Metamagnetic transitions and magnetoelectricity in the spin-1 honeycomb antiferromagnet Ni₂Mo₃O₈

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Earlier neutron scattering studies suggested the coexistence of complex stripelike and zigzaglike antiferromagnetic orders in polar honeycomb lattice $Ni₂Mo₃O₈$, while its magnetoelectric (ME) behavior as an emergent effect is thus of high interest. Here we report our observations of two metamagnetic transitions and novel ME responses of Ni2Mo3O8 single crystals against high magnetic field *H* up to ∼60 T. The *c*-axis (polar axis) spontaneous electric polarization P_{spin} , emerging at the magnetic Néel temperature $T_N \sim 5.5$ K, and its remarkable response to *H* applied along the *c* axis $(H//c)$ and *a* axis $(H//a)$, respectively, provide the clear evidence for the magnetism-driven ferroelectricity. While the magnetism exhibits the in-plane anisotropy to some extent, the magnetic field dependencies of magnetization and electric polarization in the low-field region and high-field region are distinctly different. In the low-field region where a weak spin-flop type metamagnetic transition occurs, the electric polarization response shows the parabolic dependence of magnetic field applied along both the *c* axis and in-plane *a* axis. The second metamagnetic transition happens when the magnetic field extends up to the high-field region where the magnetization and electric polarization response at low temperature are characterized by an extraordinarily broad plateau for the magnetic field along the *c* axis but roughly linear dependence for field along the *a* axis. These unusual phenomena are discussed, based on the symmetry-related local ME tensor analysis, and it is suggested that both the spin current and *p-d* hybridization mechanisms may contribute to the spontaneous electric polarization and ME responses. The present work demonstrates $Ni₂Mo₃O₈$ as a unique multiferroic and promising platform for exploring the rich spin-1 physics and ME phenomena in honeycomb lattice.

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I. INTRODUCTION

Multiferroics possessing more than one ferroic order parameter are of significant interest due to their potential application in novel electronic devices [\[1,2\]](#page-9-0). Among them, the fascinating interplay between the magnetization (*M*) or electric polarization (*P*) induced by electric field (*E*) or magnetic field (*H*), respectively, has attracted tremendous interest because of the strong magnetoelectric (ME) effect [\[3,4\]](#page-9-0). There are two types of multiferroics: in type-I multiferroics, the ferroelectricity and magnetism have different origins, and are largely independent of each other, whereas the magnetic order itself induces ferroelectricity in type-II multiferroics. Constrained by the symmetry requirements, very limited materials exhibiting strong ME coupling were found in the frustrated magnets. In particular, besides the cross coupling between ferroelectric and magnetic order, the observed topological phenomena associated with multiferroicity and magnetoelectric coupling has been intensively stimulated, such as the ME coupling in a skyrmion lattice [\[5\]](#page-9-0), topological defects in hexagonal $YMnO₃$ [\[6\]](#page-9-0), magnetoelectric monopole [\[7\]](#page-9-0), the toroidal moment $[8]$, and optical diode effect $[9,10]$.

Along with continuing efforts to realize colossalization of the ME signal, as well as the subsequent phenomena associated with multiferroicity, some ME materials exhibiting novel metapolarization transition response along with the occurrence of metamagnetic transition (e.g., spin-flop, magnetization plateau) in the trigonal or kagome lattice, such as CuFeO₂ [\[11\]](#page-9-0), Ni₃V₂O₈ [\[12\]](#page-9-0), Ni₃TeO₆ [\[13\]](#page-9-0), and PbCu₃TeO₇ [\[14\]](#page-9-0), have strongly received the interest in this topic. It has been suggested that the strong next-nearest-neighbor and next-next-nearest-neighbor interactions may play important roles in promoting the formation of exotic magnetic orders [\[15–17\]](#page-9-0). One candidate of such ME materials is magnetic nickelates in spin-1 systems, e.g., A_3 Ni₂XO₆ ($A = Li$, Na and $X = Sb$, Bi) [\[15\]](#page-9-0), Na₂Ni₂TeO₆ [\[17\]](#page-9-0), and BaNi₂V₂O₈ [\[18\]](#page-9-0), in which various types of magnetically ordered states are established. Nonetheless, quite a few concerning the research about the ME coupling in the honeycomb-based materials with fascinating physics has been investigated [\[19,20\]](#page-10-0).

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FIG. 1. (a) The schematic diagram of magnetic structure in $Ni₂Mo₃O₈$ with magnetic space group $P_Cna2₁$. The spins of the Ni₁ and $Ni₂$ sites are expressed in the magenta and green colors. (b) The crystal structure of $Ni₂Mo₃O₈$ with polar space group $P6₃mc$. (c) The XRD pattern of as-grown single crystal, and the insets show the image, Laue spots, and chemical component analysis of the crystals. (d) The XRD pattern of the $Ni₂Mo₃O₈$ powders at room temperature and the data from the corresponding Rietveld fitting.

Most recently, Morey and McQueen *et al.* reported more complex AFM order on a honeycomb lattice with a mixture of stripy and zigzag characters in polycrystalline $Ni₂Mo₃O₈$, as shown in Fig. $1(a)$ [\[21\]](#page-10-0), which is essentially different from the trigonal or kagome lattice mentioned above. Compared to the other compounds in this $M_2M_0_3O_8$ ($M = \text{Fe}$, Mn, Co) family, $Ni₂Mo₃O₈$ looks different in terms of magnetic structure [\[22\]](#page-10-0). First, $Ni₂Mo₃O₈$ has an orthogonal magnetic structure (magnetic space group P_C *na* 2_1) in the ground state, while other members exhibit hexagonal magnetic structure. Second, it is suggested that the zigzag AFM order in $Ni₂Mo₃O₈$ may be stabilized by the bond-dependent anisotropic exchange due to ligand distortion, which usually appears in the 4*d* and 5*d* ions with strong anisotropy and spin-orbit coupling [\[15\]](#page-9-0). Third, such complex magnetic structure indicates the stronger

second- and third-neighbor interactions in $Ni₂Mo₃O₈$ [\[23\]](#page-10-0), similar to the honeycomb compounds α -RuCl₃, Li₂IrO₃, and $Na₂IrO₃$ that allow a Kitaev quantum spin liquid state to emerge [\[24–26\]](#page-10-0).

In addition to these fundamental aspects, the most intriguing phenomena found in $M_2M_0^3O_8$ recently include the prominent magnetoelectric (ME) coupling between the magnetic and ferroelectric orders [\[27–30\]](#page-10-0). Crystallographically, $Ni₂Mo₃O₈$ lattice belongs to the noncentrosymmetric space group ($P6_3mc$) with Ni²⁺ ions occupying both the tetrahedral (Ni_1) and octahedral (Ni_2) sites, as shown in Fig. 1(b). It is noted that the Mo^{4+} ions appear to be nonmagnetic due to the trimerization [\[22\]](#page-10-0). The large linear ME effects have been reported in Fe₂Mo₃O₈ [\[27,28\]](#page-10-0) and Mn₂Mo₃O₈ [\[29\]](#page-10-0). One of their interesting features is that the AFM ground state appears to be fragile and can be easily switched to the ferrimagnetic (FIM) state or spin-flop state by either external magnetic field or Zn/Fe doping, which leads to the sign inverse of the ME coefficient [\[27,29\]](#page-10-0). Much recently we also reported the similar collinear magnetic structure in $Co₂Mo₃O₈$, accompanied only with the second-order ME effect [\[30\]](#page-10-0).

Although the electronic structures and ME effect of $(Fe, Mn, Co)₂Mo₃O₈$ were thoroughly investigated, it seems that no clear consensus on the microscopic origin of giant ME effect has been reached. The first-principles calculations suggest that oxygen ion displacements originated from the exchange striction in the AFM state underlie the observed giant ME coefficients [\[28\]](#page-10-0). Most recently, two scenarios based on the homogeneous distribution of $Fe²⁺$ ions among the tetrahedral and octahedral sites and charge disproportionation with somewhat exotic ionization state $1+$ at the tetrahedral sites were proposed $[23]$. Here Ni₂Mo₃O₈ is a much more attractive case. First, different from the collinear order along the *c* axis in those (Fe, Co, Mn)₂Mo₃O₈ materials, the feature of coexistence of stripe and zigzag AFM order in $Ni₂Mo₃O₈$ is clear, which enables an effort in exploring the rich physics of integer spin on the honeycomb lattice. Second, the weak metamagnetic feature in polycrystalline samples suggests a low-lying (in field) magnetic phase transition [\[21\]](#page-10-0), and hence measurements on single crystals are highly required to explore the metamagnetic transitions driven by magnetic field along different crystallographic directions, noting that the AFM orders along the *c* axis and in the *ab* plane are different [\[21\]](#page-10-0). Third, different from those $6'mm'$ (AFM) and $6m'm'$ (ferromagnetic) hexagonal magnetic point groups, the P_C na 2_1 orthogonal magnetic group allows nonlinear spiral spin order, which provides a platform to explore the potential novel physics beyond the linear ME coupling on a spin-1 honeycomb lattice [\[31\]](#page-10-0).

Motivated by these discussions, it is unambiguous to reveal the ME coupling by combining the characterizations on magnetization, specific heat, and ferroelectric polarization of high-quality $Ni₂Mo₃O₈$ single crystals over a broad range of magnetic field *H* up to 60 T in this work. The magnetic phase transition, ME coefficient, and ME phase diagram are systematically investigated. In contrast to the other members of M_2M_0 ₃O₈ family, our results reveal the distinctly different magnetic and ME coupling behaviors in the low- and highmagnetic field regions. In the low-field region, the electric polarization response shows the parabolic dependence while the magnetic structure exhibits the clear spin-flop transitions. More intriguingly, over the high-field region, the magnetization and electric polarization response exhibit extraordinarily broad plateaus against magnetic field along the *c* axis but roughly linear dependence on magnetic field along the *a* axis. Therefore, $Ni₂Mo₃O₈$ stands out as a new exotic multiferroic in this $M_2M_0_3O_8$ family.

II. EXPERIMENTAL DETAILS

The single crystals of $Ni₂Mo₃O₈$ were grown using the chemical vapor transport technique as described earlier [\[32,33\]](#page-10-0). In detail, stoichiometric mixture of high-purity NiO and $MoO₂$ powder was thoroughly ground and sealed in evacuated silica tubes at 1000 °C for 48 h to synthesize polycrystalline $Ni₂Mo₃O₈$ powder. Then the obtained power and transporting agent $TeCl₄$ with molar ratio 14:1 were sealed in evacuated silica tubes at 1000 °C in the change zone (T_1) and 850 °C in the growth zone (T_2) for 15 days to grow the single crystals [photoimaged crystals are shown in Fig. $1(c)$]. The crystal structure was characterized by x-ray diffraction (XRD) (D8 ADVANCE, Bruker) with Cu K_{α} source ($\lambda = 1.5406$ Å). The back-reflection Laue detector (MWL120, Multiwire Laboratrories, Ltd.) was used to check the quality of obtained single crystals and determine the crystallographic orientation. The electron dispersion spectroscopy (EDS) (Quanta 200, FEI) was used to analyze the chemical composition.

The well-prepared and aligned samples were submitted for a series of characterizations. First, the temperature (*T*) dependence of magnetic susceptibility (χ) were measured using the Quantum Design Superconducting Quantum Interference Device magnetometer (SQUID) from 2 to 300 K under the zero-field-cooled (ZFC) and field-cooled (FC) conditions with cooling magnetic field $H = 0.1$ T. Simultaneously, the magnetization (*M*) as a function of *H* was measured at selected *T*. For high-*H* experiments, the in-plane and out-of-plane magnetizations were measured at selected *T* in the pulsed field mode, using a coaxial pickup coil and calibrated by a comparison with the low-field data measured by SQUID. The specific heat (C_p) data were collected from $T = 2$ to 20 K using the Quantum Design Physical Properties Measurement System (PPMS).

Resembling earlier reports, the electric polarizations were obtained by conventional pyroelectric current measurement [\[34\]](#page-10-0), noting that no electric or magnetic poling of the sample is needed here, similar to $Fe₂Mo₃O₈$ [\[28\]](#page-10-0). The *T* dependence of pyroelectric current (J_c) was collected upon *T* increasing from 2 to 30 K using the Keithely 6514 programmable electrometer, with a sample warming rate of 4 K/min. In addition, to explore the ME coupling, the *H-*dependent magnetocurrent (J_m) with $H//a$ axis and $H//c$ axis modes were also recorded at a fixed T ($∼2$ K) upon *H* ramping from −9 to 9 T at a rate of 100 Oe/s. The variation in electric polarization (ΔP_c) was obtained by integrating the pyroelectric current with the time. In addition, the high field-induced ΔP_c was also measured in the $H//c$ and $H//a$ modes, noting that a small bias electric field $E = 1 \text{ kV/cm}$ was applied under the warming sequence to ensure the high-quality data. All the high-field data were measured using the 10.5 ms short-pulse magnet in the Wuhan National High Magnetic Field Center (WHMFC).

FIG. 2. (a) The measured magnetic susceptibility as a function of *T*, $\chi(T)$, and (b) the Curie-Weiss fitting χ^{-1} measured along the *c* axis and *a* axis, respectively. The inset shows the amplifying $\chi(T)$ curve and specific heat C_p details at low-*T* range.

III. RESULTS AND DISCUSSIONS

A. Crystal structure

In most cases, the as-grown crystals are naturally hexagonal in geometry, with \sim 2 mm in diameter and 1.0 to 1.5 mm in thickness, as shown in the inset of Fig. $1(c)$. The room-temperature slow-scan XRD pattern onto the naturally developed hexagonal plane is presented in Fig. [1\(c\)](#page-1-0) too, which shows very sharp diffraction peaks well indexed by the (00*l*) reflections. In addition, such a hexagonal plane is aligned with the back-reflection Laue detector, showing perfect diffraction spots from the [001] direction, demonstrating the good quality of our crystals.

The refined structure of $Ni₂Mo₃O₈$ fits the noncentrosymmetric $P6₃mc$ space group with unit cell parameters $a =$ $b = 5.757 \text{ Å}$, $c = 9.879 \text{ Å}$, in agreement with previously reported data from powder neutron scattering [\[21\]](#page-10-0), as shown in Fig. [1\(d\).](#page-1-0) Furthermore, the as-grown crystals were checked for their chemical composition using the EDS technique, giving rise to the atomic Mo:Ni ratio of $24.4 : 16.3 = 1.4969$, close to the nominal ratio 1.5. The high chemical and structural quality of the as-grown single crystals is confirmed.

B. Magnetic susceptibility and specific heat

We present the conventional characterizations on the DC magnetic properties. Figure $2(a)$ shows the measured $\chi(T)$ curves in the ZFC and FC modes with a measuring field of 0.1 T in the geometry aligned along the *c* axis and *a* axis, respectively. Several features deserve highlighting here. First, the $\chi(T)$ curves along the *c* axis $(\chi)/c$ and *a* axis $(\chi)/a$ show remarkable difference, noting that the shape anisotropy is negligible here. The larger in-plane susceptibility is consistent with the neutron scattering result [\[21\]](#page-10-0), suggesting the in-plane Ni²⁺ spin anisotropy. Second, the two $\chi(T)$ curves increase with decreasing *T* and reach the peak at $T_N \sim 5.5 \text{ K}$, the AFM Néel point, as highlighted in the inset of Fig. $2(a)$. This AFM transition is confirmed by the specific heat $C_p(T)$ curve presenting a typical λ-shaped anomaly (characteristic three-dimensional magnetic order) at $T_N \sim 5.5$ K, consistent with earlier data [\[21,35\]](#page-10-0). Third, the $\chi^{-1}(T)$ curves plotted in Fig. [2\(b\)](#page-2-0) unveil the dominant AFM interactions. By analyzing the data far above T_N (above 100 K), we find that the fitted Curie-Weiss temperature (θ) in the *H*//*c* (θ_c) and *H*//*a* (θ_a) geometries are −162.9 and −83.3 K, respectively, indicating the stronger AFM coupling along the *c* axis than the *a* axis. The derived effective paramagnetic moment μ_{eff} along the c axis and *a* axis are 3.33 and 3.30 μ_B /Ni. These values are between the expected value of spin-only moment $(2.8 \mu_B/Ni)$ and total effective moment $(5.6 \mu_B/Ni)$, implying the incomplete orbital moment quenching. Finally, the frustration ratio (f) along the two directions can be estimated too, giving rise to $f = \theta/T_N$ ∼ 29.6 for the *H*//*c* case and ∼15.1 for the *H*//*a* case, indicative of the stronger magnetic frustration in the *c* direction.

Here it should be mentioned that both the θ and μ_{eff} values extracted from our single crystals are slightly different from those from the polycrystalline samples [\[21\]](#page-10-0), and such difference can be reasonably attributed to the possible imperfections and random grain orientations with the polycrystalline samples. Indeed, a tiny amount of NiMoO4 impurity phase was found in the polycrystalline samples, as revealed by the powder neutron diffraction [\[21\]](#page-10-0).

C. Low-field magnetization

Based on the $\chi(T)$ data, it is concerning how magnetization *M* as a function of *H* evolves. We focus on the *M* responses to *H* in the $H//c$ and $H//a$ modes below T_N , namely $M_a(H)$ and $M_c(H)$. We discuss the data in the low-field and high-field regions to be defined below. The data in the lowfield region are displayed in Figs. 3(a) and 3(c) for the *H*//*c* and *H*//*a* cases.

In the low-field region, the typical AFM responses are observed, characterized by the roughly linear increasing of the $M_a(H)$ and $M_c(H)$ curves with *H*. Nevertheless, slight deviations from the linear dependencies reflect the spin-flop events. For the $M_c(H)$ data, such a deviation is marked by a small but clear jump at $H = H_{low}$, as shown in Fig. 3(a) and further illustrated by the sharp (dM_c/dH) peak in Fig. 3(b). This jump suggests the *H*-driven spin-flop event and it marks a metamagnetic transition often observed in transition metal oxides of the complicated AFM orders, where H_{low} is the critical field for such metamagnetic transition. For the $M_a(H)$ data, an obvious but weaker deviation from the linear dependence can be found at $T = 2$ and 3 K, as shown in Fig. 3(d), indicating a metamagnetic transition induced by the field. The critical field H_{low} for the $H//c$ cases are larger than that for the $H//a$ cases, and they evidently decrease with increasing *T* and disappear at $T_N \sim 5.0$ K.

FIG. 3. The low-field magnetization *M* of $\text{Ni}_2\text{Mo}_3\text{O}_8$. The magnetic field dependence of the magnetization with $H//c$ and $H//a$, namely (a) $M_c(H)$ and (c) $M_a(H)$ measured at selected temperatures. (b) and (d) Derivative of the magnetization dM_c/dH and dM_a/dH with respect to the magnetic field for fields *H*//*c* and *H*//*a* measured at $T = 2$ K.

D. High-field magnetization

Extending the measuring up to the high field region (pulse field events) unveiled some different characters, and an overall illustration of the $M_a(H)$ and $M_c(H)$ data are plotted in Figs. $4(a)$ and $4(c)$ where the low-field data from the SQUID

FIG. 4. The high-field magnetization of $Ni₂Mo₃O₈$. The *H* dependence of the magnetization (a) $M_c(H)$ with $H//c$, and (c) $M_a(H)$ with $H//a$ under selected *T*. (b) and (d) Derivative of the magnetization dM_c/dH and dM_a/dH with respect to the magnetic field for fields $H//c$ and $H//a$ measured at $T = 1.7$ K, respectively. The speculated spin arrangements located in the antiferromagnetic (AFM) phase, low-field spin-flop (SF1 $_a$ and SF1 $_c$) phase, and highfield spin-flop (SF2_{*a*} and SF2_{*c*}) phase with $H//c$ and $H//a$ are plotted on the top of the figure. The magenta arrow stands for the Ni₁ (tetrahedral site), and the green arrow denotes the $Ni₂$ (octahedral site).

measurement are inserted for comparison. It is seen that the spin-flop feature is evident in both the low-field SQUID data and pulse-field data on $M_c(H)$ and $M_a(H)$, confirming the data reliability. Besides the spin-flop jump at critical field H_{low} in the low-field region, another clear metamagnetic transition occurring at critical field *H*high is observed for both the *H*//*c* and *H*//*a* cases, respectively. The metamagnetic transitions can be more clearly identified in the (*dM*/*dH*) curves in Figs. $4(b)$ and $4(d)$, noting that the critical field (H_{high}) is used to define the low-field region and the high-field one in the present work.

Again, several features from the *M*(*H*) data deserve highlighting, taking the $M_c(H)$ data as an example. First, two metamagnetic transitions occurring at $H_{\text{low}} = 4.0 \text{ T}$ and $H_{\text{high}} = 13.8 \text{ T}$ for $T = 1.7 \text{ K}$ are unambiguously evidenced. Second, above H_{high} , a clear and broad $M_c(H)$ plateau ranging from H_{high} beyond 40 T is demonstrated at $T =$ 1.7 K. It is noted that the saturation value for this system would be \sim 5.9 μ _{*B*}/f.u. estimated from the ESR measurement on polycrystalline samples [\[21\]](#page-10-0). This plateau magnetization $(\sim 2.5 \mu_B/f.u.)$ is about half of the saturation value, indicative of the 1/2-like plateau. Third, with increasing *T*, both H_{low} and *H*high move to the low field side and the 1/2-like plateau becomes dim and almost disappeared near T_N . Similar but much more vague features can be found from the $M_a(H)$ data. While the two metamagnetic transitions occurring at relatively low fields can be identified, as shown in Figs. [4\(c\)](#page-3-0) and $4(d)$, the magnetization plateau feature is far from clear and the high-field magnetization M_a shows somehow linear increasing with *H* beyond 40 T.

While the two metamagnetic transitions are evidenced, the spin structure evolutions remain not well defined so far. As for the metamagnetic transition at ∼*H*low, the discussion on the polycrystalline samples suggests the conventional low-lying (in field) magnetic transition, which could be interpreted as differences between the in-plane and out-of-plane magnetic responses, is confirmed by our data on single crystals. Spinflop transition as one classical effect describes the magnetic field-driven first-order reorientation transition in some easyaxis antiferromagnets. In $Ni₂Mo₃O₈$, the magnetic moments of the tetrahedral $Ni₁$ (in magenta color) and the octahedral Ni₂ (in green color) are 2.64 and 1.68 μ _B, respectively. While the AFM interaction and spin frustration along the *c* axis are much stronger, here we assume that $Ni₁$ spins exhibit the stronger exchange interaction due to the larger moment, so that the spins on the $Ni₁$ sites are much more robust against the magnetic field than that on the $Ni₂$ sites.

In this case, for the low-field spin-flop transition, the *c*axis field will cause a rotation of spins on the octahedral $Ni₂$ site away from the easy axis (c axis), while the spins on the tetrahedral $Ni₁$ site do not necessarily reorientate owing to the relatively large moment and weak magnetic field. However, the nearest-neighbor $Ni₁-Ni₁$, $Ni₂-Ni₂$, and the next-nearest-neighbor Ni_1-Ni_2 along *c* axis still prefer the AFM interaction, as featured on the top of Fig. $4(a)$ by the schematic drawing of the spin structure. Similar to the discussion above, the *a*-axis field will cause a rotation of spins on the octahedral site away from the *a*-axis direction, in the low field range, noting that the Ni_1-Ni_1 , Ni_2-Ni_2 pairs still prefer the AFM interaction. Due to the strong AFM interac-

tion along the *c* axis, the critical field for spin-flop should be larger for M_c , consistent with our experimental finding that both H_{low} and H_{high} for $H//c$ cases are larger than those for *H*//*a* cases, respectively. In addition, there is a large difference in susceptibility between a longitudinal and transverse field, evidenced by dM_a/dH at AFM and SF1_a phases in Fig. $4(d)$. Exactly, Ni₂Mo₃O₈ exhibits an ordered magnetic state characterized as an admixture of stripy and zigzag order in the *ab* plane and *c* direction, respectively. In the *c* direction, the zigzag AFM order may be stabilized by a bond-dependent anisotropic exchange due to ligand distortion as reported in the $Ni₂Mo₃O₈$ polycrystalline [\[21\]](#page-10-0). Such a complex AFM order may lead to the anisotropy of the exchange interaction between the *ab* plane and *c* direction. These similar behaviors can be seen in other ME materials, such as the $Ni₃V₂O₈$ [\[12\]](#page-9-0).

However, it should be mentioned that the details of magnetic structures in responding to the high-lying spin-flop (SF2_{*a*} and SF2_{*c*}) transition at $H \sim H_{\text{high}}$ remain unclear and require more experiments to verify, such as neutron scattering experiments applied in magnetic field. Nevertheless, this does not prevent us from tentative explanation by combining the data on magnetization and field-induced polarization, which will be discussed in detail later. To this end, the low-field and high-field magnetizations are revealed to show distinctly different behaviors, benefiting to our understanding of the ME coupling while the ME responses in the low-field and high-field regions are also distinctly different. Subsequently, we turn to the main issue and discuss the ME responses in these two regions separately.

E. Low-field ME effect

 $Ni₂Mo₃O₈$ has the same polar structure as other $M₂Mo₃O₈$ systems and thus the most concerned issue is the ferroelectricity and ME coupling. Here the reason that we choose the *c* axis as the orientation along which the magnetoelectric coupling was measured, is that only the *c*-axis spontaneous polarization constrained by the *P*6₃*cm* lattice space group and *P*_Cna²₁ magnetic space group is allowed. Of course the ME tensors could also give rise to polarization along *a* axis, while unfortunately this is not investigated in this work. We do observe the emergence of pyroelectric current and thus electric polarization (P_{spin}) along the *c* axis but no other signal along the *a* axis (and *b* axis) in zero magnetic field. Our attention will be then paid to the polarization variation ΔP_c against *H*//*c* axis and *H*//*a* axis, respectively. For details, we present in Figs. $5(a)$ and $5(b)$ the pyroelectric current J_c and the evaluated P_{spin} as a function of T in the absence and presence of magnetic field. For $H = 0$, there is a sharp J_c peak emerging at T_N , indicating magnetic ordering-induced polarization, defining the ferroelectric transition point $T_{FE} \sim T_N$. With increasing H , one observes an evident shift of T_{FE} and suppression of P_{spin} , giving rise to ΔP_c . This suppression is more pronounced as *H*//*a* than that as *H*//*c*.

It is seen that P_{spin} for Ni₂Mo₃O₈ is ∼290 μ C/m² $T = 2$ K, much smaller than the values observed in sin-gle crystals of Fe₂Mo₃O₈ (∼1400 µC/m²) [\[27\]](#page-10-0), Mn₂Mo₃O₈ $(\sim 2000 \,\mu\text{C/m}^2)$ [\[29\]](#page-10-0), and Co₂Mo₃O₈ (\sim 1200 $\mu\text{C/m}^2$) [\[30\]](#page-10-0). Such large difference becomes reasonable if one notices the specific noncollinear magnetic structure of $Ni₂Mo₃O₈$,

FIG. 5. The *T*-dependent (a) pyroelectric current density (J_c) and (b) change of ferroelectric polarization (P_{spin}) against $H//c$ axis and $H//a$ axis. The *H* dependence of (a) magnetocurrent density (J_m) and variation of polarization (ΔP_c) against $H//c$ axis and $H//a$ axis at $T = 2$ K.

while other members share the collinear spin order. In fact, consulting with the well-established experimental facts and microscopic mechanism for type-II multiferroics, one understands that the spiral spin order induced electric polarization, associated with spin-orbit coupling, is generally small but the ME effect is usually remarkable, because the noncollinear spin order is more flexible against external stimuli. On the contrary, the symmetric exchange striction-induced polarization, essentially observed in materials with collinear spin order, can be much larger.

Regarding the ΔP_c as a function of *H*, i.e., the ME response, remarkable variation occurs for both the *H*//*c* and $H//a$ cases. In detail, one sees from Fig. $5(c)$ the magnetocurrent density (J_m) response against $H//c$ and $H//a$, respectively. Here the $J_m(H)$ data are collected by sweeping the magnetic field from positive to negative and then back to positive at $T = 2K$, as denoted by arrows in Fig. 5(c). Obviously there exists a significant difference between the ME effects for the $H//c$ and $H//a$ cases. For the $H//c$ case, *Jm* varies in a nearly linear behavior in response to the field, while the response for the *H*//*a* case exhibits a much stronger nonlinear character. It should be mentioned that the butterflylike ME loop for the *H*//*a* case, distinctly different from the $H//c$ case, is more or less related to the metamagnetic transition occurring at $H \sim H_{\text{high}}$.

The evaluated $\Delta P_c(H)$ data are plotted in Fig. 5(d), and the ΔP_c at *H* = 9 T can be as large as -250 and $-578 \,\mu$ C/m² for the *H*//*c* and *H*//*a* cases, respectively. This consequence is surprising since $\Delta P_c(H)$ can be one and two times larger than *P*spin, quite unusual for multiferroic materials and other $M_2M_0_3O_8$ members in which the *H*-modulated ΔP_c is much smaller than P_{spin} itself. As expected from the magnetic space group, the $\Delta P_c(H)$ behaviors for the $H//c$ and $H//a$ cases should follow the parabolic dependence, confirmed by the data in Fig. 5(d). In addition, the spin-flop triggered by *H* ∼ *H*_{high} is accompanied by a structural transition indicated by the deviation of $\Delta P_c(H)$ from the quadratic dependence.

FIG. 6. (a) and (b) The polarization (ΔP_c) with respect to the magnetic field for fields $H//a$ and $H//c$ up to 26 T measured at $T = 1.7$ K, respectively. (c) The $M_a(H)$ and $M_c(H)$ curves measured at $T = 1.7$ K with *H* up to 60 T. (d) The *H*-dependent polarization ΔP_c at $T = 1.7$ K with $H//a$ and $H//c$ up to 60 T. The dashed bright green lines and dashed magenta curve are the fits to the ΔP -*H* curve with the different functions of $\Delta P \sim \beta H^2$ and $\Delta P \sim \alpha H + \beta H^2$, where α and β are constants.

It is worth reminding that the consistency in the phase boundary between the $M(H)$ and $\Delta P(H)$ curves indicates the strong correlation between magnetism and ferroelectricity, and the anisotropic ME coupling effect is thus reasonably demonstrated. We thus transfer to the discussion of the highfield ME effect since it must be different from the low-field behavior.

F. High-field ME effect

Moving to the experimental results obtained at the Wuhan High Magnetic Field Laboratory, the measured $\Delta P_c(H)$ curves for the *H*//*a* and *H*//*c* cases at several selected *T* were collected. We first look at the data for *H* up to 26 T, as plotted in Figs. $6(a)$ and $6(b)$, respectively. While the data in the low-field regions reproduce the results in Fig. 5, the typical parabolic dependencies in the low-field region are clearly shown. We then pay more attention to the high-field region where the dependencies are quite different and exhibit the roughly linear behaviors with the one-to-one correspondences between $\Delta P_c(H)$ and $M_c(H)$. Especially plateaulike $\Delta P_c(H)$ and $M_c(H)$ features emerge for $H//c$ at low *T*, as shown in Figs. $6(b)$ and $6(c)$ for $T = 1.7$ K. Therefore, an intrinsic coupling between the electric polarization and magnetization is also confirmed in the high-field region.

It is found that the low- $T\Delta P_c$ for $H//a$ case is \sim − 1600 μ C/m² at *H* \sim 26 T, almost twice as that in *H*//*c* case, displaying the dramatic ME anisotropy. It is known that the *M* plateau is an ordinary phenomenon typically born in frustrated nonpolar materials [\[36–38\]](#page-10-0), nonetheless the coexistence of *M* plateau and *P* plateau in single-phase multiferroics is rather rare $[14,39]$ $[14,39]$. To the best of our knowledge, only the spin-1/2 honeycomb lattice antiferromagnet $Cu₂(pymca)₃(CTIO₄)$ with the *P*31*m* polar space group was reported to exhibit similar behavior [\[40\]](#page-10-0). Thus, $Ni₂Mo₃O₈$ seems to be a good platform for exploring the rich physics of integer spin on the honeycomb lattice. In addition, the plateaulike features seems to be persistent over a very broad region beyond $H \sim 60$ T, the limit of our measuring instrument, as shown in Figs. $6(c)$ and $6(d)$, respectively, indicating the considerable robustness of the AFM order along the *c* axis. For the $H//a$ case, both M_a and ΔP_c continue to increase linearly with increasing *H*, noting that the saturated magnetization is \sim 5.9 μ_B /f.u. [\[21\]](#page-10-0).

To present a quantitative measure of the ME effect, the ME coefficients are evaluated from the $\Delta P_c(H//a)$ and $\Delta P_c(H//c)$ data. In general, the $\Delta P_c(H)$ is expressed as a polynominal up to the second order:

$$
\Delta P_c = P_c - (P_{\text{latt}} + P_{\text{spin}}) = \alpha H + \beta H^2, \qquad (1)
$$

where P_{latt} is the lattice-related spontaneous polarization [\[29\]](#page-10-0), and α and β are the first-order and second-order ME coefficients, respectively.

It is already shown in Fig. $5(d)$ that the $\Delta P_c(H)$ curves below H_{high} show prefect quadratic fittings to $H₁$, as confirmed by the dashed bright green curves in Fig. [6\(d\),](#page-5-0) concomitance with the fact that the magnetic space group $(P_{C}na2_1)$ allows only the second-order ME coupling, while the linear ME effect is forbidden. In the low-field region, the fitted $\beta_{H//a} \sim -1.3 \times 10^{-17}$ s/A and $\beta_{H//c} \sim -5.1 \times$ 10^{-18} s/A are obtained in the low *T*. These values are much larger than those found for $Fe₂Mo₃O₈$ (1.81 × 10⁻²⁸ s/A) and $Co_2Mo_3O_8$ (2.9 × 10⁻¹⁸ s/A). We are more interested in the high-field region. For the $H//c$ cases, $\Delta P_c(H//c)$ exhibits the plateaulike behavior and the ME coefficients must be small or negligible. The $\Delta P_c(H//a)$ data in the high-field region can be well fitted by including the linear and $H²$ terms, as denoted as the dashed magenta curve in Fig. $6(d)$ for $T = 1.7$ K. The best-fitted ME coefficients are $\alpha_{H//a} \sim -70 \,\text{ps/m}$ and $\beta_{H//a} \sim 1.5 \times 10^{-19} \,\text{s/A}$, confirming that $\beta_{H//a}$ is indeed negligible. The linear ME coefficient $\alpha_{H//a}$ is much larger than those typical linear ME compounds, including Cr_2O_3 ($\alpha \sim 4.13 \text{ ps/m}$) [\[41\]](#page-10-0), Fe₂Mo₃O₈ $(\alpha \sim 16.2 \text{ ps/m})$ [\[27\]](#page-10-0), GaFeO₃ ($\alpha \sim 21 \text{ ps/m}$) [\[42\]](#page-10-0), Fe₂O₃ $(\alpha \sim 50 \,\text{ps/m})$ [\[43\]](#page-10-0), and NdCrTiO₅ ($\alpha \sim 2.01 \,\text{ps/m}$) [\[44\]](#page-10-0).

Such a large linear ME response seems to be unusual and we argue that it originates from the strong ME coupling due to the lattice distortions associated with the octahedral and tetrahedral coordination environments modified by high magnetic field [\[21\]](#page-10-0). This is not surprising and we recall previous reports on typical linear ME magnet LiMPO₄ ($M = Mn$, Fe, Co, Ni) where the important role of orbital moment contributions to the ferroelectricity was confirmed $[45]$. Hence, $Ni₂Mo₃O₈$ is a peculiar multiferroic that is distinct to other $M_2M_0^{3}$ O₈ members.

G. ME phase diagram

All the presented results are sufficient for us to draw the ME phase diagrams, as shown in Figs. $7(a)$ and $7(b)$, respectively. The phase regions are separated by the peak positions in the dP/dH , dM/dH , T_N (and also T_{FE} in Fig. [5\)](#page-5-0) as a function of *H* at various temperatures. The low-*T* AFM ground state is separated from the paramagnetic (PM) phase in the high-*T* range by the AFM Neel point T_N that is also the magnetic ordering point in the presence of magnetic field *H*. Our magnetization data show that this ordering point does not change

FIG. 7. The H -*T* magnetoelectric phase diagram of $Ni₂Mo₃O₈$ for (a) $H//a$, and (b) $H//c$ conditions. The results from low-field magnetization (LF)-*M*(*H*), high-field magnetization (HF)-*M*(*H*), and magnetic field response to polarization $\Delta P_c(H)$ were used to determine the phase boundary.

much against magnetic field up to ∼30 T, suggesting the vertical boundary between the PM phase and low-*T* ordered phases, as shown in Fig. 7.

In the low-*T* phase regions, the AFM ground state and low-field spin-flopped (SF1*^a* and SF1*c*) phase are separated by *H*low, as shown in Figs. [3](#page-3-0) and [4.](#page-3-0) The high-field phase (SF2*^a* and SF2*c*) and the low-field phase are separated by *H*high. These separating points constitute the phase boundaries, contributing to the phase diagrams shown in Fig. 7. It is also clearly shown that the phase diagrams for the *H*//*c* and *H*//*a* cases are quite different, the consequence of magnetic anisotropy.

IV. DISCUSSION

Finally, we present an updated discussion of the underlying mechanisms for the observed ME responses. On one hand, the *c*-axis spontaneous polarization emerging exactly at the AFM Néel point *T_N* (∼*T*_{FE}) confirms again Ni₂Mo₃O₈ as a type-I multiferroics [\[46,47\]](#page-10-0), implying that the microscopic origin for ferroelectricity and ME responses should be relevant with one of the three possible mechanisms: (1) the spin-current mechanism for noncollinear spin order $[48]$, (2) the exchange striction for collinear magnetic structure [\[49\]](#page-10-0), and (3) the *p-d* hybridization mechanism between spin and its ligand atoms with spin-orbital interaction [\[50\]](#page-10-0). Nevertheless, the experimental data available to us are not sufficient to identify which mechanism, the Dzyaloshinskii-Moriya (DM) interaction, single-ion anisotropies, or both, are responsible for the noncollinear spin structure, although it is known that the spins on the octahedral and tetrahedral sites form stripy and zigzag AFM character in the *ab* plane and *c* axis, respectively, as presented in Fig. $8(a)$.

In proceeding, we may employ the recently developed symmetry-based local ME tensor technique to discuss the microscopic origin of polarization [\[51](#page-10-0)[,52\]](#page-11-0). $Ni₂Mo₃O₈$ possesses the polar lattice symmetry and polar magnetic space

FIG. 8. (a) The schematically stripy *ab* plane and zigzag *c*direction components of the magnetic structure of $Ni₂Mo₃O₈$. The spins on each tetrahedral and octahedral sites are plotted in light magenta and olive arrows. Nearest neighbor antiferromagnetic and ferromagnetic interactions are expressed in red and blue dashed lines, and the distance Ni_1-Ni_2 is 3.384 Å and the Ni_1-Ni_1/Ni_2-Ni_2 bonds have the length of 5.759 Å. (b) The sketch of cycloidal spiral order along $2a + b$ and *c* direction. Black arrows denote directions of ferroelectric polarization arising from the DM interaction. The red arrows represent unit vector pointing from the nearest neighbor site $1i$ to site $2j$.

group *PCna*21, imposing severe symmetric constraints over the forms of local ME tensors. We start from the consideration of the nearest-neighbor and next-nearest-neighbor exchanges, as plotted in Fig. $8(a)$ where the nearest-neighbor distance $Ni₁-Ni₂$ is 3.384 Å and the $Ni₁-Ni₁/Ni₂-Ni₂$ bonds have the length of 5.759 Å. A consideration of the three types of Ni-Ni bonds should be sufficient for discussing the local electric dipoles. The total polarization generated in the $Ni₁-Ni₂$ layer, $pNi₁-Ni₂$, by the two $Ni₁-Ni₂$ nearest-neighboring interactions, can be expressed as

$$
p_{\text{Nil-Ni2}} = \sum_{\langle i,j \rangle} P_{ij}^{\alpha \beta \gamma} S_{i\alpha} S_{j\beta} = \left(\sum_{\langle i,j \rangle} P_{i,j}^{\alpha \beta \gamma} \right) S_{1\alpha} S_{2\beta}, \quad (2)
$$

where $\langle i, j \rangle$ runs all the adjacent Ni₁-Ni₂ pairs; (α, β, γ) runs over all the Cartesian coordinates, $x = 2a + b$, $y = b$, $z = c$. The magnetic point group is *mm*2, which is polar and allows for the second-order ME effect. Based on the theoretical analysis of two-spin tensor $P_{12}^{\alpha\beta\gamma}$ in the 32 point groups [\[52\]](#page-11-0), we can obtain the Ni₁-Ni₂ two-spin tensor $P_{12}^{\alpha\beta\gamma}$ as

$$
P_{12}^{\alpha\beta\gamma} = \begin{pmatrix} 0, 0, P_{12}^{xx(z)} & 0, 0, 0 & P_{12}^{xz(x)}, 0, 0 \\ 0, 0, 0 & 0, 0, P_{12}^{yy(z)} & 0, P_{12}^{yz(y)}, 0 \\ P_{12}^{zx(x)}, 0, 0 & 0, P_{12}^{zy(y)}, 0 & 0, 0, P_{12}^{zz(z)} \end{pmatrix}
$$

=
$$
\begin{pmatrix} 0, 0, f & 0, 0, 0 & f, 0, 0 \\ 0, 0, 0 & 0, 0, g & 0, g, 0 \\ h, 0, 0 & 0, k, 0 & 0, 0, l \end{pmatrix},
$$
 (3)

where f , g , h , k , l are the constants. Here the $(0, 0, l)$, $(f,$ 0, 0), and $(0, g, 0)$ denote vectors along the *z*, *x*, and *y* axis. For the zero field case, coefficient $l \neq 0$, allowing a local spontaneous polarization $p_{Ni_1\text{-}Ni_2}$ appearing in the *z* axis due to the nonzero components of $Ni₁$ and $Ni₂$ spins along the *z* axis, as shown in Figs. 8(a) and 8(b). For $H//x$ or $H//y$, the pNi_1-Ni_2 in the *z* axis can also be induced by the two-spin tensor *h* or *k*, consistent with our experiments where nonzero $\beta_{H//a} \sim -13 \times 10^{-18}$ s/A and $\beta_{H//c} \sim -5.1 \times 10^{-18}$ s/A in the AFM ground state were obtained.

For the Ni_1-Ni_1 and Ni_2-Ni_2 cases, the same form for tensor $P_{12}^{\alpha\beta\gamma}$ can also be exactly derived. From Eq. (2), the local polarization can be expressed in a form similar to the spin-current mechanism shown in Fig. 8(b):

$$
p_{\text{Nil-Ni2}} \propto e_{12} \times (S_1 \times S_2),\tag{4}
$$

here S_1 and S_2 are the spins on tetrahedral and octahedral sites, and vector e_{12} denotes the connection from S_1 to S_2 . In the framework of the spin-current model, one sees that the tetrahedral nickel carrying larger moment $(1.727 \mu_B)$ which lies mainly in the *ab* plane, and the moment $(1.431 \mu_B)$ on the octahedral site primarily points in the c axis $[21]$. Therefore, a cycloidal spiral spin order propagating along direction $x = 2a + b$ is explicitly demonstrated, as shown in Fig. 8(b). Here the spins on one cycloidal chain (highlighted in red and blue dashed rectangle) are labeled as S_{1i} , S_{2j} , S_{1k} , and S_{2l} , and thus ferroelectric polarization *P* ∼ e_{ij} × (S_{1i} × S_{2j}) could be induced along the *c* axis by breaking the inversion symmetry via the antisymmetric DM interaction [\[53\]](#page-11-0). Recently, a theoretical work using the effective spin model also confirmed the role of the DM interaction in developing the noncollinear spin order [\[54\]](#page-11-0) and thus the ME coupling effect in $Ni₂Mo₃O₈$.

Besides the DM interaction, one may also discuss the single-spin induced ferroelectric polarization p_{Ni} . It can be written as

$$
p_{\rm Ni} = \sum_{i} P_{ii}^{\alpha\beta\gamma} S_{1\alpha} S_{1\beta} + \sum_{j} P_{jj}^{\alpha\beta\gamma} S_{2\alpha} S_{2\beta},\tag{5}
$$

where S_1 spin and S_2 spin reside on the *i* sites and *j* sites, respectively. Based on the symmetry restriction, the Ni₁ and Ni₂ single-spin tensors $P_{ii}^{\alpha\beta\gamma}$ and $P_{jj}^{\alpha\beta\gamma}$ can be deduced as

$$
\sum_{i} P_{ii}^{\alpha\beta\gamma} = A \begin{pmatrix} 0, 0, P_{ii}^{xx(z)} & 0, 0, 0 & P_{ii}^{xz(x)}, 0, 0 \\ 0, 0, 0 & 0, 0, P_{ii}^{yy(z)} & 0, P_{ii}^{yz(y)}, 0 \\ P_{ii}^{zx(x)}, 0, 0 & 0, P_{ii}^{zy(y)}, 0 & 0, 0, P_{ii}^{zz(z)} \end{pmatrix}, (6)
$$

$$
\sum_{j} P_{jj}^{\alpha\beta\gamma} = -B \begin{pmatrix} 0, 0, P_{jj}^{xx(z)} & 0, 0, 0 & P_{jj}^{xz(x)}, 0, 0 \\ 0, 0, 0 & 0, 0, P_{jj}^{yy(z)} & 0, P_{jj}^{yz(y)}, 0 \\ P_{jj}^{zx(x)}, 0, 0 & 0, P_{jj}^{zy(y)}, 0 & 0, 0, P_{jj}^{zz(z)} \end{pmatrix}, (7)
$$

where *A* and *B* are constants, corresponding to the different magnitude of moment between $Ni₁$ and $Ni₂$. It is obvious that polarization $p_{\text{Ni}} \neq 0$ at $H = 0$ due to the constants $A \neq B$. In addition, no matter in which direction H is applied (x, y, or) *z* axis), the single-spin tensor from the two spin sites will no longer cancel each other out, e.g., $(A \cdot P_{ii}^{zz(z)} - B \times P_{jj}^{zz(z)}) \neq 0$ 0, leading to all allowed nonzero spontaneous polarization components and ME coefficients, consistent with our experimental results shown in Figs. $6(a)$ and $7(d)$. Therefore, the *p-d* hybridization mechanism corresponding to the single-spin mode can also qualitatively explain our observations. In such sense, the *p-d* hybridization mechanism and the spin-current model may both contribute to the ferroelectric polarization P_{spin} and the ME responses $\Delta P_c(H)$.

Surely it should be mentioned that a quantitative estimation of the polarization and ME responses is essentially hindered by the complex mathematical handling for the two-spin and single-spin ME tensors, and the discussion presented here is more or less the consequence of a number simplifications. Honestly, it is beyond our ability to determine the specific magnetic space groups of the different magnetic phases in the present case that is too much challenging, unless neutron scattering experiments in applied magnetic field or rigorous calculations can resolve the magnetic structure. Nevertheless, this does not prevent us from tentative explanation by combining the data on magnetization and field-induced polarization.

The powder neutron diffraction suggests that the ground state belongs to the 33.154 $(P_{C}na2_1)$ magnetic space group [\[21\]](#page-10-0). While the magnetic space group describes all the possibilities of atoms, it becomes far more complex to track. Here it is more convenient to use magnetic point group *mm*2 to describe the magnetic structure P_C *na* $2₁$. It is well known that by adding the time reversal symmetry operation, the magnetic point group *mm*2 can include four point groups: mm2, $mm2'$, $mm'2'$, and $m'm'2$, in which the first and second *m* denote mirror symmetry operation perpendicular to the *y* axis, and including the *y* axis, respectively. Here 2' represents a double rotational symmetry operation plus a time reversal symmetry operation. Because the antiferromagnetic order in the *c* direction creates time reversal symmetry, and the inplane mirror-symmetric antiferromagnetism cannot constitute time reversal symmetry, it is much more accurate to describe the ground state by point group *mm*2 . Following up aforementioned symmetry analysis, we will focus our attention on speculating the magnetic point groups upon the SF transitions.

First, let us discuss the case of *H*//*c*. As discussed above, we could speculate that only Ni₂ spins change their directions to be perpendicular to the c axis, while the $Ni₁$ spins could not change their directions near SF1*c*, due to the relatively weak exchange interaction of Ni₂ spins. Meanwhile, the $\Delta P_c(H)$ still follows the quadratic relation until the high-field spinflop transition at *H*high, indicating that the magnetic point group in the SF1*^c* does not change at all (*mm*2). At least it is reasonable to speculate that AFM interaction should be dominated below *H*high. Here we recall that similar behavior was once reported in $Ni₃TeO₆$ [\[13\]](#page-9-0), which exhibits different ME responses to the magnetic fields thanks to the Ni ions carrying different moments. In this case, the $Ni₁$ spins keep the AFM arrangement along c axis, while $Ni₂$ spins are still of the AFM order but perpendicular to the *c* axis, as featured on the top of Fig. $4(a)$. When the system enters into the $SF2_c$ phase, all the Ni₁ spins rotate towards the direction in perpendicular to the c axis while the Ni₂ spins prefer the ferromagnetic arrangement along the *c* axis, as the ME coupling is of the linear dependence, thus changing $2'$ to 2. In this way, the magnetic point group in the SF2*^c* phase could be *mm*2, which allows the first order ME tensor α_{33} , concomitant with our experimental results.

Second, for the $H//a$ case, similarly, only the Ni₂ spins change their direction to be perpendicular to the *a* axis with the AFM interaction in the SF1_a phase. Because the polarization $\Delta P_c(H)$ in the ground state keeps the same quadratic relation, the $mm2'$ for the SF1_a can be safely assumed. Eventually the $Ni₂$ spins would rotate to the field direction in the $SF2_a$ phase, and the Ni₁ spins have canted moments along the *a* axis, resulting in the canted ferromagnetic arrangement, as featured on the top of Fig. $4(c)$ In this case, the second mirror symmetry *m* (including the *y* axis) changes into *m* , while 2 keeps unchanged due to the AFM interaction along *c* axis. Therefore, the magnetic point group of $SF2_a$ phase is $mm'2'$, which allows the first order ME tensor α_{31} , perfectly verified by our experimental results.

Based on the analysis of the magnetic structures, one is allowed to determine which magnetoelectric coupling mechanism is dominated upon the low-field and high-field SF transitions. When the field $H < H_{\text{high}}$, the magnetic point group does not change with *H*//*a* and *H*//*c*, and noncollinear AFM magnetic structure dominates in the SF1*^a* and SF1*^c* phases. Subsequently, the *c*-axis spontaneous polarization can be reserved due to the DM interaction, see Fig. $8(b)$. On the other side, Even though it was reported that there might be a slight distortion of the octahedral and tetrahedral coordination environments from the ideal single ion crystal field to the symmetry-adapted, spin-orbit-coupled regime [\[21\]](#page-10-0), such distortion is very weak that no evident rotation of tilting are detected as shown in the refined neutron scattering at 15 and 1.5 K. Therefore, the net polarization induced by the *p-d* hybridization mechanisms would be zero if the $Ni₁$ moment on the tetrahedral site is along the *c* direction or in the *ab* plane. In this case, due to the noncollinear arrangement of $Ni₁$ spins, nonzero polarization could be observed, thus giving rise to both *c*-axis and *a*-axis polarization.

Second, for the field $H > H_{\text{high}}$, no matter $H//a$ or $H//c$, the spin-current model is no longer applicable to explain the emergence of polarization. Nevertheless, the *p-d* hybridization mechanism may apply, depending on the direction of magnetic field. In the case of $H//c$, the Ni₁ spins are perpendicular to the c axis and $Ni₂$ spins are a ferromagnetic arrangement along the *c* axis, thus there is no net polarization for S_{c} ^{Ω}*c* phase. Therefore, it is reasonable to ascribe the experimental observed polarization as the consequence of the exchange striction, as that in Fe₂Mo₃O₈ For the case of $H//a$, the Ni₁ spins on the tetrahedral site can induce both the *c*-axis and *a*-axis polarization components, due to the *p-d* hybridization, while the collinear exchange striction generates the *a*-axis polarization.

To summarize, we are allowed to propose that the magnetic point groups of SF1*c*, SF2*c*, SF1*a*, and SF2*^a* phases are $mm2', mm2, mm2', and mm2', noting that mm2' only$ allows the second-order ME tensor. On the contrary, *mm*2 and $mm'2'$ allow the first-order tensor α_{33} and α_{31} , respectively, while both of them allow the second-order tensor, consistent with our experimental results. In addition, it seems that the speculated magnetic structures can explain well the observed magnetization curves and ME effects. The *c*-axis ME coupling effect can be well explained by combination of spin-current, *p-d* hybridization, and exchange striction mechanisms, based on the low-field and high-field magnetic structures. Surely neutron scattering under magnetic field or reliable theoretical calculations are highly recommended to provide a clean identification of which mechanism or combined mechanisms are responsible for the observed spontaneous polarization and ME effects in $Ni₂Mo₃O₈$, in the future.

V. CONCLUSION

In conclusion, we have presented our systematic investigation on the multiferroic behaviors and ME responses of $Ni₂Mo₃O₈$ single crystals with a noncollinear-ordered spin-1 honeycomb lattice. The unambiguous magnetism-driven ferroelectricity behaviors have been demonstrated by the spontaneous ferroelectric polarization associated with the specific AFM ordering, as well as the quadratic ME response in the low-field region and linear ME response in the highfield region up to 60 T. In particular, the distinctly different ME responses to the in-plane and out-of-plane aligned magnetic fields reflect the characteristics of the polar magnetic symmetry and polar lattice symmetry. More intriguingly, the extraordinarily broad 1/2-like magnetization plateau against the *c*-axis aligned high magnetic field correlates the very

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different ME responses along the *c* axis and *a* axis. Both the spin-current model and *p-d* hybridization mechanism can be applied to explain the observed ferroelectric behaviors and ME responses, consistent with the symmetry-related local ME tensor analysis. The present works indicates that $Ni₂Mo₃O₈$ is a peculiar multiferroic, which provides a perfect platform to explore the rich physics of spin-1 on the honeycomb lattice. Further works, such as electromagnon resonance or nonreciprocal directional dichroism by terahertz (THz) spectroscopy, dynamic behavior of the magnetic moment under different magnetic fields via magnetic torque, and THz absorption spectroscopy measurements, *ab initio* calculations, or mean-field model, are highly recommended to unveil the ME origins as well as the hidden rich physics in this material.

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