Role of magnetoelastic coupling and magnetic anisotropy in MnTiO₃

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We report the thermodynamic properties studied by thermal expansion, magnetostriction, magnetization, and specific heat measurements as well as the low-energy magnetic excitations of MnTiO₃ and investigate how magnetoelastic coupling and magnetic anisotropy affect the evolution of long-range order and the magnetic phase diagram. Specifically, we utilize high-resolution capacitance dilatometry and antiferromagnetic resonance (AFMR) studies by means of high-frequency electron spin resonance (HF-ESR) spectroscopy. The role of anisotropy is reflected by spin reorientation at $B_{SF} \simeq 6$ T and a corresponding sign change in $\partial T_N / \partial B$. Analysis of the AFMR modes enables us to establish the zero-field excitation gap Δ as well as its temperature dependence. We derive the effective anisotropy field $B_A = 0.16(1)$ T which predominately originates from out-of-plane nearest-neighbor magnetic dipole interactions. Despite the nearly fully quenched orbital moment, our data show pronounced thermal expansion and magnetostriction anomalies at T_N and B_{SF} , respectively, which allows the experimental determination of sizable uniaxial pressure dependencies, i.e., $\partial B_{SF} / \partial p_c = -0.20(2)$ T/GPa, $\partial T_N / \partial p_b = 0.69(12)$ K/GPa, and $\partial T_N / \partial p_c = -2.0(4)$ K/GPa. While in the thermodynamic properties the presence of short-range magnetic order up to $\simeq 170$ K is detected by anisotropic lattice distortion and by the violation of a constant Grüneisen behavior, the presence of local magnetic fields is seen in ESR at even higher temperatures of at least $3T_N$.

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I. INTRODUCTION

Ilmenite-structured titanates $M \text{TiO}_3$ (M = Mn, Fe, Ni, Co) have been studied for nearly a century, with the observation of significant magnetoelectric effects in CoTiO₃, NiTiO₃ [1], and MnTiO₃ [2], placing them into the class of multiferroics and reviving interest in fundamental research on them. Besides potential applications in multiferroic technology, the ilmenite titanates show promising applications as photocatalysts [3–6], in solar cells [7,8], and as sensors [9–11]. Furthermore, interesting effects, such as NiTiO₃ thin films showing polarization-induced ferromagnetism [12] and the existence of Dirac magnons in CoTiO₃, have been reported recently [13].

In MnTiO₃, G-type long-range antiferromagnetic order evolves at $T_N = 64$ K and a linear magnetoelectric effect is observed as demonstrated by anomalies in the dielectric function ϵ for B > 0 T in the ordered phase and the appearance of finite electric polarization [2,14–18]. The material crystallizes in the space group $R\bar{3}$ where 2/3 inhabited hexagonal layers of Mn²⁺ and Ti⁴⁺ ions surrounded by distorted oxygen octahedra are stacked [19]. Notably, the magnetic susceptibility shows a broad maximum around 100 K which is attributed to in-plane two-dimensional short-range order persisting at least up to $2T_N$. This is supposed to result from accidental cancellation of the interlayer exchange interactions [14,15,20]. Below T_N , uniaxial anisotropy favors the easy magnetic axis along the crystallographic *c* direction and a spin-flop transition appears at $B_{SF}||c = 6T$ [21,22]. It has been recently shown that spin reorientation is accompanied by a flop in electric polarization *P* from the *c* axis to the *a* axis [23].

Here, we report the interplay of spin and lattice in MnTiO₃, and the role of small but finite magnetic anisotropy. We apply high-resolution capacitance dilatometry to obtain thermal expansion and magnetostriction up to 15 T on MnTiO₃ single crystals which are supported by specific heat and magnetization data. The thermodynamic response functions are combined with antiferromagnetic resonance studies of the q = 0 magnon excitations. Our experimental data enable us to quantify and elucidate the magnetoelastic coupling and the role of magnetic anisotropy. In addition, the magnetic phase diagram is established.

II. EXPERIMENTAL METHODS

Single crystals of MnTiO₃ were grown using the floatingzone technique in a high-pressure optical furnace (HKZ, SciDre) equipped with a 3500-W Xe arc lamp [24]. The precursor for crystal growth was prepared via a standard solid-state reaction of stoichiometric amounts of MnCO₃ and TiO₂, performed at 1200 °C in air. Macroscopic single crystals were grown in the argon atmosphere (flow rate 0.5 l/min) at an elevated pressure of 5 bar. Applying moderate pressure proved necessary to avoid an Mn₃O₄ impurity phase which appears with an estimated fraction of <1% upon growth under

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ambient pressure as detected by our magnetization studies [25]. In contrast, the crystal grown at 5 bar used for our study does neither show any detectable hysteresis in the magnetization curves, M(B), at low temperature, nor of any feature around \simeq 42 K in the magnetic susceptibility, $\chi(T)$, which would indicate the presence of small fractions of Mn₃O₄. The growth rate was maintained at 5 mm/h and the feed and seed rods were counter-rotated at 10 rpm. Phase purity of the resulting single crystals was confirmed using powder x-ray diffraction on a Bruker D8 Advance ECO diffractometer. Laue diffraction shows high crystallinity and was used to orient the single crystal. For the measurements reported at hand, a cuboid-shaped single crystal of dimensions of about $1.2 \times 2.1 \times 2.1 \text{ mm}^3$ was used (see Fig. S1 in Ref. [26]).

The magnetization was investigated in the temperature regime 2-300 K and in magnetic fields up to 14 T using a vibrating sample magnetometer (VSM) of Quantum Design's Physical Property Measurement System (PPMS-14). The specific heat was studied in the same device using a relaxation method. The relative length changes dL_i/L_i were studied by means of a three-terminal high-resolution capacitance dilatometer in a home-built setup [27,28], and the thermal expansion coefficients $\alpha_i = 1/L_i \times dL_i(T)/dT$ were obtained. The magnetostriction, i.e., the field-induced length change $dL_i(B)/L_i$, was measured at several fixed temperatures in magnetic fields up to 15 T, and the longitudinal magnetostriction coefficients $\lambda_i = 1/L_i \times dL_i(B)/dB$ were derived. In all dilatometric studies, the magnetic field was applied along the direction of the measured length changes. HF-ESR studies were performed in the frequency range 30 GHz $\leq \nu \leq$ 420 GHz utilising unpolarized continuous-wave microwave radiation generated by a millimetre-wave vector network analyser from AB Millimetre [29]. Transmission amplitude and phase of the microwave radiation were detected as a function of static magnetic field at a fixed microwave frequency and sample temperature by means of a lock-in amplifier. Moreover, X-Band frequency measurements were performed on a Bruker Elexsys E500 spectrometer. While all the abovementioned studies were performed on the same single crystal, auxiliary HF-ESR measurements were conducted also on a finely-ground polycrystalline sample fixed by means of eicosane.

III. RESULTS

A. Evolution of antiferromagnetic order and magnetostructural coupling at B = 0 T

The specific heat displayed as c_p/T in Fig. 1(a) shows a λ -like anomaly at $T_N = 64(1)$ K which signals a second order phase transition associated with the evolution of long-range antiferromagnetic order in MnTiO₃. This is in accordance with previous studies on MnTiO₃ [14,17,22]. Corresponding clear features in the relative length changes dL_i/L_i and the thermal expansion coefficient α_i , i.e., a kink and a λ -shaped peak, respectively, for the crystallographic directions i = b, c [Figs. 1(b) and 1(c)] show the existence of pronounced magnetoelastic coupling in MnTiO₃. Differing signs of the anomalies in α_i imply opposite uniaxial pressure dependencies of T_N , specifically a positive dependence for pressure



FIG. 1. (a) Temperature dependence of the specific heat c_p/T in zero magnetic field. The solid red line indicates the phonon specific heat $c_p^{\rm ph}$ (see the text). The vertical dashed line marks $T_{\rm N}$. Insets: Specific heat c_p at selected applied magnetic fields B||c axis around the Néel temperature and magnetic entropy obtained by integrating $(c_p - c_p^{\rm ph})/T$. (b) Relative length changes dL_i/L_i along the crystallographic *b* and *c* axes vs temperature. The data have been shifted vertically such, that they overlap at 200 K. (c) Corresponding thermal expansion coefficients α_i .

applied along the *b* axis and negative for p||c axis. The volume thermal expansion, approximated by $\beta = (\alpha_c + 2 \times \alpha_b)$, shows only a small (negative) anomaly in its temperature derivative, corresponding to negative hydrostatic pressure dependence of $T_{\rm N}$.

In order to obtain the total magnetic entropy changes $S_{\text{mag}}(T)$, we approximate the phononic specific heat c_p^{ph} with an Einstein-Debye model in a temperature regime well above the Néel temperature (i.e., above 150 K). The fit yields the characteristic temperatures $\Theta_D = 508$ K and $\Theta_E = 1254$ K and describes the data well for temperature above 140 K [see the red line in Fig. 1(a)]. Integrating $(c_p - c_p^{\text{ph}})/T$ yields the magnetic entropy changes [right inset of Fig. 1(a)], which at T > 140 K nearly agree to the full expected magnetic entropy $S_{\text{mag}}^{\text{th}} = R \ln 6 = 14.9$ J/(mol K) associated with the S = 5/2 spin system. At T_N , only 2/3 of the total magnetic entropy changes above T_N . This finding confirms the presence of significant



FIG. 2. Distortion parameter $\delta = (dL_c - dL_b)/(dL_c + dL_b)$ as a function of temperature. (Inset) Temperature derivative of the distortion parameter $\partial \delta/\partial T$.

short-range order above T_N which has been suggested to be of two-dimensional nature [14,15,20].

Thermal expansion of the b and c axis is very similar at high temperatures as shown in Fig. 1(b) but becomes anisotropic below ~ 170 K.¹ While the *b* axis shrinks upon cooling in the whole temperature regime under study, the c axis displays a minimum around $T \simeq 85$ K below which it expands towards lowest temperatures. This is reflected by the sign of the thermal expansion coefficients with $\alpha_{\rm b} >$ 0 and a sign change in α_c at 85 K [Fig. 1(c)]. In addition to the anomalies at T_N , α_c exhibits a broad minimum around 30 K which is not observed in α_b and whose origin is unclear. Note that the absence of a clear associated feature in the magnetic susceptibility (Fig. S2 of Ref. [26]) and its rather field-independent nature (Fig. S4 in Ref. [26]) do not support a magnetic origin of this feature. The anisotropy in thermal expansion is described by a distortion parameter $\delta = (dL_c - dL_b)/(dL_c + dL_b)$ shown in Fig. 2. Upon cooling, δ evolves finite values and gradually increases below ~ 170 K. At low temperatures, structural distortion saturates and becomes almost constant below 20 K. At T_N , δ exhibits a moderate anomaly which appears λ -shaped in the temperature derivative $\partial \delta / \partial T$ (see the inset of Fig. 2). It is straightforward to attribute the observed distortion to magnetoelastic coupling and spin degrees of freedom driving anisotropic thermal expansion both, in the long-range (below T_N) and short-range spin ordered regimes.

Further information on the evolution of the lowtemperature phase is obtained by comparing the entropy and the lattice changes through the (uniaxial) Grüneisen parameter



FIG. 3. Uniaxial Grüneisen ratio $\gamma_i = \alpha_i/c_p$ for i = b and c axis. (Inset) Thermal expansion coefficients α_i (left ordinate) and specific heat c_p (right ordinate) scaled such that the anomalies at T_N overlap, with α_c being multiplied with -0.7.

 $\gamma_i = c_p / \alpha_i$. Here, we consider the experimentally obtained total thermal expansion and specific heat, since our data do not allow us to extract the purely magnetic length changes because a reliable phonon background could not be obtained. The uniaxial Grüneisen parameters (see Fig. 3) are approximately constant at high temperatures, thereby indicating a single dominating energy scale [30,31]. This agrees with the fact that entropy changes in this temperature regime are attributed to phonons [see Fig. 1(a)]. The corresponding Grüneisen ratio amounts to $\gamma = 1.05(9) \times 10^{-7} \text{ mol/J}.$ The relevance of an additional (magnetic) degree of freedom below ~ 170 K yields a nonconstant behavior for both axes with opposite signs. For example, γ_b increases towards the onset of long-range antiferromagnetic order and exhibits an anomaly at the transition, followed by a nearly temperatureindependent regime down to \sim 22 K. In contrast, γ_c exhibits a more pronounced anomaly at $T_{\rm N}$ and varies more strongly upon cooling. We conclude that the lattice changes along the b axis are dominated by one magnetic energy scale at 20 K $\leq T < T_{\rm N}$, while the behavior of the c axis is driven by more than one degree of freedom. This is further illustrated in the inset of Fig. 3, where α_c shows a generally different behavior below $T_{\rm N}$ as compared to $c_{\rm p}$, while $\alpha_{\rm b}$ can be scaled to $c_{\rm p}$ by a single factor.

By exploiting the Ehrenfest relation $dT_N/dp_i = T_N V_m \Delta \alpha_i / \Delta c_p$, the uniaxial pressure dependencies of T_N can be derived from the anomalies. In order to acquire the anomaly heights in c_p and α_i , a method of equal-area construction [32] was applied (see Fig. S3 in Ref. [26]). For MnTiO₃ this method leads to $\Delta c_p = 9.0(16)$ J/(mol K), $\Delta \alpha_b = 3.0(5) \times 10^{-6}$ 1/K, and $\Delta \alpha_c = -7.5(13) \times 10^{-6}$ 1/K. Considering the molar volume $V_m = 3.2 \times 10^{-5}$ m³/mol yields the uniaxial pressure dependencies $\partial T_N / \partial p_b = 0.69(12)$ K/GPa and $\partial T_N / \partial p_c = -2.0(4)$ K/GPa for uniaxial pressure along the *b* and *c* axes, respectively.

¹The *c*-axis length changes have been shifted by $dL/L = 1.05 \times 10^{-3}$ in order to illustrate the similar temperature dependence at high temperatures.



FIG. 4. (a) Magnetization M(B) and (b) magnetostriction $dL_i(B)/L_i$ for B||b and B||c axis, respectively, at 2 K. (Inset) Corresponding isothermal susceptibility $\partial M/\partial B$ for B||c.

B. Magnetostriction and spin reorientation

The isothermal magnetization *M* at 2 K for magnetic fields up to 14 T applied along the *b* and *c* axis is plotted in Fig. 4(a). For *B*||*b* axis, the magnetization shows a linear behavior with $\chi_b = \partial M/\partial B = 0.021(1) \mu_B/(f.u. T)$. In contrast, for *B*||*c* a jump in *M* is observed at $B_{SF} = 5.9(1)$ T, separating linear behavior in *M* vs. *B* with $\chi_c \simeq \chi_b$ in high fields from $\chi_c \simeq 0$ for $B < B_{SF}$. These observations are typical for a spin-flop transition where the spins rotate from the magnetically easy *c* axis into the *ab* plane. From the magnetization jump at B_{SF} , which amounts to $\Delta M = 0.12(1) \mu_B/f.u.$, and from the ordered magnetic moment of 4.55 $\mu_B/f.u.$ [16], we infer that for $B \gtrsim B_{SF}$ the spins are predominately in the *ab* plane with only a small canting of $\simeq 1.5^{\circ}$ towards the *c* axis.

Notably, the spin reorientation is associated with a discontinuous decrease of the *c* axis [Fig. 4(b)], too, which confirms the relevance of spin-orbit coupling in MnTiO₃ and shows that the transition is not a pure spin phenomenon. We also note that in the linear regimes in *M* versus *B*, where the spins are approximately perpendicular to *B*, there is a quadratic behavior of lattice parameters *b* and *c*. Specifically, for $B||c > B_{SF}$, the *c* axis shrinks quadratically in *B*, while for B||b the *b* axis elongates up to highest fields.

The temperature dependence of the spin-flop-related anomalies driven by B||c is shown in Fig. 5 which displays the magnetostriction coefficient λ_c and the isothermal susceptibility χ_c . Upon heating, the anomalies shift to higher fields and become smaller, i.e., the associated jumps in magnetization and length decrease. At 60 K, just below T_N , we find $B_{SF} = 6.8(1)$ T. Above the Néel temperature no anomalies are observed. Thermodynamically, the observed finite slope of the phase boundary B_{SF} (T), in the presence of significant jumps ΔM , implies non-negligible entropy



FIG. 5. (a) Magnetostriction coefficient λ_c and (b) isothermal susceptibility $\chi_c = \partial M / \partial B$ at various temperatures for B||c axis.

changes associated with the phase transition. By exploiting the Clausius-Clapeyron equation and utilising the initial slope of $B_{\rm SF}(T)$ of $\simeq 0.9/{\rm mT/K}$, the observed ΔM can be used to determine the associated entropy changes of $\Delta S_{\rm SF} = |\Delta M \times (\partial T/\partial B)^{-1}| \simeq 6(1) {\rm mJ/(mol K)}$ [33]. Upon heating, this analysis yields an increase of $\Delta S_{\rm SF}$ by a factor of about 2.5, at 20 K, while it then continuously decreases upon further heating up to $T_{\rm N}$ (see details in Table S1 of Ref. [26]). Sizeable entropy changes are not expected for a pure spin-flop transition and in this respect its observation is in accordance with the observed discontinuous lattice changes at $B_{\rm SF}$ [cf. Fig. 4(b)].

The fact that the lattice responds to spin reorientation implies that uniaxial pressure can be used to shift the associated phase boundary. Qualitatively, the data show that uniaxial pressure p||c axis initially yields a decrease of $B_{\rm SF}$. For $p_c \rightarrow 0$, the uniaxial pressure dependence is associated with the jumps ΔM_c and ΔL_c in magnetization and length through the Clausius-Clapeyron relation $\partial B_{\rm SF}/\partial p_c = V_m/L_c \times \Delta L_c/\Delta M_c$. At T = 2 K, the jumps amount to $\Delta M_c = 0.12(1) \mu_{\rm B}/f.u.$ and $\Delta L_c/L_c = -4.2(4) \times 10^{-6}$, which leads to a uniaxial pressure dependence for p||cof $\partial B_{\rm SF}/\partial p_c = -0.20(2)$ T/GPa. For higher temperatures, this value only changes slightly and at T = 40 K we find $\partial B_{\rm SF}/\partial p_c = -0.16(2)$ T/GPa.

C. Antiferromagnetic resonance

ESR studies are used to investigate the paramagnetic response as well as the evolution of local magnetic fields and the long-range ordered phase. While in a paramagnetic regime ESR measurements provide information, e.g., on the evolution of short-range magnetic order, in the long-range antiferromagnetically ordered phase HF-ESR is susceptible to collective q = 0 spin excitations, i.e., it detects AFMR modes.

The room temperature X-Band data demonstrate $g_{||c} = 2.004(1)$ and $g_{\perp c} = 2.002(1)$, as expected for paramagnetic, uncorrelated Mn²⁺ ions with S = 5/2 and vanishing orbital momentum (see Fig. S5 in Ref. [26]) [34]. Decreasing the temperature results in a shift of the resonance field away from the $g \simeq 2.00$ resonance position, as attested by our high-frequency measurements. In particular, the resonance



FIG. 6. Temperature evolution of the resonance spectra recorded at fixed frequency v = 54.2 GHz and B||c axis. $v_{||}^{L}$ and $v_{||}^{SF}$ mark the lower parallel branch and the spin-flop branch, respectively (see the text). Curved dashed lines are a guide to the eye illustrating the evolution of the resonance features.

spectrum obtained at T = 200 K and $\nu = 54.2$ GHz on the single-crystal sample with B||c axis exhibits a sharp paramagnetic Lorentzian resonance signal with a small effect of wave-phase mixing and with corresponding g-factor g = 1.94(4) (see Fig. 6). The deviation from the g = 2.00resonance position at T = 200 K indicates the presence of short-range magnetic correlations at temperatures as high as $T \simeq 3T_{\rm N}$. Upon further cooling, the paramagnetic resonance feature broadens and shifts to higher magnetic fields, which clearly demonstrates a further development of the short-range magnetic order. Below $T_{\rm N}$, the single paramagnetic resonance feature is found to split, drastically broaden and shift, indicating the evolution of static internal fields. At 2 K, two distinct AFMR modes, marked $\nu_{\rm II}^{\rm L}$ and $\nu_{\rm II}^{\rm SF}$ in Fig. 6, are observed to lie at about 5 and 6 T, respectively.

HF-ESR measurements at T = 2 K and various fixed frequencies allow us to construct the frequency-magnetic-field diagram (Fig. 7). The figure combines resonance positions stemming from single-crystal measurements with B||c axis and from measurements on the polycrystalline sample. In addition to the dominant resonance features (black and red data points in Fig. 7), we find weak features forming a gapless branch with g = 2.00 and disappearing for $\nu \gtrsim 90$ GHz (see Fig. 7) which we attribute to a small number of paramagnetic impurities and do not consider for further analysis.

The presence of four dominant resonance modes, three of which merge into a single zero-field excitation gap, indicates a two-sublattice antiferromagnet (AFM) with an axial-like anisotropy. The corresponding model's free parameters are given by the zero-field excitation gap Δ and by a pair of



FIG. 7. Microwave frequencies vs. resonance magnetic fields at T = 2 K. Closed symbols depict data obtained on the single crystal with B||c axis, open symbols on the polycrystal. Black and red solid lines show fitting results from a two-sublattice uniaxial AFMR model (see the text). Light grey markers and line corresponds to a weak paramagnetic impurity feature. The vertical dashed line marks the spin-flop field B_{SF} esr as determined from the fitting procedure of the resonance modes (see the text). Inset: temperature dependence of the zero-field excitation gap Δ , whereby the vertical dashed line marks T_{N} .

effective g factors, $g_{\parallel easy}$ and $g_{\perp easy}$ [35–37]. A least-squares fitting of the four magnon branches yields $\Delta = 166(1)$ GHz, $g_{\parallel easy} = 1.98(1)$, and $g_{\perp easy} = 2.00(1)$. In the two-sublattice uniaxial AFMR model, the spin-flop field is fixed at $B_{SF}^{ESR} =$ $(h\Delta)/(g_{||easy}\mu_B)$, where h is the Planck constant and μ_B the Bohr magneton, giving $B_{SF}^{ESR} = 6.0(1)$ T. Note that using an isotropic g-factor as suggested by high-temperature X-band ESR data does not lead to a difference in the determined value of the gap. The hereby-obtained value of the spin-flop field corroborates its determination from the magnetization and magnetostriction measurements presented in Fig. 4 and is in accordance with previous studies on MnTiO₃ [21,38]. The fitting results of the magnon branches are displayed in Fig. 7 as black and red solid lines, and the resonance branches are labeled $\nu_{||}^{U}$, $\nu_{||}^{L}$, $\nu_{||}^{SF}$, and ν_{\perp} , corresponding to the upper, lower, and spin-flop branch stemming from B || easy axis, and to the perpendicular branch stemming from $B \perp$ easy axis, respectively. Further to the dominant strongly field-dependent features, almost field-independent resonance features around the position of the spin-flop field have also been detected for B||c (blue data points in Fig. 7). Such a field-independent resonance branch is expected to occur at B_{SF} , corresponding to the rotation of the spins perpendicularly to the easy axis [37].

The AFMR results confirm the orientation of the easy axis along the crystallographic c axis, as the single-crystal measurements with B||c axis contribute solely to the model's

predictions for the easy-axis resonance features. The model's above-obtained parameters contrast with earlier observations from HF-ESR in which the zero-field excitation gap was found to amount to $\Delta = 153$ GHz and the effective g factor to 2.1 [39], and to $\Delta = 156$ GHz [17], respectively; and from inelastic neutron scattering which found $\Delta \simeq 200$ GHz [20], and $\Delta \simeq 193$ GHz [40], respectively.

Having shown that the two resonance features $v_{||}^{L}$ and $v_{||}^{SF}$ can be interpreted as the lower and spin-flop branch of a uniaxial two-sublattice AFMR model, the features' temperature dependence can be revisited. As observed in Fig. 6, $v_{||}^{L}$ remains at approximately the same resonance field all the way up to 30 K. For $T \ge 40$ K, a significant shift of the resonance field to lower values is detected. Since in the model under consideration, $v_{||}^{L}$ is directly related to the zero-field splitting and to the spin-flop field, the branch's shift to lower fields signals a reduction in Δ and its eventual disappearance as $T \nearrow T_{N}$. Indeed, making use of [35]

$$\nu_{||}^{\rm L} = \left[\Delta^2 + \left(\frac{\chi_{||}}{2\chi_{\perp}} \frac{g_{||}\mu_{\rm B}B}{h}\right)^2\right]^{\frac{1}{2}} - \frac{g_{||}\mu_{\rm B}B}{h} \left(1 - \frac{\chi_{||}}{2\chi_{\perp}}\right) \quad (1)$$

where $\chi_{\parallel}(\chi_{\perp})$ is the temperature-dependent static magnetic susceptibility parallel (perpendicular) to the easy magnetization axis (Fig. S2 in Ref. [26]), the temperature evolution of the feature's position may be used to obtain the temperature dependence of Δ and of B_{SF}^{ESR} . This is done by simulating the field dependence of the resonance branch $v_{||}^{L}$ at the various temperatures under the constraint of the position of the resonance field extracted from Fig. 6. The frequency value at which the simulated resonance branch crosses B = 0 T gives $\Delta(T)$, while the field value at which the simulated branch crosses v = 0 GHz gives $B_{SF}^{ESR}(T)$. The temperature dependence of Δ is displayed in the inset of Fig. 7 and exhibits a typical behavior of a long-range-ordered AFM. A similar temperature evolution of the zero-field excitation gap was observed also in previous HF-ESR measurements on MnTiO₃ [17,39].

The resonance branch $v_{||}^{SF}$ is found to shift to higher resonance fields as temperature increases from the lowest measured temperatures, and, moreover, no resonance feature associated with $v_{||}^{SF}$ is observed for $T \ge 40$ K. A similar effect has been observed in a number of long-range-ordered AFMs, such as CuCl₂ · 2H₂O [41] or LiFePO₄ [42]. Note that the conventional mean-field spin-only AFMR model predicts a shift of $v_{||}^{SF}$ to lower resonance fields upon heating, contrary to our observation.

IV. DISCUSSION

Our thermodynamic studies presented in Figs. 1, 4, and 5, and S4 allow us to construct the magnetic phase diagram in Fig. 8. Also included in the phase diagram is B_{SF} as calculated from the temperature-dependent HF-ESR spectra (Fig. 6) by means of Eq. (1). The results show that application of $B||c < B_{SF}$ yields a small reduction of the longrange ordering temperature T_N by about 0.5 K. In contrast, $\partial T_N / \partial B_{||c}$ is positive for fields $B > B_{SF}$ and T_N increases to 65 K at 15 T. Thermodynamically, according to the Ehrenfest relation this observation implies that the phase boundary



FIG. 8. Magnetic phase diagram of MnTiO₃ for B||c axis constructed from magnetization M(T, B), dilatometry L(T, B), specific heat $C_p(T, B)$, and HF-ESR data. Lines are guides to the eye. G-type AFM, SF, PM, and SRO label the G-type antiferromagnetic (yellow), the spin-flop (blue), the high-temperature paramagnetic (white), and the paramagnetic short-range-ordered phase (grey), respectively. The inset enlarges the regime around T_N .

 $T_{\rm N}(B > B_{\rm SF})$ is associated with an increase in magnetic susceptibility upon cooling. In addition, as discussed above, $B_{\rm SF}$ is found to increase slightly upon heating (see Fig. 5). Specifically, it increases by $\simeq 0.9$ T from 5.9(1) T at 2 K to 6.8(1) T at 60 K. For fields applied along the crystallographic *b* axis, the Néel temperature does not significantly change up to the highest measured fields.

The calculated uniaxial pressure dependencies of T_N show that in-plane pressure stabilizes antiferromagnetic order, while the order is suppressed for p||c. Based on inelastic neutron scattering the net in-plane antiferromagnetic exchange interaction J_{ab} is much stronger than the net out-of-plane interaction J_c , with ratio of $J_c/J_{ab} = 0.022$ at 2 K [40]. Upon heating, this ratio further diminishes to less than 1% at T_N [40]. Assuming that long-range order depends on J_c much more sensitively than on J_{ab} , the experimentally determined uniaxial pressure dependencies of T_N reflect predominately $\partial J_c/\partial p_i$. This is consistent with our results which indicate opposite signs of the uniaxial pressure dependencies at T_N , as uniaxial pressure along c will decrease the interplanar distances, while p||ab in a simple macroscopic picture is expected to have an opposite effect.

A comparison of $\partial T_N / \partial p_i$ in MnTiO₃ with values from the literature for NiTiO₃ [43] and CoTiO₃ [44] shows that despite supposed strong differences in spin-orbit coupling for the metal ions and despite different types of antiferromagnetic ordering, the uniaxial pressure dependencies have the same order of magnitude. This again is consistent with the supposition of the dominant role of J_c for establishing long-range magnetic order in these materials. Note, however, that the three magnetic titanates have different signs of the uniaxial pressure dependence at T_N : $\partial T_N / \partial p_c$ is positive in NiTiO₃ and CoTiO₃, while it is negative in MnTiO₃; for p||ab, the uniaxial

TABLE I. Signs of the uniaxial pressure dependencies $\partial T_N / \partial p_i$ of T_N for MnTiO₃, NiTiO₃ [43], and CoTiO₃ [44], as well as the antiferromagnetic spin configuration in the respective material.

	MnTiO ₃	NiTiO ₃	CoTiO ₃
$\partial T_N / \partial p_{\perp c}$	+	_	+
$\partial T_N / \partial p_{\parallel c}$	_	+	+
spin structure	G-type (<i>c</i>)	A-type $(\perp c)$	A-type $(\perp c)$

pressure dependence is positive in MnTiO₃ and CoTiO₃, while it is negative in NiTiO₃ (see Table I). Our data in particular exclude that spin-lattice coupling in MnTiO₃ are weak as suggested previously [2]. A similar conclusion has been drawn from a recent Mössbauer study [23].

In addition to these quantitative results describing the spin-lattice coupling in MnTiO₃, our data allow conclusions about the coupling of spin and lattice to the dielectric properties. Previous measurements show that MnTiO₃ develops spontaneous polarization in c direction below T_N in finite fields B||c > 0 which increases with field up to B_{SF} and is approximately constant in temperature in the ordered phase [23,45]. At B = 0 T, no spontaneous polarization is observed. The lattice, on the other hand, exhibits significant anisotropic changes with temperature below roughly 170 K in zero magnetic field, as can be seen from the distortion parameter in Fig. 2, as well as in finite magnetic fields (see Fig. S4). Furthermore, applying magnetic field parallel to the c axis does not induce relevant lattice changes below B_{SF} as shown in Figs. 4 and S4. This implies that the strong magnetostrictive effects which we observe at B = 0 T do not drive the ferroelectric moment but additional, i.e., field-induced, symmetry breaking is essential.

By performing a linear extrapolation of the static magnetization in Fig. 4(a) to the known value of the full ordered magnetic moment in MnTiO₃, $\mu_{sat} = 4.55 \ \mu_B/f.u.[16]$, the saturation field can be roughly estimated as $B_{sat} = 213(10)$ T. In combination with the spin-flop field $B_{SF} = 5.9$ T observed in our study (cf. Figs. 4 and 7), the effective anisotropy and exchange fields, B_A and B_E , can be obtained. To do so, we take the spin Hamiltonian [37]

$$\hat{\mathcal{H}} = -D\sum_{i} \left(\hat{S}_{i}^{z}\right)^{2} + \sum_{\langle i,j \rangle} J_{ij}\hat{S}_{i} \cdot \hat{S}_{j} - g\mu_{\rm B}\sum_{i} \boldsymbol{B} \cdot \hat{S}_{i} \quad (2)$$

where the first term quantifies the anisotropy energy, the second term the exchange energy, and the third term the Zeeman energy. *D* is the uniaxial anisotropy constant, defined as $D = (g\mu_B B_A)/(2S)$ [37]. J_{ij} is the magnetic exchange constant between two magnetic neighbors occupying the sites *i* and *j*.

To solve the Hamiltonian in Eq. (2), we assume a meanfield two-sublattice model and relate macroscopic sublattice magnetizations to the above spin operators via the relation $M_{1,2} = g\mu_B N \hat{S}_{i,j}$, where N is the number of spins per unit volume [37]. Considering the exchange interaction only between nearest neighbors residing on opposite sublattices, the effective exchange interaction can be defined as $J_{\text{eff}} = (g\mu_B B_E)/(2Sz)$, where z is the number of nearest neighbors [37]. Upon minimization of the energy in Eq. (2), the expressions $B_A = B_{SF}^2/B_{sat}$, $B_E = B_{sat}/2 + B_{SF}^2/2B_{sat}$, and $\cos(\theta) = B/(2B_E - B_A)$ are obtained in terms of experimentally measurable quantities, where θ is the angle between the easy axis and the sublattice magnetization in the spin-flop phase. We find $B_A = 0.16(1)$ T and $B_E = 107(6)$ T, which correspond to D = 0.0038(3) meV and, with the number of nearest neighbors z = 3, $J_{eff} = 0.82(4)$ meV. In addition, $\theta_{SF} = 88.4(1)^\circ$ at the spin flop.

The signs of the obtained parameters from Eq. (2) signal that the system's anisotropy assumes a uniaxial character and that the dominant intersublattice exchange interaction is antiferromagnetic. Moreover, the obtained values of D and J_{eff} quantitatively corroborate their determination by means of inelastic neutron scattering: D = 0.0011 meV and $J_1 = 0.63$ meV [20], and D = 0.0045 meV and $J_1 = 0.92$ meV [40], respectively.² A good correspondence is found also between the angle θ_{SF} as estimated from the size of ΔM at the spin flop transition [cf. Fig. 4(a)] and as ascertained here from the energy minimization of the above Hamiltonian.

In a mean-field model, the effective anisotropy field can be independently calculated from $B_A = (B_{SF}^2 \times \chi_{\perp})/M_{sat}$ [46]. Here, χ_{\perp} is the static magnetic susceptibility obtained with the external field applied perpendicular to the easy magnetization axis. Our static magnetic susceptibility obtained for B = 1 T||*b* axis reveals that $\chi_b = 0.0122(2)$ erg/(G²mol) at 2 K (see Fig. S2 in Ref. [26]). Hence, the effective anisotropy field can be estimated as $B_A = 0.17(1)$ T, corroborating the above determination of its value.

The ratio of the anisotropy to exchange field amounts to $B_A/B_E \simeq 1.5 \times 10^{-3}$, thereby confirming the weak effective anisotropy field relative to the effective exchange field. This ratio is typical for Mn²⁺-containing two-sublattice AFMs such as P31*m*-phase MnSb₂O₆, P321-MnSb₂O₆, CaMnCl₃ · H₂O, or MnF₂ which have been studied by AFMR [46–49]. In MnF₂, $B_A = 0.82$ T and $B_E = 52.6$ T, resulting in the ratio $B_A/B_E \simeq 1.6 \times 10^{-2}$ [46] while the other examples feature values of $B_A/B_E \simeq 2 \times 10^{-3}$.

The energy changes associated with the spin-flop transition can be estimated by $\Delta E_{\rm SF} = \Delta M \times B_{\rm SF}$. Using ΔM and $B_{\rm SF}$ obtained at 2 K, where both quantities show only negligible temperature dependencies, yields $\Delta E_{\rm SF} \simeq 42(5) \,\mu {\rm eV}^3$ This value is of the same order but clearly larger than $E_{\rm a}^{\rm ESR} =$ $DS^2 = 24(3) \,\mu {\rm eV}$ obtained from the analysis of our HF-ESR data. This further indicates that the spin-flop transition is not governed purely by magnetic anisotropy but also includes additional contributions, i.e., from the lattice. Analysing the M(B) data obtained at higher temperatures yields the temperature dependence of $\Delta E_{\rm SF}$ as summarized in Table S1 (see also Fig. 9).

A summary of the various results on the temperature dependence of anisotropy-related quantities from our analyses is shown in Fig. 9. In addition to the anisotropy gap measured by HF-ESR and ΔE_{SF} obtained as described above, Fig. 9 also presents the mean-field estimate of the anisotropy constant

 $^{{}^{2}}J_{1}$ denotes the dominant, nearest-neighbor exchange coupling; as such, J_{eff} in the present study reflects predominately J_{1} .

³Extrapolating the temperature dependencies to T = 0 K does not change this result within the error bars.



FIG. 9. (a) $\Delta_{\rm ESR}$ from the analysis of the HF-ESR data (see the inset of Fig. 7), energy changes at the spin-flop phase boundary $\Delta E_{\rm SF} = \Delta M_{\rm SF} \times B_{\rm SF}$, and the anisotropy constant as derived from $K_1 = (\chi_{\perp} - \chi_{\parallel})(B = 1 \text{ T}) \times B_{\rm SF}^2/2$ [50]. All data are normalized to their respective values at 2 K. (b) Spin-flop field $B_{\rm SF}$ and the corresponding entropy jump $\Delta S_{\rm SF}$ (see Sec. B and Table S1 of Ref. [26]). The solid line is a phenomenological fit to $B_{\rm SF}(T)$.

 $K_1 = (\chi_{\perp} - \chi_{\parallel})B_{SF}^2/2$ [50,51]. Here, $\chi_{\perp} - \chi_{\parallel}$ is obtained from the magnetic susceptibility data at B = 1 T displayed in Fig. S2 of Ref. [26]. All data show that the anisotropy decreases upon heating and vanishes at T_N . In addition, Fig. 9(b) shows the temperature-dependence of B_{SF} and the associated entropy changes.

In searching for the origin of the anisotropy, it can be observed that due to vanishing orbital moment of the Mn²⁺ magnetic ions, spin-orbit coupling is expected to act only in second order of perturbation theory [34]. In particular, only negligible contributions to the overall anisotropy are expected to arise from single-ion effects and from anisotropic exchange interaction. In principle, both symmetric and antisymmetric exchange can lead to macroscopic anisotropy effects [34]. Based on symmetry considerations, however, the contribution from antisymmetric exchange in MnTiO₃ amounts to zero, $J_{anis}^{antisym} = 0$. The contribution due to symmetric exchange can be estimated as $J_{anis}^{sym} \sim J_{eff}(\frac{\Delta g}{g})^2$ [34]. With the room-temperature spectroscopic g-factors $g_{\parallel c} = 2.004(2)$ and $g_{\perp c} = 2.002(2)$ obtained in our X-Band measurement (cf. Fig. S5 of Ref. [26]), $\Delta g = 0.002$, resulting in the effective anisotropy energy on the order of $\sim S^2 J_{anis}^{sym} \simeq 5 \times 10^{-6} \text{ meV}.$ This is almost three orders of magnitude smaller than the calculated anisotropy constant D and hence cannot account for its value.

On the other hand, the magnetic dipole energy can be approximated as $E^{\text{dip}} = \frac{\mu_0 \mu_B^2 g_1 g_2}{4\pi |\mathbf{r}_{1,2}|^3} [3(\mathbf{S}_1 \cdot \hat{\mathbf{r}}_{1,2})(\mathbf{S}_2 \cdot \hat{\mathbf{r}}_{1,2}) - \mathbf{S}_1 \cdot \mathbf{S}_2]$ where $\mathbf{r}_{1,2}$ ($\hat{\mathbf{r}}_{1,2}$) is the (direction) vector between spins \mathbf{S}_1 and \mathbf{S}_2 with g-factors g_1 and g_2 . Using our refined Mn–Mn distances [25], we perform the sum of the magnetic dipole

energies $\sum_{i} n_i \times E_i^{\text{dip}}$, where n_i is the number of respective neighbors, up to first six distinct neighbors (i.e., i = 6, so that in total 22 neighbors are considered). To obtain the anisotropy energy due to magnetic dipole interactions, we compute the difference in the obtained dipole energy between uniaxial and planar spin orientations, $E_A^{\text{dip}} = E_{||}^{\text{dip}} - E_{\perp}^{\text{dip}}$ [52]. $E_A^{\text{dip}} \simeq$ 0.09544 meV is found, i.e., $E_A^{\text{dip}} \simeq 0.0477$ meV per magnetic ion. Based on $E \sim DS^2$, it follows that $D^{\text{dip}} \simeq 0.0076$ meV. This is on the same order of magnitude, albeit larger, than the above obtained value of D, indicating the significance of magnetic dipole interactions for the evolution of anisotropy in MnTiO₃. It may be speculated that the discrepancy to the experimental observations results from second-order effects from admixing of Mn³⁺ (3d⁴) ions [21] which may reduce the actual magnetic dipole energy.

While the low-temperature AFMR modes are explained well by the mean-field spin-only model elaborated above, the temperature dependence of the resonance mode $v_{||}^{SF}$ is not accounted for by the conventional mean-field theory. This predicts a shift of $v_{||}^{SF}$ to lower resonance fields upon heating, in contrast to our observation of the resonance's shift to higher fields (see Fig. 6). Furthermore, our results show that, despite acting only to the second order in perturbation theory, spin-orbit coupling must be also considered to fully cover the magnetic ground state properties in MnTiO₃. This is clearly evidenced by the observed spin-lattice coupling and magnetostrictive response on spin reorientation.

V. SUMMARY

In summary, we report on the static and dynamic magnetic properties and magnetoelastic coupling in MnTiO₃ single crystals in magnetic fields up to 15 T. We present the magnetic phase diagram which exhibits the spin-flop phase and show that $\partial T_{\rm N}/\partial B$ changes sign at $B_{\rm SF}$. Despite the vanishing orbital moment associated with Mn²⁺ ions in MnTiO₃, we observe pronounced thermal expansion and magnetostriction anomalies at $T_{\rm N}$ and $B_{\rm SF}$, respectively, of similar magnitude as in related Co²⁺- and Ni²⁺-based magnetic titanates. Accordingly, uniaxial pressure affects the respective phase boundaries and we obtain a finite suppression of $B_{\rm SF}$ amounting to $\partial B_{\rm SF}/\partial p_{\rm c} = -0.20(2)$ T/GPa. Longrange antiferromagnetic order is stabilized (suppressed) by p||b|(p||c), as $\partial T_N/\partial p_b = 0.69(12)$ K/GPa and $\partial T_N/\partial p_c =$ -2.0(4) K/GPa, respectively. The observed AFMR modes are fully described by a two-sublattice mean-field model with uniaxial anisotropy. The model's optimized parameters are the effective g factors $g_{\parallel easy} = 1.98(1)$ and $g_{\perp easy} = 2.00(1)$, and the zero-field excitation gap $\Delta = 166(1)$ GHz. Δ is found to decrease with increasing temperature and to vanish at $T_{\rm N}$. Our analysis yields the effective anisotropy field $B_{\rm A} =$ 0.16(1) T, corresponding to D = 0.0038(3) meV, as well as $B_{\rm A}/B_{\rm E} \simeq 1.5 \times 10^{-3}$. The anisotropy is argued to originate predominately from out-of-plane nearest-neighbor magnetic dipole interactions. Notably, the short-range magnetic order persists well above T_N as indicated by a finite lattice distortion parameter δ , the violation of a constant Grüneisen behavior, and the presence of local magnetic fields up to at least 200 K detected by HF-ESR. Our results hence elucidate the origin and the role of anisotropy and show its relevance as well as

the importance of spin-lattice coupling for static and dynamic magnetic properties in MnTiO₃.

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