)

$$J_{2} = \frac{E_{AF_{II}}(S_{AF_{I}}^{2} + 3S_{FM}^{2}) - 3E_{AF_{I}}S_{FM}^{2} - E_{FM}S_{AF_{I}}^{2}}{6[3S_{AF_{II}}^{2}S_{FM}^{2} + S_{AF_{I}}^{2}(S_{AF_{II}}^{2} + 4S_{FM}^{2})]}.$$
(28)

Note that if we set $|S_{FM}| = |S_{AF_I}| = |S_{AF_I}| = 1$ then Eqs. (27) and (28) reduce to Eqs. (20) and (21).

Exchange coupling parameters that incorporate differences in magnetization across MOs are not immediately comparable to those reported in previous studies. We find this is not so much an artefact of the assumption that $|S_{FM}| = |S_{AF_I}| =$ $|S_{AF_{II}}| = \sigma$ (where σ is a constant) so much as it reflects the assumption that $\sigma = 1$. As a comparative intermediary, therefore, we look at the exchange interaction parameters when the sum of electrons in a Ni atom corresponds directly to the magnitude of the Ni magnetic moment in the ground-state magnetic configuration of NiO (i.e., when $\sigma^2 = S_{AF_{II}}^2$ [a.u.]). Under this approach, Eqs. (27) and (28) reduce to

$$J_{1} = \frac{1}{16 S_{AF_{II}}^{2}} (E_{AF_{I}} - E_{FM}) \text{ and } (29)$$

$$J_2 = \frac{1}{48 \text{ } \text{S}_{\text{AF}_{\text{II}}}^2} (4E_{\text{AF}_{\text{II}}} - 3E_{\text{AF}_{\text{I}}} - E_{\text{FM}}).$$
(30)

For the purposes of clarity, we will refer to the first method of calculating exchange interaction parameters, those embodied by Eqs. (20) and (21), as Method A. Similarly, Eqs. (29) and (30) are henceforth referenced as Method B, and Eqs. (27) and (28) as Method C.

E. Computational details

We use different runtime parameters for computing the Hubbard parameters and applying them due to the different simulation cell and attendant Brillouin zone sizes needed for each task. This information, for all parameters that varied across tasks and MO (FM, AF_I , and AF_{II}), are laid out in Table II.

Otherwise, consistent across all calculations, we use the Perdew-Burke-Ernzerhof (PBE) GGA [85] XC functional and the Jollet-Torrent-Holzwarth PBE PAW pseudopotentials (Version 1.1) [110] generated via *atompaw* [103]. Derived from those pseudopotentials, the PAW augmentation sphere cutoff radius r_c used in, for example, Eqs. (16) and (22), is 1.81 a_0 for Ni and 1.41 a_0 for O. We use the experimental lattice parameters pertaining to each MO of NiO specified in Table II, and for those that have metallic character, we use the Marzari-Vanderbilt smearing scheme [128] with 0.005 Ha broadening. Furthermore, we enforce a cutoff energy of 33.1 Ha (44.1 Ha on the PAW atom augmentation sphere). In terms of the magnetic moment, we extract S_z in accordance with Eq. (24) using C2x [129].

For brevity, we test six Hubbard functionals: DFT+U^{*d*}, DFT+U^{*d*,*p*}, DFT+U^{*d*}_{eff}, DFT+U^{*d*,*p*}, DFT+U^{*d*} + J^{*d*}_z, DFT+U^{*d*,*p*} + J^{*d*,*p*}_z. Furthermore, we seek to understand if total energy differences between MOs are most comparable when calculated using Hubbard parameters that are (a) specific to each MO or (b) the same across MOs. To test (a), we apply each of the aforementioned Hubbard functionals with MO-specific Hubbard parameters (HPs) to its respective magnetically ordered version of NiO (that is, FM HPs applied to FM NiO via each functional, AF_I HPs to AF_I NiO, and so on). To test (b), we apply the U and J_z from the ground-state TABLE II. Calculation parameters specific to each magnetic ordering of NiO for both the calculation and application of the Hubbard parameters, respectively. We include here number of atoms, experimental lattice parameter a_0 , lattice vectors, converged energy tolerance, k-point sampling, insulating character, and Hubbard U and Hund's J_z parameters calculated for the Ni 3d and 2p subspaces respectively, calculated with the population analysis schema of Eq. (17) (see Sec. II C for a discussion of population analysis schemes).

| | DFT | LR Calculat | ions | $DFT+U(\pm J_z)$ Exchange Calculations | | | | |
|--|-----------------------|--|----------------------|--|--|---|--|--|
| | | $\mathbf{AF_{I}}$ | AFII | | $\mathbf{AF_{I}}$ | $\mathbf{AF}_{\mathbf{II}}$ | | |
| | | | | | | | | |
| # Atoms | | 64 | | 4 | | | | |
| Experimental a (Å) | 4.171^{a} | 4.168^{b} | 4.170^{b} | $4.171^{\rm a}$ | 4.168^{b} | 4.17^{b} | | |
| Lattice Vectors | | $egin{array}{c} [2,0,0] \ [0,2,0] \ [0,0,2] \end{array}$ | | $\begin{matrix} [1/2, 1/2, 0] \\ [0 \ , \ 0 \ , \ 1] \\ [1/2, -1/2, 0] \end{matrix}$ | $\begin{matrix} [1/2, 1/2, 0] \\ [0 \ , \ 0 \ , \ 1] \\ [1/2, -1/2, 0] \end{matrix}$ | $\begin{matrix} [1/2, 1/2, 1] \\ [1, 1/2, 1/2] \\ [1/2, 1, 1/2] \end{matrix}$ | | |
| Energy Tolerance (Ha) | | 1×10^{-7} | | | 1×10^{-12} | | | |
| k-Point Sampling | $3 \times 3 \times 3$ | $3 \times 3 \times 3$ | $5\times5\times5$ | $40\times 20\times 40$ | $40\times 20\times 40$ | $30\times 30\times 30$ | | |
| Insulating Character | metal | metal | insulator | insulator | insulator | insulator | | |
| $\mathbf{U}^{d} \mid \mathbf{J}^{d}_{\mathbf{z}} \; (\mathbf{eV})$ | $0.0 \mid 0.0$ | $0.0 \mid 0.0$ | $0.0 \mid 0.0$ | $6.91 \mid 0.18$ | $6.74 \mid 0.55$ | $5.58 \mid 0.47$ | | |
| $\mathbf{U}^p \mid \mathbf{J}^p_{\mathbf{z}} \; (\mathbf{eV})$ | $0.0 \mid 0.0$ | $0.0 \mid 0.0$ | $0.0 \mid 0.0$ | $11.64 \mid 1.55$ | $11.30 \mid 1.87$ | $11.37 \mid 1.82$ | | |

^a Refs. [65–73]

 AF_{II} MO to every MO of NiO via each of the aforementioned Hubbard functionals (that is, AF_{II} HPs applied to FM NiO via each functional, AF_{II} HPs to AF_{I} NiO, and so on).

1. Linear response calculations

We conduct ABINIT linear response calculations via what is now its intrinsic LRUJ algorithm [44] to determine the Hubbard parameters U and J_z for both the 3*d* Ni and 2*p* O orbitals (procedure described in Sec. II B). For each U (J_z) parameter, at least five perturbations between $\alpha = \pm 0.2 \text{ eV}$ ($\beta = \pm 0.1 \text{ eV}$) are applied to one of the 64 atoms in the supercells of each MO of NiO. Depending on how well these calculations converged, additional perturbations were applied.

As described at the end of Sec. II C, QUANTUM ESPRESSO augments the pseudo wavefunctions with core contributions without truncating at the PAW cutoff radius. Therefore, we further undertook QUANTUM ESPRESSO linear response calculations for the Hubbard U on Ni and O across all magnetic orderings.

For each U parameter calculated with QUANTUM ESPRESSO, five equally spaced perturbations between $\alpha = \pm 0.2 \text{ eV}$ are applied to one of the 64 atoms in the supercells of each MO of NiO. Runtime parameters remain the same as those for ABINIT except the following: The energy is relaxed self-consistently to within a threshold tolerance of 10^{-9} Ha, and a kinetic energy of 25 Ha for wavefunctions is administered alongside an energy cutoff of 250 Ha for the density. For all MOs, we use Monkhorst-Pack [130] grid k-point sampling of $2 \times 2 \times 2$.

III. RESULTS AND DISCUSSION

A. PAW projectors and the Hubbard parameters

Switching between PAW population analysis schemes induces noticeable numerical differences in the resulting Hubbard parameters for both Ni 3d orbitals and O 2p orbitals

(see Table III). It is encouraging that the unscreened response functions χ_0 and χ_{M_0} , pertaining to each projector choice, are reasonably alike. This is expected in accordance with the Corollary in the Appendix.

The Hubbard parameters U and J_z resulting from dmatpuopt = 2 and dmatpuopt = 3 are consistently similar, whereas dmatpuopt = 1 yields the largest in magnitude of these parameters and dmatpuopt = 4 yields the smallest. These phenomena are concordant with our understanding of the make-up of the response matrices. That is, extrapolations of occupancies beyond the PAW radius increase the overall occupancy (magnetization), thereby decreasing the inverse of the response matrices and reducing U (J_z).

For Ni, which has well-localized valence electrons, the range of Hubbard parameters is not very wide; by our calculations, 90.47% of the 3*d* atomic wavefunctions lie inside the PAW augmentation sphere. For O 2*p* orbitals, however, this overlap is significantly smaller, around 66.87%, and thus the range of achievable Hubbard parameters by various population analysis schemes is significantly larger for O than for Ni. It is important, therefore, in treating O 2*p* orbitals, to select an occupancy matrix calculation scheme that is normalized inside the PAW augmentation sphere.

In terms of numerical similarity with the QUAN-TUM ESPRESSO-derived U, only dmatpuopt = 2 and dmatpuopt = 3 remain viable candidates, with dmatpuopt = 2 barely outperforming dmatpuopt = 3 based on numerical nuances. Not only does dmatpuopt = 4 systematically underestimate U, from a theoretical standpoint (described in Sec. II C), using the integral of ϕ_0 to renormalize the AE basis function is an approximation that is not entirely justified. While the first AE partial wave listed in a PAW dataset for a particular treated subspace, which ABINIT selects as ϕ_0 , is required to be a normalized atomic eigenfunction, the second (or third or fourth, etc.) AE basis functions have no such constraints. There is no guarantee that these functions are localized in the PAW augmentation region to the same extent

^b Ref. [77]

TABLE III. PAW-based occupation determination methods [variations itemized by dmatpuopt integer for brevity: 1 refers to Eq. (14), 2 refers to Eq. (15), and 3 and 4 refer to Eq. (17)] compared with that of Quantum ESPRESSO (QE) in terms of the effect induced on Hubbard parameters, U and J_z , as well as their bare (χ_0) and screened (χ) response function components. Numbers gathered via linear response procedures implemented on both Ni and O across all magnetic orderings MO of the 64-atom NiO supercell.

| | | Ni Linear Response Parameters | | | | | | O Linear Response Parameters | | | | | |
|-----------------------------|--|-------------------------------|---------|----------------|-----------------|-------------|---------------------------|------------------------------|---------|----------------|-----------------|----------|---------|
| | | | U | | | Jz | | | U | | | Jz | |
| MO | $n^{t,\sigma}_{m,m'}$ | χ_0 | χ | \mathbf{U}^d | $\chi_{ m M_0}$ | $\chi_{_J}$ | $\mathbf{J}^d_\mathbf{z}$ | χ_0 | χ | \mathbf{U}^p | $\chi_{ m M_0}$ | χ_J | J_z^p |
| | _ 1 | -0.8064 | -0.1024 | 8.53 | -0.8054 | -0.9829 | 0.22 | -0.0856 | -0.0264 | 26.16 | -0.0856 | -0.1216 | 3.46 |
| | $\begin{bmatrix} 1 \\ 2 \end{bmatrix} 2$ | -0.9856 | -0.1359 | 6.34 | -0.9854 | -1.1746 | 0.16 | -0.1901 | -0.0595 | 11.55 | -0.1901 | -0.2697 | 1.55 |
| \mathbf{FM} | AB 3 | -0.9858 | -0.1262 | 6.91 | -0.9826 | -1.1892 | 0.18 | -0.1914 | -0.0593 | 11.64 | -0.1914 | -0.2719 | 1.55 |
| | 4 | -1.2010 | -0.1610 | 5.38 | -1.2010 | -1.3887 | 0.11 | -0.4281 | -0.1314 | 5.28 | -0.4280 | -0.6075 | 0.69 |
| | QE | -0.5008 | -0.1178 | 6.49 | | | | -0.1176 | -0.0530 | 10.36 | | | |
| | _ 1 | -0.9219 | -0.1064 | 8.31 | -0.9449 | -2.5779 | 0.67 | -0.0923 | -0.0277 | 25.27 | -0.0927 | -0.1545 | 4.31 |
| | | -1.1289 | -0.1347 | 6.54 | -1.1577 | -3.3259 | 0.56 | -0.2054 | -0.0601 | 11.52 | -0.2060 | -0.3363 | 1.88 |
| $\mathbf{AF_{I}}$ | $\begin{bmatrix} B \\ A \end{bmatrix}$ | -1.1259 | -0.1311 | 6.74 | -1.1547 | -3.2044 | 0.55 | -0.2068 | -0.0620 | 11.30 | -0.2073 | -0.3390 | 1.87 |
| | 4 | -1.3750 | -0.1583 | 5.59 | -1.4107 | -3.1137 | 0.39 | -0.4634 | -0.1417 | 4.90 | -0.4643 | -0.7652 | 0.85 |
| | QE | -0.6299 | -0.1244 | 6.45 | | | | -0.1144 | -0.0516 | 10.64 | | | |
| | _ 1 | -0.3104 | -0.0998 | 6.80 | -0.3104 | -0.3870 | 0.64 | -0.0803 | -0.0264 | 25.49 | -0.0799 | -0.1181 | 4.04 |
| | | -0.3791 | -0.1248 | 5.37 | -0.3790 | -0.4737 | 0.53 | -0.1784 | -0.0593 | 11.27 | -0.1779 | -0.2640 | 1.83 |
| $\mathbf{AF}_{\mathbf{II}}$ | $\mathbf{A}_{\mathbf{B}}$ | -0.3793 | -0.1218 | 5.58 | -0.3795 | -0.4621 | 0.47 | -0.1798 | -0.0590 | 11.37 | -0.1793 | -0.2661 | 1.82 |
| | 4 | -0.4644 | -0.1428 | 4.85 | -0.4634 | -0.5206 | 0.24 | -0.4024 | -0.1320 | 5.09 | -0.4017 | -0.5953 | 0.81 |
| | QE | -0.2607 | -0.0998 | 6.18 | | | | -0.1247 | -0.0537 | 10.59 | | | |

as ϕ_0 . By contrast, based on dmatpuopt = 1's systematic overestimation of U, it is clear that at least some normalization is necessary, especially for the oxygen 2p states.

For these reasons, we focus our attention on the subtle differences between dmatpuopt = 2 and 3. It is worth mentioning that, outside of convergence difficulties arising from the J_z calculations for the FM and AF_I MOs, dmatpuopt = 3 marginally outperforms dmatpuopt = 2 in terms of quality of linear response and ease of numerical convergence. The metrics used to evaluate this quality were (a) similarity of unscreened response functions χ_0 and χ_{M_0} ; (b) average number of SCF cycles needed to converge perturbative calculations; (c) average regression order (i.e., how often did the response functions demonstrate linearity over higher-order polynomial regressions); and (d) the error on the Hubbard parameter, calculated as follows:

$$\sigma_{\rm U} = \sqrt{\left(\frac{\sigma_{\chi_0}}{\chi_0^2}\right)^2 + \left(\frac{\sigma_{\chi}}{\chi^2}\right)^2} \tag{31}$$

where σ_{χ_0} and σ_{χ} are the unbiased root mean squared errors on the regression fit. The same error assessment is true for J_z. For dmatpuopt = 3, the average $\sigma_U(\sigma_{J_z})$ across all HPs is $\bar{\sigma}_U = 1.28\%$ ($\bar{\sigma}_{J_z} = 4.36\%$), which is slightly better than dmatpuopt = 2's $\bar{\sigma}_U = 1.41\%$ ($\bar{\sigma}_{J_z} = 5.20\%$). All in all, dmatpuopt = 3 outperforms its competitor in three of the four aforementioned metrics [(b), (c), and (d), to be precise].

There is a case to be made in opposition to dmatpuopt = 2 from a theoretical standpoint. Its projection operator,

Eq. (16), is defined using a Heaviside step function, which operationally "counts," so to speak, the overlap between two PAW radial basis functions by brute force. The ground-state eigenfunction basis for dmatpuopt = 3 is underpinned by the same assumptions and approximations that are used in the definition of the bound PAW AE partial waves. The renormalization of projections of all partial waves with a bounded, truncated AE wavefunction reintroduces a crucial, physics-based safeguard into the definition of the occupancy matrix. For this reason, we select dmatpuopt = 3 as our preferred PAW population analysis. We report the Hubbard parameters pertaining to this population analysis scheme as well as their associated errors in Table IV.

B. Intra-atomic spin moment

Having fossilized the *in situ* Hubbard parameters, we must give due consideration to the most appropriate Hubbard

TABLE IV. *In situ* Hubbard parameters calculated via linear response for FM, AF_I , and AF_{II} NiO and associated errors, calculated via Eq. (31), reported in eV.

| (eV) | \mathbf{FM} | $\mathbf{AF_{I}}$ | $\mathbf{AF}_{\mathbf{II}}$ |
|---------------------------|----------------|-------------------|-----------------------------|
| \mathbf{U}^d | 6.91 ± 0.13 | 6.74 ± 0.08 | 5.58 ± 0.09 |
| $\mathbf{J}_\mathbf{z}^d$ | 0.18 ± 0.02 | 0.55 ± 0.01 | 0.47 ± 0.02 |
| \mathbf{U}^p | 11.64 ± 0.06 | 11.30 ± 0.16 | 11.37 ± 0.12 |
| $\mathbf{J}_\mathbf{z}^p$ | 1.55 ± 0.03 | 1.87 ± 0.04 | 1.82 ± 0.05 |

TABLE V. Percent RMS errors (%RMSE). AS spin moments refer to those calculated in atom-centered augmentation spheres [Eq. (22)]; PAW refers to cumulative differences in PAW occupation matrix elements [Eq. (23)]; SD refers to use of the global spin density difference [Eq. (24)]. Cell hue represents error magnitude; the darker the blue, the smaller the error.

| | | \mathbf{S}_{z} | | S | | | |
|-------|------|------------------|-------|------|------|------|--|
| %RMSE | А | В | С | А | В | С | |
| AS | 50.2 | 105.1 | 89.8 | 50.2 | 29.7 | 29.4 | |
| PAW | 50.2 | 96.3 | 86.4 | 50.2 | 30.3 | 30.4 | |
| SD | 50.2 | 76.9 | 100.0 | 50.2 | 33.1 | 36.3 | |

functional (among the six tested using the AF_{II} HPs), intraatomic spin moment calculation (AS, PAW or SD, permutated across S_z and |**S**|), and MO-based spin parametrization method (Methods A, B, or C). Table I within the Supplemental Material(SM) [131] provides all of this raw data. To make the process of elimination simpler, however, we will focus on identifying the most effective combination of spin treatments by averaging across all six Hubbard functionals the percent RMS deviation (% RMSE) of our results for J_1 and J_2 from those of one particular experiment (exp), specifically

$$\% \text{ RMSE} = \sqrt{\frac{1}{2} \left(\left(\frac{J_1}{J_1^{\exp}} \right)^2 + \left(\frac{J_2}{J_2^{\exp}} \right)^2 \right)}.$$
 (32)

We select the inelastic neutron scattering experiment of Hutchings *et al.* [132] as our benchmark for this analysis. After averaging Eq. (32) across the six Hubbard functionals, Table I within the SM [131] reduces to Table V, where the darker shades of blue coincide with the degree to which error is minimized.

We notice immediately that the % RMS error is smaller when approximately treating intra-atomic spin as a quantum vector. Compared with its *z*-axis projection, $|\mathbf{S}|$ diminishes differences across population analysis schemes, such that $|\mathbf{S}^{AS}|$ is only subtly favored over other candidates. The preferable MO-based spin parametrization method depends crucially on the use of the quantum vector over the *z*-axis projection; under the latter treatment, Method A obtains exchange parameters more in line with experiment by a clear margin, indicating a preference for total energy differences with idealized spin-moment values. By contrast, if the magnetic moments are amplified by a factor of $\sqrt{2}$ as with the quantum vector approximation, Methods B and C, which weight the total energies from each MO according to various combinations of their atomic magnetic moments, win out.



FIG. 4. The effect of treating Ni 3*d* orbitals in the DFT+U^{*d*} functional with varying magnitude of the AF_{II} $U_{eff}^{d} = U^{d} - J_{z}^{d}$ (same HPs across all MOs). Plot (a) and (b) show average nickel and oxygen site magnetic moments μ_{z} , projected on the *z* axis and resolved via magnetic ordering (FM for ferromagnetic, AF_I for antiferromagnetic I, and AF_{II} for antiferromagnetic II), for each population analysis described in Sec. II D 2; AS = augmented sphere [Eq. (22)], PAW = PAW occupations [Eq. (23)], and SD = total spin density [Eq. (24)]. Plots (c) and (d) show, respectively, nearest-neighbor and next-nearest-neighbor Heisenberg exchange interaction parameters, calculated via different spin-parametrization methods, where intra-atomic spin is calculated as $|\mathbf{S}^{AS}| = \sqrt{2} S_{z}^{AS}$.

To understand this behavior, we recall that DFT-derived Ni magnetic moments on AF NiO do not typically exceed 2 μ_B [see Fig. 4(a)], meaning that the coefficients by which the AF total energies are weighted exceed unity, thereby amplifying the total energy differences to which the exchange interaction parameters are proportional. Most underlying Hubbard functionals, in and of themselves, generate total energy differences that are too large anyway, so the modifications provided by these magnetic moments are almost always counterproductive in obtaining results that more closely resemble experiment. The only exceptions to this trend occurred when we applied Hubbard functionals using parameters exceeding those calculated via linear response by several eV. Overestimation of the exchange parameters continues as long as DFT underestimates the magnetic moment of Ni on NiO.

According to Fig. 4(a), experimentally consistent AF moments are out of the reach of all population analysis techniques. The additional $\sqrt{2}$ factor accompanying $|\mathbf{S}|$ serves to approximately compensate where DFT falls short, lending itself to overcompensation when coupled with the population analysis techniques that typically predict higher Ni magnetic moments by default (like PAW and SD). For this reason, we see that a relatively underestimated population analysis, like that coming from AS, coupled with the overestimation attributed to the $\sqrt{2}$ factor from $|\mathbf{S}|$ can balance out to yield minimized errors.

This explanation, of course, is not a glowing testimonial for either technique, especially since S_z^{AS} performs rather poorly compared to its PAW and SD counterparts. However, the results in and of themselves, independent of their justification, offer concrete, actionable insights to the reader in terms of which spin treatment is best for NiO exchange. Certainly, Table V offers more organized navigation of a few available DFT-tractable, spin-based objects and their influence on the Heisenberg exchange constants, and the numerical forgiveness awarded by these techniques should not be overlooked. For the purposes of further analysis, therefore, we take Table V at its face value and both select and recommend $|S^{AS}| = \sqrt{2} S_z^{AS}$ as the preferred intra-atomic spin moment calculation method. Data pertaining to the S_z calculations across all population analysis methods is provided in the SM [132].

C. Comparison of spin parametrization methods for the Heisenberg model mapping

In Table VI, we resolve our $|S^{AS}|$ results to the level of the individual Hubbard functional and spin parametrization method. Principally, we report obtained values for J_1 and J_2 . Secondarily, for visual comparison, the final column shows the range of RMS errors obtained by comparing the best pair of coefficients from that functional to each of the four experiments listed at the bottom of the table. To clarify, for each functional, we take the best J_1 , J_2 pair of the three spin parametrization methods and calculate their percent RMS error with respect to each of the four experiments via Eq. (32). We then report the minimum (leftmost bar), range (stacked bar), and maximum (digit) of all those errors to compare across other functionals. The color of the bar corresponds to the spin parametrization method used to obtain that best (J_1, J_2) pair (i.e., tan for Method A, blue for Method B, and orange for Method C)

We prepare an identical error analysis scheme in Table VII, where we compare the exchange interaction parameters J_1 and J_2 reported in the literature [12,54,117–119,125,136–140], and experimental [131,133–135] to our own results, obtained from the various spin parametrization methods proposed in Sec. II D. For values reported in the literature, we reverse engineered the exchange interaction parameters for Methods B and C based on those reported for Method A (the conventional method) in addition to the reported values for the Ni magnetic moment for each configuration.

It should be clarified that we use RMS error in Tables V, VI, and VII as a measure of proximity to the reported values of the four different experiments cited. The RMS error is not asserting the proximity of our computed parameters to the true, physical values of these parameters (e.g., a % RMS error of 20% on $J_1 = 0.7$ meV does not mean we claim to know the value of J_1 to within ± 0.14 meV on the basis of calculations.

All Hubbard functionals we tested produced exchange coupling parameters that are highly competitive with those from other investigations. Generally, however, the literature reports a wide range of values for this particular magnetic property, in part due to large variation in calculation of the magnetic moments (approximately half use some form of the total spin density difference integrals, others use atom-centered sphere integration up to inconsistent maximum radii; all use S_z). Only a few computations use PAW, and many calculate the exchange coefficients with methods other than that involving total energy differences. It is crucial to recognize that the variation in spin moment integrations severely limits the reliability of any comparison across works of literature, and it is these incongruities in procedure that reinforce our inclination to benchmark with experiment.

Now for the analysis, referencing Tables V and VI, for converged total energies and their corresponding magnetic moments, we find that spin parametrization Methods B and C generally predict exchange interaction values smaller in magnitude, and closer to experiment, than the method that uses idealized values for Ni magnetization. This much is also demonstrated by the curves (the exchange parameters as functions of $AF_{II} U^d$ variation) in Figs. 4(c) and 4(d) as U^d approaches its *in situ* values of $U^d = 5.58 \text{ eV}$ and $U^d_{eff} = 5.11 \text{ eV}$.

Table VI shows that Method C routinely predicts exchange parameters smaller in magnitude than those from Methods B and A. This may be attributed to the presence of and a heavy reliance on a strong magnetic moment on FM NiO—that, or the AF_{II} moment is comparatively too weak. Whatever the explanation, the message is clear; if the spin moments are large enough, then it is well worth the effort to weight the total energies with them, as spin parametrization Methods B and C predict exchange parameters more closely in line with experiment. That much is also evident in the % RMSE in row 1 of Table V. The same AS data demonstrate a slight preference for Method B over Method C; however, the difference is minute.

The same conclusions cannot be drawn from Table VII. While J_2 is consistently smallest in magnitude under Method C, J_1 actually performs best under Method A. This is TABLE VI. Comparison of exchange coupling parameters from various Hubbard functionals, where $U_{eff} = U - J_z$. The "AF_{II} NiO Parameters" header refers to functionals tested using the Hubbard parameters arising from linear response on the ground-state AF_{II} magnetic ordering of NiO (i.e., column 4 in Table IV). "MO-Specific Parameters" refers to the use of the Hubbard parameters pertaining specifically to each magnetic ordering (i.e., FM HPs applied to FM ordering via Hubbard functional, AF_I parameters applied to AF_I ordering, etc.). Histogram shows minimum (leftmost bar), range (stacked bar) and maximum (digit) of all % RMS errors with respect to J_1 , J_2 pairs from each of the four experiments listed above. Color of bar corresponds to spin parametrization method used to obtain that best J_1 , J_2 pair (i.e., tan for Method A, blue for Method B, and orange for Method C).

| | Method A | | Method B | | Method C | | |
|---|---|---------------------|-------------------|-----------------------|-----------------------|--------------------|--------------------------|
| | $\mathbf{S}_{\mathbf{FM}} = \mathbf{S}_{\mathbf{AF_I}}$ | $= S_{AF_{II}} = 1$ | $S_{FM} = S_{AF}$ | $s_{I} = s_{AF_{II}}$ | $S_{FM} \neq S_{AFI}$ | $\neq S_{AF_{II}}$ | |
| | J_1 | J_2 | J_1 | J_2 | J_1 | | % RMS Error |
| Functional | (meV) | (meV) | (meV) | (meV) | (meV) | (meV) | <u>0 10 20 30 40 5</u> 0 |
| DFT (PBE) | 8.23 | -29.32 | 9.73 | -34.68 | 1.37 | -30.48 | 191.3 |
| AF _{II} NiO Parameters | | | | | | | |
| $DFT + U^d$ | 0.95 | -10.20 | 0.67 | -7.22 | 0.60 | -6.95 | 17.7 |
| $DFT + U^{d,p}$ | 1.53 | -7.11 | 1.07 | -5.00 | 1.00 | -4.88 | 47.1 |
| $\mathrm{DFT}+\mathrm{U}^{d}_{\mathrm{eff}}$ | 1.03 | -11.10 | 0.74 | -8.00 | 0.66 | -7.68 | 12.9 |
| $\mathrm{DFT} + \mathrm{U}_{\mathrm{eff}}^{d,p}$ | 1.52 | -8.41 | 1.09 | -6.04 | 1.00 | -5.86 | 42.1 |
| $DFT + U^d + J^d_z$ | 0.96 | -10.24 | 0.69 | -7.31 | 0.62 | -7.04 | 16.9 |
| $\mathrm{DFT} + \mathrm{U}^{d,p} + \mathrm{J}^{d,p}_{\mathrm{z}}$ | 1.00 | -7.35 | 0.71 | -5.22 | 0.66 | -5.09 | 32.4 |
| MO-Specific Parameters | | | | | | | 0 1000 2000 |
| $DFT + U^d$ | -2.60 | -43.53 | -1.84 | -30.79 | -1.72 | -29.27 | 299 |
| $DFT + U^{d,p}$ | -18.01 | -41.77 | -12.67 | -29.38 | -11.96 | -28.28 | 1306 |
| $\mathrm{DFT}+\mathrm{U}^{d}_{\mathrm{eff}}$ | -10.19 | -45.89 | -7.35 | -33.09 | -6.54 | -31.29 | 764 |
| $\mathrm{DFT} + \mathrm{U}^{d,p}_{\mathrm{eff}}$ | -40.63 | -46.36 | -29.15 | -33.26 | -26.75 | -31.79 | 2818 |
| $DFT + U^d + J^d_z$ | -19.24 | -44.87 | -13.73 | -32.01 | -12.39 | -30.40 | 1352 |
| $\mathrm{DFT} + \mathrm{U}^{d,p} + \mathrm{J}^{d,p}_{\mathrm{z}}$ | -49.96 | -45.16 | -35.48 | -32.07 | -32.83 | -30.81 | 3440 |
| Experiment | | | | | | | |
| INS Hutchings $(1972)[132]$ | 0.69 | -9.50 | | | | | |
| Thermo Shanker $(1973)[133]$ | -0.69 ^a | -8.66 | | | | | |
| Magnon Lockwood $(1992)[134]$ | 0.69 | -9.61 | | | | | |
| Magnon Dietz (1971)[135] | _ | -9.18 | | | | | |

^a Reference [133] reports a preference for antiferromagnetic alignment for both J_1 and J_2 .

consistent with our own data, recalling that the literature analyzed here exclusively calculates the magnetic moment as a projection on the z axis, whereas values from our own Hubbard functionals reported in Table VI use S^{AS} as per the conclusions of Sec. III B. If we better align our calculations with the literature by using S_z^{AS} , we can see from Table V that indeed, Method A minimizes the error compared to the other spin parametrization methods. This demonstrates the sensitivity of the interplay between the magnetic moments and total energies of each MO of NiO. Under our preferred spin treatment procedure, our results indicate that Methods B and C are, somewhat interchangeably, the most reliable in obtaining realistic exchange interaction parameters for NiO.

D. Effect of MO-specific Hubbard parameter calculation

Going back to Table VI, we see the uncorrected PBE XC functional predicts, at best, $J_1 = 1.37 \text{ meV}$ under Method C and $J_2 = -29.32 \text{ meV}$ under Method A, values which are far from corroborated by our experimental benchmarks of $J_1 = 0.69 \text{ meV}$ and $J_2 = -9.50 \text{ meV}$ [132]. The error arising from parametrization schemes B and C can be attributed in part to PBE's incorrect prediction of the Ni magnetic moments under

the AS population analysis ($\mu_{\text{FM}} = 1.09 \ \mu_B, \ \mu_{\text{AF}_{\text{I}}} = 1.56 \ \mu_B, \ \mu_{\text{AF}_{\text{I}}} = 1.30 \ \mu_B$).

PBE nonetheless correctly predicts the preference for ferromagnetic alignment in J_1 and antiferromagnetic alignment in J₂, unlike all functionals incorporating MO-specific parameters (center of Table VI). These performed poorly, not only quantitatively-note the inflated % RMS error scale, dilated by two orders of magnitude-but qualitatively, as they failed to predict the correct magnetic alignment for J_1 . This decisively answers the query posed earlier; while the U and J_z for a particular subspace in FM NiO differ from those calculated for other MOs (see Tables III and IV), the total energies resulting from the application of these magnetic ordering-specific parameters to their corresponding system are not necessarily comparable to one another. We must emphasize, however, that this will not necessarily be the case for any and all Hubbard functionals; there remain quite a few functionals untested in this article, and further testing is needed to determine if this ineptitude is an intrinsic property of Hubbard corrective protocol.

This conclusion is somewhat unexpected. The MO of a particular material system influences the magnitude of its atomic TABLE VII. Comparison of exchange interaction parameters from different XC functionals (or experiment) across literature works for each spin-parametrization method. Parameters for Methods B and C are reverse engineered from past works based on reported values for MO-dependent magnetization on Ni atoms and J_1 and J_2 values, translated according to Hamiltonian conventions outlined in Table I. Reported values from each work are in bold. Histogram shows minimum (leftmost bar), range (stacked bar) and maximum (digit) of all % RMS errors with respect to J_1 , J_2 pairs from each of the four experiments listed. Color of bar corresponds to spin parametrization method used to obtain that best J_1 , J_2 pair (i.e., tan for Method A, blue for Method B, and orange for Method C). Asterisk (*) indicates that these exchange parameters were calculated via a method other than that using total energy (i.e., Green's functions).

| | | | Method A | | Method B | | Method C | | |
|----------------------------------|---------------------|-------|-----------------------|------------------|-------------------|--------------------|------------------------|--------------------|----------------------------------|
| | | | $S_{FM} = S_{AF_{T}}$ | $= S_{AFII} = 1$ | $S_{FM} = S_{AI}$ | $S_{T} = S_{AFTT}$ | $S_{FM} \neq S_{AF_1}$ | $\neq S_{AF_{11}}$ | |
| | U | J_z | J_1 | J_2 | J_1 | J_2 | J_1 | J_2 | $\% \mathrm{RMS} \mathrm{Error}$ |
| XC [Ref] | (eV) | (eV) | (meV) | (meV) | (meV) | (meV) | (meV) | (meV) | 0 100 200 |
| PBE | | | 8.23 | -29.32 | 9.73 | -34.68 | 1.37 | -30.48 | 191.3 |
| $PBE+U^d$ | 5.58 | | 0.95 | -10.20 | 0.67 | -7.22 | 0.60 | -6.95 | 17.7 |
| $PBE+U_{eff}^d$ | 5.58 | 0.47 | 1.03 | -11.10 | 0.74 | -8.00 | 0.66 | -7.68 | 12.9 |
| $PBE+U^d+J^d_z$ | 5.58 | 0.47 | 0.96 | -10.24 | 0.69 | -7.31 | 0.62 | -7.04 | 16.9 |
| PBE+U [118] | 6.30 | 1.00 | 1.10 | -11.90 | 1.66 | -17.96 | 2.05 | -17.83 | 49.7 |
| PBE [117] | | | 0.60 | -22.25 | 1.32 | -48.95 | 2.32 | -44.02 | 111.3 |
| HSE [117] | | | 1.15 | -10.5 | 1.69 | -15.46 | 1.34 | -14.52 | 49.5 |
| PSIC [117] | | | 1.65 | -12.35 | 2.55 | -19.10 | 2.37 | -18.40 | 102.9 |
| $SIC-LSD^*$ [136] | | | 1.15 | -6.0 | 1.89 | -9.88 | 1.53 | -9.12 | 54.1 |
| GGA+U [119] | 6.30 | 1.00 | 0.87 | -9.54 | 1.23 | -13.55 | | | 19.8 |
| LDA [137] | | | 5.95 | -35.65 | 17.13 | -102.65 | | | 582.4 |
| UHF $[137]$ | | | 0.4 | -2.3 | 0.44 | -2.53 | | | 58.1 |
| Fock-35 [137] | | | 0.95 | -9.85 | 1.24 | -12.90 | | | 28.4 |
| Fock-50 [137] | | | 0.75 | -5.3 | 0.92 | -6.49 | | | 32.3 |
| B3LYP [137] | | | 1.2 | -13.35 | 1.73 | -19.19 | | | 64.8 |
| LSIC [138] | | | 1.42 | -6.95 | 2.02 | -9.87 | | | 77.3 |
| $LSIC^*$ [138] | | | 0.15 | -6.92 | 0.21 | -9.83 | | | 50.1 |
| $LSDA^*$ [12] | | | 0.54 | -21.50 | 0.64 | -25.18 | | | 105.9 |
| $LSDA+U^{*}$ [12] | 8.2 | 0.95 | -0.03 | -6.80 | -0.03 | -7.97 | | | 74.8 |
| LSDA+DMFT [*] [12] | | | -0.04 | -6.53 | -0.05 | -7.65 | | | 77.2 |
| LDA^{*} [125] | | | 0.3 | -28.3 | 0.82 | -77.50 | | | 165.3 |
| $PBE+U^{d,p}$ [54] | 7.63^{a} | | 2.36 | -25.15 | 4.25 | -45.42 | | | 217.8 |
| $\text{LDA}+\text{U}^{d,p}$ [54] | 7.49^{b} | | 1.48 | -20.04 | 4.05 | -54.86 | | | 123.2 |
| LDA+U [139] | 8.0 | 0.95 | 2.4 | -28.0 | | | | | 235.9 |
| $LDA+U^{*}$ [140] | 8.0 | 0.9 | 0.2 | -9.4 | | | | | 50.6 |
| | Exper | iment | - | | | | | | |
| | INS | [132] | 0.69 | -9.50 | | | | | |
| Т | hermo | [133] | -0.69 | -8.66 | | | | | |
| N | lagnon | [134] | 0.69 | -9.61 | | | | | |
| N | lagnon | [135] | _ | -9.18 | | | | | |

^a Applied $U^d = 7.63 \text{ eV}$ and $U^p = 3.00 \text{ eV}$.

^b Applied $U^d = 7.49$ eV and $U^p = 2.60$ eV.

magnetic moments, as is demonstrated in Figs. 4(a) and 4(b). These moments are direct functions of valence subspace occupations, and so changing the magnetic moments in turn alters the amount of self-interaction intrinsic to each subspace and, therefore, the Hubbard parameters and the efficacy of their host Hubbard functionals. If the total energies of each system are individually more accurate by application of a

Hubbard corrective functional, then is it not expected that the total energy differences between these systems become more accurate, as well?

The linearity condition of linear response is based heavily on the quadratic nature of SIE (i.e., the deviation from piece-wise linearity of the total energy with respect to fractional charge). It is worth noting, however, that the response behavior in the first-principles calculation of the Hubbard parameters for certain Ni 3d subspaces—notably upon application of the beta perturbation to the FM and AF_I MOs—noticeably deviated from linearity. This nonlinear response suggests that the SIE in these systems may require a correction of order higher than quadratic, possibly as a result of the added complexity in enforcing the system's nonground-state MOs.

We might be tempted to conclude then that applying different sets of Hubbard parameters to these systems is akin to changing its functional entirely, to the degree that the resulting total energies are no longer comparable (i.e., asserting that a DFT+ $U^d = 5.58 \text{ eV}$ functional is inherently different to a $DFT+U^d = 6.91 \text{ eV}$ functional). However, this justification is not compatible with our understanding of the Hubbard parameters as intrinsic, ground-state properties of the degree to which SIE and SCE infect the system. In maintaining the integrity of this understanding, then, it is more plausible that we have simply not found/tested a Hubbard-like functional adequately equipped to deal with long-range magnetic ordering of this nature. In applying the same Hubbard parameters inside this particular set of functionals to all structures, it is possible that the results benefit from a black-box cancellation of errors that manifests in the total energy difference domain. Whatever the case, it is clear that calculating and applying MO-ordering specific Hubbard parameters is a disadvantageous complexity for deriving the exchange coupling parameters for magnetic NiO, at least for the six Hubbard functionals tested here.

E. Effect of Hubbard functional choice

Whatever ails the MO-specific functionals does not affect the Hubbard functionals employing only the AF_{II} NiO parameters, in which the same Hubbard functionals are applied to all structures regardless of MO. Like PBE, these functionals correctly predict the preference for ferromagnetic alignment in J_1 and antiferromagnetic alignment in J_2 . This remains true as long as the Hubbard parameters are reasonable in magnitude, as demonstrated in Figs. 4(c) and 4(d); under the standard DFT+U^d functional, both the NN and the NNN exchange interactions gradually weaken with increasing U^d, coinciding with experimental values around the first-principles value of U^d.

In particular, the DFT+ U_{eff}^d yields J_1 and J_2 coefficients strikingly in line with experimental values under the Method B spin parametrization, contributing to shared RMS errors ranging from 7.4% to 12.9%. It is closely followed in performance by the same functional under Method C spin parametrization (8.6–14.5% error), and then by the Liechtenstein DFT+ $U^d + J_z^d$ functional—in which we apply both HPs to the Ni 3*d* subspace only—under Method B (11.0–16.9% error). This ranking remains stable even as the intra-atomic spin population analysis technique varies, although only when considering $|\mathbf{S}|$. When using instead S_z , the Liechtenstein DFT+ $U^d + J_z^d$ under Method A performs best.

The J_1 , J_2 pairs from these Hubbard functionals are highly competitive with those from other semilocal or hybrid functionals reported in the literature. Looking back at Table VII, the best DFT prediction for J_1 on NiO remains that from Fock-35 results of Ref. [137], with an individual parameter RMS error of 8.7%. For J_2 , the best predictions are held by the GGA + U_{eff} = 5.3 eV calculation of Ref. [119] (J_2 RMS errors ranging from 0.4% to 10.2%) and the LDA + U^d_{eff} = 7.1 eV conducted in Ref. [140] (J_2 errors 1.1–8.6%). In terms of pair-wise performance, however, the *in situ* Hubbard functionals monopolize the best error minimization rankings, reflecting positively on the error compensation and spin parametrization techniques. At minimum, this demonstrates that the Hubbard functionals are competitive with the more costly hybrid functionals in achieving total energy difference results that reflect experiment.

F. Mechanisms of Hubbard functional choice

It is possible to rationalize the dependence of J_1 and J_2 on the choice of Hubbard functional and parameter strength in a relatively simple way. To see this, we start with the energy expressions for Method B, Eqs. (29) and (30). Taking the former, J_1 , we first note that a sufficient U^d will change the FM and AF_I systems from metallic to insulating. This will tend to reduce the relevant total energy difference; the total energy of the metallic state depends significantly, through the kinetic energy, on the Fermi densities of the two systems, whereas in the insulating systems, it will primarily depend on weaker differences of exchange and correlation effects. As we can see from Fig. 4(a), the local moments become more similar in the two states at larger U values, which moderates the tendency for the different Hubbard energy contributions to increase. As these local moments increase with U^d , furthermore, so too the denominator in Eq. (29) increases quadratically, which further brings down the J_1 . Thus, we can visualize how increasing U^d tends to bring down the predicted J_1 . Of course, that effect is moderated in the Dudarev scheme when U^d is reduced to U^d_{eff} by an increasing J_{τ}^{d} .

Meanwhile, it is interesting that the exchange parameters arising from DFT+U^d and the Liechtenstein DFT+U^d + J^d_z are so similar. In fact, the similarity between the exchange parameters arising from either functional is tenacious, remaining even as we vary U^d and J^d_z (keeping the first-principles ratio between U^d and J^d_z constant). The relative agnosticism of the interatomic, Heisenberg exchange parameters to the addition of a nonzero J_z parameter suggests that the spin localization facilitated by the U^d term is already quite robust. The addition of U^p_{eff} largely mitigates the effect of U^d_{eff} on J₁, and not in a beneficial way when using the first-principles parameters. We return to discuss this effect further in the next section.

As for the substantial moderation of J_2 by the addition of U^d , the same arguments and trends largely hold. The total energy differences in Eq. (30) are reduced when all three systems are insulating. Once again, as the moments are increased with increasing U_{eff}^d , so the J_2 decreases due to the quadratic expression in the denominator. This effect is weakened by J_z^d in the Dudarev functional, by definition, and beneficially so when J_2 is compared against experiment. The effects of J_z^d and J_z^p are much more weak in the Liechtenstein functional, where it is also more difficult to interpret. The main difference in the trends observed between J_1 and J_2

is a different sign in the change with the addition of finite U_{eff}^p , so that increasing U_{eff}^p reduces the magnitude of J_2 , as we go on to discuss further. Once again, however, the addition of the first-principles U_{eff}^p is detrimental to the quality of J_2 .

G. Effect of Hubbard parameters on oxygen

Introduction of HPs on the oxygen 2p orbitals, across all Hubbard functionals, seems to have the effect of strengthening the NN ferromagnetic exchange while simultaneously weakening the antiferromagnetic coupling between layers, similar to how HSE and PSIC perform compared to raw PBE [117]. For example, relative to DFT+U^d + J^d_z, the additional U^p and J^p_z in DFT+U^{d,p} + J^{d,p}_z increase J₁ to 1 meV, but bring J₂ down in magnitude to -7.35 meV. These are typically unwelcome modifications under Methods B and C, for both J₁ and J₂, since the latter is chronically underestimated by a large margin and the former hovers tightly around the experimental value of 0.69 eV.

Nonetheless, the regularity of the phenomenon is intriguing and worth some discussion. In the much simplified picture of Eq. (26), the spin dependence of the AF_{II} total energy is informed exclusively by J_2 interactions, and the FM total energy is informed heavily, although not exclusively, by J_1 interactions. Moreover, as discussed in Sec. II D 1, the crystal field splitting of orbitals is heavily implicated in the direct and superexchange mechanisms that collectively give rise to interatomic exchange. We can perhaps look for clues in the FM and AF_{II} NiO density of states (DOS) plots, shown in Fig. 5 as functions of U^{*p*}, keeping U^{*d*} = 5.58 eV constant.

In FM NiO, the occupied minority spin states on oxygen and the unoccupied minority spin states on Ni are shifted higher in energy with increasing U^p , although the former at a faster rate than the latter. By itself, this means we should expect the exchange energy, and thus J_1 , to gradually reduce in magnitude. However, simultaneous to this feature, we notice two pivotal details about the shifting occupations: (1) the t_{2g} states are decreasing in occupation, so the Coulombic repulsion that contributes to a strong AF exchange interaction is weakening; and (2) the oxygen 2p orbitals are increasing in occupation, as demonstrated by the gradually inflating area underneath the pDOS curve as well as the inset graphs in Fig. 5. Thus, the intra-atomic Coulomb repulsion grows, thereby strengthening the FM preference of the superexchange mechanism. Because intra-atomic exchange overwhelms its interatomic counterpart by several orders of magnitude, we expect these local oxygen dynamics to eclipse the gradual reduction in exchange energy coming from the evolution of the minority spin pDOS. Collectively, these justifications point to an overall increasing FM preference for J_1 .

For the AF_{II} NiO case, the Ni and O subspace occupancies both become more integer-like upon application of HPs on O in addition to those already applied on Ni—more integer-like and, thus, more localized spatially. As the orbitals cling more closely to their respective nuclei, there is less overlap between neighboring atomic orbitals. In the NNN case, less orbital overlap translates to a decrease in virtual hopping and therefore a weaker exchange bridge; by consequence, J_2 decreases in magnitude. Yet, unlike in 180° superexchange, the 90° superexchange is mediated by the Coulombic exchange between orthogonal orbitals on the same oxygen atom (see Fig. 3). As orthogonal orbitals on the same atom become more localized, the exchange increases in magnitude, thereby intensifying the ferromagnetic preference embodied by J_1 . So, J_1 increases in magnitude and J_2 decreases in magnitude, precisely as observed. In this way, it is possible that the application of the oxygen HPs on top of those applied on Ni serves to partially relocalize the atomic orbitals.

In Fig. 5, it is interesting that the AF_{II} pDOS character varies almost imperceptibly with changes in oxygen's U^{*p*}, which changes only E_F in a manner rigidly linear with the magnitude of the Hubbard parameter. We mention this to preemptively assuage concerns that it belies a plotting error and tentatively propose that the phenomenon is related to the fact that the oxygen atoms in AF_{II} NiO have absolutely no net magnetic moment. When we set U^{*p*} to a large, finite number, the DOS undergoes significant changes.

We emphasize here that the pDOS shown in Fig. 5, in addition to SM Figs. 2 and 3, are truncated within the PAW augmentation spheres, with no interstitial contribution, based on the PAW population analysis. It therefore exhibits sharper, more atomic-like features than would be observed experimentally. An increased smearing would be needed for comparison to experiment. Furthermore, only the conduction band edge should be interpreted, and features no higher in energy thereof, due to insufficient basis resolution. Our principle objective here is to understand the mechanisms of interatomic exchange interactions.

Despite these limitations, our AF_{II} NiO pDOS does not deviate significantly in its features from those of prior DFT+U calculations [141–146]. The upper band is clearly of primarily Ni e_{ρ} character, whereas the lower band edge is characterized by the O 2p, followed by non-negligible contributions of Ni t_{2g} and e_g character. The failure of DFT+U-like methods to replicate other features of the experimental pDOS is well documented across the literature. For example, our oxygen 2p spectrum exhibits a two-peaked structure (with edges at-5.1 eV and 0.6 eV) instead of the broad peak feature attributed to oxygen from angle-resolved photoemission spectra [147,148]. This result is also at odds with the results of DFT + DMFT calculations [13,14]. The width of the valence band Ni 3d spectrum is also narrower overall than might be expected, as well as appearing sharper for the aforementioned reasons. However, the same experiments also indicate that the NiO valence-band spectrum has two distinct peaks, one around-2 eV and the other around-5 eV, a behavior reflected in our own plots, where the peak-to-peak splitting across the Fermi energy is approximately 4.6 eV, a value comparable to that derived from experiment (4.3 eV) [149,150] and DMFT [13,14]. We refer the reader to the SM [132] for an analogous figure to Fig. 3, where instead the U^d is varied (Fig. 2 within the SM [132]), as well as for an indicative plot of the total DOS (Fig. 3 within the SM [132]) with no PAW sphere truncation, for the best performing functional that used $U_{eff}^d = 5.11 \text{ eV}$. Figure 3 within the SM [132], in particular, displays more clearly the low-energy peak at-5.1 eV of primarily Ni e_g character, which has the



FIG. 5. Overlaid atomic (AE) orbital projected density of states (pDOS) on two particular atoms of the four-atom unit cell, resolved by crystal field splitting characterization on Ni (e_g blue, top row; t_{2g} light green, middle row) and O (2*p* orbital orange, bottom row) for each magnetic ordering (FM left column; AF_I middle column; AF_{II} right column) of NiO. Positive values indicate majority spin DOS, negative values indicate minority spin. Only the component within the PAW augmentation sphere cutoff r_c for each species is included, with no interstitial contributions, hence the relatively sharp (atomic-like) features of the pDOS. Moreover, only the conduction band edge position should be interpreted here, as the Kohn-Sham basis is not thought to be complete for the conduction band for all MOs in the energy range shown. Insets show corresponding relative changes in majority spin ($\sigma = \uparrow$) and minority spin ($\sigma = \downarrow$) occupancies $\Delta n^{\sigma} = 100(n^{\sigma}|_{U^{p}} - n^{\sigma}|_{U^{p=0.45}})/n^{\sigma}|_{U^{p=0.45}}$ as a function of U^{*p*} [eV] for that particular MO and crystal field characterization. Opacity of plot is a function of U^{*p*} (eV) in the Dudarev *et al*. DFT+U^{*d*,*p*} functional, where U^{*d*} = 5.58 eV is kept constant and applied to all MOs. Variations in the AF_{II} case are present but barely discernible.

distinct sharpness as the spectral feature attributed to Ni in experiment.

IV. SUMMARY AND CONCLUSIONS

In this paper, we analyzed the NN and NNN Heisenberg exchange coupling coefficients of the prototypical Mott-Hubbard insulator NiO, calculating its electronic structure in its three main enforceable magnetic configurations (FM, AF_I , and AF_{II}) in a Hubbard corrected DFT context, for comparison with experiment. First, we examined the sensitivity of the first-principles-derived Hubbard parameters U and J_z to the PAW projectors. Here, we found that dmatpuopt = 3, the population analysis corresponding to the oncenormalized projection of the PAW basis functions on bound AE atomic eigenfunctions and truncated at the PAW augmentation radius, renders a numerically well-behaved linear response procedure that yields robust Hubbard parameters. These go on to provide experimentally relevant J_1 and J_2 exchange pairs once applied to NiO via a range of host Hubbard functionals.

Our results highlight just how valuable to the simulation of NiO and its total energetic properties are its *in situ*-derived U and J_z . We report rigorously defined PBE Hubbard parameters

U and J_z for the Ni 3*d* and O 2*p* subspaces of NiO. The technical platform on which we build these investigations, the open-source plane-wave DFT suite ABINIT [42,43], allowed us to at once assess the numerical performance of the code's linear-response implementation for calculating the Hubbard U and to extend its utility to also calculate the intraatomic exchange coupling corrective parameter J_z from first principles.

We continued to probe the minutiae of population analysis procedures to reevaluate spin-moment parametrization conventions, both inter- and intra-atomic, in the Heisenberg model. In doing so, we systematically charted a particular region of DFT-accessible spin-moment parameter space to make navigation of these techniques more manageable. Our optimization of this parameter space corresponds to the particular combination of (i) a rudimentary-quantum, atomic-sphere-restricted intra-atomic spin moment ($|S^{AS}| =$ $\sqrt{2}$ S^{AS}, and (ii) a spin-parametrization scheme that assumes $|S_{FM}| = |S_{AFT}| = |S_{AFT}|$. We recommend this particular spin treatment for magnetic properties and total energy differences pertaining to NiO, given that the resulting J_1, J_2 exchange pairs give % RMS errors (with respect to experiment) as low as 13%, a considerable improvement on state-of-the-art DFT simulations conducted thus far. Regardless of the Hubbard functional used, this spin treatment produced exchange coupling constants that are satisfactorily in line with experiment and highly competitive with other, more computationally taxing hybrid functionals, which were methodically categorized in a comprehensive literature review.

Throughout these investigations are implicit suggestions towards best practices in calculating Hubbard parameters and use of Hubbard functionals for nonground-state magnetic orderings of TMOs. Notably, we highlight the fact that the Hubbard functionals tested here, despite the bespoke nature of these parameters to their respective magnetic environments, did not result in comparable total energies, so we generally disadvise their use for such comparisons in NiO. Alternatively, these results highlight the need for more advanced Hubbard functionals designed to accommodate differences in magnetic environment. Notwithstanding, it is clear from Table VI that use of the J_z parameter on both the Ni and the O is critical for reducing % RMSE of the Heisenberg exchange coupling parameters in NiO with standard functionals.

Lastly, through extensive testing of a suite of 6+ different corrective Hubbard functionals, we draw attention to the effect of the O 2p Hubbard parameter pairs on the NiO exchange coupling coefficients and provide a justification for the phenomenon thereabouts based on the crystal-field resolved DOS and occupation analyses.

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APPENDIX: EQUIVALENCY OF UNSCREENED RESPONSE FUNCTIONS FOR α AND β PERTURBATIONS

It is important to note, for verification purposes, that the unscreened response matrices χ_0 and χ_{M_0} are equivalent. The proof of this is as follows.

Suppose we apply, to separate spin channels, an identical potential perturbation $dV_{\text{ext}}^{\uparrow} = dV_{\text{ext}}^{\downarrow} = dV_{\text{ext}} = d\alpha$ to a subspace on an atom. The unscreened change in spin-up occupancy n_0^{\uparrow} as a result of this perturbation is

$$\frac{dn_0^{\uparrow}}{dV_{\text{ext}}} = \frac{dn_0^{\uparrow}}{dV_{\text{ext}}^{\uparrow}} \frac{dV_{\text{ext}}^{\uparrow}}{dV_{\text{ext}}} + \frac{dn_0^{\uparrow}}{dV_{\text{ext}}^{\downarrow}} \frac{dV_{\text{ext}}^{\downarrow}}{dV_{\text{ext}}}.$$
 (A1)

The second term of Eq. (A1) disappears because the occupancy of a spin channel does not change as a result of a potential perturbation on the opposing spin channel (i.e., $dn_0^{\uparrow}/dV_{\text{ext}}^{\downarrow} = 0$). Then $dV_{\text{ext}}^{\uparrow}/dV_{\text{ext}} = 1$, leaving

$$\frac{dn_0^{\uparrow}}{dV_{\text{ext}}} = \frac{dn_0^{\uparrow}}{dV_{\text{ext}}^{\uparrow}} = \frac{dn_0^{\uparrow}}{d\alpha}.$$
 (A2)

Similarly, via the same procedure, we find that

$$\frac{dn_0^{\downarrow}}{dV_{\text{ext}}} = \frac{dn_0^{\downarrow}}{d\alpha} \tag{A3}$$

and thus the unscreened change in total occupancy of the subspace, where $N_0 = n_0^{\uparrow} + n_0^{\downarrow}$, is

$$\chi_0 = \frac{dN_0}{dV_{\text{ext}}} = \frac{d(n_0^{\uparrow} + n_0^{\downarrow})}{d\alpha}.$$
 (A4)

Now consider that on a completely identical subspace on an identical atom, we apply a potential perturbation of $dV_{\text{ext}}^{\uparrow} = \beta$ to the spin-up channel and $dV_{\text{ext}}^{\downarrow} = -\beta$ to the spindown channel. Here, we observe the unscreened change in magnetization $\mu_0 = n_0^{\uparrow} - n_0^{\downarrow}$ as a result of this perturbation, which is

$$\chi_{M_0} = \frac{d\mu_0}{dV_{ext}} = \frac{d(n_0^{\uparrow} - n_0^{\downarrow})}{d\beta}$$
$$= \frac{dn_0^{\uparrow}}{d\beta} - \frac{dn_0^{\downarrow}}{d\beta}.$$
(A5)

The change in spin-up occupancy with respect to β is found to be identical to Eq. (A1). But because a $-\beta$ perturbation is applied to the spin-down channel potential,

$$\frac{dn_0^{\downarrow}}{dV_{\text{ext}}} = -\frac{dn_0^{\downarrow}}{d\beta}.$$
 (A6)

- A. Chakraborty, M. Dixit, D. Aurbach, and D. T. Major, Predicting accurate cathode properties of layered oxide materials using the SCAN meta-GGA density functional, npj Comput. Mater. 4, 60 (2018).
- [2] K. Saritas, E. R. Fadel, B. Kozinsky, and J. C. Grossman, Charge density and redox potential of LiNiO₂ using *ab initio* diffusion quantum Monte Carlo, J. Phys. Chem. C **124**, 5893 (2020).
- [3] J. Shu, T.-F. Yi, M. Shui, Y. Wang, R.-S. Zhu, X.-F. Chu, F. Huang, D. Xu, and L. Hou, Comparison of electronic property and structural stability of LiMn₂O₄ and LiNi_{0.5}Mn_{1.5}O₄ as cathode materials for lithium-ion batteries, Comput. Mater. Sci. 50, 776 (2010).
- [4] Y. S. Meng and M. E. Arroyo-de Dompablo, First principles computational materials design for energy storage materials in lithium ion batteries, Energy Environ. Sci. 2, 589 (2009).
- [5] M. Shishkin and H. Sato, Redox, magnetic, and structural properties of α -NaMnO₂ cathode material analyzed by fitting-free DFT+U calculations, parameterized by the linear response approach, J. Phys. Chem. C **125**, 1531 (2021).
- [6] F. S. Hegner, J. R. Galán-Mascarós, and N. López, A database of the structural and electronic properties of prussian blue, prussian white, and berlin green compounds through density functional theory, Inorg. Chem. 55, 12851 (2016).
- [7] A. Aziz, Electronic structure simulations of energy materials: chalcogenides for thermoelectrics and metal-organic frameworks for photocatalysis, Ph.D. thesis, University of Reading, 2017.
- [8] S. Siebentritt and S. Schorr, Kesterites—A challenging material for solar cells, Progr. Photovol. 20, 512 (2012).
- [9] A. Polman, M. Knight, E. C. Garnett, B. Ehrler, and W. C. Sinke, Photovoltaic materials: Present efficiencies and future challenges, Science 352, aad4424 (2016).
- [10] C. Yan, K. Sun, J. Huang, S. Johnston, F. Liu, B. P. Veettil, K. Sun, A. Pu, F. Zhou, J. A. Stride, M. A. Green, and X. Hao, Beyond 11% efficient sulfide kesterite Cu₂Zn_xCd_{1-x}SnS₄ solar cell: Effects of cadmium alloying, ACS Energy Lett. 2, 930 (2017).
- [11] L. Zhang, P. Staar, A. Kozhevnikov, Y.-P. Wang, J. Trinastic, T. Schulthess, and H.-P. Cheng, DFT + DMFT calculations of the complex band and tunneling behavior for the transition metal monoxides MnO, FeO, CoO, and NiO, Phys. Rev. B 100, 035104 (2019).
- [12] Y. O. Kvashnin, O. Granas, I. Di Marco, M. I. Katsnelson, A. I. Lichtenstein, and O. Eriksson, Exchange parameters of strongly correlated materials: Extraction from spin-polarized

Thus,

$$\chi_{M_0} = \frac{d\mu_0}{dV_{ext}} = \frac{dn_0^{\uparrow}}{d\beta} - \left(-\frac{dn_0^{\downarrow}}{d\beta}\right)$$

$$\therefore \chi_{M_0} = \frac{d(n_0^{\uparrow} + n_0^{\downarrow})}{d\beta}.$$
 (A7)

In the event that $\alpha = \beta$, therefore, $\chi_0 = \chi_{M_0}$, a conclusion following Eqs. (A4) and (A7).

density functional theory plus dynamical mean-field theory, Phys. Rev. B **91**, 125133 (2015).

- [13] S. Mandal, K. Haule, K. M. Rabe, and D. Vanderbilt, Systematic beyond-DFT study of binary transition metal oxides, npj Comput. Mater. 5, 115 (2019).
- [14] J. Kuneš, V. I. Anisimov, S. L. Skornyakov, A. V. Lukoyanov, and D. Vollhardt, NiO: Correlated band structure of a chargetransfer insulator, Phys. Rev. Lett. 99, 156404 (2007).
- [15] S. Tanaka, Variational quantum monte carlo approach to the electronic structure of NiO, J. Phys. Soc. Jpn. 64, 4270 (1995).
- [16] S. Zhang, F. D. Malone, and M. A. Morales, Auxiliary-field quantum Monte Carlo calculations of the structural properties of nickel oxide, J. Chem. Phys. 149, 164102 (2018).
- [17] C. Mitra, J. T. Krogel, J. A. O. Santana, and F. A. Reboredo, Many-body *ab initio* diffusion quantum Monte Carlo applied to the strongly correlated oxide NiO, J. Chem. Phys. 143, 164710 (2015).
- [18] K. Saritas, J. T. Krogel, P. R. C. Kent, and F. A. Reboredo, Diffusion Monte Carlo: A pathway towards an accurate theoretical description of manganese oxides, Phys. Rev. Mater. 2, 085801 (2018).
- [19] Y. Gao, Q. Sun, J. M. Yu, M. Motta, J. McClain, A. F. White, A. J. Minnich, and Garnet Kin-Lic Chan, Electronic structure of bulk manganese oxide and nickel oxide from coupled cluster theory, Phys. Rev. B 101, 165138 (2020).
- [20] H. Takaki, S. Inoue, and Y. Matsumura, Colorationdecoloration properties and mechanisms of nickel oxide films, Chem. Phys. Lett. 774, 138624 (2021).
- [21] D. Hait, N. M. Tubman, D. S. Levine, K. B. Whaley, and M. Head-Gordon, What levels of coupled cluster theory are appropriate for transition metal systems? A study using nearexact quantum chemical values for 3d transition metal binary compounds, J. Chem. Theory Comput. 15, 5370 (2019).
- [22] A. Schron, C. Rodl, and F. Bechstedt, Energetic stability and magnetic properties of MnO in the rocksalt, wurtzite, and zincblende structures: Influence of exchange and correlation, Phys. Rev. B 82, 165109 (2010).
- [23] M. Cococcioni and N. Marzari, Energetics and cathode voltages of LiMPO₄ olivines (M = Fe, Mn) from extended Hubbard functionals, Phys. Rev. Mater. **3**, 033801 (2019).
- [24] W. Zhang, D. G. Truhlar, and M. Tang, Explanation of the source of very large errors in many exchange-correlation functionals for vanadium dimer, J. Chem. Theory Comput. 10, 2399 (2014).
- [25] A. J. Cohen, P. Mori-Sánchez, and W. Yang, Insights into current limitations of density functional theory, Science 321, 792 (2008).

- [26] P. Mori-Sánchez, A. J. Cohen, and W. Yang, Many-electron self-interaction error in approximate density functionals, J. Chem. Phys. 125, 201102 (2006).
- [27] E. B. Isaacs, S. Patel, and C. Wolverton, Prediction of Li intercalation voltages in rechargeable battery cathode materials: Effects of exchange-correlation functional, van der Waals interactions, and Hubbard U, Phys. Rev. Mater. 4, 065405 (2020).
- [28] A. Georges, L. d. Medici, and J. Mravlje, Strong electronic correlations from Hund's coupling, Annu. Rev. Condens. Matter Phys. 4, 137 (2013).
- [29] D. D. O'Regan, M. C. Payne, and A. A. Mostofi, Subspace representations in *ab initio* methods for strongly correlated systems, Phys. Rev. B 83, 245124 (2011).
- [30] P. Mori-Sánchez, A. J. Cohen, and W. Yang, Localization and delocalization errors in density functional theory and implications for band-gap prediction, Phys. Rev. Lett. **100**, 146401 (2008).
- [31] A. J. Cohen, P. Mori-Sánchez, and W. Yang, Development of exchange-correlation functionals with minimal many-electron self-interaction error, J. Chem. Phys. 126, 191109 (2007).
- [32] A. Ruzsinszky, J. P. Perdew, G. I. Csonka, O. A. Vydrov, and G. E. Scuseria, Density functionals that are one- and two- are not always many-electron self-interaction-free, as shown for H₂+, He₂+, LiH+, and Ne₂+, J. Chem. Phys. **126**, 104102 (2007).
- [33] J. P. Perdew and A. Zunger, Self-interaction correction to density-functional approximations for many-electron systems, Phys. Rev. B 23, 5048 (1981).
- [34] O. K. Orhan and D. D. O'Regan, First-principles Hubbard U and Hund's J corrected approximate density-functional theory predicts an accurate fundamental gap in rutile and anatase TiO₂, Phys. Rev. B **101**, 245137 (2020).
- [35] D. S. Lambert and D. D. O'Regan, Use of DFT + U + J with linear response parameters to predict non-magnetic oxide band gaps with hybrid-functional accuracy, Phys. Rev. Res. 5, 013160 (2023).
- [36] A. C. Burgess, E. Linscott, and D. D. O'Regan, DFT + U-type functional derived to explicitly address the flat plane condition, Phys. Rev. B 107, L121115 (2023).
- [37] M. Cococcioni and S. de Gironcoli, A linear response approach to the calculation of the effective interaction parameters in the LDA+U method, Phys. Rev. B 71, 035105 (2005).
- [38] J. W. Bennett, B. G. Hudson, I. K. Metz, D. Liang, S. Spurgeon, Q. Cui, and S. E. Mason, A systematic determination of Hubbard U using the GBRV ultrasoft pseudopotential set, Comput. Mater. Sci. **170**, 109137 (2019).
- [39] B. Kim, K. Kim, and S. Kim, Quantification of coulomb interactions in layered lithium and sodium battery cathode materials, Phys. Rev. Mater. 5, 035404 (2021).
- [40] K. Yu and E. A. Carter, Communication: Comparing *ab initio* methods of obtaining effective U parameters for closed-shell materials, J. Chem. Phys. **140**, 121105 (2014).
- [41] P. E. Blochl, Projector augmented-wave method, Phys. Rev. B 50, 17953 (1994).
- [42] X. Gonze, B. Amadon, P. M. Anglade, J. M. Beuken, F. Bottin, P. Boulanger, F. Bruneval, D. Caliste, R. Caracas, M. Côté *et al.*, ABINIT: First-principles approach to material and nanosystem properties, Comput. Phys. Commun. **180**, 2582 (2009).

- [43] B. Amadon, F. Jollet, and M. Torrent, γ and β cerium: LDA + U calculations of ground-state parameters, Phys. Rev. B 77, 155104 (2008).
- [44] L. MacEnulty and D. J. Adams, Hubbard U and Hund's J parameters with Cococcioni and de Gironcoli's approach: Determine the Hund's J for Ni 3d in NiO with LRUJ (2023), ABINIT, Accessed: 2023-09-22.
- [45] V. I. Anisimov, J. Zaanen, and O. K. Andersen, Band theory and Mott insulators: Hubbard U instead of Stoner I, Phys. Rev. B 44, 943 (1991).
- [46] V. I. Anisimov and O. Gunnarsson, Density-functional calculation of effective coulomb interactions in metals, Phys. Rev. B 43, 7570 (1991).
- [47] V. I. Anisimov, I. V. Solovyev, M. A. Korotin, M. T. Czyżyk, and G. A. Sawatzky, Density-functional theory and nio photoemission spectra, Phys. Rev. B 48, 16929 (1993).
- [48] Y.-C. Wang, B.-L. Liu, Y. Liu, H.-F. Liu, Y. Bi, X.-Y. Gao, J. Sheng, H.-Z. Song, M.-F. Tian, and H.-F. Song, The underestimation of high pressure in DFT + U simulation for the wide range cold-pressure of lanthanide metals, arXiv:2111.12863.
- [49] M. Yu, S. Yang, C. Wu, and N. Marom, Machine learning the Hubbard U parameter in DFT+U using Bayesian optimization, npj Computational Mathematics 6, 180 (2020).
- [50] B. Dorado, B. Amadon, M. Freyss, and M. Bertolus, DFT + U calculations of the ground state and metastable states of uranium dioxide, Phys. Rev. B 79, 235125 (2009).
- [51] B. Dorado, G. Jomard, M. Freyss, and M. Bertolus, Stability of oxygen point defects in UO_2 by first-principles DFT + *U* calculations: Occupation matrix control and Jahn-Teller distortion, Phys. Rev. B **82**, 035114 (2010).
- [52] D. A. Tompsett, D. S. Middlemiss, and M. S. Islam, Importance of anisotropic coulomb interactions and exchange to the band gap and antiferromagnetism of β -MnO₂ from DFT + U, Phys. Rev. B **86**, 205126 (2012).
- [53] C. E. Patrick and K. S. Thygesen, Hubbard-U corrected Hamiltonians for non-self-consistent random-phase approximation total-energy calculations: A study of ZnS, TiO₂, and NiO, Phys. Rev. B **93**, 035133 (2016).
- [54] P. Gopal, R. D. Gennaro, M. S. dos Santos Gusmao, R. A. R. A. Orabi, H. Wang, S. Curtarolo, M. Fornari, and M. B. Nardelli, Improved electronic structure and magnetic exchange interactions in transition metal oxides, J. Phys.: Condens. Matter 29, 444003 (2017).
- [55] O. Le Bacq, A. Pasturel, C. Lacroix, and M. D. Nunez-Regueiro, First-principles determination of exchange interactions in delafossite YCuO_{2.5}, Phys. Rev. B **71**, 014432 (2005).
- [56] K. L. Materna, A. M. Beiler, A. Thapper, S. Ott, H. Tian, and L. Hammarström, Understanding the performance of NiO photocathodes with alkyl-derivatized cobalt catalysts and a push-pull dye, ACS Appl. Mater. Interfaces 12, 31372 (2020).
- [57] R. A. Wahyuono, M. Braumüller, S. Bold, S. Amthor, D. Nauroozi, J. Plentz, M. Wächtler, S. Rau, and B. Dietzek, Localizing the initial excitation—A case study on nio photocathodes using ruthenium dipyridophenazine complexes as sensitizers, Spectrochimica Acta A 252, 119507 (2021).
- [58] C.-Y. Hsu, W.-T. Chen, Y.-C. Chen, H.-Y. Wei, Y.-S. Yen, K.-C. Huang, K.-C. Ho, C.-W. Chu, and J. T. Lin, Charge transporting enhancement of NiO photocathodes for ptype dye-sensitized solar cells, Electrochim. Acta 66, 210 (2012).

- [59] X. Xia, L. Fu, Z. Luo, J. Zhu, W. Yang, D. Li, and L. Zhou, Discharge behavior of nio as thermal battery cathode at ultrahigh temperature, Electrochem. Comm. 134, 107185 (2022).
- [60] A. Manthiram, A reflection on lithium-ion battery cathode chemistry, Nat. Commun. **11**, 1550 (2020).
- [61] M. D. Irwin, D. B. Buchholz, A. W. Hains, R. P. H. Chang, and T. J. Marks, p-type semiconducting nickel oxide as an efficiency-enhancing anode interfacial layer in polymer bulkheterojunction solar cells, Proc. Natl. Acad. Sci. USA 105, 2783 (2008).
- [62] H. Yang, Q. Tao, X. Zhang, A. Tang, and J. Ouyang, Solidstate synthesis and electrochemical property of SnO₂/NiO nanomaterials, J. Alloys Compd. 459, 98 (2008).
- [63] M. Shi, T. Qiu, B. Tang, G. Zhang, R. Yao, W. Xu, J. Chen, X. Fu, H. Ning, and J. Peng, Temperature-controlled crystal size of wide band gap nickel oxide and its application in electrochromism, Micromachines 12, 80 (2021).
- [64] J. Hugel and C. Carabatos, Band structure and optical properties of NiO. I. Band structure calculations, J. Phys. C 16, 6713 (1983).
- [65] A. P. Cracknell and S. J. Joshua, The space group corepresentations of antiferromagnetic NiO, Math. Proc. Cambridge Philos. Soc. 66, 493 (1969).
- [66] W. L. Roth, Multispin axis structures for antiferromagnets, Phys. Rev. 111, 772 (1958).
- [67] W. L. Roth, Magnetic structures of MnO, FeO, CoO, and NiO, Phys. Rev. 110, 1333 (1958).
- [68] T. Eto, S. Endo, M. Imai, Y. Katayama, and T. Kikegawa, Crystal structure of NiO under high pressure, Phys. Rev. B 61, 14984 (2000).
- [69] A. M. Balagurov, I. A. Bobrikov, J. Grabis, D. Jakovlevs, A. Kuzmin, M. Maiorov, and N. Mironova-Ulmane, Neutron scattering study of structural and magnetic size effects in NiO, IOP Conf. Ser.: Mater. Sci. Eng. 49, 012021 (2013).
- [70] A. M. Balagurov, I. A. Bobrikov, S. V. Sumnikov, V. Y. Yushankhai, J. Grabis, A. Kuzmin, N. Mironova-Ulmane, and I. Sildos, Neutron diffraction study of microstructural and magnetic effects in fine particle NiO powders, Phys. Status Solidi (b) 253, 1529 (2016).
- [71] N. Smith, The structure of thin films of metallic oxides and hydrates, J. Am. Chem. Soc. 58, 173 (1936).
- [72] M. A. Peck and M. A. Langell, Comparison of nanoscaled and bulk NiO structural and environmental characteristics by XRD, XAFS, and XPS, Chem. Mater. 24, 4483 (2012).
- [73] A. K. Cheetham and D. A. O. Hope, Magnetic ordering and exchange effects in the antiferromagnetic solid solutions Mn_xNi₁-x₀, Phys. Rev. B 27, 6964 (1983).
- [74] V. Fernandez, C. Vettier, F. de Bergevin, C. Giles, and W. Neubeck, Observation of orbital moment in NiO, Phys. Rev. B 57, 7870 (1998).
- [75] W. Neubeck, C. Vettier, V. Fernandez, F. de Bergevin, and C. Giles, Observation of orbital moment in NiO using magnetic x-ray scattering, J. Appl. Phys. 85, 4847 (1999).
- [76] E. Brok, K. Lefmann, P. P. Deen, B. Lebech, H. Jacobsen, G. J. Nilsen, L. Keller, and C. Frandsen, Polarized neutron powder diffraction studies of antiferromagnetic order in bulk and nanoparticle NiO, Phys. Rev. B 91, 014431 (2015).
- [77] Y. Shimomura, M. Kojima, and S. Saito, Crystal structure of ferromagnetic nickel oxide, J. Phys. Soc. Jpn. 11, 1136 (1956).

- [78] J. Hubbard and B. H. Flowers, Electron correlations in narrow energy bands, Proc. R. Soc. London A 276, 238 (1963).
- [79] S. L. Dudarev, G. A. Botton, S. Y. Savrasov, C. J. Humphreys, and A. P. Sutton, Electron-energy-loss spectra and the structural stability of nickel oxide: An LSDA+U study, Phys. Rev. B 57, 1505 (1998).
- [80] B. Himmetoglu, A. Floris, S. de Gironcoli, and M. Cococcioni, Hubbard-corrected dft energy functionals: The LDA+U description of correlated systems, Int. J. Quantum Chem. 114, 14 (2014).
- [81] D. D. O'Regan, N. D. M. Hine, M. C. Payne, and A. A. Mostofi, Linear-scaling DFT+U with full local orbital optimization, Phys. Rev. B 85, 085107 (2012).
- [82] V. I. Anisimov, F. Aryasetiawan, and A. I. Lichtenstein, Firstprinciples calculations of the electronic structure and spectra of strongly correlated systems: The LDA+U method, J. Phys.: Condens. Matter 9, 767 (1997).
- [83] H. J. Kulik, Perspective: Treating electron over-delocalization with the DFT+U method, J. Chem. Phys. 142, 240901 (2015).
- [84] A. I. Liechtenstein, V. I. Anisimov, and J. Zaanen, Densityfunctional theory and strong interactions: Orbital ordering in Mott-Hubbard insulators, Phys. Rev. B 52, R5467(R) (1995).
- [85] J. P. Perdew, K. Burke, and M. Ernzerhof, Generalized gradient approximation made simple, Phys. Rev. Lett. 77, 3865 (1996).
- [86] H. J. Kulik, M. Cococcioni, D. A. Scherlis, and N. Marzari, Density functional theory in transition-metal chemistry: A self-consistent Hubbard U approach, Phys. Rev. Lett. 97, 103001 (2006).
- [87] W. E. Pickett, S. C. Erwin, and E. C. Ethridge, Reformulation of the LDA + U method for a local-orbital basis, Phys. Rev. B **58**, 1201 (1998).
- [88] G. Moynihan, G. Teobaldi, and D. D. O'Regan, A selfconsistent ground-state formulation of the first-principles Hubbard U parameter validated on one-electron selfinteraction error, arXiv:1704.08076.
- [89] E. B. Linscott, D. J. Cole, M. C. Payne, and D. D. O'Regan, The role of spin in the calculation of Hubbard *U* and Hund's *J* parameters from first principles, Phys. Rev. B 98, 235157 (2018).
- [90] F. Aryasetiawan, M. Imada, A. Georges, G. Kotliar, S. Biermann, and A. I. Lichtenstein, Frequency-dependent local interactions and low-energy effective models from electronic structure calculations, Phys. Rev. B 70, 195104 (2004).
- [91] F. Aryasetiawan, K. Karlsson, O. Jepsen, and U. Schonberger, Calculations of Hubbard *U* from first-principles, Phys. Rev. B 74, 125106 (2006).
- [92] I. Timrov, N. Marzari, and M. Cococcioni, Hubbard parameters from density-functional perturbation theory, Phys. Rev. B 98, 085127 (2018).
- [93] M. Cococcioni, A LDA+U study of selected iron compounds, Ph.D. thesis, Scuola Internazionale Superiore di Studi Avanzati, 2002.
- [94] Q. Wu and T. Van Voorhis, Constrained density functional theory and its application in long-range electron transfer, J. Chem. Theory Comput. 2, 765 (2006).
- [95] D. D. O'Regan and G. Teobaldi, Optimization of constrained density functional theory, Phys. Rev. B 94, 035159 (2016).

- [96] P. H. Dederichs, S. Blügel, R. Zeller, and H. Akai, Ground states of constrained systems: Application to cerium impurities, Phys. Rev. Lett. 53, 2512 (1984).
- [97] Abinit makes available to users a variety of mixing schemes to aid convergence of systems. That which we use herein is a dielectric density mixing scheme, which is governed by input variables diemix and diemixmag (briefly, defined as the percentage of the total density and/or magnetic density pertaining to the current SCF iteration to be mixed with those of prior iterations. Based on Abinit's program architecture, the density of the first SCF iteration is mixed before the unscreened occupations/magnetizations are calculated and harvested, meaning that the printed unscreened occupations/magnetizations are actually fractions of those that responded to the potential perturbation. For this reason, we must divide the calculated values of χ_0 and χ_{M_0} by the density mixing parameters (i.e., diemix = 0.15 for our α perturbations and diemixmag = 0.1 for our β perturbations. Furthermore, this stipulation requires we be careful in representing our linear response data and regressions. Therefore, what is plotted is the data points and polynomial fit corresponding to $[X(\omega) + (\theta -$ 1)X(0)]/ θ , where $X = \chi_0, \chi_{M_0}, \omega = \alpha, \beta$ and $\theta = \text{diemix} =$ 0.15, diemixmag = 0.1 are Abinit-specific convergence parameters for U, J_H calculations.
- [98] B. Himmetoglu, R. M. Wentzcovitch, and M. Cococcioni, First-principles study of electronic and structural properties of CuO, Phys. Rev. B 84, 115108 (2011).
- [99] M. Bernardi, Physical origin of the one-quarter exact exchange in density functional theory, J. Phys.: Condens. Matter 32, 385501 (2020).
- [100] G. Moynihan, A self-contained ground-state approach for the correction of self-interaction error in approximate densityfunctional theory, Ph.D. thesis, Trinity College Dublin, 2018.
- [101] G. Kresse and D. Joubert, From ultrasoft pseudopotentials to the projector augmented-wave method, Phys. Rev. B 59, 1758 (1999).
- [102] N. Holzwarth, Updated comments on projector augmented wave (PAW) implementations within various electronic structure code packages, Comput. Phys. Commun. 243, 25 (2019).
- [103] N. Holzwarth, A. Tackett, and G. Matthews, A Projector Augmented Wave (PAW) code for electronic structure calculations, Part I: atompaw for generating atom-centered functions, Comput. Phys. Commun. 135, 329 (2001).
- [104] M. Torrent, N. Holzwarth, F. Jollet, D. Harris, N. Lepley, and X. Xu, Electronic structure packages: Two implementations of the projector augmented wave (PAW) formalism, Comput. Phys. Commun. 181, 1862 (2010).
- [105] G. Geneste, B. Amadon, M. Torrent, and G. Dezanneau, DFT+U study of self-trapping, trapping, and mobility of oxygen-type hole polarons in barium stannate, Phys. Rev. B 96, 134123 (2017).
- [106] O. Bengone, M. Alouani, P. Blochl, and J. Hugel, Implementation of the projector augmented-wave LDA+U method: Application to the electronic structure of NiO, Phys. Rev. B 62, 16392 (2000).
- [107] Abinit, PAW input variables: dmatpuopt (2021), https://docs. abinit.org/variables/paw/#dmatpuopt.
- [108] PAW datasets generated via the *atompaw* code [103] will always feature a normalized, bound atomic eigenfunction as the first atomic wavefunction of a PAW dataset. Step 4 on

Page 2 of Ref. [109] states clearly that *atompaw* mandates the use of "atomic eigenfunctions related to valence electrons (bound states)" as the partial waves included in the PAW basis. Therefore, all PAW datasets generated by *atompaw*, including the JTH sets on PseudoDojo,[110] have atomic eigenfunctions as the first atomic wavefunctions of the correlated subspace.

Their normalization, however, depends on the pseudo partialwave generation scheme. *atompaw* provides two options for this scheme: the Vanderbilt or the Blöchl. We are able to reasonably infer that the JTH table of PAW datasets [110], available on the PseudoDojo website, match all criteria as a suitable dataset with which one may use dmatpuopt=1.

- [109] N. A. W. Holzwarth, Notes for revised form of *atompaw* code (2010), http://users.wfu.edu/natalie/papers/pwpaw/notes/ atompaw/atompawEqns.pdf.
- [110] F. Jollet, M. Torrent, and N. Holzwarth, Generation of projector augmented-wave atomic data: A 71 element validated table in the XML format, Comput. Phys. Commun. 185, 1246 (2014).
- [111] Y.-C. Wang, Z.-H. Chen, and H. Jiang, The local projection in the density functional theory plus *U* approach: A critical assessment, J. Chem. Phys. **144**, 144106 (2016).
- [112] I. Timrov, N. Marzari, and M. Cococcioni, Self-consistent Hubbard parameters from density-functional perturbation theory in the ultrasoft and projector-augmented wave formulations, Phys. Rev. B 103, 045141 (2021).
- [113] P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo *et al.*, Quantum espresso: a modular and open-source software project for quantum simulations of materials, J. Phys.: Condens. Matter **21**, 395502 (2009).
- [114] P. Giannozzi, O. Andreussi, T. Brumme, O. Bunau, M. B. Nardelli, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, M. Cococcioni *et al.*, Advanced capabilities for materials modelling with quantum espresso, J. Phys.: Condens. Matter 29, 465901 (2017).
- [115] X. He, N. Helbig, M. J. Verstraete, and E. Bousquet, TB2J: A python package for computing magnetic interaction parameters, Comput. Phys. Commun. 264, 107938 (2021).
- [116] R. Skomski, *Simple Models of Magnetism* (Oxford University Press, Oxford, 2008).
- [117] T. Archer, C. D. Pemmaraju, S. Sanvito, C. Franchini, J. He, A. Filippetti, P. Delugas, D. Puggioni, V. Fiorentini, R. Tiwari, and P. Majumdar, Exchange interactions and magnetic phases of transition metal oxides: Benchmarking advanced *ab initio* methods, Phys. Rev. B 84, 115114 (2011).
- [118] R. Logemann, A. N. Rudenko, M. I. Katsnelson, and A. Kirilyuk, Exchange interactions in transition metal oxides: The role of oxygen spin polarization, J. Phys.: Condens. Matter 29, 335801 (2017).
- [119] W.-B. Zhang, Y.-L. Hu, K.-L. Han, and B.-Y. Tang, Pressure dependence of exchange interactions in NiO, Phys. Rev. B 74, 054421 (2006).
- [120] P. Garcia-Fernandez, J. C. Wojdel, J. Iniguez, and J. Junquera, Second-principles method including electron and lattice degrees of freedom, Phys. Rev. B 93, 195137 (2016).
- [121] V. R. Eisberg and R. R. J. Wiley, *Quantum Physics of Atoms, Molecules, Solids, Nuclei and Particles* (John Wiley & Sons, Hoboken, NJ, 1976), pp. 127–128.

- [122] J. Coey and J. Coey, *Magnetism and Magnetic Materials*, Magnetism and Magnetic Materials (Cambridge University Press, Cambridge, 2010).
- [123] X. Li, H. Yu, L. Feng, F. J.S., M.-H. Whangbo, and H. Xiang, Spin Hamiltonians in magnets: Theories and computations, Molecules 26, 803 (2021).
- [124] W. A. Harrison, Heisenberg exchange in the magnetic monoxides, Phys. Rev. B 76, 054417 (2007).
- [125] T. Kotani and M. van Schilfgaarde, Spin wave dispersion based on the quasiparticle self-consistent *GW* method: NiO, MnO and α -MnAs, J. Phys.: Condens. Matter **20**, 295214 (2008).
- [126] X. Wan, Q. Yin, and S. Y. Savrasov, Calculations of magnetic exchange interactions in mott-hubbard systems, Phys. Rev. Lett. 97, 266403 (2006).
- [127] Equations (27) and (28) are true only in atomic units, where $\mu_B = 1/2$ and the numerical value of these units is absorbed into the other constants of the equations. Thus, total energies expressed in eV must each be multiplied by a factor of four for Eqs. (27) and (28) to hold true.
- [128] N. Marzari, *Ab initio* molecular dynamics for metallic systems, Ph. D. thesis, University of Cambridge, 1996.
- [129] M. Rutter, C2x: A tool for visualisation and input preparation for Castep and other electronic structure codes, Comput. Phys. Commun. 225, 174 (2018).
- [130] H. J. Monkhorst and J. D. Pack, Special points for Brillouinzone integrations, Phys. Rev. B 13, 5188 (1976).
- [131] M. T. Hutchings and E. J. Samuelsen, Measurement of spinwave dispersion in NiO by inelastic neutron scattering and its relation to magnetic properties, Phys. Rev. B 6, 3447 (1972).
- [132] See Supplemental Material at http://link.aps.org/ supplemental/10.1103/PhysRevB.108.245137 for (1) an expansion of Table II containing J_1 and J_2 pairs corresponding to the specific Hubbard functional used to calculate it; (2) plot showing the evolution of the pDOS with respect to increasing values of U^d in DFT+U^d; and (3) total pDOS over all atoms in the unit cell.
- [133] R. Shanker and R. A. Singh, Analysis of the exchange parameters and magnetic properties of NiO, Phys. Rev. B 7, 5000 (1973).
- [134] D. J. Lockwood, M. G. Cottam, and J. H. Baskey, Oneand two-magnon excitations in NiO, J. Magn. Magn. Mater. 104-107, 1053 (1992).
- [135] R. E. Dietz, W. F. Brinkman, A. E. Meixner, and H. J. Guggenheim, Raman scattering by four magnons in NiO and KNiF₃, Phys. Rev. Lett. **27**, 814 (1971).
- [136] D. Kodderitzsch, W. Hergert, W. M. Temmerman, Z. Szotek, A. Ernst, and H. Winter, Exchange interactions in NiO and at the NiO(100) surface, Phys. Rev. B 66, 064434 (2002).

- [137] I. d. P. R. Moreira, F. Illas, and R. L. Martin, Effect of Fock exchange on the electronic structure and magnetic coupling in NiO, Phys. Rev. B 65, 155102 (2002).
- [138] G. Fischer, M. Dane, A. Ernst, P. Bruno, M. Lueders, Z. Szotek, W. Temmerman, and W. Hergert, Exchange coupling in transition metal monoxides: Electronic structure calculations, Phys. Rev. B 80, 014408 (2009).
- [139] A. Jacobsson, B. Sanyal, M. Lezaic, and S. Blugel, Exchange parameters and adiabatic magnon energies from spin-spiral calculations, Phys. Rev. B 88, 134427 (2013).
- [140] D. M. Korotin, V. V. Mazurenko, V. I. Anisimov, and S. V. Streltsov, Calculation of exchange constants of the Heisenberg model in plane-wave-based methods using the Green's function approach, Phys. Rev. B 91, 224405 (2015).
- [141] G. Trimarchi, Z. Wang, and A. Zunger, Polymorphous band structure model of gapping in the antiferromagnetic and paramagnetic phases of the Mott insulators MnO, FeO, CoO, and NiO, Phys. Rev. B 97, 035107 (2018).
- [142] M. J. Han, T. Ozaki, and J. Yu, O(n)LDA + U electronic structure calculation method based on the nonorthogonal pseudoatomic orbital basis, Phys. Rev. B 73, 045110 (2006).
- [143] I. Naz, F. Ahmad, J. Jang, and J. Y. Rhee, Ab-initio investigation of electronic structures of α -BiFeO3 with different exchange-correlation functionals, AIP Adv. **8**, 125216 (2018).
- [144] K. O. Egbo, C. P. Liu, C. E. Ekuma, and K. M. Yu, Vacancy defects induced changes in the electronic and optical properties of NiO studied by spectroscopic ellipsometry and first-principles calculations, J. Appl. Phys. **128**, 135705 (2020).
- [145] J. A. Dawson, Y. Guo, and J. Robertson, Energetics of intrinsic defects in NiO and the consequences for its resistive random access memory performance, Appl. Phys. Lett. 107, 122110 (2015).
- [146] W. B. Zhang, N. Yu, W. Y. Yu, and B. Y. Tang, Stability and magnetism of vacancy in NiO: A GGA+U study, Eur. Phys. J. B 64, 153 (2008).
- [147] Z.-X. Shen, C. K. Shih, O. Jepsen, W. E. Spicer, I. Lindau, and J. W. Allen, Aspects of the correlation effects, antiferromagnetic order, and translational symmetry of the electronic structure of NiO and CoO, Phys. Rev. Lett. 64, 2442 (1990).
- [148] Z.-X. Shen, R. S. List, D. S. Dessau, B. O. Wells, O. Jepsen, A. J. Arko, R. Barttlet, C. K. Shih, F. Parmigiani, J. C. Huang, and P. A. P. Lindberg, Electronic structure of NiO: Correlation and band effects, Phys. Rev. B 44, 3604 (1991).
- [149] G. A. Sawatzky and J. W. Allen, Magnitude and origin of the band gap in NiO, Phys. Rev. Lett. 53, 2339 (1984).
- [150] O. Tjernberg, S. Söderholm, G. Chiaia, R. Girard, U. O. Karlsson, H. Nylén, and I. Lindau, Influence of magnetic ordering on the NiO valence band, Phys. Rev. B 54, 10245 (1996).