

Magnetic, thermodynamic, and magnetotransport properties of CeGaGe and PrGaGe single crystals

Daloo Ram ¹, Sudip Malick ¹, Zakir Hossain,^{1,*} and Dariusz Kaczorowski ^{2,†}

¹*Department of Physics, Indian Institute of Technology, Kanpur 208016, India*

²*Institute of Low Temperature and Structure Research, Polish Academy of Sciences, ulica Okólna 2, 50-422 Wrocław, Poland*



(Received 22 May 2023; accepted 11 July 2023; published 28 July 2023)

We investigate the physical properties of high-quality single crystals CeGaGe and PrGaGe using magnetization, heat capacity, and magnetotransport measurements. Gallium-indium binary flux was used to grow these single crystals that crystallize in a body-centered tetragonal structure. Magnetic susceptibility data reveal a magnetic phase transition around 6.0 and 19.4 K in CeGaGe and PrGaGe, respectively, which is further confirmed by heat capacity and electrical resistivity data. A number of additional anomalies have been observed below the ordering temperature in the magnetic susceptibility data, indicating a complex magnetic structure. The magnetic measurements also reveal a strong magnetocrystalline anisotropy in both compounds. Our detailed analysis of the crystalline electric field (CEF) effect as observed in magnetic susceptibility and heat capacity data suggests that the $J = 5/2$ multiplet of CeGaGe splits into three doublets, while the $J = 4$ degenerate ground state of PrGaGe splits into five singlets and two doublets. The estimated energy levels from the CEF analysis are consistent with the magnetic entropy.

DOI: [10.1103/PhysRevB.108.024428](https://doi.org/10.1103/PhysRevB.108.024428)

I. INTRODUCTION

The competing interactions resulting from the hybridization of $4f$ and conduction electrons in the strongly correlated electron system yields remarkable physical features such as the Kondo effect, heavy-fermion behavior, intermediate valency, crystalline electric field (CEF) effect, magnetic ordering, superconductivity, and so on [1–8]. Therefore, rare-earth-based intermetallic compounds have always been appealing because they provide a platform for investigating such exotic physical properties. The introduction of strong correlation into a nontrivial topological system results in more intriguing ground states. For instance, $\text{Ce}_3\text{Bi}_4\text{Pd}_3$ is reported to be a Weyl-Kondo heavy-fermion semimetal [9,10], whereas $\text{Ce}_3\text{Bi}_4\text{Pt}_3$ displays a topological Kondo insulating state [10,11]. On the other hand, the coexistence of relativistic fermions and magnetism in these compounds manifests anomalous transport [12–15].

The ternary tetragonal compounds $RAiX$ ($R = \text{La–Nd}$ and Sm ; $X = \text{Si}$ and Ge) have gained attention recently since this class of compounds displays a Weyl semimetal (WSM) state [16–26]. Interestingly, $RAiX$ crystallizes in two space groups: LaPtSi -type noncentrosymmetric space group $I4_1md$ (No. 109) and $\alpha\text{-ThSi}_2$ -type centrosymmetric space group $I4_1/amd$ (No. 141). LaAlGe crystallizes in a noncentrosymmetric structure and exhibits a type-II WSM state as determined by angle-resolved photoemission spectroscopy [17]. Moreover, magnetic compounds are more intriguing because they provide a framework for investigating the interaction between relativistic Weyl fermions and magnetism. For instance, CeAlSi is a noncollinear ferromagnetic (FM) WSM that

exhibits an anisotropic anomalous Hall effect [23] and PrAlSi is a centrosymmetric ferromagnet manifesting an anomalous Hall effect for magnetic field applied along the c axis [25]. In addition, the substitution of Ge with Si in PrAlGe results in a crossover from intrinsic to extrinsic anomalous Hall conductivity [27]. Further, this family of compounds also exhibits strong magnetocrystalline anisotropy and the CEF effect [22–26,28]. Moreover, a number of recent reports reveal the presence of a WSM state exhibiting complex magnetic structure and anomalous transport in CeAlSi [23], CeAlGe [29], PrAlSi [25], NdAlSi [30], and NdAlGe [28,31].

In this report, we present a thorough investigation of the physical properties of CeGaGe and PrGaGe single crystals. CeGaGe is a FM Kondo lattice system with an ordering temperature of 5.5 K as determined in polycrystalline samples [32,33]. However, to the best of our knowledge there is no report on PrGaGe. Here, we have grown single crystals of $RGaGe$ ($R = \text{Ce}$ and Pr) as well as its nonmagnetic counterpart, LaGaGe , using the flux method. Powder x-ray diffraction (XRD) patterns of crushed single crystals revealed that $RGaGe$ ($R = \text{La–Pr}$) crystallizes in a noncentrosymmetric structure with space group $I4_1md$. Magnetic susceptibility data along with the heat capacity and electrical resistivity data suggest a complex magnetic ordering and CEF effect in CeGaGe and PrGaGe. Further, both compounds exhibit negative magnetoresistance near the magnetic ordering temperature.

II. EXPERIMENTAL DETAILS

High-quality single crystals of $RGaGe$ ($R = \text{La–Pr}$) were grown using gallium-indium binary flux. Constituent elements La (99.9%, Alfa Aesar), Ce (99.9%, Alfa Aesar), Pr (99.9%, Alfa Aesar), Ga (99.9999%, Alfa Aesar), Ge (99.999%, Alfa Aesar), and In (99.99%, Alfa Aesar) are taken in the molar

*zakir@iitk.ac.in

†d.kaczorowski@intibs.pl

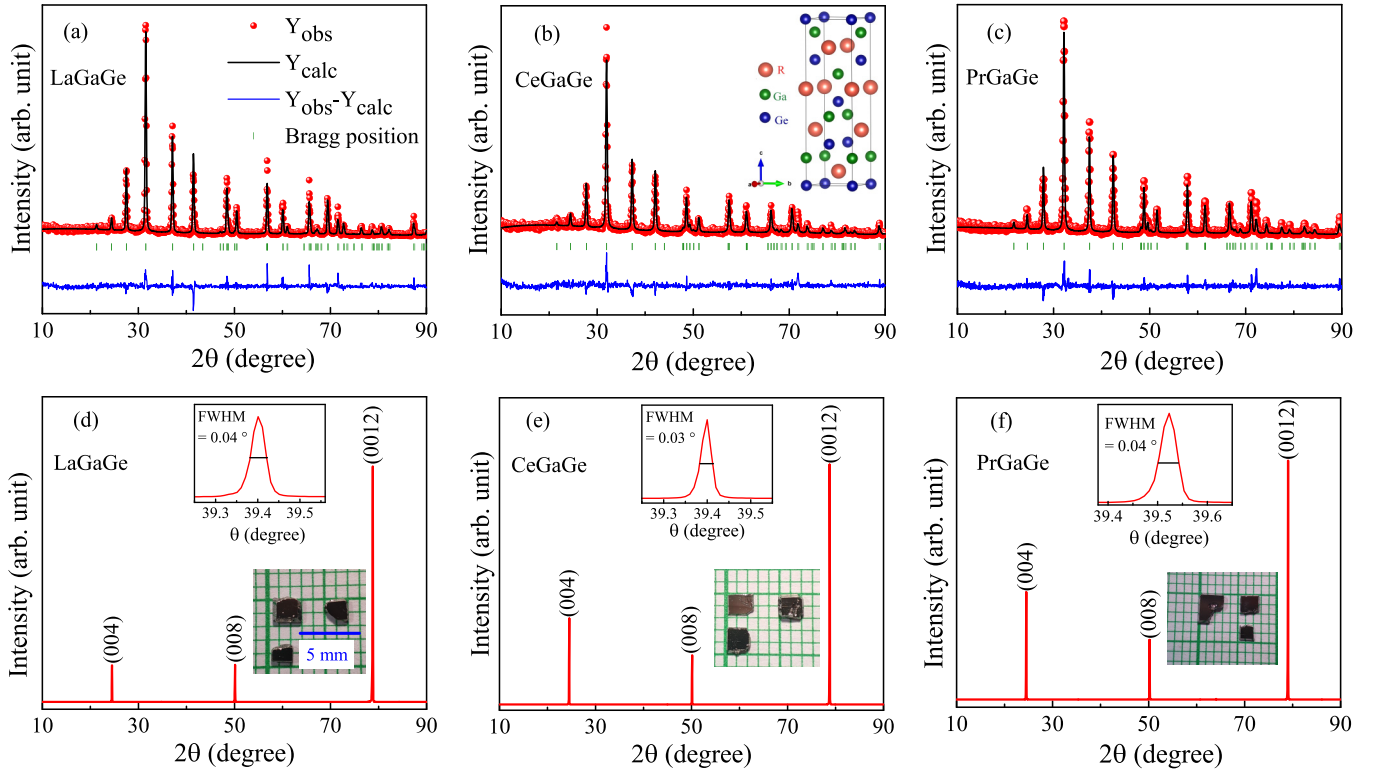


FIG. 1. Powder XRD patterns along with Rietveld refinement of crushed single crystals (a) LaGaGe, (b) CeGaGe, and (c) PrGaGe recorded at room temperature. The solid black line and red circles represent the calculated pattern and experimental data. The blue line shows difference between experimental and calculated intensities. The olive bars mark the Bragg positions. Inset of (b) displays the crystal structure of $RGaGe$. Single-crystal XRD patterns are presented for (d) LaGaGe, (e) CeGaGe, and (f) PrGaGe. Insets of (d), (e), and (f) show optical images of single crystals (lower) and the rocking curves (upper).

ratio R:Ga:Ge:In = 1:2:1:8. The elements were mixed up properly and placed in an alumina crucible. Next, the crucible was sealed inside a quartz tube with partial argon pressure. The sealed quartz tube was put into a muffle furnace and heated to 1050 °C for 24 h. Then the furnace was slowly cooled down to 500 °C at a rate of 3 °C/h. At this point the excess flux was removed by centrifuging. Platelike shiny single crystals with a typical dimension of $3 \times 3 \times 0.4 \text{ mm}^3$ were obtained as shown in the lower insets of the bottom panel of Fig. 1.

The structural characterization of the obtained crystals was carried out using the XRD method in a PANalytical X'Pert PRO diffractometer with Cu $K_{\alpha 1}$ radiation. The chemical compositions of the crystals were checked by energy-dispersive spectroscopy (EDS) in a JEOL JSM-6010LA electron microscope. Electrical resistivity and magnetotransport measurements were performed in a Quantum Design physical property measurement system (PPMS) utilizing the usual four-probe method. Heat capacity measurements were performed using the relaxation method in the heat capacity option of PPMS. The magnetic properties were measured using a Quantum Design magnetic property measurement system.

III. RESULTS AND DISCUSSION

A. Crystal structure

Powder XRD patterns of crushed crystals, recorded at room temperature, are shown in Figs. 1(a)–1(c). The

analysis of the powder XRD patterns using the Rietveld refinement method (see Supplemental Material [34]) suggest all three $RGaGe$ ($R = \text{La–Pr}$) compounds crystallize in a body-centered tetragonal structure with space group $I4_1md$ (No. 109) [35]. The estimated lattice parameters as presented in Table I agree well with the previous report [32]. A schematic diagram of the crystal structure is displayed in the inset of Fig. 1(b). The structure consists of four formula units per unit cell, in which all three atoms (R , Ga, and Ge) have the $4a$ Wyckoff site. There is stacking of R , Ga, and Ge layers along the (001) direction. This structure has two vertical mirror planes (σ_v) but lacks a horizontal mirror plane (σ_h), resulting in inversion-symmetry breaking. The observed single-crystal XRD pattern of $RGaGe$ can be indexed by (00 l) Miller indices, as depicted in Figs. 1(d)–1(f), indicating that the crystallographic c axis is perpendicular to the flat plane of single crystals. Moreover, the peaks are extremely sharp, suggesting the high quality of the single crystals. Further, the

TABLE I. The obtained lattice parameters of $RGaGe$ from powder XRD data recorded at room temperature.

Lattice parameters	LaGaGe	CeGaGe	PrGaGe
a (Å)	4.3542(2)	4.2911(5)	4.2599(4)
b (Å)	4.3542(2)	4.2911(5)	4.2599(4)
c (Å)	14.581(1)	14.576(2)	14.521(2)
V (Å ³)	276.43(3)	268.39(6)	263.51(5)

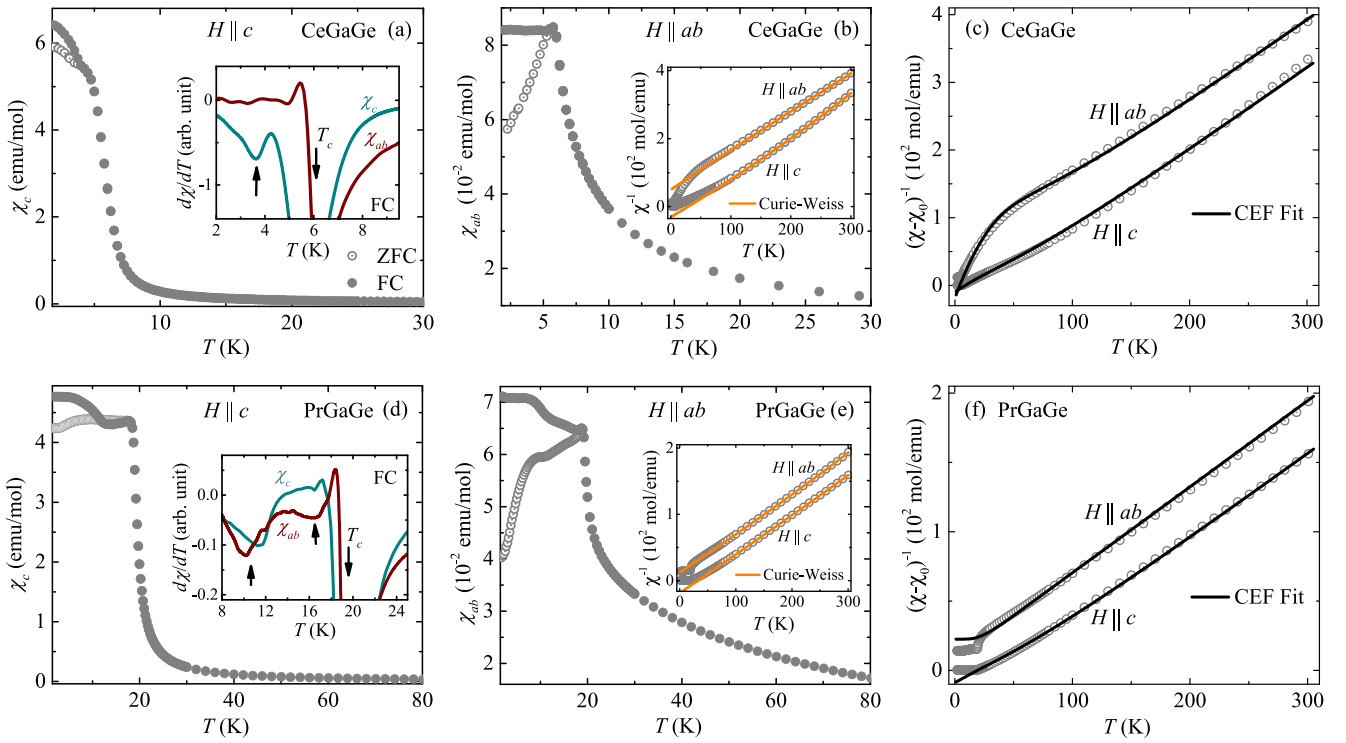


FIG. 2. The temperature-dependent magnetic susceptibility (χ) measured at 0.1 T along (a) $H \parallel c$ and (b) $H \parallel ab$ of CeGaGe. The open and filled circles represent magnetic susceptibility measured in ZFC and FC configurations, respectively. The $\chi(T)$ of PrGaGe for ZFC and FC configurations was measured at 0.1 T for (d) $H \parallel c$ and (e) $H \parallel ab$, respectively. The insets (a) and (d) display the temperature-dependent $d\chi/dT$ for FC configuration for CeGaGe and PrGaGe, respectively. The insets of (b) and (e) depict the inverse magnetic susceptibilities of CeGaGe and PrGaGe at 0.1 T, respectively, as a function of temperature. The orange lines represent Curie-Weiss fitting. The temperature-dependent inverse magnetic susceptibility of (c) CeGaGe and (f) PrGaGe at field 0.1 T. The black solid lines in (c) and (f) represent the calculated inverse susceptibility using CEF model.

full width at half maximum (FWHM) of the rocking curve [see upper insets of Figs. 1(d)–1(f)] for the peak (0012) is about 0.04° , which is very small, confirming that all single crystals are of excellent quality. EDS data collected from multiple points and areas on the surface of crystals reveal expected stoichiometry (see Supplemental Material [34]).

B. Magnetic properties

The temperature-dependent magnetic susceptibilities χ_c ($H \parallel c$) and χ_{ab} ($H \parallel ab$) for zero-field-cooled (ZFC) and field-cooled (FC) configurations at 0.1 T for both compounds are shown in Figs. 2(a), 2(b), 2(d), and 2(e). The sudden upturn in χ_c around $T_C = 6.0$ K in CeGaGe and $T_C = 19.4$ K in PrGaGe suggests a FM phase transition. However, χ_{ab} slowly increases with decreasing temperature down to T_C in both compounds. Interestingly, the ZFC and FC data exhibit a bifurcation below T_C along both crystallographic directions, more prominent along $H \parallel ab$. Moreover, χ_{ab} for ZFC drops below T_C in an AFM fashion. The value of χ_c/χ_{ab} is about 10 above T_C and reaches a value of ~ 105 at 2 K in both compounds, which suggests strong magnetocrystalline anisotropy in these compounds. A similar temperature dependence with strong magnetic anisotropy was observed in the same family of compounds $RAiX$ ($R = \text{Ce-Nd}$; $X = \text{Ge and Si}$) [23–26]. In the magnetically ordered state, $\chi(T)$ shows

an anomaly around 3.7 K in CeGaGe, and two anomalies around 11.0 and 16.5 K in PrGaGe, as is evident from the $d\chi/dT$ data, presented in the insets of Figs. 2(a) and 2(d). The observed anomalies may be associated with the transition from incommensurate to commensurate order, which has recently been observed in WSM NdAlSi [26,30] and NdAlGe [28,31]. Therefore, a comprehensive investigation utilizing neutron diffraction is required to reveal the complex magnetic structure of CeGaGe and PrGaGe. The inverse magnetic susceptibility data are plotted as a function of temperature and fitted well above T_C with the modified Curie-Weiss law, $\chi(T) = \chi_0 + C/(T - \Theta_P)$, where C is the Curie constant, Θ_P is the paramagnetic Curie temperature, and χ_0 is temperature-independent magnetic susceptibility. The insets of Figs. 2(b) and 2(e) illustrate the fitting of the Curie-Weiss law for CeGaGe and PrGaGe, respectively. The obtained fitting parameters are listed in Table II. The negative values of χ_0 may appear due to the diamagnetic contribution of the sample holder. The positive Θ_P along the c axis indicates the dominant FM exchange interactions whereas negative Θ_P along the ab plane suggests an AFM coupling. Similar values of Θ_P were also observed in isostructural compounds such as CeAlGe [22], CeAlSi [23], and PrAlGe [21,24]. The calculated effective moments μ_{eff} of CeGaGe and PrGaGe are very close to Ce^{3+} ($2.54 \mu_B$) and Pr^{3+} ($3.58 \mu_B$) ions, respectively. Below 100 K, the inverse susceptibility deviates from the

TABLE II. The estimated value of χ_0 , μ_{eff} , and Θ_p from modified Curie-Weiss fit of CeGaGe and PrGaGe.

	CeGaGe		PrGaGe	
	$H \parallel c$	$H \parallel ab$	$H \parallel c$	$H \parallel ab$
χ_0 (10^{-4} emu/mol)	-3.97	-0.16	-1.23	-3.71
μ_{eff} (μ_B)	2.71	2.59	3.70	3.78
Θ_p (K)	28.7	-41.9	32.3	-21.9

Curie-Weiss law and exhibits a hump around 50 K, as depicted in the insets of Figs. 2(b) and 2(e). Such a hump appears due to the CEF effect as observed in numerous rare-earth compounds such as CeAgAs₂ [4], PrSi [36], and Pr₂Re₃Si₅ [37].

Next, we have analyzed the magnetic susceptibility data using CEF schemes. For a tetragonal system, the $(2J + 1)$ levels of $J = 5/2$ (Ce^{3+}) split into three doublets and the $J = 4$ (Pr^{3+}) multiplet splits into two doublets along with five singlets [38]. The CEF Hamiltonian corresponding to tetragonal site symmetry and C_{4v} point symmetry is as follows,

$$\mathcal{H}_{\text{CEF}} = B_2^0 O_2^0 + B_4^0 O_4^0 + B_4^4 O_4^4 + B_6^0 O_6^0 + B_6^4 O_6^4, \quad (1)$$

where B_p^q and O_p^q are the CEF parameters and the Stevens operators, respectively [39,40]. The magnetic susceptibility based on the CEF scheme, χ_{CEF}^i ($i = x, y, z$), is given by

$$\chi_{\text{CEF}}^i = \frac{N_A (g_J \mu_B)^2}{Z} \left[\sum_n \beta |\langle n | J_i | n \rangle|^2 e^{-\beta E_n} + \sum_{n \neq m} |\langle m | J_i | n \rangle|^2 \frac{e^{-\beta E_n} - e^{-\beta E_m}}{E_m - E_n} \right], \quad (2)$$

where g_J is the Landé factor, $Z = \sum_n e^{-\beta E_n}$, and $\beta = 1/k_B T$. $|n\rangle$ is the n th eigenfunction and E_n is the corresponding eigenvalue. J_i is the component of angular momentum [41]. To analyze the CEF, we have plotted inverse susceptibility as a function of temperature, subtracting the temperature-independent term (χ_0) as shown in Figs. 2(c) and 2(f). The magnetic susceptibility ($\chi_i - \chi_0^i$), including the molecular field contribution λ_i and the CEF contribution, can be ex-

pressed as

$$(\chi_i - \chi_0^i)^{-1} = (\chi_{\text{CEF}}^i)^{-1} - \lambda_i. \quad (3)$$

The calculated temperature-dependent magnetic susceptibilities along different crystallographic directions using the above equation, presented in Figs. 2(c) and 2(f), agree well with the experimental data. The obtained values of CEF parameters and the energy levels of CeGaGe and PrGaGe are given in Tables III and IV, respectively. The estimated parameters and energy levels are comparable to those of other Ce- and Pr-based compounds [42,43]. We have also recalculated the CEF parameter B_2^0 directly from the paramagnetic Curie-Weiss temperatures to check the consistency of our analysis using the following expression [44],

$$\Theta_p^{ab} - \Theta_p^c = \frac{3}{10} B_2^0 (2J - 1)(2J + 3). \quad (4)$$

The calculated value of B_2^0 is -7.3 and -2.4 K for CeGaGe and PrGaGe, respectively, which is in good agreement with the estimated value from the CEF model fit.

Furthermore, we measured the isothermal magnetizations $M(H)$ of CeGaGe and PrGaGe, as shown in Figs. 3(a) and 3(b), respectively. Magnetization along $H \parallel c$ increases sharply and saturates at 0.15 and 0.35 T for CeGaGe and PrGaGe, respectively. The saturated value of magnetization is $1.55 \mu_B$ for CeGaGe and $2.95 \mu_B$ for PrGaGe, which is less than the value of free Ce^{3+} ($g_J J \mu_B = 2.16 \mu_B$) and Pr^{3+} ($g_J J \mu_B = 3.20 \mu_B$) ions. Such low value of saturation magnetization may be due to the presence of the CEF effect [21,23,24], whereas magnetization along $H \parallel ab$ is substantially lower and does not saturate even at 7 T. Such a variation in magnetization in different directions indicates strong magnetocrystalline anisotropy. Moreover, $M(H)$ data also indicate that the c axis is the magnetic easy axis as magnetization saturates easily at low applied field. A small hysteresis is observed in both measured directions in the magnetization data, supporting the FM nature in CeGaGe and PrGaGe. We have also determined the isothermal magnetization considering the CEF scheme using the expression

$$M_i = \frac{g_J \mu_B}{Z} \sum_n |\langle n | J_i | n \rangle| e^{-\beta E_n}. \quad (5)$$

TABLE III. CEF parameters, energy levels, and the corresponding wave functions for CeGaGe.

CEF Parameters						
B_2^0 (K)	B_4^0 (K)	B_4^4 (K)	B_6^0 (K)	B_6^4 (K)		
-6.40 ± 0.20	-0.418 ± 0.035	3.32 ± 0.270	0	0		
$\lambda_z = 10.0 \pm 1.7$ (mol/emu)		$\lambda_{x,y} = 18.7 \pm 1.8$ (mol/emu)				
Energy Levels and Wave Functions						
E (K)	$ +5/2\rangle$	$ +3/2\rangle$	$ +1/2\rangle$	$ -1/2\rangle$	$ -3/2\rangle$	$ -5/2\rangle$
251.4 ± 12.6	-0.38	0	0	0	-0.92	0
251.4 ± 12.6	0	-0.92	0	0	0	-0.38
126.9 ± 7.4	0	0	1	0	0	0
126.9 ± 7.4	0	0	0	1	0	0
0	0.92	0	0	0	-0.38	0
0	0	-0.38	0	0	0	0.92

TABLE IV. CEF parameters, energy levels, and the corresponding wave functions for PrGaGe.

CEF Parameters									
B_2^0 (K)	B_4^0 (K)	B_4^4 (K)	B_6^0 (K)	B_6^4 (K)	λ_i (mol/emu)				
-2.752 ± 0.338	0.050 ± 0.012	0.106 ± 0.019	0 ± 0.0002	0.006 ± 0.0018	$\lambda_z = 8.9 \pm 2.1, \lambda_{x,y} = 7.5 \pm 1.9$				
Energy Levels and Wave Functions									
E (K)	$ +4\rangle$	$ +3\rangle$	$ +2\rangle$	$ +1\rangle$	$ 0\rangle$	$ -1\rangle$	$ -2\rangle$	$ -3\rangle$	$ -4\rangle$
202.8 ± 29.2	-0.177	0	0	0	-0.968	0	0	0	-0.177
158.9 ± 23.0	0	0	0	0.996	0	0	0	0.089	0
158.9 ± 23.0	0	0.089	0	0	0	0.996	0	0	0
77.8 ± 10.5	0	0	$-1/\sqrt{2}$	0	0	0	$-1/\sqrt{2}$	0	0
68.9 ± 10.3	0	0	$-1/\sqrt{2}$	0	0	0	$1/\sqrt{2}$	0	0
48.9 ± 26.1	$-1/\sqrt{2}$	0	0	0	0	0	0	0	$1/\sqrt{2}$
38.6 ± 26.2	0.685	0	0	0	-0.250	0	0	0	0.685
0	0	0	0	0.089	0	0	0	-0.996	0
0	0	-0.996	0	0	0	0.089	0	0	0

The eigenvalue and associated eigenfunction of the above expression can be calculated by diagonalizing the given total Hamiltonian

$$\mathcal{H} = \mathcal{H}_{\text{CEF}} - g_J \mu_B J_i (H + \lambda_i M_i), \quad (6)$$

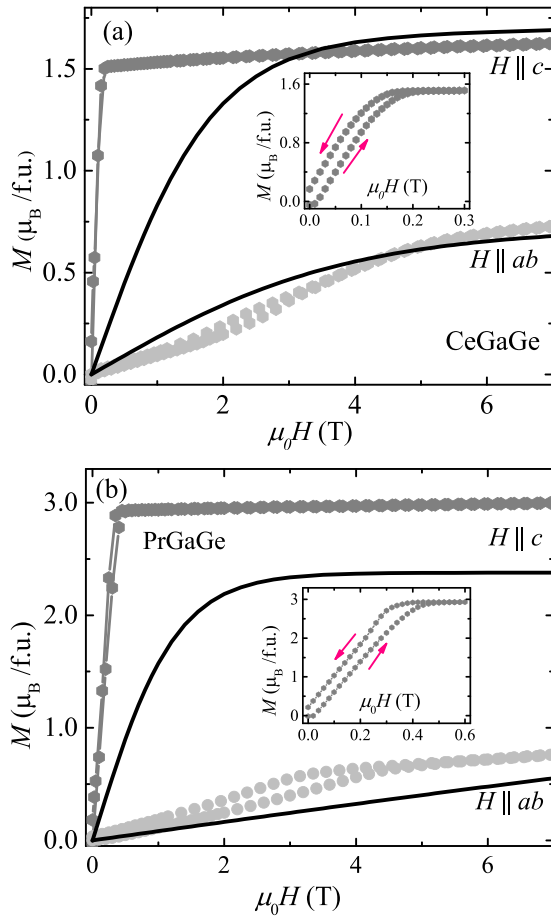


FIG. 3. The isothermal magnetization of (a) CeGaGe and (b) PrGaGe was measured at 1.7 K as a function of the applied magnetic fields. Insets of (a) and (b) show a zoomed view of low-field magnetization for the $H \parallel c$ of CeGaGe and PrGaGe, respectively. The black lines are the result of the CEF calculations.

where the first term of the Hamiltonian is defined in Eq. (1). The second and third terms of the above Hamiltonian are the Zeeman and the molecular field term, respectively. The calculated magnetizations of CeGaGe and PrGaGe are represented in Figs. 3(a) and 3(b) by solid black lines obtained by solving Eqs. (5) and (6). The above calculation roughly reproduces the magnetization as we neglected the exchange interaction in the Hamiltonian. However, this calculation perfectly recognizes the easy and hard axes of magnetization. Furthermore, we have estimated the saturation magnetization (M^{sat}) using the Zeeman term [41,45]

$$\mathcal{H}_{x,z} = g_J \mu_B \langle \Gamma_{\text{mix},1} | J_{x,z} | \Gamma_{\text{mix},1} \rangle B_z, \quad (7)$$

where $\Gamma_{\text{mix},1}$ is the ground state wave function. The estimated and observed value of saturation magnetization is presented in Table V. A slight discrepancy between the computed and observed numbers is due to the use of the simplified CEF model [37].

C. Specific heat

The temperature-dependent heat capacity (C_p) of single-crystalline $R\text{GaGe}$ ($R = \text{La}-\text{Pr}$) at a constant pressure is presented in Figs. 4(a)–4(c), respectively. A sharp anomaly at 6.0 and 19.4 K for CeGaGe and PrGaGe, respectively, in the C_p data confirms the bulk magnetic ordering as observed in the magnetic susceptibility data. The heat capacity reaches an expected value of Dulong-Petit limits $C_p = 3nR = 74.83$ J/mol K for all three compounds, where n is the number of atoms in formula units and R is the universal gas constant. The $C_p(T)$ of the nonmagnetic LaGaGe can be expressed well

TABLE V. The calculated saturation magnetizations using Eq. (7) and observed value at 1.7 K and 7 T.

	CeGaGe		PrGaGe	
	Estimated	Observed	Estimated	Observed
$M_{\parallel c}^{\text{sat}} (\mu_B)$	1.64	1.63	2.38	2.96
$M_{\perp c}^{\text{sat}} (\mu_B)$	0.68	0.72	0	0.74

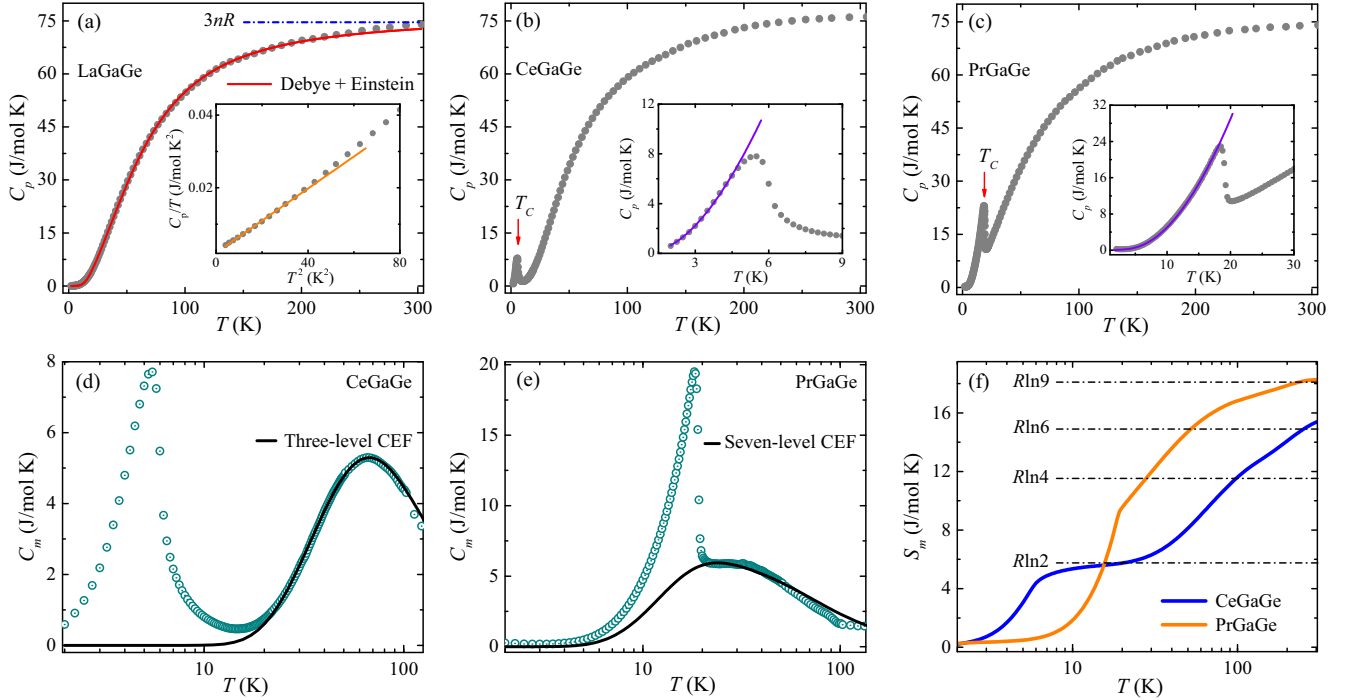


FIG. 4. The temperature-dependent heat capacity of single crystals (a) LaGaGe, (b) CeGaGe, and (c) PrGaGe. The red solid line in (a) represents the fitting of the Debye plus Einstein model. The inset of (a) shows linear fitting to C_p/T vs T^2 at low temperatures. The purple solid line in the insets of (b) and (c) is the fitting of Eq. (11) to the C_p data below T_c . The temperature dependence of magnetic contribution to C_p of (d) CeGaGe and (e) PrGaGe. The solid black line shows the fitting of $C_m(T)$ data using different levels CEF scheme. (f) The magnetic entropy of CeGaGe and PrGaGe calculated as a function of temperature.

by considering the Debye (C_D) and Einstein (C_E) modes of heat capacity as presented by the formula

$$C_p(T) = \gamma T + mC_D(T) + (1 - m)C_E(T), \quad (8)$$

where m is the weight factor and γ is the Sommerfeld coefficient. $C_D(T)$ and $C_E(T)$ are defined as

$$C_D(T) = 9nR \left(\frac{T}{\Theta_D} \right)^3 \int_0^{\Theta_D/T} \frac{x^4 e^x}{(e^x - 1)^2} dx \quad (9)$$

and

$$C_E(T) = 3nR \left(\frac{\Theta_E}{T} \right)^2 \frac{e^{\Theta_E/T}}{(e^{\Theta_E/T} - 1)^2}, \quad (10)$$

where Θ_D and Θ_E are the Debye and Einstein temperatures, respectively [46,47]. The obtained fitting parameters are $\Theta_D = 295$ K, $\Theta_E = 100$ K, and $m = 0.76$. We have estimated the γ from the C_p data of LaGaGe by applying the formula $C_p/T = \gamma + \beta T^2$ in the low-temperature regime ($2 \text{ K} \leq T \leq 7 \text{ K}$). The linear fit of the C_p/T vs T^2 curve as shown in the inset of Fig. 4(a) gives $\gamma = 2(2) \text{ mJ/mol K}^2$ and $\beta = 0.44(1) \text{ mJ/mol K}^4$. The estimated value γ is comparable to the previous report [32]. Furthermore, to estimate γ for CeGaGe and PrGaGe, we have fitted the C_p data in the magnetically ordered state as presented in the inset of Figs. 4(b) and 4(c) using the following formula,

$$C_p(T) = \gamma T + \beta T^3 + \delta T^{3/2} e^{-\Delta/T}. \quad (11)$$

The last term of the above equation is the contribution to the C_p due to the spin wave for a ferromagnet with an energy gap

Δ in the magnon spectrum [48]. To perform the above fitting we have used the same β obtained from LaGaGe, which reduces the fitting parameter. The so-obtained fitting parameters are listed in Table VI. The estimated value of γ for CeGaGe and PrGaGe is comparable to that of other FM compounds such as CeIr_2B_2 [49], $\text{Pr}_2\text{Rh}_2\text{Ga}$ [50], and $\text{Ce}_{11}\text{Pd}_4\text{In}_9$ [51]. However, it is less than that of typical Kondo/heavy-fermion compounds like CeIn_3 [52] and $\text{PrV}_2\text{Al}_{20}$ [53].

The magnetic contribution to the C_p for CeGaGe and PrGaGe was calculated by subtracting the C_p of the non-magnetic reference LaGaGe, assuming that the phonon contribution in CeGaGe and PrGaGe is approximately equal to that of LaGaGe. The temperature dependence of magnetic heat capacity [$C_m(T)$] is presented in Figs. 4(d) and 4(e). In addition to the prominent peak at low temperatures due to the magnetic phase transition, a broad hump around 60 and 30 K for CeGaGe and PrGaGe is seen in $C_m(T)$, respectively. The broad peak can be attributed to a Schottky-type anomaly, which arises from the CEF-induced splitting of energy levels Ce^{3+} and Pr^{3+} . The generalized formula of the Schottky contribution to the heat capacity for the multilevel CEF scheme is

TABLE VI. Obtained fitting parameters of low-temperature C_p data of CeGaGe and PrGaGe using Eq. (11).

	γ (mJ/mol K ²)	δ (J/mol K ^{5/2})	Δ (K)
CeGaGe	13	1.54	3.90
PrGaGe	35	0.58	14.42

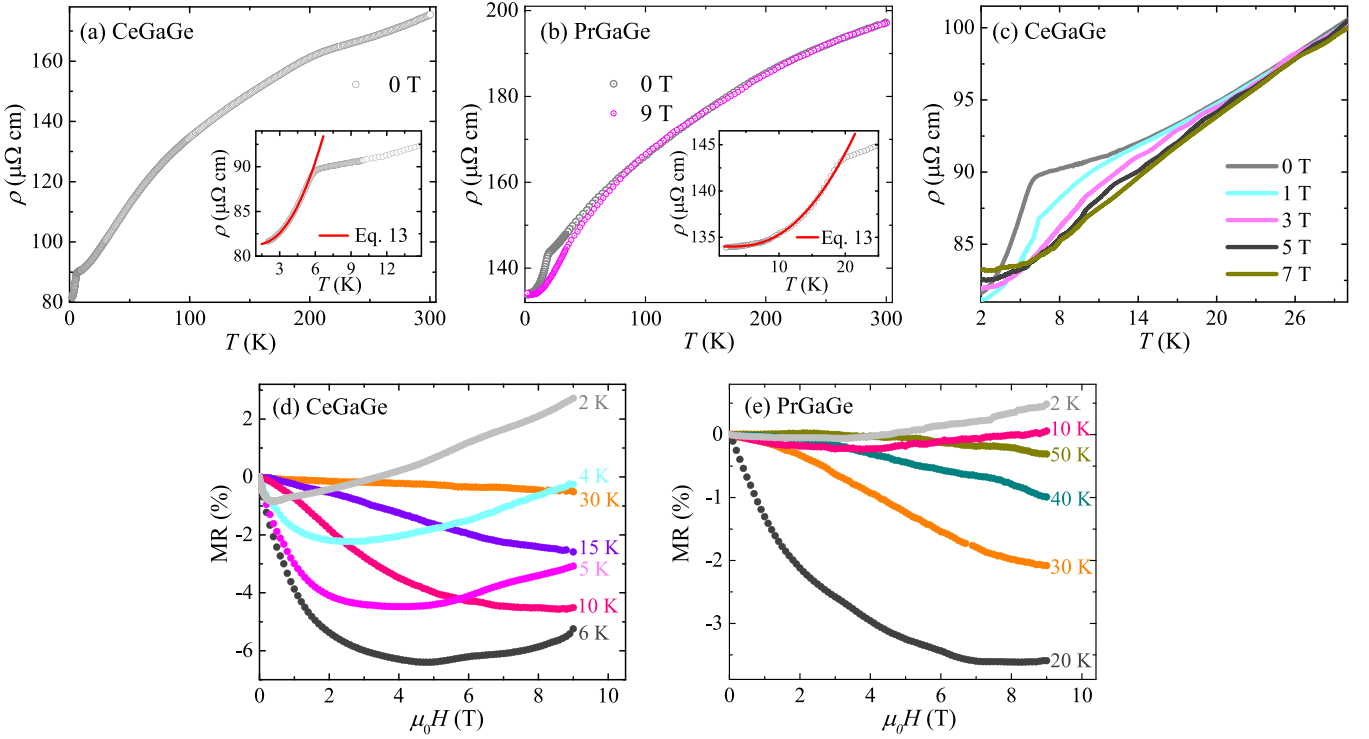


FIG. 5. Temperature-dependent electrical resistivity of (a) CeGaGe and (b) PrGaGe measured in between 2–300 K. The insets show the fitting of Eq. (13) in the magnetically ordered state. (c) The temperature dependence of electrical resistivity of CeGaGe measured under external magnetic fields. The MR of (d) CeGaGe and (e) PrGaGe as a function of the magnetic field for different temperatures.

given by

$$C_{\text{Sch}}(T) = \left(\frac{R}{T^2} \right) \left[\sum_i g_i e^{-\Delta_i/T} \sum_i g_i \Delta_i^2 e^{-\Delta_i/T} - \left(\sum_i g_i \Delta_i e^{-\Delta_i/T} \right)^2 \right] \left(\sum_i g_i e^{-\Delta_i/T} \right)^{-2}, \quad (12)$$

where g_i is the degeneracy of the i th state with Δ_i energy gap splitting [48]. Here, we have used the same CEF parameters obtained from CEF analysis of the magnetic susceptibility data to reproduce the Schottky anomaly observed in $C_m(T)$. Interestingly, the calculated Schottky anomaly matches well with the experimental data for both compounds, as presented in Figs. 4(d) and 4(e). This justifies the validity of the CEF parameters and the energy level splittings of both compounds.

Next, we have calculated the magnetic entropy using the expression $S_m(T) = \int \frac{C_m}{T} dT$. The magnetic entropy as a function of temperature for CeGaGe and PrGaGe is presented in Fig. 4(f). According to CEF analysis, six energy levels in CeGaGe split into three doublets, resulting in three plateaus in $S_m(T)$ near $R\ln 2$, $R\ln 4$, and $R\ln 6$. In the case of PrGaGe, the $S_m(T)$ reaches a value of 7.9 J/mol K at T_C , which lies between a value of magnetic entropy of ground state $R\ln 2$ and $R\ln 4$. Therefore, a strong thermal population of the excited levels is indicated, and the first excited level is probably slightly higher in energy than T_C . The total magnetic entropy $R\ln 6$ for CeGaGe and $R\ln 9$ for PrGaGe is released at 252 and 210 K, respectively, indicating the overall splitting of CEF levels within this energy window. The observed $S_m(T)$ is consistent with our CEF analysis.

D. Magnetotransport

The electrical resistivity $\rho(T)$ measured within the ab plane in CaGaGe and PrGaGe single crystals in the temperature range of 2–300 K is shown in Figs. 5(a) and 5(b), respectively. The $\rho(T)$ of CeGaGe and PrGaGe show metallic behavior down to 2 K with a sharp anomaly around 6.0 and 19.4 K, respectively. The low-temperature anomaly is associated with the magnetic phase transition where resistivity drops rapidly due to reduction of spin scattering as systems order magnetically. The residual resistivity ratio (RRR) of CeGaGe and PrGaGe is ~ 2.1 and ~ 1.5 , respectively, which is small compared to typical metals. Such value of RRR may appear due to site disorder between Ga and Ge. Similar RRR has also been observed in the $RA1X$ ($R = \text{La-Nd}$ and $X = \text{Ge, Si}$) family for site disorder between Al and Ge/Si [22–24]. On the other hand, $\rho(T)$ of CeGaGe exhibits a broad hump from 220 to 50 K, which may arise from the CEF splitting, which is also evident from thermodynamic measurement. Similarly, a concave curvature is observed in the $\rho(T)$ data of PrGaGe. Such behavior of $\rho(T)$ is commonly seen in rare-earth compounds like CeAuGa₃ [54], Ce₂NiSi₃ [7], and Pr₇Ru₃ [55]. For both compounds, the $\rho(T)$ in the magnetically ordered state can be described using the following expression,

$$\rho(T) = \rho_0 + AT^2 \exp\left(-\frac{\Delta}{T}\right), \quad (13)$$

where ρ_0 is the residual resistivity, and the second term is the contribution to resistivity due to the FM magnon. A is related to the strength of electron-magnon scattering, and Δ is the energy gap in the magnon spectrum [56]. The so-obtained

TABLE VII. The estimated fitting parameters ρ_0 , A , and Δ from the resistivity data using Eq. (13).

	ρ_0 ($\mu\Omega$ cm)	A ($\mu\Omega$ cm/ K^2)	Δ (K)
CeGaGe	81.3	0.45	3.33
PrGaGe	134.0	0.05	13.36

values of fitting parameters from Eq. (13) are listed in Table VII. The estimated value of Δ is very close to that obtained from the heat capacity data, confirming the consistency of our analysis.

The $\rho(T)$ of CeGaGe and PrGaGe under various external magnetic fields is presented in Figs. 5(b) and 5(c). The low-temperature anomaly in $\rho(T)$ due to magnetic ordering for both compounds is getting smeared out by the applied field, and the magnitude of electrical resistivity is reduced near T_C . The field dependence of the magnetoresistance (MR) of CeGaGe and PrGaGe at various temperatures for $H \parallel c$ is represented in Figs. 5(d) and 5(e), respectively. The MR is defined as $[\rho(H) - \rho(0)]/\rho(0)$, where $\rho(H)$ and $\rho(0)$ are the resistivities in the presence and absence of a magnetic field, respectively. The MR of both compounds becomes negative close to the magnetic ordering temperature and remains negative throughout the paramagnetic region. The maximum value of MR reaches about -6.5% and -3.8% for CeGaGe and PrGaGe, respectively. The negative MR for both samples near T_C results from the suppression of spin scattering by the applied magnetic field, which is the typical MR behavior for a FM [57,58]. Interestingly, at low temperatures ($T < T_C$), the MR for both samples in a weak-field regime is negative, but as the field intensity increases, the MR rises and becomes positive. The magnetic moments become polarized at low applied fields (~ 0.4 T), and MR becomes negative due to suppression of spin disorder; however, as the field strength increases, the

Lorentz force becomes dominant, and MR starts to increase, as observed in PrAlGe [24] and CeIr₂B₂ [49].

IV. SUMMARY

We have studied the magnetic, thermodynamic, and magnetotransport properties of RGe (R = Ce and Pr) single crystals synthesized using gallium-indium flux. The powder XRD patterns indicated that these crystals crystallize in a body-centered tetragonal structure with space group $I4_1md$ (No. 109). The temperature-dependent magnetic susceptibility and resistivity measurements show a magnetic phase transition in CeGaGe and PrGaGe at $T_C = 6.0$ and 19.4 K, respectively. Further, heat capacity data show a sharp peak at the magnetic transition, which confirms the bulk nature of the magnetic ordering. The temperature and field-dependent magnetization data reveal strong magnetic anisotropy and the presence of CEF in both compounds. The CEF analysis for the inverse magnetic susceptibility and heat capacity data of CeGaGe and PrGaGe reveals that the $J = 5/2$ degenerate ground state of the Ce³⁺ ion splits into three doublets, whereas the $J = 4$ multiplet of the Pr³⁺ ion splits into two doublets and five singlets. The field-dependent MR for both compounds is positive at low temperatures and high fields, it switches to negative near the magnetic transition temperature, and it remains negative in the paramagnetic region.

ACKNOWLEDGMENTS

We acknowledge IIT Kanpur and the Department of Science and Technology, India [Order No. DST/NM/TUE/QM-06/2019 (G)], for financial support. We thank Suman Sanki for valuable discussion. D.K. acknowledges financial support from the National Science Centre (Poland) under Research Grant No. 2021/41/B/ST3/01141.

- [1] L. Wang, C. Wang, Z. Liu, J. Cheng, S. Miao, Y. Song, Y. Shi, and Y.-f. Yang, *Phys. Rev. B* **100**, 085122 (2019).
- [2] D. Das, D. Gnida, Ł. Bochenek, A. Rudenko, M. Daszkiewicz, and D. Kaczorowski, *Sci. Rep.* **8**, 16703 (2018).
- [3] F. Steglich, J. Aarts, C. D. Bredl, W. Lieke, D. Meschede, W. Franz, and H. Schäfer, *Phys. Rev. Lett.* **43**, 1892 (1979).
- [4] R. Mondal, R. Bapat, S. K. Dhar, and A. Thamizhavel, *Phys. Rev. B* **98**, 115160 (2018).
- [5] V. K. Anand, Z. Hossain, G. Behr, G. Chen, M. Nicklas, and C. Geibel, *J. Phys.: Condens. Matter* **19**, 506205 (2007).
- [6] N. D. Mathur, F. M. Grosche, S. R. Julian, I. R. Walker, D. M. Freye, R. K. W. Haselwimmer, and G. G. Lonzarich, *Nature (London)* **394**, 39 (1998).
- [7] S. Malick, D. Das, and Z. Hossain, *J. Magn. Magn. Mater.* **482**, 108 (2019).
- [8] P. Opletal, E. Duverger-Nédellec, K. Miliyanchuk, S. Malick, Z. Hossain, and J. Custers, *J. Alloys Compd.* **927**, 166941 (2022).
- [9] H.-H. Lai, S. E. Grefe, S. Paschen, and Q. Si, *Proc. Natl. Acad. Sci. U.S.A.* **115**, 93 (2018).
- [10] S. Dzsaber, L. Prochaska, A. Sidorenko, G. Eguchi, R. Svagera, M. Waas, A. Prokofiev, Q. Si, and S. Paschen, *Phys. Rev. Lett.* **118**, 246601 (2017).
- [11] D. J. Campbell, Z. E. Brubaker, C. Roncaioli, P. Saraf, Y. Xiao, P. Chow, C. Kenney-Benson, D. Popov, R. J. Zieve, J. R. Jeffries, and J. Paglione, *Phys. Rev. B* **100**, 235133 (2019).
- [12] M. Hirschberger, S. Kushwaha, Z. Wang, Q. Gibson, S. Liang, C. A. Belvin, B. A. Bernevig, R. J. Cava, and N. P. Ong, *Nat. Mater.* **15**, 1161 (2016).
- [13] R. Singha, S. Roy, A. Pariari, B. Satpati, and P. Mandal, *Phys. Rev. B* **99**, 035110 (2019).
- [14] S. Wolgast, C. Kurdak, K. Sun, J. W. Allen, D.-J. Kim, and Z. Fisk, *Phys. Rev. B* **88**, 180405(R) (2013).
- [15] E. Cheng, W. Xia, X. Shi, H. Fang, C. Wang, C. Xi, S. Xu, D. C. Peets, L. Wang, H. Su, L. Pi, W. Ren, X. Wang, N. Yu, Y. Chen, W. Zhao, Z. Liu, Y. Guo, and S. Li, *Nat. Commun.* **12**, 6970 (2021).
- [16] H. Su, X. Shi, J. Yuan, Y. Wan, E. Cheng, C. Xi, L. Pi, X. Wang, Z. Zou, N. Yu, W. Zhao, S. Li, and Y. Guo, *Phys. Rev. B* **103**, 165128 (2021).
- [17] S.-Y. Xu, N. Alidoust, G. Chang, H. Lu, B. Singh, I. Belopolski, D. S. Sanchez, X. Zhang, G. Bian, H. Zheng, M.-A. Husanu, Y. Bian, S.-M. Huang, C.-H. Hsu, T.-R. Chang, H.-T. Jeng, A. Bansil, T. Neupert, V. N. Strocov, H. Lin *et al.*, *Sci. Adv.* **3**, e1603266 (2017).

- [18] W. Cao, Y. Su, Q. Wang, C. Pei, L. Gao, Y. Zhao, C. Li, N. Yu, J. Wang, Z. Liu, Y. Chen, G. Li, J. Li, and Y. Qi, *Chin. Phys. Lett.* **39**, 047501 (2022).
- [19] J. Zhao, W. Liu, A. Rahman, F. Meng, L. Ling, C. Xi, W. Tong, Y. Bai, Z. Tian, Y. Zhong, Y. Hu, L. Pi, L. Zhang, and Y. Zhang, *New J. Phys.* **24**, 013010 (2022).
- [20] P. Puphal, V. Pomjakushin, N. Kanazawa, V. Ukleev, D. J. Gawryluk, J. Ma, M. Naamneh, N. C. Plumb, L. Keller, R. Cubitt, E. Pomjakushina, and J. S. White, *Phys. Rev. Lett.* **124**, 017202 (2020).
- [21] D. Destraz, L. Das, S. S. Tsirkin, Y. Xu, T. Neupert, J. Chang, A. Schilling, A. G. Grushin, J. Kohlbrecher, L. Keller, P. Puphal, E. Pomjakushina, and J. S. White, *npj Quantum Mater.* **5**, 5 (2020).
- [22] H. Hodovanets, C. J. Eckberg, P. Y. Zavalij, H. Kim, W.-C. Lin, M. Zic, D. J. Campbell, J. S. Higgins, and J. Paglione, *Phys. Rev. B* **98**, 245132 (2018).
- [23] H.-Y. Yang, B. Singh, J. Gaudet, B. Lu, C.-Y. Huang, W.-C. Chiu, S.-M. Huang, B. Wang, F. Bahrami, B. Xu, J. Franklin, I. Sochnikov, D. E. Graf, G. Xu, Y. Zhao, C. M. Hoffman, H. Lin, D. H. Torchinsky, C. L. Broholm, A. Bansil *et al.*, *Phys. Rev. B* **103**, 115143 (2021).
- [24] B. Meng, H. Wu, Y. Qiu, C. Wang, Y. Liu, Z. Xia, S. Yuan, H. Chang, and Z. Tian, *APL Mater.* **7**, 051110 (2019).
- [25] M. Lyu, J. Xiang, Z. Mi, H. Zhao, Z. Wang, E. Liu, G. Chen, Z. Ren, G. Li, and P. Sun, *Phys. Rev. B* **102**, 085143 (2020).
- [26] J.-F. Wang, Q.-X. Dong, Z.-P. Guo, M. Lv, Y.-F. Huang, J.-S. Xiang, Z.-A. Ren, Z.-J. Wang, P.-J. Sun, G. Li, and G.-F. Chen, *Phys. Rev. B* **105**, 144435 (2022).
- [27] H.-Y. Yang, B. Singh, B. Lu, C.-Y. Huang, F. Bahrami, W.-C. Chiu, D. Graf, S.-M. Huang, B. Wang, H. Lin, D. Torchinsky, A. Bansil, and F. Tafti, *APL Mater.* **8**, 011111 (2020).
- [28] H.-Y. Yang, J. Gaudet, R. Verma, S. Baidya, F. Bahrami, X. Yao, C.-Y. Huang, L. DeBeer-Schmitt, A. A. Aczel, G. Xu, H. Lin, A. Bansil, B. Singh, and F. Tafti, *Phys. Rev. Mater.* **7**, 034202 (2023).
- [29] X. He, Y. Li, H. Zeng, Z. Zhu, S. Tan, Y. Zhang, C. Cao, and Y. Luo, *Sci. China Phys., Mech. Astron.* **66**, 237011 (2023).
- [30] J. Gaudet, H.-Y. Yang, S. Baidya, B. Lu, G. Xu, Y. Zhao, J. A. Rodriguez-Rivera, C. M. Hoffmann, D. E. Graf, D. H. Torchinsky, P. Nikolić, D. Vanderbilt, F. Tafti, and C. L. Broholm, *Nat. Mater.* **20**, 1650 (2021).
- [31] C. Dhital, R. L. Dally, R. Ruvalcaba, R. Gonzalez-Hernandez, J. Guerrero-Sanchez, H. B. Cao, Q. Zhang, W. Tian, Y. Wu, M. D. Frontzek, S. K. Karna, A. Meads, B. Wilson, R. Chapai, D. Graf, J. Baesa, R. Jin, and J. F. DiTusa, *Phys. Rev. B* **107**, 224414 (2023).
- [32] S. Dhar, S. Patalwar, and R. Vijayaraghavan, *Phys. B: Condens. Matter* **186-188**, 491 (1993).
- [33] N. Patil and S. Dhar, *Phys. B: Condens. Matter* **223-224**, 359 (1996).
- [34] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevB.108.024428> for EDS data and XRD refinement.
- [35] P. Puphal, C. Mielke, N. Kumar, Y. Soh, T. Shang, M. Medarde, J. S. White, and E. Pomjakushina, *Phys. Rev. Mater.* **3**, 024204 (2019).
- [36] P. K. Das, A. Bhattacharyya, R. Kulkarni, S. K. Dhar, and A. Thamizhavel, *Phys. Rev. B* **89**, 134418 (2014).
- [37] S. Sanki, V. Sharma, S. Sasmal, V. Saini, G. Dwari, B. B. Maity, R. Kulkarni, R. Prakash Pandeya, R. Mondal, A. Lakshan, S. Ramakrishnan, P. Pratim Jana, K. Maiti, and A. Thamizhavel, *Phys. Rev. B* **105**, 165134 (2022).
- [38] U. Walter, *J. Phys. Chem. Solids* **45**, 401 (1984).
- [39] K. W. H. Stevens, *Proc. Phys. Soc., Sect. A* **65**, 209 (1952).
- [40] M. Hutchings, *Solid State Phys.* **16**, 227 (1964).
- [41] J. Banda, B. K. Rai, H. Rosner, E. Morosan, C. Geibel, and M. Brando, *Phys. Rev. B* **98**, 195120 (2018).
- [42] N. Van Hieu, T. Takeuchi, H. Shishido, C. Tonohiro, T. Yamada, H. Nakashima, K. Sugiyama, R. Settai, T. D. Matsuda, Y. Haga, M. Hagiwara, K. Kindo, S. Araki, Y. Nozue, and Y. Ōnuki, *J. Phys. Soc. Jpn.* **76**, 064702 (2007).
- [43] Y. Nakano, F. Honda, T. Takeuchi, K. Sugiyama, M. Hagiwara, K. Kindo, E. Yamamoto, Y. Haga, R. Settai, H. Yamagami, and Y. Ōnuki, *J. Phys. Soc. Jpn.* **79**, 024702 (2010).
- [44] Y.-L. Wang, *Phys. Lett. A* **35**, 383 (1971).
- [45] H. Jin, W. Cai, J. Coles, J. R. Badger, P. Klavins, S. Deemyad, and V. Taufour, *Phys. Rev. B* **106**, 075131 (2022).
- [46] S. Malick, J. Singh, A. Laha, V. Kanchana, Z. Hossain, and D. Kaczorowski, *Phys. Rev. B* **105**, 045103 (2022).
- [47] D. Ram, J. Singh, M. K. Hooda, O. Pavlosiuk, V. Kanchana, Z. Hossain, and D. Kaczorowski, *Phys. Rev. B* **107**, 085137 (2023).
- [48] E. S. R. Gopal, *Specific Heats at Low Temperatures* (Plenum, New York, 1966).
- [49] A. Prasad, V. K. Anand, U. B. Paramanik, Z. Hossain, R. Sarkar, N. Oeschler, M. Baenitz, and C. Geibel, *Phys. Rev. B* **86**, 014414 (2012).
- [50] B. Sahu, [arXiv:2102.04106](https://arxiv.org/abs/2102.04106).
- [51] D. Das and D. Kaczorowski, *J. Magn. Magn. Mater.* **471**, 315 (2019).
- [52] H. Shishido, T. Shibauchi, K. Yasu, T. Kato, H. Kontani, T. Terashima, and Y. Matsuda, *Science* **327**, 980 (2010).
- [53] M. Tsujimoto, Y. Matsumoto, T. Tomita, A. Sakai, and S. Nakatsuji, *Phys. Rev. Lett.* **113**, 267001 (2014).
- [54] B. Lv and B. Liu, *Nucl. Anal.* **1**, 100002 (2022).
- [55] S. Krolak, H. Swiatek, K. Gornicka, M. Winiarski, W. Xie, R. Cava, and T. Klimczuk, *J. Alloys Compd.* **929**, 167279 (2022).
- [56] D. Kaczorowski and A. Szytuła, *J. Alloys Compd.* **614**, 186 (2014).
- [57] V. K. Anand, D. T. Adroja, and A. D. Hillier, *Phys. Rev. B* **85**, 014418 (2012).
- [58] S. Nallamuthu, A. Dzubinska, M. Reiffers, J. R. Fernandez, and R. Nagalakshmi, *Phys. B: Condens. Matter* **521**, 128 (2017).