Three-gap superconductivity in two-dimensional InB₂/InB₄ films

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In recent years, multigap superconductors have attracted much attention since the discovery of novel two-gap superconductivity with transition temperature $T_c \sim 39$ K in bulk MgB₂. Based on the first-principles calculation and anisotropic Migdal-Eliashberg theory, we conduct a study on a series of two-dimensional (2D) boron-based materials with doped metal atoms to search for multigap superconductors. We find that InB₂ monolayer films and InB₄ trilayer films are dynamically stable but not synthesized experimentally yet. An evident three-gap superconductor with high $T_c \sim 41.5$ K is obtained in a InB₂ monolayer film. Similarly, InB₄ trilayer film is a novel superconductor with three distinct superconducting gaps and a high $T_c \sim 53$ K. The superconductivity in both 2D films originates mainly from the covalent-state-driven metallization. In addition, the effect of biaxial strain on the superconducting behavior of InB₄ trilayer films is also involved. The InB₄ trilayer films stay dynamically stable under biaxial tensile strain of -3%-6%, and the highest T_c boosts to 64 K under the biaxial tensile strain of $\sim 4\%$. Meanwhile, we also find that the GeB₄ and ZnB₄ trilayer films are two-gap superconductors with high $T_c \sim 48.5$ and 32 K, respectively.

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I. INTRODUCTION

Since the discovery of two-gap superconductivity ($T_c \sim$ 39 K) in bulk MgB₂ [1], plenty of studies have been conducted to explore the origin of these gaps and their effects [1–8]. For bulk MgB₂, the electronic states formed by boron orbital planes strongly couple with corresponding phonon modes and B- σ bands contribute strongly to the density of states at Fermi level, which is favorable for electron-phonon-coupling (EPC) formation. This reveals the origin of high T_c and multiple gaps in bulk MgB₂ [4,9]. The Fermi surface (FS) of MgB₂ is formed by bands with completely different electron orbital projections. The in-plane $\sigma(s + p_{x,y})$ -bonding state, together with out-of-plane $\pi(p_z)$ lead to the separation of superconducting gaps; the σ gap $\Delta_{\sigma} \sim 7$ meV is obtained by the former and the weaker π gap $\Delta_{\pi} \sim 2-3$ meV is obtained by the latter.

The multigap superconductors have many novel physical effects such as anomalous vortical and skyrmionic states [10,11], giant-paramagnetic response [12], hidden criticality [13,14], time-reversal symmetry breaking [8,15], etc. Many efforts are taken to search for robust multigap superconductors (especially beyond two-gap) to confirm these predictions, yet only a few candidates were discovered, such as CaC₆, hole-doped graphane C₂H₂ [16], CaBeSi crystal (which is isostructural and isoelectronic to MgB₂) [17], and two molecular hydrogen structures under extremely high pressure [18]. However, the superconducting gap separation is indistinguishable, and the T_c is extremely low (~0.4 K) [17].

Similar to bulk MgB₂, monolayer MgB₂ is a three-gap superconductor with a critical temperature of 20 K through theoretical predictions, which can be increased to 50 K under biaxial strain of $\sim 4\%$ [19]. Compared with the two-gap superconductor bulk MgB₂, the surface states of monolayer MgB₂ are comparable in electronic density to the bulklike σ and π bands. The three-gap superconductivity nature of monolayer MgB_2 originates from the emergent surface contribution [19]. Two AA-stacked honeycomb boron planes with a triangular Mg metal layer as an interlayer can form MgB₄ trilayer film, which is predicted to have a novel four-gap superconductivity with high $T_c \sim 52$ K [20]. The separation between the EPC strength distributions on the σ_1 and σ_2 sheets leads to the evident four-region distribution characteristics, which are responsible for the high- T_c four-gap superconductivity [20]. Atomic doping and substitution are used to search for new superconductors. It is found that defects in the boron sublattice (nonstoichiometric $MgB_{2-\nu}$) are not favorable for superconductivity [21]. However, hole doping of MgB₂ or isoelectronic substitution of Mg is a feasible way to find other boron-based superconductors. It is shown that AlB₂ monolayers reveal a two-gap superconducting nature with a high $T_c \sim 26.5$ K. When increasing the number of boron layers (AlB_{2+x}) , the surface boron atom layers on both sides enhance the superconductivity; the T_c of B₂-Al-B₂-type AlB₄ boosts to 47 K with robust three-gap superconducting behavior [22].

Although several multigap superconductors such as MgB_4 [20] and AlB_{2+x} [22] have been discovered, many stable structures composed of boron atom layers and other metal atom layers (especially transition metals) have not been reported yet. A novel multigap superconducting nature may exist in

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these structures. In this paper, we conduct a first-principles study and use the fully anisotropic Migdal-Eliashberg theory to investigate the pairing mechanism and multigap superconducting nature in a series of boron-based two-dimensional (2D) materials, in particular, the sandwich-type trilayer films. The most important discovery is that the stoichiometric InB₂ monolayer films and InB₄ trilayer films have three separated superconducting gaps with high $T_c \sim 41.5$ K and $T_c \sim 53$ K, respectively. Moreover, the T_c of InB₄ trilayer films can be increased to ~ 64 K under biaxial tensile strain of 4%. We also find the two-gap superconductor ZnB₄ with $T_c \sim 48.5$ K.

II. METHODOLOGY

We performed first-principles calculations based on density-functional theory [23,24] with local density approximation. The subsequent lattice vibration properties are based on density-functional perturbation theory calculations [25,26], as coded in the QUANTUM ESPRESSO package [27]. We use plane waves with a kinetic energy cutoff of 80 Ry to simulate the valence electrons, and norm-conserving pseudopotentials to simulate the ion cores. A larger vacuum layer of 18 Å is used to remove the interlayer interaction. The kinetic energy cutoff (Ry) for charge density is set to 220 Ry. All the 2D film structures are fully optimized until the Hellman-Feynman force on each atom is under 10^{-6} Ry/bohr. A 40 × 40×1 k-point mesh with a Methfessel-Paxton smearing [28] of 0.02 Ry is set to simulate the Brillouin zone (BZ) integral in the wave-vector space, and a q-point mesh of $10 \times 10 \times 1$ is set in the subsequent lattice dynamics calculations.

The superconducting properties of these films are investigated through the EPW code [29-31]. We solve fully anisotropic Migdal-Eliashberg equations through an imaginary axis at the fermion Matsubara frequencies $\omega_i = (2j + i)$ $1\pi T$ (*j* is an integer) for a series of temperatures with the same interval. The electronic-phonon interpolation technique applied in the EPW code, which is based on the maximum localized Wannier function, is an efficient and accurate method to calculate the superconducting gaps and T_c . The precedent computations of the electronic wave functions required for the Wannier interpolations are carried out within a uniform unshifted BZ k-mesh of $20 \times 20 \times 1$. An interpolated k-point grid of $300 \times 300 \times 1$ and a q grid of $150 \times 150 \times 1$ are set to solve the anisotropic Migdal-Eliashberg equations. The fermion Matsubara frequencies cutoff is set to be 1.0 eV. The Morel-Anderson pseudopotential μ_c^* is a semiempirical parameter to solve the numerical solution of the Eliashberg equation, defined as $\mu_c^* = \mu^* / [1 + \mu^* \ln(\omega_{el} / \omega_{ph})]$, where ω_{el} and ω_{ph} are the characteristic electron and phonon energy, respectively, and μ^* is the Coulomb potential [7]. A reasonable value is in the range of 0.1–0.2 [7]. In this work, we set $\mu_c^* = 0.13$, the same as that used in the previous materials [19,20,22].

III. RESULTS AND DISCUSSION

In this work, we construct a series of boron-based films through the replacement of the Mg atoms by metals in the fourth and fifth periods of the periodic table in the monolayer MgB_2 films and MgB_4 trilayer films. We first investigate the phonon spectrum of these films to verify the dynamic stability. We find that only the InB₄, ZnB₄, AgB₄, and GeB₄ trilayer films and the InB2 monolayer films are dynamically stable. So far, these films have not been reported experimentally. We mainly focus on InB2 and InB4 to reveal the formation mechanism of the three-gap superconductivity and the regulation of the superconductivity under biaxial strain. In general, the superconductivity of the trilayer films is better than that of the monolayer film [20]. For example, the T_c of theMgB₄ trilayer film is higher than monolayer film MgB₂. Interestingly, The InB₄ trilayer film has a higher $T_c \sim 53$ K than that of monolayer InB₂ ($T_c \sim 42$ K), while monolayer InB₂ shows a three-gap superconducting nature with a slight split in the strongest superconducting gap, which shows more interesting multigap superconductivity.

Figure 1 shows the crystal structures, FSs, corresponding band structures with orbital projections, and the electronic density of states (EDOS) of monolayer InB₂ and trilayer InB₄. As is shown in Fig. 1(a), the monolayer InB₂ film consists of a triangular indium lattice and a honeycomb boron layer. There are four bands crossing the FS; the hexagonal and petalshaped Fermi sheets surrounding the Γ point are contributed mainly by B- σ electrons (marked as the σ_1 and σ_2 sheets, respectively). This gives rise to two hole pockets surrounding the BZ center and causes a covalent-state-driven metallization, similar to the case in bulk MgB₂ [32-35], boron-doped diamond [36-38], and 2D boron sheets [38-41]. The circle Fermi sheet near the BZ boundary represents the band with B- π states (marked as π sheet); the outermost petal-shaped Fermi sheet at the edge of the BZ boundary represents the band with the hybridized states of the B- π and In- σ orbitals (marked as μ sheet). This is verified by the energy band with orbital projections in Fig. 1(c). The coexistence of σ_1 , σ_2 , μ , and π sheets indicates that the InB₄ trilayer film may show novel properties of multigap superconductivity. Compared with monolayer InB₂, InB₄ trilayer film has an extra honeycomb boron plane, which has a B2-In-B2 sandwich structure. The FS of trilayer InB₄ is similar to that of monolayer InB₂. There are six bands crossing the FS. Due to the existence of two boron planes, $B-\sigma_1$ and $B-\sigma_2$ sheets form two approximately hexagonal and two petal-shaped Fermi sheets respectively, which forms two pairs of hole pockets near the BZ center.

We proceed to concentrate on the vibrational properties and EPC of the monolayer InB₂ and trilayer InB₄ films. The phonon spectra with the projection of EPC parameters λ_{av}^{ph} , together with the phonon density of states (PDOS), and total and mode-resolved corresponding Eliashberg spectral function $\alpha^2 F(\omega)$ with the cumulative EPC strength $\lambda(\omega) =$ $2\int_0^{\omega} \alpha^2 F(\omega')/\omega' d\omega'$ are shown in Fig. 2. We can see that the monolayer InB₂ and trilayer InB₄ exhibit similar vibration properties. The absence of imaginary frequency in the phonon spectra indicates that monolayer InB₂ monolayer film is dynamically stable. As for InB₄ trilayer film, there is a slight imaginary (~ 0.44 meV) frequency at the Γ point. To further verify whether trilayer InB₄ is dynamically stable, we recalculate the phonon spectrum and superconducting properties of InB₄ using another similar type of pseudopotential. We find that the slight imaginary frequency of InB₄ disappears and

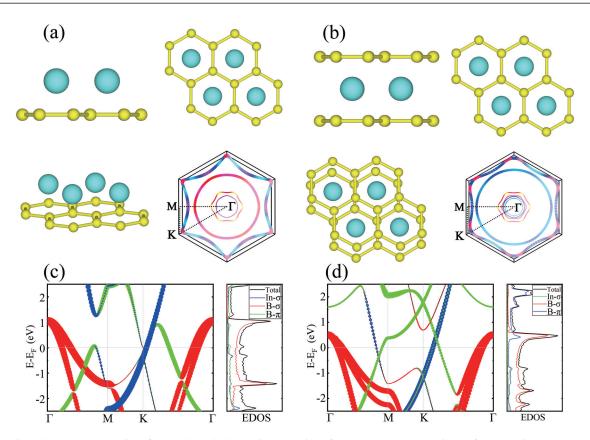


FIG. 1. Crystal structure, Fermi surface (FS), and electronic properties of InB_4 and InB_2 . The views of the atomic structure and FSs of (a) monolayer InB_2 and (b) trilayer InB_4 . Energy band structures with orbital projections: EDOS of monolayer (c) InB_2 and (d) trilayer InB_4 . The red, green, and blue bands represent the Fermi sheets formed by the B- σ states, pure B- π states, and the hybridized states of B- π , and $In-\sigma$ orbitals, respectively.

shows the same superconducting properties, as is shown in Supplemental Material Fig. S1 in Ref. [42]. This indicates that the InB_4 trilayer film is more likely to be a stable superconductor.

For InB₄ trilayer film, the partial PDOS demonstrates that the phonon modes with frequency under 23 meV are completely dominated by In atoms, while the remaining high frequency modes are mainly dominated by B atoms. Different from monolayer MgB₂ film [19], the stretch of in-plane, out-of-plane B-B atoms of InB₄, which are labeled E₂, B₁ modes (B_{1u} , B_{1g} , two degenerate E_{2u} , and two degenerate E_{2g} modes; see Fig. S2 in Ref. [42]), respectively, cause the optical vibration modes which are softened drastically due to the presence of Kohn anomalies [43] in the Γ point. The region of optical phonon softening in the BZ is highly consistent with the diameter of the hole Fermi sheets caused by the σ -bonding surface boron orbitals, which shows a typical feature of Kohn effect. In addition, the soft mode in InB₄ trilayer films is \sim 40 meV, which is comparable with that in other boron-doped diamonds [36,37]. Therefore, the E_2 and B_1 modes near the Γ point dominate the maximum phonon linewidth projection, as shown in Fig. 2(a). In addition, the in-plane modes of the internal B atoms originating from the coupling of $B-\pi$ orbitals and In-s orbitals also contribute to the projections linewidth, the frequencies of which range from 20 to 60 meV, although the contribution is relatively slight compared with the former. The mode-resolved $\alpha^2 F(\omega)$ and corresponding $\lambda(\omega)$ in Fig. 2(c) further verifies these results. Three low-frequency acoustic phonon branches ($\nu = 1-3$) contribute only 11% of the total EPC coupling strength, while the six high-frequency optical phonon branches ($\nu = 10-15$) contribute 43% and the remaining modes ($\nu = 4-9$) contribute 46%, which is dominant in EPC coupling strength. It is worth noting that contributions of the phonon modes above 23 meV to the EPC strength are consistent with the vibration PDOS distribution. Under the common influence of these phonon modes, the total isotropic EPC parameter $\lambda_{iso}^{ph} = 2 \int_0^\infty \alpha^2 F(\omega)/\omega \, d\omega$ reaches the value of 0.97 in the InB₄ trilayer film.

The phonon vibration and EPC properties of monolayer InB₂ are almost consistent with that of trilayer InB₄. The B (In) atoms dominate the high- (low-) frequency phonon modes. The E₂ soft modes (two degenerate E₂ modes; see Fig. S3 in Ref. [42]) lead to strong electron-phonon coupling near the Γ point. As shown in Fig. 2(f), the acoustic phonon branches ($\nu = 1-3$) contribute 48% of the total EPC strength, while the high-frequency optical phonon branches ($\nu = 7-9$) and the remaining modes ($\nu = 4-6$) contribute 30% and 22%, respectively, which follows the distribution of PDOS. The calculated EPC parameter $\lambda_{iso}^{ph} = 0.96$ is slightly lower than that of trilayer InB₄.

By solving the fully anisotropic Migdal-Eliashberg equation for self-consistent solution with the Coulomb

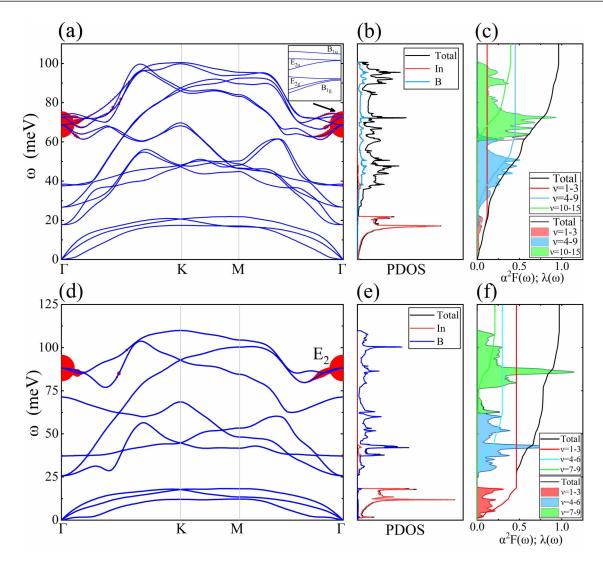


FIG. 2. Lattice dynamics and phonon EPC characteristics of trilayer InB₄ and monolayer InB₂. Phonon spectra with a projection of EPC parameters $\lambda_{q\nu}^{\text{ph}}$ (red), phonon density of states (PDOS), and total and mode-resolved isotropic Eliashberg function $\alpha^2 F(\omega)$ with the corresponding EPC strength $\lambda(\omega)$ of (a)–(c) trilayer InB₄ and (d)–(g) monolayer InB₂.

pseudopotential $\mu_c^* = 0.13$, we can analyze the superconducting gaps based on the normalized superconducting density of states (SDOS), momentum-resolution gaps Δ_{nk} on the FS, and the energy distribution of the gap $\Delta_{nk}(\omega = 0)$ versus temperature *T*. The normalized SDOS is obtained from the isotropic gap $\Delta(\omega)$ on the real axis via [7,29] $\frac{N_S(\omega)}{N_F} = \text{Re}[\frac{\omega}{\sqrt{\omega^2 - \Delta^2(\omega)}}]$; the $N_S(\omega)$ and N_F in the formula represent the electron DOS of the superconducting state and the normal DOS at the Fermi level. The calculated energy distribution of the gap $\Delta_{nk}(\omega = 0)$ versus temperature *T*, SDOS(*T* = 5 K), electronic EPC parameters λ_{nk}^{el} , and momentum-resolved superconducting gaps $\Delta_{nk}(\omega = 0)$ on the FS are shown in Fig. 3.

For trilayer InB₄, the regions of superconducting gaps on the FS are consistent with the distributions of electronic EPC parameters λ_{nk}^{el} , as shown in Figs. 3(e) and 3(f), which is associated with the InB₄ band structure in Fig. 1(a). The hexagonal and petal-shape B- σ Fermi sheets lead to the strongest superconducting gap (called σ gap Δ_{σ}). The outermost hybridized-state-caused petal-shaped Fermi sheets cause the secondary superconducting gap (called μ gap Δ_{μ}). The circle Fermi sheet gives rise to the the weakest gap (called π gap Δ_{π}). At the low temperature of 5 K, the three gaps Δ_{σ} , Δ_{μ} , and Δ_{π} have values in the ranges of 9.31–10.73, 4.40–5.58, and 3.24–3.52 meV with the corresponding average values of 10.02, 4.99, and 3.38 meV, respectively. The SDOS (T = 5 K) are shown in Fig. 3(c), There are three distinct peaks related to different gaps on the FS, which are labeled Δ_{σ} , Δ_{μ} , and Δ_{π} . As the temperature rises, each energy gap gradually decreases to zero, which follows BCS theory. A high $T_c \sim 53$ K in trilayer InB₄ is obtained. The T_c is the highest among boron-based 2D materials ever discovered; much higher than that of monolayer MgB₂ ($T_c \sim 20$ K) and trilayer AlB₄ ($T_c \sim 47$ K).

All the electronic EPC parameters λ_{nk}^{el} range from $\lambda_{nk}^{el} = 0.48$ to $\lambda_{nk}^{el} = 1.64$. Obviously, the normalized distribution $\rho(\lambda_{nk}^{el})$ of λ_{nk}^{el} verifies the anisotropy of the EPC parameters. The B- σ Fermi sheets have the largest EPC parameters λ_{nk}^{el} and strength distributions $\rho_{nk}^{el,\sigma} \in [1.26, 1.64]$.

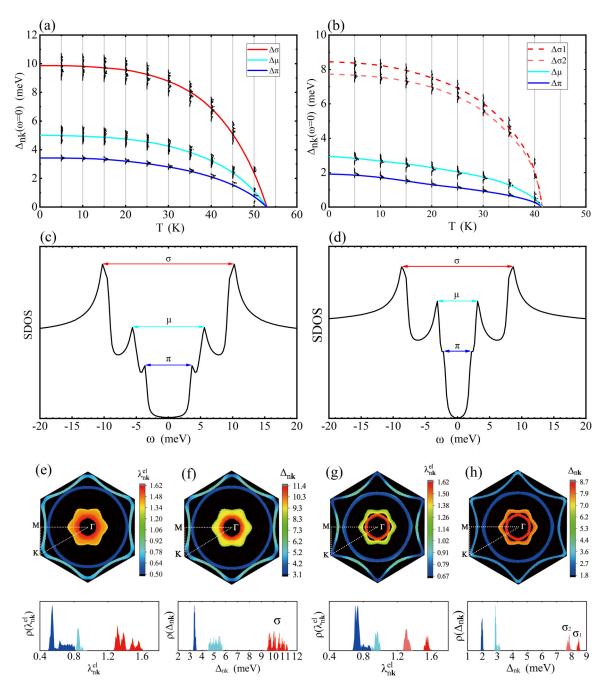


FIG. 3. Energy distribution of the gaps Δ_{nk} versus *T* for the (a) trilayer InB₄ and (b) monolayer InB₂. The red, cyan-blue, and blue curves represent the average values of Δ_{σ} , Δ_{μ} , and Δ_{π} , respectively. Superconducting density of states (SDOS) at T = 5 K of (c) trilayer InB₄ and (d) monolayer InB₂. The momentum-resolved EPC parameters λ_{nk}^{el} on FS (upper), normalized distribution $\rho(\lambda_{nk}^{el})$ (lower) of (e) monolayer InB₂ and (g) trilayer InB₄. Momentum-resolved superconducting gaps $\Delta_{nk}(\omega = 0)$ on FS (upper), normalized distribution $\rho(\Delta_{nk})$ of energy gaps at 5 K for (f) monolayer InB₂ and (h) trilayer InB₄.

The weaker EPC parameters λ_{nk}^{el} come from μ and π sheets with strength distributions of $\rho(\lambda_{nk}^{el,\mu}) \in [0.83, 0.95]$ and $\rho(\lambda_{nk}^{el,\pi}) \in [0.48, 0.80]$, respectively, which lead to an average electronic EPC parameter of 1.1. It shows that the strongest electronic EPC on the B- σ sheet around the Γ point is created by the optical soft phonons.

We now turn to the multigap superconductivity in monolayer InB₂. The electronic EPC parameters λ_{nk}^{el} on FS and the strength distribution $\rho(\lambda_{nk}^{el})$ of monolayer InB₂ are shown in Fig. 3(g). In general, the distribution of λ_{nk}^{el} on the FS coincides with the regions of Fermi sheets on the FS. Interestingly, the B- σ Fermi sheets form two separated EPC parameters $\lambda_{nk}^{el,\sigma_1}$ and $\lambda_{nk}^{el,\sigma_2}$ with the strength distributions $\rho(\lambda_{nk}^{el,\sigma_1}) \in [1.51, 1.60]$ and $\rho(\lambda_{nk}^{el,\sigma_2}) \in [1.25, 1.38]$, while the weaker EPC parameters λ_{nk}^{el} with the strength distributions $\rho(\lambda_{nk}^{el,\mu}) \in [0.67, 0.89]$ and $\rho(\lambda_{nk}^{el,\pi}) \in [0.89, 1.00]$ are observed on the

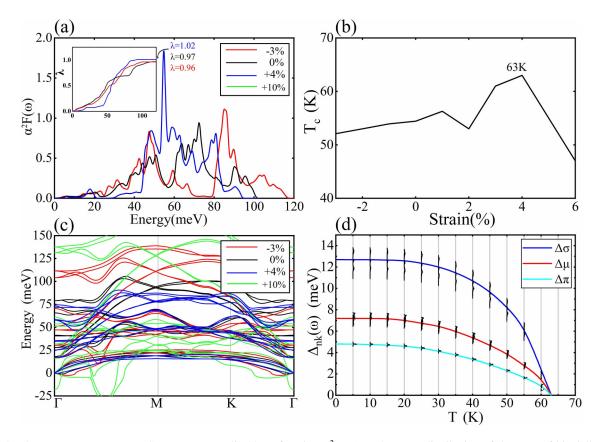


FIG. 4. Phonons spectrum, T_c , EPC parameter λ , Eliashberg function $\alpha^2 F(\omega)$, and energy distribution of the gaps of biaxially strained InB₄. (a) The phonon dispersion under the strain of 2%, 4%, 10%, -3%; tensile strain leads to higher phonon frequencies, while compressive strain leads to lower phonon frequencies. (b) T_c versus strain for InB₄ trilayer films. (c) The isotropic Eliashberg function $\alpha^2 F(\omega)$ and EPC parameter λ of InB₄ trilayer films under different strains. (d) Energy distribution of the gaps Δ_{nk} versus T for the InB₄ trilayer film under the tensile strain of 4%.

 μ and π Fermi sheets. There is no gap between $\lambda_{nk}^{el,\mu}$ and $\lambda_{nk}^{el,\pi}$. However, the $\rho_{nk}^{el,\pi}$ formed by the μ Fermi sheet appears blue at the BZ boundary, which has a distance with the π Fermi sheet. The separation of electronic EPC parameters verifies the anisotropy superconductivity.

As is shown in Fig. 3(h), the energy distribution $\rho(\Delta_{nk})$ has four distinct peaks, which represent four energy gaps on FS. At a low temperature of 5 K, the largest two gaps (called σ_1 gap Δ_{σ_1} and σ_2 gap Δ_{σ_2}) formed by the σ_1, σ_2 Fermi sheets have values of 8.31-8.60 and 7.56-8.00 meV, respectively. The weaker gaps (called μ gap Δ_{μ} and π gap Δ_{π}) observed on the μ Fermi sheet and the π Fermi sheet have values of 2.81-3.18 and 1.87-2.10 meV, respectively. The average value of Δ_{σ_1} , Δ_{σ_2} , Δ_{μ} , and Δ_{π} are 8.46, 7.78, 3.00, and 1.99 meV, respectively. Obviously, Δ_{π} and Δ_{μ} , and Δ_{μ} and Δ_{σ} are distinctly separated with the separation gap of 0.71 and 4.38 meV, respectively. However, the separation gap between Δ_{σ_1} and Δ_{σ_2} is only 0.31 meV, which is too small to define Δ_{σ_1} and $\bar{\Delta}_{\sigma_2}$ as two superconducting gaps. The SDOS (T = 5 K) only demonstrates three distinct peaks which represent Δ_{σ} , Δ_{μ} , and Δ_{π} . Therefore, monolayer InB₂ is a three-gap superconductor with a slight separation in σ gap Δ_{σ} . As is shown in Fig. 3(b), we use red ($\Delta_{\sigma 1}$) and pink $(\Delta_{\sigma 2})$ dotted lines to describe the slight separation of the σ

gap. The trend of superconducting gaps versus temperature is consistent with the temperature dependence of BCS theory. The gaps Δ_{nk} reveal a four-gap-like superconducting nature at temperatures below 15 K. The Δ_{σ_1} and Δ_{σ_2} gradually become indistinguishable at temperatures above 15 K. The gaps finally vanish at $T_c \sim 41.5$ K.

Besides this, we also calculate the superconductivity in monolayer InB₂ and trilayer InB₄ through the Allen-Dynes formula with $\mu_c^* = 0.13$. A $T_c \sim 13.2$ K is obtained in monolayer InB₂, which is only 31.8% of the value analyzed from multiband theory. For trilayer InB₄, the Allen-Dynes T_c is 28.3 K; much lower than the value ($T_c \sim 53$ K) at temperatures for which the gaps vanish. The T_c 's obtained through these two methods demonstrate a huge divergence. Bulk MgB₂ is found to be a two-gap superconductor with high $T_c \sim 39$ K in experiment and the superconducting gaps of bulk MgB₂ vanished at 39.7 K under multigap theoretical calculations [1,9]. This indicates that the multigap theory can predict a higher and more accurate T_c than the Allen-Dynes formula as compared with experiment.

Due to the lattice mismatch in two-dimensional materials, the effect of biaxial strain is considered on the novel three-gap superconductivity in the InB₄ trilayer film. We focus on the biaxial strain applied relative to the in-plane cell parameters, which is the In-In distance with a balance value of a = 3.04 Å. In Fig. 4(a), we compare the equilibrium phonon band structure with the unidirectional strain. The interatomic charge density is depleted as the distance between atoms increases under tensile strain. As a result, the interatomic bonds of structure become hard, leading to the decrease of phonon frequency. However, under compressive strain, the situation is exactly opposite. As is shown in Fig. 4(c), the shift of the $\alpha^2 F(\omega)$ peak to lower energy (in line with the general trend of the phonons) and amplification of the $\alpha^2 F(\omega)$ peak due to tensile strain cause a great enhancement to the EPC. In general, the tensile strain over the InB₄ trilayer film reduces the overall frequency of the phonon modes, thereby enhancing the EPC strength. Since $\lambda = 2 \int_0^\infty d\omega \, \omega^{-1} \alpha^2 F(\omega)$ [44], the peaks in lower frequency have a larger contribution to λ . Electron-doped and hole-doped graphenes show a similar trend in the EPC under the influence of biaxial strain in the previous work [45–47]. We further conduct first-principles calculations on the InB_4 trilayer film under strain. The FS is almost consistent with the equilibrium case, which causes the retention of the characteristics of three-gap superconductivity in the InB_4 trilayer film. As shown in Fig. 4(d), for the tensile strain of +4%, the average gaps are $\Delta_{\sigma} = 12.7$ meV, $\Delta_{\mu} = 7.3$ meV, and $\Delta_{\pi} = 4.8$ meV, which are much larger than the equilibrium case, the corresponding EPC parameter $\lambda = 1.02$, leading to the highest critical temperature T_c of 64 K. In the compression case, the critical temperature T_c decreases, but in a much less pronounced way. When the tensile strain exceeds 6% or the compressive stress exceeds 3%, the trilayer InB_4 phonon spectrum shows a large imaginary frequency, and the system becomes dynamically unstable.

In addition, we also find that the GeB_4 and ZnB_4 trilayer films have attractive superconductivity. The momentumresolved superconducting gaps Δ_{nk} versus T and the SDOS at T = 5 K for trilayers GeB₄ and ZnB₄ are shown in Fig. 5. Obviously, the GeB₄ trilayer film is predicted to be a two-gap superconductor with high $T_c \sim 48.5$ K. The σ gap (Δ_{σ}) and π gap (Δ_{π}) formed by the σ and π Fermi sheets have values in the ranges of 9.97-11.2 and 5.5-7.5 meV, respectively. As shown in Figs. 5(d)-5(f), the energy gap distributions and two distinct peaks of the SDOS indicate that the trilayer ZnB₄ is also a two-gap superconductor with $T_c \sim 28$ K. The σ gap (Δ_{σ}) and π gap (Δ_{π}) originating from the σ and π Fermi sheets have values of 3.51-5.19 and 2.15-2.38 meV, respectively. The calculated Allen-Dynes T_c of trilayer GeB₄ and ZnB₄ is 27.0 and 16.0 K, respectively; much lower than that obtained from multigap theory, which further demonstrates the great influence of the multigap nature in capturing higher T_c .

IV. CONCLUSION

In conclusion, we use the fully anisotropic Migdal-Eliashberg theory to investigate the superconductivity in a series of boron-based 2D materials. We find that only the monolayer InB₂, trilayer InB₄, ZnB₄, GeB₄, and AgB₄ are dynamically stable. The InB₄ trilayer film shows a novel three-gap superconducting behavior with high $T_c \sim 53$ K, which originates from the evident three-region distributed strong EPC parameters λ_{nk}^{el} . Three energy gaps Δ_{σ} , Δ_{μ} , and

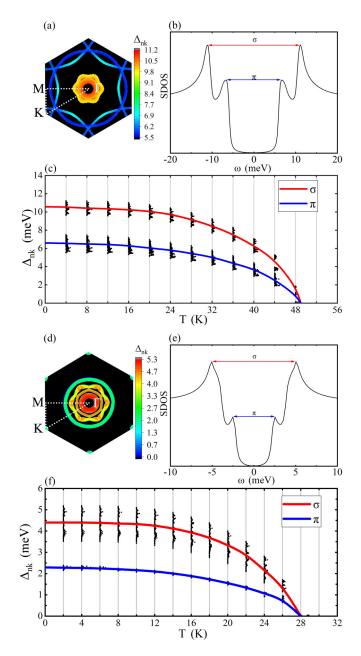


FIG. 5. Momentum-resolved superconducting gaps $\Delta_{nk}(\omega = 0)$ of (a) trilayer GeB₄ and (d) ZnB₄ on FS. Superconducting density of states (SDOS) at T = 5 K of (b) trilayer GeB₄ and (e) ZnB₄. Energy distribution of the gaps Δ_{nk} versus T for the (c) trilayer GeB₄ and (f) ZnB₄; the red and blue curves represent the average values of Δ_{σ} and Δ_{π} , respectively.

 Δ_{π} have average values of 10.02, 4.99, and 3.38 meV. We reveal that the metallization of covalent states in the boron layer and the in-plane phonon modes of surface boron are responsible for the superconductivity. In addition, the T_c can be boosted to 64 K under the tensile strain of 5%. Similarly, the InB₂ monolayer film is a three-gap superconductor with a slight separation in σ gap Δ_{σ} . Compared with InB₄, the Δ_{σ} slightly splits as Δ_{σ_1} and Δ_{σ_2} with an energy gap of 0.31 meV; the corresponding electronic EPC parameter λ_{nk}^{el} in the σ_1 and σ_2 Fermi sheets shows distinct separation, indicating the novel multigap superconducting nature. In addition, we also discover that the trilayer GeB₄ and ZnB₄ are two-gap superconductors with $T_c \sim 48.5$ and 28 K. These findings will promote the search of multigap superconductors.

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