

Ta₄CoSi: A tantalum-rich superconductor with a honeycomb network structureLingyong Zeng,¹ Xunwu Hu,² Shu Guo,^{3,4} Gaoting Lin,⁵ Jing Song,⁶ Kuan Li,¹ Yiyi He,¹ Yanhao Huang,¹ Chao Zhang,¹ Peifeng Yu,¹ Jie Ma,⁵ Dao-Xin Yao,^{2,4} and Huixia Luo^{1,*}¹*School of Materials Science and Engineering, State Key Laboratory of Optoelectronic Materials and Technologies, Key Lab of Polymer Composite & Functional Materials, Guangzhou Key Laboratory of Flexible Electronic Materials and Wearable Devices, Sun Yat-Sen University, No. 135, Xingang Xi Road, Guangzhou 510275, People's Republic of China*²*School of Physics, Center for Neutron Science and Technology, Guangdong Provincial Key Laboratory of Magnetoelectric Physics and Devices, State Key Laboratory of Optoelectronic Materials and Technologies, Sun Yat-Sen University, Guangzhou 510275, People's Republic of China*³*Shenzhen Institute for Quantum Science and Engineering, Southern University of Science and Technology, Shenzhen 518055, China*⁴*International Quantum Academy, Shenzhen 518048, China*⁵*Key Laboratory of Artificial Structures and Quantum Control, Shenyang National Laboratory for Materials Science, School of Physics and Astronomy, Shanghai Jiao Tong University, Shanghai 200240, China*⁶*Institute of Physics, Chinese Academy of Sciences, Beijing 100190, China*

(Received 18 May 2022; revised 16 September 2022; accepted 19 September 2022; published 3 October 2022)

The crystal structure of solid-state materials with unique lattices is considered one of the main factors determining physical properties, including superconductivity. Materials with honeycomb lattices have inspired intense research interest for their novel properties. A previously unknown compound, Ta₄CoSi, is discussed herein, along with its crystal structure and superconducting properties. It crystallizes in a CuAl₂-type structure (space group *P4/mcc*, No. 124), with the lattice parameters $a = b = 6.17755(7)$ Å and $c = 4.98443(8)$ Å, featuring honeycomb networks of Ta-Ta in the (110) plane. Superconductivity is observed below the critical temperature of ~ 2.45 K, while lower and upper critical magnetic fields are ~ 9.86 mT and 0.84 T, respectively. Experiments and theoretical calculations show that the honeycomb lattices significantly influence the superconducting properties. This material may thus provide a new platform for investigating the exotic superconductivity of honeycomb networks.

DOI: [10.1103/PhysRevB.106.134501](https://doi.org/10.1103/PhysRevB.106.134501)**I. INTRODUCTION**

Ternary silicide intermetallic compounds $A_xB_ySi_z$ ($A, B =$ transition metal or rare-earth metal) have shown very diverse and interesting crystal structures and physical properties. Depending on the constituent element, properties like superconductivity, heavy fermions, quantum spin liquids, large magnetocaloric effect, large magnetoresistance, etc., have been observed in this family [1–11]. The search for new superconductors in ternary silicide intermetallic systems has always been one of the research hotspots in the field of condensed matter. And many novel superconductors have been reported in this family, such as a wide variety of noncentrosymmetric superconductors ($ABSi_3$) [1,6–8], multigap superconductors ($Lu_2Fe_3Si_5$ and $Sc_5Ir_4Si_{10}$) [2,3], the first heavy-fermion superconductor ($CeCu_2Si_2$) [4], and the strong correlation superconductor ($Er_2Fe_3Si_5$) with the coexistence of antiferromagnetism and superconductivity [5]. Besides, equiatomic

ternary silicide intermetallics ($ABSi$) are noticeable, which are well-known superconductors with a different superconducting transition temperature (T_c) [12–16].

Furthermore, another family of layered materials with a Kagome structure also attracts attention. Recently, the ternary silicide intermetallic $LaRu_3Si_2$ is a nodeless, moderate coupling Kagome superconductor, with $T_c \sim 7.3$ K [17,18]. The structure of $LaRu_3Si_2$ contains distorted Kagome layers of ruthenium sandwiched between lanthanum layers and silicon layers with a honeycomb network. Meanwhile, honeycomb lattices are also an attractive playground for superconductivity. Although the superconducting compounds in ternary silicide intermetallic systems are abundant, there is still a high potential to investigate and discover novel types of superconductors in these systems.

Nb_4MSi (where $M = Fe, Co, \text{ or } Ni$) are the first representatives of ternary silicide intermetallic compounds with a structure of the CuAl₂ type. Two kinds of space groups of *P4/mcc* and *I4/mcm* were previously reported for these CuAl₂-type compounds [19]. And previous studies have demonstrated that the *P4/mcc* space group is a

*luohx7@mail.sysu.edu.cn

superconducting phase. In addition, there are few reports on superconducting cobalt-based compounds. Nb_4CoSi is one of the cobalt-based superconductors ($T_c = 6.0$ K) that includes Co as a principal constituent element [20]. And an interesting feature is that although binary compounds of the CuAl_2 type are unknown for niobium, tantalum forms the compounds Ta_2Si and Ta_2Co , exhibiting structures of this type. It is possible that these compounds will present a CuAl_2 -type structure in the ternary Ta-Co-Si system, which has not yet been studied, and that restricted solid solutions of cobalt in Ta_2Si will be found in the ternary Ta-Co-Si systems.

In this work, we have successfully synthesized a new intermetallic compound in the Ta-Co-Si ternary system. Ta_4CoSi has a Ta-Ta honeycomb network in the (110) plane. It displays superconductivity with $T_c = 2.45$ K. Electric resistivity, magnetic susceptibility, and specific heat measurements have been carried out to determine the Ta_4CoSi alloy's superconducting parameters. First-principles calculations indicate that the Ta $5d$ state mainly dominates the electronic structure of the Ta_4CoSi intermetallic compound.

II. EXPERIMENTAL DETAILS

Elemental tantalum (99%, 22 mesh, Alfa Aesar), cobalt (99.5%, 325 mesh, Alfa Aesar), and silicon (99.9%, 100 mesh, Alfa Aesar) for Ta_4CoSi were utilized as raw materials and synthesized by the arc-melting method. A total of 250 mg of the elemental powders in a molar ratio of 3.2:1:1 was pressed into a piece, with tantalum powder on the top and silicon powder on the bottom, then arc-melted under 0.5 atm of argon gas. The ingots were turned over and melted several times from both sides of the ingot to ensure homogeneity. The resulting air-stable material was annealed at 1050 °C for 48 h under a high vacuum.

The crystal structure of the sample was determined by powder x-ray diffraction (PXRD) using a conventional x-ray spectrometer equipped with Cu $K\alpha$ radiation. The intensity data were obtained over the 2θ range of 10 to 100 °, with a step width of 0.01 °. The lattice parameters and the occupancy of the atoms were refined using the Rietveld method. The Fullprof suite package was used for Rietveld fitting of the PXRD data [21]. The elemental ratio was determined by scanning electron microscope combined with energy-dispersive x-ray spectroscopy (SEM-EDXS). Electrical transport, magnetization, and specific heat measurements were measured in a Quantum Design PPMS-14T. The resistivity was performed with the standard four-probe configuration by attaching platinum electrodes with silver paste on the sample.

We performed first-principles calculations implemented in the Vienna Ab initio Simulation Package (VASP) [22,23]. The generalized gradient approximation in the Perdew–Burke–Ernzerhof form was used in treating the electron exchange–correlation [24]. The pseudopotentials for the projector augmented-wave method (where valence electron configurations of each atom are $3d^8 4s^1$ for cobalt, $3s^2 3p^2$ for silicon, and $5p^6 6s^2 5d^3$ for tantalum) with a basis set cutoff energy of 500 eV were used to describe the ion–electron interaction [25]. The lattice constants of polycrystalline refined from x-ray diffraction were fixed for comparison with the experiment. In contrast, atomic positions were fully relaxed

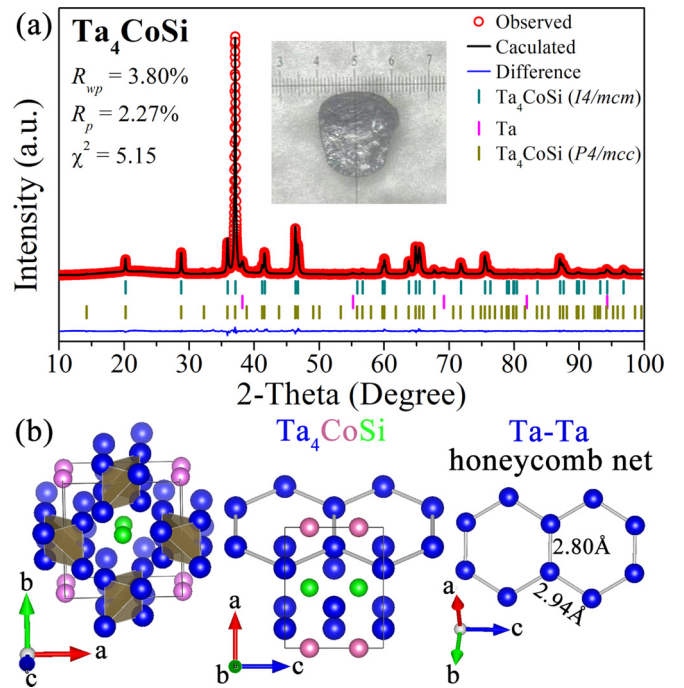


FIG. 1. Crystal structure characterization of Ta_4CoSi . (a) Rietveld refinement profile of the PXRD of the new superconducting phase Ta_4CoSi . (b) The crystal structure of Ta_4CoSi with space group $P4/mcc$. The tantalum atoms form a honeycomb network in the (110) plane.

until the Hellmann–Feynman force on each atom was less than 0.01 eV/Å in the ionic relaxation loop. The k -point sampling of the first Brillouin zone is $5 \times 5 \times 7$ Γ -centered k -mesh to ensure convergence.

III. RESULTS AND DISCUSSION

PXRD characterization of the ground-polycrystalline Ta_4CoSi is shown in Fig. 1(a). Ta_4CoSi crystallizes in a CuAl_2 -type structure, the same as other reported 411-phases [20,26] and the recent high-entropy alloys superconductors [27]. Consistent with Nb_4NiSi [19,20], there may also exist two space groups- $P4/mcc$ and $I4/mcm$ -in the Ta_4CoSi compound. And the diffraction peaks of these two space groups basically coincide. The difference between these two space groups is whether the silicon and cobalt atoms are ordered or disordered. When refining the PXRD data only with the $P4/mcc$ space group, the refinement parameters R_{wp} , R_p , and χ^2 are 4.51%, 3.23%, and 7.26, respectively. However, we could obtain smaller refinement parameters by applying a two-space group model [see in Fig. 1(a)]. And the refinement parameters R_{wp} , R_p , and χ^2 are 3.80%, 2.65%, and 5.15, respectively. The fitting results show that the main phase is about 76% of the $P4/mcc$ phase and about 15% of the $I4/mcm$ phase. Some extra peaks originate from the excess tantalum in the raw material. The lattice parameters of $P4/mcc$ space group in Ta_4CoSi from the Rietveld-fitted PXRD profile are $a = b = 6.1775(7)$ Å, $c = 4.98443(8)$ Å. Table I lists the refined structural parameters of the $P4/mcc$ phase. The refined crystal structure with the $P4/mcc$ space group of Ta_4CoSi is

TABLE I. Structural parameters of superconducting Ta₄CoSi at 300 K. The space group is $P4/mcc$ (No. 124), $a = b = 6.17755(7)$ Å, $c = 4.98443(8)$ Å.

Atom	x	Y	z	Wyckoff	Occupancy
Ta	0.16233(2)	0.65777(2)	0	$8m$	1.0
Co	0	0	0.25	$2a$	1.0
Si	0.5	0.5	0.25	$2c$	1.0

shown in Fig. 1(b). As shown in Fig. 1(b), the nearest tantalum atoms form a one-dimensional zigzag along the c -axis. The nearest cobalt and silicon atoms form linear chains along the c -axis. The tantalum atoms in Ta₄CoSi are stretched to the direction of the (110) plane of the crystal structure, forming a hexagonal tantalum-chain network, similar to graphite, with two different Ta-Ta band lengths (2.80 Å and 2.94 Å). The atom having a honeycomb structure is also observed in MgB₂ [28], noncentrosymmetric TaRh₂B₂ [29], and Kagome lattice superconductors [30]. The back-scattered electron (BSE) and EDXS were characterized to check the homogeneity and the chemical composition of Ta₄CoSi. As displayed in Fig. 2, all constituent elements distribute homogeneously, and the real chemical formula is Ta_{4.05}Co_{0.89}Si_{1.06}.

The temperature-dependent electrical resistivity from 300 to 1.8 K of the polycrystalline Ta₄CoSi sample is presented in Fig. 3. The sample displays metallic behavior. The resistivity changes slightly above T_c , and the residual resistivity ratio $= \rho(300 \text{ K})/\rho(5 \text{ K}) \approx 1.25$ is a low ratio, comparable to that observed for highly disordered intermetallic or non-stoichiometric compounds. The inset in Fig. 3 displays the low-temperature behavior in the vicinity of T_c . The onset temperature of T_c appears at about 2.52 K, whereas zero resistivity is reached at about 2.41 K, characteristic of superconductivity. The sharp resistivity transition covers only a 0.11 K temperature range. The $T_c \approx 2.45$ K is determined from the midpoint of the resistive transition. The zero-field-cooled

(ZFC) magnetization of Ta₄CoSi is displayed in Fig. 4(a). Measurements were carried out between 1.8 K and 10 K under a 20-Oe field. The field-dependent volume magnetization was performed from 0 to 200 Oe for Ta₄CoSi at 1.8 K, and was fitted by the formula $M_{\text{fit}} = a + bH$, where b represents the slope of the fit line. The slope of the linear fit is -0.11082 . The procedure, $-b = 1/[4\pi(1-N)]$, was then used to estimate the demagnetization factor, N , which depends on the shape of the sample and how the sample is oriented concerning the magnetic fields. And the resulting N is about 0.28 for Ta₄CoSi for the measured sample, close to the theoretical value (0.33) for a rectangular cuboid sample [31]. The actual value of N (0.28) has been used to correct the magnetic susceptibility data, $4\pi\chi_v(1-N) = 0.78$. The PXRD refinement indicates that the sample has a superconducting phase ($P4/mcc$) of about 74%, which is close to the magnetic susceptibility data (78%). The strong diamagnetic response at 1.8 K approaches the ideal value of $4\pi\chi_v(1-N) = -1$ (-1 represents the ideal value of fully superconducting materials) for the Ta₄CoSi superconductor, implying bulk superconductivity. The critical temperature of $T_c = 2.44$ K was determined as the value at the point where the linearly approximated slope crosses the normal state magnetization [see the arrow in Fig. 4(a)].

The new tantalum-rich superconductor Ta₄CoSi was further performed with field dependence of magnetization measurements at various temperatures below T_c , as shown in Fig. 4(b). With increasing magnetic fields, magnetization rapidly decreases. The low-field linear fitted the magnetization data (M_{fit}) were used to construct the $M-M_{\text{fit}}$ plot in Fig. 4(c). The field where the magnetization starts to deviate from the linear response is the uncorrected lower critical field, $\mu_0H_{c1}^*$ at that temperature. All values with the corresponding temperatures are displayed in Fig. 4(d) and are fitted to the formula $\mu_0H_{c1}^*(T) = \mu_0H_{c1}^*(0)[1 - (T/T_c)^2]$, which is illustrated by the solid purple line. The $\mu_0H_{c1}^*(0)$ was calculated to be 7.10 mT. Furthermore, when considering the N value of 0.28, the lower critical field is calculated by the equation $\mu_0H_{c1}(0) = \mu_0H_{c1}^*(0)/(1 - N)$. Therefore, the

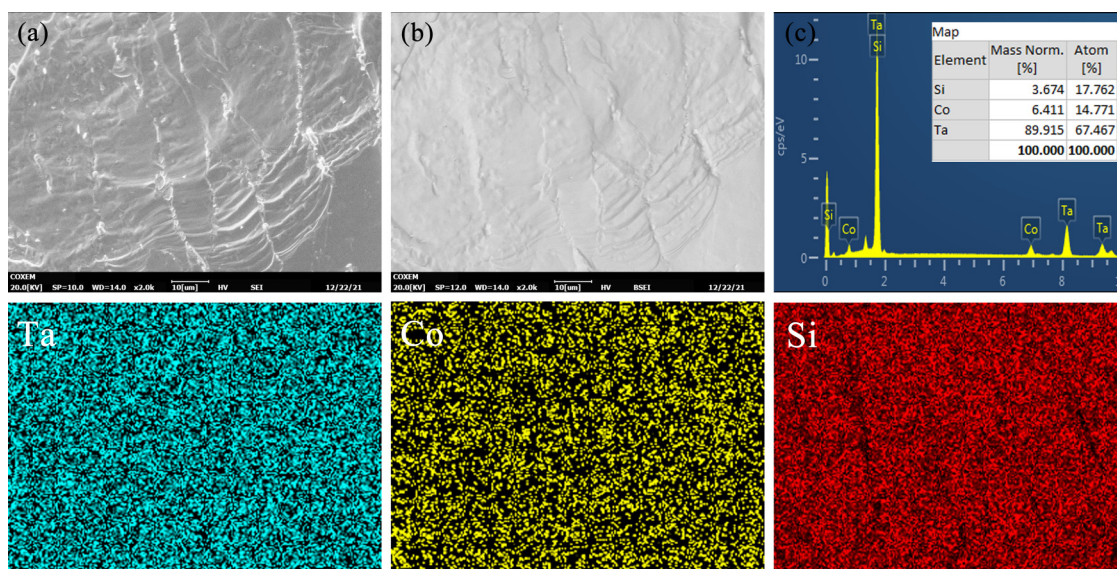


FIG. 2. The SEM (a), BSE (b), EDXS spectrum (c), and mapping of the Ta₄CoSi sample.

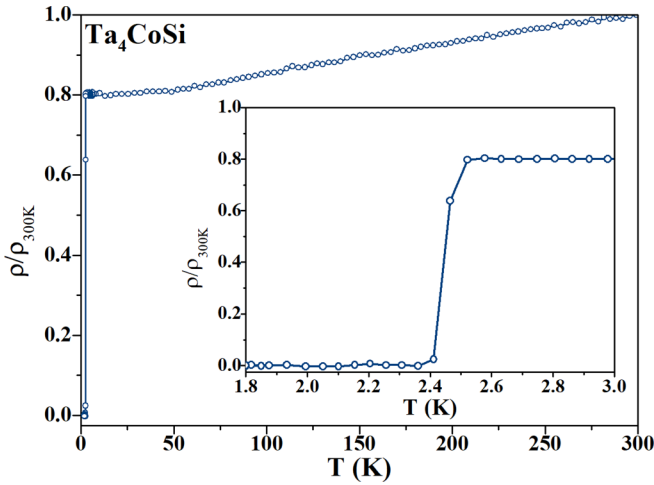


FIG. 3. Temperature dependence of normalized $\rho/\rho_{300\text{K}}$ of the Ta_4CoSi sample. The inset shows the low-temperature behavior in the vicinity of T_c .

corrected lower critical field $\mu_0 H_{c1}(0)$ is 9.86 mT, which is somewhat smaller than that of the Nb_4NiSi sample (10.9 mT) [20].

To understand the superconducting state further, the magnetoresistivity and the magnetic field-dependent magnetization under different temperatures were measured. The resistivity under different magnetic fields is presented in Fig. 5(a). T_c at various magnetic fields is determined at the 50% drop from the normal state resistivity value. Figure 5(b) presents the H_{c2} - T phase diagram of the superconducting

states for Ta_4CoSi alloys. The resulting slope ($d\mu_0 H_{c2}/dT$) is -0.4952 T/K for Ta_4CoSi . The upper critical field $\mu_0 H_{c2}(0)$ can then be obtained via the Werthamer–Helfand–Hohenberg (WHH) formula: $\mu_0 H_{c2}(0) = -0.693 T_c \left(\frac{d\mu_0 H_{c2}}{dT} \right) \Big|_{T=T_c}$. Using $T_c = 2.45$ K, the dirty limit $\mu_0 H_{c2}(0)^{\text{WHH}}$ value was calculated to be 0.84 T, with the Pauli limit transition field $\mu_0 H^{\text{Pauli}} = 1.85 T_c = 4.53$ T.

After observing both the zero resistivity and the Meissner effect in the magnetization, to confirm the observed superconductivity is intrinsic in Ta_4CoSi , specific heat measurements were carried out from 0.6 to 10 K, as shown in the inset in Fig. 6. An apparent anomaly at $T_c = 2.34$ K in the zero-field heat capacity corresponds to the emergence of superconductivity. It is well consistent with the T_c determined from the resistivity and magnetization measurements. The data were fitted with the equation $C_p/T = \gamma + \beta T^2$, yielding the electronic specific heat coefficient $\gamma = 15.1$ mJ/mol/K² and the phonon specific heat coefficient $\beta = 0.28$ mJ/mol/K⁴. After subtracting the phonon contribution, the electronic specific heat (C_{el}) is isolated and plotted as $C_{\text{el}}/\gamma T$ against T/T_c in Fig. 6. The $C_{\text{el}}/\gamma T$ jumps are analyzed by a modified BCS model, the so-called α model [32]. Using the equal-area construction method, the normalized heat capacity jump ($\Delta C_{\text{el}}/\gamma T_c$) is estimated to be 0.58, which is substantially smaller than the expected BCS value of 1.43 [32]. The small jump in specific heat indicates that the quasiparticles participating in the superconductivity condensation experience strong elastic scattering because inelastic scattering usually enhances the jump ratio [33,34]. Besides, the residual γ_0 value obtained by the extrapolation of the C_p/T curves linearly to

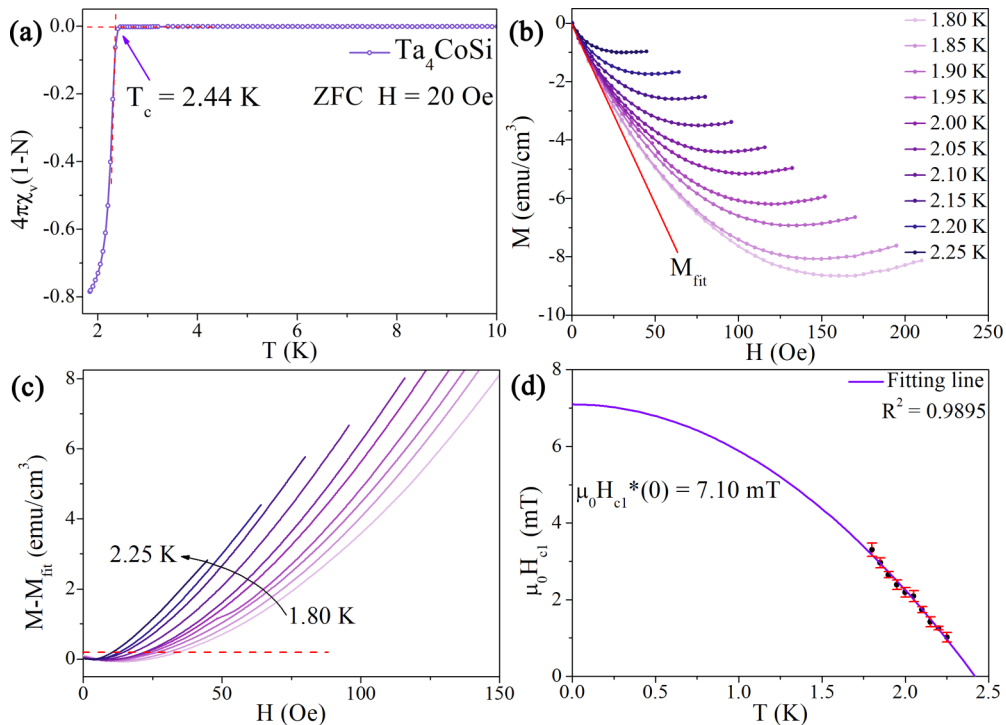


FIG. 4. (a) Temperature-dependent ZFC magnetization under the field of $H = 20$ Oe, exhibiting a transition to the diamagnetic Meissner state below 2.44 K. The volume susceptibility χ_v is corrected with the demagnetization factor derived from M versus H measurements. (b) Field-dependent magnetization curves measured at 1.80 to 2.25 K. (c) The $M - M_{\text{fit}}$ curves as a function of the magnetic fields. (d) Lower critical field estimation.

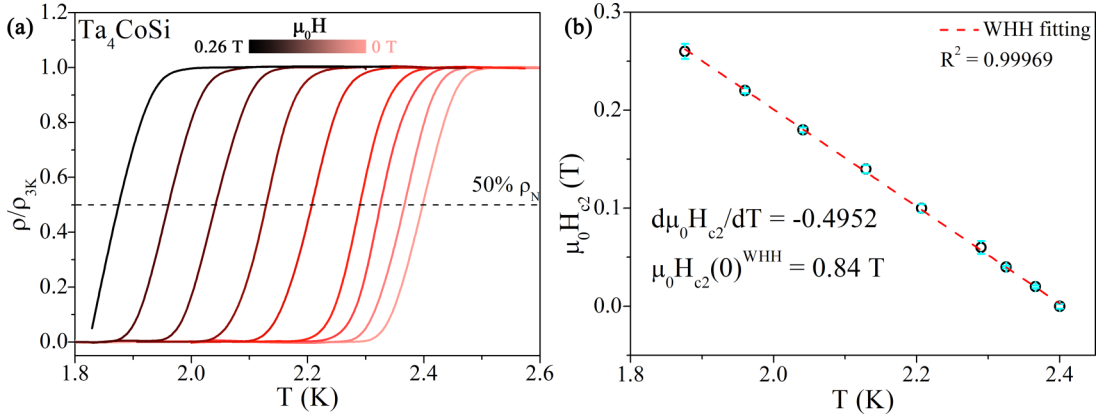


FIG. 5. (a) The magnetic field dependence of the superconducting transition for $0 T \leq \mu_0 H \leq 0.26 T$. The 50% criterion is depicted as the black dashed line. (b) The $\mu_0 H-T$ phase diagram.

0 K as shown in Fig. 6 is $\gamma_0 \sim 10.77 \text{ mJ/mol/K}^2$, which is about 71% of γ , also supports the presence of strong scatterers [35]. The large γ_0 value indicates a large residual density of states at the Fermi energy E_F . Phenomenological theories suggest that only the Fermi surface of the majority band opens the superconducting gap, and the minority band remains at a normal state below T_c . And the larger value of γ_0/γ in the case of Ta₄CoSi might indicate that the present superconductivity is extremely sensitive to a small amount of impurity. It can be seen from the PXRD refinement that there is a small part of the $I4/mcm$ phase in the Ta₄CoSi alloy [see Fig. 1(a)]. And previous studies have demonstrated that the $I4/mcm$ space group is a nonsuperconducting phase in the same CuAl₂-type compounds [19,20]. Furthermore, the nonsuperconducting contribution may also originate from defects in arc-melted material. Similar behaviors have also been observed in other systems [20,34,36–38], such as ternary silicide intermetallic compounds (Nb₄NiSi and URu₂Si₂) [20,36], a transition-metal chalcogenide superconductor (Nb₂Pd_{0.81}S₅) [37], and Fe-based superconductors

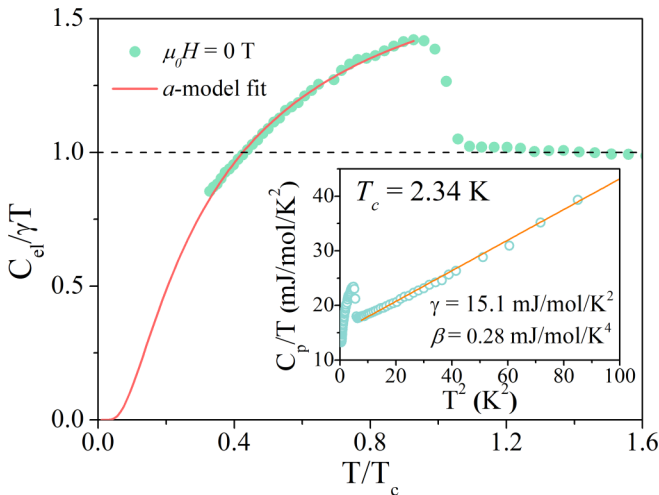


FIG. 6. Normalized electronic specific heats plotted a function of T/T_c . The inset displays the C_p/T versus T^2 for Ta₄CoSi at 0.6 to 10 K.

(LiFeAs, NdFeAsO_{1-x}F_x) [34,38]. The Debye temperature (Θ_D) of Ta₄CoSi was then calculated to be 346 K using the β value in the formula $\Theta_D = (12\pi^4 nR/5\beta)^{1/3}$, where R represents the ideal gas constant and n represents the number of atoms per formula unit ($n = 6$). Moreover, assuming $\mu^* = 0.13$, the electron-phonon coupling constant (λ_{ep}) is further estimated to be 0.51, following the inverted McMillan equation $\lambda_{ep} = \frac{1.04 + \mu^* \ln(\frac{\Theta_D}{1.45T_c})}{(1 - 0.62\mu^*) \ln(\frac{\Theta_D}{1.45T_c}) - 1.04}$ [39]. Finally, Table II summarizes all the superconducting parameters.

As is shown in Fig. 7, the projected and total density of states (PDOS) of Ta₄CoSi was calculated using first principles. The density of states (DOS) passing through the Fermi level demonstrates its metallic properties. The PDOS diagram shows that the $5d$ electron of tantalum is the main contribution near the Fermi level, although the $3d$ -electron of cobalt has some contribution. These results indicate that the superconductivity may mainly originate from the $5d$ -electrons of tantalum.

The new Ta₄CoSi superconductor has unique and elegant honeycomb tantalum networks in the (110) plane, as displayed in Fig. 1(b). And the first-principal calculations indicate that the superconductivity may mainly originate from the $5d$ -electrons of tantalum. Thus, we compare the new Ta₄CoSi superconductor with other hexagonal metal-chain-network superconductors and the remarkably high T_c MgB₂ superconductor with the B-B honeycomb network. These compounds have a single-atom honeycomb network structure. Table III summarizes the parameters associated with superconductivity

TABLE II. Observed superconducting parameters of Ta₄CoSi.

Parameter	Unit	Value
T_c	K	2.45
$\mu_0 H_{c1}(0)$	mT	9.86
$\mu_0 H_{c2}(0)$	T	0.84
$\mu_0 H^{\text{Pauli}}$	T	4.53
$\Delta C/\gamma T_c$		0.58
Θ_D	K	346
λ_{ep}		0.51

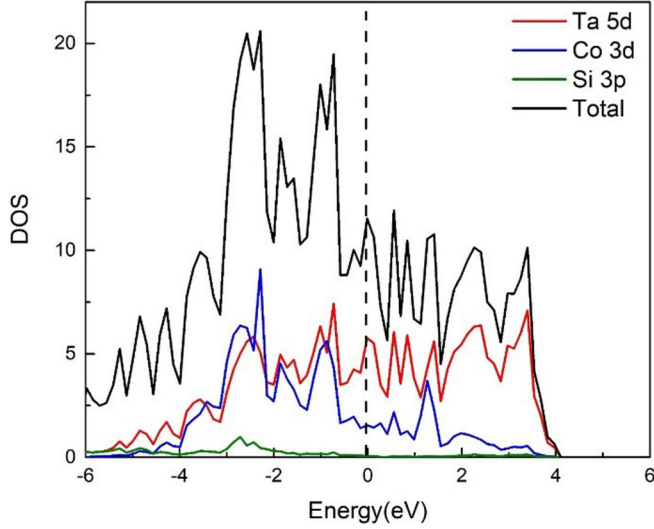


FIG. 7. Total and projected density of states calculated for Ta_4CoSi . The black dashed line indicates the Fermi level.

and the structures of hexagonal metal-chain-network-based compounds and the MgB_2 superconductor. The T_c as a function of the honeycomb lattice in honeycomb network superconductors is shown in Fig. 8. Interestingly, the T_c is roughly linearly related to the honeycomb lattice. The new Ta_4CoSi superconductor follows the relationship between T_c and the honeycomb lattice function. T_c gradually decreases from MgB_2 with the boron honeycomb network to KV_3Sb_5 with the antimony honeycomb network as the honeycomb lattice from 1.781 to 3.160 Å. At the same time, the Θ_D decreases with the increase of the honeycomb lattice, which is displayed in the illustration in Fig. 8. The phenomenon that the T_c is negatively correlated with the bond length is also observed in niobium-based superconductors [20]. Besides, the shrinkage of the iron honeycomb lattice of layered FePX_3 ($X = \text{S}$ and Se) under high pressure leads to the emergence of superconductivity and the increase of T_c [41]. Thus, reducing the honeycomb lattice parameter may effectively increase the T_c and Θ_D in the honeycomb network superconducting materials. It will be worth searching for more new superconductors with a honeycomb network and confirming this relationship.

TABLE III. Comparison between the superconducting and structural parameters among honeycomb metal-chain-network superconductors and the MgB_2 superconductor (honeycomb network of boron atoms).

	MgB_2	$\alpha\text{-Ga}$	Nb_4NiSi	TaRh_2B_2	Ta_4CoSi	KV_3Sb_5
	[28]	[40]	[20]	[29]	[This work]	[30]
T_c (K)	39	10	7.7	5.8	2.45	0.93
Honeycomb lattice (Å)	1.781	2.720	2.779	2.795	2.943	3.160
Θ_D (K)	884	–	415	379	346	141
λ_{ep}	0.680	–	0.600	0.624	0.510	0.460

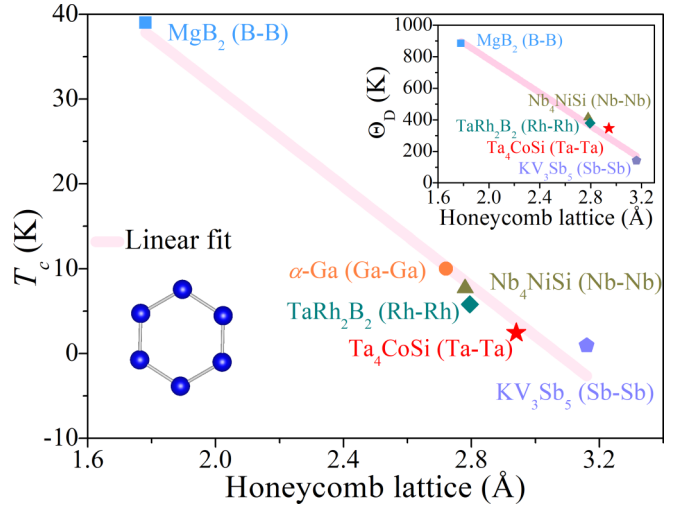


FIG. 8. The honeycomb lattice parameter dependence of T_c . The inset shows the honeycomb lattice dependence of Θ_D .

IV. CONCLUSION

In conclusion, we have studied the crystal structure and its superconducting properties of the novel superconductor Ta_4CoSi , prepared by the arc-melting method. The Ta_4CoSi compound adapts a CuAl_2 -type structure (space group $P4/mcc$, No. 124), featuring the honeycomb networks of Ta-Ta in the (110) plane. Our studies show that Ta_4CoSi is a type-II superconductor with $T_c \sim 2.45$ K, $\mu_0 H_{c1}(0) \sim 9.86$ mT, and $\mu_0 H_{c2}(0) \sim 0.84$ T. This work shows that reducing the honeycomb lattice parameter (adding pressure) may be an effective method of increasing the T_c in honeycomb network superconducting materials. Furthermore, Ta_4CoSi is a new platform for studying a unique crystal structure and superconductivity, which could lead to insights into the physics of Kagome superconductors, high- T_c superconductors, and topological superconductors [42,43].

ACKNOWLEDGMENTS

This work is supported by National Natural Science Foundation of China (11922415, 12274471, 11974432, 92165204, and U2032213), the National Key Research and Development Program of China (2017YFA0206203, 2018YFA0306001), the Guangdong Basic and Applied Basic Research Foundation (2022A1515011168, 2019A1515011718, 2019A1515011337), the Fundamental Research Funds for the Central Universities (19lgzd03), the Key Research & Development Program of Guangdong Province, China (2019B110209003), the Shenzhen International Quantum Academy (Grant No. SIQA202102), the Leading Talent Program of Guangdong Special Projects (201626003), the Pearl River Scholarship Program of Guangdong Province Universities and Colleges (20191001). J.S. is thankful for the funding from Institute of Physics Chinese Academy of Sciences Startup Funds (BR202118). G.T.L. is thankful for the funding from the China Postdoctoral Science Foundation (Grant No. 2022T150414).

- [1] M. Isobe, H. Yoshida, K. Kimoto, M. Arai, and E. Takayama-Muromachi, *Chem. Mater.* **26**, 2155 (2014).
- [2] T. Tamegai, Y. Nakajima, T. Nakagawa, G. J. Li, and H. Harima, *Sci. Technol. Adv. Mater.* **9**, 044206 (2008).
- [3] Y. Nakajima, T. Nakagawa, T. Tamegai, and H. Harima, *Phys. Rev. Lett.* **100**, 157001 (2008).
- [4] G. Seyfarth, A.-S. Rüetschi, K. Sengupta, A. Georges, D. Jaccard, S. Watanabe, and K. Miyake, *Phys. Rev. B* **85**, 205105 (2012).
- [5] S. Noguchi and K. Okuda, *Phys. B* **194–196**, 1975 (1994).
- [6] N. Kimura, K. Ito, K. Saitoh, Y. Umeda, H. Aoki, and T. Terashima, *Phys. Rev. Lett.* **95**, 247004 (2005).
- [7] I. Sugitani, Y. Okuda, H. Shishido, T. Yamada, and Y. Onuki, *J. Phys. Soc. Jpn.* **75**, 043703 (2006).
- [8] E. Bauer, R. T. Khan, H. Michor, E. Royanian, A. Grytsiv, N. Melnychenko-Koblyuk, P. Rogl, D. Reith, R. Podloucky, E. W. Scheidt, W. Wolf, and M. Marsman, *Phys. Rev. B* **80**, 064504 (2009).
- [9] R. R. Chowdhury, S. Dhara, I. Das, B. Bandyopadhyay, and R. Rawat, *J. Magn. Magn. Mater.* **451**, 625 (2018).
- [10] S. Tencé, S. Gorse, E. Gaudin, and B. Chevalier, *Intermetallics* **17**, 115 (2009).
- [11] F. Kneidinger, H. Michor, E. Bauer, A. Griбанov, A. Lipatov, Y. Seropegin, J. Sereni, and P. Rogl, *Phys. Rev. B* **88**, 024423 (2013).
- [12] N. Kase, H. Suzuki, T. Nakano, and N. J. S. S. Takeda, *Supercond. Sci. Tech.* **29**, 035011 (2016).
- [13] S. Yashiro, A. Kasahi, R. Kasai, H. Samata, and Y. Nagata, *J. Alloy. Compd.* **309**, 51 (2000).
- [14] H. Suzuki, N. Kase, T. Nakano, and N. Takeda, *Phys. Procedia* **81**, 57 (2016).
- [15] F. Kneidinger, H. Michor, A. Sidorenko, E. Bauer, I. Zeiringer, P. Rogl, C. Blaas-Schenner, D. Reith, and R. Podloucky, *Phys. Rev. B* **88**, 104508 (2013).
- [16] D. I. Walicka, Z. Guguchia, J. Lago, O. Blacque, K. Y. Ma, H. Liu, R. Khasanov, and F. O. von Rohr, *Phys. Rev. Res.* **3**, 033192 (2021).
- [17] C. Mielke, Y. Qin, J. X. Yin, H. Nakamura, D. Das, K. Guo, R. Khasanov, J. Chang, Z. Q. Wang, S. Jia, S. Nakatsuji, A. Amato, H. Luetkens, G. Xu, M. Z. Hasan, and Z. Guguchia, *Phys. Rev. Mater.* **5**, 034803 (2021).
- [18] B. Li, S. Li, and H.-H. Wen, *Phys. Rev. B* **94**, 094523 (2016).
- [19] E. I. Gladyshevskii and Y. B. Kuz'ma, *J. Struct. Chem.* **6**, 60 (1965).
- [20] G. Ryu, S. W. Kim, S. Matsuishi, H. Kawaji, and H. Hosono, *Phys. Rev. B* **84**, 224518 (2011).
- [21] J. Rodríguez-Carvajal, *Phys. B* **192**, 55 (1993).
- [22] G. Kresse and J. Hafner, *Phys. Rev. B* **47**, 558 (1993).
- [23] G. Kresse and J. Furthmüller, *Phys. Rev. B* **54**, 11169 (1996).
- [24] J. P. Perdew, K. Burke, and M. Ernzerhof, *Phys. Rev. Lett.* **77**, 3865 (1996).
- [25] P. E. Blöchl, *Phys. Rev. B* **50**, 17953 (1994).
- [26] V. Babizhetskyy, Y. Lomnytska, M. Dzevenko, C. Zheng, V. Smetana, and A. V. Mudring, *J. Alloy. Compd.* **864**, 158122 (2021).
- [27] F. von Rohr, M. J. Winarski, J. Tao, T. Klimczuk, and R. J. Cava, *Proc. Natl. Acad. Sci. USA* **113**, E7144 (2016).
- [28] J. Nagamatsu, N. Nakagawa, T. Muranaka, Y. Zenitani, and J. Akimitsu, *Nature (London)* **410**, 63 (2001).
- [29] E. M. Carnicom, W. Xie, T. Klimczuk, J. Lin, and R. J. Cava, *Sci. Adv.* **4**, 7969 (2018).
- [30] B. R. Ortiz, L. C. Gomes, J. R. Morey, M. Winarski, M. Bordelon, J. S. Mangum, I. W. H. Oswald, J. A. Rodriguez-Rivera, J. R. Nelison, S. D. Wilson, E. Ertekin, T. M. McQueen, and E. S. Toberer, *Phys. Rev. Mater.* **3**, 094407 (2019).
- [31] R. Prozorov and V. G. Kogan, *Phys. Rev. Appl.* **10**, 014030 (2018).
- [32] D. C. Johnston, *Supercond. Sci. Tech.* **26**, 115011 (2013).
- [33] Y. Bang and A. V. Balatsky, *Phys. Rev. B* **69**, 212504 (2004).
- [34] D.-J. Jang, J. B. Hong, Y. S. Kwon, T. Park, K. Gofryk, F. Ronning, J. D. Thompson, and Y. Bang, *Phys. Rev. B* **85**, 180505(R) (2012).
- [35] Y. Bang, H.-Y. Choi, and H. Won, *Phys. Rev. B* **79**, 054529 (2009).
- [36] A. Gallagher, K.-W. Chen, C. M. Moir, S. K. Cary, F. Kametani, N. Kikugawa, D. Graf, T. E. Albrecht-Schmitt, S. C. Riggs, A. Shekhter, and R. E. Baumbach, *Nat. Commun.* **7**, 10712 (2016).
- [37] Q. Zhang, G. Li, D. Rhodes, A. Kiswandhi, T. Besara, B. Zeng, J. Sun, T. Siegrist, M. Johannes, and L. Balicas, *Sci. Rep.* **3**, 1682 (2013).
- [38] U. Welp, R. Xie, A. E. Koshelev, W. K. Kwok, P. Cheng, L. Fang, and H. H. Wen, *Phys. Rev. B* **78**, 140510(R) (2008).
- [39] W. L. McMillan, *Phys. Rev.* **167**, 331 (1968).
- [40] M. Petrov, J. Bekaert, and M. V. Milošević, *2D Mater.* **8**, 035056 (2021).
- [41] Y. Wang, J. Ying, Z. Zhou, J. Sun, T. Wen, Y. Zhou, N. Li, Q. Zhang, F. Han, Y. Xiao, P. Chow, W. Yang, V. V. Struzhkin, Y. Zhao, and H.-K. Mao, *Nat. Commun.* **9**, 1914 (2018).
- [42] Z. Zhu, M. Papaj, X.-A. Nie, H.-K. Xu, Y.-S. Gu, X. Yang, D. Guan, S. Wang, Y. Li, C. Liu, J. Luo, Z.-A. Xu, H. Zheng, L. Fu, and J.-F. Jia, *Science* **374**, 1381 (2021).
- [43] T. Murtaza *et al.*, *Phys. Rev. B* **106**, L060501 (2022).