Magnetic exchange interactions in the van der Waals layered antiferromagnet MnPSe₃

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Two-dimensional van der Waals compounds with magnetic ions on a honeycomb lattice are hosts to a variety of exotic behavior. The magnetic interactions in one such compound, MnPSe₃, are investigated with elastic and inelastic neutron scattering. Magnetic excitations are observed in the magnetically ordered regime and persist to temperatures well above the ordering temperature, $T_{\rm N}=74~\rm K$, consistent with low dimensional magnetic interactions. The inelastic neutron scattering results allow a model spin Hamiltonian to be presented that includes dominant intralayer interactions of $J_{1ab}=0.45~\rm meV$, $J_{2ab}=0.03~\rm meV$, $J_{3ab}=0.19~\rm meV$, consistent with theoretical predictions. Despite the quasi-2D behavior, appreciable interlayer interactions of $J_c=0.031(5)~\rm meV$ are required to model the data. No evidence for anisotropy in the form of a spin gap is observed in the data collected. The measurements on MnPSe₃ are contrasted with those on MnPS₃ and reveal a large increase in the interlayer exchange interaction in MnPSe₃ that may stabilize the similar ordering temperatures in the bulk compounds.

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I. INTRODUCTION

Materials composed of two-dimensional (2D) layers, either in bulk compounds or reduced to a few or single layers, provide avenues for enhanced quantum phenomena. In this context 2D monolayer graphene, formed from the exfoliation of weakly connected van der Waals (vdW) bonded layers in graphite, has ignited widespread interest due to a variety of novel behavior [1-3]. General 2D material research has extended beyond graphene to the broader class of cleavable vdW materials [4]. A particular focus has been on materials with magnetic ions on 2D layers with predictions of topological and enhanced quantum behavior in the bulk compounds down to the monolayer. In the 2D atomic layer limit for Heisenberg spins the Mermin-Wagner theorem prohibits magnetic order [5], however, Onsager's theorem for Ising spins supports the stabilization of long range 2D order, which would apply to materials with strong crystalline anisotropy [6]. Therefore, an understanding of the exchange interactions and anisotropy provides indications on the stability and behavior of low dimensional magnetism.

The class of 2D vdW materials has begun expanding and includes $Fe_{3-x}GeTe_2$ [7–13], $CrXTe_3$ (X = Si,Ge,Sn) [14–17], $CrPS_4$ [18,19], $CrCl_3$ [20], and VI_3 [21,22]. Ferromagnetism in atomically thin 2D layers has been shown in CrI_3 and $CrGeTe_3$ [23–25]. One of the initial classes of 2D vdW materials is MPX_3 (M = Fe, Mn, Co, Ni and X = S, Se) [26–31]. These are characterized by 2D hexagonal layers of magnetic ions and host behavior including observation of antiferromagnetism in single layer $FePS_3$ [23] to ferrotoroidicity in MnPS₃ [32].

MnPSe₃ belongs to this class of magnetic 2D vdW materials; however, it has undergone limited research, particularly when compared to the sulfur analog MnPS₃. MnPSe₃ is a semiconductor with an optical energy gap of 2.27 eV measured from absorption data [33]. This is consistent with earlier reports from optical absorption of 2.5 eV [34]. MnPSe₃ shows potential for valleytronics, with a study of single layer MnPSe₃ on YMnO₃ [35]. MnPSe₃ has magnetic ions forming the same hexagonal motif as graphene with crystal space group $R\bar{3}$ (#148) [36]. Antiferromagnetic order of the Mn²⁺ ions $(3d^5)$, in the high-spin state S = 5/2, has been reported at 74 ± 2 K from neutron scattering [36]. The size of the magnetic and crystallographic unit cell are the same due to the propagation vector of $\mathbf{k} = (000)$. The reduced T_N compared to a Curie-Weiss temperature of -201 K [37] supports the low dimensional nature of the magnetism. Considering the reduction to monolayer theoretically the magnetic ground state of 2D MnPSe₃ remains antiferromagnetic with a predicted ordering temperature of 88 K, close to the value of the bulk system [38]. Experimental indications show that the magnetic ordering temperature remains similar from bulk to a few nanometer [39]. Considering the magnetic structure in Ref. [36] for bulk MnPSe₃ concludes from neutron diffraction that the spins are confined to the ab plane but cannot rule out a nonzero component along the c axis. A more recent study using powder diffraction also concluded the spins for MnPSe₃ are confined to the basal plane [40]. The moment direction has implications for an understanding of the anisotropic, Ising, XY, or 3D Heisenberg nature of the spins and any competition between single ion anisotropy and dipolar anisotropy.

Comparisons of MnPSe₃ and related MnPS₃ have led to intriguing findings which form a broader context for this study. Despite the analogous Mn²⁺ 2D lattice and similar magnetic ordering temperatures, $T_N(MnPSe_3) = 74$ K and

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 $T_{\rm N}({\rm MnPS_3}) = 78$ K, the magnetism is altered from predominantly isotropic in MnPS₃ to indications of unusually large XY anisotropy in MnPSe₃ along with a flipping of the spin direction [41]. The magnetic anisotropy in Ref. [41] of MnPSe₃ was proposed as a consequence of the zero-field splitting of the Mn-ion ground state (S = 5/2, L = 0) from a combination of spin-orbit coupling and the axial crystal field due to a trigonal distortion of the MnX_6 octahedra, where X = S or Se. The octahedral structural distortion, however, is reduced in MnPSe₃ compared to MnPS₃. Indeed, there is debate in the literature regarding the origin of the anisotropy in MnPS₃. A spin gap at the magnetic Brillouin zone center of 0.5 meV was measured using inelastic neutron scattering (INS) [42]. This was modeled within a framework of linear spin-wave theory with the introduction of an easy-axis term to account for the anisotropy, rather than a dipolar coupling term. It was noted that the calculated difference between the two routes was negligible for the data. One signature of the dipoledipole coupling is a splitting of the magnon excitation [43]. Indications of this were observed with the high resolution available of neutron spin echo [44]. Consequently, MnPS₃ is best described as hosting both single-ion anisotropy and dipolar coupling, with continued research aimed at further insights, for example through Raman studies [45].

Here we present powder neutron diffraction and INS measurements on MnPSe₃ to access the anisotropy and exchange interactions in this bulk vdW layered material. Magnetic excitations, consistent with spin waves, are observed in the magnetically ordered regime that dampen but persist above $T_{\rm N}$. Analysis of the data provides a model spin Hamiltonian that requires the inclusion of both intralayer interactions J_{1ab} , J_{2ab} , J_{3ab} and interlayer exchange interaction J_c . The data does not reveal any observable spin gap, indicating weak anisotropy. The results reveal that while the exchange interactions in MnPSe₃ are dominated by intralayer interactions, the appreciable interlayer (J_c) interactions contrast to that observed for MnPS₃, with an alteration of an order of magnitude.

II. EXPERIMENTAL DETAILS

A. Sample synthesis

Powdered samples of MnPSe₃ and MnPS₃ were synthesized through annealing of their constituent elements: Mn (99.95%, Alfa Aesar), P (99.999 + %, Alfa Aesar), and Se (99.999%, Alfa Aesar) for MnPSe₃, or S (99.9999%, Puratronic) for MnPS₃. For each sample, under argon atmosphere to avoid oxygen and moisture effects, stoichiometric amounts of the elements were measured out to equal 6 g and ground in an agate mortar and pestle until reaching a fine texture and uniform color. The fine powder was then loaded into a pellet press die and transferred to ambient atmosphere for pressing. Once compressed, the resulting pellet was sealed in a fused-quartz tube under a constant vacuum of 10^{-2} Torr, before being placed in a muffle furnace for annealing. To avoid overpressure of the tube due to volatilization of P, S, and Se, the samples were slowly heated to 730 °C over three days, then held at that temperature for one week, before allowing to cool to room temperature. The resulting fine powders of bright green MnPS₃ and wine red MnPSe₃ were verified for

correct crystal structure on an Empyrean x-ray diffractometer (Malvern Panalytical).

B. Neutron powder diffraction

Neutron powder diffraction was carried out on the HB-2A powder diffractometer at the High Flux Isotope Reactor (HFIR), Oak Ridge National Laboratory (ORNL) [46,47]. A vertically focusing germanium monochromator was used to select a wavelength of 1.12 Å from the Ge(117) reflection and 1.54 Å from the Ge(115) reflection. HB-2A has a Debye-Scherrer geometry and the diffraction pattern was collected by scanning a 120° bank of 44 ³He detectors in 0.05° steps to give 2θ coverage from 5 to 130° . Soller collimators of 12'are located before each detector. Measurements with a 1.12 Å wavelength were collected at 295 K, 100 K, 75 K, and 15 K for four hours. Measurements with a 1.54 Å wavelength were collected from 15 to 110 K in 5 K temperature steps for one hour. A 5 g sample of MnPSe₃ was contained in a vanadium sample holder with diameter 0.9 cm. The data was normalized using a vanadium measurement. The wavelengths and detector positions were calibrated with a Si measurement. The neutron data collected was reduced using Mantid [48].

C. Inelastic neutron scattering

INS measurements were performed on the time-of-flight direct geometry spectrometer Sequoia [49] at the Spallation Neutron Source, ORNL. Data were collected on MnPSe₃ and MnPS₃ powder samples. The samples were loaded into a 1 cm diameter cylindrical aluminium can and measured utilizing the three-sample changer in the closed cycle refrigerator (CCR). An identical empty aluminium can was measured and used to subtract the background from the sample environment. Data were collected with incident energies of $E_i = 20 \text{ meV}$ and $E_i = 8$ meV. Both of these energies were in high resolution mode. For $E_i = 20$ meV the Fermi chopper frequency was 240 Hz with a T₀ chopper frequency of 60 Hz giving an energy resolution at the elastic line of 0.48 meV. For E_i 8 meV the Fermi chopper frequency was 120 Hz with a T_0 chopper frequency of 30 Hz giving an energy resolution at the elastic line of 0.18 meV. The energy resolution was calculated using DGS resolution [50] and confirmed by taking cuts of the elastic line in the data. The momentum resolution Q was based on fitting the resolution of the Bragg peaks at the elastic line. Measurements were collected for six hours at 200, 100, 70, and 15 K. Shorter measurements of one hour from 22 K to 95 K in steps of 7 K were also collected with $E_i = 8$ meV. The neutron data collected were reduced using Mantid [48]. The data was corrected for the Bose thermal population factor, $[1 - \exp(-E/k_BT)]^{-1}$, where k_B is the Boltzmann factor. Data reduction and analysis utilized the DAVE software [51]. Modeling was performed with SpinW [52].

III. RESULTS AND DISCUSSION

A. Neutron powder diffraction measurements

Neutron powder diffraction measurements were performed on MnPSe₃ to probe the crystal and magnetic structure as a function of temperature, as shown in Figs. 1(a)-1(c).

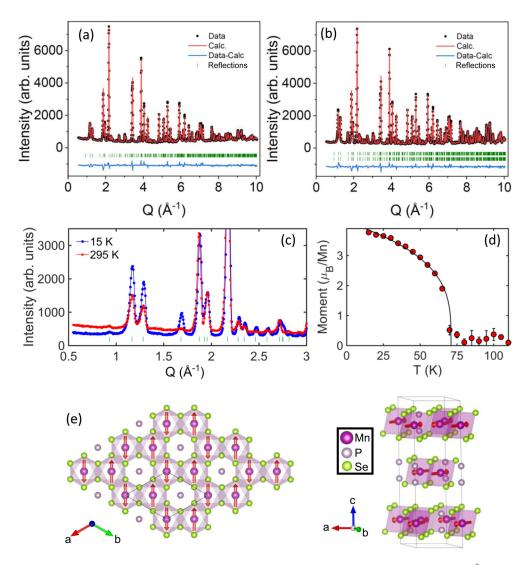


FIG. 1. Refinements of neutron powder diffraction measurements on MnPSe₃ with a wavelength of $\lambda = 1.12$ Å at (a) 295 K with the $R\bar{3}$ and (b) 15 K with $R\bar{3}$ space group (top ticks) and Γ_4 IR (bottom ticks). The P-1' magnetic space group refinement gave identical results. (c) Comparison of the data collected at 15 K and 295 K at low Q to reveal the magnetic reflections. Tick marks indicate allowed overlapping reflections. (d) Refined Mn²⁺ magnetic moment at 15 K to 110 K in 5 K temperature steps through the magnetic ordering transition. The solid line is a fit to a power law. (e) Crystal and magnetic structure of MnPSe₃ produced using VESTA [53]. The unit cell is indicated by the solid lines.

To initially investigate the crystal symmetry a short wavelength of 1.12 Å was used to give a large Q coverage up to 10 Å^{-1} . This allowed detailed measurements of the lattice, atomic positions, and thermal parameters with Rietveld refinement, which was performed using the Fullprof software [54]. The refinements did not require the inclusion of any preferred orientation parameter. No indication of any structural symmetry lowering from the $R\bar{3}$ (#148) was found with these measurements, or with the measurements using the longer wavelength of 1.54 Å. MnPSe₃ consists of Mn layers stacked in an ABC stacking sequence separated by 6.65 Å at 15 K. The Se layers are stacked in an ABAB sequence. The trigonally distorted MnSe₆ octahedra have Mn-Se bond distances of 2.757(8) Å and 2.721(7) Å.

We turn now to the magnetic structure, which is evidenced by additional intensity in Fig. 1(c) where the reflections correspond to both magnetic and nuclear. As previously reported

measurements indicated, the spins are in the ab plane with a propagation vector $\mathbf{k} = (000)$ [36,40]. We consider this further based on symmetry arguments using two complimentary approaches: representational analysis and magnetic space groups. The representational analysis approach was performed using the SARAh software [55]. This assumes the high temperature paramagnetic space group remains unaltered in the magnetic phase. This is consistent with the refinements of the crystal structure at all temperatures measured, see Fig. 1(a). Under this case the irreducible representations (IRs) contain magnetic structures with spins either confined to the ab plane or magnetic structures with spins along the c axis, see Table I. To allow for a magnetic structure with a component of the spin out of either the basal plane or c axis requires the combination of more than one IR. This goes against Landau theory which states that for a continuous transition only one IR is required. The IR Γ_4 was used to determine the magnetic

TABLE I. Basis vectors for the space group $R\bar{3}$:H with $\mathbf{k}=(000)$ in Kovalev's notation. The decomposition of the magnetic representation for the Mn site (0, 0, 0.16817) is $\Gamma_{\text{Mag}} = 1\Gamma_1^1 + 1\Gamma_2^1 + 1\Gamma_3^1 + 1\Gamma_4^1 + 1\Gamma_5^1 + 1\Gamma_6^1$. The atoms of the nonprimitive basis are defined according to 1: (0, 0, 0.16817), 2: (0, 0, 0.83183).

IR	BV	Atom	BV components		
			$\overline{m_{\parallel a}}$	$m_{\parallel b}$	$m_{\parallel c}$
Γ_1	ψ_1	1	0	0	1
		2	0	0	1
Γ_2	ψ_2	1	0	0	1
		2	0	0	-1
Γ_3	ψ_3	1	1	1	0
		2	1	1	0
Γ_4	ψ_4	1	1	1	0
	, .	2	-1	-1	0
Γ_5	ψ_5	1	1	-1	0
	, 3	2	1	-1	0
Γ_6	ψ_6	1	1	-1	0
Ŭ	, 0	2	-1	1	0

structure in this approach. Conversely, using the magnetic space group approach we find no maximally allowed magnetic space group compatible with the data. The only maximally allowed magnetic space groups found with the Bilbao Crystallographic Server [56] have spins only along the c axis. Therefore we were required to go to a further lower symmetry subgroup to find the magnetic space group of P-1' (#2.6), in BNS notation. Within this magnetic space group the spins are symmetry allowed to have nonzero and different magnetic components along the a, b, and c axis. Results from both the IR and magnetic space group approaches were refined against the data, shown in Fig. 1(b). No distinction was observed and therefore we follow the highest symmetry approach since no breaking of the a = b symmetry was measured. The magnetic structure therefore confines the spins to the ab plane with arbitrary direction within the plane. The refined magnetic moment was determined to be 3.83(3) μ_B/Mn^{2+} . We note that the P-1' magnetic space group allows both a definite spin direction within the ab plane as well as a component along the c axis, moreover this space group supports magnetoelectric coupling. Further measurements will be of interest to probe for symmetry breaking within the layer to provide more insight into the magnetic structure.

To follow the evolution of the magnetic ordering, the magnetic and crystallographic scattering were refined from measurements in 5 K steps from 15 to 110 K, shown in Fig. 1(d). No anomalies of the crystal lattice were observed within this temperature range. The temperature dependence of the ordered Mn^{2+} moment was fit to a power law, shown in Fig. 1(d). The onset of long range order is consistent with the reported transition temperature of 74 K. We note that there exists diffuse magnetic intensity above T_N that is commonly observed due to short range correlations and is evidenced in the inelastic data below. The best fit critical exponent was $2\beta = 0.14(1)$. This is consistent with 2D ordering, as has been observed in related bulk vdW compounds [57].

The neutron powder diffraction data therefore further supports the conclusion that MnPSe₃ is a quasi-2D vdW material. We note, however, that while there is evidence of diffuse magnetic scattering above T_N , which is excepted for low dimensional systems with strong interactions in the layers, the magnetic Bragg peak width in the ordered regime shows no strong broadening compared to the nuclear reflections. In particular the peaks do not show a strong Warren type peak shape associated with 2D ordering and instead indicate that once MnPSe₃ orders the intralayer and interlayer correlation lengths are comparable. This may indicate that the magnetic structure can be tuned with layer reduction, however recent studies suggest a limited impact on the number of layers to the onset of magnetic ordering temperature [39]. To further understand this behavior and gain quantitative insights we consider the exchange interactions with inelastic neutron scattering.

B. Inelastic neutron scattering measurements

To access the magnetic exchange interactions and anisotropy, inelastic neutron scattering measurements were performed on MnPSe₃ at various temperatures through the $T_{\rm N}=74$ K magnetic ordering transition, see Fig. 2. Complimentary data, shown in Fig. 3, at the same incident energy and temperature conditions were collected for MnPS₃, $T_{\rm N}=78$ K, to allow for a direct comparison of the data between MnPSe₃ and MnPS₃. The MnPS₃ scattering within the magnetically ordered phase, as expected, is consistent with the single crystal data in the literature [42]. The data presented here allows the temperature dependence to be observed.

The data shown in Fig. 2 for the inelastic scattering in MnPSe₃ have temperature and Q dependence behavior consistent with magnetic scattering. The well defined excitations within the magnetically ordered regime broaden but remain robust at 100 K and even persist to 200 K, the highest temperatures measured. Note the scale is the same in Figs. 2(a)-2(d), but altered in Fig. 2(e) to highlight the scattering at 200 K. This temperature behavior indicates low dimensional exchange interactions with the presence of in-plane correlations to high temperatures This provides further support for the quasi-2D behavior in the bulk compound and points to the potential for magnetism in a free standing layer. Contrasting the INS data for MnPSe₃ with MnPS₃ shows a reduced energy scale with a maximum energy of 8.5 meV for MnPSe₃ compared to 11.5 meV for MnPS₃, despite the similar ordering temperatures. Nevertheless, similar temperature behavior is observed for MnPS₃ and MnPSe₃, with well defined magnetic excitations dampening, but remaining present above the magnetic transition. To follow the low energy temperature dependence through T_N in Fig. 2(h) and Fig. 3(h) the Bose corrected intensity in a fixed Q and energy transfer window are plotted. The $T < T_N$ results are fit to $\chi''(T) \propto (-\Delta/k_B T)$. A pronounced change in the spectral weight occurs at the expected ordering temperatures consistent with $T_N(MnPSe_3) =$ 74 K and $T_N(MnPS_3) = 78 \text{ K}.$

Focusing on low energy and inspecting the data at the magnetic Bragg positions ($Q = 1.2\text{--}1.3 \,\text{Å}^{-1}$) reveals an intriguing difference in the spectra of MnPSe₃ compared to MnPS₃. The MnPS₃ inelastic scattering signal extends down to $E = 0.5 \,\text{meV}$, see Figs. 3(a) and 3(f) and Ref. [42]. For MnPSe₃

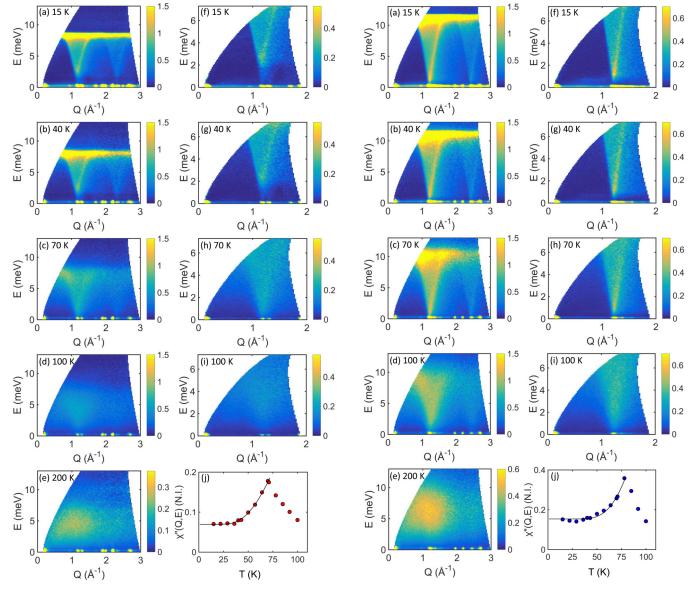


FIG. 2. INS data for MnPSe₃. $E_{\rm i}=20$ meV Bose factor corrected data at (a) 15 K, (b) 40 K, (c) 70 K, (d) 100 K, and (e) 200 K. The scale has been altered for clarity of the weaker intensity at 200 K. $E_{\rm i}=8$ meV Bose factor corrected data at (f) 15 K, (g) 40 K, (h) 70 K, (i) 100 K. (j) Bose corrected intensity in region E=0.3-0.8 meV and Q=1.135-1.235 Å⁻¹ to follow the evolution of the low energy scattering around the magnetic Bragg point.

the low energy magnetic scattering is qualitatively different, with strong scattering cutting off below 2 meV. This can be seen most clearly in the $S(Q,\omega)$ scattering in Fig. 4(a) and constant momentum Fig. 4(b) and constant energy Fig. 4(c) cuts taken at low energy in MnPSe₃. While it is tempting to assign this behavior to a spin gap due to increased anisotropy, as predicted from magnetic susceptibility measurements [41], closer inspection reveals weak scattering dispersing from the magnetic Bragg points at all energies. Therefore any spin gap is smaller than the energy resolution. Instead, as we show below when the data are modeled, the magnetic excitation spectra indicates an alteration of the interlayer interactions

FIG. 3. INS data for MnPS₃. $E_i = 20$ meV at (a) 15 K, (b) 40 K, (c) 70 K, (d) 100 K, and (e) 200 K. The scale has been altered for clarity of the weaker intensity at 200 K. $E_i = 8$ meV at (f) 15 K, (g) 40 K, (h) 70 K, (i) 100 K. (j) Bose corrected intensity in region E = 0.3–0.8 meV and Q = 1.17–1.27 Å⁻¹ to follow the evolution of the low energy scattering around the magnetic Bragg point.

between MnPSe₃ and MnPS₃. Such an alteration could have implications on the ordering temperatures in the bulk compound as well as the possibilities for stabilizing single layer magnetism.

C. Magnetic exchange interactions

To gain a quantitative understanding of the magnetic excitations we turn to modeling the data with linear spin wave theory to extract exchange interactions and anisotropy. Inelastic neutron scattering on single crystals is the premier method to achieve this due to the well understood scattering cross section $S(Q, \omega)$ and access to four-dimensional energy and momentum space with Q = H, K, L. Indeed this

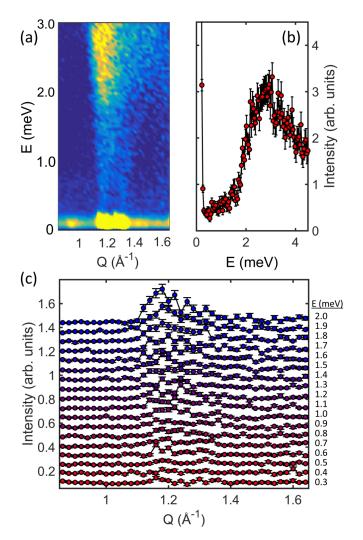


FIG. 4. (a) Low energy inelastic scattering of MnPSe₃ at the magnetic Bragg points with $E_i = 8$ meV at 15 K. (b) Inelastic intensity in the Q range 1.12 Å⁻¹ to 1.22 Å⁻¹. (c) Constant energy cuts with an energy step of 0.1 meV. The energies of each cut are indicated. The cuts have been offset by a constant factor.

was done on MnPS₃ and so the fitting is not repeated here. The reproduced results are shown in Fig. 5 to confirm the validity of the model for the MnPS₃ powder sample. However, we note that for low dimensional vdW compounds the hierarchy of intralayer and interlayer exchange interactions can allow the extraction of detailed Hamiltonians from the $S(|Q|, \omega)$ powder data, as has been shown in previous studies [19,28,58,59]. We therefore follow this approach to provide a spin Hamiltonian for MnPSe₃. Moreover, the high symmetry of MnPSe₃, as confirmed above with neutron powder diffraction, simplifies the required exchange parameter further and makes the development of a model spin Hamiltonian for MnPSe₃ more straightforward. Namely, MnPS₃ has a distorted hexagonal 2D layer which leads to further exchange interactions as compared to MnPSe3, which has identical bond lengths in the layer.

The inelastic measurements presented on MnPSe₃ revealed no spin gap (Fig. 4) and consequently we do not consider any anisotropy in the Hamiltonian. This is consistent with

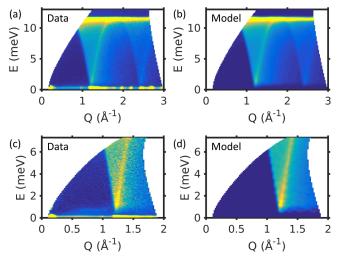


FIG. 5. INS data and spin wave model for MnPS₃ based on previously determined exchange interactions [42]. Data collected at 15 K with (a) an $E_i = 20$ meV and (b) corresponding model simulation. (c) Data collected at 15 K with an $E_i = 8$ meV and (d) corresponding model simulation.

the expected reduced single-ion anisotropy from the less distorted octahedral environment of the Mn ion. Moreover, the magnetic structure of spins confined to the *ab* plane indicates reduced anisotropy. Our initial model was based on a 2D Hamiltonian on a honeycomb lattice,

$$\mathcal{H} = \sum_{i \neq j} J_{ij} \mathbf{S}_i \cdot \mathbf{S}_j = \sum_{\langle ij \rangle} 2J_{ij} \mathbf{S}_i \cdot \mathbf{S}_j,$$

that included J_{1ab} , J_{2ab} , and J_{3ab} intralayer exchange interactions, as well as the full spin (S=5/2), shown in Fig. 6(a). Note the exchange interactions have been reported to be consistent with parameters in the literature and allow comparisons between MnPS₃ and MnPS₃ [42,60–62]. Alternative notation related by J/2 is used for MnPS₃ in Ref. [45].

We initially considered a 2D model of nearest neighbor intralayer interactions out to the third neighbor. However, this did not reproduce the observed INS scattering in MnPSe₃ at low energy. This was the case with or without single-ion anisotropy incorporated in the Hamiltonian. Therefore we extended the Hamiltonian to include interlayer exchange interactions (J_c) to give a 3D model isotropic Hamiltonian.

We begin with a confirmation that the previous reported results for MnPS₃ in Ref. [42] can be reproduced by the powder data. These measurements make the approximation of a higher symmetry in which the layers are undistorted with $J_1 = 0.77$ meV, $J_2 = 0.07$ meV, $J_3 = 0.18$ meV, and $J_c = -0.0038$ meV. An easy-axis single ion anisotropy term of 0.0086 meV is also included to model the observed spin gap in MnPS₃ [42]. Figure 5 shows the powder data are well reproduced using these parameters.

We now turn to modeling the exchange interactions in MnPSe₃ that have not been experimentally determined with INS. Exchange interactions based on a J_{1ab} - J_{2ab} - J_{3ab} 2D model without an interlayer J_c or anisotropy terms have been theoretically reported in several studies [60–63]. These

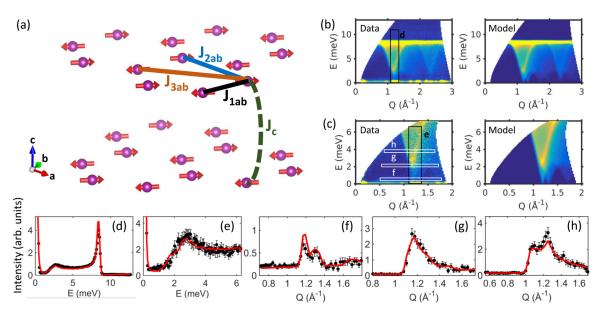


FIG. 6. (a) Exchange interactions between the Mn ions (purple spheres) in MnPSe₃ used in the spin Hamiltonian. Inelastic neutron scattering data and model based on the best fit exchange interaction parameters for (b) $E_i = 20$ meV and (c) $E_i = 8$ meV. All data was collected at 15 K. The boxes highlight the data cuts shown in the rest of the figure. Data (circles) and model fit (line) for (d) cut along energy with $E_i = 20$ meV in the range $1.12 \le Q$ (Å⁻¹) ≤ 1.22 . (e) Cut along energy with $E_i = 8$ meV in the range $1.12 \le Q$ (Å⁻¹) ≤ 1.22 . (f) Cut along Q with $E_i = 8$ meV in the range $0.5 \le E$ (meV) ≤ 0.8 . (g) Cut along Q with $E_i = 8$ meV in the range $0.5 \le E$ (meV) ≤ 0.8 . (g) Cut along Q with $E_i = 8$ meV in the range $0.5 \le E$ (meV) ≤ 0.8 . (h) Cut along Q with $E_i = 8$ meV in the range $0.5 \le E$ (meV) ≤ 0.8 . (c)

considered the required antiferromagnetic structure of MnPSe₃ on the 2D hexagonal lattice with the hierarchy of $J_1 > J_3 > J_2$ and all interactions positive (antiferromagnetic). The different studies show broad agreement. Therefore after initial confirmation of these parameters, the focus of this study will be on the interlayer interactions J_c .

Starting with the values reported in Refs. [60,61] gives very close agreement to the measured excitation energy range, within 1.5%. We therefore applied a scale factor and kept these values fixed in the subsequent analysis at J_{1ab} = 0.45 meV, $J_{2ab} = 0.03$ meV, and $J_{3ab} = 0.19$ meV. To model the data, we compared measured and calculated $S(|Q|, \omega)$, as well as constant energy and momentum cuts. Simulations were performed with SpinW [52] using 100 000 random orientations per Q value. The form factor of Mn^{2+} was included in the model. To rule out any single-ion anisotropy term being able to model the data, we applied this to a modified Hamiltonian; however, the results did not show agreement. Therefore we applied the J_c term and performed least squares fitting to find the best-fit J_c parameter to the data. The results are shown in Fig. 6. The measured INS data are best modeled with exchange interactions $J_{1ab} = 0.45 \text{ meV}$, $J_{2ab} = 0.03 \text{ meV}$, $J_{3ab} =$ 0.19 meV, and $J_c = 0.031(5)$ meV.

Comparing the exchange interactions we have determined for MnPSe₃ with those measured for MnPSe₃ reveals an enhanced interlayer interaction energy in MnPSe₃ of an order of magnitude greater than reported for MnPS₃ from INS [42]. This is coupled to a reduced energy scale for intralayer J_{1ab} - J_{2ab} - J_{3ab} interactions in MnPSe₃ compared to MnPS₃. While any anisotropy present in MnPSe₃ is beyond the limits of our measurement, adding single ion anisotropy to the spin Hamiltonian indicates that the energy is likely on the same scale or smaller than that measured in MnPS₃. This

suggests that interlayer coupling plays a more significant role than anisotropy in MnPSe₃ compared to MnPS₃, which could have implications on stabilizing single layers magnetism in MnPSe₃.

The extraction of a model spin Hamiltonian that best fits the INS data reveals MnPSe₃ as having dominant intralayer interactions with non-negligible interlayer interactions. Conversely, MnPS₃ is closer to the 2D limit in the bulk compound. This balance of competing interactions can explain the similar magnetic ordering temperatures, with lower intralayer exchange interactions in MnPSe₃ that drive a lower ordering temperature being offset by higher interlayer exchange interactions that stabilize long range three-dimensional order in the bulk. Further theoretical and experimental investigations of MnPSe₃ from the bulk down to the monolayer, where $J_c = 0$, will be of interest to follow the crossover from the weakly three-dimensional exchange interaction model to purely 2D in-plane interactions.

IV. CONCLUSIONS

The magnetic structure and interactions in the layered vdW material MnPSe₃ have been investigated in detail with neutron scattering on a powder sample. Neutron powder diffraction found no symmetry lowering from $R\bar{3}$ for the crystal structure at all temperatures measured. Consequently a magnetic model of spins confined to the *ab* plane is consistent with the data, as previously reported. However, a magnetic space group approach indicates a symmetry lowering to P-1', which allows for a unique definition of the in-plane spin direction as well as a nonzero spin component out of the plane. Within this magnetic space group magnetoelectric coupling is additionally supported. Further measurements will be of interest to

probe the magnetic ground state. The temperature evolution of the magnetic intensity further supported the 2D nature of the magnetic interactions. To provide quantitative information on the spin interactions, inelastic neutron measurements on MnPSe₃ revealed well-defined spin waves that allowed model magnetic exchange interactions to be extracted within linear spin wave theory. This was despite the measurements being on a powder sample that often makes extraction of quantitative information challenging or ambiguous. In low dimensional materials these hurdles are often lowered, as was found here for MnPSe₃. The scattering was contrasted with MnPS₃ and showed a clear departure in behavior at low energy. MnPSe₃ contained no observable spin gap, highlighting a reduced anisotropy in this material compared to MnPS₃, likely due to the less distorted octahedral environment of the Mn²⁺ ion. Despite the quasi-2D vdW nature of the material, inelastic excitations consistent with interlayer interactions were observed. While the intralayer J_{1ab} - J_{2ab} - J_{3ab} interactions dominate the magnetic behavior, extension beyond a 2D Hamiltonian was required with the inclusion of a non-negligible interlayer interaction $J_c = 0.031(5)$ meV to model the data. This interlayer interaction is significantly stronger than MnPS₃ indicating that in the bulk MnPSe₃ is further from the 2D limit than MnPS₃. This highlights the impact of tuning the nonmag-

netic ions can have on the observed behavior. Collectively the results show that MnPSe₃ falls into the class of quasi-2D magnetic vdW materials that are of great interest for novel applied and fundamental phenomena.

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- [1] K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, M. I. Katsnelson, I. V. Grigorieva, S. V. Dubonos, and A. A. Firsov, Nature (London) 438, 197 (2005).
- [2] Y. B. Zhang, Y. W. Tan, H. L. Stormer, and P. Kim, Nature (London) 438, 201 (2005).
- [3] R. Yu, W. Zhang, H.-J. Zhang, S.-C. Zhang, X. Dai, and Z. Fang, Science 329, 61 (2010).
- [4] G. R. Bhimanapati, Z. Lin, V. Meunier, Y. Jung, J. Cha, S. Das, D. Xiao, Y. Son, M. S. Strano, V. R. Cooper, L. Liang, S. G. Louie, E. Ringe, W. Zhou, S. S. Kim, R. R. Naik, B. G. Sumpter, H. Terrones, F. Xia, Y. Wang, J. Zhu, D. Akinwande, N. Alem, J. A. Schuller, R. E. Schaak, M. Terrones, and J. A. Robinson, ACS Nano 9, 11509 (2015).
- [5] N. D. Mermin and H. Wagner, Phys. Rev. Lett. 17, 1133 (1966).
- [6] L. Onsager, Phys. Rev. 65, 117 (1944).
- [7] H.-J. Deiseroth, K. Aleksandrov, C. Reiner, L. Kienle, and R. K. Kremer, Eur. J. Inorg. Chem. 2006, 1561 (2006).
- [8] A. F. May, S. Calder, C. Cantoni, H. Cao, and M. A. McGuire, Phys. Rev. B 93, 014411 (2016).
- [9] Z. Fei, B. Huang, P. Malinowski, W. Wang, T. Song, J. Sanchez, W. Yao, D. Xiao, X. Zhu, A. F. May, W. Wu, D. H. Cobden, J.-H. Chu, and X. Xu, Nat. Mater. 17, 778 (2018).
- [10] C. Tan, J. Lee, S.-G. Jung, T. Park, S. Albarakati, J. Partridge, M. R. Field, D. G. McCulloch, L. Wang, and C. Lee, Nat. Commun. 9, 1554 (2018).
- [11] O. Johansen, V. Risinggård, A. Sudbø, J. Linder, and A. Brataas, Phys. Rev. Lett. 122, 217203 (2019).
- [12] S. Calder, A. I. Kolesnikov, and A. F. May, Phys. Rev. B 99, 094423 (2019).
- [13] X. Wang, J. Tang, X. Xia, C. He, J. Zhang, Y. Liu, C. Wan, C. Fang, C. Guo, W. Yang, Y. Guang, X. Zhang, H. Xu, J. Wei, M. Liao, X. Lu, J. Feng, X. Li, Y. Peng, H. Wei, R. Yang, D. Shi,

- X. Zhang, Z. Han, Z. Zhang, G. Zhang, G. Yu, and X. Han, Sci. Adv. 5, eaaw8904 (2019).
- [14] V. Carteaux, F. Moussa, and M. Spiesser, Europhys. Lett. 29, 251 (1995).
- [15] T. J. Williams, A. A. Aczel, M. D. Lumsden, S. E. Nagler, M. B. Stone, J.-Q. Yan, and D. Mandrus, Phys. Rev. B 92, 144404 (2015).
- [16] J. Zhang, X. Cai, W. Xia, A. Liang, J. Huang, C. Wang, L. Yang, H. Yuan, Y. Chen, S. Zhang, Y. Guo, Z. Liu, and G. Li, Phys. Rev. Lett. 123, 047203 (2019).
- [17] N. Ito, T. Kikkawa, J. Barker, D. Hirobe, Y. Shiomi, and E. Saitoh, Phys. Rev. B 100, 060402(R) (2019).
- [18] A. Louisy, G. Ouvrard, D. Schleich, and R. Brec, Solid State Commun. 28, 61 (1978).
- [19] S. Calder, A. V. Haglund, Y. Liu, D. M. Pajerowski, H. B. Cao, T. J. Williams, V. O. Garlea, and D. Mandrus, Phys. Rev. B 102, 024408 (2020).
- [20] M. A. McGuire, G. Clark, S. KC, W. M. Chance, G. E. Jellison, Jr., V. R. Cooper, X. Xu, and B. C. Sales, Phys. Rev. Materials 1, 014001 (2017).
- [21] J. Yan, X. Luo, F. C. Chen, J. J. Gao, Z. Z. Jiang, G. C. Zhao, Y. Sun, H. Y. Lv, S. J. Tian, Q. W. Yin, H. C. Lei, W. J. Lu, P. Tong, W. H. Song, X. B. Zhu, and Y. P. Sun, Phys. Rev. B 100, 094402 (2019).
- [22] S. Son, M. J. Coak, N. Lee, J. Kim, T. Y. Kim, H. Hamidov, H. Cho, C. Liu, D. M. Jarvis, P. A. C. Brown, J. H. Kim, C.-H. Park, D. I. Khomskii, S. S. Saxena, and J.-G. Park, Phys. Rev. B 99, 041402(R) (2019).
- [23] J.-U. Lee, S. Lee, J. H. Ryoo, S. Kang, T. Y. Kim, P. Kim, C.-H. Park, J.-G. Park, and H. Cheong, Nano Lett. 16, 7433 (2016).
- [24] B. Huang, G. Clark, E. Navarro-Moratalla, D. R. Klein, R. Cheng, K. L. Seyler, D. Zhong, E. Schmidgall, M. A. McGuire,

- D. H. Cobden, W. Yao, D. Xiao, P. Jarillo-Herrero, and X. Xu, Nature (London) **546**, 270 (2017).
- [25] C. Gong, L. Li, Z. Li, H. Ji, A. Stern, Y. Xia, T. Cao, W. Bao, C. Wang, Y. Wang, Z. Q. Qiu, R. J. Cava, S. G. Louie, J. Xia, and X. Zhang, Nature (London) 546, 265 (2017).
- [26] P. A. Joy and S. Vasudevan, Phys. Rev. B 46, 5425 (1992).
- [27] K. C. Rule, A. R. Wildes, R. I. Bewley, D. Visser, and T. J. Hicks, J. Phys.: Condens. Matter 21, 124214 (2009).
- [28] A. R. Wildes, K. C. Rule, R. I. Bewley, M. Enderle, and T. J. Hicks, J. Phys.: Condens. Matter 24, 416004 (2012).
- [29] K. C. Rule, G. J. McIntyre, S. J. Kennedy, and T. J. Hicks, Phys. Rev. B 76, 134402 (2007).
- [30] A. R. Wildes, V. Simonet, E. Ressouche, R. Ballou, and G. J. McIntyre, J. Phys.: Condens. Matter 29, 455801 (2017).
- [31] C. R. S. Haines, M. J. Coak, A. R. Wildes, G. I. Lampronti, C. Liu, P. Nahai-Williamson, H. Hamidov, D. Daisenberger, and S. S. Saxena, Phys. Rev. Lett. 121, 266801 (2018).
- [32] E. Ressouche, M. Loire, V. Simonet, R. Ballou, A. Stunault, and A. Wildes, Phys. Rev. B 82, 100408(R) (2010).
- [33] V. Grasso and L. Silipigni, J. Opt. Soc. Am. B 16, 132 (1999).
- [34] R. Brec, D. M. Schleich, G. Ouvrard, A. Louisy, and J. Rouxel, Inorg. Chem. 18, 1814 (1979).
- [35] Q. Pei and W. Mi, Phys. Rev. Appl. 11, 014011 (2019).
- [36] A. Wiedenmann, J. Rossat-Mignod, A. Louisy, R. Brec, and J. Rouxel, Solid State Commun. 40, 1067 (1981).
- [37] G. Le Flem, R. Brec, G. Ouvard, A. Louisy, and P. Segransan, J. Phys. Chem. Solids 43, 455 (1982).
- [38] X. Li, X. Wu, and J. Yang, J. Am. Chem. Soc. 136, 11065 (2014).
- [39] P. Liu, Z. Xu, H. Huang, J. Li, C. Feng, M. Huang, M. Zhu, Z. Wang, Z. Zhang, D. Hou, Y. Lu, and B. Xiang, J. Alloys Compd. 828, 154432 (2020).
- [40] A. Bhutani, J. L. Zuo, R. D. McAuliffe, C. R. dela Cruz, and D. P. Shoemaker, Phys. Rev. Materials 4, 034411 (2020).
- [41] P. Jeevanandam and S. Vasudevan, J. Phys.: Condens. Matter 11, 3563 (1999).
- [42] A. R. Wildes, B. Roessli, B. Lebech, and K. W. Godfrey, J. Phys.: Condens. Matter 10, 6417 (1998).
- [43] C. Pich and F. Schwabl, J. Magn. Magn. Mater. 148, 30 (1995).
- [44] T. Hicks, T. Keller, and A. Wildes, J. Magn. Magn. Mater. **474**, 512 (2019).
- [45] D. Vaclavkova, A. Delhomme, C. Faugeras, M. Potemski, A. Bogucki, J. Suffczyński, P. Kossacki, A. R. Wildes, B. Grémaud, and A. Saúl, 2D Mater. 7, 035030 (2020).

- [46] V. O. Garlea, B. C. Chakoumakos, S. A. Moore, G. B. Taylor, T. Chae, R. G. Maples, R. A. Riedel, G. W. Lynn, and D. L. Selby, Appl. Phys. A 99, 531 (2010).
- [47] S. Calder, K. An, R. Boehler, C. R. Dela Cruz, M. D. Frontzek, M. Guthrie, B. Haberl, A. Huq, S. A. J. Kimber, J. Liu, J. J. Molaison, J. Neuefeind, K. Page, A. M. dos Santos, K. M. Taddei, C. Tulk, and M. G. Tucker, Rev. Sci. Instrum. 89, 092701 (2018).
- [48] O. Arnold, J. Bilheux, J. Borreguero, A. Buts, S. Campbell, L. Chapon, M. Doucet, N. Draper, R. F. Leal, M. Gigg, V. Lynch, A. Markvardsen, D. Mikkelson, R. Mikkelson, R. Miller, K. Palmen, P. Parker, G. Passos, T. Perring, P. Peterson, S. Ren, M. Reuter, A. Savici, J. Taylor, R. Taylor, R. Tolchenov, W. Zhou, and J. Zikovsky, Nucl. Instrum. Methods Phys. Res. 764, 156 (2014).
- [49] G. E. Granroth, A. I. Kolesnikov, T. E. Sherline, J. P. Clancy, K. A. Ross, J. P. C. Ruff, B. D. Gaulin, and S. E. Nagler, J. Phys.: Conf. Ser. 251, 012058 (2010).
- [50] rez.mcvine.ornl.gov
- [51] R. T. Azuah, L. R. Kneller, Y. Qiu, P. L. W. Tregenna-Piggott, C. M. Brown, J. R. D. Copley, and R. M. Dimeo, J. Res. Natl. Inst. Stand. Technol. 114, 341 (2009).
- [52] S. Toth and B. Lake, J. Phys.: Condens. Matter 27, 166002 (2015).
- [53] K. Momma and F. Izumi, J. Appl. Crystallogr. 44, 1272 (2011).
- [54] J. Rodríguez-Carvajal, Phys. B: Condens. Matter 192, 55 (1993).
- [55] A. Wills, Physica B 276-278, 680 (2000).
- [56] J. Perez-Mato, S. Gallego, E. Tasci, L. Elcoro, G. de la Flor, and M. Aroyo, Annu. Rev. Mater. Res. 45, 217 (2015).
- [57] A. R. Wildes, V. Simonet, E. Ressouche, G. J. McIntyre, M. Avdeev, E. Suard, S. A. J. Kimber, D. Lançon, G. Pepe, B. Moubaraki, and T. J. Hicks, Phys. Rev. B 92, 224408 (2015).
- [58] D. Lançon, R. A. Ewings, T. Guidi, F. Formisano, and A. R. Wildes, Phys. Rev. B 98, 134414 (2018).
- [59] C. Kim, J. Jeong, P. Park, T. Masuda, S. Asai, S. Itoh, H.-S. Kim, A. Wildes, and J.-G. Park, Phys. Rev. B 102, 184429 (2020).
- [60] N. Sivadas, M. W. Daniels, R. H. Swendsen, S. Okamoto, and D. Xiao, Phys. Rev. B 91, 235425 (2015).
- [61] J. Yang, Y. Zhou, Q. Guo, Y. Dedkov, and E. Voloshina, RSC Adv. 10, 851 (2020).
- [62] B. L. Chittari, Y. Park, D. Lee, M. Han, A. H. MacDonald, E. Hwang, and J. Jung, Phys. Rev. B 94, 184428 (2016).
- [63] Q. Pei, X.-C. Wang, J.-J. Zou, and W.-B. Mi, Front. Phys. 13 (2018).