

Fate of multipolar physics in $5d^2$ double perovskites

Ahmed Rayyan¹,^{*} Xiaoyu Liu,² and Hae-Young Kee^{1,3,*}

¹*Department of Physics, University of Toronto, Toronto, Ontario, Canada M5S 1A7*

²*Department of Materials Science and Engineering, University of Washington, Seattle, Washington 98195, USA*

³*CIFAR Program in Quantum Materials, Canadian Institute for Advanced Research, Toronto, Ontario, Canada M5G 1M1*



(Received 18 May 2023; accepted 12 July 2023; published 31 July 2023)

In a cubic environment, the ground state of spin-orbit coupled $5d^2$ ions is a non-Kramers E_g doublet, which hosts quadrupole and octupole moments. A series of $5d^2$ osmium double perovskites Ba_2MOsO_6 ($M = Mg, Ca, Zn, Cd$) have recently been proposed to exhibit multipolar orders. We investigate the structural properties of these materials using *ab initio* calculations and find that the cubic structure is unstable for the Cd compound while the Mg, Ca, and Zn materials retain $Fm\bar{3}m$ symmetry. We show that Ba_2CdOsO_6 favors a rhombohedral $R\bar{3}$ structure characterized by $a^-a^-a^-$ octahedral tiltings as indicated by unstable T_{1g} phonon modes. Trigonal distortions split the excited T_{2g} triplet into an E'_g doublet and an A_g singlet, which may cross energy levels with the E_g doublet and suppress the multipolar physics. We find a window where E_g remains the lowest energy state under trigonal distortion, enabling the emergence of multipole phases in noncubic crystal environments.

DOI: [10.1103/PhysRevB.108.045149](https://doi.org/10.1103/PhysRevB.108.045149)

I. INTRODUCTION

Strongly correlated materials feature many distinct phases often classified by different order parameters associated with their broken symmetries. A textbook example is the magnetic order described by various arrangements of magnetic dipole moments. The magnetic order parameter can be probed using several experimental techniques such as neutron scattering, and its onset would accompany thermodynamic phase transitions. Under certain conditions, the dipole moment is absent yet higher-rank moments may transition into an ordered state. However, due to their multipolar nature, they can be “hidden” from some experimental probes [1]. A physical framework encompassing these cases motivates studying the ordering mechanisms of states with nontrivial multipolar moments.

The f -electron systems have been a natural platform to explore multipolar physics, as the rare-earth ions carry total angular momentum J via strong spin-orbit coupling (SOC) [2,3]. Examples include the $4f^2$ Pr materials where the Pr^{3+} ions carry a $J = 4$ multiplet [4–9]. The ninefold degeneracy is lifted by octahedral or tetrahedral crystal electric fields (CEFs) and yields a doublet where the Kramers degeneracy does not apply due to the even number of electrons. The resulting non-Kramers doublet lacks a dipole moment yet carries quadrupole and octupole moments. A Landau theory for Pr^{3+} 1-2-20 materials has been developed in recent years [10–13] and it suggests that the hidden octupole moment can be revealed within magnetoelastic experiments by applying a [111] magnetic field [14].

The situation in d -electron systems is more subtle. In the $3d$ Mott insulators, the light magnetic ions carry a much weaker SOC so that the orbital angular momentum is often

quenched by CEFs. In the case where the orbital degrees of freedom may yet fluctuate, such as for one hole or one electron in e_g states (e.g., $3d^9$ or low-spin $3d^7$, respectively), then orbital ordering is usually found via the Kugel-Khomskii mechanism [15]. The resulting orbital ordering is equivalent to a motif of charge quadrupoles and can be accompanied by a structural transition via the cooperative Jahn-Teller effect [16]. The ordering of higher multipoles, including octupolar moments, is difficult to achieve in the lighter transition metals.

However, higher-rank multipoles may be relevant in the heavier transition metal compounds such as the $5d^1$ and $5d^2$ double perovskites (DPs) $A_2BB'O_6$ [17]. The combination of strong SOC, large separation between magnetic $B'O_6$ octahedra, and high cubic symmetry satisfies the necessary preconditions for local multipolar physics [17–23]. A promising platform for octupolar ordering lies within the $5d^2$ DPs, where the magnetic B' ion features a $J = 2$ SOC multiplet that is split into a low-lying non-Kramers doublet (E_g) and an excited triplet (T_{2g}) via orbital t_{2g} - e_g mixing [24,25]. Similar to the $4f^2$ case, the non-Kramers E_g doublet carries multipolar moments with interactions obtained by projecting d orbital hopping channels onto the E_g doublet to form an effective pseudospin-1/2 model. These include compass quadrupole and Ising octupole interactions [26] where the strength (and sign) of these terms depends on the details of each material under consideration [27].

The $5d^2$ barium osmate DPs with Ba_2MOsO_6 and $M \in \{Mg, Ca, Zn, Cd\}$ have been proposed as a series of compounds which features multipolar orderings [28–31]. These materials exhibit at most a single transition at temperature T^* that does not coincide with a reduction of cubic symmetry. Various candidates for the ordering observed in these materials include antiferro-quadrupolar [26], ferri-octupolar [32], and ferro-octupolar [27,33–35]. The octupolar order hypothesis is favored for the Mg, Ca, and Zn compounds,

*hy.kee@utoronto.ca

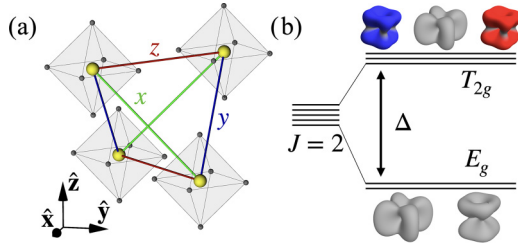


FIG. 1. (a) Os⁶⁺ ions (yellow) enclosed in oxygen octahedral cages (gray) arranged in the fcc structure. The crystallographic *xyz* coordinate system is shown along with fcc nearest-neighbor bonds of type *x* (green), *y* (blue), and *z* (red). The *MO*₆ octahedra and BaO₁₂ cuboctahedra are not shown. (b) The single-ion level scheme for 5*d*² electrons in an octahedral CEF, where the low-lying *E_g* doublet is separated from the excited *T_{2g}* triplet by a residual CEF splitting Δ ∼ 10 meV. The corresponding *E_g* and *T_{2g}* states are shown, with regions of red and blue denoting nonzero spin density. [Reproduced from Fig. 1(b) in Ref. [36].]

as the ordering at $T^* \sim 30\text{--}50$ K coincides with the onset of muon-spin relaxation (μ SR) oscillations, signaling a loss of time-reversal symmetry [29,30]. On the other hand, the Cd compound does not feature thermodynamic anomalies or μ SR oscillations down to $T^* = 0.47$ K [30], and its low-temperature structural properties are yet to be determined by the high-resolution synchrotron x-ray diffraction as in Ref. [31]. Thus, a set of questions arise naturally which we aim to resolve in this work: Is the cubic structure of the 5*d*² Os DPs Ba₂MOsO₆ stable at low temperatures, and if not, what is the fate of the multipolar physics in the low-symmetry environment?

The paper is organized as follows. In Sec. II we briefly review how the *E_g* doublet arises out of the *J* = 2 states and discuss challenges in identifying the crystal structure in some 5*d* DPs. In Sec. III we investigate the phonon spectrum of each compound using *ab initio* density functional theory (DFT) simulations, and find that the Cd compound favors a rhombohedral structure characterized by a set of octahedral tiltings. In Sec. IV we evaluate how the *E_g* and *T_{2g}* states are modified in the rhombohedral structure where each OsO₆ octahedron is trigonally compressed along the [111] direction. We conclude with a summary of our findings and their implications, as well as avenues of future work.

II. *E_g* AND *T_{2g}* STATES IN 5*d*² OSMIUM DOUBLE PEROVSKITES

The 5*d*² Os DPs Ba₂MOsO₆ form a rocksalt pattern of alternating *MO*₆ and OsO₆ corner-sharing octahedra that enclose Ba²⁺ ions. The Os⁶⁺ ions carry the relevant magnetic degrees of freedom and form an fcc lattice [see Fig. 1(a)]. The octahedral CEF lifts the electronic *d* orbital degeneracy into irreps of the octahedral group *O_h* as $\Gamma_{l=2} = t_{2g} \oplus e_g$, with the low-lying *t_{2g}* triplet separated from the excited *e_g* doublet by energy gap 10*Dq*. The two *t_{2g}* electrons form a high-spin configuration with *S* = *L* = 1 via Hund's rules.

The strong SOC $-\lambda|\mathbf{L} \cdot \mathbf{S}$ of the 5*d* ions with $|\lambda| < 10Dq$ favors a total *J* = 2 multiplet with magnetic moment $\mathbf{M} = -\mathbf{L} + 2\mathbf{S} = \mathbf{J}/2$ with magnitude $\sim 1.25\mu_B$ [17]. However,

heat capacity, μ SR, and inelastic neutron scattering experiments find a low-lying doublet separated from excitations by a gap of Δ ∼ 10 meV with a small magnetic moment of $\mathcal{O}(0.1\mu_B)$, in apparent contradiction with the *J* = 2 picture [30,31]. Some mechanisms for a residual CEF that generates a low-lying *E_g* doublet and an excited *T_{2g}* triplet as shown in Fig. 1(b) have been proposed to lift the *J* = 2 degeneracy, including *t_{2g}*-*e_g* mixing via SOC and Hund's coupling, or nonspherical Coulomb interactions in a *t_{2g}*-only model [24]. Defining $|m\rangle \equiv |J = 2; J^z = m\rangle$ where $m \in \{2, 1, 0, -1, -2\}$, the *E_g* and *T_{2g}* states in the cubic harmonic basis are linear combinations of the states shown in Fig. 1(b), and are given by

$$\begin{aligned} |\uparrow\rangle &= \frac{1}{\sqrt{2}}(|-2\rangle + |2\rangle), & |T_x\rangle &= \frac{i}{\sqrt{2}}(|-1\rangle + |1\rangle), \\ |\downarrow\rangle &= |0\rangle, & |T_y\rangle &= \frac{1}{\sqrt{2}}(|-1\rangle - |1\rangle), \\ & & |T_z\rangle &= \frac{i}{\sqrt{2}}(|-2\rangle - |2\rangle). \end{aligned} \quad (1)$$

The *E_g* doublet is of non-Kramers type and has vanishing dipole moments $\langle \mathcal{P}_{E_g}^\dagger \mathbf{J} \mathcal{P}_{E_g} \rangle = 0$ where $\mathcal{P}_{E_g} = \sum_{\omega \in \{\uparrow, \downarrow\}} |\omega\rangle \langle \omega|$ projects onto the *E_g* states. Three higher-rank multipoles retain a finite moment: The quadrupole operators $Q_{z^2} = \frac{1}{\sqrt{3}}(3J_z^2 - \mathbf{J}^2)$ and $Q_{x^2-y^2} = J_x^2 - J_y^2$, and the octupole operator $T_{xyz} = \frac{\sqrt{15}}{6} \overline{J_x J_y J_z}$ where the overline symbol denotes symmetrization [2]. These multipolar operators have the same matrix elements as the Pauli matrices within the *E_g* doublet and can be considered as effective pseudospin-1/2 operators. Microscopic models of the *E_g* doublets feature a variety of multipolar-ordered ground states including antiferro-quadrupolar (AFQ) and ferro-octupolar (FO) orders [26,27,32–35]. The lack of an observed structural transition for the Mg, Ca, and Zn compounds seems to imply that AFQ is unlikely to be the low-temperature phase. On the other hand, the predicted antiferro-distortions tend to generate tiny structural deformations which can be missed in some diffraction experiments.

For example, consider the 5*d*¹ DP Ba₂MgReO₆ [37,38] which features a $j_{\text{eff}} = 3/2$ pseudospin on each Re⁶⁺ ion and two thermodynamic anomalies at temperatures $T_q > T_m$ where $T_q = 33$ K ($T_m = 18$ K) corresponds to the onset of quadrupolar (dipolar) order [39–42]. In Ref. [39] the μ SR data suggests that there are two inequivalent oxygen sites which hints at an underlying tetragonal distortion, yet no such effect is detected via neutron diffraction. On the other hand, a very small cubic-to-tetragonal distortion below T_q was detected using the high-resolution synchrotron x-ray diffraction on a sample with high crystallinity, and is predominantly associated with the ordering of alternating $Q_{x^2-y^2}$ quadrupoles near T_q [41]. This highlights the challenges faced in the accurate structural determination of DPs, especially the detection of quadrupolar-induced antiferro-distortions in 5*d* materials [21,41]. It is also worthwhile to note that in the case of CeB₆ ($4f^1$) there is no change in B₆ positions despite the established AF quadrupole order [43–45].

While the lack of a structural transition in the Ba₂MOsO₆ for the *M* = Mg, Ca, Zn compounds has been verified using

synchrotron x-ray diffraction [31], it is worthwhile to examine the stability of the high-symmetry cubic structure. Note that the non-Kramers doublet degeneracy is sensitive to general structural deformation which may result in the loss of multipolar physics. Furthermore, the situation regarding the $M = \text{Cd}$ material is not yet fully determined. In the next section we investigate the phonon spectra of the four $5d^2$ Os DPs to examine the stability of the cubic structure.

III. STRUCTURAL PROPERTIES OF $5d^2$ OSMIUM DOUBLE PEROVSKITES

Early approaches to analyze perovskite structural properties include Goldschmidt's tolerance factor, which is adapted for DPs $A_2BB'O_6$ as

$$t = \frac{(r_A + r_O)}{\sqrt{2}\left(\frac{r_B + r_{B'}}{2} + r_O\right)}, \quad (2)$$

where r_l is the radius of ion l ; $t = 1$ would then correspond to the high-symmetry cubic structure with space group symmetry $Fm\bar{3}m$ (No. 225) for the rocksalt formation [see Fig. 2(a)] [46,47]. Deviations from $t = 1$ indicate a mismatch between the size of the A and B/B' ions, inducing structural distortions that lower the crystal symmetry. The Ba_2MOsO_6 materials we consider fall into two doppelgänger pairs since the radii of the $\text{Mg}^{2+}/\text{Zn}^{2+}$ (0.72/0.74 Å) and $\text{Ca}^{2+}/\text{Cd}^{2+}$ (1.00/0.95 Å) ions are roughly equivalent, resulting in tolerance factors of $t_{\text{Mg,Zn}} \sim 1.04$ and $t_{\text{Ca,Cd}} \sim 0.985$ [48]. As a result, one may expect the structural properties within each pair of doppelgängers to be equivalent.

We test this hypothesis by investigating the phonon dispersion of each material in its electronic ground state, which can be calculated using DFT. We first perform a structural optimization where the ions are relaxed from the initial $Fm\bar{3}m$ structure. The fcc conventional unit cell has Os at representative Wyckoff position $4a = (0, 0, 0)$, M at $4b = (\frac{1}{2}, \frac{1}{2}, \frac{1}{2})$, Ba at $8c = (\frac{1}{4}, \frac{1}{4}, \frac{1}{4})$, and O at $24e = (w, 0, 0)$, with $w \sim 0.23$ – 0.24 depending on the choice of M . We relax the primitive unit cell which has lattice constant $a \sim 5.7$ – 6.0 Å and contains one formula unit, i.e., ten atoms. Once the optimized structure is obtained, we calculate the interatomic force constants using density functional perturbation theory (DFPT) [49,50], which, after Fourier interpolation, yields the dynamical matrix at nonzero \mathbf{q} . Diagonalization of the dynamical matrix then yields the dispersion $\omega_s(\mathbf{q})$ for a given phonon branch $s = 1, \dots, 30$. The computational details of this three-step procedure are given in Appendix A.

The phonon dispersion for the Ca/Cd doppelgänger pair in the $Fm\bar{3}m$ structure is shown in Fig. 2; the Mg/Zn pair are given in Appendix A for completeness. We will use calligraphic font to distinguish phonon irreps from the states in Eq. (1), i.e., the E_g spin-orbital doublet vs the \mathcal{E}_g phonon mode. At the Γ point the 27 optical phonon modes decompose into irreps of O_h as $A_1g \oplus \mathcal{E}_g \oplus \mathcal{T}_{1g} \oplus 2\mathcal{T}_{2g} \oplus 4\mathcal{T}_{1u} \oplus \mathcal{T}_{2u}$ [51]. The clearest difference among the Ca/Cd pair is the \mathcal{T}_{1g} mode which takes on an imaginary frequency at the Γ point for the Cd compound [see Fig. 2(d)]. This occurs when the phonon spectrum is computed about an unstable structural equilibrium [52]; in this case, our *ab initio* DFT analysis suggests that the ideal cubic structure of

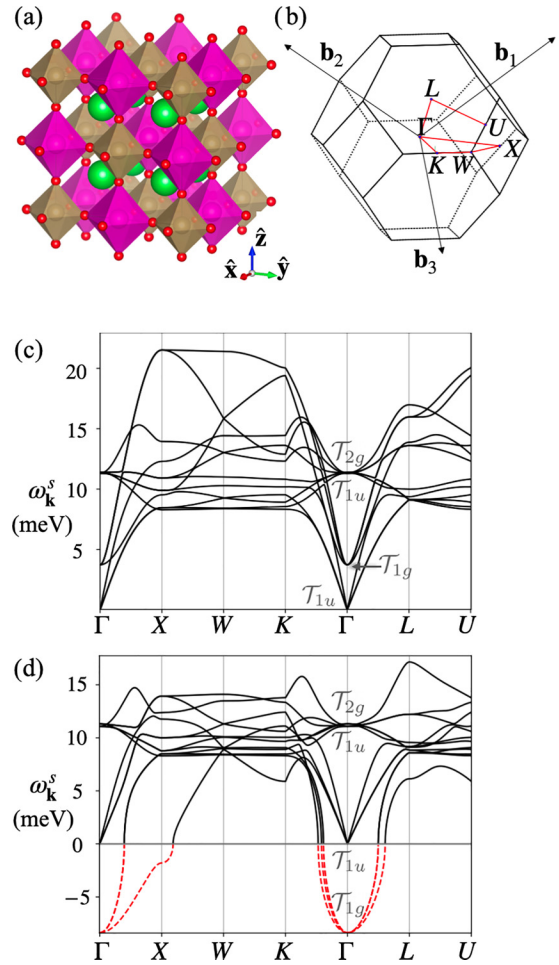


FIG. 2. (a) Ba_2MOsO_6 in the ideal rocksalt structure with space group $Fm\bar{3}m$, where OsO_6 (light brown) and MO_6 (fuschia) corner-sharing octahedra alternate in an fcc pattern surrounding Ba ions (green). (b) First Brillouin zone of the fcc primitive reciprocal lattice vectors along with points of high symmetry. Low-energy phonon spectrum for (c) $\text{Ba}_2\text{CaOsO}_6$ and (d) $\text{Ba}_2\text{CdOsO}_6$ in the $Fm\bar{3}m$ structure as calculated by the *ab initio* approach detailed in Appendix A; the phonon irrep at the Γ point is indicated in gray. The k path traversed is indicated in (b). The presence of unstable \mathcal{T}_{1g} modes for the cubic $M = \text{Cd}$ material (dashed red) signals a structural distortion instability induced by octahedral tilting at low temperatures.

$\text{Ba}_2\text{CdOsO}_6$ with $Fm\bar{3}m$ symmetry is unstable and undergoes a structural transition to a state with lower symmetry at low temperatures.

The correct structure for $\text{Ba}_2\text{CdOsO}_6$ is obtained by displacing the atoms according to the three unstable eigenvectors associated with the \mathcal{T}_{1g} modes. This corresponds to fixing the positions of the Ba, Cd, and Os atoms while rotating the oxygen ions about the $[111]$ direction. The resulting structure is shown in Fig. 3(a) and consists of uniformly staggered octahedral canting, i.e., $a^-a^-a^-$ in Glazer notation with space group $R\bar{3}$ (No. 148) [53,54]. The tilting reduces the Cd-O-Os bond angle to $\psi = 148.3^\circ$ as shown in Fig. 3(a). While the space group symmetry is lowered, the primitive unit cell still contains one formula unit and maintains its shape; i.e., rhombohedral with $\alpha = 60^\circ$. We repeat the *ab initio* analysis for $\text{Ba}_2\text{CdOsO}_6$ in the low-symmetry structure and find that all

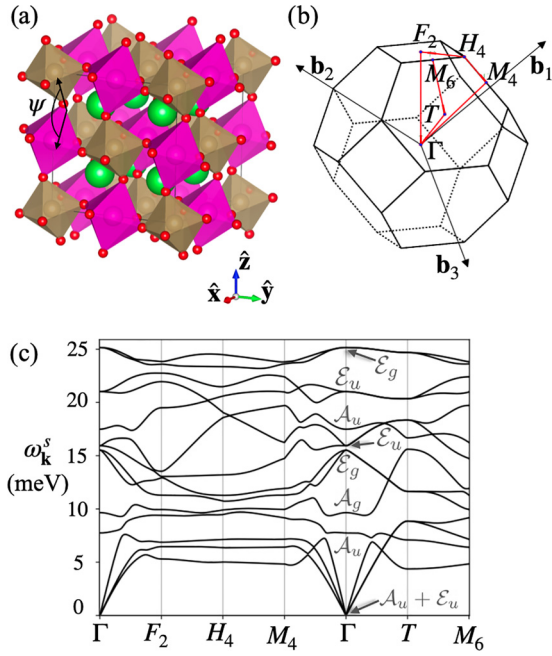


FIG. 3. (a) Crystal structure of the $a^-a^-a^-$ distorted $\text{Ba}_2\text{CdOsO}_6$ with space group $R\bar{3}$, where each OsO_6 octahedron undergoes a uniform tilting in size and orientation. The Cd-O-Os angle is given by $\psi = 148.3^\circ$. (b) First Brillouin zone of the rhombohedral primitive reciprocal lattice vectors along with points of high symmetry; the rhombohedral setting is used. (c) Low-energy phonon spectrum for the crystal structure shown in (a) calculated by *ab initio* techniques. The k path traversed is indicated in (b). The lack of imaginary phonon frequencies indicates that the $R\bar{3}$ structure is dynamically stable at low temperatures for the $M = \text{Cd}$ compound.

optical phonon modes carry nonzero frequency [see Fig. 3(c)], indicating that the $R\bar{3}$ structure is stable.

IV. E_g AND T_{2g} STATES UNDER TRIGONAL DISTORTION

In the previous section we argued that the stable structure for $\text{Ba}_2\text{CdOsO}_6$ has space group $R\bar{3}$, which breaks most O_h point-group operations at each Os^{6+} ion yet retains the C_3 symmetry about the $[111]$ axis. This is compatible with trigonal deformation of the OsO_6 octahedra; for $\text{Ba}_2\text{CdOsO}_6$ we find a compression of the OsO_6 cage along the $[111]$ axis (see Fig. 6 in Appendix B). In the trigonal environment the t_{2g} orbital degeneracy is reducible into $t_{2g} = a_g \oplus e'_g$ irreps of S_6 . The E_g non-Kramers doublet arising from $J = 2$ is not protected by the time-reversal symmetry and may be sensitive to the breaking of cubic crystalline symmetries. However, as the trigonal distortion preserves C_3 symmetry, the multipolar physics of the E_g doublet may yet survive if the excited T_{2g} states (which decompose as $T_{2g} = A_g \oplus E'_g$ in analogy with t_{2g} orbitals) do not cross energies with the E_g doublet.

We investigate the fate of the multipolar physics by modeling the trigonal CEF and using exact diagonalization (ED) to solve for the single-ion spectrum of the electronic $5d^2$ configuration. The CEF Hamiltonian is given by

$$H_{\text{CEF}}(\delta) = 10Dq \sum_{\alpha\sigma;\beta\sigma'} [\Xi^{\alpha\beta}(\delta) \otimes \mathbb{1}_{\sigma\sigma'}] d_{\alpha\sigma}^\dagger d_{\beta\sigma'}, \quad (3)$$

where $d_{\alpha\sigma}^\dagger$ creates a single electron with spin $\sigma \in \{+, -\}$ in d orbital α , and $10Dq = 4$ eV as is typical for the materials of interest [27]. δ is an angle parametrizing the degree of trigonal distortion with $\delta > 0$ ($\delta < 0$) corresponding to trigonal elongation (compression) (see Fig. 6); the relaxed structure for $\text{Ba}_2\text{CdOsO}_6$ obtained in Sec. III has $\delta = -2.87^\circ$, equivalent to a reduction of Os-O bonds by $\sim 2\%$. The orbital level scheme is dictated by the 5×5 matrix Ξ which we estimate within the point-charge approximation in Appendix B. The local physics is also governed by SOC

$$H_{\text{SOC}} = \xi \sum_{\alpha\sigma;\beta\sigma'} [\mathbf{l}^{\alpha\beta} \cdot \mathbf{s}_{\sigma\sigma'}] d_{\alpha\sigma}^\dagger d_{\beta\sigma'}, \quad (4)$$

where \mathbf{l} and \mathbf{s} are angular momentum operators of $l = 2$ and $s = 1/2$, respectively, and ξ is the single-particle SOC strength, and the Kanamori-Hubbard interactions

$$\begin{aligned} H_{\text{int}} = & U \sum_{\alpha} n_{\alpha+} n_{\alpha-} + U' \sum_{\alpha \neq \beta} n_{\alpha+} n_{\beta-} \\ & + (U' - J_H) \sum_{\alpha < \beta, \sigma} n_{\alpha\sigma} n_{\beta\sigma} \\ & - J_H \sum_{\alpha \neq \beta} (d_{\alpha+}^\dagger d_{\alpha-} d_{\beta-}^\dagger d_{\beta+} + d_{\alpha+}^\dagger d_{\beta-} d_{\alpha-}^\dagger d_{\beta+}), \end{aligned} \quad (5)$$

where $n_{\alpha\sigma} = d_{\alpha\sigma}^\dagger d_{\alpha\sigma}$, J_H is Hund's coupling strength, and U ($U' = U - 2J_H$) is the intraorbital (interorbital) Hubbard parameter. For a given δ the local Hamiltonian $H_{\text{int}} + H_{\text{SOC}} + H_{\text{CEF}}(\delta)$ can be diagonalized within the basis of $\binom{10}{2} = 45$ possible d^2 states with $10Dq = 4$ eV, $U = 2.5$ eV, $\xi = 0.4$ eV, and finite J_H . At $\delta = 0$ the five lowest states are the E_g doublet and T_{2g} triplet and we choose $J_H = 0.2U$ so that the E_g - T_{2g} splitting is $\Delta \sim 10$ meV. For our choice of U this corresponds to $J_H = 0.5$ eV, which is a reasonable upper bound for the Hund's coupling in the $5d$ series [55]. The result for finite δ is given in Fig. 4; within the set of parameters considered the five lowest eigenvalues are gapped from the rest of the spectrum.

Unlike the tetragonal case, trigonal distortions do not split the non-Kramers E_g doublet due to the presence of C_3 symmetry, while the excited triplet decomposes as $T_{2g} = A_g \oplus E'_g$. The E_g - E'_g doublets undergo level repulsion with the E_g states lower in energy for all δ . On the other hand, the A_g singlet is the lowest eigenvalue at larger distortions, crossing the E_g doublet and suppressing the multipolar physics within. In the case of elongation, A_g is lower in energy than E'_g and the singlet crosses the E_g doublet at a relatively small distortion of $\delta \sim +0.18^\circ$. The case of trigonal compression is more interesting, as the E_g doublet remains the state of lowest energy until $\delta \sim -2.1^\circ$.

The asymmetry between compression and elongation can be traced back to filling the a_g and e'_g single-electron orbital states. Trigonal elongation lowers the e'_g doublet and the resulting two-electron state has its orbital degrees of freedom quenched. On the other hand, trigonal compression lowers the a_g singlet, and the two-electron state is predominantly composed of (a) the filled singlet and (b) the state where both a_g and e'_g are singly occupied. Thus there is a competition between the distortion and Hund's coupling, with the latter seeking to eliminate orbital fluctuations. This frustration manifests itself in the relatively small energy difference between

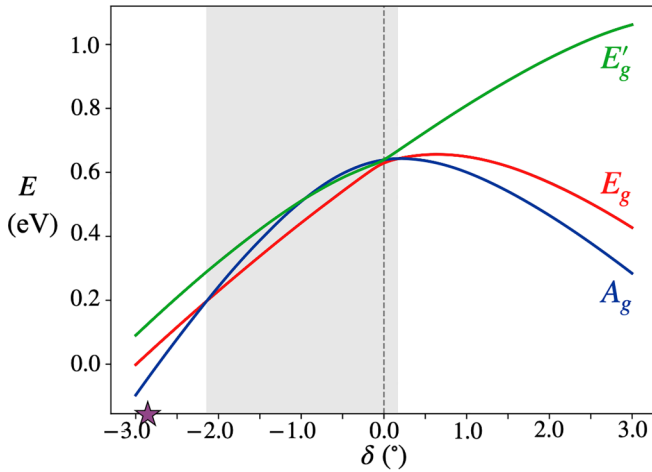


FIG. 4. ED calculation of the five lowest eigenvalues of $H_{\text{int}} + H_{\text{SOC}} + H_{\text{CEF}}(\delta)$ as δ is varied with $10Dq = 4$ eV, $U = 2.5$ eV, $J_H = 0.5$ eV, and $\xi = 0.4$ eV. $\delta > 0$ ($\delta < 0$) corresponds to trigonal elongation (compression). At $\delta = 0$ the T_{2g} triplet is separated by a gap of $\Delta \sim 10$ meV from the E_g doublet (red) and splits into an A_g singlet (blue) and E'_g doublet (green). The shaded region corresponds to the range of distortion where the E_g doublet remains the lowest energy level. The purple star corresponds to the distortion in $\text{Ba}_2\text{CdOsO}_6$ of $\delta = -2.85^\circ$.

the E'_g and A_g states when $\delta \lesssim 0$ (see Fig. 4), and is finally resolved in the limit of strong distortions by stabilizing the A_g singlet. The trigonal compression in $\text{Ba}_2\text{CdOsO}_6$ is given by $\delta = -2.85^\circ$, which is just beyond the A_g - E_g crossing. This suggests that the multipolar physics in rhombohedral $\text{Ba}_2\text{CdOsO}_6$ is likely to be revealed in pressure-tuned experiments by approaching the cubic limit.

V. SUMMARY AND DISCUSSION

In summary, we investigated the structural properties of four $5d^2$ Os DPs Ba_2MOsO_6 using an *ab initio* calculation of the phonon spectrum. We found that while the $M = \text{Mg}$, Zn , Ca compounds have a stable cubic structure with space group $Fm\bar{3}m$, the $M = \text{Cd}$ material forms the rhombohedral $R\bar{3}$ structure at low temperatures. The transition to the low-symmetry state is indicated by softening of the \mathcal{T}_{1g} phonon modes which results in out-of-phase tilting of the oxygen octahedra in each of three crystallographic directions, i.e., $a^-a^-a^-$. The rhombohedral structure generally allows for trigonal distortion of the OsO_6 octahedra, which modifies the physics of the $J = 2$ states by splitting the excited T_{2g} triplet into an E'_g doublet and A_g singlet. There exists a finite window where the E_g doublet is the lowest energy level, allowing for the ordering of various multipolar orders. A strong enough deformation stabilizes the A_g singlet and suppresses the E_g doublet's multipolar magnetism.

It is interesting to note that while $r_{\text{Ca}} > r_{\text{Cd}}$ by only 0.05 \AA , $\text{Ba}_2\text{CaOsO}_6$ remains cubic at low temperatures despite having a lower tolerance factor than $\text{Ba}_2\text{CdOsO}_6$. This is likely due to the nonionic nature of DPs which cannot be accounted for by the tolerance factor calculated from tabulated ionic radii [21]. One likely explanation for the difference between the Ca and Cd compounds is the number of d electrons

present on the M^{2+} ion, i.e., d^0 and d^{10} , respectively. The capacity for π bonding in $\text{Ba}_2\text{CdOsO}_6$ favors the reduction of the Cd-O-Os bond angles, while the lack of such for $\text{Ba}_2\text{CaOsO}_6$ results in linear Ca-O-Os bonds [56–58]. While Eq. (2) provides a good first approach to the classification of perovskite structural properties, the fact that $\text{Ba}_2\text{CaOsO}_6$ and $\text{Ba}_2\text{CdOsO}_6$ form a doppelgänger pair but have different low-temperature structures motivates the need to go beyond the tolerance factor by considering each compound's chemical properties.

The presence of trigonal distortion allows for new interorbital exchange pathways including xy - xz / yz hopping (parametrized by t_4 in Ref. [27]) for bonds parallel to the xy plane. The parameters of the E_g pseudospin-1/2 effective model are modified in the weak hopping limit, and the dependence on t_4 in $5d^2$ DPs with small trigonal distortion is an interesting subject left for future study. We find that the distortion in $\text{Ba}_2\text{CdOsO}_6$ is large enough to place the material outside the region where the effective E_g model is valid. Since the singlet crosses the doublet and becomes the lowest state separated by a small gap from the E_g , the intersite exchange interaction may become important in $\text{Ba}_2\text{CdOsO}_6$. While the singlet itself has no moment, the intersite exchange interaction may induce a magnetic moment.

Our first main conclusion is that $\text{Ba}_2\text{CdOsO}_6$ should feature a cubic-rhombohedral structural transition at low temperatures. Investigation of this scenario can be determined using high-resolution synchrotron x-ray diffraction, a reliable technique for detecting antiferro-distortions in $5d$ DPs [31,41]. The cubic-rhombohedral structural transition has not been considered in previous studies of the $5d^2$ Ba_2MOsO_6 series yet is common in other DPs, for example the strontium osmate $\text{Sr}_2\text{CrOsO}_6$ [59–61] and a series of strontium antimony oxides $\text{Sr}_2B''\text{SbO}_6$ [62–64]. The structural properties of these materials have been explored using x-ray and neutron powder diffraction measurements, and display a series of structural transitions (monoclinic \rightarrow rhombohedral \rightarrow cubic) as temperature is increased. In particular, the $R\bar{3} \rightarrow Fm\bar{3}m$ transition is observed to be of second order as predicted by Landau theory [54] and corresponds to the suppression of the cubic (642) peak below the ordering temperature [62–65]. Similar observations in the low-temperature phase of $\text{Ba}_2\text{CdOsO}_6$ would serve as experimental evidence for rhombohedral symmetry in this material. Another signature of the symmetry lowering would be the splitting of triplet phonon modes $\mathcal{T}_{1g,2g}$ at the Γ point into irreps of S_6 as $\mathcal{T}_{1g,2g} = A_g \oplus E_g$. In Fig. 3(c) we see that the A_g - E_g splitting is roughly 5 meV; such an energy difference can be resolved via Raman scattering. Moreover, inelastic scattering experiments can reveal signatures of strong vibronic coupling in the excitation spectrum as recently demonstrated for K_2IrCl_6 [66]. In this case, the spin-orbit-lattice entangled vibronic state does not result in static structural distortions due to dynamic Jahn-Teller considerations; we leave this scenario open for future research and analysis. Our findings highlight the importance of considering the interplay between crystal structure and electronic states when exploring the magnetic properties of $5d$ spin-orbit coupled magnets.

The second main conclusion is that multipolar physics can survive in $5d^2$ materials with trigonal distortions. The E_g

doublet does not split under trigonal distortions and there exists a window of finite distortion where it remains the lowest energy level, especially for the case of trigonal compressions. This suggests that multipolar physics can be found in structures with lower symmetry such as $R\bar{3}$. Interestingly, this space group is common amongst several honeycomb compounds including CrI_3 [67]. A recent proposal has suggested that $5d^2$ honeycomb compounds could host exotic ordered and disordered multipolar phases, including the Kitaev multipolar liquid [36]. The identification of a $5d^2$ honeycomb material with, e.g., $R\bar{3}$ symmetry and small trigonal compression is an excellent first step to the realization of the Kitaev multipolar liquid and forms an interesting direction for future work.

Note added. Recently, we became aware of the study by Voleti *et al.* [68] which also features the phonon dispersion for the $M = \text{Ca}$ material in the $Fm\bar{3}m$ structure. Our results match those of Ref. [68] and include the $M = \text{Mg, Zn, Cd}$ materials in addition.

ACKNOWLEDGMENTS

A.R. is grateful for helpful discussions with D. Churchill and S. Voleti. We acknowledge support from the Natural Sciences and Engineering Research Council of Canada Discovery Grant No. 2022-04601. H.Y.K. also acknowledges support from the Canadian Institute for Advanced Research and the Canada Research Chairs Program. Computations were performed on the Niagara supercomputer at the SciNet HPC Consortium. SciNet is funded by the Canada Foundation for Innovation; the Government of Ontario; Ontario Research Fund - Research Excellence; and the University of Toronto.

APPENDIX A: AB INITIO CALCULATIONS AND PHONON DISPERSIONS FOR $M = \text{Mg/Zn}$

The DFT calculations are performed using the Vienna *Ab initio* Simulation Package (VASP) with the Perdew-Burke-Ernzerhof exchange-correlation functional and cutoff energy of 600 eV [69,70]. The crystal structures of all four materials considered in this work are obtained from the Materials project [71]. Structural relaxation of the primitive unit cell in the initial $Fm\bar{3}m$ structure is performed using a Γ -centered $8 \times 8 \times 8 k$ mesh with maximum force/atom of 10^{-4} eV/Å. In the ionic relaxation all degrees of freedom are varied including ionic positions, cell shape, and cell volume; care is taken to eliminate artifacts associated with the Pulay stress by repeated re-relaxation of the intermediate ionic positions. We then used VASP to calculate interatomic force constants using DFPT simulations on a $2 \times 2 \times 2$ supercell with a Γ -centered $4 \times 4 \times 4 k$ mesh and an energy convergence threshold of 10^{-8} eV. The supercell contains 8 f.u. and has a (super)lattice constant of roughly 12 Å; a large supercell is chosen to eliminate artifacts arising from long-range forces. Finally, phonopy performs the diagonalization of the dynamical matrix at \mathbf{q} commensurate with the supercell and the Fourier interpolation at general \mathbf{q} [72,73]. In Fig. 5 we present the phonon dispersion for the Mg/Zn doppelgänger pair. VASPKIT was used to prepare the VASP simulations such as in creation of the position and ionic relaxation/DFPT input files [74]. Crystal structures shown in Figs. 2(a) and 3(a) were visualized

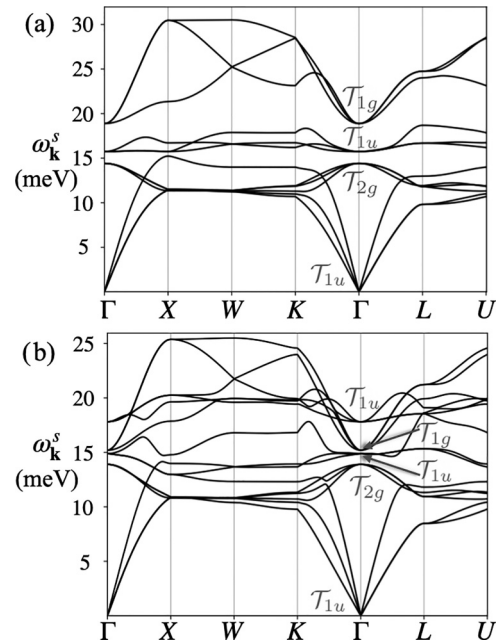


FIG. 5. Low-energy phonon spectrum calculated via an *ab initio* approach for the (a) $\text{Ba}_2\text{MgOsO}_6$ and (b) $\text{Ba}_2\text{ZnOsO}_6$ materials in the $Fm\bar{3}m$ structure, with symmetry points shown in Fig. 2. The phonon irrep at the Γ point is shown in gray. The lack of imaginary phonon frequencies indicates the stability of the $Fm\bar{3}m$ structure at low temperatures for the $M = \text{Mg, Zn}$ compounds.

using VESTA [75]. The Brillouin zones in Figs. 2(b) and 3(b) were generated using the ATOMIC SIMULATION ENVIRONMENT package [76] with symmetry points labeled according to the HPKOT convention [77].

APPENDIX B: CONSTRUCTION OF THE TRIGONAL CEF

We can estimate the effect of trigonal distortion on electrons in the $5d$ shell by constructing a trigonal CEF Hamiltonian in the point-charge limit. We will only consider trigonal compression and elongation of the OsO_6 cage, but other deformations are allowed in principle [78]. The oxygen positions can be parametrized as

$$\begin{aligned} \mathbf{r}_1 &= \frac{b}{\sqrt{2}}\hat{\mathbf{x}} + f(\delta)\hat{\mathbf{c}}, & \mathbf{r}_4 &= -\mathbf{r}_1, \\ \mathbf{r}_2 &= \frac{b}{\sqrt{2}}\hat{\mathbf{y}} + f(\delta)\hat{\mathbf{c}}, & \mathbf{r}_5 &= -\mathbf{r}_2, \\ \mathbf{r}_3 &= \frac{b}{\sqrt{2}}\hat{\mathbf{z}} + f(\delta)\hat{\mathbf{c}}, & \mathbf{r}_6 &= -\mathbf{r}_3, \end{aligned} \quad (\text{B1})$$

where $\hat{\mathbf{c}} = \frac{1}{\sqrt{3}}(\hat{\mathbf{x}} + \hat{\mathbf{y}} + \hat{\mathbf{z}})$, b is the length of the octahedron edge in the ideal limit, and

$$f(\delta) = \frac{b}{4} \left[\tan\left(60^\circ + \frac{\delta}{2}\right) - \tan(60^\circ) \right], \quad (\text{B2})$$

where $\delta > 0$ ($\delta < 0$) corresponds to the case of trigonal elongation (compression) (see Fig. 6). For $\text{Ba}_2\text{CdOsO}_6$ we find that $b = 2.94$ Å and $\delta = -2.85^\circ$, equivalent to a reduction in Os-O bonds by $\sim 2\%$. Note that the xyz coordinates in

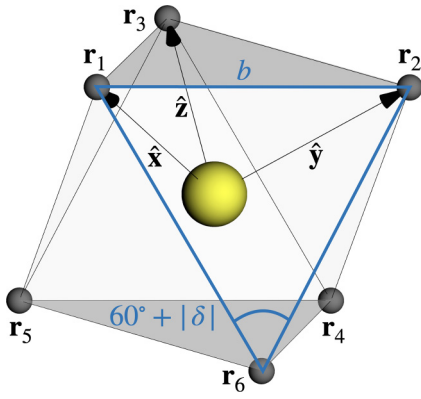


FIG. 6. Geometry used to derive the CEF in the case of trigonal compression ($\delta < 0$), where the $[111]$ and $[\bar{1}\bar{1}\bar{1}]$ faces (shaded) are pressed towards each other. In $\text{Ba}_2\text{CdOsO}_6$ the three oxygen ions at $\mathbf{r}_{1,2,6}$ form an isosceles triangle (blue) with base $b = 2.94 \text{ \AA}$ and $\delta = -2.85^\circ$. The xyz coordinates in this figure (i.e., the local octahedral coordinates at each Os atom) differ from the xyz coordinates shown in Fig. 3(a) (the crystallographic directions) by a rotation about the $[111]$ direction due to the octahedral canting in the $R\bar{3}$ structure.

Eq. (B1) refer to the local octahedral coordinates at each osmium atom, which differ from the crystallographic xyz coordinates by a rotation about the $[111]$ direction. This is due to the octahedral tilting within the $R\bar{3}$ structure which affects all OsO_6 octahedra uniformly. With this in mind, we will continue to use the xyz symbols to simplify notation.

The potential energy of the $5d^2$ electrons in each Os^{6+} ion in the presence of the O^{2-} point charges can be written in a multipole expansion given by

$$V(\mathbf{r}) = \sum_{k=0}^{\infty} \sum_{p=-k}^k r^k \left[\frac{4\pi A}{2k+1} \sum_{i=1}^6 \frac{Y_{kp}(\theta_i, \phi_i)^*}{r_i^{k+1}} \right] Y_{kp}(\theta, \phi), \quad (\text{B3})$$

for $r \equiv |\mathbf{r}| < |\mathbf{r}_i|$ [79]. The proportionality constant $A > 0$ is fixed so that the t_{2g} - e_g splitting at $\delta = 0$ is $10Dq = 4 \text{ eV}$. We then evaluate matrix elements of Eq. (B3) between hydrogenic wave functions $R_{n=5,l=2}(r)Y_{l=2,l_z}(\theta, \phi)$. Since $l = 2$ we may restrict the k summation to $k \leq 4$ and we further ignore the $k = 0$ contribution as it does not carry an angular dependence. When $\delta \neq 0$ the t_{2g} degeneracy is lifted and it becomes more appropriate to write the trigonal CEF in the following basis of single-electron creation operators [80]:

$$\begin{aligned} d_{a_g}^\dagger &= \frac{1}{\sqrt{3}}(d_{xy}^\dagger + d_{yz}^\dagger + d_{xz}^\dagger), \\ d_{e_g^-}^\dagger &= \frac{1}{\sqrt{3}}(d_{xy}^\dagger + \omega^2 d_{yz}^\dagger + \omega d_{xz}^\dagger), \\ d_{e_g^+}^\dagger &= \frac{-1}{\sqrt{3}}(d_{xy}^\dagger + \omega d_{yz}^\dagger + \omega^2 d_{xz}^\dagger), \\ d_{e_g^\mp}^\dagger &= \frac{\pm 1}{\sqrt{2}}(d_{z^2}^\dagger \mp i d_{x^2-y^2}^\dagger), \end{aligned} \quad (\text{B4})$$

where $\omega = \exp(2\pi i/3)$ and the spin index is suppressed. For $\delta = -2.85^\circ$ as in $\text{Ba}_2\text{CdOsO}_6$ we find that the trigonal CEF is given by

$$\Xi^{\alpha\beta}(\delta) = \begin{pmatrix} -0.197 & 0 & 0 & 0 & 0 \\ 0 & 0.0406 & 0 & 0.145 & 0 \\ 0 & 0 & 0.0406 & 0 & 0.145 \\ 0 & 0.145 & 0 & 1.058 & 0 \\ 0 & 0 & 0.145 & 0 & 1.058 \end{pmatrix}, \quad (\text{B5})$$

and the numerical values are in units of $10Dq$. By diagonalizing Eq. (B5) one can check that the trigonal compression $\delta < 0$ yields a low-lying a_g singlet whereas elongation $\delta > 0$ prefers the lowering of the e_g' doublet. Equation (B5) demonstrates the mixing of e_g and e_g' states of the same parity under trigonal deformations.

[1] V. Tripathi, P. Chandra, and P. Coleman, *Nat. Phys.* **3**, 78 (2007).
[2] P. Santini, S. Carretta, G. Amoretti, R. Caciuffo, N. Magnani, and G. H. Lander, *Rev. Mod. Phys.* **81**, 807 (2009).
[3] Y. Kuramoto, H. Kusunose, and A. Kiss, *J. Phys. Soc. Jpn.* **78**, 072001 (2009).
[4] A. Sakai and S. Nakatsuji, *J. Phys. Soc. Jpn.* **80**, 063701 (2011).
[5] T. U. Ito, W. Higemoto, K. Ninomiya, H. Luetkens, C. Baines, A. Sakai, and S. Nakatsuji, *J. Phys. Soc. Jpn.* **80**, 113703 (2011).
[6] T. Onimaru, K. T. Matsumoto, Y. F. Inoue, K. Umeo, T. Sakakibara, Y. Karaki, M. Kubota, and T. Takabatake, *Phys. Rev. Lett.* **106**, 177001 (2011).
[7] T. J. Sato, S. Ibuka, Y. Nambu, T. Yamazaki, T. Hong, A. Sakai, and S. Nakatsuji, *Phys. Rev. B* **86**, 184419 (2012).
[8] M. Tsujimoto, Y. Matsumoto, T. Tomita, A. Sakai, and S. Nakatsuji, *Phys. Rev. Lett.* **113**, 267001 (2014).
[9] A. M. Konic, R. B. Adhikari, D. L. Kunwar, A. A. Kirmani, A. Breindel, R. Sheng, M. B. Maple, M. Dzero, and C. C. Almasan, *Phys. Rev. B* **104**, 205139 (2021).

[10] K. Hattori and H. Tsunetsugu, *J. Phys. Soc. Jpn.* **83**, 034709 (2014).
[11] K. Hattori and H. Tsunetsugu, *J. Phys. Soc. Jpn.* **85**, 094001 (2016).
[12] S. B. Lee, S. Trebst, Y. B. Kim, and A. Paramekanti, *Phys. Rev. B* **98**, 134447 (2018).
[13] T. Ishitobi and K. Hattori, *Phys. Rev. B* **104**, L241110 (2021).
[14] A. S. Patri, A. Sakai, S. Lee, A. Paramekanti, S. Nakatsuji, and Y. B. Kim, *Nat. Commun.* **10**, 4092 (2019).
[15] K. I. Kugel and D. I. Khomskii, *Sov. Phys. Usp.* **25**, 231 (1982).
[16] G. A. Gehring and K. A. Gehring, *Rep. Prog. Phys.* **38**, 1 (1975).
[17] G. Chen and L. Balents, *Phys. Rev. B* **84**, 094420 (2011).
[18] G. Chen, R. Pereira, and L. Balents, *Phys. Rev. B* **82**, 174440 (2010).
[19] T. Dodds, T.-P. Choy, and Y. B. Kim, *Phys. Rev. B* **84**, 104439 (2011).
[20] H. Ishizuka and L. Balents, *Phys. Rev. B* **90**, 184422 (2014).

- [21] S. Vasala and M. Karppinen, *Prog. Solid State Chem.* **43**, 1 (2015).
- [22] J. Romhányi, L. Balents, and G. Jackeli, *Phys. Rev. Lett.* **118**, 217202 (2017).
- [23] C. Svoboda, W. Zhang, M. Randeria, and N. Trivedi, *Phys. Rev. B* **104**, 024437 (2021).
- [24] S. Voleti, D. D. Maharaj, B. D. Gaulin, G. Luke, and A. Paramakanti, *Phys. Rev. B* **101**, 155118 (2020).
- [25] T. Takayama, J. Chaloupka, A. Smerald, G. Khaliullin, and H. Takagi, *J. Phys. Soc. Jpn.* **90**, 062001 (2021).
- [26] G. Khaliullin, D. Churchill, P. P. Stavropoulos, and H.-Y. Kee, *Phys. Rev. Res.* **3**, 033163 (2021).
- [27] D. Churchill and H.-Y. Kee, *Phys. Rev. B* **105**, 014438 (2022).
- [28] K. Yamamura, M. Wakeshima, and Y. Hinatsu, *J. Solid State Chem.* **179**, 605 (2006).
- [29] C. M. Thompson, J. P. Carlo, R. Flacau, T. Aharen, I. A. Leahy, J. R. Pollicheimi, T. J. S. Munsie, T. Medina, G. M. Luke, J. Munevar, S. Cheung, T. Goko, Y. J. Uemura, and J. E. Greedan, *J. Phys.: Condens. Matter* **26**, 306003 (2014).
- [30] C. A. Marjerrison, C. M. Thompson, A. Z. Sharma, A. M. Hallas, M. N. Wilson, T. J. S. Munsie, R. Flacau, C. R. Wiebe, B. D. Gaulin, G. M. Luke, and J. E. Greedan, *Phys. Rev. B* **94**, 134429 (2016).
- [31] D. D. Maharaj, G. Sala, M. B. Stone, E. Kermarrec, C. Ritter, F. Fauth, C. A. Marjerrison, J. E. Greedan, A. Paramakanti, and B. D. Gaulin, *Phys. Rev. Lett.* **124**, 087206 (2020).
- [32] S. W. Lovesey and D. D. Khalayavin, *Phys. Rev. B* **102**, 064407 (2020).
- [33] S. Voleti, A. Haldar, and A. Paramakanti, *Phys. Rev. B* **104**, 174431 (2021).
- [34] L. V. Pourovskii, D. F. Mosca, and C. Franchini, *Phys. Rev. Lett.* **127**, 237201 (2021).
- [35] D. Fiore Mosca, L. V. Pourovskii, and C. Franchini, *Phys. Rev. B* **106**, 035127 (2022).
- [36] A. Rayyan, D. Churchill, and H.-Y. Kee, *Phys. Rev. B* **107**, L020408 (2023).
- [37] A. W. Sleight, J. Longo, and R. Ward, *Inorg. Chem.* **1**, 245 (1962).
- [38] K. G. Bramnik, H. Ehrenberg, J. K. Dehn, and H. Fuess, *Solid State Sci.* **5**, 235 (2003).
- [39] C. A. Marjerrison, C. M. Thompson, G. Sala, D. D. Maharaj, E. Kermarrec, Y. Cai, A. M. Hallas, M. N. Wilson, T. J. S. Munsie, G. E. Granroth, R. Flacau, J. E. Greedan, B. D. Gaulin, and G. M. Luke, *Inorg. Chem.* **55**, 10701 (2016).
- [40] D. Hirai and Z. Hiroi, *J. Phys. Soc. Jpn.* **88**, 064712 (2019).
- [41] D. Hirai, H. Sagayama, S. Gao, H. Ohsumi, G. Chen, T.-H. Arima, and Z. Hiroi, *Phys. Rev. Res.* **2**, 022063(R) (2020).
- [42] H. Arima, Y. Oshita, D. Hirai, Z. Hiroi, and K. Matsubayashi, *J. Phys. Soc. Jpn.* **91**, 013702 (2022).
- [43] H. Nakao, K.-i. Magishi, Y. Wakabayashi, Y. Murakami, K. Koyama, K. Hirota, Y. Endoh, and S. Kunii, *J. Phys. Soc. Jpn.* **70**, 1857 (2001).
- [44] Y. Tanaka, U. Staub, K. Katsumata, S. Lovesey, J. Lorenzo, Y. Narumi, V. Scagnoli, S. Shimomura, Y. Tabata, Y. Onuki, Y. Kuramoto, A. Kikkawal, T. Ishikawal, and H. Kitamura, *Europhys. Lett.* **68**, 671 (2004).
- [45] P. Fazekas, *Lecture Notes on Electron Correlation and Magnetism* (World Scientific, Singapore, 1999).
- [46] V. Goldschmidt, *Naturwissenschaften* **14**, 477 (1926).
- [47] D. Babel, R. Haegele, G. Pausewang, and F. Wall, *Mater. Res. Bull.* **8**, 1371 (1973).
- [48] J. R. Rumble, *CRC Handbook of Chemistry and Physics*, 103rd ed. (CRC Press, Boca Raton, FL, 2022).
- [49] P. Giannozzi, S. de Gironcoli, P. Pavone, and S. Baroni, *Phys. Rev. B* **43**, 7231 (1991).
- [50] X. Gonze and C. Lee, *Phys. Rev. B* **55**, 10355 (1997).
- [51] E. Kroumova, M. Aroyo, J. Perez-Mato, A. Kirov, C. Capillas, S. Ivantchev, and H. Wondratschek, *Phase Trans.* **76**, 155 (2003).
- [52] A. A. Maradudin, E. W. Montroll, G. H. Weiss, and I. P. Ipatova, *Solid State Physics 3. Theory of Lattice Dynamics in the Harmonic Approximation* (Academic Press, New York, 1971).
- [53] A. M. Glazer, *Acta Cryst. B* **28**, 3384 (1972).
- [54] C. J. Howard, B. J. Kennedy, and P. M. Woodward, *Acta Cryst. B* **59**, 463 (2003).
- [55] D. Khomskii, *Transition Metal Compounds* (Cambridge University Press, Cambridge, UK, 2014).
- [56] P. J. Saines, B. J. Kennedy, and M. M. Elcombe, *J. Solid State Chem.* **180**, 401 (2007).
- [57] P. J. Saines, J. R. Spencer, B. J. Kennedy, and M. Avdeev, *J. Solid State Chem.* **180**, 2991 (2007).
- [58] P. J. Saines, J. R. Spencer, B. J. Kennedy, Y. Kubota, C. Minakata, H. Hano, K. Kato, and M. Takata, *J. Solid State Chem.* **180**, 3001 (2007).
- [59] Y. Krockenberger, M. Reehuis, M. Tovar, K. Mogare, M. Jansen, and L. Alff, *J. Magn. Magn. Mater.* **310**, 1854 (2007).
- [60] Y. Krockenberger, K. Mogare, M. Reehuis, M. Tovar, M. Jansen, G. Vaitheeswaran, V. Kanchana, F. Bultmark, A. Delin, F. Wilhelm, A. Rogalev, A. Winkler, and L. Alff, *Phys. Rev. B* **75**, 020404(R) (2007).
- [61] R. Morrow, J. R. Soliz, A. J. Hauser, J. C. Gallagher, M. A. Susner, M. D. Sumption, A. A. Aczel, J. Yan, F. Yang, and P. M. Woodward, *J. Solid State Chem.* **238**, 46 (2016).
- [62] A. Faik, M. Gateshki, J. Igartua, J. Pizarro, M. Insausti, R. Kaindl, and A. Grzechnik, *J. Solid State Chem.* **181**, 1759 (2008).
- [63] A. Faik, E. Iturbe-Zabalo, I. Urcelay, and J. Igartua, *J. Solid State Chem.* **182**, 2656 (2009).
- [64] B. Orayech, A. Faik, and J. Igartua, *Polyhedron* **123**, 265 (2017).
- [65] B. Orayech, I. Urcelay-Olabarria, G. A. López, O. Fabelo, A. Faik, and J. M. Igartua, *Dalton Trans.* **44**, 13867 (2015).
- [66] N. Iwahara and W. Furukawa, [arXiv:2305.05853](https://arxiv.org/abs/2305.05853).
- [67] B. Morosin and A. Narath, *J. Chem. Phys.* **40**, 1958 (1964).
- [68] S. Voleti, K. Pradhan, S. Bhattacharjee, T. Saha-Dasgupta, and A. Paramakanti, [arXiv:2211.07666](https://arxiv.org/abs/2211.07666).
- [69] G. Kresse and J. Furthmüller, *Phys. Rev. B* **54**, 11169 (1996).
- [70] J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).
- [71] A. Jain, S. P. Ong, G. Hautier, W. Chen, W. D. Richards, S. Dacek, S. Cholia, D. Gunter, D. Skinner, G. Ceder, and K. A. Persson, *APL Mater.* **1**, 011002 (2013).
- [72] A. Togo and I. Tanaka, *Scr. Mater.* **108**, 1 (2015).
- [73] A. Togo, *J. Phys. Soc. Jpn.* **92**, 012001 (2023).
- [74] V. Wang, N. Xu, J.-C. Liu, G. Tang, and W.-T. Geng, *Comput. Phys. Commun.* **267**, 108033 (2021).

- [75] K. Momma and F. Izumi, *J. Appl. Cryst.* **41**, 653 (2008).
- [76] A. H. Larsen, J. J. Mortensen, J. Blomqvist, I. E. Castelli, R. Christensen, M. Dulak, J. Friis, M. N. Groves, B. Hammer, and C. Hargus, *J. Phys.: Condens. Matter* **29**, 273002 (2017).
- [77] Y. Hinuma, G. Pizzi, Y. Kumagai, F. Oba, and I. Tanaka, *Comput. Mater. Sci.* **128**, 140 (2017).
- [78] P. P. Stavropoulos, X. Liu, and H.-Y. Kee, *Phys. Rev. Res.* **3**, 013216 (2021).
- [79] G. B. Arfken, H. J. Weber, and F. E. Harris, *Mathematical Methods for Physicists* (Academic Press, New York, 2012).
- [80] S. Sugano, Y. Tanabe, and H. Kamimura, *Multiplets of Transition-Metal Ions in Crystals* (Academic Press, New York, 1970).