








Disentanglement of intrinsic and extrinsic side-jump scattering induced spin Hall effect in N-implanted Pt

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The rapidly evolving utilization of spin Hall effect (SHE) arising from spin-orbit coupling in $5d$ transition metals and alloys has made giant strides in the development of designing low-power, robust, and nonvolatile magnetic memory. Recent studies, on incorporating nonmetallic lighter elements such as oxygen, nitrogen, and sulfur into $5d$ transition metals, have shown an enhancement in dampinglike torque efficiency θ_{DL} due to the modified SHE, but the mechanism behind this enhancement is not clear. In this paper, we study θ_{DL} at different temperatures (100–293 K) to disentangle the intrinsic and extrinsic side-jump scattering induced SHE in N-implanted Pt. We observe a crossover of intrinsic to extrinsic side-jump mechanism as the implantation dose increases from 2×10^{16} to 1×10^{17} ions/cm². A sudden decrease in the intrinsic spin Hall conductivity is counterbalanced by the increase in the extrinsic side-jump induced SHE efficiency. These results conclude that studying θ_{DL} as a function of implantation dose, and as a function of temperature, is important to understand the physical mechanism contributing to SHE.

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I. INTRODUCTION

Spin Hall effect (SHE) [1–4] has been garnering much attention in the development of spin-orbit torque magnetic random-access memory (SOT-MRAM) [5–8] due to its low power consumption and efficient magnetization switching. A charge current density j_c in a heavy metal (HM) is converted into a spin current density j_s via SHE, which then exerts an in-plane dampinglike spin-orbit torque (SOT) τ_{DL} on the magnetization of the adjacent ferromagnet (FM) [9,10]. The ratio of j_s to j_c is called the dampinglike torque (DLT) efficiency, θ_{DL} (also termed as the charge-to-spin conversion efficiency or spin Hall angle). With the strength of spin-orbit coupling (SOC) depending on the atomic number Z , $5d$ transition metals such as Pt, Ta, and W [9,11–15] have been improved by alloying with other heavy metals with large Z such as Au and Pd [16–18]. Alternatively, nonmetallic elements (impurities with smaller Z) have been incorporated into $5d$ transition metals (hosts with larger Z). The difference in Z between the host and the impurity has been found to result in an enhancement of SHE [19]. θ_{DL} has been enhanced by incorporating nonmetallic elements into $5d$ transition metals such as sulfur (S) in Pt [20], oxygen (O) in Pt [21–24], Ta [25],

and W [26], and nitrogen (N) in Pt [27], Ta [28], and W [29]. However, the effects of nitrogen (N) incorporation are still underexplored, especially in controlling the longitudinal resistivity ρ_{xx} , which is an important yardstick to be considered for SOT-MRAM applications. Incorporation of nitrogen has led to an undesirable increase in ρ_{xx} of Ta(N) and W(N) [28,29]. Furthermore, the origin responsible for the enhancement in SHE in nonmetallic doped Pt, Ta, and W has not yet been confirmed explicitly [25,27–29]. Studies on the temperature dependence of the SHE in $5d$ transition metals having nonmetallic impurities are scarce. Most room temperature studies have reported a limited variation of θ_{DL} with ρ_{xx} .

There have been extensive efforts to enhance the SHE, mainly via two mechanisms: intrinsic and extrinsic SHE [30–34]. The intrinsic SHE depends on the Berry curvature of the material, in which an anomalous velocity arises from a momentum-space Berry phase [31]. It leads to an elastic event in which the wave vector \vec{k} of the up-spin and down-spin electrons generated from charge current is conserved [32,33], and is typically seen in $4d$ and $5d$ transition metals. The extrinsic SHE arises when impurities are introduced in the HM and can be further classified into side jump and skew scattering [34]. For side-jump scattering, a discontinuous sideways displacement is created near the impurities for the up-spin and down-spin electrons generated from charge current, leading to an elastic event due to the cancellation of \vec{k} , and is found in

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materials with a high amount of impurity concentrations in the host. The skew scattering, however, is different from the intrinsic and side-jump mechanism as scattering bends or skews the trajectories of up-spin and down-spin electrons in different directions and is found in superclean or low resistivity materials [32]. This leads to a condition where \vec{k} is not conserved, resulting in an inelastic event. Given the separation of SHE based on elastic and inelastic events, a strong correlation between spin Hall conductivity, $\sigma_{\text{SH}}^{\text{xy}}$ and momentum relaxation time T_{relax} can be obtained. Intrinsic SHE and side-jump scattering share the same scaling, with $\sigma_{\text{SH}}^{\text{xy}}$ being independent of T_{relax} . Skew scattering shows $\sigma_{\text{SH}}^{\text{xy}} \propto T_{\text{relax}}$ scaling. Therefore, it is hard to disentangle the contribution of the intrinsic from the side-jump scattering. Despite tremendous efforts [18,27–29], a clear separation between the contributions of intrinsic and side-jump scattering to the SHE in nonmetallic element doped HM has eluded us so far. The limited variation of θ_{DL} with resistivity ρ_{xx} at room temperature and the choice of host/impurity combinations, especially for incorporation of nonmetallic elements in the HM, need to be addressed.

In this paper, we present a successful disentanglement of intrinsic and extrinsic side-jump scattering by studying the SHE using a nonmetallic nitrogen (N) implanted in Pt, at 100–293 K using spin-torque ferromagnetic resonance (ST-FMR) line-shape (spectral) analysis. We observe a crossover from an intrinsic to extrinsic side-jump scattering mechanism as the N-ion dose increases from 2×10^{16} to 1×10^{17} ions/cm². This happens due to a decrease in intrinsic spin Hall conductivity, $\sigma_{\text{SH}}^{\text{int}}$, which is counterbalanced by an increase in side-jump induced SHE efficiency, $\theta_{\text{SH}}^{\text{sj}}$. Our results offer an interesting opportunity to understand the underlying SHE mechanism for incorporation of nonmetallic elements in HM, by use of a less explored approach of ion implantation.

II. EXPERIMENTAL DETAILS

Thin films of Pt (10 nm)/MgO (10 nm)/Al₂O₃ (10 nm) layers were deposited on a Si/SiO₂ substrate at room temperature using an ultrahigh vacuum sputtering. The thin film stacks were implanted sequentially with doses of 2×10^{16} , 5×10^{16} , and 1×10^{17} ions/cm² by an N-ion source beam having an energy of 20 keV. After ion implantation, the capping layers of MgO and Al₂O₃ were removed by Ar⁺ ion milling and then an FM layer of NiFe (5 nm) was sputtered on these samples (see Appendix A). Hereafter, they will be referred to as Pt(N) 2×10^{16} , Pt(N) 5×10^{16} , and Pt(N) 1×10^{17} . All three bilayer samples were then patterned into rectangular microstrips using photolithography. Thereafter, Ti (10 nm)/Al (200 nm) electrodes were deposited. The design of the coplanar waveguides for ST-FMR measurements is shown in Fig. 1(a). The temperature dependent ST-FMR measurements were performed in the range 100–293 K.

III. RESULTS AND DISCUSSION

A. Dose-dependent ST-FMR measurements at room temperature (293 K)

To study the influence of dose on SHE in Pt(N), we first performed the ST-FMR based line-shape analysis to determine θ_{DL} . Figure 1(a) shows the schematic of the ST-FMR measurement setup. In this technique, when a microwave

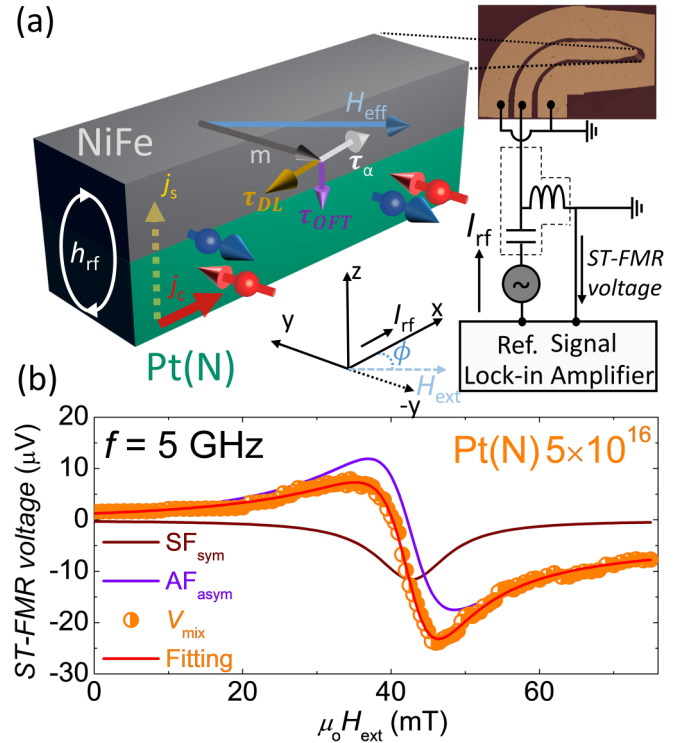


FIG. 1. ST-FMR measurement setup and line shape: (a) Schematic showing ST-FMR measurement technique and detection principle for a bilayer thin film, along with an optical image of the microdevice. (b) ST-FMR spectra (V_{mix}) for $f = 5$ GHz obtained for Pt(N) 5×10^{16} , fitted using Eq. (1). Deconvolution fitting of V_{mix} into symmetric and antisymmetric components displayed by brown and violet solid lines, respectively.

current I_{rf} flows in the longitudinal direction of the HM/FM bilayer, a transverse spin current density j_s is generated, which exerts an in-plane DLT, τ_{DL} , on the local magnetization of the FM. So, an I_{rf} was passed in the longitudinal direction (along x-axis) of the HM/FM bilayer with an applied power of 10 dBm. The I_{rf} generates a rf Oersted field h_{rf} (along y-axis), which simultaneously exerts an Oersted field torque τ_{OFT} . An external magnetic field $\mu_o H_{\text{ext}}$ was swept in the range ± 240 mT at an angle of $\phi = 45^\circ$ with respect to the longitudinal direction of the HM/FM bilayer. At resonance condition, both τ_{DL} and τ_{OFT} drive the magnetization precession in the FM, which results in a periodically varying resistance ΔR due to the anisotropic magnetoresistance (AMR) of NiFe. The mixing of the oscillating ΔR and I_{rf} produces a ST-FMR voltage, which is detected using a lock-in amplifier via a bias tee. The I_{rf} is amplitude modulated by a low frequency sinusoidal wave signal which serves as a trigger since it is provided as a reference signal into the reference port of lock-in amplifier. The ST-FMR spectrum here is the low frequency voltage with the I_{rf} being amplitude modulated, and can be detected using the phase locking technique. This ST-FMR (spectrum) voltage is expressed as [9,20,25,26]

$$V_{\text{mix}} = SF_{\text{sym}}(H_{\text{ext}}) + AF_{\text{asym}}(H_{\text{ext}}), \quad (1)$$

where $F_{\text{sym}}(H_{\text{ext}}) = \frac{(\Delta H)^2}{(H_{\text{ext}} - H_0)^2 + (\Delta H)^2}$, is the symmetric part of the V_{mix} spectrum, $F_{\text{asym}}(H_{\text{ext}}) = \frac{\Delta H(H_{\text{ext}} - H_0)}{(H_{\text{ext}} - H_0)^2 + (\Delta H)^2}$ is the

antisymmetric part, ΔH and H_0 are the half width at half maximum (linewidth) and the resonance field, and S and A are the weight factors of the symmetric and antisymmetric spectra, respectively. For the observed spectra, while the symmetric component is dominated by the τ_{DL} contribution (from j_s), the antisymmetric component is primarily dominated by τ_{OFT} (from j_c). Figure 1(b) shows the deconvoluted ST-FMR spectra of V_{mix} measured at frequency $f = 5$ GHz for Pt(N) 5×10^{16} . Please refer to Appendix B to see the deconvoluted ST-FMR spectra obtained for other samples. The broad range of ST-FMR spectra obtained for applied $f = 5\text{--}11$ GHz is shown for all the samples (Figs. 6(a)–6(e) in Appendix B). The Gilbert damping parameter α which depends on linewidth ΔH is estimated using [21]

$$\Delta H = \Delta H_0 + \frac{2\pi f}{\gamma} \alpha, \quad (2)$$

where γ is the gyromagnetic ratio and ΔH_0 is the inhomogeneous linewidth broadening which is independent of f . Referring to Appendix B [Fig. 6(f)], α is estimated from the slope of ΔH plotted as a function of f . The value of α is higher for Pt(N) 5×10^{16} as compared to that of Pt(N) 2×10^{16} and Pt(N) 1×10^{17} . To quantify θ_{DL} , we performed the line-shape analysis of the ST-FMR spectrum, using Eq. (3) [9]:

$$\theta_{DL} = \frac{S e \mu_0 M_s t d}{A \hbar} \sqrt{1 + \frac{M_{\text{eff}}}{H_0}}, \quad (3)$$

where e is the elementary charge, \hbar is the reduced Planck constant, t is the thickness of the NiFe layer, d is the thickness of the HM layer, and the effective magnetization M_{eff} is obtained from Kittel fitting. The figure in Appendix B [Fig. 6(g)] shows the obtained values of θ_{DL} for the studied frequency range $f = 5\text{--}11$ GHz. θ_{DL} is found to be invariant with frequency, implying a negligible role of thermal effect and noncontrolled relative phase between I_{rf} and h_{rf} that arises from sample design [21,35,36]. The average θ_{DL} values obtained are 0.119 ± 0.002 for Pt(N) 2×10^{16} , 0.132 ± 0.008 for Pt(N) 5×10^{16} , and 0.098 ± 0.008 for Pt(N) 1×10^{17} . Noticeably, the θ_{DL} of pure sample (Pt/NiFe) is found to be 0.052 ± 0.004 . It demonstrates that ion implantation provides a better alternative to incorporate nitrogen in Pt when compared to the sputtering method, as even a small dose of 2×10^{16} ions/cm² in Pt leads to ~ 2.3 times enhancement in θ_{DL} from 0.052 to 0.119. We find a ~ 2.5 times enhancement in θ_{DL} from 0.052 (Pt) to 0.132 [Pt(N) 5×10^{16}]. However, we observe a nonmonotonic dependence of θ_{DL} on the implantation dose, similar to Xu *et al.* [27] where nitrogen was incorporated in Pt via sputtering. On the contrary, we recently observed a monotonic dependence of θ_{DL} on oxygen (O) implantation dose [21]. We also performed the angular dependent ST-FMR measurements (see Appendix C). The unbroken twofold and mirror symmetries of the torques along with the negligible spin-pumping (see Appendix D) contribution allows us to use the line-shape analysis to quantify the θ_{DL} .

To picturize the correlation among ρ_{xx} , α , $\mu_0 M_{\text{eff}}$, spin mixing conductance $g_{\text{eff}}^{\uparrow\downarrow}$ and θ_{DL} for Pt(N), we plot these parameters as a function of the implantation dose as shown in Fig. 2. First, ρ_{xx} is found to be monotonically increasing

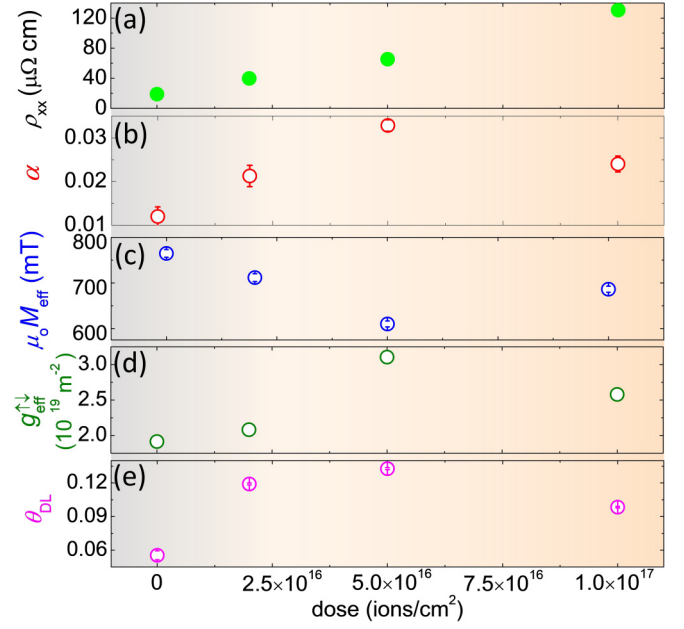


FIG. 2. Dose-dependent ST-FMR measurements: (a) ρ_{xx} , (b) α , (c) $\mu_0 M_{\text{eff}}$, (d) $g_{\text{eff}}^{\uparrow\downarrow}$, and (e) θ_{DL} for different doses of nitrogen in Pt, at room temperature (293 K).

in Fig. 2(a), like a previous report [27]. Second, α shows a nonmonotonic dependence with an increase of the N-ion dose, where a maximum value of 0.032 is obtained for the dose of 5×10^{16} [Fig. 2(b)]. Third, $\mu_0 M_{\text{eff}}$ decreases with increasing N-ion dose [Fig. 2(c)]. Noticeably, a minimum $\mu_0 M_{\text{eff}} = 610$ mT is seen for Pt(N) 5×10^{16} as compared to 711 mT for Pt(N) 2×10^{16} and 686 mT for Pt(N) 1×10^{17} , indicating a change in the perpendicular magnetic anisotropy field H_p . The $\mu_0 M_{\text{eff}}$ of pure Pt is found to be 765 mT, indicating less H_p . Fourth, the $g_{\text{eff}}^{\uparrow\downarrow}$ shows a similar trend as obtained for α [Fig. 2(d)], indicating an enhanced j_s at the HM/FM interface. Fifth, θ_{DL} in Fig. 2(e) shows a similar trend as obtained for α and $g_{\text{eff}}^{\uparrow\downarrow}$. Summarizing the dose dependent results, both α and θ_{DL} increase monotonically from 0 to 5×10^{16} ions/cm² and then suddenly decrease for 1×10^{17} ions/cm², like the results obtained by Xu *et al.* [27]. In agreement with the highest α and $g_{\text{eff}}^{\uparrow\downarrow}$ for Pt(N) 5×10^{16} , θ_{DL} is found to be maximum for Pt(N) 5×10^{16} [37]. However, due to the limited variation of θ_{DL} with ρ_{xx} , studying the θ_{DL} for different doses of impurities at room temperature alone may be insufficient in understanding the underlying mechanism. Hence, it is also important to investigate the dependence of θ_{DL} and associated properties on temperature.

B. Temperature-dependent ST-FMR measurements

To gain a deeper understanding of the enhancement in SHE of Pt(N), we performed temperature (T) dependent ST-FMR measurements in the range 100–293 K. A linear increase of ρ_{xx} as a function of T (for $T = 10\text{--}293$ K) is observed in Fig. 3(a), which confirms a metallic behavior [31]. Fitting a straight line to the data and extrapolating to $T = 0$ K allowed us to deduce the residual resistivity $\rho_{xx,0}$, which is summarized in Table II. Second, the linewidth ΔH and

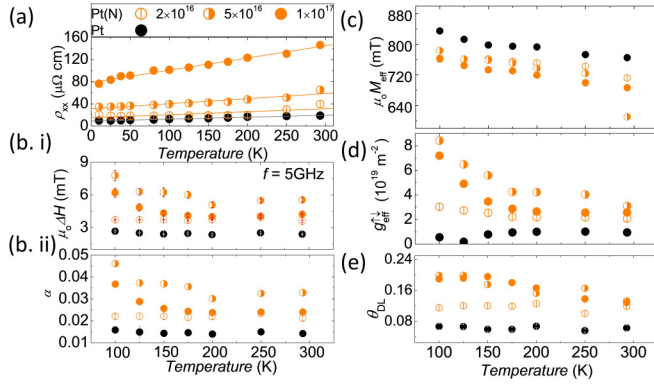


FIG. 3. T -dependent ST-FMR measurements: (a) ρ_{xx} , (b) (i) $\mu_o\Delta H$, (b) (ii) α , (c) $\mu_o M_{\text{eff}}$, (d) $g_{\text{eff}}^{\uparrow\downarrow}$, and (e) θ_{DL} for different doses of nitrogen in Pt plotted as a function of temperature. The solid line in 3(a) represents the linear fitting.

Gilbert damping parameter α increase at lower T , especially for higher doses, i.e., Pt(N) 5×10^{16} and Pt(N) 1×10^{17} , as shown in Fig. 3(b) (i and ii). This enhancement in α may be due to two reasons: an increase of j_s at lower T , and the increase of magnetic damping at the surface contribution. We consider the latter to be unlikely as it arises in a ferromagnet at a very low temperature range and with low thickness [38]. Third, $\mu_o M_{\text{eff}}$ increases at lower T , as seen in Fig. 3(c). This is in accordance with $M_s \propto \frac{1}{T}$, assuming that H_p remains invariant with temperature [39]. Fourth, to confirm the high j_s created at HM/FM interface [37], $g_{\text{eff}}^{\uparrow\downarrow}$ (see Appendix E) is found to increase with decreasing T , as shown in Fig. 3(d), especially for the higher doses of Pt(N) 5×10^{16} and Pt(N) 1×10^{17} . Last, θ_{DL} is plotted as a function of T in Fig. 3(e), and is found to increase with decreasing T for Pt(N) 5×10^{16} and Pt(N) 1×10^{17} . Consequently, a high θ_{DL} of 0.18–0.19 is obtained for Pt(N) 5×10^{16} and Pt(N) 1×10^{17} . The θ_{DL} is found to be invariant with T for Pt(N) 2×10^{16} and Pt. Such a kind of increase/decrease of θ_{DL} with T hints at the possibility of an intrinsic and/or extrinsic side-jump contribution [33,40,41].

C. Contribution to SHE

To elucidate the explicit contribution from intrinsic and extrinsic mechanism (side-jump/skew scattering) to the SHE, the spin Hall conductivity, $\sigma_{\text{SH}}^{\text{xy}}$ is plotted as a function of longitudinal conductivity, σ_{xx} in the inset of Fig. 4(a). The anomalous Hall effect (AHE) and SHE share the same analogy as demonstrated experimentally by Moriya *et al.* [42]. The range of σ_{xx} in Pt(N) lies very close to $\sim 10^4 \Omega^{-1} \text{cm}^{-1}$. Therefore, in analogy to AHE, there has been power scaling between $\sigma_{\text{SH}}^{\text{xy}}$ and σ_{xx} , i.e., $\sigma_{\text{SH}}^{\text{xy}} \propto \sigma_{\text{xx}}^{1.6-1.8}$ due to intrinsic SHE for the bad metal/dirty metal regime ($\sigma_{\text{xx}} \lesssim 10^4 \Omega^{-1} \text{cm}^{-1}$), whereas, for the good metal regime ($10^4 \lesssim \sigma_{\text{xx}} \lesssim 10^6 \Omega^{-1} \text{cm}^{-1}$), $\sigma_{\text{SH}}^{\text{xy}}$ has been found to be independent of σ_{xx} due to intrinsic SHE [30,31]. We cannot obtain a power scaling, i.e., $\sigma_{\text{SH}}^{\text{xy}} \propto \sigma_{\text{xx}}^{1.6-1.8}$ or $\sigma_{\text{SH}}^{\text{xy}}$ being independent of σ_{xx} for Pt(N) [see inset of Fig. 4(a)]. Therefore, a simple model that can account for both the good metal and bad metal regime may not reveal the

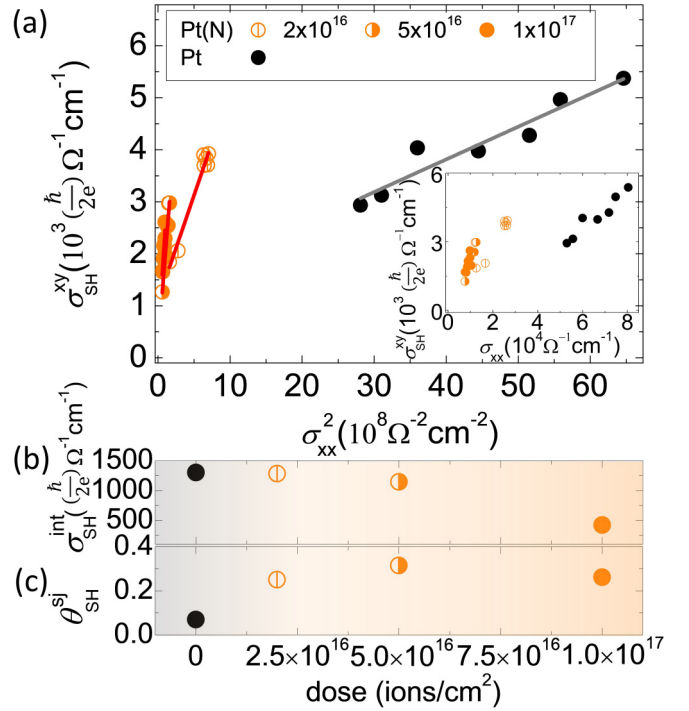


FIG. 4. Contribution to SHE: (a) $\sigma_{\text{SH}}^{\text{xy}}$ plotted as a function of σ_{xx}^2 for implantation (orange data points) and pure Pt (black data points). The solid lines represent the fitting using Eq. (6). Inset shows $\sigma_{\text{SH}}^{\text{xy}}$ plotted as a function of σ_{xx} . (b) $\sigma_{\text{SH}}^{\text{int}}$, and (c) $\theta_{\text{SH}}^{\text{sj}}$ plotted as a function of N-ion dose.

exact contribution of intrinsic and extrinsic (skew/side-jump) scattering induced SHE in our samples.

To understand the contribution of extrinsic (skew/side-jump) scattering, we again focus on the range of σ_{xx} in our samples. In analogy to AHE, skew scattering arises in the higher conductivity range of superclean metals ($10^6 \lesssim \sigma_{\text{xx}} \lesssim 10^8 \Omega^{-1} \text{cm}^{-1}$) [30,31,43]. Additionally, impurity induced skew scattering shows a T -independent θ_{DL} which is not observed for the higher doses in our samples [Fig. 3(e)] [34,40,41]. Hence, skew scattering is not a possible mechanism in our samples. Next, to sort out the assumption of considering the extrinsic side-jump scattering in our samples, we checked an independent method employed by Asomoza *et al.* [44], Niimi and co-workers [34,41], and Ramaswamy *et al.* [45]. Upon removing the intrinsic SHE contribution, we obtain the $\rho_{\text{SH}}^{\text{imp}} \propto \rho_{\text{imp}}^2$ trend, suggesting the influence of side-jump scattering induced SHE in our samples. Here, $\rho_{\text{SH}}^{\text{imp}}$ is the spin Hall resistivity from impurities, and ρ_{imp} is the impurity induced resistivity. Please see Appendix F for details. Therefore, after safely excluding the skew scattering and confirming the influence of side jump, we try to separate the intrinsic and extrinsic side-jump induced SHE.

In AHE, the side-jump term, ρ_{sj} or σ_{sj} (proposed by Berger) [46] arising from extrinsic effect was confusingly viewed as an intrinsic term, ρ_{int} or σ_{int} (proposed by Karplus and Luttinger) [47]. This happened due to both the ρ_{int} and ρ_{sj} being proportional to ρ_{xx}^2 (or, simply, $\sigma_{\text{int}} \propto \sigma_{\text{xx}}^2$ and $\sigma_{\text{sj}} \propto \sigma_{\text{xx}}^2$), where ρ_{xx}^2 is the square of resistivity and σ_{xx}^2 is the square of conductivity. This was accepted until the concept of residual

TABLE I. α , $\mu_o M_{\text{eff}}$, $g_{\text{eff}}^{\uparrow\downarrow}$, θ_{DL} , and ρ_{xx} for different N-ion doses at 293 K.

Sample	α	$\mu_o M_{\text{eff}}$ (mT)	$g_{\text{eff}}^{\uparrow\downarrow}$ (10^{19} m^{-2})	θ_{DL}	ρ_{xx} at 293 K ($\mu\Omega \text{ cm}$)
Pure Pt	0.012	765	1.91	0.052	18.8
Pt(N) 2×10^{16} ions/cm ²	0.022	711	2.08	0.119	39.7
Pt(N) 5×10^{16} ions/cm ²	0.032	610	3.10	0.132	65.3
Pt(N) 1×10^{17} ions/cm ²	0.024	686	2.57	0.098	146.3

resistivity arising from impurities, $\rho_{\text{xx},0}$, was introduced by Tian *et al.* [48]. They proposed that the proper scaling for the AHE should involve not only the ρ_{xx} (or σ_{xx}), but also an important term, the residual resistivity $\rho_{\text{xx},0}$. As mentioned earlier, the AHE and SHE share the same analogy. Therefore, the total spin Hall conductivity $\sigma_{\text{SH}}^{\text{xy}}$ can be expressed as a sum of intrinsic and extrinsic SHE (side jump and skew scattering) [30,33,48,49]:

$$|\sigma_{\text{SH}}^{\text{xy}}| = \sigma_{\text{SH}}^{\text{int}} + |\sigma_{\text{SH}}^{\text{sj}} + \sigma_{\text{SH}}^{\text{ss}}|, \quad (4)$$

$$= \sigma_{\text{SH}}^{\text{int}} + |(\theta_{\text{SH}}^{\text{sj}} \rho_{\text{xx},0} + \theta_{\text{SH}}^{\text{ss}} \rho_{\text{xx},0}) \sigma_{\text{xx}}^2|, \quad (5)$$

where $\sigma_{\text{SH}}^{\text{int}}$ is intrinsic spin Hall conductivity, $\sigma_{\text{SH}}^{\text{sj}}$ is spin Hall conductivity due to side-jump scattering, $\sigma_{\text{SH}}^{\text{ss}}$ is spin Hall conductivity due to skew scattering, $\theta_{\text{SH}}^{\text{sj}}$ is side-jump induced SHE efficiency, $\theta_{\text{SH}}^{\text{ss}}$ is skew scattering induced SHE efficiency, and σ_{xx} is conductivity. After excluding skew scattering in our samples, as discussed in the previous paragraph, Eq. (5) can be expressed as

$$|\sigma_{\text{SH}}^{\text{xy}}| = \sigma_{\text{SH}}^{\text{int}} + |(\theta_{\text{SH}}^{\text{sj}} \rho_{\text{xx},0}) \sigma_{\text{xx}}^2|. \quad (6)$$

To probe the exact contributions from intrinsic and side-jump scattering, $\sigma_{\text{SH}}^{\text{xy}}$ is plotted as a function of σ_{xx}^2 in Fig. 4(a). Using the value of $\rho_{\text{xx},0}$, a $\sigma_{\text{SH}}^{\text{int}}$ of $1303.29(\frac{\hbar}{2e})\Omega^{-1} \text{ cm}^{-1}$ is obtained for Pt, which is found to be remarkably close to the theoretical value of $1300(\frac{\hbar}{2e})\Omega^{-1} \text{ cm}^{-1}$ reported by Tanaka *et al.* [50]. Further, for Pt(N), $\sigma_{\text{SH}}^{\text{xy}}$ is fitted to Eq. (6) and a $\sigma_{\text{SH}}^{\text{int}}$ of $1283.55(\frac{\hbar}{2e})\Omega^{-1} \text{ cm}^{-1}$ is obtained for lower dose Pt(N) 2×10^{16} as shown in Fig. 4(b). This is close to the $\sigma_{\text{SH}}^{\text{int}}$ of Pt, hinting at the dominant intrinsic SHE mechanism. However, the surprising result is the lower $\sigma_{\text{SH}}^{\text{int}}$ of $1146.01(\frac{\hbar}{2e})\Omega^{-1} \text{ cm}^{-1}$ and $422.72(\frac{\hbar}{2e})\Omega^{-1} \text{ cm}^{-1}$ for Pt(N) 5×10^{16} and Pt(N) 1×10^{17} , respectively [see Fig. 4(b)]. The $\sigma_{\text{SH}}^{\text{int}}$ decreases with increasing ρ_{xx} (Tables I and II), which is predicted by Kontani *et al.* [51]. As the amount of N impurity in Pt increases, the intrinsic contribution decreases and, in turn, the extrinsic contribution increases. Simply, a sudden decrease in $\sigma_{\text{SH}}^{\text{int}}$ is

counterbalanced by an increase of $\theta_{\text{SH}}^{\text{sj}}$ due to increase in ρ_{xx} and $\rho_{\text{xx},0}$ (Tables I and II). The $\theta_{\text{SH}}^{\text{sj}}$ is found to be 0.31 and 0.26 for Pt(N) 5×10^{16} and Pt(N) 1×10^{17} , respectively [Fig. 4(c)]. Pt is a spin Hall material having a positive SHE sign [9,11,50–52] and so, the positive sign of $\sigma_{\text{SH}}^{\text{int}}$ indicates that intrinsic SHE still has some contribution to the SHE [30,33,50–52]. The increase in θ_{DL} in Pt (N) is also influenced by extrinsic side-jump scattering, especially for the high implanted dose samples Pt(N) 5×10^{16} and Pt(N) 1×10^{17} . Hence, with the increase in dose, we observe an increase in extrinsic side-jump contribution to SHE, which could play a significant role in the enhancement of SHE. This also leads to a reduction in the intrinsic SHE. Finally, a crossover of intrinsic to extrinsic side-jump induced SHE is observed as the implantation of the N dose from 2×10^{16} ions/cm² to 1×10^{17} ions/cm² is increased in the Pt layer. The successful disentanglement of intrinsic and extrinsic side jump induced by SHE could be a promising approach to understand the mechanism for enhancement in SHE.

IV. CONCLUSION

In conclusion, we utilized a less explored approach of ion implantation to incorporate nitrogen in Pt by varying the dose from 2×10^{16} to 1×10^{17} ions/cm². We studied the dependence of θ_{DL} on both the N-ion dose and temperature, to disentangle the intrinsic and extrinsic side-jump scattering mechanism in SHE. We found a crossover of intrinsic to extrinsic side-jump induced SHE as the N-ion dose increased from 2×10^{16} ions/cm² to 1×10^{17} ions/cm². A decrease in $\sigma_{\text{SH}}^{\text{int}}$ is observed due to the increase in ρ_{xx} , and subsequently, $\rho_{\text{xx},0}$. These results indicate that studying the θ_{DL} as a function of implantation dose, and as a function of temperature, is important to understand the underlying physical phenomenon contributing to SHE. We believe that such a deep comprehension of enhancement in SHE may help us in revealing the host and impurity combination to unlock the full potential of SHE in 5d transition metals for SOT-MRAM application.

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 TABLE II. $\rho_{\text{xx},0}$, $\sigma_{\text{SH}}^{\text{int}}$, and $\theta_{\text{SH}}^{\text{sj}}$ for different N-ion doses.

Sample	$\rho_{\text{xx},0}$ ($\mu\Omega \text{ cm}$)	$\sigma_{\text{SH}}^{\text{int}}$ [$(\frac{\hbar}{2e})\Omega^{-1} \text{ cm}^{-1}$]	$\theta_{\text{SH}}^{\text{sj}}$
Pure Pt	9.0	1303.29	0.069
Pt(N) 2×10^{16} ions/cm ²	14.8	1283.55	0.251
Pt(N) 5×10^{16} ions/cm ²	31.1	1146.01	0.315
Pt(N) 1×10^{17} ions/cm ²	78.8	422.72	0.262

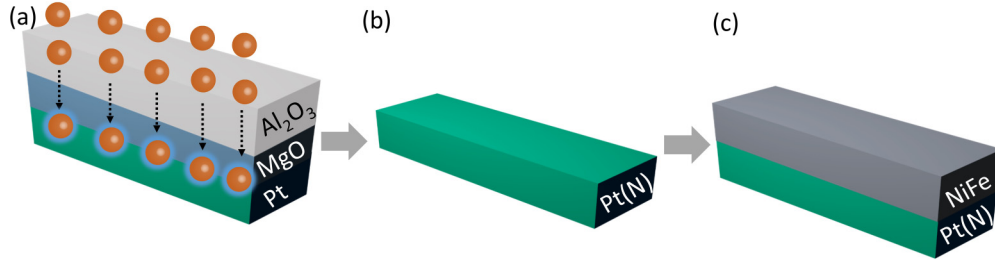


FIG. 5. (a) Visualization of ion implantation followed by removal of protective layers (b) and (c) after the deposition of NiFe on top of Pt.

Singapore, respectively. The authors thank R. Kumar, Inter University Accelerator Center, for his support in nitrogen ion implantation using the Tabletop Accelerator.

APPENDIX A: ION IMPLANTATION METHOD

The present experiment was carried out using a 30-kV ion accelerator/implanter installed at the Inter-University Accelerator Centre (IUAC), New Delhi. This system is comprised of an ion source and a cold plasma-based Penning ion generator, assembled in a nylon housing and connected to a 30-kV power supply which is capable of delivering a stable current of up to $350 \mu\text{A}$. An Einzel lens and an electrostatic quadrupole are used for focusing the ion beams. The required ions are selected by adjusting the magnetic field strength of a bending magnet which provides a uniform, variable field of 0.21–0.35 T. The ion beam spot size of $15 \times 15 \text{ mm}^2$ can be scanned upon the target in the implantation chamber having a vacuum of $1.3 \times 10^{-4} \text{ Pa}$.

The ion dose or fluence is defined as the number of ions implanted per unit area of the target. The number of ions is determined by allowing the beam to hit the inner wall of a Faraday cup (conductive) and measuring the resulting current by a current integrator. Hence, the dose or fluence (ions/cm^2) is calculated by using the relation

$$\text{Dose} = \frac{\int I dt}{qe(\text{Area})} (\text{ions}/\text{cm}^2), \quad (7)$$

where I is the beam current, t is the time, q is the charge state of the ion, e is the electronic or elementary charge, and Area is the area of the target.

The deposited multilayer stack of Pt (10 nm)/MgO (10 nm)/Al₂O₃(10 nm) were implanted with a 20-keV N ion beam, as shown in Fig. 5(a). The two protective layers (MgO and Al₂O₃) were deposited above Pt for the following two reasons: First, the protective layers allowed the least disturbance to the target Pt layer while ensuring the uniform distribution of N ions in Pt following a similar recipe of

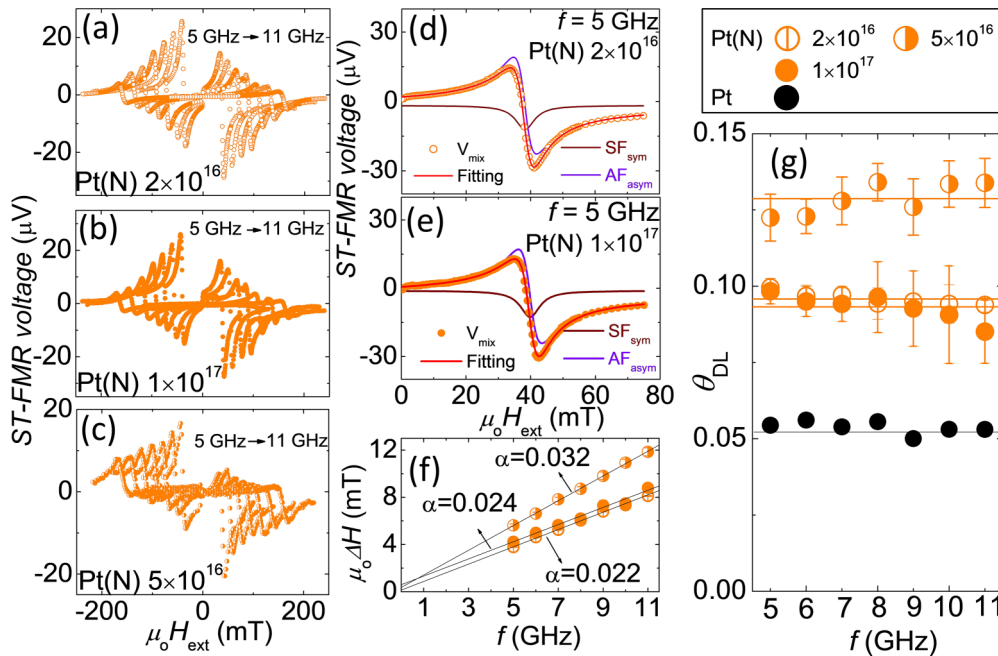


FIG. 6. ST-FMR spectra V_{mix} obtained for (a) Pt(N) 2×10^{16} , (b) Pt(N) 1×10^{17} , (c) Pt(N) 5×10^{16} . Deconvolution fitting of V_{mix} measured at $f = 5 \text{ GHz}$ into symmetric ($S F_{\text{sym}}$) and antisymmetric ($A F_{\text{asym}}$) component displayed by brown and violet solid lines, respectively, for (d) Pt(N) 2×10^{16} and (e) Pt(N) 1×10^{17} , (f) $\mu_0 \Delta H$ vs f (with solid black lines as linear fit), and (g) f invariant θ_{DL} obtained for Pt(N) 2×10^{16} , Pt(N) 5×10^{16} , Pt(N) 1×10^{17} , and pure Pt. The lines in (g) represent the average value.

our previous work [20]. Second, the MgO provides a good end-point detector signal during the Ar⁺ ion milling whereas Al₂O₃ protects the hygroscopic MgO from self-sputtering in the implantation process [20]. Then, oxide capping layers of MgO and Al₂O₃ were removed by ion milling [Fig. 5(b)] and the top FM layer of NiFe (5 nm) was then sputtered on these samples [Fig. 5(c)].

APPENDIX B: LINE-SHAPE ANALYSIS AT 293 K (ROOM TEMPERATURE)

We performed the ST-FMR based line-shape analysis in a wide range of frequency $f = 5\text{--}11$ GHz to extract the symmetric and antisymmetric components from V_{mix} , i.e., $F_{\text{sym}}(H_{\text{ext}})$ and $F_{\text{asym}}(H_{\text{ext}})$, multiplied with the weight factors S and A . From V_{mix} , linewidth ($\mu_0\Delta H$) and S/A are used to calculate the α and θ_{DL} , respectively (see Fig. 6).

APPENDIX C: ANGULAR DEPENDENT ST-FMR

The line-shape analysis, which uses the ratio of weight factors of symmetric (S) to antisymmetric (A) component at one fixed angle of $\phi = 45^\circ$, could be a hindrance in revealing the complete picture of exerted spin-orbit torques [21,36]. There may be hidden effects apart from SHE, such as effective field with different spin polarization, poor device designs, Nernst heating, etc., which may serve as an artifact and lead to an unreliable assessment of spin-orbit torque (SOT). So, we performed the angular ST-FMR measurements by varying the angle between applied H_{ext} and along the length of HM/FM bilayer axis from $\phi = 0^\circ$ to 360° . By deconvolution of V_{mix} using Eq. (1) into S and A , we fitted the data with the anticipated $\sin 2\phi \cos \phi$ for the implanted sample Pt(N) 2×10^{16} , Pt(N) 5×10^{16} , and Pt(N) 1×10^{17} [Figs. 7(a)–7(c)]. It shows that the SOT traces a similar line shape when the magnetization is rotated by 180° which implies that there is no breaking of the twofold ($180^\circ + \phi$) and mirror ($180^\circ - \phi$) symmetries of torques, affirming SHE as the only origin of the rectified voltage [21] obtained by ST-FMR. The $\cos \phi$ arises from τ_{DL} and τ_{OFT} in S and A , respectively, while $\sin 2\phi$ arises from AMR.

APPENDIX D: SPIN PUMPING CONTRIBUTION

The spin pumping contribution V_{SP} in the symmetric component of the ST-FMR spectrum may have a role to play in the high values of α , $g_{\text{eff}}^{\uparrow\downarrow}$, and θ_{DL} for Pt(N) 5×10^{16} , and therefore it might be naive to not identify this contribution. To confirm that the enhanced α , $g_{\text{eff}}^{\uparrow\downarrow}$, and θ_{DL} in the N-implanted sample may be attributed to the enhanced DLT, and is not due to the contribution of spin pumping voltage in the symmetric component of the ST-FMR spectrum, we investigated the spin pumping contribution V_{SP} using the derived values of $g_{\text{eff}}^{\uparrow\downarrow}$ and $\theta_{\text{DL}}^{\text{LS}}$ by [11]

$$V_{\text{SP}} = \frac{\theta_{\text{DL}} l \lambda_{\text{sd}}}{d \sigma_{\text{Pt(N)}} + t \sigma_{\text{NiFe}}} \tanh\left(\frac{d}{2\lambda_{\text{sd}}}\right) \left(\frac{2e}{h}\right) j_s \sin(\phi), \quad (8)$$

where, l is the length of the device, λ_{sd} is the spin diffusion length of the Pt(N) layer, d and $\sigma_{\text{Pt(N)}}$ are the thickness and conductivity of Pt(N), t and σ_{NiFe} are the thickness and con-

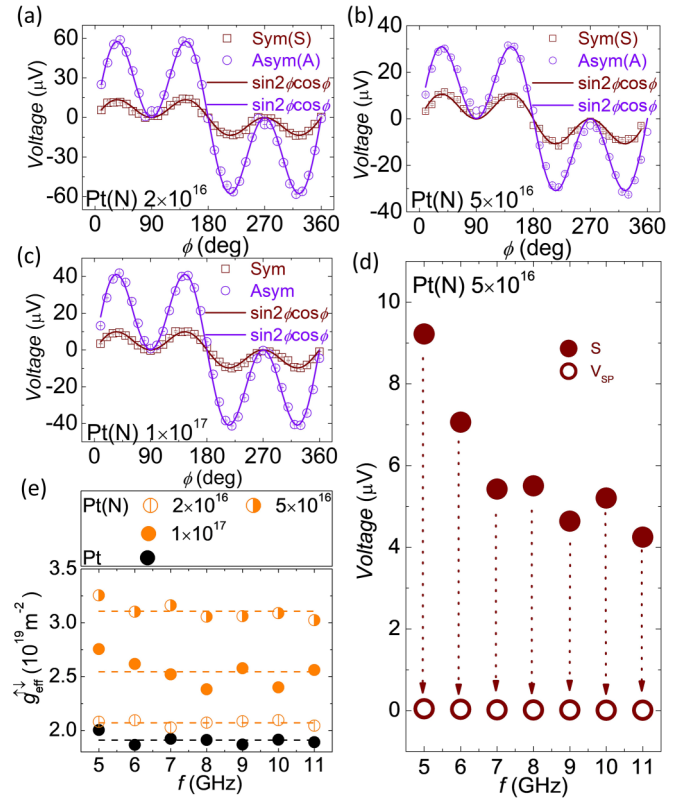


FIG. 7. (a)–(c) Angular dependence of S and A components in ST-FMR spectra for Pt(N) 2×10^{16} , Pt(N) 5×10^{16} , and Pt(N) 1×10^{17} ($f = 5$ GHz) with solid lines fitted by $\sin 2\phi \cos \phi$. (d) S and V_{SP} plotted as a function of f for Pt(N) 5×10^{16} . (e) $g_{\text{eff}}^{\uparrow\downarrow}$ obtained as a function of f for Pt(N) 2×10^{16} , Pt(N) 5×10^{16} , Pt(N) 1×10^{17} along with pure Pt. The dashed lines represent the average value in (e).

ductivity of NiFe, and ϕ is the angle between H_{ext} and I_{rf} applied in the longitudinal direction of the HM/FM bilayer (45°), j_s is the spin current density from the precessing NiFe into the Pt(N), given as $j_s = \frac{h}{2} f \sin^2(\theta_c) g_{\text{eff}}^{\uparrow\downarrow}$, where the precession cone angle θ_c is given by $\theta_c = \frac{1}{dR/d\phi} \frac{2}{I_{\text{rf}}} \sqrt{S^2 + A^2}$, $dR/d\phi$ is obtained from AMR, and I_{rf} is the current in the HM/FM bilayer. The ratio of the spin pumping voltage V_{SP} to the symmetric component S is found to be 0.39% for Pt(N) 2×10^{16} , 0.49% for Pt(N) 5×10^{16} , and 0.25% for Pt(N) 1×10^{17} , which are all less than 1%. Figure 7(d) shows S and V_{SP} plotted for Pt(N) 5×10^{16} as a function of frequency, confirming the negligible contribution of V_{SP} as compared to S .

APPENDIX E: SPIN MIXING CONDUCTANCE

Spin mixing conductance $g_{\text{eff}}^{\uparrow\downarrow}$ is an important parameter that provides a better picture of transversely generated j_s created at the HM/FM interface. Based on the theory of spin pumping, assuming that there is no significant spin memory loss, the $g_{\text{eff}}^{\uparrow\downarrow}$ [21] can be estimated from the linewidth difference δ of the ST-FMR spectra ($\delta = \Delta H_{\text{Pt(N)/NiFe}} - \Delta H_{\text{NiFe}}$), given by

$$g_{\text{eff}}^{\uparrow\downarrow} = \left(\frac{\gamma}{2\pi f}\right) \left(\frac{4\pi M_s t \delta}{g \mu_0 \mu_B}\right), \quad (9)$$

where, g is the Landé g factor, μ_0 is the permeability of free space, M_s is the saturation magnetization of NiFe, and μ_B is the Bohr magneton constant. The average value of $g_{\text{eff}}^{\uparrow\downarrow}$ is found to be $2.08 \times 10^{19} \text{ m}^{-2}$ for Pt(N) 2×10^{16} , $3.10 \times 10^{19} \text{ m}^{-2}$ for Pt(N) 5×10^{16} , and $2.57 \times 10^{19} \text{ m}^{-2}$ for Pt(N) 1×10^{17} as shown by the dashed lines in Fig. 7(e). Most importantly, Pt(N) 5×10^{16} is found to have a higher $g_{\text{eff}}^{\uparrow\downarrow}$ value in comparison to Pt(N) 2×10^{16} and Pt(N) 1×10^{17} . For a HM layer much thicker than its λ_{sd} , θ_{DL} is found to be proportional to $g_{\text{eff}}^{\uparrow\downarrow}$ [37]. This is linked to the fact that the Gilbert damping parameter $g_{\text{eff}}^{\uparrow\downarrow}$ and θ_{DL} of Pt(N) 5×10^{16} is found to be larger than that of Pt(N) 2×10^{16} and Pt(N) 1×10^{17} .

APPENDIX F: INFLUENCE OF EXTRINSIC SIDE-JUMP SCATTERING

To sort out the approach of separating the extrinsic side-jump scattering from intrinsic SHE, we removed the intrinsic SHE contribution. So, for extrinsic skew scattering, $\rho_{\text{SH}}^{\text{imp}}$ is proportional to the ρ_{imp} , i.e., $\rho_{\text{SH}}^{\text{imp}} \propto \rho_{\text{imp}}$, whereas, for extrinsic side-jump scattering, $\rho_{\text{SH}}^{\text{imp}}$ is proportional to the square of ρ_{imp} , i.e., $\rho_{\text{SH}}^{\text{imp}} \propto \rho_{\text{imp}}^2$ [34,41,44,45]. Here, $\rho_{\text{SH}}^{\text{imp}}$ and ρ_{imp} are defined as

$$\rho_{\text{SH}}^{\text{imp}} = \rho_{\text{SH}}^{\text{Pt(N)}} - \rho_{\text{SH}}^{\text{Pt}}, \quad (10)$$

$$\rho_{\text{imp}} = \rho_{\text{Pt(N)}} - \rho_{\text{Pt}}, \quad (11)$$

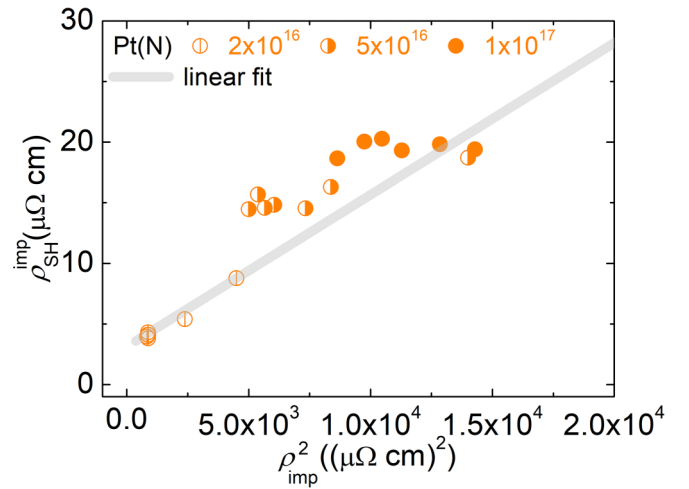


FIG. 8. $\rho_{\text{SH}}^{\text{imp}}$ vs ρ_{imp}^2 behavior for Pt(N). Solid grey line shows the linear fit.

where $\rho_{\text{SH}}^{\text{Pt(N)}}$ is the spin Hall resistivity of Pt(N), and $\rho_{\text{SH}}^{\text{Pt}}$ is the spin Hall resistivity of pure Pt. Also, $\rho_{\text{Pt(N)}}$ and ρ_{Pt} are the longitudinal resistivity of Pt(N) and pure Pt, respectively. Therefore, the $\rho_{\text{SH}}^{\text{imp}} \propto \rho_{\text{imp}}^2$ trend for all the Pt(N) samples confirms the influence of extrinsic side-jump scattering induced SHE (see Fig. 8).

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