Enhanced Sm spin projection in $Gd_xSm_{1-x}N$

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The rare-earth nitrides form a series of structurally simple *intrinsic* ferromagnetic semiconductors, a rare class of both fundamental interest and application potential. Within the series there is a wide range of magnetic properties relating to the spin/orbit contributions to the ferromagnetic ground states. We report an x-ray magnetic circular dichroism investigation of the spin/orbit magnetic dipole alignments of Sm and Gd ions in epitaxial $Gd_xSm_{1-x}N$ films. The Sm spin-alignment expectation value $\langle S_z \rangle$ is seen to be strengthened by the Gd/Sm exchange interaction, providing guidance that the composition for an angular momentum compensation point, where the volume-averaged total angular momentum of a film is zero, lies near $x = 0.24$.

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I. INTRODUCTION

The lanthanides are the series of the periodic table across which the $4f$ shell is sequentially filled, imbuing them with the strongest angular momentum and magnetic moments among all stable elements. The strong spin-orbit coupling within the $4f$ shell ties the orbital moment firmly to the spin so that the net magnetic moment is of mixed spin and orbital character. Furthermore, the limited extent of the $4f$ wave function reduces the orbital angular momentum quenching that is a feature of transition metal compounds; indeed the orbital moment can dominate the net magnetic moment.

The rare-earth nitrides (*R*N, with *R* a lanthanide element) form in the very simple NaCl structure which provides a fertile ground to investigate the spin/orbit magnetism that stems from the partially full $4f$ electron levels [\[1\]](#page-5-0). The chemical similarity of the lanthanides and the wellmatched lattice constants in their nitrides invites studies in the mixed-cation environments offered by multilayers and solid solutions. Multilayer structures of GdN/SmN and of GdN/NdN have already shown a striking competition between the exchange and Zeeman interactions across interfaces

[\[2–5\]](#page-5-0). The present report investigates the effects of that competition among randomly sited Gd^{3+}/Sm^{3+} ions in the solid solution $Gd_xSm_{1-x}N$ using x-ray magnetic circular dichroism (XMCD).

The magnetism of the parent materials, GdN $(x = 1)$ and SmN $(x = 0)$ illustrates the range of magnetism in the *RN* series. Both are ferromagnetic, with Curie temperatures (T_C) of ∼70 and ∼30 K, respectively, resulting from an indirect exchange mechanism that links the $4f$ alignments via the R 5*d* and N 2*p* states [\[1,6–8\]](#page-5-0). Despite some contribution to the magnetism by electrons in the 5*d* band, the moments of GdN and SmN are overwhelmingly dominated by electrons in the 4 f levels. The Gd^{3+} ions in GdN have a half-filled $4f$ shell ([Xe] $4f^7$), and in the ground state all seven spins align, giving a spin number $S = 7/2$. The half-full shell further features zero orbital angular momentum $(L = 0)$ so that the material behaves as a spin-only ferromagnet with $\langle M_z \rangle = 2\mu_B \langle S_z \rangle = 7\mu_B$ per Gd³⁺ ion. In contrast, the Sm³⁺ ions in SmN possess two fewer electrons, giving an electronic configuration of $[Xe]4f^5$. Interion exchange aligns the 4 *f* spins, but opposing magnetic contributions from the spin and orbital angular momenta very nearly cancel to yield a net magnetization of $\approx 0.035 \mu_B$ per Sm³⁺ ion in the ferromagnetic ground state $[8,9]$ $[8,9]$. Crucially, the magnetic moment $\langle M_z \rangle = \mu_B(\langle L_z \rangle + 2\langle S_z \rangle)$ is weakly orbital dominated, with $\langle L_z \rangle / 2 \langle S_z \rangle < -1$ [\[10\]](#page-6-0), so that the spin contribution to the magnetic moment is opposed to an applied field. It is this opposition of the spin magnetic moment (μ_S) to the applied field that provides a competition between the Zeeman and

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FIG. 1. Schematic showing the alignment of the magnetic moments of Sm^{3+} and Gd^{3+} in (a) the parent materials where the Sm-Gd exchange is absent and (b) the solid solution $Gd_{x}Sm_{1-x}N$ where the Sm-Gd exchange dominates the Zeeman interaction and aligns the spins. Panel (c) shows the net (low-temperature, volume averaged) values of the magnetization (M) and angular momentum (\bar{J}) as the composition (x) interpolates between the parent materials. This schematic refers to to the most simple model and is shown in Fig. [8,](#page-5-0) adapted in light of our XMCD results.

exchange interactions when SmN and GdN are paired in multiple layers [\[3](#page-5-0)[,10\]](#page-6-0).

Similarly to multilayer structures, the Gd-Sm exchange interaction in Gd*x*Sm1−*^x*N disrupts the tendency of the Zeeman interaction to align the Sm^{3+} orbital magnetic moment (μ_L) with an applied field. The alignment (antialignment) of the Gd^{3+} (Sm³⁺) spin magnetic moments with an applied magnetic field [Fig. $1(a)$] is in direct competition with the ferromagnetic interion exchange attempting to align the Gd^{3+} and Sm^{3+} spins [Fig. 1(b)]. Because the Gd^{3+} magnetic moment is two orders of magnitude larger, most values of *x* show the alignment of the the Gd^{3+} spin moment with the applied field. Yet, for extremely dilute concentrations of Gd in solution (that is, $x \approx 0$) it must be that the net Sm³⁺ magnetic moments align with the applied field, rather than the Gd^{3+} , due to the same exchange. It is natural to question how uniform would be the Sm^{3+} spin alignment in this dilute limit, where the many Sm^{3+} ions are farther than a few neighbors separated from the nearest Gd^{3+} ion. For this purpose we rely on the observation that in SmN/GdN mulilayers the Sm^{3+} spins rotate over distances of a few nm $[2,3]$, a typical distance to the nearest Gd^{3+} ion at a concentration of $x \sim 0.005$.

The uniform-alignment model can thus be expected to lose accuracy at sub-1% Gd concentrations.

The simplest model for the magnetic state of $Gd_xSm_{1-x}N$ would treat the Sm^{3+} and Gd^{3+} ion moments as fixed at their values in SmN and GdN. This model suggests that under full ferromagnetic order the saturation magnetization is linear in *x* and for some appropriate composition there exists a net zero magnetization: a magnetization compensation point. Within that model the net magnetization (\bar{M}) ranges from $-0.035\mu_B$ to 7μ _B per *R* ion (with respect to the spin-moment direction in the solid solution) with the compensation point occurring at $x = 0.005$ [Fig. 1(c)]. The XMCD results in this paper indicate that the Sm^{3+} spin-alignment does not remain as adopted in SmN, but nearly doubles in Gd-rich films.

Furthermore, the same behavior seen for the net magnetic moment must also manifest for the net angular momentum, \bar{J} , meaning that under full ferromagnetic order \bar{J} must also pass through zero as *x* varies from 0 to 1, giving an angular momentum compensation point [Fig. $1(c)$]. Here again the simplest model uses a linear interpolation between the ground state angular momenta of GdN and SmN, but unlike the magnetization there is no simple measurement of the angular momenta of the parent materials. The spin-only magnetism in GdN clearly indicates that $J = S = 7/2$, and Hund's rules suggest that $J = L - S = 5/2$ in SmN. However, an atom-centred treatment of the magnetism of SmN based on realistic crystal-field and exchange interactions and reproducing the orbital-dominant ground state moment of $0.035\mu_B$ per Sm^{3+} ion yields an eigenstate of neither L_z or S_z [\[11\]](#page-6-0). Rather, it shows projections $\langle L_z \rangle \approx 2\hbar$ and $\langle S_z \rangle \approx -\hbar$, substantially smaller than the maximum possible values provided by Hund's rules ($\langle L_z \rangle = 5\hbar$ and $\langle S_z \rangle = -5\hbar/2$, treating the spin moment direction as positive, once more). Importantly, the spin-orbit coupling is still strong enough that the spin and orbital angular momenta remain opposed.

This description of the magnetism of the Sm^{3+} ion, in which the $4f$ spins are ferromagnetically aligned in the ground state, but without their maximum potential magnitudes, accommodates the factor of ≈2 increase in the Sm spin polarization that is seen in XMCD measurements performed on GdN/SmN superlattices $[2,11]$ $[2,11]$. We thus introduce a parameter $\alpha = \frac{\langle S_z \rangle}{\langle 5\hbar/2 \rangle}$, which indicates the expectation value of the spin alignment $\langle S_z \rangle$ in the 4*f* shell of the Sm³⁺ ion. Appealing to the ion-based model of McNulty *et al.* [\[11\]](#page-6-0), in homogeneous SmN $\alpha \approx 0.4$, while at the SmN/GdN interface $\alpha \approx 0.6$. One might then anticipate, as seen below, that the effective volume-averaged $α$ increases with Gd concentration *x*. The simple model shown in Fig. 1 illustrates the dependence of the net magnetic moment and angular momentum on *x*, but assumes that α is independent of *x*. In the upcoming discussion we show the equivalent diagram with $\alpha(x)$ as implied by the XMCD results (Fig. [8\)](#page-5-0).

Finally we note that the present system, indeed the *R*N in general, show ferromagnetism only at cryogenic temperatures. Nonetheless the system described here has wider interest than simply the fundamentals; there is a recognized demand for cryogenic spintronics, especially as regards magnetic memory for integration with superconductor interconnects and central processors [\[12–15\]](#page-6-0). The *R*N series' magnetic behavior generally, and specifically the presence

FIG. 2. X-ray diffraction data for the $Gd_{0.76}Sm_{0.24}N$ film. The XRD peak around 31 \degree , corresponding to the Gd_{*x*}Sm_{1−*x*}N (111) crystal direction, is typical of these thin films. The exact location of this peak is dependent on both the Gd/Sm ratio and the incidence of nitrogen vacancies in the crystal. The inset shows a rocking curve of the same Gd*x*Sm1−*^x*N crystallites; the intensity scale is linear. The two peaks on the right of the 2θ scan show peaks corresponding to the AlN buffer layer and the GaN cap.

of such features as magnetization and angular momentum compensation points, are of clear importance in these applications. Note that in these applications it is exactly (i) the net volume-averaged magnetic moment that is of importance for the resulting fringe field, and (ii) the volume-averaged angular momentum that determines the torque required to reverse the magnetization.

II. EXPERIMENTAL DETAILS

Gd*x*Sm1−*^x*N films were grown in a Riber 32p molecular beam epitaxy (MBE) system on 100 nm thick (0001) AlN buffered Si, heated during growth to between 450 and $700\degree$ C depending on the composition. The growth rate was held to 100 nm/h. High purity rare-earth metals were evaporated from conventional effusion cells to achieve the desired composition under nitrogen rich conditions ($P_N \approx$ 2.1×10^{-5} Torr). The composition of the solid-solutions was measured to within $x \pm 0.04$ using x-ray fluorescence (XRF) spectroscopy. X-ray diffraction confirmed (111)-oriented FCC NaCl structure, typical of the rare-earth nitride series (Fig. 2).

Field in-plane SQUID (superconducting quantum interference device) magnetometry was performed in a Quantum Design Magnetic Property Measurements System (MPMS) capable of temperatures as low as 5 K and fields up to 7 Te. These measurements determine the temperature- and hysteretic-field-dependent net magnetic moment across the para- to ferromagnetic transition. In the present system that net magnetic moment is overwhelmingly dominated by the Gd ions, motivating support of the ionic-species resolution capability of XMCD to investigate the Sm^{3+} alignment separately. That alignment relates specifically to the 4 *f* -shell spin and orbital states, suggesting the use of XMCD at the *M*⁴,⁵ edge, which interrogates alignment in the $4f$ shell. However, the surface sensitivity of XAS at the ∼1 keV *M* edge renders it unsuitable for the present study, and we have used instead XMCD at the *L*²,³ edges of Gd and Sm. The hard-x-ray

lanthanide *L* edge measurements probed the full \approx 100 nm thickness of the films.

The *L* edge involves excitation into the 5*d* band, coupled to the $4f$ spin alignment by the strong $5d-4f$ exchange interaction. Thus, we report here the results of XMCD at the Gd and Sm $L_{2,3}$ edges to probe separately the alignment on the two cations. The measurements were performed at temperatures down to 5 K and fields up to 6 T at the BL39XU beamline at the SPring-8 synchrotron in Japan. All XMCD was performed at 10° from grazing incidence.

XMCD at the two edges, $L_{2,3}$, separated by the spin-orbit interaction in the 2*p* core-hole state, can in principle yield information about the spin and orbital alignments in the final 5*d* state, though the failure of the XMCD sum rules for the $L_{2,3}$ edges prevents their use in the present case [\[16\]](#page-6-0). Furthermore, the Sm *L*² edge in these materials is masked by magnetic extended x-ray absorption fine structure (EXAFS) from the Gd L_3 edge [\[17\]](#page-6-0). We thus focus the present discussion on the relatively weak L_3 edge of Sm and the stronger L_2 edge of Gd, interpreted in terms of the temperature- and compositiondependent spin alignments.

III. RESULTS AND DISCUSSION

A. Magnetization in the para- and ferromagnetic phases

To aid the interpretation of the XMCD results, we present field-cooled magnetization and hysteresis loop measurements (Fig. [3\)](#page-3-0) on the same three films for which XMCD was performed. The temperature traces identified for each sample a single ferromagnetic transition between the 30 and 70 K Curie temperatures of SmN and GdN $[7,8,18]$ $[7,8,18]$, as seen in Fig. $3(b)$. The most Gd-rich film $(x = 0.92)$ has the smallest coercive and saturation fields, comparable with epitaxial GdN films. With \approx 1/4 of the Gd³⁺ ions replaced by non-sphericallysymmetric Sm^{3+} there are increases in both the coercive and saturation fields. Both trends, which are further enhanced in the most Sm-rich film, follow from the combination of (i) a weaker Zeeman interaction that results from a reduced saturation magnetization and (ii) a larger crystal anisotropy experienced by the $J = 5/2$ ground state of Sm³⁺. All films show a full alignment above \sim 5 T, dictating that the spectra below were collected at the maximum available XMCD field of 6 T.

B. XMCD

The near-zero magnetic-moment of Sm in the ferromagnetic state of SmN ensures that the magnetization of Fig. [3](#page-3-0) relates directly to the Gd alignment. We thus show first in Fig. [4](#page-3-0) the Gd L_2 -edge XAS (a) and XMCD (b) in the three films taken at saturation, 5 K and 6 T. A single feature dominates here, with the nearly identical magnitudes of the XMCD for different sample compositions providing a reminder that normalizing the XMCD to the XAS signal causes the XMCD to reflect the *alignment* of the magnetic moments rather than their concentration. The few-percent reduction in the most Sm-rich film then signals that the Gd^{3+} ions are less perfectly aligned in that film. This less-than-perfect alignment is discussed further where we present the temperature-dependence of the XMCD of the films.

FIG. 3. (a) Hysteresis loops of the three Gd*x*Sm1−*^x*N films taken at 5 K. The inset shows a zoom of the region close to zero, emphasising the different coercivities of the films. (b) The inverse susceptibility of the three films with fits included to show the different Curie temperatures extracted from the measurements.

Figure [5](#page-4-0) shows Sm *L*³ XAS and XMCD spectra for the three Gd*x*Sm1−*^x*N films, with XMCD taken from a homogeneous SmN film [\[10\]](#page-6-0) shown for comparison in Fig. [5\(b\).](#page-4-0) The spectra in the solid-solution films have signs opposite to that seen for pure SmN, a clear signature of the Zeeman-exchange conflict. As discussed in the Introduction, under solely the Zeeman interaction in SmN the 4f *spin* magnetic moment aligns in opposition to the applied field. This is reversed in the $Gd_{x}Sm_{1-x}N$ films by Gd -Sm exchange that couples the Sm spin alignment to the strongly Zeeman-aligned Gd^{3+} spins. Moreover, the magnitude of the XMCD at the Sm *L*³ in $Gd_xSm_{1-x}N$ clearly increases with the value of *x*, indicating the increasing alignment of $\langle S_z \rangle$ in films with increasing Gd content.

A subtle contrast between the $Gd_{x}Sm_{1-x}N$ and the SmN XMCD [Fig. $5(b)$] is an enhanced strength across the range 6.715–6.720 keV in the present films, seen most strongly in the most Gd-rich film $(x = 0.92)$. Significantly, even in SmN the weak signal at those energies was not predicted by an XMCD-based calculation [\[10\]](#page-6-0). It is, however, exactly the range across which there are XAS spectral lines from Sm^{2+} [\[19–21\]](#page-6-0), suggesting that their presence here could relate to a low concentration of divalent Sm. The signal is exceedingly weak, \approx 1% of the XAS signal at the normalization point (\approx 6.735 keV), so it is no surprise that there appears no

FIG. 4. Comparison of XMCD spectra for the three Gd*x*Sm1−*^x*N films at the Gd L_2 edge at 5 K. All measurements were cooled to 5 K in an applied field of 6 T. Panel (a) shows the average of the x-ray absorption measurements for left- and right-circularly-polarized light, normalized to the continuum edge jump. Each subsequent dataset is offset by 0.5. Panel (b) shows the XMCD, defined as the difference between the left- and right-circularly-polarized absorption. Each subsequent dataset is offset by 10%.

clear signature of this feature in the XAS spectra. In order to limit interference from the possible divalent signal in our interpretation of the Sm^{3+} XMCD signal strength, for our temperature-dependent discussion below we have chosen to use the XMCD magnitude across the 6.722–6.7256 keV transition as a measure of the spin alignment in the Sm^{3+} final states.

The feature that appears at ≈ 6.709 keV in Fig. [5\(b\)](#page-4-0) is also not predicted in a calculated spectrum, and it lies at an energy that would imply an empty state deep within the valance band, well below the Fermi energy. It was previously identified as an electric quadrupole (labelled EQ in Fig. 5) $2p$ to $4f$ transition $[10]$, in which the final $4f$ state appears, as expected within XAS, at the $4f$ energy reduced in the presence of the $2p$ core hole $[16]$. Such features have been attributed to electric quadrupole absorption in many rare-earth insulators [\[22\]](#page-6-0), and this attribution has been confirmed in measurements of dysprosium using angular resolved XAS [\[23\]](#page-6-0). It is natural to regard this feature in XMCD spectra as proportional to the spin alignment within the Sm^{3+} 4*f* shell, and indeed it does strengthen across the three films, though it is not quite commensurate in strength with the 5*d* signals (Fig. [6\)](#page-4-0). An analogous situation is seen also in SmN/GdN superlattices, and signals that the spin alignment of the Sm^{3+} 5*d* states is enhanced by neighboring Gd^{3+} ions. The contraction of the radial $5d$ wave function that stems from $4f - 5d$ exchange interaction leads to an enhancement of the $2p \rightarrow 5d$ transition

FIG. 5. Comparison of XMCD spectra for the three Gd_xSm_{1−*x*}N films at the $Sm L_3$ edge at 5 K. All measurements were cooled to 5 K in an applied field of 6 T. Panel (a) shows the average of the x-ray absorption measurements for left- and right-circularly-polarized light, normalized to the continuum edge jump. Each subsequent dataset is offset by 0.5. Panel (b) shows the XMCD, defined as the difference between the left- and right-circularly-polarized absorptions. Each subsequent dataset is offset by 2%. Data taken for a SmN film in a previous study by Anton *et al.* [\[10\]](#page-6-0) are included.

 $[16,24,25]$, so it is no surprise to see that the increased spin alignment of the 4 *f* states is reflected with a more pronounced intensity in the electric dipole absorption channel. It is this same contraction that prevents the use of the usual XMCD sum rules to separate the orbital and spin contributions to the magnetic moment [\[16,26\]](#page-6-0).

The influence of the solid-solution composition on exchange interactions is also seen in the temperature-dependent XMCD strengths, shown in Fig. 7 as the spin alignments inferred by the Gd L_2 and the Sm L_3 edges. Looking first at Gd, there is a factor of 2 drop of the 80 K alignment from that at low-temperature in both of the films with high Gd concentration. Recalling that these films show a Curie temperature in the range of 60–70 K, and that the XMCD was performed at a field of 6 T, there is no surprise that the alignment remains strong to this temperature. In contrast, the film with only 0.21 Gd shows a much more rapid drop. The Gd concentration for this film is within uncertainty of the site percolation threshold on the fcc lattice $[27]$; and this suggests the potential for a reliance in this film on the weaker Gd-Sm-Gd exchange for full ferromagnetic spin alignment on

FIG. 6. Magnitude of the Sm *L*³ absorption features as a function of the Gd fraction, *x*. Panel (a) shows the magnitude for the conventional Sm^{3+} absorption channel. Panel (b) shows the magnitude of the electric quadrupole absorption. The star marker represents the data taken from Anton *et al.* [\[10\]](#page-6-0).

FIG. 7. Strengths of the (a) $Gd^{3+} L_2$ -edge and (b) $Sm^{3+} L_3$ -edge features measured using XMCD as a function of temperature. Error bars are smaller than the markers where not visible.

FIG. 8. Model of the net (volume-averaged) angular momentum, \bar{J} , as a function of Gd fraction, \bar{x} . The simple model is a linear interpolation of total angular momentum between $\langle J_z \rangle = -1$ (\hbar per *R* ion) for SmN (where $\langle L_z \rangle = -2\hbar$ and $\langle S_z \rangle = \hbar$) and $\langle J_z \rangle = J = 7/2$ for GdN. Taking into account the variation of $\alpha(x)$ implied by Fig. [6](#page-4-0) alters the net angular momentum, shifting the compensation point.

the Gd ions.The dominance of the exchange interaction over the Zeeman interaction on the Sm ions leads to a situation in which the temperature dependence of the Sm alignment follows that of Gd ions rather closely in all films.

Returning to the results of Fig. [6,](#page-4-0) a consequence of the varying Sm^{3+} spin alignment is its influence on the Gd concentration of the angular momentum compensation point of Fig. [1\(c\).](#page-1-0) That compensation occurs at $x \approx 0.2$ for the $\langle S_z \rangle \approx$ \hbar found in SmN, but the stronger Sm³⁺ alignment in these solid solutions shifts the compensation point to higher Gd concentration. A quantitative interpretation of the shift relies on a measure of the enhancement of the 4*f* spin alignment $\langle S_z(x) \rangle$, for which we rely on the increase of the $4f$ quadrupole feature at ≈ 6.709 ≈ 6.709 ≈ 6.709 keV shown in Fig. 6. Those data imply that the magnitude of the Sm^{3+} electric quadrupole feature is enhanced by \sim 1.5 between *x* = 0 and *x* → 1. The strong spin-orbit interaction ensures the Sm^{3+} net angular momentum (which is negative with respect to the spin direction, due to the orbital-dominant magnetism) will also increase linearly with *x*, leading ultimately to the shift of the compensation point by ≈ 0.02 as shown in Fig. 8.

- [1] F. Natali, B. J. Ruck, N. O. Plank, H. J. Trodahl, S. Granville, C. Meyer, and W. R. Lambrecht, [Prog. Mater. Sci.](https://doi.org/10.1016/j.pmatsci.2013.06.002) **58**, 1316 (2013).
- [2] J. F. McNulty, E.-M. Anton, B. J. Ruck, F. Natali, H. Warring, F. Wilhelm, A. Rogalev, M. M. Soares, N. Brookes, and H. J. Trodahl, Phys. Rev. B **91**[, 174426 \(2015\).](https://doi.org/10.1103/PhysRevB.91.174426)
- [3] E.-M. Anton, W. Holmes-Hewett, J. McNulty, F. Natali, F. Bramley, Y. Choi, D. Haskel, B. J. Ruck, and H. J. Trodahl, AIP Adv. **11**[, 015348 \(2021\).](https://doi.org/10.1063/9.0000099)
- [4] E.-M. Anton, S. Granville, A. Engel, S. V. Chong, M. Governale, U. Zülicke, A. G. Moghaddam, H. J. Trodahl, F. [Natali, S. Vézian, and B. J. Ruck,](https://doi.org/10.1103/PhysRevB.94.024106) Phys. Rev. B **94**, 024106 (2016).

IV. CONCLUSIONS

We have probed the alignment of the moments in $Gd_xSm_{1-x}N$ by *L*-edge XMCD tracing out the spin alignments $\langle S_z \rangle$ independently on the Sm³⁺ and Gd³⁺ ions. Special care was taken to follow the XMCD signal amplitudes and the spin alignments $\langle S_z \rangle$ of the two ionic species as a function of the composition across the GdN/SmN solid solutions.

The XMCD study was supported by thorough magnetization studies of the films. It is noted that the magnetization is overwhelmingly dominated by the Gd^{3+} spin alignment, since the magnetic moment on the Sm^{3+} is more than two orders of magnitude smaller. Magnetic measurements revealed the temperature-dependent magnetization as well and complete hysteresis curves in the ferromagnetic phase that were used to inform XMCD measurements and limit those to magnetic fields adequate to ensure saturated magnetization. There was excellent agreement between the magnetization and Gd $\langle S_z \rangle$, but it is only XMCD that is capable of revealing the spin alignment $\langle S_z \rangle$ on the Sm³⁺ ions. One important outcome of the study is that the Sm³⁺ ions experience a factor of \sim 2 enhanced spin alignment as a result of the Gd/Sm exchange interaction.

The sign of the Sm XMCD, opposite to that in SmN, immediately identifies that the alignment of the Sm and Gd spins by the Sm-Gd exchange interaction dominates the Sm^{3+} Zeeman interaction, which on its own acts to align the *orbital* magnetic moment in SmN $[2,10]$ $[2,10]$. The data further show that the exchange dominance strengthens the Sm^{3+} spin alignment in the presence of Gd³⁺ ions; the projection $\langle S_z \rangle \approx \hbar$ in SmN, rising to $\langle S_z \rangle \approx 3\hbar/2$ for isolated Sm³⁺ ions in a predominantly GdN host. The composition-dependent alignment specifies that the composition at which an angular-momentum compensation can be expected is at higher values of *x* than implied by measurements of pure SmN films. The shift is significant but relatively small, from 0.22 to ∼0.24.

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- [5] J. F. McNulty, E.-M. Anton, B. J. Ruck, M. Suzuki, M. [Mizumaki, and H. J. Trodahl,](https://doi.org/10.1103/PhysRevB.100.094441) Phys. Rev. B **100**, 094441 (2019).
- [6] F. Hulliger, [J. Magn. Magn. Mater.](https://doi.org/10.1016/0304-8853(78)90121-X) **8**, 183 (1978).
- [7] S. Granville, B. J. Ruck, F. Budde, A. Koo, D. J. Pringle, F. Kuchler, A. R. H. Preston, D. H. Housden, N. Lund, A. Bittar, [G. V. M. Williams, and H. J. Trodahl,](https://doi.org/10.1103/PhysRevB.73.235335) Phys. Rev. B **73**, 235335 (2006).
- [8] C. Meyer, B. J. Ruck, J. Zhong, S. Granville, A. R. H. Preston, [G. V. M. Williams, and H. J. Trodahl,](https://doi.org/10.1103/PhysRevB.78.174406) Phys. Rev. B **78**, 174406 (2008).
- [9] [R. Moon and W. Koehler,](https://doi.org/10.1016/0304-8853(79)90135-5) J. Magn. Magn. Mater. **14**, 265 (1979).
- [10] E.-M. Anton, B. J. Ruck, C. Meyer, F. Natali, H. Warring, F. [Wilhelm, A. Rogalev, V. N. Antonov, and H. J. Trodahl,](https://doi.org/10.1103/PhysRevB.87.134414) Phys. Rev. B **87**, 134414 (2013).
- [11] [J. F. McNulty, B. J. Ruck, and H. J. Trodahl,](https://doi.org/10.1103/PhysRevB.93.054413) Phys. Rev. B **93**, 054413 (2016).
- [12] [D. S. Holmes, A. L. Ripple, and M. A. Manheimer,](https://doi.org/10.1109/TASC.2013.2244634) **IEEE Trans.** Appl. Supercond. **23**, 1701610 (2013).
- [13] M. A. Manheimer, [IEEE Trans. Appl. Supercond.](https://doi.org/10.1109/TASC.2015.2399866) **25**, 1 (2015).
- [14] M.-H. Nguyen, G. J. Ribeill, M. V. Gustafsson, S. Shi, S. V. Aradhya, A. P. Wagner, L. M. Ranzani, L. Zhu, R. Baghdadi, B. Butters, E. Toomey, M. Colangelo, P. A. Truitt, A. Jafari-Salim, D. McAllister, D. Yohannes, S. R. Cheng, R. Lazarus, [O. Mukhanov, K. K. Berggren](https://doi.org/10.1038/s41598-019-57137-9) *et al.*, Sci. Rep. **10**, 248 (2020).
- [15] [E. Garzón, A. Teman, and M. Lanuzza,](https://doi.org/10.3390/electronics11010061) Electronics **11**, 61 (2022).
- [16] J.-C. Parlebas, K. Asakura, A. Fujiwara, I. Harada, and A. Kotani, [Phys. Rep.](https://doi.org/10.1016/j.physrep.2006.05.002) **431**, 1 (2006).
- [17] G. Schütz and D. Ahlers, in *Spin-Orbit-Influenced Spectroscopies of Magnetic Solids* (Springer, Berlin, 1996), pp. 229–257.
- [18] B. Ludbrook, I. Farrell, M. Kuebel, B. J. Ruck, A. Preston, H. J. [Trodahl, L. Ranno, R. Reeves, and S. Durbin,](https://doi.org/10.1063/1.3211290) J. Appl. Phys. **106**, 063910 (2009).
- [19] [E. Beaurepaire, J. P. Kappler, and G. Krill,](https://doi.org/10.1103/PhysRevB.41.6768) Phys. Rev. B **41**, 6768 (1990).
- [20] N. Yoshikane, K. Matsui, T. Nakagawa, A. G. Terzidou, Y. Takabayashi, H. Yamaoka, N. Hiraoka, H. Ishii, J. Arvanitidis, and K. Prassides, [Mater. Chem. Front.](https://doi.org/10.1039/D0QM00707B) **4**, 3521 (2020).
- [21] N. Yoshikane, T. Nakagawa, K. Matsui, H. Yamaoka, N. [Hiraoka, H. Ishii, J. Arvanitidis, and K. Prassides,](https://doi.org/10.1016/j.jpcs.2020.109822) J. Phys. Chem. Solids **150**, 109822 (2021).
- [22] C. Neumann, B. Hoogenboom, A. Rogalev, and J. Goedkoop, [Solid State Commun.](https://doi.org/10.1016/S0038-1098(99)00089-7) **110**, 375 (1999).
- [23] J. C. Lang, G. Srajer, C. Detlefs, A. I. Goldman, H. König, X. [Wang, B. N. Harmon, and R. W. McCallum,](https://doi.org/10.1103/PhysRevLett.74.4935) Phys. Rev. Lett. **74**, 4935 (1995).
- [24] F. Baudelet, C. Giorgetti, S. Pizzini, C. Brouder, E. Dartyge, A. [Fontaine, J. Kappler, and G. Krill,](https://doi.org/10.1016/0368-2048(93)80012-B) J. Electron Spectrosc. Relat. Phenom. **62**, 153 (1993).
- [25] [B. Harmon and A. Freeman,](https://doi.org/10.1103/PhysRevB.10.1979) Phys. Rev. B **10**, 1979 (1974).
- [26] P. Carra, B. N. Harmon, B. T. Thole, M. Altarelli, and G. A. Sawatzky, [Phys. Rev. Lett.](https://doi.org/10.1103/PhysRevLett.66.2495) **66**, 2495 (1991).
- [27] S. C. van der Marck, Phys. Rev. E **55**[, 1514 \(1997\).](https://doi.org/10.1103/PhysRevE.55.1514)