



**Prediction of ferroelectric superconductors with reversible superconducting diode effect**Baoping Zhai <sup>1</sup>, Bohao Li,<sup>1</sup> Yao Wen,<sup>1</sup> Fengcheng Wu <sup>1,2,\*</sup> and Jun He<sup>1,2,†</sup><sup>1</sup>Key Laboratory of Artificial Micro and Nanostructures of Ministry of Education, and School of Physics and Technology, Wuhan University, Wuhan 430072, China<sup>2</sup>Wuhan Institute of Quantum Technology, Wuhan 430206, China

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A noncentrosymmetric superconductor can have a superconducting diode effect, where the critical current in opposite directions is different when time-reversal symmetry is also broken. We theoretically propose that a ferroelectric superconductor with coexisting ferroelectricity and superconductivity can support a ferroelectric reversible superconducting diode effect. Through first-principles calculation, we predict that monolayer  $\text{CuNb}_2\text{Se}_4$  (i.e., bilayer  $\text{NbSe}_2$  intercalated with Cu) is such a ferroelectric superconductor, where ferroelectricity controls the layer polarization as well as the sign of spin-orbit-coupling-induced spin splittings. Because the nonreciprocal effect of the critical current is proportional to the spin splittings, the superconducting diode effect is reversible upon electric switch of ferroelectricity. While we use  $\text{CuNb}_2\text{Se}_4$  as a model system, the predicted effect can appear in a class of two-dimensional superconducting bilayers with ferroelectricity induced by interlayer sliding. Our work opens the door to studying the interplay between superconductivity and ferroelectricity in two-dimensional materials.

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**Introduction.** A noncentrosymmetric material can support nonreciprocal charge transport, where the electrical resistance becomes different if the direction of the charge current is reversed. Recently, nonreciprocal phenomena in superconductors have become an active research topic [1–18]. In bulk metals without the inversion symmetry, nonreciprocal charge transport occurs when the time-reversal symmetry is also broken. This nonreciprocity induced by the magnetochiral anisotropy is significantly enhanced for the paraconductivity near the superconducting transition temperature  $T_c$  because of the superconducting fluctuation [1]. Moreover, in the superconducting state below  $T_c$ , the critical current along opposite directions differs, i.e.,  $j_c(\hat{n}) \neq j_c(-\hat{n})$ , where  $j_c(\hat{n})$  represents the magnitude of the critical current along direction  $\hat{n}$ . This nonreciprocity results in the superconducting diode effect (SDE) [7], where the system is superconducting in one direction but resistive in the opposite direction if the applied current has a magnitude between  $j_c(\hat{n})$  and  $j_c(-\hat{n})$ . The SDE was recently observed experimentally in several systems [7–12,19], including an artificial superlattice  $[\text{Nb}/\text{V}/\text{Ta}]_n$  [7] and a heterostructure of the twisted trilayer graphene and  $\text{WSe}_2$  [19]. Theory on the SDE was developed based on Ginzburg-Landau free energy as well as microscopic calculation [14–18].

In this Letter, we introduce a new type of noncentrosymmetric superconductors, i.e., ferroelectric superconductors, where ferroelectricity acts as a new knob in tuning superconductivity. A ferroelectric material breaks the inversion symmetry with a spontaneous electric polarization that can

be reversed by an applied electric field. While ferroelectric semiconductors/insulators have been widely studied, ferroelectricity can also exist in metals as exemplified by few-layer  $\text{WTe}_2$  [20]. When a ferroelectric metal (also known as polar metal) becomes superconducting at low temperatures, a ferroelectric superconductor forms and supports the SDE if time-reversal symmetry is further broken. We predict that the superconducting direction of the diode can be reversed upon ferroelectric reversal, which we term as the reversible SDE. This prediction represents an example on the controlling of superconductivity through ferroelectricity.

For material realization, we propose monolayer  $\text{CuNb}_2\text{Se}_4$  to be a ferroelectric superconductor with the reversible SDE. Monolayer  $\text{CuNb}_2\text{Se}_4$  can be viewed as a 2H bilayer  $\text{NbSe}_2$  intercalated by Cu atoms, as illustrated in Fig. 1. We establish both ferroelectricity and superconductivity in this material through first-principles calculations. The ferroelectricity in monolayer  $\text{CuNb}_2\text{Se}_4$  controls the layer and spin degrees of freedom of low-energy states across the Fermi energy in  $\pm K$  valleys (two corners of the hexagonal Brillouin zone), where both the layer polarization and the valley-dependent spin splittings [induced by spin-orbit coupling (SOC)] are reversed by ferroelectric reversal. In the superconducting state, the nonreciprocal factor  $\eta$  [Eq. (8)] of the critical current is proportional to the SOC-induced spin splittings. Therefore,  $\eta$  changes sign by the reverse of ferroelectricity, which leads to the reversible SDE. While we study monolayer  $\text{CuNb}_2\text{Se}_4$  in detail to elucidate the physics, reversible SDE is generally expected in ferroelectric superconductors, which can be realized in van der Waals (vdW) superconducting bilayers with ferroelectricity arising from interlayer sliding [21]. Our work not only presents a novel approach to detect ferroelectricity

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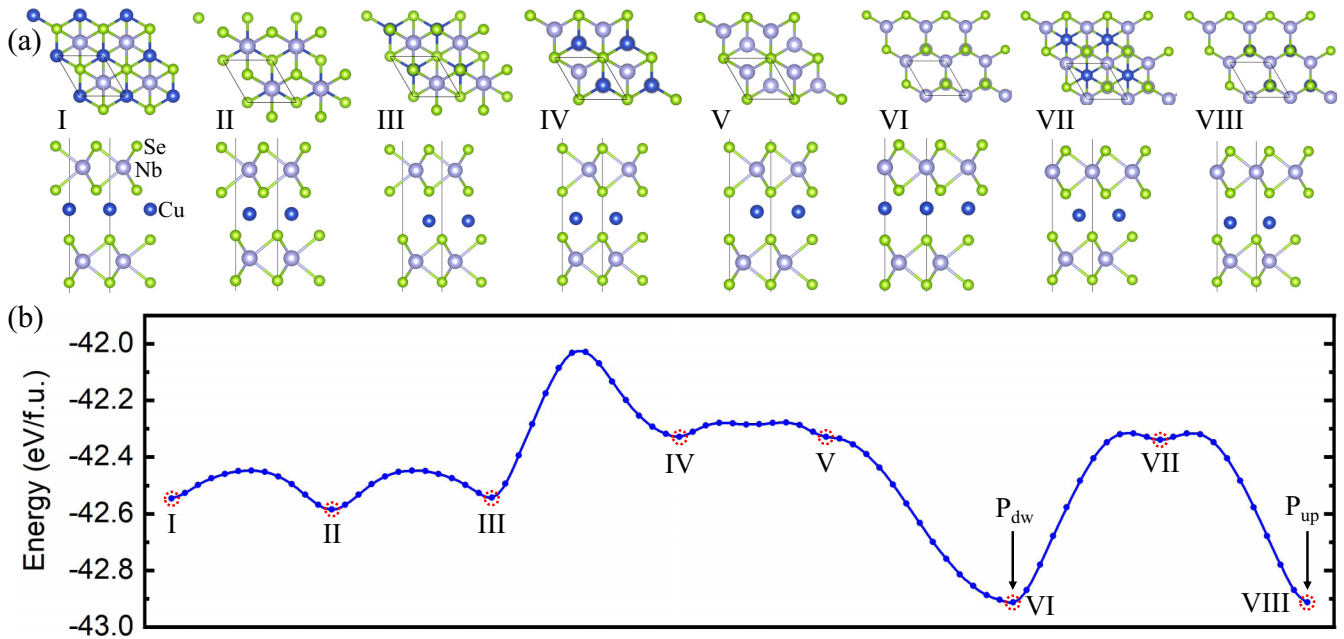


FIG. 1. (a) Top and side views of eight high-symmetry structures of monolayer CuNb<sub>2</sub>Se<sub>4</sub>. In structures I to III, Nb atoms are aligned vertically. In structures IV and V, Se atoms are aligned vertically. In structures VI to VIII, Nb atoms are aligned with Se atoms vertically. Structures II and VII are inversion symmetric, while other structures are not. (b) The energy per formula unit for different lattice structures of monolayer CuNb<sub>2</sub>Se<sub>4</sub>. The red dashed circles mark the high-symmetry structures shown in (a). The transition structures between two neighboring high-symmetry structures are generated by the nudged-elastic band method.

through superconducting transport, but also opens up new opportunities for the construction of electrically controllable and nondissipative diodes.

**Ferroelectricity.** The vdW materials, such as graphite and transition metal dichalcogenides (e.g., NbS<sub>2</sub>, NbSe<sub>2</sub>, MoS<sub>2</sub>), can be the intercalation host, which provides a powerful approach to induce a variety of exotic quantum phenomena [22–28] including ferroelectricity [29]. Intercalation has also been achieved in the two-dimensional (2D) limit [30–32]. Here we theoretically study monolayer CuNb<sub>2</sub>Se<sub>4</sub>. The strategy is to start from a prototypical 2D superconductor (i.e., bilayer NbSe<sub>2</sub>) [33], which we show to develop ferroelectricity upon Cu intercalation while remaining superconducting.

We demonstrate ferroelectricity in monolayer CuNb<sub>2</sub>Se<sub>4</sub> by studying eight high-symmetry structures, as illustrated in Fig. 1(a). In the monolayer CuNb<sub>2</sub>Se<sub>4</sub> under study, the top and bottom NbSe<sub>2</sub> layers are rotated by 180°. The eight structures can be distinguished by the in-plane relative positions of the top NbSe<sub>2</sub> layer, the middle Cu layer and the bottom NbSe<sub>2</sub> layer. We perform lattice relaxation for each structure using first-principles calculation implemented in the Vienna *ab initio* simulation package (VASP) [34] and obtain the corresponding energy. The energy landscape plotted in Fig. 1(b) shows that two different structures, i.e., structures VI and VIII, have the same lowest energy. We note that VI and VIII structures are inversion partners, although each of them on its own lacks inversion symmetry. Therefore, the VI and VIII structures have the same energy, but opposite layer polarizations, which gives rise to ferroelectricity. We further perform *ab initio* molecular dynamics (AIMD) simulations and phonon spectrum calculations for these two structures to verify

structural stability. The computational details are presented in the Supplemental Material (SM) [35] (see, also, Refs. [36–39] therein). In AIMD simulations, the energy fluctuates slightly and the structure maintains integrity after 5 ps at 300 K, indicating the thermodynamic stability. Meanwhile, the phonon dispersion calculated using the PHONOPY code [40] has no virtual frequency in the whole Brillouin zone, implying the dynamic stability.

We use Bader charge analysis [41] to quantitatively characterize the ferroelectricity. For structure VI, we find that the average number of electrons transferred from one Cu atom to the top and bottom NbSe<sub>2</sub> layers is 0.23 and 0.1, respectively [35], which are unequal because the Cu atoms occupy noncentrosymmetric sites. Thus, the structure VI possesses a downward electric dipole moment. By contrast, the structure VIII has an upward electric dipole moment. Hereafter, we refer to VI and VIII structures as  $P_{dw}$  and  $P_{up}$  structures, respectively. We also confirm the electric polarization by calculating the electrostatic potential difference across the monolayer, as discussed in the SM [35]. We investigate the ferroelectric transition process using the nudged-elastic band method [42]. The transition barrier [Fig. 1(b)] between  $P_{dw}$  and  $P_{up}$  is about 0.6 eV per formula unit, which is comparable to that of monolayer In<sub>2</sub>Se<sub>3</sub> [43]. Therefore, ferroelectric reversal by an applied out-of-plane electric field is feasible in monolayer CuNb<sub>2</sub>Se<sub>4</sub>.

**Band Structure.** The band structures including SOC effects are plotted in Fig. 2. Because  $P_{up}$  and  $P_{dw}$  structures are inversion partners and time-reversal symmetry is preserved, their band structures have identical energy dispersion. At the Fermi energy, there are eight Fermi pockets, of which four are in the  $\Gamma$  valley, two in the  $+K$  valley, and two in the  $-K$  valley

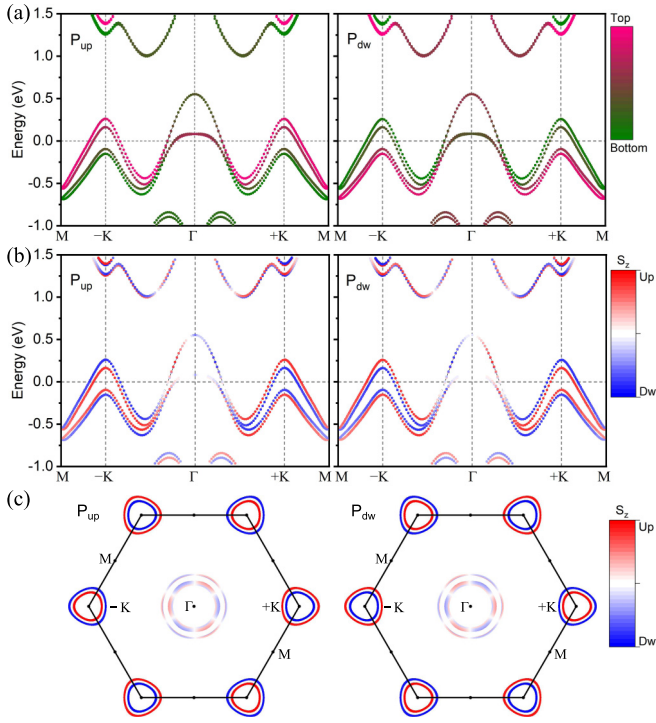


FIG. 2. (a) The layer-projected band structure of monolayer  $\text{CuNb}_2\text{Se}_4$  with SOC effect for  $P_{up}$  (left panel) and  $P_{dw}$  (right panel). (b) The spin-projected band structure. The spin projection is along out-of-plane  $\hat{z}$  axis with red representing spin up and blue representing spin down. Fermi energy is set to be 0 in (a) and (b). (c) The spin-projected Fermi surfaces.

[Fig. 2(c)]. Here  $\Gamma$  and  $\pm K$  represent, respectively, the center and two inequivalent corners of the Brillouin zone.

The opposite electric polarization in  $P_{up}$  and  $P_{dw}$  structures results in differences in electronic states regarding the layer and spin degrees of freedom. The layer-projected band structures in Fig. 2(a) show that the two bands crossing the Fermi energy  $E_F$  in  $\pm K$  valleys are mainly localized in the top (bottom)  $\text{NbSe}_2$  layer for the  $P_{up}$  ( $P_{dw}$ ) structure. The opposite layer polarization in combination with the  $180^\circ$  rotation between the two layers leads to ferroelectric reversible spin-valley coupling. To elaborate on this feature, we first focus on the  $P_{up}$  structure. In  $P_{up}$ , the higher and lower energy bands across  $E_F$  carry, respectively, up and down spin polarization in the  $+K$  valley, but down and up spin polarization in the  $-K$  valley dictated by time-reversal symmetry [Fig. 2(b)]. Here the spin polarization is along the out-of-plane  $\hat{z}$  axis. This is the well-known valley-dependent spin splitting effect in transition metal dichalcogenides [44]. We now turn to the  $P_{dw}$  structure, where the spin splitting in a given valley is opposite compared to that in the  $P_{up}$  structure. Therefore, the spin-valley coupling is controlled by the layer polarization, which is, in turn, controlled by the ferroelectricity.

The spin- and valley-dependent band structure in the  $\pm K$  valleys can be described by the following effective Hamiltonian:

$$\mathcal{H}_0 = -\frac{\hbar^2 \mathbf{k}^2}{2m^*} + \lambda_w (k_x^3 - 3k_x k_y^2) \tau_z + \Delta_{\text{SOC}}^{(\ell)} \tau_z \sigma_z - E_F, \quad (1)$$

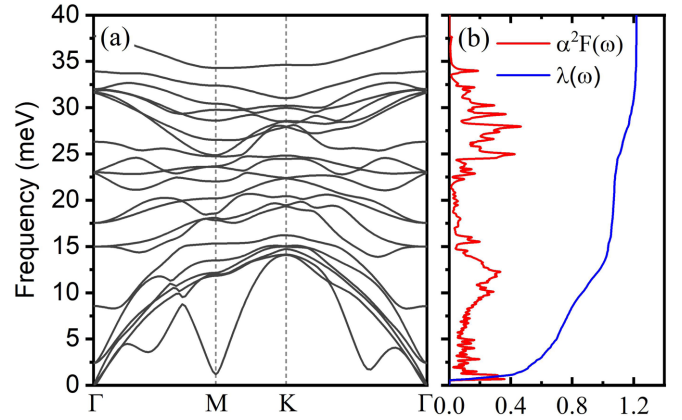


FIG. 3. (a) Phonon dispersion of the monolayer  $\text{CuNb}_2\text{Se}_4$  in  $P_{up}$  ( $P_{dw}$ ) structure. (b) Eliashberg function  $\alpha^2 F(\omega)$  (red line) and cumulative electron-phonon interaction strength  $\lambda(\omega)$  (blue line).

where  $\tau_z = \pm$  for  $\pm K$  valleys,  $\sigma_z = \pm$  for spin up ( $\uparrow$ ) and down ( $\downarrow$ ),  $\mathbf{k} = (k_x, k_y)$  is the momentum defined relative to the  $\tau_z K$  point,  $m^*$  is the effective mass,  $\lambda_w$  is the parameter of the trigonal warping of the Fermi surfaces, and  $\Delta_{\text{SOC}}^{(\ell)}$  is the spin splitting. To capture the dependence on layer polarization, we take  $\Delta_{\text{SOC}}^{(\ell)} = \ell \Delta_{\text{SOC}}$  where  $\ell = +1$  in the  $P_{up}$  structure and  $\ell = -1$  in the  $P_{dw}$  structure. By fitting to the band structure, we obtain  $m^* \approx 0.46m_0$ ,  $\lambda_w \approx 7.5 \text{ meV} \cdot \text{nm}^3$ ,  $\Delta_{\text{SOC}} \approx 50 \text{ meV}$ , and  $E_F \approx -0.2 \text{ eV}$ , where  $m_0$  is the free electron mass. The SOC-induced spin splitting is much weaker in the  $\Gamma$  valley, which we do not analyze in detail.

**Superconductivity.** Since monolayer  $\text{CuNb}_2\text{Se}_4$  is metallic, it can become superconducting at low temperatures. We note that the intercalation of Cu atoms into *bulk*  $\text{NbS}_2$  [45] and  $\text{NbSe}_2$  [46] was achieved experimentally, and superconductivity persists after the intercalation. Here we consider phonon-mediated superconductivity for the monolayer in the  $P_{up}$  ( $P_{dw}$ ) structure and estimate its superconducting transition temperature  $T_c$  based on the McMillan-Allen-Dynes formula [47,48]

$$k_B T_c = \frac{\hbar \omega_{\text{log}}}{1.2} \exp\left(-\frac{1.04(1+\lambda)}{\lambda - \mu^*(1+0.62\lambda)}\right), \quad (2)$$

where

$$\omega_{\text{log}} = \exp\left(\frac{2}{\lambda} \int_0^\infty d\omega \frac{\alpha^2 F(\omega)}{\omega} \log \omega\right),$$

$$\lambda = 2 \int_0^\infty \frac{\alpha^2 F(\omega)}{\omega} d\omega. \quad (3)$$

Here  $\omega$  is the phonon frequency,  $\omega_{\text{log}}$  is the logarithmic average of the phonon frequencies,  $\alpha^2 F(\omega)$  is the Eliashberg function [49],  $\lambda$  is the electron-phonon coupling strength, and  $\mu^*$  is the parameter accounting for the Coulomb repulsion. We recalculate the phonon spectra using the density functional perturbation theory (DFPT) as coded in QUANTUM ESPRESSO [50]. The obtained phonon dispersion shown in Fig. 3(a) is consistent with that calculated by the PHONOPY code. The electron-phonon coupling is then calculated using ‘‘Electron-phonon Wannier’’ (EPW) code [51,52], and the results are

presented in Fig. 3(b), where  $\lambda$  is found to be 1.22. Taking the empirical parameter  $\mu^*$  to be 0.15 [49], we obtain a  $T_c$  of 3.04 K. This estimation is consistent with the experimental  $T_c$  measured in electron-doped bilayer NbSe<sub>2</sub> [53], which provides a strong support for our prediction of superconductivity in monolayer CuNb<sub>2</sub>Se<sub>4</sub>.

**Superconducting diode effect.** A ferroelectric superconductor lacks spatial inversion symmetry and supports SDE provided that time-reversal symmetry  $\hat{T}$  is also broken. To break the  $\hat{T}$  symmetry, we consider a minimal model with a spin-splitting term added to the Hamiltonian,  $\mathcal{H} = \mathcal{H}_0 + \Delta_z \sigma_z$ , where  $\mathcal{H}_0$  is given by Eq. (1). The  $\Delta_z \sigma_z$  term can be induced by the proximity effect of a ferromagnetic insulator [54–57]. Here we take  $\Delta_z$  as a phenomenological parameter to demonstrate the effect. A magnetic-field-free superconducting diode effect was recently demonstrated in Ref. [56], where the time-reversal symmetry breaking for superconductors is generated through proximitized magnetization. This experiment [56] supports our proposal of using proximitized magnetization to induce the superconducting diode effect.

To theoretically analyze the SDE, we focus on states in the  $\pm K$  valleys, which have strong SOC-induced spin splittings and contribute most significantly to the SDE. We introduce an order parameter  $\Delta_{q,\tau_z}$  for intralayer pairing between  $(\tau_z, \uparrow)$  and  $(-\tau_z, \downarrow)$  states, and  $\mathbf{q}$  is the center-of-mass momentum of the Cooper pair. The free energy per area for  $\Delta_{q,\tau_z}$ , derived in the SM [35] is given by

$$\begin{aligned} \mathcal{F}[\Delta_{q,\tau_z}] &= \alpha_{q,\tau_z} |\Delta_{q,\tau_z}|^2 + \frac{\beta}{2} |\Delta_{q,\tau_z}|^4, \\ \alpha_{q,\tau_z} &= \alpha_0 + \gamma_{\tau_z} q^2 + \kappa_{\tau_z} (q_x^3 - 3q_x q_y^2), \\ \alpha_0 &= \nu \frac{T - T_c}{T_c}, \\ \gamma_{\tau_z} &= \frac{7\zeta(3)}{4} \nu (\tau_z \Delta_{\text{SOC}}^{(\ell)} - E_F) \frac{\hbar^2}{(\pi k_B T)^2 4m^*}, \\ \kappa_{\tau_z} &= -\tau_z \nu \lambda_w \Delta_z \frac{93\zeta(5)}{16} \frac{(\tau_z \Delta_{\text{SOC}}^{(\ell)} - E_F)^2}{(\pi k_B T)^4}, \end{aligned} \quad (4)$$

where  $T$  is the temperature,  $\nu$  is the density of states per spin and per valley,  $\zeta(n)$  is the Riemann zeta function, and  $\beta = 7\zeta(3)\nu/8(\pi k_B T)^2$ . We minimize  $\mathcal{F}[\Delta_{q,\tau_z}]$  with respect to  $\Delta_{q,\tau_z}$  and the free energy becomes

$$F_{q,\tau_z} = -\alpha_{q,\tau_z}^2 / (2\beta). \quad (5)$$

The supercurrent carried by the Cooper pairs with momentum  $\mathbf{q}$  is calculated as

$$\mathbf{j} = \frac{2e}{\hbar} \sum_{\tau_z} \nabla_{\mathbf{q}} F_{q,\tau_z} = \frac{2e}{\hbar} \frac{1}{\beta} \sum_{\tau_z} |\alpha_{q,\tau_z}| \nabla_{\mathbf{q}} \alpha_{q,\tau_z}, \quad (6)$$

where  $2e < 0$  is the charge of a Cooper pair, and therefore,  $\mathbf{j}$  is antiparallel to  $\mathbf{q}$ . We parametrize  $\mathbf{q}$  as  $-q(\cos \theta, \sin \theta)$ . The critical current is obtained by maximizing  $|\mathbf{j}|$  with respect to  $q$  for  $q > 0$ , which leads to the following orientation depen-

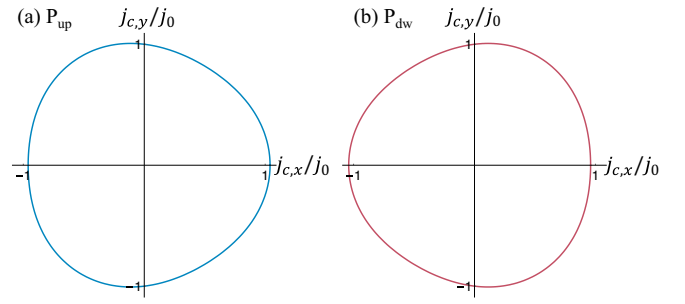


FIG. 4. The angle dependence of the critical current in (a)  $P_{\text{up}}$  state and (b)  $P_{\text{dw}}$  state. The nonreciprocal factor  $\eta$  is taken to be  $0.04\ell$ , where  $\ell$  is +1 for  $P_{\text{up}}$  and  $-1$  for  $P_{\text{dw}}$ , respectively.

dence of the critical current:

$$j_c(\theta) = j_0(1 + \eta \cos 3\theta), \quad (7)$$

where  $j_0$  is the isotropic part of the critical current. In Eq. (7), the nonreciprocal factor  $\eta$  is given by

$$\begin{aligned} \eta &= -\sqrt{\frac{|\alpha_0|}{3}} \frac{\sum_{\tau_z} \kappa_{\tau_z} / \gamma_{\tau_z}}{\sum_{\tau_z} \sqrt{\gamma_{\tau_z}}} \\ &\approx \sqrt{\frac{6}{7}} \frac{31\zeta(5)}{7\pi [\zeta(3)]^{3/2}} \frac{\lambda_w k_F^3 \Delta_{\text{SOC}}^{(\ell)} \Delta_z}{|E_F|^2 k_B T} \sqrt{\frac{T_c - T}{T_c}} \\ &\approx 0.04\ell \frac{\Delta_z}{k_B T} \sqrt{\frac{T_c - T}{T_c}}, \end{aligned} \quad (8)$$

where  $k_F$  is defined as  $\sqrt{2m^*|E_F|}/\hbar$ . The factor  $\eta$  is proportional to  $\Delta_{\text{SOC}}^{(\ell)}$ ,  $\Delta_z$  and  $\lambda_w$ , which characterize, respectively, SOC-induced spin splittings due to inversion symmetry breaking, spin splittings due to time-reversal symmetry breaking, and trigonal warping. The critical current is generally different for opposite directions since  $j_c(\theta) \neq j_c(\theta + \pi)$  for a generic  $\theta$ , as shown in Fig. 4. We assume  $\Delta_z$  is positive for definiteness in the following. In the  $P_{\text{up}}$  state,  $\ell = +1$ , which renders  $\eta > 0$  and  $j_c(0) > j_c(\pi)$ ; therefore, the system is superconducting for a current with a magnitude in the range of  $j_0(1 - |\eta|, 1 + |\eta|)$  along  $+\hat{x}$  direction, but resistive along the  $-\hat{x}$  direction, which leads to the SDE. By contrast,  $\ell = -1$  and  $\eta < 0$  in the  $P_{\text{dw}}$  state; the superconducting direction is changed to the  $-\hat{x}$  direction for a current with a magnitude in the same range. Therefore, the SDE is reversed upon ferroelectric reversal. As shown by Eq. (8),  $\eta$  increases with decreasing  $T$ . If we take  $\Delta_z = k_B T = 0.1 k_B T_c$ ,  $\eta \approx 0.04\ell$ , which represents an experimentally measurable effect [7].

**Conclusions.** In summary, we proposed a microscopic mechanism for ferroelectric reversible SDE using monolayer CuNb<sub>2</sub>Se<sub>4</sub> as a model system. In addition to CuNb<sub>2</sub>Se<sub>4</sub>, we expect ferroelectricity can widely exist in 2D superconductors. In particular, sliding ferroelectricity has been shown to be ubiquitous in 2D vdW stacked layers, where the layer polarization can be switched by the in-plane interlayer sliding [21, 58–63]. Thus, a superconducting vdW bilayer can naturally host coexisting superconductivity and ferroelectricity. A promising candidate is bilayer MoTe<sub>2</sub> in the  $T_d$  structure, which carries the sliding ferroelectricity and becomes superconducting at  $T_c \sim 2\text{K}$  [64]. The bilayer



$T_d - \text{MoTe}_2$  has Rashba spin splittings [65], and an in-plane magnetic field can effectively break the time-reversal symmetry. As in  $\text{CuNb}_2\text{Se}_4$ , ferroelectricity controls the sign of the Rashba spin splittings as well as the SDE in the bilayer  $T_d - \text{MoTe}_2$ . With the recent rapid developments in the study of 2D ferroelectric materials and nonreciprocal superconducting transport, we anticipate that our proposed ferroelectric reversible SDE should soon be experimentally realizable. In a broader prospective, our work establishes a new type of superconductor, in which ferroelectricity acts as a tuning knob in controlling the superconductivity properties. Further theoretical, computational, and experimental works are expected to substantially broaden the material candidates and device functionalities of ferroelectric superconductors.

*Note added.* Controlling of superconductivity through ferroelectricity was recently demonstrated in twisted bilayer graphene aligned with hBN [66].

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