# In-plane electron-phonon coupling anisotropy and multiple charge density wave orders in the superconductor Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>

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(Received 15 November 2022; revised 13 June 2023; accepted 20 June 2023; published 12 July 2023)

The ternary chalcogenide superconductor  $Bi_2Rh_3Se_2$  was discovered to have charge density wave (CDW) order and electron-phonon coupling (EPC). However, it is still debated whether the phase transition at ~240 K is a CDW one. Here, we systematically performed Raman scattering investigations on single-crystal  $Bi_2Rh_3Se_2$ . Angle-resolved polarized Raman spectroscopic measurements reveal the in-plane anisotropic EPC in  $Bi_2Rh_3Se_2$ , which inhibits the Raman intensity. Temperature-dependent x-ray diffraction and Raman measurements indicate the existence of the incommensurate (ICCDW) and commensurate (CCDW) CDW orders below ~250 and ~170 K, respectively, in  $Bi_2Rh_3Se_2$ . The appearance of zone-folded phonon modes at ~170 K confirms the CCDW-induced 2 × 2 reconstruction of the Brillouin zone. Semiquantitative analysis based on Landau theory of second-order phase transitions of CDW-correlated and uncorrelated phonon modes reflects extensive coupling between the phonons and CDW orders. Our findings elucidate the existence of CCDW and ICCDW and confirm the critical role of EPC in CDW transitions in  $Bi_2Rh_3Se_2$ .

DOI: 10.1103/PhysRevB.108.045118

# I. INTRODUCTION

Charge density wave (CDW) and superconductivity are two very important and closely linked collective excitations in solids, and their interactions have been investigated widely [1–4]. BCS superconducting theory [5] and the CDW formation mechanism [6,7] point out that both superconductivity and CDW order originate from electron-phonon coupling (EPC) and compete with each other. However, recent investigations on  $2H - NbSe_2$  demonstrate that CDW order does not simply compete with superconductivity [8]. There is increasing evidence for CDW order existing in high- $T_c$  cuprate superconductors [1,3,9,10]. The competition or coexistence relationship between CDW and high- $T_c$  superconductivity remains unclear and deserves more in-depth investigations.

Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> offers a potential platform to investigate the interaction between CDW order and superconductivity, since it is reported to exhibit a superconducting transition at  $T_c \sim 0.7$  K and a CDW order below  $T_s \sim 240$  K [11–13]. Measurements such as electrical resistivity, specific heat, and temperature-dependent x-ray diffraction (TDXRD) [11] as well as optical conductivity [12] and angle-resolved photoelectron spectroscopy (ARPES) [13] suggest that the phase transition at  $\sim 240$  K is a CDW one, showing evidence of EPC [12,13]. However, pressure-dependent electrical resistivity measurements illustrate an unexpected rise in anomaly temperature with pressure. Moreover, temperature-dependent

selected-area electron diffraction (SAED) measurements demonstrate that it is a purely structural transition [14]. Thus, disclosing whether the phase transition at  $\sim$ 240 K is a CDW one or a purely structural one is the first and most essential step. Raman spectroscopy is an effective method to investigate the physical properties such as EPC and phase transitions: Angle-resolved polarized Raman spectroscopy (ARPRS) can reveal the in-plane anisotropy [15,16], and temperature-dependent Raman spectroscopy (TDRS) can unveil the phonon anharmonicity, purely structural phase transition, as well as second-order phase transitions, such as CDW and magnetic phase transitions, etc. [17–34].

In this work, we report on the systematic Raman scattering investigations on single crystalline Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>. XRD, scanning electron microscopy (SEM), and high-resolution transmission electron microscopy (HRTEM) measurements have been carried out to demonstrate the high quality of the as-grown Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> crystals. ARPRS measurements reveal the in-plane phonon and EPC anisotropies. An excitation wavelength-dependent EPC is also observed. TDRS combined with TDXRD measurements from 80 to 300 K disclose two distinct second-order phase transitions at  ${\sim}170\pm10$  K  $(T_1)$  and  $\sim 250 \pm 10$  K  $(T_2$ , close to  $T_s$ ) in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>, respectively. The reversible appearance and disappearance of zone-folded phonon modes provide clear evidence of the Brillouin zone (BZ) reconstruction following CCDW transition at  $\sim 170$  K. Quantitative analysis of the evolution of peak frequencies and intensities with temperature shows an interaction between phonons and CDWs. Our findings demonstrate the worth and urgency of further studies of CDWs and their interactions with superconductivity in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>.

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FIG. 1. (a) Crystal structure of  $Bi_2Rh_3Se_2$ . (b) XRD pattern of as-grown  $Bi_2Rh_3Se_2$  crystal. The inserted optical image shows the  $Bi_2Rh_3Se_2$  crystal in a millimeter-scale. (c) SEM image of a freshly cleaved  $Bi_2Rh_3Se_2$  crystal. Black dotted arrows present crystallographic *a*- and *b*- axes. (d) HRTEM image of an exfoliated  $Bi_2Rh_3Se_2$  flake. The inset shows the corresponding fast Fourier transform pattern.

### **II. RESULTS AND DISCUSSION**

Normal-state bulk  $Bi_2Rh_3Se_2$  with a space group C12/m1has the parkerite-type structure with standardized unit-cell parameters of a = 1.141 nm, b = 0.837 nm, c = 0.822 nm, and  $\beta = 133.148^{\circ}$  [35], as schematically shown in Fig. 1(a). Rh atoms (blue) are covalently bonded with Se (yellow), Bi (red), and Rh atoms to form anisotropic pseudo-two-dimensional networks, implying there possibly exists anisotropic properties in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> [11,14]. The room-temperature XRD pattern of the millimeter-scale as-grown Bi2Rh3Se2 crystals (optical image inserted) shown in Fig. 1(b) only displays five intense peaks at  $2\theta = 14.84^{\circ}$ ,  $29.86^{\circ}$ ,  $45.42^{\circ}$ ,  $61.94^{\circ}$ , and 80.04°, in good agreement with previous reports [11]. The interplanar spacing corresponding to the first peak is calculated to be 1.193 nm, which matches well with  $2 \times c \times \sin((180 - 10^{-1}))$  $(133.148)^{\circ}) = 1.199$  nm. Thus, the five peaks are assigned to (002), (004), (006), (008), and (0010) planes of Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>, respectively, revealing that the as-grown Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> are single crystals. A representative SEM image on a freshly cleaved  $Bi_2Rh_3Se_2$  flake in Fig. 1(c) shows an atomically smooth surface and long straight step edges, confirming the layered structure of Bi2Rh3Se2. The angle included by the two long straight edges (marked by black and white solid arrows) is  $\sim 73^{\circ}$  instead of 90° by vectors **a** and **b**, indicating that Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> crystals prefer to fracture along the  $[\bar{1}\bar{2}0]$  and  $[1\bar{2}0]$  directions [marked in Fig. 1(a)] rather than crystallographic a- and b-axes. Similar results have been reported in MoO<sub>2</sub> flakes [36,37]. A typical HRTEM image in Fig. 1(d) shows clear parallel stripes with an interplanar spacing of  $\sim 0.29$  nm, consistent with a a/4 lattice constant of Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>. The corresponding fast Fourier transform pattern



FIG. 2. Calculated phonon dispersion (a) and the corresponding DOS (b) of normal-state Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>. (c) Raman spectrum of asgrown Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> crystals under 532 nm excitation at 300 K.  $A_g^{10}$  mode performed by Lorentzian (d) and BWF functions (e). Ramanactive modes in regions I (f) and II (g).

inserted at the lower-right corner in Fig. 1(d) shows a rectangular symmetry and indicates that the first bright spot along the [100] direction corresponds to the  $(40\overline{2})$  plane. For more details, see Fig. S1 in the supplemental material [38].

Figure 2(a) shows the calculated phonon dispersion of the normal-state Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> (for details, see the subsection that discusses DFPT calculations in Sec. IV), which exhibits real eigenvalues through all the high-symmetry points in the first BZ (the BZ of Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> is shown in Fig. S2 [38]), confirming its dynamical stability. At the  $\Gamma$  point, there are three acoustic and 39 optical phonon modes, whose calculated frequencies and symmetries are summarized in Table I. Among them, there are  $10 A_g$  and  $8 B_g$  Raman-active modes in 0-300 cm<sup>-1</sup>. The corresponding density of states (DOS) is displayed in Fig. 2(b), extremely similar to the unpolarized Raman spectrum of as-grown Bi2Rh3Se2 crystals under 532 nm excitation at 300 K in Fig. 2(c). The spectral ranges from  $\sim$ 50 to 100 and  $\sim$ 150 to 210 cm<sup>-1</sup> are named regions I and II, respectively. The isolated peak at  $\sim 233 \,\mathrm{cm}^{-1}$  is assigned to  $A_{g}^{10}$  mode (according to Table I).

 $A_g^{10}$  mode in Fig. 2(d) shows an asymmetric line shape and is performed poorly by the Lorentzian function (the coefficient of determination,  $R^2$ , is ~0.958). Similar asymmetric Raman line shapes have been observed in other superconductors and heavily *p*-doped semiconductor Si, all arising from Fano-type discrete-continuous resonances between phonons and electrons [17,18,22,23,39]. The modified Raman peak intensity (*I*) can be given by the Breit-Wigner-Fano (BWF)

TABLE I. Calculated frequencies and symmetries of phonon modes in  $Bi_2Rh_3Se_2$ .

Mode	$A_{\rm g}$	$A_{\mathrm{u}}$	$B_{ m g}$	B <sub>u</sub>
Frequency (cm <sup>-1</sup> )	$66.3(A_g^1)$	$0.1(A_{\rm u}^1)$	$61.7(B_g^1)$	$-0.1(B_{\rm u}^1)$
	$70.3(A_{g}^{2})$	$59.7(A_{\rm u}^2)$	$78.6(B_{g}^{2})$	$0.1(B_{\rm u}^2)$
	$76.3(A_{o}^{\bar{3}})$	$79.1(A_{\rm u}^{3})$	$79.6(B_{g}^{5})$	$62.7(B_{\rm u}^3)$
	$79.9(A_{g}^{\tilde{4}})$	$88.7(A_{\rm u}^{4})$	$91.3(B_{g}^{4})$	$67.7(B_{\rm u}^{4})$
	$93.1(A_{\sigma}^{5})$	$104.5(A_{\rm u}^5)$	$126.1(B_{g}^{5})$	$72.3(B_{\rm u}^5)$
	$117.3(\tilde{A_{g}^{6}})$	$123.2(A_{\rm u}^{6})$	$180.3(B_{g}^{\delta})$	$81.2(B_{\rm u}^{6})$
	$140.8(A_{g}^{5})$	$141.4(A_{\rm u}^{\bar{7}})$	$217.2(B_{g}^{5})$	$95.9(B_{\rm u}^{\bar{7}})$
	$169.2(A_{g}^{8})$	$193.0(A_{\rm u}^8)$	$264.5(B_{g}^{8})$	$103.2(B_{\rm u}^8)$
	$174.2(A_{g}^{5})$	$219.8(A_{\rm u}^{\bar{9}})$	/	$118.1(B_{\rm u}^{\rm 9})$
	$234.1(A_{g}^{10})$	$262.0(A_{\rm u}^{10})$	/	$147.9(B_{\rm u}^{10})$
	/	/	/	$173.3(B_{u}^{\bar{1}1})$
	/	/	/	$198.8(B_{\rm u}^{12})$
	/	/	/	$223.4(B_{\rm u}^{13})$
	/	/	/	$263.3(B_{\rm u}^{14})$

function and written as [22,23,40]

$$I(\omega) = A \frac{[q + 2(\omega - \omega_0)/F]^2}{1 + [2(\omega - \omega_0)/F]^2},$$
(1)

where A is the intensity parameter,  $\omega_0$  is renormalized phonon frequency, q is a parameter that quantitatively describes asymmetry, 1/q can describe EPC strength, and F is the full width at half-maximum (FWHM) parameter. As displayed in Fig. 2(e),  $A_g^{10}$  mode can be better described by the BWF function ( $R^2 \sim 0.980$ ), and the extracted q is  $\sim -8.78$ , suggesting the existence of EPC in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>, as demonstrated in our previous report [13]. Then, six intense Raman-active modes with BWF line shapes shown in Figs. 2(f) and 2(g) were resolved at ~60, 68, 75, 168, 178, and 185 cm<sup>-1</sup> in regions I ( $R^2 \sim 0.999$ ) and II ( $R^2 \sim 0.981$ ), corresponding to  $A_g^1, A_g^2, A_g^3, A_g^8, A_g^9$ , and  $B_g^6$  modes, respectively, according to Fig. 2(a) and Table I. The three extremely weak peaks at ~50, 84, and 92 cm<sup>-1</sup> (U1, U2, and U3) are difficult to identify due to the lack of experimental evidence and interference from dense phonon modes.

### A. In-plane phonon and EPC anisotropies in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>

Figure 3(a) shows the 3D colormap surfaces with the projection of the normalized *I* according to the ARPRS spectra of Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> under 532 and 633 nm excitation in a parallel polarization configuration, respectively. The polarization angle ( $\theta$ ) is defined as 0° when the incident polarization is aligned with the black solid arrow in Fig. 1(c). The *I* of the phonon modes varies periodically with  $\theta$ , indicating in-plane phonon anisotropy. For clarity, the extracted *I* of the corresponding mode is plotted in Figs. 3(b) and S3 [38]. According to the classical Placzek approximation [41], *I* is proportional to  $|e_i \cdot R \cdot e_s^{\tau}|^2$ , where *R* is the Raman tensor, and  $e_i$  and  $e_s$ are the unit polarization vectors of incident and scattering light, respectively.  $e_i = e_s = (\cos \theta, \sin \theta, 0)$  in a parallel polarization configuration. Raman tensors of the  $A_g$  and  $B_g$ modes in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> are expressed as

$$R_{A_{g}} = \begin{pmatrix} ue^{i\varphi_{u}} & de^{i\varphi_{d}} & 0\\ de^{i\varphi_{d}} & ve^{i\varphi_{v}} & 0\\ 0 & 0 & we^{i\varphi_{w}} \end{pmatrix},$$
(2)  
$$R_{R} = \begin{pmatrix} 0 & fe^{i\varphi_{f}} & 0\\ fe^{i\varphi_{f}} & 0 & 0 \end{pmatrix},$$
(3)

0

0

0



FIG. 3. (a) 3D colormap surfaces with the projection of ARPRS spectra of as-grown Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> crystal under 532 and 633 nm excitation in parallel polarization configuration. Polar plots of *I* (b), 1/q (c), and corresponding anisotropic ratios ( $\frac{\text{the absolute value}}{\text{the absolute minimum value}}$ ) (d) for  $A_g^{10}$  mode. Solid lines are performed curves. Green and red dots represent 532 and 633 nm excitation, respectively. The extracted experimental data in (c) and (d) are fitted by a function of  $a_1 \sin\theta + a_2 \cos\theta$ .

where u, v, w, d, and f are Raman tensor elements, and  $\varphi$  is the phase of the Raman tensor element. I is associated with  $A_g$ and  $B_g$  modes in a parallel polarization configuration, which are written as

$$I_{A_g} = u^2 \cos^4\theta + v^2 \sin^4\theta + 4d^2 \sin^2\theta \cos^2\theta + 4ud \cos^3\theta \sin\theta \cos\varphi_{ud} + 4vd \sin^3\theta \cos\theta \cos\varphi_{vd} + 2uv \sin^2\theta \cos^2\theta \cos 2\varphi_{uv}, \qquad (4)$$

$$I_{B_e} = f^2 \sin^2 2\theta, \tag{5}$$

where  $\varphi_{ud} = \varphi_u - \varphi_d$ ,  $\varphi_{vd} = \varphi_v - \varphi_d$ ,  $\varphi_{uv} = \varphi_u - \varphi_v$ . The polar plots of *I* in Figs. 3(b) and S3 [38] are performed well by Eqs. (4) and (5). The  $B_g^6$  mode shows a four-lobed shape with intensity maxima at ~82°, 172°, 262°, and 352° regardless of excitation wavelength. Under 532 and 633 nm excitation, the  $A_g^3$ ,  $A_g^8$ , and  $A_g^{10}$  modes all show a bow-tie-like shape. The intensity maxima of  $A_g^3$  and  $A_g^8$  modes are along ~ 37° (217°) (*b*-axis), and the intensity maximum of  $A_g^{10}$  mode is along ~ 127° (307°) (*a*-axis). Interestingly, under 532 nm excitation, the  $A_g^1$ ,  $A_g^2$ , and  $A_g^9$  modes show a propeller-shaped pattern with two intensity maxima along the *b*- and *a*-axes, respectively. Under 633 nm excitation, all still maintain a propeller-shaped pattern, but they show a secondary maximum in the normal direction of the corresponding intensity maximum, revealing excitation-wavelength-dependent polarization responses for  $A_g^1$ ,  $A_g^2$ , and  $A_g^9$  modes.

It has been reported that the polarizations of certain phonon modes are modified by optical birefringence, interference effects, and EPC when the excitation wavelength is changed in anisotropic materials [15,16,27,42–44]. The measured Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> crystal is hundreds of microns thick (much larger than the excitation wavelength) and is put on copper tape. Thus, the excitation-wavelength dependent EPC is most likely the determinant factor for the anomalous polarizations of  $A_g^1$ ,  $A_g^2$ , and  $A_g^9$  modes in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>, while the optical birefringence and interference effects may play a minor role and can be ignored [44]. Excitation wavelength-dependent EPCs have been directly and widely observed by ARPRS [15,22,23,42]. We adopt the semiquantitative analysis of 1/qto investigate the excitation-wavelength dependence of EPC in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>.

Figure 3(c) plots the 1/q of the  $A_g^{10}$  mode under 532 and 633 nm excitation, respectively. Under both excitations, 1/q varies with  $\theta$  in a bow-tie-like mode, revealing the EPC anisotropy. Surprisingly, the intensity maximum of 1/q is along the *b*-axis, perpendicular rather than parallel to the *I* intensity maximum of the  $A_g^{10}$  mode (*a*-axis). This anomaly seems to indicate that EPC tends to inhibit I in  $Bi_2Rh_3Se_2$ , similar to the case in superconducting Nd(O, F)BiS<sub>2</sub> [22], but contrary to the fact that EPC is the dominant contributor to the I of the  $A_g$  mode in black phosphorus when the optical effects can be ignored [15,42]. Along the *b*-axis, the absolute value of 1/q excited by 633 nm (~0.20) is about twice that excited by 532 nm ( $\sim$ 0.10), indicating a stronger EPC under 633 nm excitation. A decrease in the strength of EPC with increasing excitation photon energy has been observed in other superconductors [22]. To compare the relative strengths of I and 1/q for the  $A_{o}^{10}$  mode under 532 and 633 nm excitation, the anisotropic ratios ( $\frac{\text{the absolute value}}{\text{the absolute minimum value}}$ ) of *I* and 1/q as a function of  $\theta$  were compared quantitatively, as shown in Fig. 3(d). The ratios of both *I* and 1/q under 633 nm excitation are larger than those under 532 nm excitation, indicating stronger inplane anisotropy under 633 nm excitation. Extracting reliable parameters of EPC from  $A_g^1$ ,  $A_g^2$ , and  $A_g^9$  modes in the future may make the analysis more rigorous.

#### B. Multiple CDW orders in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>

To investigate the temperature evolution of Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> phonons, the single-crystal TDRS (no polarizer) and TDXRD [along the (0,0,L) direction] measurements were performed from 80 to 300 K. The TDRS results are reversible during cooling-down and warming-up cycles at different incident polarizations, as shown in Fig. S4 [38]. A cooling cycle of TDRS measurement (incident polarization along the *b*-axis) is shown in Fig. 4(a). The Raman spectrum at 300 K (above  $T_s$ ) is substantially different from that at 80 K (below  $T_s$ ), indicating that phase transitions have occurred. Figure 4(b) shows the second derivative image of Fig. 4(a), where two distinct phase transitions are clearly observed at  ${\sim}170\pm10$  K (T<sub>1</sub>) and  ${\sim}250\pm$ 10 K ( $T_2$ ). Around  $T_2$ , the  $A_g^9$  mode disappears, and a slight mode change at  $\sim 125 \text{ cm}^{-1}$  can also be observed (zoom-in Raman spectra; see Fig. S5 [38]), which corresponds to the widely reported phase transition at  $T_s \sim 240$  K in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> [11–14], demonstrating the existence of a symmetry-breaking phase below  $T_2$ . Sakamoto *et al.* and Chen *et al.* have performed a systematic powder TDXRD study in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>, and only one additional diffraction peak was found in their respective results [11,14]. Figures S6(a)-(f) show the complete data and the zoomed-in diffraction peaks of (002), (004), (006), (008), and (0010) of Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> from our single-crystal TDXRD results, which do not reflect the apparent difference in the unit-cell parameter c with a temperature increase [38]. The extracted peak positions of (008) and (0010) peaks in Figs. S7(a) and (b) only display two slight changes at  $T_1$  and  $T_2$ , respectively, well consistent with the TDRS results [38]. Our TDXRD measurements indicate that Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> does not undergo a structural phase transition like Bi<sub>2</sub>Rh<sub>3</sub>S<sub>2</sub> [45]. The optical conductivity measurement indicates the formation of an energy gap with associated spectral change only at low energies in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>, supporting a CDW transition at 240 K [12]. ARPES measurement observes the temperature dependence of the electronic structures in Bi2Rh3Se2 and reveals CDW-induced band folding, band splitting, and the opening of the CDW gaps at and away from the Fermi level [13]. Thus, a pure structural phase transition is not supported in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>. A similar Raman temperature dependence for a CDW transition has been observed in  $TaSe_3$  [46]. Therefore, the phase transition around  $T_2$  is considered to be a CDW transition [11–13].

As the temperature decreases further below  $T_1$ , in region I, the  $A_g^1$  and  $A_g^2$  modes merge into a continuum (C1) broadly centered at ~62 cm<sup>-1</sup>, and U1 blueshifts to ~55 cm<sup>-1</sup> as a shoulder. For details, see the zoomed-in spectra in Fig. 4(c) and the analysis of selected peaks in Fig. S8 [38]. Another obvious change can be observed in region II and in the corresponding zoomed-in spectra in Fig. 4(d). All of the Raman



FIG. 4. (a) 3D colormap surface with the projection of unpolarized TDRS measurements of as-grown  $Bi_2Rh_3Se_2$  crystal under 532 nm excitation. (b) Second derivative image of panel (a). Zoomed-in spectra of regions I (c) and II (d) in the cooling-down cycle. (e) Selected peak analysis based on BWF function at 80 K,  $T_1$ ,  $T_2$ , and 300 K. (f) Zoomed-in calculated phonon dispersion from 150 to 220 cm<sup>-1</sup>. (g) Phase diagram of  $Bi_2Rh_3Se_2$  as a function of temperature.

peaks increase drastically with *I*, and two additional asymmetric peaks named *P*1 and *P*2 appear at ~192 and 207 cm<sup>-1</sup>, respectively, below  $T_1$  [peaks fitting is shown in Fig. 4(e)], indicating that there is another symmetry-breaking phase at lower temperatures. The above observations confirm that another phase transition occurs at  $T_1$ , which has never been reported before.

Quantitative analysis of the Raman modes in region II is used to distinguish the two symmetry-breaking phases, as shown in Fig. 4(e). At 80 K, both P1 and P2 show asymmetric line shapes with the extracted q of  $\sim -5$  and -3, respectively, suggesting EPC. Our previous ARPES report [13] has revealed that  $Bi_2Rh_3Se_2$  is in a 2 × 2 CCDW phase at 30 K, leading to a BZ reconstruction. Unfortunately, the threshold temperature for the CCDW phase transition could not be determined directly, due to the limited experimental resolution [13]. Due to the BZ reconstruction, the phonon modes are folded from the  $M_2$  point to the central  $\Gamma$  point (reconstructed BZ; see Fig. S9 [38]). Some of them exhibiting Raman activity could be assigned to the newly observed Raman peaks (such as P1 and P2). P1 and P2 are assigned to the two folded phonon modes of those at ~197 and 204 cm<sup>-1</sup> at the  $M_2$ point, highlighted by black and magenta dots, respectively, in Fig. 4(f). Both P1 and P2 begin to appear at  $T_1$ , indicating that the 2  $\times$  2 CCDW reconstruction exists below  $T_1$ . Coincidentally, Lin et al. reported that a CDW amplitude mode in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> can be observed only below 170 K  $(T_1)$  [12]. Above  $T_1$ , the decrease of I in region II and the disappearance of P1 and P2 are similar to the observations of the CCDW to ICCDW transition in TaSe<sub>2</sub> and TaS<sub>2</sub>, etc., possibly due to the loss of translation symmetry of the lattice in the ICCDW phase [25,47]. Thus, it seems that Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> has an ICCDW phase from  $T_1$  to  $T_2$ . Conclusively, the phase diagram of Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> is proposed in Fig. 4(g).

CDW transitions typically result in an abnormal  $\omega$  shift and *I* decrease of the phonon modes in Raman measurements, providing clear evidence for EPC [48–50]. Figure 5(a) emphasizes the temperature dependence of the extracted *I* (after background subtraction) of the CCDW-related *P*1 and *P*2. For both modes, *I* varies nonlinearly with increasing temperature and decays rapidly at  $\sim T_1$ . According to Landau theory of second-order phase transitions, taking the CDW gap as the order parameter ( $\eta$ ), the temperature dependence of  $\eta$  can be written as [27,49–51]

$$\eta \sim \left(1 - \frac{T}{T_{\rm CDW}}\right)^{\beta},\tag{6}$$

where  $\beta$  is the critical exponent. The temperature dependence of the Raman mode characteristic parameters X (X = I, F, and  $\omega$ , etc.) is proportional to the square of  $\eta$  [52]. If the interactions of multiple CDWs are neglected [50], their temperature dependence is

$$X(T) = X_0 + A_i \left( 1 - \frac{T}{T_{\text{CDW}i}} \right)^{2\beta} \quad (i = 1, 2, 3, \dots), \quad (7)$$

where  $X_0$  is a temperature-independent constant and  $A_i$  is a coupling constant. The solid fitting curves using Eq. (7)



FIG. 5. Temperature-dependent *I* of the *P*2 (a), the *P*1 (inset), as well as C1,  $A_g^1$ , and  $A_g^2$  modes (b). (c) Temperature-dependent *I* and  $\omega$  of  $A_g^0$  mode. The solid lines are fitting curves using Eq. (7).

in Fig. 5(a) and the inset agree well with the experimental data, and the  $T_{CCDW}^{P1}$  of ~173 K and  $T_{CCDW}^{P2}$  of ~170 K are excellently close to  $T_1$ .  $\beta$  of I of P1 and P2 are extracted to be  $\beta_{T_{CCDW}^{I}}^{I} = 0.145$  and  $\beta_{T_{CCDW}^{I}}^{I} = 0.147$ , which are much smaller than the experimentally reported  $\beta$  of ~0.3 in ErTe<sub>3</sub> [50] and the ideal  $\beta$  of 0.5 [51], suggesting a more drastic change at  $T_1$  in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>. Figure 5(b) shows the temperature dependence of I of the CCDW- and ICCDW-related phonon modes below ~70 cm<sup>-1</sup> in region I. CCDW-related C1 splits into ICCDW-related  $A_g^1$  and  $A_g^2$  modes above  $T_1$ . The I of C1 (green stars) rapidly decays at ~ $T_1$ , and those of  $A_g^1$  and  $A_g^2$  modes (black dots and red triangles) display similar behaviors at ~ $T_2$ . Such decay can be basically described by Eq. (7).

Further analysis of the CDW-unrelated  $A_g^{10}$  mode is shown in Fig. 5(c). *I* and  $\omega$  both shift discontinuously with temperature, and their slopes suddenly change at  $T_1$  and  $T_2$ . Obviously, the anharmonicity effect cannot describe them. Following Eq. (7), the temperature dependence of *I* and  $\omega$  is presented basically. All extracted important fitting parameters are summarized in Table S1 [38]. Note that both CDW-related and unrelated phonon modes change at  $T_1$  and  $T_2$ . The underlying physical mechanism deserves further investigations. It implies that there is a coupling between electrons and phonons when the CCDW and ICCDW transitions occur in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>. In addition, the temperature dependence of EPC is observed semiquantitatively by *q*. The extracted *q* (<0) from  $A_g^{10}$  mode, as shown in Fig. S10(a) [38], unexpectedly increases from  $\sim -22$  to  $\sim -12$  above  $T_1$ , indicating that EPC is enhanced. The extracted *q* (<0) from  $A_g^3$  mode shows clearly different values in CCDW, ICCDW, and normal phases, indicating that EPC is different in different phases in  $Bi_2Rh_3Se_2$ , as shown in Fig. S10(b) [38]. The above results reveal that EPC plays a vital role in the CDW transitions in  $Bi_2Rh_3Se_2$ .

### **III. CONCLUSION**

We have performed systematic Raman scattering investigations on the single-crystal Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub>. ARPRS measurements that reveal the in-plane anisotropic phonons and EPC. A consistent physical picture of newly observed Raman peaks in TDRS measurements and the zone-folding of the phonon modes indicates that a 2 × 2 CCDW phase exists in Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> below ~170 K. Furthermore, an ICCDW phase exists from ~170 to ~250 K. Temperature-dependent *q* discloses different EPC in the above phases. Our findings suggest that Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> is a possible material to investigate the interaction between CDW and superconductivity.

### **IV. MATERIALS AND METHODS**

### A. Experimental section

High-quality Bi<sub>2</sub>Rh<sub>3</sub>Se<sub>2</sub> single crystals were grown by the self-flux method [11,13], whose crystallographic structure was examined by XRD (Rigaku Smartlab) and HRTEM (jeol2100f). TDXRD measurements were performed at special temperatures in a warming cycle. All Raman measurements were taken at an InVia Qontor system (Renishaw, UK) with an 1800 lines/mm grating in the backscattering configuration under 532 and 633 nm laser excitation with a spot size less than 1 µm. The laser power was kept low enough to avoid the heating effect. The ARPRS measurements were performed from  $0^{\circ}$  to  $360^{\circ}$  with a step of  $15^{\circ}$ at room temperature. The TDRS measurements were taken from 80 to 300 K with a step of 10 K using the Raman spectroscope combined with a variable temperature sample manipulator (Linkam THMS600, UK) with 0.01 K accuracy and stability. All experiments were performed on the freshly cleaved samples, and all temperature-dependent Raman data were normalized by a divided intensity of  $A_{g}^{3}$  (~75 cm<sup>-1</sup>) mode at 80 K. The I of ARPRS in Fig. S3 is extracted by a Lorentzian function. In regions I and II, the peaks are very close to each other and change substantially with polarization angle  $\theta$ , resulting in complications in resolving individual Raman-active modes and difficulty in extracting reliable qusing the BWF function. It is reported that such regions being performed with a sum of Lorentzian functions can simplify the problem, although EPC presents [27,53]. For TDRS, the spectral ranges to fit the isolated  $A_g^{10}$  mode, regions I and II, with Fano functions are taken in the range of 200–260, 45–100, and 150–220 cm<sup>-1</sup>, respectively.

# **B. DFPT calculations**

The force constants of all atoms in the normal state are determined using VASP. In calculated dynamical properties, we include contributions from cubic interatomic force constants to phonon frequencies. Phonon calculations were performed in the supercell with 56 atoms containing  $2 \times 2 \times$ 2 conventional unit cells. The phonon-dispersion curves and phonon density of states (DOS) are calculated using PHONOPY software. We acknowledge the financial support from the National Natural Science Foundation of China (Grants No. 11874427 and No. 12074436). J.-Q.M. would like to ac-

- knowledge support from the science and technology innovation program of Hunan Province (2022RC3068). This work was carried out in part using computing resources at the High Performance Computing Center of Central South University.
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