

Two-dimensional magnetoelectric multiferroics in a MnSTe/In₂Se₃ heterobilayer with ferroelectrically controllable skyrmions

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The magnetoelectric effect and skyrmions are two fundamental phenomena in the field of condensed-matter physics. Here, using first-principles calculations and Monte Carlo simulations, we propose that strong magnetoelectric coupling can be demonstrated in a multiferroic heterobilayer consisting of two-dimensional (2D) MnSTe and α -In₂Se₃. As the electric polarization in ferroelectric α -In₂Se₃ is switched, the creation and annihilation of the topological magnetic phase can be achieved in this multiferroic heterobilayer, giving rise to the intriguing ferroelectrically controllable skyrmions. This feature is further revealed to be closely related to the physical quantity of $D^2/|KJ|$, which is generally applicable for describing the required conditions of such physics. Moreover, the evaluations of their topological magnetic phases with temperature are systematically discussed. These insights not only greatly enrich the research on 2D magnetoelectric multiferroics, but also pave a promising avenue to realize new skyrmionic device concepts.

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I. INTRODUCTION

Two-dimensional (2D) multiferroics are a special class of materials that hold simultaneously two or more primary ferroic (i.e., ferromagnetic, ferroelectric, and ferroelastic) orders [1–6]. Among numerous multiferroics, magnetoelectric multiferroics has attracted intensive attention because of its novel physics and potential applications in next-generation spintronics and memory devices [5–12]. The current research on magnetoelectric multiferroics in 2D lattice has been mainly established in the paradigm of ferroelectrically manipulation of magnetic ground state [9,13–15], magnetization orientation [11,16,17], or magnetic moment distribution [5,18,19], leaving other magnetic behaviors less explored in this field [20–22]. Ideal 2D magnetoelectric multiferroics requires ferromagnetic and ferroelectric orders to be strongly coupled [23]. Until now, however, reported 2D magnetoelectric multiferroics are rather scarce [11–19], most with coexisting yet loosely coupled ferromagnetic and ferroelectric orders [12–19] owing to the inherent exclusion between ferroelectricity and magnetism. 2D multiferroics with strong magnetoelectric coupling, especially on novel magnetic properties, remains highly desired from both scientific and technological impact.

Compared with conventional magnetic properties, magnetic skyrmions are a newly emerging topological magnetic order, which, however, spur rapid development in both fundamental research and device applications [24–29]. The unique fingerprint of magnetic skyrmions can be characterized by the spin swirling vortexlike texture, which is protected by a real-space topological number [30,31]. Magnetic skyrmions have been proposed experimentally in B20 compounds, such as

MnSi [27] and FeGe [32]. Recently, with the advance of long-range magnetism in 2D CrI₃ [33], Cr₂Ge₂Te [34], and VSe₂ [35,36], impressive progress has been made in realizing magnetic skyrmions in 2D lattice. Typical examples include 2D Janus TMDs [37–39], CrGe(Se, Te)₃ [40], Cr(*I*, *X*)₃ (*X* = Cl, Br) [41], CrN [42], and so on [43–45]. To make skyrmionic devices, the essential step is to control skyrmions [46–49]. With respect to the dynamic strategies [50], undoubtedly, the possible coupling between ferroelectrics and magnetic skyrmions, which adds skyrmions into magnetoelectric effect, could provide an unprecedented opportunity to control skyrmions in a robust way and enable novel physical phenomena. Through highly valuable, so far, 2D magnetoelectric multiferroics with ferroelectrically controllable skyrmions has been rarely explored [20,21].

In this work, we design a multiferroic heterobilayer consisting of magnetic MnSTe and ferroelectric α -In₂Se₃. Using first-principles calculations and Monte Carlo simulations, we find that significant magnetoelectric effect can be realized in this multiferroic heterobilayer. Upon reversing the electric polarization in ferroelectric α -In₂Se₃, the topological magnetic phase can be switched on and off, which leads to the exotic ferroelectrically controlled creation and annihilation of magnetic skyrmions. We further unveil that such phenomenon closely correlates to the physical quantity of $D^2/|KJ|$, and this quantity can be employed to estimate the required conditions of such physics. Finally, we discuss the effect of temperature on the skyrmion number and topological magnetic phases of such multiferroic heterobilayer. The explored phenomena and mechanism are useful for fundamental research in 2D magnetoelectric multiferroics and skyrmionics.

II. COMPUTATIONAL METHOD

Our first-principles calculations are performed based on density functional theory as implemented in the Vienna *ab*

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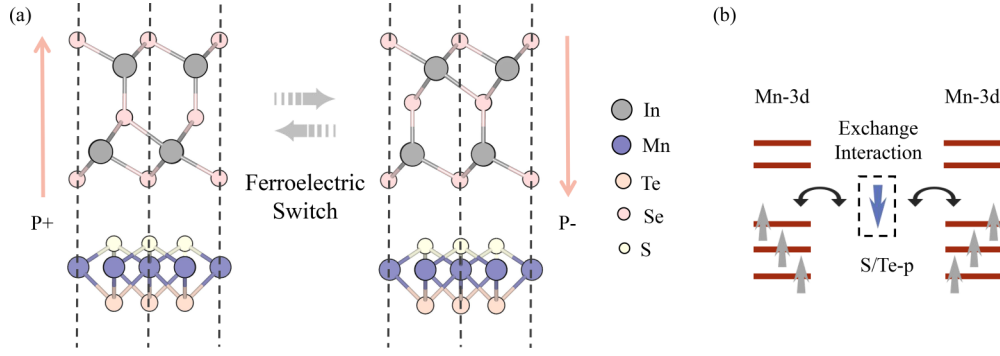


FIG. 1. (a) Crystal structures of MnSTe/In₂Se₃ heterobilayer under $P+$ and $P-$ configurations. (b) Schematic diagram of d -orbital splitting under distorted octahedral coordination environment and the superexchange coupling between $3d$ electrons of the Mn atoms in the heterobilayer.

initio simulation package [51]. The projector augmented wave method [52] is adopted to treat the ionic potential. The Perdew-Burke-Ernzerhof functional of generalized gradient approximation [53] is used for the exchange-correction interactions. Following the previous works [38,54,55], we employ effective Hubbard [56] $U = 2$ eV to describe the strongly correlated electrons in the $3d$ orbital of the Mn atom. The convergence tolerances for the residual force and energy are set to 0.001 eV/Å and 10^{-6} eV, respectively. The plane-wave cutoff energy is set to 520 eV. The Monkhorst-Pack k -point mesh [57] of $13 \times 13 \times 1$ is adopted to sample the Brillion zone. The DFT-D3 method [58] is employed for treating van der Waals interaction in all calculations. The ferroelectric transition pathway and energy barrier are calculated on the basis of the nudged elastic band (NEB) method [59].

Using the magnetic parameters obtained from first-principles calculations, we employ parallel tempering Monte Carlo (MC) simulations [60] with the Metropolis algorithm to obtain the energy minimum spin textures. The spin textures are obtained based on a $150 \times 150 \times 1$ supercell with 100 000 MC steps performed at each temperature (from 660 K cooled down to the investigated low temperature).

III. RESULTS AND DISCUSSION

The heterobilayer that we designed consists of a monolayer MnSTe and a monolayer α -In₂Se₃. While monolayer MnSTe harbors topological magnetic order [37,38], monolayer α -In₂Se₃ is known for its out-of-plane ferroelectric polarization [61]. The lattice constants of MnSTe and α -In₂Se₃ are optimized to be 3.55 and 4.02 Å, respectively, which agree well with the previous studies [38,61]. For the heterobilayer, the 2×2 supercell of MnSTe is used to match the $\sqrt{3} \times \sqrt{3}$ supercell of α -In₂Se₃, which results in a rather small lattice mismatch of 1.9%. The contacted surface of MnSTe with α -In₂Se₃ could be either a S- or Te-terminated surface, and they share roughly similar results. In the following, we only focus on the S-terminated case. For more information on the Te-terminated case, please see Supplemental Material [62]. Given the switchable polarization of α -In₂Se₃, the heterobilayer naturally displays two types: the first one with polarization pointing away from the interface ($P+$), while the second one with polarization pointing to the interface ($P-$).

For each type, different stacking patterns are considered, and the energy minimum patterns are shown in Fig. 1(a).

The interlayer distances for $P+$ and $P-$ phases of the heterobilayer are found to be 0.106 and 0.108 Å, respectively, and the corresponding electric polarizations are calculated to be 3.263×10^{-12} and 3.092×10^{-11} C/m². Obviously, they display different absolute values for the electric polarizations. Nevertheless, as we will show later, $P+$ and $P-$ phases can be considered as two ferroelectric states of the heterobilayer and the ferroelectric transition between them is feasible. The valence electron configuration of Mn atom is $3d^5 4s^2$. By coordinating with three S and three Te atoms, four electrons are transferred to the neighboring atoms, leading to the electron configuration of $3d^3 4s^0$. Under the distorted octahedral coordination environment, the d orbitals roughly split into two groups, i.e., the high-lying e_g and low-lying t_{2g} orbitals. As displayed in Fig. 1(b), the three left electrons of the Mn atom would half fill the t_{2g} orbitals, giving rise to a magnetic moment of $3 \mu_B$. As expected, our spin-polarized calculations show that $P+$ and $P-$ phases are spin polarized and the magnetic moments are mainly distributed on the Mn atoms. The magnetic moment on the Mn atom is calculated to be $3.51 \mu_B$ in MnSTe monolayer and $3.48 \mu_B$ in heterostructures, and the delocalized magnetic moments result from the metallic nature [37].

To investigate the magnetic interactions of the $P+$ and $P-$ phases, we adopt the following spin Hamiltonian:

$$H = - \sum_{\langle i,j \rangle} \mathbf{D}_{ij} \cdot (\mathbf{S}_i \times \mathbf{S}_j) - J \sum_{\langle i,j \rangle} (\mathbf{S}_i \cdot \mathbf{S}_j) - \lambda \sum_{\langle i,j \rangle} (S_i^z S_j^z) - K \sum_i (S_i^z)^2 - mB \sum_i S_i^z. \quad (1)$$

Here, \mathbf{S}_i and \mathbf{S}_j are the unit vectors representing local spins of the i th and j th Mn atoms, respectively. S_i^z and S_j^z indicate the projection of the vector \mathbf{S}_i and \mathbf{S}_j in z direction, respectively. \mathbf{D}_{ij} is the Dzyaloshinskii-Moriya interaction (DMI), J is the isotropic exchange coupling, λ is anisotropic exchange coupling, and K is the single ion anisotropy. The last term is the Zeeman energy. Based on Moriya's rule [63], because there is a mirror plane passing through the middle of the bond between two neighboring Mn atoms, \mathbf{D}_{ij} for the nearest-neighboring Mn atoms is perpendicular to their bond. As the out-of-plane component of \mathbf{D}_{ij} is negligible, we only consider

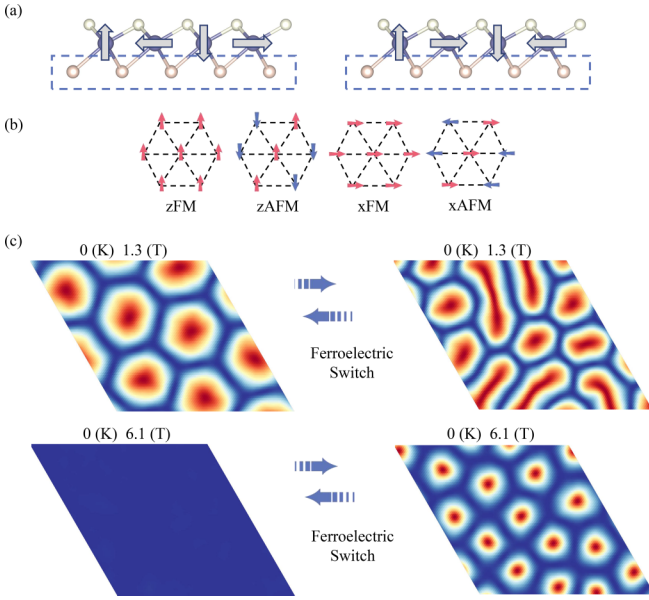


FIG. 2. (a) Illustrations of right- and left-hand spin-spiral configurations adopted to obtain D_{ij} . (b) Illustrations of spin configurations used to estimate J , K , and λ . (c) Spin textures for $P+$ (left panel) and $P-$ (right panel) phases under external magnetic field of 1.3 T (upper panel) and 6.1 T (bottom panel) at 0 K.

its in-plane component. To calculate the DMI parameter, a 4×1 supercell is constructed, and two different noncollinear magnetic orders are set [i.e., left-hand and right-hand orders; see Fig. 2(a)]. The calculated D_{ij} are listed in Table I. For free-standing MnSTe, D_{ij} is calculated to be 1.875 meV. Such a large value can be attributed to the combined effect of the large electronegativity difference between S and Te atoms and strong spin-orbital coupling (SOC) strength within the Te atom. By forming the heterobilayers, due to the interfacing effect, D_{ij} is further enhanced to 2.181 and 2.502 meV for $P+$ and $P-$ phases, respectively. The significantly difference in D_{ij} for $P+$ and $P-$ can be attributed to their distinct discrepancy in electric polarization.

To get J , K , and λ , four magnetic configurations illustrated in Fig. 2(b) are considered, and the corresponding results are listed in Table I. The isotropic exchange interaction J is calculated to be 9.369 and 8.086 meV, respectively, for $P+$ and $P-$ phases, suggesting ferromagnetic (FM) exchange interaction among the magnetic moments. Such FM coupling can be explained by the Stoner model. According to the Stoner model, the itinerant ferromagnetism occurs if $I D(E_F) > 1$ is satisfied [64,65]. The Stoner parameter I indicates the strength of exchange interaction. We evaluate the average exchange splitting $\langle \epsilon_k \rangle$ as the average differences between two bands around

TABLE I. Magnetic parameters of monolayer MnSTe, $P+$ phase, and $P-$ phase.

| | D (meV) | J (meV) | λ (meV) | K (meV) | D/J | $D^2/ KJ $ |
|-------|-----------|-----------|-----------------|-----------|-------|------------|
| MnSTe | 1.875 | 10.228 | -0.034 | 0.401 | 0.183 | 0.857 |
| $P+$ | 2.181 | 9.369 | -0.238 | 0.393 | 0.233 | 1.292 |
| $P-$ | 2.520 | 8.086 | -0.320 | 0.415 | 0.312 | 1.892 |

the Fermi level in every k point and obtain the value of I from $\langle \epsilon_k \rangle = Im$ (m indicates the magnetic moment of the Mn atom). $D(E_F)$ is the non-spin-polarized density of state at Fermi level. We obtain $I = 0.734/0.645$ and $D(E_F) = 10.810/4.937$ for the $P+$ / $P-$ phase, satisfying the criterion. Therefore, these two structures exhibit ferromagnetic nature. Also, similar to the case of DMI, the different J values in $P+$ and $P-$ phases correlate to their opposite electric polarizations. It should be that as the main physics arises from the interlayer coupling in the van der Waals heterobilayer, it can also be attributed to a ferroelectric proximity effect from In_2Se_3 (see Fig. S1 of the Supplemental Material [62]).

Different from the cases of DMI and isotropic exchange interaction, as listed in Table I, the anisotropic exchange coupling λ is significantly enhanced in the heterobilayers with respect to that of monolayer MnSTe. The negative values of λ indicate that the Heisenberg exchange interaction favors the in-plane magnetization for $P+$ and $P-$ phases. However, as compared with the Heisenberg exchange interaction, the single ion anisotropy K is obviously larger. As listed in Table I, the single ion anisotropy exhibits positive values, which suggests out-of-plane magnetization. The competition between them leads to the out-of-plane magnetization orientation to be more favorable for both systems.

By featuring different magnetic parameters, different spin textures are expected in the $P+$ and $P-$ phases, indicating the possibility for realizing ferroelectrically controllable magnetic behaviors. To verify this scenario, parallel tempering MC simulations [60] are performed to explore the spin textures in the heterobilayers, which is based on the first-principles calculations parametrized Hamiltonian in Eq. (1). The spin textures for $P+$ and $P-$ phases in the absence of an external magnetic field are presented in Fig. S2 [62], which displays ferromagnetic states with a labyrinthlike pattern, called labyrinth domains. As shown in Table I and Fig. S2 [62], the magnetic parameters, as well as the magnetic spin textures, of $P+$ and $P-$ phases without magnetic field are different. This firmly confirm that, in the absence of magnetic field, the magnetic properties of MnSTe/ In_2Se_3 can be controlled by ferroelectricity as well, giving rise to magnetoelectric multiferroics. The labyrinth domains can be broken when applying external magnetic field. Remarkably, we find that under an external magnetic field of 1.3 T, the labyrinth domain is transformed into the intriguing skyrmion lattice state in the $P+$ phase [see Fig. 2(c)] and the radius of skyrmion is ~ 15 nm. In contrast to the $P+$ phase, although some isolated skyrmions appear in the $P-$ phase under the magnetic field of 1.3 T, the fragmented labyrinth domains are observed simultaneously, as shown in Fig. 2(c). Such mixed phase of skyrmion and fragmented labyrinth domains in the $P-$ phase renders the isolated skyrmions in it technologically useless. As $P+$ and $P-$ phases can be considered as two ferroelectric states of the heterobilayer, the skyrmion lattice phase can be switched on and off by ferroelectricity, thus realizing the strong magnetoelectric effect of ferroelectrically controllable skyrmionics.

Moreover, by applying an external magnetic field of 6.1 T, as shown in Fig. 2(c), neither a fragmented labyrinth domain nor skyrmion lattice is observed in $P+$ phase, giving rise to a trivial ferromagnetism state, while for the $P-$ phase

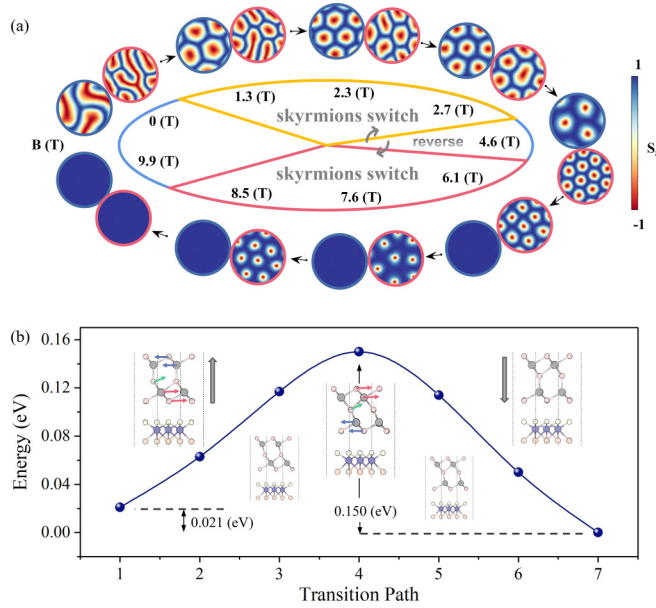


FIG. 3. (a) Spin textures for $P+$ (with blue edge) and $P-$ (with red edge) phases under various external magnetic fields at 0 K. (b) Ferroelectric transition pathway and energy barrier between $P+$ and $P-$ phases of the heterobilayer.

under an external magnetic field of 6.1 T, interestingly, the labyrinth pattern transforms into skyrmion lattice state, and the radius of the skyrmion is found to be ~ 8 nm; see Fig. 2(c). Apparently, similar to the case of 1.3 T, the skyrmion phase can be switched on and off by reversing the electric polarization, leading to the ferroelectrically controllable skyrmionics. However, the “switch-on” and “switch-off” states are reversed with respect to that of 1.3 T.

To explore the robustness of such magnetoelectric effect in terms of external magnetic field, we further investigate their spin textures under various magnetic fields. The corresponding results are presented in Fig. 3(a). Interestingly, under the external magnetic fields of 1.3–2.7 T and 6.1–8.5 T, the magnetoelectric effect of ferroelectrically controllable skyrmionics in this heterobilayer can be preserved, suggesting the phenomenon is rather robust. For the cases of 1.3–2.7 T, with increasing the external magnetic field, the density of skyrmions in the $P+$ phase is roughly maintained and the size of skyrmion is reduced, while the fragmented labyrinth domains in the $P-$ phase become more and more isolated. For the cases of 6.1–8.5 T, with the increasing external magnetic field, while the $P+$ phase preserves a trivial ferromagnetic state, the density of skyrmions in the $P-$ phase decreases and the size of skyrmion is roughly maintained. Under the external magnetic field of 2.7–6.1 T, as shown in Fig. 3(a), the skyrmion lattice state is achieved for both the $P+$ and $P-$ phases, excluding the magnetoelectric effect of ferroelectrically controllable skyrmionics in this heterobilayer. And when increasing the magnetic field larger than 8.5 T, both phases favor the trivial ferromagnetic state, also prohibiting the realization of such magnetoelectric effect.

Having established the magnetoelectric effect of ferroelectrically controllable skyrmionics, we then investigate the feasibility of ferroelectricity in this heterobilayer. To this end,

we employ the nudged elastic band (NEB) method to study the ferroelectric switching process. The $P+$ phase is lower in energy than the $P-$ phase by 21 meV per f.u. of In_2Se_3 . There are two paths for the transition between the $P+$ and $P-$ phases; see Fig. S3 [62]. The minimum-energy path is found to be path II. As shown in Fig. 3(c), the transition between the $P+$ phase and paraelectric state along path II is realized by the left shifting (right shifting) of the upper (lower) In-Se layer and the upper-inclined shifting of the inner Se layer. And the transition between the paraelectric state and the $P-$ phases is caused by the right shifting (left shifting) of the upper (lower) In-Se layer and the upper-inclined shifting of the inner Se layer. The energy barriers for the ferroelectric switching from the $P+$ ($P-$) to $P-$ ($P+$) phases along path II are estimated to be 129 (150) meV per f.u. of In_2Se_3 , and it is 2.93 (2.95) eV for path I. These values in path II are significantly smaller than those of VOCl_2 (180 meV/f.u.) [66], Sc_2CO_2 (520 meV/f.u.) [67], and vacancy-doped CrI_3 (650 meV/f.u.) [68], but larger than those of NaOH (75 meV/f.u.) [69] and CuCrP_2S_6 (100 meV/f.u.) [70], indicating the feasibility of the ferroelectricity and thus the strong magnetoelectric effect in this heterobilayer. The ferroelectric transition barrier for In_2Se_3 (CuInP_2S_6) monolayer is estimated to be 0.066 (0.43) eV in theory [61,71], and the corresponding electric field needed in experiment is 200 (700) kV/cm [72,73]. Accordingly, the electric field required for flipping the polarization of α - In_2Se_3 on MnSTe is estimated to be between 200 and 700 kV/cm.

To get more physics into the magnetoelectric effect in this heterobilayer, we discuss the relationship between topological spin texture and magnetic parameters of D , J , and K . Among these three parameters, J and K are inclined to force the magnetic moments to be arranged parallelly, while D is prone to induce noncolinear alignments of magnetic moments. Based on this fact, we find that $D^2/|KJ|$ can describe the magnetoelectric effect in this heterobilayer. In detail, $D^2/|KJ|$ can be used to qualitatively estimate the required external magnetic field for realizing the skyrmion state. Taking $D^2/|KJ|$ of 1.892 for the $P-$ phase as reference, we set $D^2/|KJ|$ to be 1.135, 1.514, 2.270, and 2.649, respectively, corresponding to 0.6, 0.8, 1.2, and 1.4 times 1.892. The corresponding spin textures and phase diagram for them under external magnetic field are shown in Figs. 4(a) and 4(b). For all these cases, the labyrinth domain state can be transformed into the skyrmion lattice state and then to the trivial ferromagnetic state by applying external magnetic field. With decreasing the value of $D^2/|KJ|$, the critical magnetic fields required for such transitions decreases. As $P+$ and $P-$ phases exhibit different $D^2/|KJ|$, by applying an appropriate external magnetic field, the magnetoelectric effect of ferroelectrically controllable skyrmionics can be realized.

At last, we take $P+$ and $P-$ phases under the external magnetic field of 1.3 and 6.1 T, respectively, as examples to discuss the temperature effect on the skyrmion lattice state. To quantitatively characterize such effect, the topological index is employed, which is defined as [74]

$$Q = \frac{1}{4\pi} \int \mathbf{m} \cdot \left(\frac{\partial \mathbf{m}}{\partial x} \times \frac{\partial \mathbf{m}}{\partial y} \right) dx dy.$$

Here, m is the normalized magnetization. Q describes the number of unit spheres that the magnetic moments can be

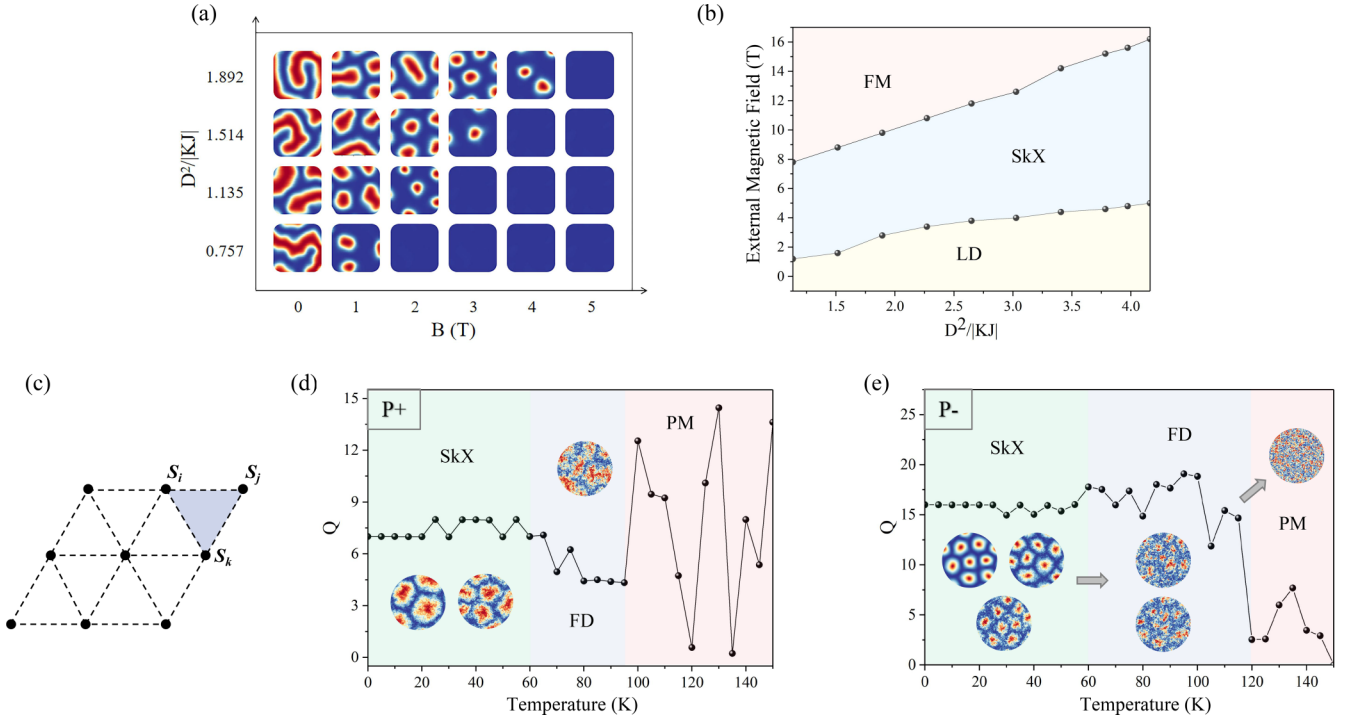


FIG. 4. (a) Spin textures and (b) phase diagram as functions of external magnetic field and $D^2/|KJ|$ at 0 K. LD in (b) indicates labyrinth domain state. (c) Equilateral triangle lattice employed for calculating skyrmion number Q . The variation of Q as a function of temperature and the corresponding phase diagrams for (d) $P+$ phase at 1.3 T and (e) $P-$ phase at 6.1 T. SkX, FD, and PM in (b), (d), and (e) indicate skyrmion crystal, fluctuation disorder, and paramagnetic states, respectively.

wrapped around. $Q = \pm 1$ refers to one skyrmion. This formula is in continuous integral form, which is not convenient for discrete lattice. According to the method proposed by Berg and Luscher *et al.* [74], we partition the lattice into many unit right triangles [75], as shown in Fig. 4(c). All unit triangles are employed to calculate the Q as follows:

$$Q = \frac{1}{4\pi} \sum_l qn$$

$$\tan \frac{qn}{2} = \frac{\mathbf{S}_i^n \cdot (\mathbf{S}_j^n \times \mathbf{S}_k^n)}{1 + \mathbf{S}_i^n \cdot \mathbf{S}_j^n + \mathbf{S}_j^n \cdot \mathbf{S}_k^n + \mathbf{S}_k^n \cdot \mathbf{S}_i^n}.$$

Here, \mathbf{S}_i^n , \mathbf{S}_j^n , \mathbf{S}_k^n are the magnetic moments of the three atoms at the vertices of n th equilateral triangle in the counterclockwise lattice.

The variation of Q as a function of temperature and the corresponding phase diagrams are displayed in Figs. 4(d) and 4(e). For the $P+$ phase, as shown in Fig. 4(d), the skyrmion lattice state can be preserved with temperature reaching up to 60 K. In the skyrmion lattice state, Q is almost unchanged with increasing temperature, arising from the edge of the skyrmion becoming vague. When further increasing the temperature, the skyrmion lattice state is transformed into the fluctuation disorder (FD) state and finally paramagnetic (PM) state. It should be noted that in the FD and PM states, Q oscillates with increasing temperature. This phenomenon can be attributed to the competition between the following two factors. First, since Q is essentially the product of polarity and vorticity, the breaking of skyrmions caused by temperature increases the number of little vortices in the spin textures, leading to the increase of Q . Second, the temperature-induced vanishing of the vortex

would decrease Q . The roughly similar scenario is shared by the phase of $P-$; see Fig. 4(e). Figures S4–S6 present the spin textures of $P+$ and $P-$ phases under different magnetic fields at 20, 40, and 60 K [62]. Obviously, the magnetic skyrmions can be preserved with temperatures reaching up to 60 K. Also, the creation and annihilation of skyrmions can be realized at 20, 40, and 60 K.

Besides magnetic-field-required skyrmion lattice in MnSTe/In₂Se₃ bilayer, it is worth emphasizing that the metastable isolated skyrmions are found for the $P+$ phase without external magnetic field; see Fig. S7 [62]. The nontrivial topology of these skyrmions is confirmed by the calculated skyrmion number $Q = 2$. This state is 0.014 meV/f.u. higher in energy than the labyrinth domain state. Such state does not appear in the $P-$ phase without magnetic field. Therefore, reversing the ferroelectric polarization could control the creation and annihilation of metastable skyrmions without external magnetic field in MnSTe/In₂Se₃ heterobilayer.

IV. CONCLUSION

In conclusion, using first-principles calculations and Monte Carlo simulations, we report the discovery of a multiferroic heterobilayer with strong magnetoelectric effect. We reveal that with switching the electric polarization in ferroelectric α -In₂Se₃, the creation and annihilation of skyrmion lattice state can be realized, giving rise to the exotic ferroelectrically controllable skyrmionics. Such phenomenon is revealed to be closely related to the physical quantity of $D^2/|KJ|$, and this quantity can describe the required conditions for such physics. In addition, the evolution of Q with temperature in

the heterobilayers and the corresponding phase diagram are investigated. These results shed light on the research of 2D magnetoelectric multiferroics and skyrmionics.

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