Large-gap quantum anomalous Hall states induced by functionalizing buckled Bi-III monolayer/Al₂O₃

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Chiral edge modes inherent to the topological quantum anomalous Hall (QAH) effect are a pivotal topic of contemporary condensed matter research aiming at future quantum technology and application in spintronics. A large topological gap is vital to protecting against thermal fluctuations and thus enabling a higher operating temperature. From first-principles calculations, we propose Al₂O₃ as an ideal substrate for atomic monolayers consisting of Bi and group-III elements, in which a large-gap quantum spin Hall effect can be realized. Additional half-passivation with nitrogen then suggests a topological phase transition to a large-gap QAH insulator. By effective tight-binding modeling, we demonstrate that Bi-III monolayer/Al₂O₃ is dominated by p_x , p_y orbitals, with subdominant p_z orbital contributions. The topological phase transition into the QAH is induced by Zeeman splitting, where the off-diagonal spin exchange does not play a significant role. The effective model analysis promises utility far beyond Bi-III monolayer/Al₂O₃, as it should generically apply to systems dominated by p_x , p_y orbitals with a band inversion at Γ .

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I. INTRODUCTION

Since the celebrated discovery of the quantum anomalous Hall (QAH) effect [1,2], the study of two-dimensional (2D) topological quantum systems and the search for new material platforms with a large topological gap has gained significant attention due to their strong potential in spintronic applications [3-6]. The chiral edge state in QAH systems travels dissipationless along the edge and is believed to provide a promising solution to the Moore's Law issue in the silicon industry. As the analog of the integer quantum Hall effect under a strong magnetic field, the QAH acquires the quantized Hall conductivity by intrinsic magnetism without Landau levels. Since the theoretical proposal by Haldane [7], however, little progress was made in experiments on QAH systems before the field was cross-fertilized by the discovery of the 2D quantum spin Hall (QSH) effect [8,9] and 3D topological insulators (TIs) [5,10–13]. Note that the QAH is distinctly different from Chern insulator realizations in classical platforms, where the dissipationless character of the topological edge modes is not found [14]. The discovery of the QSH and 3D TIs implicitly paved the way for the realization of the QAH in experiments, as the field turned its focus to

materials with significant spin-orbit coupling (SOC) and a careful analysis of bulk and surface effect in semiconductor composites [2,15–19].

QSH systems and 3D TIs are time-reversal (TR) invariant topological systems. Their Hamiltonian is block-diagonal with different angular momenta $j = l \pm 1/2$ being TR partners. They form two subsystems that are also individually topological. Each subsystem can be viewed as a OAH system. Their combination under TR symmetry generates the QSH or the surface states in 3D TIs. Thus, breaking TR can potentially revert the above combination to yield the QAH. The essential condition of this process is to retain one of the subsystem's topological characters while dissolving the other one [20-23]. Theoretical proposals on magnetically doped HgTe/CdTe [24] and InAs/GaSb [25] quantum wells nicely fulfilled the required criteria. Unfortunately, experimental attempts did not succeed in realizing the QAH due to the failure in establishing long-range ferromagnetic order. Summarizing this direction of attempts, the immediate manipulation of the 2D QSH states does not seem particularly promising in achieving QAH at first sight. Instead, the final breakthrough was made along the other route, i.e., magnetically doping the surface states of 3D TIs [1,2]. Experiments found that doping tetradymite semiconductors Bi2Te3, Bi2Se3, and Sb2Te3 with Cr, Mn, and V can significantly enhance the bulk spin susceptibility and establish a ferromagnetic insulating phase through the van Vleck mechanism [1,2,26-30]. At the same time,

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the ferromagnetic exchange coupling makes one copy of the QAH subsystem revert to normal band order and, therefore, lose its nontrivial topology. Consequently, only one edge state survives, leading to the QAH. In recent years, the MnBi₂Te₄-family materials have also provided a platform for QAH with intrinsic magnetism from the inserted MnTe bilayer [31]. The interplay between the magnetic coupling and the topologically nontrivial bands endows the MnBi₂Te₄-family materials with rich topological phases [32–43].

Despite the successful realization of the QAH in magnetically doped tetradymite semiconductors and MnBi2Te4family materials, the operating temperature is limited to a relatively low value. Therefore, new QAH materials or QSH parent compounds with large topological gaps are strongly sought after. A large gap QSH state was experimentally realized in the monolayer bismuthene system [44,45]. This system is geometrically equivalent to graphene, while a completely different mechanism characterizes its low-energy excitation. The σ -bond states form an effectively two-orbital Kane-Mele model, which encourages a large on-site SOC that supports a topological gap as large as 0.8 eV. Bismuthene is a QSH system with the so-far largest topological gap confirmed in the experiment, which thus can potentially be used in roomtemperature spintronic device applications.

This paper revisits 2D QSH parent compounds as a hitherto less successful recipe to reach the QAH. However, as a gamechanging starting point, we employ the room-temperature QSH in bismuthene as the parent state for our QAH design. We start from a large-gap QSH system and introduce the long-range ferromagnetism by breaking the complete shell configuration of certain atoms, such that TR will be broken. Our idea is inspired by recent studies on several 2D materials with a honeycomb structure of elements of groups IV [46-48], V [49-53], and III-V [51,54-60]. QAH states were found in buckled Bi-III honeycomb systems by chemically decorating one side with hydrogen and the other side with nitrogen. As a result, a net magnetic moment can be established [61]. Furthermore, we employ the fact that functionalizing semiconductors with nitrogen is an experimentally feasible operation; for example, it induces a magnetic moment in 3D TI Bi₂Te₃ [62]. Thus, introducing charge imbalance and breaking complete shell configurations by functionalizing large-gap QSH insulators with nitrogen atoms may result in a large-gap QAH scenario.

We propose the experimentally feasible monolayer/ substrate combination, i.e., Bi-III monolayer on Al₂O₃, as a new platform for realizing a large-gap QSH. It represents a promising material candidate similar to recent proposals [63,64], with high experimental feasibility for epitaxial synthesis. We further tune these QSH systems to large-gap QAH insulators by functionalizing them with nitrogen with the largest bandgap as large as 405 meV. In particular, we provide a unified, effective Hamiltonian description of QSH and QAH states that originate from 2D honeycomb systems with low-energy excitations dominated by σ bonds. Our paper is organized as follows. In Sec. II, we propose α -Al₂O₃ as the appropriate substrate for Bi-III monolayer that a large-gap QSH can be realized. Based on these systems, we further modify them by functionalizing bismuth with nitrogen to introduce long-range ferromagnetism. In Sec. III, we provide a detailed understanding of the QSH and QAH in the Bi-III monolayer/ Al_2O_3 , as well as the topological transition between them by explicitly constructing effective tight-binding models.

II. MATERIAL PLATFORMS

Corundum, i.e., $\alpha - Al_2O_3$, is a typical substrate material, and α is its most stable phase in nature. It is widely used in experiments to epitaxial grow topological thin films [65–68]. The experimental lattice constant of Al_2O_3 is 4.761 Å [69]. The calculated equilibrium lattice constant of the three Bi-III monolayer systems studied in this work are 4.928 Å (BiTl), 4.805 Å (BiIn), and 4.521 Å (BiGa) [54]. Thus, the applied strains on Bi-III monolayers range from -5% to 3.5%, indicating high feasibility for their experimental growth on Al_2O_3 .

first-principles calculations are based on the Our generalized-gradient approximation in the Perdew-Burke-Ernzerhof form [70-74] within the framework of the density-functional theory (DFT) using projector-augmentedwave [75] wave functions as implemented in the Vienna Ab Initio Simulation Package (VASP) [76,77]. The effect of Van der Waals (VdW) interactions was taken into account by using the empirical correction scheme of Grimme (DFT-D2) [78]. The cut-off energy for the wave functions was set at 500 eV. To simulate the experimental situation, we only allow the atomic positions of atom Bi and group III elements to relax while keeping the lattice constant and the atomic positions of the substrate unchanged. Atomic positions were optimized for each lattice constant value considered until the residual forces were less than 5×10^{-3} eV/Å. The self-consistent convergence threshold for total energy was set at 10^{-5} eV. A vacuum layer of at least 20 Åalong the z direction was used to simulate thin films. A Γ-centered Monkhorst-Pack [79] grid of $9 \times 9 \times 1$ was used for 2D integrations in the Brillouin zone (BZ). In the phonon calculation, we use $5 \times 5 \times 1$ supercell, and the self-consistent convergence threshold for total energy was set at 10^{-7} eV. Finally, the edge states were calculated by using our in-house code LTM (Library for Topological Material calculations) with the iterative Green's function approach [80] based on the maximally localized Wannier functions [81] obtained through the VASP2WANNIER90 [82].

A. QSH

As Bi-III buckled monolayers alone are QSH systems [54], an appropriate substrate should less correlate with the Bi-III monolayers to maximally keep the topology of the latter. We, thus, consider the Al-terminated (0001) surface of Al_2O_3 as observed in experiments in our present work. If an Oterminated surface is used, additional hydrogen passivation would be needed [83]. All three Bi-III/Al₂O₃ systems studied in our work share similar conclusions. Therefore, we take BiIn/Al₂O₃ as an example in the main text. Results on other systems can be found in the Supplemental Material [84].

Figures 1(a) and 1(b) illustrate the crystal structure of BiIn/Al₂O₃. Here, we adopt a symmetric structure to simulate a semi-infinite substrate. Otherwise, the other side of the



FIG. 1. (a), (b) Structural model of $BiIn/Al_2O_3$ in top and side views. (c) Half-passivation of BiIn with nitrogen. (d) The 2D Brillouin zone and the high-symmetry *k* path adopted in the calculation. (e) Electronic structure of $BiIn/Al_2O_3$ without SOC (blue dashed line) and with SOC (red solid line). (f) Calculations of topological edge mode of a ribbon with zigzag termination. (g) and (h) are same as (e) and (f), but for nitrogen half-passivated $BiIn/Al_2O_3$.

substrate needs to be carefully saturated to avoid any dangling state to interfere with the Fermi level. The BiIn monolayer binds to the Al-terminated surface of Al_2O_3 via VdW force. Full relaxation of the BiIn layer results in a separation of 1.68 Å from the substrate surface and a bucking of 1.20 Å amplitude between the Bi and In layers.

The corresponding bulk electronic structure is displayed in Fig. 1(e) with the BZ shown in Fig. 1(d). In the energy range shown in Fig. 1(e), all states are from BiIn. The substrate states are far away from the Fermi level without interfering with the topology of the BiIn monolayer. Without SOC, the system is a semimetal with two parabolic bands touching at the Γ point. When SOC is included, the system becomes a semiconductor and opens a bandgap as large as 420 meV, which is smaller than bismuthene/SiC(0001) [44], but much larger than HgTe [8,9] and InSb/GaAs [85]. This gap is topological nontrivial. A direct calculation of a semi-infinite ribbon with a zigzag boundary gives two counterpropagating edge modes connecting the valence and conduction bands, which is the characteristic of 2D QSH insulators. The topological edge states reside right in the middle of the bulk bandgap, which is of high utility for device application.

In contrast to bismuthene/SiC(0001), the nontrivial topology here stems from the band inversion of $p_{x/y}$ orbitals. In Fig. S4, we show the orbital components for the states around the Fermi level, which are governed by all three p orbitals of Bi and In. Thus, the low-energy model description of BiIn/Al₂O₃ will not be the simple p_x and p_y model [44,45]. Due to the buckling structure of the BiIn monolayer, p_z enters as an active orbital. However, the band inversion is between $p_{x/y}$. Thus, the p_z orbital plays a marginal role here. Nevertheless, due to the participation of the p_z orbital, the topological gap is no longer proportional to the onsite strength of $l_z s_z$ SOC. Thus, the size of topological gaps is reduced in these buckled system as compared to bismuthene/SiC(0001) [44].

B. QAH from chemical functionalization

There are two strategies to break TR symmetry and introduce QAH in 2D systems, i.e., by passivation with functionality and magnetic doping with transition-metal atoms. So far, the candidate QAH materials theoretically proposed include HgTe quantum well magnetically doped with Mn [86,87], (Bi, Sb)₂Te₃ doped with Cr and V [1], group-IV and V honeycomb monolayers passivated by transition atoms [51,88–94] or proximity to magnetic substrates [95] and the intrinsic magnetic intercalation in MnBi₂Te₄ family materials [32–43]. Some of them have been experimentally examined and confirmed [2,15–17]. Their success stimulates our search for the QAH candidate systems with a larger topological gap and higher operating temperatures. After obtaining the large-gap QSH, we further explored the half-passivation of BiIn monolayer with H, OH, F, Cl, Br, I, and N. We find that only N-adsorption yields a substantial charge transfer to nitrogen atoms from bismuth and indium atoms. A charge imbalance between the spin-up and spin-down electrons on nitrogen forms which breaks the TR symmetry. Electrons at different nitrogen sites do not hop directly but through indium and bismuth atoms. Thus, the ferromagnetic couplings are mediated by the superexchange mechanism and favored by the Goodenough-Kanamori rule [96,97].

As illustrated in Fig. 1(c), the nitrogen atom connects to bismuth but not to indium, i.e., half-passivation. To remove the influence of electric dipole moment, we took a structure of Fig. 1(c) with space inversion symmetry by sandwiching Al_2O_3 with N-Bi-III on both sides. Spin-polarized band



FIG. 2. Electron localization function (ELF) of $BiIn/Al_2O_3$ before and after nitrogen adsorption. The crystal model overlays on the ELF map showing the valence charge centers. Al_2O_3 substrate displays the expected ionic-bonding nature with the positive and negative valences at Al and O sites, respectively. Bi-In and In-N bonds share valence electrons displaying a strong covalent bond nature. The signed number on the shoulder of each element denotes its valence.

calculation without SOC shows that chemical adsorption of nitrogen leads to an intrinsic magnetic moment of 2.0 μ_B per unit cell due to the uneven distribution of the additional charge between the spin-up and spin-down states of the nitrogen atom. The ferromagnetic state is energetically more stable than the nonmagnetic and antiferromagnetic states. Local moments mainly form at N and In. The nearest neighbor spin exchange coupling and the magnetic anisotropy energies are estimated as -8.65 meV and 13.7 meV. By simulating a classical Heisenberg model with the ESpinS package [98], we estimate the Curie temperature as $T_c \sim 110$ K. As a consequence of the ferromagnetism, we obtain a half-metal, i.e., spin-up bands are gapless while spin-down bands are gapped. Two spin-up bands degenerate at the Γ point.

Before discussing the topological nature, we first explain how the local magnetic moments form due to the halfpassivation of nitrogen. To this end, we calculated the electron localization function and the ionic charge of Bi, In, and N from the Bader-Charge method [99–102]. The former helps to understand the bonding nature of Bi, In, and N. The latter directly tells us the change of valence charge after nitrogen adsorption. Without nitrogen, bismuth/indium gains/loses valence electrons, respectively, forming valence Bi0.48- and $In^{0.49+}$. Such valence configurations are far from their formal valence. Thus, they will not form an ionic bond. As shown in Fig. 2(a), bismuth and indium share valence charges and form a covalent bond with strong metallic nature in between. The Al-O bond in the substrate, however, remains ionic, and BiIn monolayer bonds with Al₂O₃ via VdW bonding. After adsorption with nitrogen, the valence of indium does not change, but bismuth transfers valence electrons to nitrogen. Similar to the Bi-In covalent bond shown in Fig. 2(a), after nitrogen adsorption Bi-In bond remains covalent with a metallic nature. They form a positive valence center bonding with the negative valence center $N^{0.52-}$, indicating a strong hybridization of the electronic states on these atoms. Thus, the charge polarization at the nitrogen site (see Table I) will lift the spin degeneracy of the electronic structures. As a result, bismuth and indium stay in the intrinsic magnetic field created by nitrogen charge polarization, and the electronic structure becomes spin-polarized. The strong hybridization of the nitrogen, bismuth, and indium orbitals can also be seen from the projective electronic structure in Fig. S3 of the Supplemental Material [84].

We also find that, except for the different binding energy levels, the spin-up and spin-down band curvatures are slightly different, implying a weak spin-exchange interaction between the spin-up and spin-down electrons. Only Zeeman splitting plays a dominant role. As BiIn is a QSH insulator, both spinup and spin-down bands are topologically nontrivial. Each contributes a dissipationless edge mode. The two edge currents propagate in opposite directions and relate by the TR symmetry. Changing such a QSH state to a QAH state is sufficient to shift one spin component away from the Fermi level. Thus, there is no need for exchange coupling to transform topologically trivial bands into nontrivial ones.

After understanding the origin of the ferromagnetic long-range order and the fully spin-polarized low-energy excitations, we further calculated the edge states with a zigzag termination in Fig. 1(h). A helical edge state exists connecting the valence band and conduction band corresponding

TABLE I. The gain/loss of valence electrons (-/+) before and after nitrogen-adsorption, the charge polarization, and the magnetic moments of N, Bi and the III group elements.

		Ionic charge change		Charge Pola.	$M(\mu_B)$
	N		0.57-	+1.07	0.822
N-BiTl	Bi	0.20 -	0.25 +	-0.14	-0.012
	Tl	0.22 +	0.38 +	+0.07	-0.003
	Ν		0.52-	+1.11	0.877
N-BiIn	Bi	0.48 -	0.10 +	-0.16	-0.021
	In	0.49 +	0.47 +	-0.03	-0.039
	Ν		0.56-	+1.08	0.827
N-BiGa	Bi	0.42 -	0.18 +	-0.12	-0.011
	Ga	0.49+	0.41 +	+0.05	-0.005

to a Chern number C = 1. More importantly, for the BiIn topological bandgap shown in Fig. 1(g) remains as large as 266 meV, which still promises an excellent chance for the observation of the QAH in experiments. We note that nitrogen doping/adsorption is feasible in experiments and has successfully induced ferromagnetic order in various semiconductors [103–106]. The QAH states in nitrogen adsorbed BiIn/Al₂O₃ benefit from two features of the *N* functionalization. First, the nitrogen spin polarization breaks the TR symmetry; Second, it does not induce additional states at the Fermi level. Consequently, a topological gap formed by spin-polarized band structure is obtained, which leads to the single helical edge mode.

III. TIGHT-BINDING MODEL

To better understand the low-energy physics and the topological transition between the QSH and QAH in N-BiIn monolayer/Al₂O₃, we analytically construct a tight-binding model (TBM) based on all *p* orbitals of Bi and In, which reproduces the DFT electronic structure and the topological nature of the above DFT calculations. We took the basis functions $|p_{x\sigma}^{Bi}\rangle$, $|p_{y\sigma}^{Bi}\rangle$, $|p_{z\sigma}^{In}\rangle$, $|p_{y\sigma}^{In}\rangle$, $|p_{z\sigma}^{In}\rangle$, and construct an effective Hamiltonian with the following form:

$$\mathcal{H} = H_0 + H_M + H_{SO},$$

$$H_0 = \sum_{i\alpha\sigma} \varepsilon^{\alpha}_{i\sigma} c^{\alpha,\dagger}_{i\sigma} c^{\alpha}_{i\sigma} + \sum_{\langle i,j\rangle\alpha\beta\sigma} t^{\alpha\beta}_{ij\sigma} (c^{\alpha,\dagger}_{i\sigma} c^{\beta}_{j\sigma} + \text{h. c.}),$$

$$H_M = -\sum_{i\sigma\sigma'} \lambda^i_m c^{\dagger}_{i\sigma} c_{i\sigma'} [\hat{\mathbf{m}} \cdot \hat{\mathbf{s}}]_{\sigma\sigma'},$$

$$H_{SO} = \sum_{i\alpha\beta\sigma\sigma'} \langle \alpha\sigma | \lambda_{SO} \vec{L} \cdot \vec{S} | \beta\sigma' \rangle c^{\alpha,\dagger}_{i\sigma} c^{\beta}_{i\sigma'}.$$
(1)

 H_0 denotes the TBM resulting only from single-particle hoppings. $\varepsilon_{i\sigma}^{\alpha}$, $c_{i\sigma}^{\alpha,\dagger}$ represent on-site energy, electron annihilation/creation operators at the α orbital and the *i*th atom, respectively. $t_{ij\sigma}^{\alpha\beta}$ is the coupling strength of electrons with spin σ at the α orbital of the *i*th atom with those electrons at the β orbital of the *j*th atom, which can be easily calculated from Slater-Koster integrals [107]. H_M and H_{SOC} are the corresponding magnetic coupling and SOC terms. \vec{L} and \vec{S} denote the orbital and spin angular momentum operators, and λ_{SO} is the strength of SOC. The potential difference between A and B sublattices reflects the corresponding environmental difference around Bi and In, breaking sublattice symmetry. We note that H_0 and H_{SOC} are similar to the TBM of Bi-III monolayer on SiO_2 [63]. Additionally, in the current model, a TR symmetry breaking term H_M is also introduced, accounting for the Zeeman splitting, which shifts the two spin-degenerate bands up and down with an equal amount of energy. $\hat{\mathbf{m}}$ is the direction of magnetic polarization and λ_m^i is the magnitude of magnetic polarization.

Because the presence of the intrinsic magnetization breaks time-reversal symmetry, we use two different sets of parameters to construct the spin-up and spin-down components in H_0 . Discarding all terms breaking spin conservation which are not important at the level of our discussion, the Hamiltonian H_0 (12 × 12) can be cast as a direct sum of two 6 × 6 Hamiltonian for the two spin sectors:

$$H_0 = \begin{pmatrix} H_{\uparrow\uparrow} & 0\\ 0 & H_{\downarrow\downarrow} \end{pmatrix}$$
(2)

with

$$H_{\sigma\sigma} = \begin{pmatrix} \epsilon^{\sigma}_{Bi,px} & 0 & 0 & h^{\sigma}_{xx} & h^{\sigma}_{xy} & h^{\sigma}_{xz} \\ 0 & \epsilon^{\sigma}_{Bi,py} & 0 & h^{\sigma}_{yx} & h^{\sigma}_{yy} & h^{\sigma}_{yz} \\ 0 & 0 & \epsilon^{\sigma}_{Bi,pz} & h^{\sigma}_{zx} & h^{\sigma}_{zy} & h^{\sigma}_{zz} \\ \vdots & \vdots & \vdots & \epsilon^{\sigma}_{In,px} & 0 & 0 \\ \vdots & \vdots & \vdots & 0 & \epsilon^{\sigma}_{In,py} & 0 \\ \vdots & \vdots & \vdots & 0 & 0 & \epsilon^{\sigma}_{In,pz} \end{pmatrix},$$
(3)

where † represents complex conjugation. We found that, to decently reproduce the DFT electronic structure, it is sufficient to consider only the nearest neighbor hopping between different sites. More details of the model parameters can be found in the Supplemental Material [84].

In N-BiIn/Al₂O₃, the direction of magnetic polarization $\hat{\mathbf{m}} = (0, 0, 1)$ is normal to the monolayer plane. Consequently, the Zeeman term only contributes diagonal elements to the effective Hamiltonian, i.e., intrinsic magnetization only leads to the spin-dependent energy shift and does not change the shape of the bands. We denote the new on-site energy level as

$$\Delta_{\alpha}^{\sigma} = \epsilon_{\alpha}^{\sigma} \mp \lambda_{m}^{\alpha}, \tag{4}$$

where - and + correspond to spin-up and spin-down electrons.

As for the SOC, we will only consider the simplest atomic SOC contribution, while discarding the Rashba contribution. The latter only affects the details of the electronic structure, but is not essential to the topological phase transition:

$$\langle p_y | \vec{L} \cdot \vec{S} | p_x \rangle = i\sigma_z,$$
 (5a)

$$\langle p_z | \vec{L} \cdot \vec{S} | p_x \rangle = -i\sigma_y,$$
 (5b)

$$\langle p_z | \vec{L} \cdot \vec{S} | p_y \rangle = i\sigma_x.$$
 (5c)

Due to the presence of p_z orbital in this model, additional SOC terms between p_z and $p_{x/y}$ orbitals appear as compared to the bismuthene model [44,45]. Adding SOC to the TBM opens a large global gap and induces a band inversion at Γ .

To obtain these model parameters, we mainly fit three spin-up bands around the Fermi level and three spin-down ones around 1 eV, as shown in Figs. 3(a) and 3(b). Other bands stay at higher binding energies. Thus, the quality of fitting on these bands is not crucial. Here, we primarily fit the spin-up bands as it determines the topological nature and gap size of the system. On the other hand, the spin-down bands stay away from the Fermi level and are less critical to the low-energy model. In Fig. 3, DFT bands are shown as blue dashed lines. Bands from the fitted TBM are shown as solid red lines. Spin-up and spin-down bands are different in shape, but degeneracy and symmetry remain identical. Using two different parameter sets for the spin-up and spin-down electrons will capture such a difference. When SOC is included, band inversion between the p_x and p_y orbitals occurs in the spinup and spin-down bands, see Fig. S4 for all three systems.



FIG. 3. Comparison of the *p*-orbital model in Eq. (1) and the DFT calculations. In (a)–(c), the blue dashed lines and the solid red lines correspond to the DFT results and the tight-binding fitting, respectively. (a), (b) Spin-polarized electronic structure without SOC. (c) Spin-polarized electronic structure with SOC. (d) The topological edge states calculations using the tight-binding model in a ribbon geometry with a zigzag edge. The blue and yellow circles represent the probability of the wave functions residing at the left edge and the right edge.

However, as the Zeeman field shifts the electronic structure of the spin-down electrons upwards by 1 eV, only band inversion in the spin-up sector remains at the Fermi level. Consequently, we effectively have a spin-polarized electronic structure with low-energy physics dominated by spin-up electrons only. The topological gap, thus, stems only from one spin component. As a result, the topological nature and the nontrivial edge mode depend only on one spin sector, i.e., a single helical edge mode is obtained. We confirm the above analysis by directly calculating the edge states in Fig. 3(d), where blue and yellow bands correspond to the edge modes residing at the two different terminations in a ribbon calculation. Along each edge, there is only one topological mode consistent with the DFT calculations shown in Fig. 1(h) and confirms the QAH nature of the proposed model in Eq. (1).

The proposed model correctly captures the QAH nature of N-BiIn/Al₂O₃, and it is generic to all four material systems studied in this work. In Fig. S7, we further show the comparison of the fitted model to the DFT calculations for other materials and the calculated helical edge modes from the model. The first three rows correspond to the spin-up, spin-down, and SOC bands. Our tight-binding model can explain the topology of a large variety of material systems. Replacing Bi with other group-V elements, such as Sb and As, is expected to result in similar band structures and the same QAH states. Although Al_2O_3 may not be a good substrate candidate anymore in those cases due to lattice mismatch, the model we proposed most likely still applies.

IV. DISCUSSIONS AND CONCLUSIONS

We have proposed using two different parameter sets to reproduce the DFT electronic structure. In this section, we further simplify the proposed model and explain the topological phase transition more transparently.

To obtain a universal effective modeling through simplification, we have identified two aspects of the full-scale generic model in Eq. (1) that turn out to be irrelevant for the topological nature and QSH-QAH phase transition in honeycomb-type electronic p-orbital models: (i) the adoption of different parameter sets for spin-up and spindown bands, and (ii) the p_z orbitals. Discarding them from Eq. (1), we arrive at a model on the following basis. $|p_{x\uparrow}^{Bi}\rangle$, $|p_{y\uparrow}^{Bi}\rangle$, $|p_{y\downarrow}^{Ri}\rangle$, $|p_{y\downarrow}^{Bi}\rangle$, $|p_{y\downarrow}^{$

$$H_{\uparrow\uparrow} = H_{\downarrow\downarrow} = \begin{pmatrix} \epsilon_{Bi,px} & 0 & h_{xx} & h_{xy} \\ 0 & \epsilon_{Bi,py} & h_{yx} & h_{yy} \\ \dagger & \dagger & \epsilon_{X,px} & 0 \\ \dagger & \dagger & 0 & \epsilon_{X,py} \end{pmatrix}.$$
 (6)

The matrix elements can be found in Eq. (S1) [84]. Here, we have neglected the spin dependence of all matrix elements and require them to be identical in both spin sectors. The Zeeman splitting and the SOC term also take the same form as in Eq. (4) and Eq. (5).

This simplified model is sufficient to explain the topological phase transition induced by nitrogen absorption. We note that such a TBM is generic to all two-dimensional spinpolarized p_{xy} -orbital system [90–94]. In Fig. 4, we show the electronic structure of the simplified model. Without SOC and Zeeman splitting, this model shows two parabolic bands touching at the Fermi level as displayed in Fig. 4(a). Each band is doubly degenerate due to spin. SOC relieves the band degeneracy at Γ , leading to a topological QSH. This model contains two counterpropagating edge modes inside the bulk energy gap, as shown in Fig. 4(e). Half-passivation of nitrogen leads to the onset of a long-range ferromagnetic order, which induces the Zeeman splitting in our model. Upon turning on the Zeeman splitting, the spin-degenerate bands shown in Fig. 4(a) start to separate. We keep the spin-up band at the Fermi level by modifying the chemical potential, see Fig. 4(b). The separation of the two bands does not modify the topology of either band. When SOC is further included in Fig. 4(d), both bands become gapped and attain a topological character. Because the Fermi level is now only governed by the spin-up bands, however, even though the spin-down bands at 2 eV are topological as well, this model will only demonstrate one helical edge mode on each edge, as shown in Fig. 4(f). Thus, in Bi-III monolayer/Al₂O₃, the topological phase transition between QSH and QAH is mainly induced by Zeeman splitting. Both spin-up and spin-down bands remain topological, but one of them is removed from the Fermi level by the Zeeman field. Consequently, the aspired QAH state



FIG. 4. Electronic structure and topological phase transition in a $p_{x/y}$ + Zeeman model. (a), (c), (e) and (b), (d), (f) correspond to the TR symmetric and symmetry-broken cases. Electronic structures in (a)–(d) correspond to the calculations of the simplified model with (a) H_0 , (b) $H_0 + H_M$, (c) $H_0 + H_{SO}$, and (d) $H_0 + H_M + H_{SO}$ terms, respectively. (e) and (f) show the topological edge modes for the QSH and QAH states, respectively.

is reached. Here, magnetic order does a trivial job, and it induces the topological phase transition only by shifting the topological gap of one spin sector away from the Fermi level, while not destroying its topology. The latter would require strong off-diagonal spin-exchange couplings, which are small in the materials we have considered.

In summary, through first-principles calculations, we propose that Al_2O_3 can be a promising substrate candidate to grow binary monolayers consisting of group III elements (Al, Ga, In, Tl) and bismuth (Bi). These systems belong to the same category of the large-gap QSH states as bismuthene/SiC(0001), with topological gaps of hundreds of meV. When further half-passivated with nitrogen, long rang

ferromagnetic order is induced by breaking the complete shell configuration of N valence electrons. Consequently, a transformation from QSH to QAH phases occurs. The topological gap remains considerably large in the QAH state, which promises a great chance to be in reach for application in spintronic devices.

We have further provided an analytical understanding of their low-energy topological physics by constructing a generic TBM, which reveals that the topological phase transition between QSH and QAH is essentially induced by Zeeman splitting. The nontrivial character of both the spin-up and spin-down bands remain unaffected by the onset of ferromagnetism. At the same time, the topological gap of one spin is removed from the Fermi level, which effectively creates a spin-polarized half-metal and, thus, the QAH under SOC. Our work provides a generalization of the bismuthene platform to the case of broken TR symmetry. We believe that the proposed TBM applies to all similar systems dominated by p_x , p_y orbitals, which will allow us to supplement the experimental effort with theoretical guidance along with the search for large-gap QAH systems.

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- R. Yu, W. Zhang, H.-J. Zhang, S.-C. Zhang, X. Dai, and Z. Fang, Science **329**, 61 (2010).
- [2] C.-Z. Chang, J. Zhang, X. Feng, J. Shen, Z. Zhang, M. Guo, K. Li, Y. Ou, P. Wei, L.-L. Wang, Z.-Q. Ji, Y. Feng, S. Ji, X. Chen, J. Jia, X. Dai, Z. Fang, S.-C. Zhang, K. He, Y. Wang *et al.*, Science **340**, 167 (2013).
- [3] A. Bansil, H. Lin, and T. Das, Rev. Mod. Phys. 88, 021004 (2016).
- [4] X.-L. Qi and S.-C. Zhang, Rev. Mod. Phys. 83, 1057 (2011).
- [5] M. Z. Hasan and C. L. Kane, Rev. Mod. Phys. 82, 3045 (2010).
- [6] N. Nagaosa, J. Sinova, S. Onoda, A. H. MacDonald, and N. P. Ong, Rev. Mod. Phys. 82, 1539 (2010).
- [7] F. D. M. Haldane, Phys. Rev. Lett. 61, 2015 (1988).

- [8] B. A. Bernevig, T. L. Hughes, and S.-C. Zhang, Science 314, 1757 (2006).
- [9] M. Kon"ig, S. Wiedmann, C. Brüne, A. Roth, H. Buhmann, L. W. Molenkamp, X.-L. Qi, and S.-C. Zhang, Science 318, 766 (2007).
- [10] H. Zhang, C.-X. Liu, X.-L. Qi, X. Dai, Z. Fang, and S.-C. Zhang, Nat. Phys. 5, 438 (2009).
- [11] Y. L. Chen, J. G. Analytis, J.-H. Chu, Z. K. Liu, S.-K. Mo, X. L. Qi, H. J. Zhang, D. H. Lu, X. Dai, Z. Fang, S. C. Zhang, I. R. Fisher, Z. Hussain, and Z.-X. Shen, Science 325, 178 (2009).
- [12] J. G. Analytis, J.-H. Chu, Y. Chen, F. Corredor, R. D. McDonald, Z. X. Shen, and I. R. Fisher, Phys. Rev. B 81, 205407 (2010).

- [13] Y. Zhang, K. He, C.-Z. Chang, C.-L. Song, L.-L. Wang, X. Chen, J.-F. Jia, Z. Fang, X. Dai, W.-Y. Shan *et al.*, Nat. Phys. 6, 584 (2010).
- [14] Z. Wang, Y. Chong, J. D. Joannopoulos, and M. Soljačić, Nature (London) 461, 772 (2009).
- [15] J. Checkelsky, R. Yoshimi, A. Tsukazaki, K. Takahashi, Y. Kozuka, J. Falson, M. Kawasaki, and Y. Tokura, Nat. Phys. 10, 731 (2014).
- [16] X. Kou, S.-T. Guo, Y. Fan, L. Pan, M. Lang, Y. Jiang, Q. Shao, T. Nie, K. Murata, J. Tang, Y. Wang, L. He, T.-K. Lee, W.-L. Lee, and K. L. Wang, Phys. Rev. Lett. **113**, 137201 (2014).
- [17] C.-Z. Chang, W. Zhao, D. Y. Kim, H. Zhang, B. A. Assaf, D. Heiman, S.-C. Zhang, C. Liu, M. H. Chan, and J. S. Moodera, Nat. Mater. 14, 473 (2015).
- [18] S. Grauer, K. M. Fijalkowski, S. Schreyeck, M. Winnerlein, K. Brunner, R. Thomale, C. Gould, and L. W. Molenkamp, Phys. Rev. Lett. **118**, 246801 (2017).
- [19] K. M. Fijalkowski, N. Liu, M. Hartl, M. Winnerlein, P. Mandal, A. Coschizza, A. Fothergill, S. Grauer, S. Schreyeck, K. Brunner, M. Greiter, R. Thomale, C. Gould, and L. W. Molenkamp, Phys. Rev. B 103, 235111 (2021).
- [20] J. Wang, B. Lian, and S.-C. Zhang, Phys. Scr. T164, 014003 (2015).
- [21] C.-X. Liu, S.-C. Zhang, and X.-L. Qi, Annu. Rev. Condens. Matter Phys. 7, 301 (2016).
- [22] X. Kou, Y. Fan, M. Lang, P. Upadhyaya, and K. L. Wang, Solid State Commun. 215-216, 34 (2015).
- [23] X.-L. Qi, T. L. Hughes, and S.-C. Zhang, Phys. Rev. B 78, 195424 (2008).
- [24] J.-c. Chen, J. Wang, and Q.-f. Sun, Phys. Rev. B 85, 125401 (2012).
- [25] Q.-Z. Wang, X. Liu, H.-J. Zhang, N. Samarth, S.-C. Zhang, and C.-X. Liu, Phys. Rev. Lett. 113, 147201 (2014).
- [26] Y. S. Hor, P. Roushan, H. Beidenkopf, J. Seo, D. Qu, J. G. Checkelsky, L. A. Wray, D. Hsieh, Y. Xia, S.-Y. Xu, D. Qian, M. Z. Hasan, N. P. Ong, A. Yazdani, and R. J. Cava, Phys. Rev. B 81, 195203 (2010).
- [27] X. Kou, M. Lang, Y. Fan, Y. Jiang, T. Nie, J. Zhang, W. Jiang, Y. Wang, Y. Yao, L. He, and K. L. Wang, ACS Nano 7, 9205 (2013).
- [28] X. Kou, L. He, M. Lang, Y. Fan, K. Wong, Y. Jiang, T. Nie, W. Jiang, P. Upadhyaya, Z. Xing, Y. Wang, F. Xiu, R. N. Schwartz, and K. L. Wang, Nano Lett. 13, 4587 (2013).
- [29] M. Li, C.-Z. Chang, L. Wu, J. Tao, W. Zhao, M. H. W. Chan, J. S. Moodera, J. Li, and Y. Zhu, Phys. Rev. Lett. **114**, 146802 (2015).
- [30] J. S. Dyck, C. Drašar, P. Lošt'ák, and C. Uher, Phys. Rev. B 71, 115214 (2005).
- [31] Z. Luo, C. Lin, W.-D. Cheng, W.-I. Zhang, Y.-b. Li, Y. Yang,
 H. Zhang, and Z. He, Cryst. Growth Des. 13, 4118 (2013).
- [32] M. M. Otrokov, I. P. Rusinov, M. Blanco-Rey, M. Hoffmann, A. Y. Vyazovskaya, S. V. Eremeev, A. Ernst, P. M. Echenique, A. Arnau, and E. V. Chulkov, Phys. Rev. Lett. **122**, 107202 (2019).
- [33] D. Zhang, M. Shi, T. Zhu, D. Xing, H. Zhang, and J. Wang, Phys. Rev. Lett. **122**, 206401 (2019).
- [34] J. Li, Y. Li, S. Du, Z. Wang, B.-L. Gu, S.-C. Zhang, K. He, W. Duan, and Y. Xu, Sci. Adv. 5, eaaw5685 (2019).
- [35] J. Ge, Y. Liu, J. Li, H. Li, T. Luo, Y. Wu, Y. Xu, and J. Wang, Natl. Sci. Rev. 7, 1280 (2020).

- [36] Y.-J. Hao, P. Liu, Y. Feng, X.-M. Ma, E. F. Schwier, M. Arita, S. Kumar, C. Hu, R. Lu, M. Zeng, Y. Wang, Z. Hao, H.-Y. Sun, K. Zhang, J. Mei, N. Ni, L. Wu, K. Shimada, C. Chen, Q. Liu *et al.*, Phys. Rev. X 9, 041038 (2019).
- [37] B. Li, J.-Q. Yan, D. M. Pajerowski, E. Gordon, A.-M. Nedić, Y. Sizyuk, L. Ke, P. P. Orth, D. Vaknin, and R. J. McQueeney, Phys. Rev. Lett. **124**, 167204 (2020).
- [38] A. Tan, V. Labracherie, N. Kunchur, A. U. B. Wolter, J. Cornejo, J. Dufouleur, B. Büchner, A. Isaeva, and R. Giraud, Phys. Rev. Lett. **124**, 197201 (2020).
- [39] B. Lian, Z. Liu, Y. Zhang, and J. Wang, Phys. Rev. Lett. 124, 126402 (2020).
- [40] X. Wu, J. Li, X.-M. Ma, Y. Zhang, Y. Liu, C.-S. Zhou, J. Shao, Q. Wang, Y.-J. Hao, Y. Feng, E. F. Schwier, S. Kumar, H. Sun, P. Liu, K. Shimada, K. Miyamoto, T. Okuda, K. Wang, M. Xie, C. Chen *et al.*, Phys. Rev. X **10**, 031013 (2020).
- [41] R. C. Vidal, H. Bentmann, J. I. Facio, T. Heider, P. Kagerer, C. I. Fornari, T. R. F. Peixoto, T. Figgemeier, S. Jung, C. Cacho, B. Büchner, J. van den Brink, C. M. Schneider, L. Plucinski, E. F. Schwier, K. Shimada, M. Richter, A. Isaeva, and F. Reinert, Phys. Rev. Lett. **126**, 176403 (2021).
- [42] H. Sun, B. Xia, Z. Chen, Y. Zhang, P. Liu, Q. Yao, H. Tang, Y. Zhao, H. Xu, and Q. Liu, Phys. Rev. Lett. **123**, 096401 (2019).
- [43] R. Lu, H. Sun, S. Kumar, Y. Wang, M. Gu, M. Zeng, Y.-J. Hao, J. Li, J. Shao, X.-M. Ma, Z. Hao, K. Zhang, W. Mansuer, J. Mei, Y. Zhao, C. Liu, K. Deng, W. Huang, B. Shen, K. Shimada *et al.*, Phys. Rev. X **11**, 011039 (2021).
- [44] F. Reis, G. Li, L. Dudy, M. Bauernfeind, S. Glass, W. Hanke, R. Thomale, J. Schäfer, and R. Claessen, Science 357, 287 (2017).
- [45] G. Li, W. Hanke, E. M. Hankiewicz, F. Reis, J. Schäfer, R. Claessen, C. Wu, and R. Thomale, Phys. Rev. B 98, 165146 (2018).
- [46] Y. Xu, B. Yan, H.-J. Zhang, J. Wang, G. Xu, P. Tang, W. Duan, and S.-C. Zhang, Phys. Rev. Lett. 111, 136804 (2013).
- [47] B.-H. Chou, Z.-Q. Huang, C.-H. Hsu, F.-C. Chuang, Y.-T. Liu, H. Lin, and A. Bansil, New J. Phys. 16, 115008 (2014).
- [48] T.-C. Wang, C.-H. Hsu, Z.-Q. Huang, F.-C. Chuang, W.-S. Su, and G.-Y. Guo, Sci. Rep. 6, 1 (2016).
- [49] C.-C. Liu, S. Guan, Z. Song, S. A. Yang, J. Yang, and Y. Yao, Phys. Rev. B 90, 085431 (2014).
- [50] Z. Song, C.-C. Liu, J. Yang, J. Han, M. Ye, B. Fu, Y. Yang, Q. Niu, J. Lu, and Y. Yao, NPG Asia Mater. 6, e147 (2014).
- [51] K.-H. Jin and S.-H. Jhi, Sci. Rep. 5, 1 (2015).
- [52] L. Chen, G. Cui, P. Zhang, X. Wang, H. Liu, and D. Wang, Phys. Chem. Chem. Phys. 16, 17206 (2014).
- [53] C.-H. Hsu, Z.-Q. Huang, C. P. Crisostomo, L.-Z. Yao, F.-C. Chuang, Y.-T. Liu, B. Wang, C.-H. Hsu, C.-C. Lee, H. Lin *et al.*, Sci. Rep. 6, 1 (2016).
- [54] F.-C. Chuang, L.-Z. Yao, Z.-Q. Huang, Y.-T. Liu, C.-H. Hsu, T. Das, H. Lin, and A. Bansil, Nano Lett. 14, 2505 (2014).
- [55] C. P. Crisostomo, L.-Z. Yao, Z.-Q. Huang, C.-H. Hsu, F.-C. Chuang, H. Lin, M. A. Albao, and A. Bansil, Nano Lett. 15, 6568 (2015).
- [56] L.-Z. Yao, C. P. Crisostomo, C.-C. Yeh, S.-M. Lai, Z.-Q. Huang, C.-H. Hsu, F.-C. Chuang, H. Lin, and A. Bansil, Sci. Rep. 5, 1 (2015).
- [57] L. Li, X. Zhang, X. Chen, and M. Zhao, Nano Lett. 15, 1296 (2015).

- [58] M. Zhao, X. Chen, L. Li, and X. Zhang, Sci. Rep. 5, 1 (2015).
- [59] Y. Ma, X. Li, L. Kou, B. Yan, C. Niu, Y. Dai, and T. Heine, Phys. Rev. B 91, 235306 (2015).
- [60] C.-C. Liu, J.-J. Zhou, and Y. Yao, Phys. Rev. B 91, 165430 (2015).
- [61] C. P. Crisostomo, Z.-Q. Huang, C.-H. Hsu, F.-C. Chuang, H. Lin, and A. Bansil, npj Comput. Mater. **3**, 1 (2017).
- [62] K.-H. Jin and S.-H. Jhi, J. Phys.: Condens. Matter 24, 175001 (2012).
- [63] Y. Xia, S. Jin, W. Hanke, R. Claessen, and G. Li, arXiv:2112.13483.
- [64] M. Chen and F. Liu, Natl. Sci. Rev. 8, nwaa241 (2020).
- [65] J. Hellerstedt, M. T. Edmonds, N. Ramakrishnan, C. Liu, B. Weber, A. Tadich, K. M. O'Donnell, S. Adam, and M. S. Fuhrer, Nano Lett. 16, 3210 (2016).
- [66] C.-Z. Chang, K. He, L.-L. Wang, X.-C. Ma, M.-H. Liu, Z.-C. Zhang, X. Chen, Y.-Y. Wang, and Q.-K. Xue, SPIN 01, 21 (2011).
- [67] N. Bansal, M. R. Cho, M. Brahlek, N. Koirala, Y. Horibe, J. Chen, W. Wu, Y. D. Park, and S. Oh, Nano Lett. 14, 1343 (2014).
- [68] Y. Cheng, S. Yu, M. Zhu, J. Hwang, and F. Yang, Phys. Rev. Lett. 123, 237206 (2019).
- [69] M. F. Peintinger, M. J. Kratz, and T. Bredow, J. Mater. Chem. A 2, 13143 (2014).
- [70] P. Hohenberg and W. Kohn, Phys. Rev. 136, B864 (1964).
- [71] W. Kohn and L. J. Sham, Phys. Rev. 140, A1133 (1965).
- [72] D. M. Ceperley and B. J. Alder, Phys. Rev. Lett. 45, 566 (1980).
- [73] J. P. Perdew and A. Zunger, Phys. Rev. B 23, 5048 (1981).
- [74] J. P. Perdew, K. Burke, and M. Ernzerhof, Phys. Rev. Lett. 77, 3865 (1996).
- [75] G. Kresse and D. Joubert, Phys. Rev. B 59, 1758 (1999).
- [76] G. Kresse and J. Hafner, Phys. Rev. B 47, 558 (1993).
- [77] G. Kresse and J. Furthmüller, Phys. Rev. B 54, 11169 (1996).
- [78] S. Grimme, J. Comput. Chem. 27, 1787 (2006).
- [79] H. J. Monkhorst and J. D. Pack, Phys. Rev. B 13, 5188 (1976).
- [80] M. P. L. Sancho, J. M. Lopez Sancho, J. M. L. Sancho, and J. Rubio, J. Phys. F 15, 851 (1985).
- [81] N. Marzari and D. Vanderbilt, Phys. Rev. B 56, 12847 (1997).
- [82] A. A. Mostofi, J. R. Yates, Y.-S. Lee, I. Souza, D. Vanderbilt, and N. Marzari, Comput. Phys. Commun. 178, 685 (2008).
- [83] B. Huang, Q. Xu, and S.-H. Wei, Phys. Rev. B 84, 155406 (2011).
- [84] See Supplemental Material at http://link.aps.org/ supplemental/10.1103/PhysRevB.106.125151 for more details on the crystal, electronic stuctures, and the topological analysis.

- [85] I. Knez and R.-R. Du, Front. Phys. 7, 200 (2012).
- [86] C.-X. Liu, X.-L. Qi, X. Dai, Z. Fang, and S.-C. Zhang, Phys. Rev. Lett. 101, 146802 (2008).
- [87] H.-C. Hsu, X. Liu, and C.-X. Liu, Phys. Rev. B 88, 085315 (2013).
- [88] Z. Qiao, S. A. Yang, W. Feng, W.-K. Tse, J. Ding, Y. Yao, J. Wang, and Q. Niu, Phys. Rev. B 82, 161414(R) (2010).
- [89] T. P. Kaloni, N. Singh, and U. Schwingenschlögl, Phys. Rev. B 89, 035409 (2014).
- [90] T. P. Kaloni, J. Phys. Chem. C 118, 25200 (2014).
- [91] S.-C. Wu, G. Shan, and B. Yan, Phys. Rev. Lett. 113, 256401 (2014).
- [92] S.-P. Chen, Z.-Q. Huang, C. P. Crisostomo, C.-H. Hsu, F.-C. Chuang, H. Lin, and A. Bansil, Sci. Rep. 6, 31317 (2016).
- [93] Z.-Q. Huang, W.-C. Chen, G. M. Macam, C. P. Crisostomo, S.-M. Huang, R.-B. Chen, M. A. Albao, D.-J. Jang, H. Lin, and F.-C. Chuang, Nanoscale Res. Lett. 13, 43 (2018).
- [94] Z.-Q. Huang, C.-H. Hsu, C. P. Crisostomo, G. Macam, J.-R. Su, H. Lin, A. Bansil, and F.-C. Chuang, Chin. J. Phys. 67, 246 (2020).
- [95] Z. Qiao, W. Ren, H. Chen, L. Bellaiche, Z. Zhang, A. H. MacDonald, and Q. Niu, Phys. Rev. Lett. 112, 116404 (2014).
- [96] J. B. Goodenough, Phys. Rev. 100, 564 (1955).
- [97] J. Kanamori, J. Phys. Chem. Solids 10, 87 (1959).
- [98] N. Rezaei, M. Alaei, and H. Akbarzadeh, Comput. Mater. Sci. 202, 110947 (2022).
- [99] M. Yu and D. R. Trinkle, J. Chem. Phys. 134, 064111 (2011).
- [100] G. Henkelman, A. Arnaldsson, and H. Jónsson, Comput. Mater. Sci. 36, 354 (2006).
- [101] E. Sanville, S. D. Kenny, R. Smith, and G. Henkelman, J. Comput. Chem. 28, 899 (2007).
- [102] W. Tang, E. Sanville, and G. Henkelman, J. Phys.: Condens. Matter 21, 084204 (2009).
- [103] P. Błoński, J. Tuček, Z. Sofer, V. Mazánek, M. Petr, M. Pumera, M. Otyepka, and R. Zbořil, J. Am. Chem. Soc. 139, 3171 (2017).
- [104] J. W. Chiou, S. C. Ray, S. I. Peng, C. H. Chuang, B. Y. Wang, H. M. Tsai, C. W. Pao, H.-J. Lin, Y. C. Shao, Y. F. Wang, S. C. Chen, W. F. Pong, Y. C. Yeh, C. W. Chen, L.-C. Chen, K.-H. Chen, M.-H. Tsai, A. Kumar, A. Ganguly, P. Papakonstantinou *et al.*, J. Phys. Chem. C **116**, 16251 (2012).
- [105] Y. Liu, N. Tang, X. Wan, Q. Feng, M. Li, Q. Xu, F. Liu, and Y. Du, Sci. Rep. 3, 2566 (2013).
- [106] Q. Zhang, Z. Ren, N. Wu, W. Wang, Y. Gao, Q. Zhang, J. Shi, L. Zhuang, X. Sun, and L. Fu, npj 2D Materials and Applications 2, 1 (2018).
- [107] J. C. Slater and G. F. Koster, Phys. Rev. 94, 1498 (1954).