Chiral polarization textures induced by the flexoelectric effect in ferroelectric nanocylinders

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Polar chiral structures have recently attracted much interest within the scientific community, as they pave the way towards innovative device concepts similar to the developments achieved in nanomagnetism. Despite the growing interest, many fundamental questions related to the mechanisms controlling the appearance and stability of ferroelectric topological structures remain open. In this context, ferroelectric nanoparticles provide a flexible playground for such investigations. Here, we present a theoretical study of ferroelectric polar textures in a cylindrical core-shell nanoparticle. The calculations reveal a chiral polarization structure containing two oppositely oriented diffuse axial domains located near the cylinder ends, separated by a region with a zero-axial polarization. We name this polarization configuration "flexon" to underline the flexoelectric nature of its axial polarization. Analytical calculations and numerical simulation results show that the flexon's chirality can be switched by reversing the sign of the flexoelectric coefficient. Furthermore, the anisotropy of the flexoelectric coupling is found to critically influence the polarization texture and domain morphology. The flexon rounded shape, combined with its distinct chiral properties and its localization near the surface, are reminiscent of chiral bobber structures in magnetism. In the azimuthal plane, the flexon displays the polarization state of a vortex with an axially polarized core region, i.e., a meron. The flexoelectric effect, which couples the electric polarization and elastic strain gradients, plays a determining role in the stabilization of these chiral states. We discuss similarities between this interaction and the recently predicted ferroelectric Dzyaloshinskii-Moriya interaction leading to chiral polarization states.

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I. INTRODUCTION

Research on ferroelectric materials has received growing interest over the past years, driven in part by the potential of these material systems for low-power technological applications in a broad spectrum of domains $[1,2]$, ranging from high-density data storage to optical nanodevices. A central aspect of this field of research is the formation of ferroelectric domain structures [\[3\]](#page-7-0), and more generally the micro- and nanoscale structure of the polarization field [\[4\]](#page-7-0). Traditionally, research on ferroelectrics is centered on the study of bulk materials and thin films [\[5](#page-7-0)[–7\]](#page-8-0), but recently ferroelectric nanoparticles have also attracted increasing interest [\[8–14\]](#page-8-0). In ferroelectric thin films and nanoparticles, the polarization structure is strongly affected by electrostatic (depolarizing) fields [\[15–18\]](#page-8-0), as well as by strain and strain gradients [\[19–22\]](#page-8-0) via the flexoelectric effect [\[23–26\]](#page-8-0).

Although the foundations for the theoretical description of ferroelectrics have been established decades ago [\[27\]](#page-8-0), under-

standing the complex physical properties of these material systems remains a challenge for fundamental research. Recent progress in this field, achieved to a large extent through advanced imaging techniques [\[28\]](#page-8-0) and by employing modern numerical simulations [\[29\]](#page-8-0), includes the discovery of highly complex polarization structures, such as flux closure $[5,30]$ $[5,30]$ and bubble domains $[31]$, meandering $[32,33]$ and/or labyrinthine $[11,34]$ structures, non-Ising type chiral domain walls [\[35\]](#page-8-0), polarization vortices in thin layers [\[36–38\]](#page-9-0), nanodots [\[39\]](#page-9-0) or nanopillars [\[40\]](#page-9-0), or polar skyrmions [\[41,42\]](#page-9-0).

While skyrmions and other chiral structures have dominated the past decade of research in magnetism [\[43\]](#page-9-0), these topological states have received less attention by the ferroelectric community. Only recently a strong interest has emerged in chiral polarization structures, which can be attributed to the observation of skyrmion states in ferroelectrics [\[41,42\]](#page-9-0). However, the theoretical understanding of these structures is not as advanced as it is in the case of their magnetic counterparts, and the mechanism that underpins the formation of skyrmions in ferroelectrics is not fully understood. The fundamental interaction stabilizing the magnetic version of these structures in chiral ferromagnets [\[44,45\]](#page-9-0) is the Dzyaloshinskii-Moriya Interaction (DMI). The DMI favors the formation of helical structures with a well-defined handedness as they occur,

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e.g., along the radial direction of skyrmions. As scientists working on ferroelectrics hope to replicate the success that chiral structures have witnessed in magnetism, the possibility of a "ferroelectric DMI" has recently been discussed [\[46\]](#page-9-0). However, Erb and Hlinka [\[47\]](#page-9-0) showed that only very few exotic ferroelectrics could theoretically sustain an intrinsic DMI-type interaction since it requires particular symmetry properties of the crystal lattice. Here we discuss the flexoelectric coupling as an alternative mechanism that can generate chiral polarization states in ferroelectrics.

The thermodynamic description of the flexoelectric effect is given by a Lifshitz invariant in the free-energy expansion [\[22\]](#page-8-0). It is known that, in magnetic materials, the occurrence of similar Lifshitz invariants converts directly into an antisymmetric coupling known as the DMI [\[48,49\]](#page-9-0), which favors the formation of helicoidal structures with a specific chirality. The existence of a ferroelectric counterpart of the DMI was recently predicted by first-principles simulations [\[46\]](#page-9-0). The ferroelectric analog of the DMI was discussed in the context of Lifshitz invariants by Strukov and Levanyuk [\[50\]](#page-9-0), and more recently by Erb and Hlinka [\[47\]](#page-9-0), who argued that a ferroelectric DMI can exist. In addition to the remarkable similarity in the mathematical form of the flexoelectric Lifshitz invariant and DMI, the flexoelectric term appears to have a similar impact as the DMI in terms of the formation of chiral structures.

By means of the finite-element modeling (FEM) based on the Landau-Ginzburg-Devonshire (LGD) theory, this paper shows that an anisotropic flexoelectric effect can give rise to a previously unexplored type of polarization state with distinct chiral properties. Remarkably, these homochiral properties are not induced by a DMI term. This finding suggests that the recently discussed DMI in ferroelectrics is not the only possible mechanism for the formation of homochiral polarization states, and that anisotropic flexoelectric effects offer an alternative pathway to stabilize such structures in ferroelectric nanostructures. We discuss common aspects of the DMI and the flexoelectric effect, which are both derived from Lifshitz invariants in the framework of the Landau theory of secondorder phase transitions [\[22\]](#page-8-0).

II. CONSIDERED PROBLEM AND MATERIAL PARAMETERS

Using a LGD phenomenological approach along with electrostatic equations and elasticity theory, we model the polarization, the internal electric field, and the elastic stresses and strains in a core-shell nanoparticle using FEM, where the ferroelectric core is made of $BaTiO₃$ and has a cylindrical shape. The aspect ratio of the nanocylinder radius *R* to its length *h* is significantly higher than unity. The *z* axis is parallel to the cylinder axis (Fig. 1). The shell is an elastically soft paraelectric or high-*k* semiconductor with a thickness $\Delta R \ll R$ and screening length $\Lambda \geq 1$ nm. The coverage can be artificial (e.g., a soft organic semiconductor or vacancy-enriched $SrTiO₃$ or natural, and in the latter case it would originate from the polarization screening by surrounding media. The core-shell nanoparticle is placed in a very soft elastic medium.

The LGD free-energy functional *G* of the nanoparticle core includes a Landau-type energy—an expansion on pow-

FIG. 1. A cylindrical ferroelectric nanoparticle (core) of radius *R*, covered with an elastically soft semiconducting shell with a thickness $\Delta R \ll R$ and screening length Λ of 1 nm, placed in an isotropic elastically soft effective medium. The direction of axial polarization P_3 is shown by the straight orange arrow, and lateral components $P_{1,2}$ are shown by the curled red-blue arrow to highlight their vortex-type structure.

ers of 2-4-6 of the polarization (*Pi*), *G*Landau; a polarization gradient energy, G_{grad} ; an electrostatic energy, G_{el} ; an elastic, electrostriction contribution *Ges*; a flexoelectric contribution, G _{flexo}; and a surface energy, G_S . It has the form [\[51\]](#page-9-0)

$$
G = G_{\text{Landau}} + G_{\text{grad}} + G_{el} + G_{\text{flexo}} + G_{\text{flexo}} + G_S,
$$
\n(1a)

$$
G_{\text{Landau}} = \int_{Vc} d^3r \left[a_i P_i^2 + a_{ij} P_i^2 P_j^2 + a_{ijk} P_i^2 P_j^2 P_k^2 \right], \quad (1b)
$$

$$
G_{\text{grad}} = \int_{V_c} d^3 r \frac{g_{ijkl}}{2} \frac{\partial P_i}{\partial x_j} \frac{\partial P_k}{\partial x_l},\tag{1c}
$$

$$
G_{el} = -\int_{Vc} d^3r \Big(P_i E_i + \frac{\varepsilon_0 \varepsilon_b}{2} E_i E_i \Big), \tag{1d}
$$

$$
G_{es} = -\int_{V_c} d^3r \left(\frac{s_{ijkl}}{2}\sigma_{ij}\sigma_{kl} + Q_{ijkl}\sigma_{ij}P_kP_l\right), \qquad (1e)
$$

$$
G_{\text{flexo}} = -\int_{Vc} d^3r \frac{F_{ijkl}}{2} \left(\sigma_{ij} \frac{\partial P_k}{\partial x_l} - P_k \frac{\partial \sigma_{ij}}{\partial x_l} \right), \tag{1f}
$$

$$
G_S = \frac{1}{2} \int_S d^2 r a_{ij}^{(S)} P_i P_j. \tag{1g}
$$

Here V_C is the core volume. The coefficient a_i linearly depends on temperature *T*, $a_i(T) = \alpha_T [T - T_C]$, where α_T is the inverse Curie-Weiss constant and T_C is the ferroelectric Curie temperature renormalized by surface tension/intrinsic surface stresses [\[52–54\]](#page-9-0) and surface bond contraction [\[55,56\]](#page-9-0). Tensor components a_{ij} are regarded as temperature independent. The tensor a_{ij} is positively defined if the ferroelectric material undergoes a second-order transition to the paraelectric phase and negative otherwise. The higher nonlinear tensor a_{ijk} and the gradient coefficients tensor *gijkl* are positively defined and regarded as temperature independent. In Eq. (1e), σ_{ij} is the stress tensor, s_{ijkl} is the elastic compliances tensor, and Q_{ijkl} is the electrostriction tensor. In the Lifshitz invariant, $[Eq. (1f)],$ F_{ijkl} is the flexoelectric tensor.

Landau-Khalatnikov equations [\[57,58\]](#page-9-0) obtained from a variation of the free energy (1), mathematical formulation of

FIG. 2. Distribution of the polarization component *P*³ (top row) inside a cylindrical nanoparticle and a magnified view on the flexon structure (bottom row). The arrows show the orientation of polarization vector *P*. The images are calculated without electrostriction (Q_{ii} = 0) and flexoelectric ($F_{ij} = 0$) couplings (a); with electrostriction coupling ($Q_{ij} \neq 0$) and negative (b), or zero (c), or positive (d) values of flexoelectric coefficients F_{ij} . The values of F_{ij} and all other parameters are given in Table SI, $T = 300$ K. Note the different scales for P_3 distributions in plots (a)–(d) in order to maintain a contrast between the different regions.

the electrostatic and elastic subproblem (see, e.g., Ref. [\[59\]](#page-9-0)), initial and boundary conditions (see, e.g., Refs. [\[60–62\]](#page-9-0)), sensitivity to the shape of the cylinder ends, polarization gradient coefficients, shell dielectric permittivity and semiconducting properties, and other details of FEM are given in Appendix A of Supplemental Material. [\[63\]](#page-9-0). The ferroelectric, dielectric, and elastic properties of the $BaTiO₃$ core are collected from Refs. [\[64–69\]](#page-9-0) and given in Table SI.

III. RESULTS OF FINITE-ELEMENT MODELING

A. FEM results at room temperature

Images in Figs. $2(a)$ and $3(a)$ are calculated without electrostriction ($Q_{ij} = 0$) and flexoelectric ($F_{ij} = 0$) couplings between the electric polarization and elastic stresses. In this case, a very prolate dipolar kernel oriented along the z axis appears inside the cylindrical core. The kernel has relatively thin 180 ° domain walls, which are mostly uncharged because they are parallel to the kernel axis and cylinder lateral surface. The bound charges appear at the walls only in a small spatial region near the kernel that is in contact with the cylinder ends, where the 180° walls become counter head-to-head walls. The axial polarization P_3 inside the kernel is high, $P_3 \sim -(20-25) \mu C/cm^2$ (this is very close to the bulk polar-

ization of BaTiO₃ \sim 26 μ C/cm²), and the surrounding core has relatively small axial polarization of the opposite sign, $P_3 \sim (0-5) \mu C/cm^2$. The lateral components of polarization, *P*¹ and *P*2, form a two-dimensional vortex without a central empty core, because a dipolar kernel evolves instead (Fig. S4 [\[63\]](#page-9-0)). The two symmetrical Bloch points with $P = 0$ are located at the junction of the dipolar kernel with the cylinder ends. The "up" or "down" orientation of polarization component *P*³ inside the kernel is determined by random noise in the initial conditions.

Images in Figs. $2(b)-2(d)$ and $3(b)-3(e)$ are calculated for a nonzero electrostriction coupling ($Q_{ij} \neq 0$) and either negative, zero, or positive values of the flexoelectric coefficients F_{ij} . In the presence of electrostriction coupling the dipolar kernel disappears completely [Figs. $2(c)$ and $3(c)$]. The flexoelectric effect induces an axial component of polarization consisting of two oppositely oriented diffuse P_3 domains located near the cylinder ends and separated by a region with $P_3 \approx 0$ [Figs. 2(b), 2(d) and [3\(b\),](#page-3-0) [3\(d\)\]](#page-3-0).

The diffuseness of the P_3 -domain walls is dictated by the need to decrease the depolarization field produced by the bound charges of the head-to-head domain walls. The *P*³ domains are located near the cylinder ends, and their length (about 10 nm) and lateral size (about 5 nm) are almost independent on the cylinder length if $h \gg 5$ nm. The

FIG. 3. Distribution of the polarization component *P*³ (top row) in the *X Z* cross section of the nanoparticle. Images are calculated without electrostriction $(Q_{ij} = 0)$ and flexoelectric $(F_{ij} = 0)$ coupling (a); with electrostriction coupling $(Q_{ij} \neq 0)$ and negative (b), zero (c), positive (d), or high-positive (e) values of flexoelectric coefficients F_{ij} . The bottom part (f) is the dependence of the maximal (red curve, P_{max}) and minimal (blue curve, P_{min}) values of P_3 on the relative amplitude of the flexoelectric coupling strength f . The green curve is the extremal (maximal or minimal) value P_e in the center of the top axial P_3 domain. Here $F_{ij} = fF_{ij}^0$; the values of F_{ij}^0 and all other parameters are given in Table SI, $T = 300$ K. The *Z* profile of the polarization topological index $n(z)$ is shown in the inset (g) for zero (black line), positive (red curve), and negative (blue curve) F_{ij} . Note the different scales for P_3 in the plots (a) and (b)–(e) in order to maintain a contrast between the different regions.

component *P*₃ is very small ($|P_3| \le 0.4 \mu$ C/cm²), but it increases up to 1.2μ C/cm² with the flexoelectric coupling increase [Figs. $3(e)$] and then saturates [Fig. $3(f)$]. The axial *P*³ domains, which have opposite direction of polarization, change their direction under the transformation $F_{ij} \rightarrow -F_{ij}$ [compare the position of red and blue diffuse spots of the *P*³ distributions in Figs. $3(b)$ and $3(d)$], while the distribution of the lateral components $P_{1,2}$ and the polarization magnitude P are virtually independent of the F_{ij} sign and magnitude (Figs. S4 and S5 [\[63\]](#page-9-0)).

The maximal (P_{max}) and minimal (P_{min}) values of P_3 are shown by the red and blue curves in Fig. $3(f)$. The values *P*max and *P*min are even functions of the flexoelectric coupling strength *f*, where $F_{ij} = fF_{ij}^0$ and the reference values of F_{ij}^0 are given in Table SI. The extremal (maximal or minimal) value P_e in the center of the diffuse axial P_3 domain is an odd function of *f*, which is zero at $F_{ij} = 0$ [the green

curve in Fig. $3(f)$]. Note that the P_e value frequently differs from P_{max} and P_{min} values due to the presence of the small 16 *P*³ domains localized near the top and bottom junction of the sidewall with the cylinder ends (bottom row in Fig. S5 [\[63\]](#page-9-0)).

For the remainder of the paper, we refer to the localized polarization structure near the wire ends as a "flexon" for the sake of brevity and to underline the flexoelectric nature of its axial polarization. The main effect of a change of sign in the flexoelectric coefficients is the reorientation of the flexon axial polarization. The polarization structures at the wire ends shown in Figs. $2(b)-2(d)$ and Figs. $3(b)-2(d)$ display localized chiral structures with different chirality on opposite ends of the wire, and their chirality changes upon reversal of the sign of the flexoelectric coupling constant.

To understand the chirality change, we derived in Appendix E [\[63\]](#page-9-0) an approximate analytical expression for the polarization distribution inside the flexon:

$$
P_1(\rho, \varphi, z) \approx p(\rho, z) \sin \varphi, P_2(\rho, \varphi, z) \approx -p(\rho, z) \cos \varphi
$$
\n(2a)
\n
$$
P_3(\rho, \varphi, z) \approx \frac{\frac{Q_{44}}{S_{44}} p(\rho, z) [u_{13}(\rho, \varphi, z) \sin \varphi - u_{23}(\rho, \varphi, z) \cos \varphi] - \frac{F_{11} - F_{44} - F_{12}}{S_{11} - S_{12}} \frac{\partial}{\partial z} u_{33}(\rho, \varphi, z)
$$
\n(2b)
\n
$$
P_3(\rho, \varphi, z) \approx \frac{\frac{Q_{44}}{S_{44}} p(\rho, z) [u_{13}(\rho, \varphi, z) \sin \varphi - u_{23}(\rho, \varphi, z) \cos \varphi] - \frac{F_{11} - F_{44} - F_{12}}{S_{11} - S_{12}} \frac{\partial}{\partial z} u_{33}(\rho, \varphi, z)
$$
\n(2b)

where $\{\rho, \varphi, z\}$ are cylindrical coordinates, the function $p(\rho, z) \sim \tanh(\frac{\rho}{L_c^x})$, L_c^x and L_c^z are lateral and axial correlation lengths. The functions $u_{ij}(\rho, \varphi, z)$ are elastic strains, s_{ij} are elastic compliances; Q_{ij} are electrostriction tensor components, g_{ij} are polarization gradient coefficients written in Voigt notations. The first term in Eq. $(2b)$ is induced by the electrostriction coupling, and the second term, proportional to $\frac{F_{11}-F_{44}-F_{12}}{s_1-s_1} \frac{\partial}{\partial z} u_{33}(\rho, \varphi, z)$, is the flexon.

In order to quantify the chirality of the polarization structure and its variation along the cylinder axis, in Appendix F $[63]$ we calculate the topological index = $\frac{1}{4\pi} \int_S \vec{p} \left[\frac{\partial \vec{p}}{\partial x} \times \frac{\partial \vec{p}}{\partial y} \right] dxdy$ [\[70\]](#page-9-0) of the unit polarization orientation $\vec{p} = \frac{\vec{p}}{P}$ for the integration over the cylinder cross section {*x*, *y*}. For the case of $P_3(\rho = R, z) \rightarrow 0$, the *z* dependence of the topological index is

$$
n(z) = -\frac{P_3(\rho = 0, z)}{2P(\rho = 0, z)} \approx -\frac{\text{sign}[f]z}{2\sqrt{1 + (z^2/B)}}.
$$
 (3)

Here sign[f] is the sign of the flexoelectric coefficients F_{ij} , *B* is a positive constant, which depends on the absolute value of $|F_{ij}|$. $n(z)$ is a normalized profile of $P_3(\rho = 0, z)$, and so $n(z) = 0$ for $F_{ij} = 0$, and its sign is defined by the sign of F_{ij} . The dependence $n(z)$ is shown in Fig. [3\(g\)](#page-3-0) and Fig. S10(a) for zero, positive, and negative F_{ij} . Since the value $P(0, z)$ is very close to the $P_3(0, z)$ near the cylinder ends [Figs. S10(a) and $S10(b)$], and $P_3(0, z)$ vanishes in the central part of a nanoparticle, the topological index continuously changes from −1/2 to +1/2 with a *z*-coordinate change from one cylinder end to the other. The result clearly shows the localization of the chiral structures—the flexons—at the ends of the wires. The topological index, which can be interpreted as the degree to which a structure is chiral, changes sign from one end to the other, and changes sign upon reversal of the sign of F_{ij} . It also increases in magnitude with increasing absolute value of $|F_{ii}|$. These properties are evidence of an obvious correlation between the flexoelectric effect and the formation of chiral polarization structures.

The revealed type of isolated chiral polarization structures, i.e., flexons, display topological features of a threedimensional meron. In this sense, the polarization vortex in the *XY* plane can be interpreted as the Bloch-like transition region of a meron connecting polarization directions of opposite P_3 sign in the core region and in the outer cylindrical shell (Fig. [2\)](#page-2-0). The flexon polarization \vec{P} develops a characteristic drop shape with a chiral structure localized near the surfaces of the cylinder that is reminiscent of the chiral-bobber state found in noncentrosymmetric magnetic films [\[71\]](#page-9-0) and nanoparticles [\[72\]](#page-10-0). It is worth noting that similar, skyrmionlike configurations at the ends of cylindrical nanowires have

also been predicted analytically [\[73\]](#page-10-0) and numerically [\[74\]](#page-10-0) in the case of nonchiral ferromagnetic materials, but only in the form of transient configurations appearing during the dynamic magnetization reversal process. Here, the skyrmionlike polarization structures appear as stable states in the ferroelectrics, owing to a chiral symmetry-breaking effect of the flexoelectric coupling. In contrast to previous findings [\[75–77\]](#page-10-0), the flexon structure is chiral [\[78\]](#page-10-0) and almost uncharged because div $\vec{P} \cong$ 0 (Fig. S9(b) [\[63\]](#page-9-0)).

As a rule, the flexoelectric tensor component *F*⁴⁴ is either poorly known from experiments or ill-defined from *ab initio* calculations; therefore, we can vary it over a wide range to determine the degree by which the flexoelectric coupling anisotropy influences the morphology of the polarization state. Corresponding FEM results are shown in Fig. [4.](#page-5-0) The top and middle rows illustrate that the P_3 distribution changes very strongly when *F*⁴⁴ varies from high negative to high positive values, while the other components of the flexoelectric tensor are fixed and equal to the tabulated values $F_{11} = 2.4 \times 10^{-11} \text{m}^3/\text{C}$ and $F_{12} = 0.5 \times 10^{-11} \text{m}^3/\text{C}$.

The flexon contains two pronounced axial domains located near the cylinder ends, which have thick diffuse domain walls and opposite polarization directions, and exist at high negative [Fig. $4(a)$] and high positive [Figs. $4(b)$ and $4(e)$] F_{44} values. The P_3 domains become smaller and more diffuse with a decrease of $|F_{44}|$, but they are still visible and practically do not change their shape, size, or polarization distribution for small $|F_{44}|$ values over the range $|F_{44}| \leq 0.06$ [Fig. [4\(b\)\]](#page-5-0). The flexon becomes faint and almost disappears when *F*⁴⁴ approaches the value $F_{44} = F_{11} - F_{12} = 1.9 \times 10^{-11}$ m³/C corresponding to the isotropic symmetry of F_{ii} [Fig. [4\(c\)\]](#page-5-0). The value will be referred to as "isotropic" below.

The dependence of the maximal (red curve, P_{max}) and minimal (blue curve, P_{min}) values of the polarization component P_3 on the relative amplitude f of the flexoelectric coefficient F_{44} is shown in Fig. [4\(f\),](#page-5-0) where $F_{44} = fF_{44}^0$ and $F_{44}^0 = 0.06 \times 10^{-11}$ m³/C. The values P_{max} and P_{min} reach a very diffuse plateaulike minimum and maximum, respectively, at the isotropic value $F_{44} = F_{11} - F_{12}$. The green curve in Fig. $4(f)$ is the extremal value P_e in the center of the bottom axial P_3 domain. The extremal value P_a in the center of the diffuse P_3 domain changes its sign in the immediate vicinity of the isotropic value $F_{44} = F_{11} - F_{12}$. The values P_{max} , P_{min} , and *Pe* have no definite parity, because they are neither odd nor even functions of the flexoelectric coefficient *F*⁴⁴ amplitude *f* . From Fig. [4\(f\)](#page-5-0) we can conclude that the anisotropy of the flexoelectric coupling critically influences the morphology of the flexon, where the axial part of the flexon polarization is proportional to $-\frac{F_{11}-F_{44}-F_{12}}{s_{11}-s_{12}}$ (Eq. (2b)]; this proportionality along with Fig. S9 qualitatively describes the curves' behavior in Fig. [4\(f\).](#page-5-0)

Flexoelectric coefficient F_{44} amplitude f (rel. units)

FIG. 4. Distribution of the polarization component *P*³ in the *X Z* cross section of cylindrical core (top row). Images are calculated for the fixed values $F_{11} = 2.4 \times 10^{-11}$ m³/C and $F_{12} = 0.5 \times 10^{-11}$ m³/C, while the value of F_{44} varies from −6 to 6 (in 10⁻¹¹m³/C) as indicated in the legends. The bottom part (f) is the dependence of the maximal (red curve, P_{max}) and minimal (blue curve, P_{min}) values of P_3 on the relative amplitude *f* of the flexoelectric coefficient F_{44} in the core. The green curve is the extremal (maximal or minimal) value P_e in the center of the bottom axial P_3 domain. Here $F_{44} = fF_{44}^0$ and $F_{44}^0 = 0.06 \times 10^{-11}$ m³/C. The electrostriction coupling coefficients Q_{ij} and all other parameters are listed in Table SI, $T = 300$ K. Note the different scales for P_3 in the plots (a)–(e) in order to maintain a contrast between the different regions.

B. Temperature behavior of the flexon-type polarization distribution

To define the temperature interval in which flexons exist as stable or metastable states, we performed FEM in the temperature range from 50 to 400 K using different initial distribution of polarization in a cylindrical core. Typical FEM results are shown in Fig. 5 , where the columns (a)–(e) correspond to the temperature increase from 240 to 370 K; the structure of the azimuthal components of the polarization vector, P_1 and P_2 , is vortexlike and shows weak variations when approaching the surface over the same temperature range (see the direction of arrows at the bottom image of Fig. S7 [\[63\]](#page-9-0)).

A bidomain configuration of P_3 is stable at temperatures lower than 250 K [Fig. $5(a)$]. The bidomain structure has a relatively thin uncharged 180° domain wall inside the cylinder, which transforms into a flux-closure domain near the electrically open cylinder ends. An initial four-domain polarization distribution relaxes to a flexonlike domain structure in the temperature range 260 K $\langle T \rangle$ < 360 K [Figs. [5\(b\)–5\(d\)\]](#page-6-0). The flexon gradually disappears at $T > 370$ K [the middle image in Fig. $5(e)$]. The ferroelectric polarization inside the core significantly decreases at $T > 370$ K and completely disappears at $T \sim 400$ K [the top image in Fig. [5\(e\)\]](#page-6-0). The structure becomes faint with a temperature increase above 370 K [Fig. $5(e)$]; hence, the flexon-type polarization distribution exists in a relatively wide temperature range $260 K <$ *T* < 360 K. The axial counterdomains inherent to flexons are the most pronounced feature over the narrower range 290 K \lt $T < 340$ K.

IV. DISCUSSION

Any deviation from a four-quadrants domain configuration in the flexon-type polarization distribution is found to be metastable. This is because the antiparallel bidomain-type polarization distribution (starting from a random noise) has a lower free energy in a $BaTiO₃$ cylindrical nanoparticle. The derived energy values at room temperature are G_{fl} = -3.6×10^{-18} J and $G_{bd} = -4.0 \times 10^{-18}$ J in the flexon and the bidomain structure, respectively. The energy difference

FIG. 5. Distribution of polarization component P_3 in *XZ* cross sections of the nanoparticle core. Different columns are calculated for the temperatures $T = 240, 270, 300, 340,$ and 370 K: (a), (b), (c), (d), and (e), respectively. All other parameters are listed in Table SI. Note the different scales for P_3 in the plots (a) and (b)–(e) in order to maintain a contrast between the different regions.

between these states, $\Delta G = 4 \times 10^{-19}$ J is much higher (about 100 $k_B T$) than the thermal energy barrier $k_B T$ at room temperature. However, the ratio $\frac{\Delta G}{k_B T}$ strongly decreases as the temperature increases. The linear relative dielectric permittivity in both states is about 110 at room temperature and it strongly increases with temperature. Furthermore, our numerical simulations show that a spontaneous off-field transition from the flexon to the bidomain polarization state does not occur, whereas the in-field transition is possible (corresponding hysteresis loops are shown in Appendix D in Supplemental Material [\[63\]](#page-9-0)). Thus, the bidomain and flexon states of a prolate core-shell ferroelectric nanoparticle can be considered as the excited and ground state of a two-level system suitable for information recording. The two-level system can imitate qubits operating in the temperature range where $1 < \frac{\Delta G}{k_B T}$ < 5. Furthermore, the bidomain polarization state corresponds to an antiferroelectriclike state of the nanoparticle polarization, which can be represented as two antiparallel nanoscopic dipoles. The flexon is a much more complex achiral vortexlike configuration containing two counterdipole nanodomains with diffuse relaxorlike polar properties. Thus, an ensemble of prolate core-shell ferroelectric nanoparticles, where a given nanoparticle is either in a flexon or a bidomain state, can be an alternative media for information processing. The media may exhibit unusual properties including antiferroelectric and/or relaxorlike polarization states, which can lead to additional functionalities. Note that the appearance of the antiferroelectric and relaxorlike glass states, as well as a newly discovered liquid glass state [\[79\]](#page-10-0) with additional (anti)ferroelectric ordering and other cross-talk effects, are possible in a suspension of the prolate core-shell ferroelectric nanoparticles.

The relatively wide temperature range (about 100 K) corresponding to the stability or metastability of the flexon-type polarization distribution gives us the hope that the domain

morphology can be observed experimentally. Specifically, the measurements of local vertical displacement by piezoresponse force microscopy (PFM) visualize the distribution of $P_3(\vec{r})$ at distances ∼10 nm from the ends of a nanoparticle, but the resolution procedure for the local piezoresponse of diffuse do-mains under the surface is so far not straightforward [\[80,81\]](#page-10-0). This is because PFM is a near-field method. A complementary tool to probe chiral polar textures is far-field nonlinear optical microscopy [\[82\]](#page-10-0), which has a comparatively much lower resolution than PFM, yet optimum focusing methods and the experimental geometry allow for overcoming the diffraction limit. For example, second-harmonic generation microscopy was successfully used by the community to precisely study semiconductor nanowires [\[83\]](#page-10-0). This method should also be capable of providing complementary information on the 3D ferroelectric domain structure (being sensitive to $P_{1,2}$ and P_3) by using polarimetry analysis (see, e.g., Ref. [\[84\]](#page-10-0)). Another promising method is resonant elastic soft x-ray scattering, a synchrotron-based method sensitive to chiral polar arrange-ments through dichroism effects [\[85,86\]](#page-10-0). This method was successfully applied to detect different topological structures, including vortices $[36]$, skyrmions $[41]$, and chiral domain walls [\[87\]](#page-10-0).

Note that Liu *et al.* [\[39\]](#page-9-0) revealed that an axial polarization component of the vortex can appear in ferroelectric $PbTiO₃$ nanodots due to the flexoelectric effect. Thus, Ref. [\[39\]](#page-9-0) and this work predict different flexosensitive vortexlike states with an axial polarization in ferroelectrics nanoparticles of various geometry. This work reveals the critical influence of the flexoelectric coefficients sign and anisotropy on the appearance and properties of the axial polarization, and, most important, on the chirality of a ferroelectric vortex. Qualitatively, both works, Ref. [\[39\]](#page-9-0) and this one, illustrate that flexoelectricity can change the chiral state of a polarization texture, and this work studies the behavior of a topological index (in fact a skyrmion number) that quantifies the change [see Fig. $3(g)$].

The main features characterizing polarization structures stabilized by DMI-type interactions are their breaking of chiral symmetry and their incommensurability, i.e., a long-period modulation in space that is unrelated to the crystalline lattice parameter. The appearance of such chiral incommensurate phases can generally be attributed to specific energy terms in the Landau-Ginzburg potential, known as Lifshitz invariants [\[49](#page-9-0)[,88\]](#page-10-0). Only a few ferroelectrics have crystalline structures whose symmetry allows such Lifshitz invariants; therefore, an interaction directly analogous to the magnetic DMI is generally not found in ferroelectrics. However, although not identical in its mathematical form, the energy density due to the flexoelectric coupling [\[23\]](#page-8-0) is similar to a DMI-type energy term related to a linear Lifshitz invariant. Accordingly, we argue that the flexoelectric coupling can lead to polarization states with properties similar to those generated by a DMI-type interaction. Such a connection between flexoelectric coupling, Lifshitz invariants, and DMI has been discussed before in the case of liquid crystals [\[89\]](#page-10-0). It was found that flexoelectricity in liquid crystals can play a central role in the development of modulated phases that are analogous to those known from chiral ferromagnets [\[90\]](#page-10-0). Our study shows that—similar to liquid crystals, where elastic strain fields couple to electric fields—the flexoelectric effect through which strain fields couple to the electric polarization field can lead to comparable modulated phases with chiral properties in a ferroelectric nanoparticle.

V. SUMMARY AND CONCLUSIONS

Using FEM simulations based on the phenomenological LGD approach alongside electrostatic equations and elasticity theory, we identify a characteristic polarization structure developing between two oppositely oriented axial domains located near the cylinder ends. This polar structure, which we named flexon, displays chiral features that are connected to the flexoelectric coupling. In the azimuthal plane, the flexon polarization forms a localized chiral structure resembling a meron, or a vortex with a central kernel. Analytical calculations and FEM prove that the flexon axial polarization, and thus its chirality, switches upon a change of the sign of the flexoelectric coefficients. We also observe that the anisotropy of the flexoelectric coupling critically influences the flexon formation and the related domain morphology. This observation corroborates the link between chirality and flexocoupling, and it identifies the flexoelectric effect as the driving force stabilizing these structures.

While in magnetic systems with strong DMI, similar localized chiral structures have been reported [\[71\]](#page-9-0), the polarization state discussed here is formed without any ferroelectric counterpart of the DMI [\[46\]](#page-9-0). We recall that, like the DMI, the flexoelectric coupling is derived as a Lifshitz invariant [Eq. $(1f)$] in the context of the Landau theory of phase transitions [\[22\]](#page-8-0), and that such linear Lifshitz invariants generally play a key role in the formation of helical structures [\[49](#page-9-0)[,88\]](#page-10-0). The fact that both the DMI and the flexoelectric stabilize structures with a specific chirality demonstrates an analogy between these two interactions which appears to have been overlooked in the literature of ferroelectric solids. An important difference compared to the classical DMI is that the flexoelectric coupling is ubiquitous in ferroelectrics, whereas the ferroelectric DMI is forbidden by symmetry in most material types. Therefore, a chiral interaction mediated by the flexoelectric effect can potentially be found in all ferroelectrics. The coupling of the electric polarization and elastic strain gradients could thus be a much more commonly accessible alternative interaction for the formation of chiral and achiral structures [\[75,76\]](#page-10-0). This coupling could also open the possibility of generating and dissolving chiral polarization states through strain engineering [\[91\]](#page-10-0).

We predict that the pronounced flexon-type polarization distribution with two axial counterdomains exists in the temperature range 290 K $< T < 340$ K. The relatively wide temperature range (about 50 K) corresponding to the stability or metastability of the flexon-type polarization distribution gives us the hope that the flexons can be observed experimentally. However, the analysis of the hysteresis loops leads to the conclusion that flexons and other domain configurations cannot be resolved from macroscopic measurements of the average polarization in a homogeneous electric field. We anticipate that flexons can be reliably observed, e.g., by the local methods using a strong gradient of electric field, such as PFM, which gives us the information about the distribution of polarization with a nanoscale resolution.

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- [1] [J. F. Scott, Applications of modern ferroelectrics,](https://doi.org/10.1126/science.1129564) Science **315**, 954 (2007).
- [2] *Physics of Ferroelectrics*, 1st ed., edited by K. M. Rabe, C. H. Ahn, and J.-M. Triscone (Springer, Berlin, 2007).
- [3] A. K. Tagantsev, L. E. Cross, and J. Fousek, *Domains in Ferroic Crystals and Thin Films*, 1st ed. (Springer, New York, 2010).
- [4] S. V. Kalinin, Y. Kim, D. Fong, and A. N. Morozovska, Surfacescreening mechanisms in ferroelectric thin films and their effect [on polarization dynamics and domain structures,](https://doi.org/10.1088/1361-6633/aa915a) Rep. Prog. Phys. **81**, 036502 (2018).
- [5] C. T. Nelson, B. Winchester, Y. Zhang, S. J. Kim, A. Melville, C. Adamo, C. M. Folkman, S. H. Baek, C. B. Eom, D. G.

Schlom, L. Q. Chen, and X. Pan, Spontaneous vortex nan[odomain arrays at ferroelectric heterointerfaces,](https://doi.org/10.1021/nl1041808) Nano Lett. **11**, 828 (2011).

- [6] C.-L. Jia, K. W. Urban, M. Alexe, D. Hesse, and I. Vrejoiu, Direct observation of continuous electric dipole rotation in fluxclosure domains in ferroelectric Pb $(Zr, Ti)O₃$, Science 331, 1420 (2011).
- [7] Y. L. Tang, Y. L. Zhu, X. L. Ma, A. Y. Borisevich, A. N. Morozovska, E. A. Eliseev, W. Y. Wang, Y. J. Wang, Y. B. Xu, Z. D. Zhang, and S. J. Pennycook, Observation of a periodic array of flux-closure quadrants in strained ferroelectric $PbTiO₃$ films, Science **348**[, 547 \(2015\).](https://doi.org/10.1126/science.1259869)
- [8] S. Lin, T. Lü, C. Jin, and X. Wang, Size effect on the dielectric properties of BaTiO₃ nanoceramics in a modified Ginsburg-[Landau-Devonshire thermodynamic theory,](https://doi.org/10.1103/PhysRevB.74.134115) Phys. Rev. B **74**, 134115 (2006).
- [9] Yu. A. Barnakov, I. U. Idehenre, S. A. Basun, T. A. Tyson, and D. R. Evans, Uncovering the mystery of ferroelectric[ity in zero dimensional nanoparticles,](https://doi.org/10.1039/C8NA00131F) Nanoscale Adv. **1**, 664 (2019).
- [10] E. A. Eliseev, A. V. Semchenko, Y. M. Fomichov, M. D. Glinchuk, V. V. Sidsky, V. V. Kolos, Yu M. Pleskachevsky, M. V. Silibin, N. V. Morozovsky, and A. N. Morozovska, Surface and finite size effects impact on the phase diagrams, polar, and dielectric properties of (Sr, Bi)Ta₂O₉ ferroelectric nanoparticles, J. Appl. Phys. **119**[, 204104 \(2016\).](https://doi.org/10.1063/1.4952707)
- [11] E. A. Eliseev, Y. M. Fomichov, S. V. Kalinin, Y. M. Vysochanskii, P. Maksymovich, and A. N. Morozovska, Labyrinthine domains in ferroelectric nanoparticles: Manifesta[tion of a gradient-induced morphological phase transition,](https://doi.org/10.1103/PhysRevB.98.054101) Phys. Rev. B **98**, 054101 (2018).
- [12] A. N. Morozovska, Y. M. Fomichov, P. Maksymovych, Y. M. Vysochanskii, and E. A. Eliseev, Analytical description of domain morphology and phase diagrams of ferroelectric nanoparticles, Acta Mater. **160**[, 109 \(2018\).](https://doi.org/10.1016/j.actamat.2018.08.051)
- [13] A. N. Morozovska, E. A. Eliseev, R. Hertel, Y. M. Fomichov, V. Tulaidan, V. Yu. Reshetnyak, and D. R. Evans, Electric field control of three-dimensional vortex states in core-shell ferroelectric nanoparticles, Acta Mater. **200**[, 256 \(2020\).](https://doi.org/10.1016/j.actamat.2020.09.003)
- [14] A. N. Morozovska, E. A. Eliseev, Y. M. Fomichov, Y. M. Vysochanskii, V. Yu. Reshetnyak, and D. R. Evans, Controlling the domain structure of ferroelectric nanoparticles using tunable shells, [Acta Mater.](https://doi.org/10.1016/j.actamat.2019.11.012) **183**, 36 (2020).
- [15] A. N. Morozovska, E. A. Eliseev, and M. D. Glinchuk, Ferroelectricity enhancement in confined nanorods: Direct variational method, Phys. Rev. B. **73**[, 214106 \(2006\).](https://doi.org/10.1103/PhysRevB.73.214106)
- [16] A. N. Morozovska, M. D. Glinchuk, and E. A. Eliseev, Phase transitions induced by confinement of ferroic nanoparticles, Phys. Rev. B **76**[, 014102 \(2007\).](https://doi.org/10.1103/PhysRevB.76.014102)
- [17] I. S. Vorotiahin, E. A. Eliseev, Q. Li, S. V. Kalinin, Y. A. Genenko, and A. N. Morozovska, Tuning the polar states of [ferroelectric films via surface charges and flexoelectricity,](https://doi.org/10.1016/j.actamat.2017.07.033) Acta Mater. **137**, 85 (2017).
- [18] E. A. Eliseev, I. S. Vorotiahin, Y. M. Fomichov, M. D. Glinchuk, S. V. Kalinin, Y. A. Genenko, and A. N. Morozovska, Defect [driven flexo-chemical coupling in thin ferroelectric films,](https://doi.org/10.1103/PhysRevB.97.024102) Phys. Rev. B **97**, 024102 (2018).
- [19] G. Catalan, B. Noheda, J. McAneney, L. J. Sinnamon, and J. M. [Gregg, Strain gradients in epitaxial ferroelectrics,](https://doi.org/10.1103/PhysRevB.72.020102) Phys. Rev. B **72**, 020102(R) (2005).
- [20] M. S. Majdoub, P. Sharma, and T. Cagin, Enhanced sizedependent piezoelectricity and elasticity in nanostructures due to the flexoelectric effect, Phys. Rev B **77**[, 125424 \(2008\).](https://doi.org/10.1103/PhysRevB.77.125424)
- [21] F. Ahmadpoor and P. Sharma, Flexoelectricity in two[dimensional crystalline and biological membranes,](https://doi.org/10.1039/C5NR04722F) Nanoscale **7**, 16555 (2015).
- [22] E. A. Eliseev, A. N. Morozovska, M. D. Glinchuk, and R. Blinc, Spontaneous flexoelectric/flexomagnetic effect in nanoferroics, Phys. Rev. B **79**[, 165433 \(2009\).](https://doi.org/10.1103/PhysRevB.79.165433)
- [23] *Flexoelectricity in Solids: From Theory to Applications*, edited by A. K. Tagantsev and P. V. Yudin (World Scientific, New Jersey, 2016).
- [24] A. N. Morozovska, E. A. Eliseev, Y. A. Genenko, I. S. Vorotiahin, M. V. Silibin, Ye Cao, Y. Kim, M. D. Glinchuk, and S. V. Kalinin, Flexocoupling impact on the size effects of piezo- response and conductance in mixed-type ferroelectrics[semiconductors under applied pressure,](https://doi.org/10.1103/PhysRevB.94.174101) Phys. Rev. B **94**, 174101 (2016).
- [25] A. N. Morozovska, E. A. Eliseev, C. M. Scherbakov, and Y. M. Vysochanskii, The influence of elastic strain gradient on the upper limit of flexocoupling strength, spatially-modulated [phases and soft phonon dispersion in ferroics,](https://doi.org/10.1103/PhysRevB.94.174112) Phys. Rev. B **94**, 174112 (2016).
- [26] B. Wang, Y. Gu, S. Zhang, and L.-Q. Chen, Flexoelectricity in [solids: Progress, challenges, and perspectives,](https://doi.org/10.1016/j.pmatsci.2019.05.003) Prog. Mater Sci. **106**, 100570 (2019).
- [27] [A. F. Devonshire, Theory of ferroelectrics,](https://doi.org/10.1080/00018735400101173) Adv. Phys. **3**, 85 (1954).
- [28] A. Gruverman, M. Alexe, and D. Meier, Piezoresponse force [microscopy and nanoferroic phenomena,](https://doi.org/10.1038/s41467-019-09650-8) Nat. Commun. **10**, 1661 (2019).
- [29] J.-J. Wang, B. Wang, and L.-Q. Chen, Understanding, predicting, and designing ferroelectric domain structures and [switching guided by the phase-field method,](https://doi.org/10.1146/annurev-matsci-070218-121843) Ann. Rev. Mater. Res. **49**, 127 (2019).
- [30] R. G. P. McQuaid, L. J. McGilly, P. Sharma, A. Gruverman, and J. M. Gregg, Mesoscale flux-closure domain formation in single-crystal BaTiO₃, [Nat. Commun.](https://doi.org/10.1038/ncomms1413) **2**, 404 (2011).
- [31] Q. Zhang, L. Xie, G. Liu, S. Prokhorenko, Y. Nahas, X. Pan, L. Bellaiche, A. Gruverman, N. Valanoor, A. Gruverman, and N. Valanoor, Nanoscale bubble domains and topological tran[sitions in ultrathin ferroelectric films,](https://doi.org/10.1002/adma.201702375) Adv. Mater. **29**, 1702375 (2017).
- [32] E. A. Eliseev, A. N. Morozovska, C. T. Nelson, and S. V. Kalinin, Intrinsic structural instabilities of domain walls driven by gradient couplings: Meandering anferrodistortive[ferroelectric domain walls in BiFeO3,](https://doi.org/10.1103/PhysRevB.99.014112) Phys. Rev. B **99**, 014112 (2019).
- [33] M. J. Han, E. A. Eliseev, A. N. Morozovska, Y. L. Zhu, Y. L. Tang, Y. J. Wang, X. W. Guo, and X. L. Ma, Mapping gradientdriven morphological phase transition at the conductive domain [walls of strained multiferroic films,](https://doi.org/10.1103/PhysRevB.100.104109) Phys. Rev. B **100**, 104109 (2019).
- [34] Y. Nahas, S. Prokhorenko, J. Fischer, B. Xu, C. Carrétéro, S. Prosandeev, M. Bibes, S. Fusil, B. Dkhil, V. Garcia, and L. Bellaiche, Inverse transition of labyrinthine domain pat[terns in ferroelectric thin films,](https://doi.org/10.1038/s41586-019-1845-4) Nature (London) **577**, 47 (2020).
- [35] S. Cherifi-Hertel, H. Bulou, R. Hertel, G. Taupier, K. D. Dorkenoo, C. Andreas, J. Guyonnet, I. Gaponenko, K. Gallo,

and P. Paruch, Non-Ising and chiral ferroelectric domain walls [revealed by nonlinear optical microscopy,](https://doi.org/10.1038/ncomms15768) Nat. Commun. **8**, 15768 (2017).

- [36] B. J. Rodriguez, X. S. Gao, L. F. Liu, W. Lee, I. I. Naumov, A. M. Bratkovsky, D. Hesse, and M. Alexe, Vortex polariza[tion states in nanoscale ferroelectric arrays,](https://doi.org/10.1021/nl8036646) Nano Lett. **9**, 1127 (2009).
- [37] N. Balke, B. Winchester, W. Ren, Y. H. Chu, A. N. Morozovska, E. A. Eliseev, M. Huijben, R. K. Vasudevan, P. Maksymovych, J. Britson, S. Jesse, I. Kornev, R. Ramesh, L. Bellaiche, L. Q. Chen, and S. V. Kalinin, Enhanced electric conductivity at ferroelectric vortex cores in BiFeO3, Nat. Phys. **8**[, 81 \(2012\).](https://doi.org/10.1038/nphys2132)
- [38] A. K. Yadav, C. T. Nelson, S. L. Hsu, Z. Hong, J. D. Clarkson, C. M. Schlepütz, A. R. Damodaran, P. Shafer, E. Arenholz, L. R. Dedon, D. Chen, A. Vishwanath, A. M. Minor, L. Q. Chen, J. F. Scott, L. W. Martin, and R. Ramesh, Observation [of polar vortices in oxide superlattices,](https://doi.org/10.1038/nature16463) Nature (London) **530**, 198 (2016).
- [39] C. Liu, J. Wang, G. Xu, Ma. Kamlah, and T.-Y. Zhang, An isogeometric approach to flexoelectric effect in ferroelectric materials, [Int. J. Solids Struct.](https://doi.org/10.1016/j.ijsolstr.2018.12.008) **162**, 198 (2019).
- [40] Y. Li, K. Chu, C. Liu, P. Jiang, K. Qu, P. Gao, J. Wang, F. Ren, Q. Sun, L. Chen, and J. Li, Superelastic oxide micropillars enabled by surface tension–modulated 90 ° domain switching [with excellent fatigue resistance,](https://doi.org/10.1073/pnas.2025255118) Proc. Natl Acad. Sci. **118**, e2025255118 (2021).
- [41] S. Das, Y. L. Tang, Z. Hong, M. A. P. Gonçalves, M. R. McCarter, C. Klewe, K. X. Nguyen, F. Gómez-Ortiz, P. Shafer, E. Arenholz, V. A. Stoica, S.-L. Hsu, B. Wang, C. Ophus, J. F. Liu, C. T. Nelson, S. Saremi, B. Prasad, A. B. Mei, D. G. Schlom, J. Íñiguez, P. García-Fernández, D. A. Muller, L. Q. Chen, J. Junquera, L. W. Martin, and R. Ramesh, Observation of [room-temperature polar skyrmions,](https://doi.org/10.1038/s41586-019-1092-8) Nature (London) **568**, 368 (2019).
- [42] J. Hlinka, and P. Ondrejkovic, Skyrmions in ferroelectric materials, [Solid State Phys.](https://doi.org/10.1016/bs.ssp.2019.09.005) **70**, 143 (2019).
- [43] G. Finocchio, F. Büttner, R. Tomasello, M. Carpentieri, and M. Kläui, Magnetic skyrmions: From fundamental to applications, [J. Phys. D:Appl. Phys.](https://doi.org/10.1088/0022-3727/49/42/423001) **49**, 423001 (2016).
- [44] I. Dzyaloshinsky, A thermodynamic theory of "weak" ferro[magnetism of antiferromagnetics,](https://doi.org/10.1016/0022-3697(58)90076-3) J. Phys. Chem. Solids **4**, 241 (1958).
- [45] T. Moriya, Anisotropic superexchange interaction and weak ferromagnetism, Phys. Rev. **120**[, 91 \(1960\).](https://doi.org/10.1103/PhysRev.120.91)
- [46] H. J. Zhao, P. Chen, S. Prosandeev, S. Artyukhin, and L. Bellaiche, Dzyaloshinskii–Moriya-like interaction in ferroelectrics and antiferroelectrics, Nat. Mater. **20**[, 341 \(2021\).](https://doi.org/10.1038/s41563-020-00821-3)
- [47] K. C. Erb and J. Hlinka, Vector, bidirector, and Bloch skyrmion phases induced by structural crystallographic symmetry breaking, Phys. Rev. B **102**[, 024110 \(2020\).](https://doi.org/10.1103/PhysRevB.102.024110)
- [48] A. N. Bogdanov and D. A. Yablonskii, Thermodynamically stable "vortices" in magnetically ordered crystals, The mixed state of magnets, Sov. Phys. JETP **68**, 101 (1989).
- [49] A. Bogdanov, On the stability of localized states in nonlinear [field models with Lifshitz invariants,](https://doi.org/10.1134/1.567866) J. Exp. Theor. Phys. Lett. **68**, 317 (1998).
- [50] B. A. Strukov and A. P. Levanyuk, *Ferroelectric Phenomena in Crystals* (Springer-Verlag, Berlin, Heidelberg, 1998).
- [51] E. A. Eliseev, A. N. Morozovska, R. Hertel, H. V. Shevliakova, Y. M. Fomichov, V. Yu. Reshetnyak, and D. R. Evans, Flexo-

elastic control factors of domain morphology in core-shell [ferroelectric nanoparticles: Soft and rigid shells,](https://doi.org/10.1016/j.actamat.2021.116889) Acta Mater. **212**, 116889 (2021).

- [52] W. Ma, Surface tension and Curie temperature in ferroelectric nanowires and nanodots, [Appl. Phys. A](https://doi.org/10.1007/s00339-009-5246-7) **96**, 915 (2009).
- [53] J. J. Wang, E. A. Eliseev, X. Q. Ma, P. P. Wu, A. N. Morozovska, and Long-Qing Chen, Strain effect on phase transitions of BaTiO₃ nanowires, Acta Mater. **59**[, 7189 \(2011\).](https://doi.org/10.1016/j.actamat.2011.08.015)
- [54] J. J. Wang, X. Q. Ma, Q. Li, J. Britson, and Long-Qing Chen, Phase transitions and domain structures of ferroelectric nanoparticles: Phase field model incorporating strong elastic and dielectric inhomogeneity, Acta Mater. **61**[, 7591 \(2013\).](https://doi.org/10.1016/j.actamat.2013.08.055)
- [55] H. Huang, C. Q. Sun, and P. Hing, Surface bond contraction [and its effect on the nanometric sized lead zirconate titanate,](https://doi.org/10.1088/0953-8984/12/6/111) J. Phys.: Condens. Matter **12**, L127 (2000).
- [56] H. Huang, C. Q. Sun, Z. Tianshu, and P. Hing, Grain-size effect on ferroelectric Pb($Zr_{1-x}Ti_x$)O₃ solid solutions induced by surface bond contraction, Phys. Rev. B **63**[, 184112 \(2001\).](https://doi.org/10.1103/PhysRevB.63.184112)
- [57] L. D. Landau and I. M. Khalatnikov, On the anomalous absorption of sound near a second order phase transition point, Dokl. Akad. Nauk SSSR **96**, 469 (1954).
- [58] J. Hlinka, Mobility of ferroelectric domain walls in barium titanate, [Ferroelectrics](https://doi.org/10.1080/00150190701260694) **349**, 49 (2007).
- [59] S. P. Timoshenko and J. N. Goodier, *Theory of Elasticity* (McGraw-Hill, New York, 1970).
- [60] A. K. Tagantsev, G. Gerra, and N. Setter, Short-range and long-range contributions to the size effect in metal-ferroelectricmetal heterostructures, Phys. Rev. B **77**[, 174111 \(2008\).](https://doi.org/10.1103/PhysRevB.77.174111)
- [61] M. Stengel and N. A. Spaldin, Origin of the dielectric dead layer in nanoscale capacitors, [Nature \(London\)](https://doi.org/10.1038/nature05148) **443**, 679 (2006).
- [62] M. Stengel, D. Vanderbilt, and N. A. Spaldin, First-principles modeling of ferroelectric capacitors via constrained displacement field calculations, Phys. Rev. B **80**[, 224110 \(2009\).](https://doi.org/10.1103/PhysRevB.80.224110)
- [63] See Supplemental Material at [http://link.aps.org/supplemental/](http://link.aps.org/supplemental/10.1103/PhysRevB.104.054118) 10.1103/PhysRevB.104.054118 for calculation details.
- [64] J. Hlinka and P. Márton, Phenomenological model of 90-degree domain wall in BaTiO₃ type ferroelectrics, Phys. Rev. B 74, 104104 (2006).
- [65] A. J. Bell, Phenomenologically derived electric fieldtemperature phase diagrams and piezoelectric coefficients for single crystal barium titanate under fields along different axes, [J. Appl. Phys.](https://doi.org/10.1063/1.1352682) **89**, 3907 (2001).
- [66] D. Berlincourt and H. Jaffe, Elastic and piezoelectric coeffi[cients of single-crystal barium titanate,](https://doi.org/10.1103/PhysRev.111.143) Phys. Rev. **111**, 143 (1958).
- [67] I. Ponomareva, A. K. Tagantsev, and L. Bellaiche, Finitetemperature flexoelectricity in ferroelectric thin films from first principles, Phys. Rev. B **85**[, 104101 \(2012\).](https://doi.org/10.1103/PhysRevB.85.104101)
- [68] G. Shirane and A. Takeda, Transition energy and volume [change at three transitions in barium titanate,](https://doi.org/10.1143/JPSJ.7.1) J. Phys. Soc. Jap. **7**, 1 (1952).
- [69] H. D. Megaw, Temperature changes in the crystal structure of barium titanium oxide, [Proc. R. Soc. London, Ser. A](https://doi.org/10.1098/rspa.1947.0038) **189**, 261 (1947).
- [70] K. Everschor-Sitte, J. Masell, R. M. Reeve, and M. Kläui, Perspective: Magnetic skyrmions—Overview of recent progress in an active research field, J. Appl. Phys. **124**[, 240901 \(2018\).](https://doi.org/10.1063/1.5048972)
- [71] F. N. Rybakov, A. B. Borisov, S. Blügel, and N. S. Kiselev, New [Type of Stable Particle-Like States in Chiral Magnets,](https://doi.org/10.1103/PhysRevLett.115.117201) Phys. Rev. Lett. **115**, 117201 (2015).
- [72] S. A. Pathak and R. Hertel, Three-dimensional chiral mag[netization structures in FeGe nanospheres,](https://doi.org/10.1103/PhysRevB.103.104414) Phys. Rev. B **103**, 104414 (2021).
- [73] H. B. Braun, Nucleation in ferromagnetic nanowires [magnetostatics and topology,](https://doi.org/10.1063/1.370034) J. Appl. Phys. **85**, 6172 (1999).
- [74] R. Hertel and J. Kirschner, Magnetic drops in a soft-magnetic cylinder, [J. Magn. Magn. Mater.](https://doi.org/10.1016/j.jmmm.2004.02.032) **278**, L291 (2004).
- [75] P. V. Yudin, A. K. Tagantsev, E. A. Eliseev, A. N. Morozovska, and N. Setter, Bichiral structure of ferroelectric domain walls driven by flexoelectricity, Phys. Rev. B **86**[, 134102 \(2012\).](https://doi.org/10.1103/PhysRevB.86.134102)
- [76] E. A. Eliseev, P. V. Yudin, S. V. Kalinin, N. Setter, A. K. Tagantsev, and A. N. Morozovska, Structural phase transitions and electronic phenomena at 180-degree domain walls in rhombohedral BaTiO3, Phys. Rev. B **87**[, 054111 \(2013\).](https://doi.org/10.1103/PhysRevB.87.054111)
- [77] Y. Gu, M. Li, A. N. Morozovska, Yi Wang, E. A. Eliseev, V. Gopalan, and L.-Q. Chen, Non-ising character of a ferroelectric [wall arises from a flexoelectric effect,](https://doi.org/10.1103/PhysRevB.89.174111) Phys. Rev. B **89**, 174111 (2014).
- [78] We define an achiral structure as one that is equal to its mirror image, and a chiral structure as one that displays a handedness such that it is distinctly different from its mirror image. Systems governed by chiral symmetry-breaking interactions, such as the DMI, favor the formation of chiral structures with one specific handedness. Those systems are sometimes called homochiral.
- [79] J. Roller, A. Laganapan, J.-M. Meijer, M. Fuchs, and A. Zumbusch, Observation of liquid glass in suspensions of ellipsoidal colloids, [Proc. Natl. Acad. Sci.](https://doi.org/10.1073/pnas.2018072118) **118**, e2018072118 (2021).
- [80] S. V. Kalinin, E. Karapetian, and M. Kachanov, Nanoelectrome[chanics of piezoresponse force microscopy,](https://doi.org/10.1103/PhysRevB.70.184101) Phys. Rev. B **70**, 184101 (2004).
- [81] D. A. Scrymgeour and V. Gopalan, Nanoscale piezoelectric response across a single antiparallel ferroelectric domain wall, Phys. Rev. B **72**[, 024103 \(2005\).](https://doi.org/10.1103/PhysRevB.72.024103)
- [82] S. Cherifi-Hertel, C. Voulot, U. Acevedo-Salas, Y. Zhang, O. Crégut, K. D. Dorkenoo, and R. Hertel, Shedding light on non-Ising polar domain walls: Insight from second harmonic [generation microscopy and polarimetry analysis,](https://doi.org/10.1063/5.0037286) J. Appl. Phys. **129**, 081101 (2021).
- [83] G. Bautista, J. Mäkitalo, Ya Chen, V. Dhaka, M. Grasso, L. Karvonen, H. Jiang *et al.*, Second-harmonic generation imaging [of semiconductor nanowires with focused vector beams,](https://doi.org/10.1021/nl503984b) Nano Lett. **15**, 1564 (2015).
- [84] M. Timofeeva, A. Bouravleuv, G. Cirlin, I. Shtrom, I. Soshnikov, M. Reig Escalé, A. Sergeyev, and R. Grange, Polar second-harmonic imaging to resolve pure and mixed crystal phases along GaAs nanowires, Nano Lett. **16**[, 6290 \(2016\).](https://doi.org/10.1021/acs.nanolett.6b02592)
- [85] P. Shafer, P. García-Fernández, P. Aguado-Puente, A. R. Damodaran, A. K. Yadav, C. T. Nelson, S. L. Hsu, J. C. Wojdeł, J. Íñiguez, L. W. Martin, and E. Arenholz, Emergent chirality [in the electric polarization texture of titanate superlattices,](https://doi.org/10.1073/pnas.1711652115) Proc. Natl. Acad. Sci. **115**, 915 (2018).
- [86] S. W. Lovesey and G. van der Laan, Resonant x-ray diffraction [from chiral electric-polarization structures,](https://doi.org/10.1103/PhysRevB.98.155410) Phys. Rev. B **98**, 155410 (2018).
- [87] J. Y. Chauleau, T. Chirac, S. Fusil, V. Garcia, W. Akhtar, J. Tranchida, P. Thibaudeau, I. Gross, C. Blouzon, A. Finco, and M. Bibes, Electric and antiferromagnetic chiral textures at multiferroic domain walls, Nat. Mater. **19**[, 386 \(2020\).](https://doi.org/10.1038/s41563-019-0516-z)
- [88] Y. A. Izyumov, Modulated, or long-periodic, magnetic structures of crystals, [Sov. Phys. Usp.](https://doi.org/10.1070/PU1984v027n11ABEH004120) **27**, 845 (1984).
- [89] A. Sparavigna, A. Strigazzi, and A. Zvezdin, Electric-field ef[fects on the spin-density wave in magnetic ferroelectrics,](https://doi.org/10.1103/PhysRevB.50.2953) Phys. Rev. B **50**, 2953 (1994).
- [90] A. Sparavigna, Role of Lifshitz invariants in liquid crystals, Materials **2**[, 674 \(2009\).](https://doi.org/10.3390/ma2020674)
- [91] A. R. Damodaran, J. C. Agar, S. Pandya, Z. Chen, L. Dedon, R. Xu, B. Apgar, S. Saremi, and L. W. Martin, New modalities [of strain-control of ferroelectric thin films,](https://doi.org/10.1088/0953-8984/28/26/263001) J. Phys.: Condens. Matter **28**, 263001 (2016).