

Multiferroic bismuth ferrite: Perturbed angular correlation studies on its ferroic α - β phase transition

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 (Received 12 May 2020; revised 13 October 2020; accepted 14 December 2020; published 29 December 2020)

Work of numerous research groups has shown different outcomes of studies of the transition from the ferroelectric α -phase to the high temperature β -phase of the multiferroic, magnetoelectric perovskite Bismuth Ferrite (BiFeO₃ or BFO). Using the perturbed angular correlation (PAC) method with ^{111m}Cd as the probe nucleus, the α to β phase transition was characterized. The phase transition temperature, the change of the crystal structure, and its parameters were supervised with measurements at different temperatures using a six detector PAC setup to observe the γ - γ decay of the ^{111m}Cd probe nucleus. The temperature dependence of the hyperfine parameters shows a change in coordination of the probe ion, which substitutes for the bismuth site, forecasting the phase transition to β -BFO by either increasing disorder or formation of a polytype transition structure. A visible drop of the quadrupole frequency ω_Q at a temperature of about $T_c \approx 820^\circ\text{C}$ indicates the α - β phase transition. For a given crystal symmetry, the DFT-calculations yield a specific local symmetry and electric field gradient value of the probe ion. The *Pbnm* (β -BFO) crystal symmetry yields calculated local electric field gradients, which very well match our experimental results. The assumption of other crystal symmetries results in significantly different computed local environments not corresponding to the experiment.

DOI: [10.1103/PhysRevB.102.224110](https://doi.org/10.1103/PhysRevB.102.224110)

I. INTRODUCTION

Significant work on multiferroic materials started to take place through Soviet research in the late 1950s. Generally, a multiferroic can be described as a material that exhibits a combination of at least two of the following ferroic forms of ordering, namely, ferromagnetism (which here includes every kind of magnetic appearance with a macroscopic magnetic moment M plus antiferromagnetism), ferroelectricity, and ferroelasticity, in the same phase [1]. Great focus lies on materials which show a combination of ferromagnetic and ferroelectric properties, as the simultaneous appearance of these forms opens new ways of exploiting material properties like electrically controlled magnetization and vice versa [2].

Wide fields of application are promised in spintronic and data storage devices as well as in sensoric and actuatoric systems [3]. In this aspect, among all discovered multiferroics, Bismuth Ferrite (also referred to as BFO), owing to its high antiferromagnetic Néel temperature of $T_N = 370^\circ\text{C}$ and high ferroelectric Curie temperature of $T_c \approx 825^\circ\text{C}$ [4], turned out to be a very promising candidate because it could be trained to exhibit a magnetoelectric effect even at room temperature. The α -phase of BFO appears in a rhombohedral structure with the cell parameters $a = 5.58 \text{ \AA}$ and $c = 13.87 \text{ \AA}$ in the hexagonal settings at room temperature. It is a quasicubic canted perovskite structure of the LiNbO₃ type [4]. The α -phase exists up to 820°C where BFO shows a phase transition to the β -phase, which is accompanied by a sudden contraction of the unit cell volume. Its character still is the subject of numerous discussions and is mentioned to appear in cubic, trigonal, and even tetragonal structures, as can be seen in Fig. 1 [5–11]. Eventually, Arnold *et al.* [9] suggested the β -phase to appear in the orthorhombic *Pbnm* space group with the cell parameters $a = 5.61 \text{ \AA}$, $b = 5.64 \text{ \AA}$, and $c = 7.97 \text{ \AA}$. To characterize the α - β phase transition in BFO, we applied

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	RT	T_N	T_C	β
[5] Raman	R3c AFM	R3c PM	Pm3m	
[6] DTA, XRD, Raman	R3c AFM	R3c PM	P2mm	
[7] XRD	R3c AFM	R3c PM	P21m	
[8] Ab initio calc.	R3c AFM	R3c PM	I4/mcm	
[9] PND	R3c AFM	R3c PM	Pbnm	
[10] XRD	R3c AFM	R3c PM	R3c	
[11] XRD	R3c AFM	R3c PM	Pbnm	
This work	R3c AFM	R3c PM	Pbnm	

FIG. 1. Summary of selected works addressing the crystallographic settings of the β -phase of BFO.

a combination of two fundamentally different experimental methods. Whereas perturbed angular correlation (PAC) is a local method, providing information on the nuclear environments, i.e., of one or two unit cells distance to the probe ion, x-ray diffraction (XRD) is a macroscopic method, unveiling details about the crystal structure over extended crystal regions. All PAC measurements were conducted at the isotope mass separator on-line (ISOLDE) facilities [12,13] at CERN in Geneva, Switzerland. XRD measurements were done at the X-Ray Center at the Vienna University of Technology in Vienna, Austria. Using a probe ion in an ionic crystal lattice allows to investigate the very local probe environment. In the isochemical case, e.g., using Mössbauer spectroscopy on ^{57}Fe in an iron containing a compound like BiFeO_3 , reflects the status of the crystal itself [14]. Using an isovalent but chemically different ion will induce mechanical distortions of the probe, which typically is of different ionic size, but no electrical effect is expected. Depending on the kind of electronic shell, also an isovalent ion may interfere with bonding, e.g., if in the lattice magnetically coupled d -orbitals (or f -orbitals) partake in bonding and a probe is introduced which does not provide the same d -orbitals. Then additional distortions arise due to the unmatched partially filled d -orbitals. For a heterovalent probe ion additional distortions may arise due to the electrostatic interaction. For solid solutions this induces local electric potentials and may attract other defects like oxygen vacancies. For large concentrations beyond the solid solution range, crystal phase transitions may be induced [15]. In ferroelectrics such probe defects are associated with yet another neighboring defect, typically an oxygen vacancy [16]. It is thus of fundamental interest to understand the role of such defects also from a local perspective. In a first-order phase transition a thermal hysteresis arises at this very phase transition [17]. Bismuth ferrite is a material with high ferroelectric Curie temperature and a strongly first-order phase transition into its paraelectric high temperature phase (β -phase). It is thus a prime candidate to investigate the local probe environments at phase transitions. Due to the high temperature of the phase transition, it is also close to the temperature where BiFeO_3 decomposes into $\text{Bi}_2\text{Fe}_4\text{O}_9$ under liberation of the volatile Bi_2O_3 [18]. For a significant local probe study, a short-lived isotope is advantageous at high radioactive decay rate to keep measurement times as short as possible. As we

showed in another work of Schell *et al.*, long-lived PAC probes like ^{181}Hf will suffer from significant changes of the sample during measurement at high temperature and are thus not suitable for this temperature range [19]. As the structural polar ordering of BiFeO_3 is constituted by the lone pair 6s-electrons of Bi, the discussion on its contribution to crystal structure transformation has been ongoing [20]. Using a local probe, such instabilities can be monitored locally. Additional density functional theory (DFT) calculations can give hints on the structural ordering from a local perspective. *Experimentally establishing a relation of such local order to the macroscopic crystal phase structure is intended in this work.* ^{111m}Cd is a short-lived and excellent PAC isotope. As will be discussed later, it occupies the Bi-site and thus represents a heterovalent probe ion in the crystal sublattice, which generates the ferroelectric ordering. It is thus directly exposed to the changes of lone pair orientation in the unit cell, which yields a change in symmetry of the site in case of nonalignment at instability points.

II. EXPERIMENTAL DETAILS

The polycrystalline BiFeO_3 samples were synthesized at the University of Duisburg-Essen from a stoichiometric mixture of finely powdered bismuth oxide (Bi_2O_3 , 99.9% Acros Organics) and iron oxide (Fe_2O_3 , 99.99% Alfa Aesar). Following a calcination process for 3 hours at 820°C , the powder was pressed to pellets and sintered for six hours under air atmosphere again at 820°C . The samples were shipped to the ISOLDE/CERN facility, where, to perform the PAC measurements, the Cadmium isotope ^{111m}Cd was implanted at the ISOLDE GLM beamline with an accelerator voltage of 30 keV. According to simulations with the SRIM software package [21], the implantation depth distribution shows a maximum at around 116 Å. The PAC measurements were done immediately after implantation in the facilities of the ISOLDE Solid State Physics Collaboration [13] using a fully digital six detector spectrometer [22]. The furnace with an included sample holder allowed measurements at all desired temperature points. All measurements were performed at temperatures reached from below. No PAC measurement was done on cooling due to the time limits in the experiment. PAC, being a hyperfine interaction technique, allows the investigation of the electromagnetic fields interacting with a nuclear probe located in the material [23,24]. The probe nucleus in its excited state decays to its ground state via the sensitive intermediate state through a γ - γ cascade decay. Through an angular correlation between the first and the second γ -ray, distinguished through their energies and detection times, hyperfine interactions between the probe nuclei and their surrounding atomic structure can be monitored. A fixed detector, which detects a first quantum γ_1 , defines a reference axis, conventionally designated as the z -axis. A number of nuclei with their initial spin I_i aligned to this direction is selected. This nuclear alignment leads to an anisotropic radiation of γ_2 with regard to γ_1 [25]. If the probe nucleus quadrupole momentum is disturbed through an electric field gradient (EFG) while being in the intermediate state, its spin I will experience a torque which forces a spin-precession around the before-referenced z -axis. Caused by this precession, the

population of the intermediate state sublevels changes with time, which again affects the spacial anisotropy of the γ_1 - γ_2 correlation. An electric field gradient is caused by charge distributions which deviate from an ideal spherically symmetric arrangement around the probe atom. It is defined as the second spacial derivative of the electrostatic potential. Through a diagonalization of the interaction Hamiltonian, rotating the tensor into its principal component coordinate system and the use of the Laplace equation, the EFG can be described by its largest component V_{zz} and the asymmetry parameter

$$\eta = \frac{V_{xx} - V_{yy}}{V_{zz}} \quad (1)$$

only. The quadrupole interaction frequency is calculated through

$$\omega_q = \frac{eQV_{zz}}{4I(2I-1)\hbar}, \quad (2)$$

where e is the electron charge, Q the quadrupole moment of the intermediate state of the probe nucleus, I its angular momentum, and \hbar the reduced Planck's constant. For an anisotropy coefficient η of zero, the lowest detectable substate-transition frequency is

$$\omega_0 = k\omega_q, \quad (3)$$

with $k = 6$ for half-integer values of spin momentum, like the $I = 5/2$ level of our probe isotope ^{111m}Cd . The higher-ordered frequencies are calculated through $\omega_n = n\omega_0$. If η is different from zero, the transition frequencies ω_n are functions of ω_q and η itself:

$$\omega_1 = 2\sqrt{3}\alpha\omega_q \sin\left(\frac{1}{3} \arccos(\beta)\right), \quad (4)$$

$$\omega_2 = 2\sqrt{3}\alpha\omega_q \sin\left(\frac{1}{3}[\pi - \arccos(\beta)]\right), \quad (5)$$

$$\omega_3 = 2\sqrt{3}\alpha\omega_q \sin\left(\frac{1}{3}[\pi + \arccos(\beta)]\right), \quad (6)$$

with

$$\alpha = \sqrt{\frac{28(3 + \eta^2)}{3}}, \quad (7)$$

and

$$\beta = \frac{80(1 - \eta^2)}{\alpha^3}. \quad (8)$$

These transition frequencies represent the transitions between the substate levels created through the hyperfine splitting of the intermediate state. For a $I = 5/2$ state and a pure quadrupole moment, the intermediate state is split into three sublevels $E_Q(M)$, resulting in three transition frequencies ω_n visible in the perturbation function [Eqs. (4) to (6)]. The normalized values of the sublevel energy states $E_Q(M)$ as well as of the transition frequencies ω_n and their change with rising asymmetry parameter η are shown in Figs. 2 and 3. Following Gerda [26], the theoretical perturbation factor for the fit of polycrystalline samples is described by

$$G_{kk}(\eta, t) = \sum_{n=0}^3 s_{kn}(\eta) \cos[\omega_n(\eta, V_{zz})t] \exp(-\delta\omega_n t), \quad (9)$$

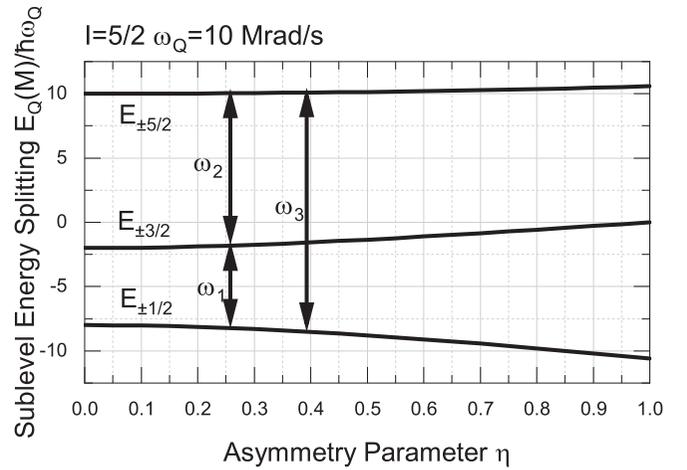


FIG. 2. Change of sublevel energies $E_Q(M)$ with increasing asymmetry parameter. Reference values only valid for $I = 5/2$.

with δ being the Lorentzian EFG distribution width and $s_{kn}(\eta)$ being weight functions depending on the asymmetry parameter η . Exact formulas for the weight functions $s_{kn}(\eta)$ for isotopes with an intermediate state of $I = 5/2$ can also be found in Friedsam [27]. The perturbation function of the γ - γ angular correlation is finally represented by the anisotropy function

$$\begin{aligned} R(t) &= A_{kk} G_{kk}(t) \\ &= A_{kk} \sum_{n=0}^3 s_{kn}(\eta) \cos[\omega_n(\eta, V_{zz})t] \exp(-\delta\omega_n t), \end{aligned} \quad (10)$$

with A_{kk} being correlation coefficients depending on the nuclear cascade and G_{kk} containing all information about the EFG [23,27–29].

III. RESULTS AND DISCUSSION

A. Simulation results

As mentioned before, the published data on the α - β phase transition temperature and the structure of the β -phase

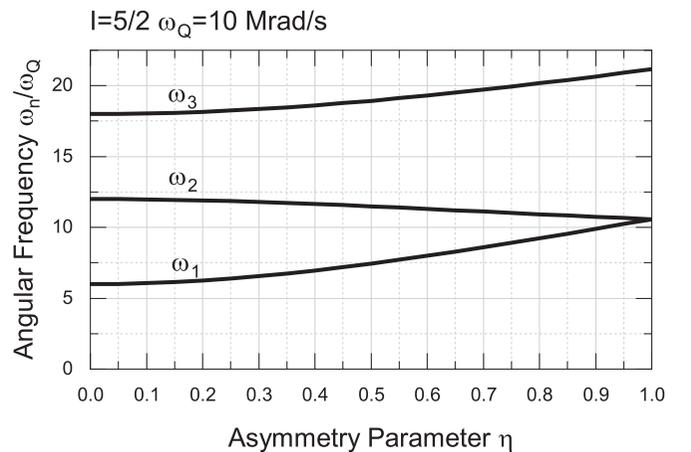


FIG. 3. Resulting change of angular frequencies ω_n calculated with Eqs. (4) to (6). Reference values only valid for $I = 5/2$.

TABLE I. Simulated values of V_{zz} and ΔH as well as the calculated value of ω_0 . The range of the results of ω_0 is caused by the uncertainty of Q .

System		$V_{zz}^{\text{DFT}} [\frac{10^{21} \text{V}}{\text{m}^2}]$	$\eta^{\text{DFT}} [1]$	$\Delta H^{\text{DFT}} [\text{eV}]$	$\omega_0^{\text{DFT}} [\text{Mrad/s}]$
α -BFO (FM/AFM)	Cd @ Bi	5.72/5.37	0/0	8.4/2.6	80.3–86.8/75.4–81.5
	Cd @ Fe	5.46/4.07	0/0	8.8/5.5	76.7–82.9/57.1–61.8
β -BFO (FM/AFM)	Cd @ Bi	−7.51/−6.88	0.39/0.26	4.2/1	105.4–113.97/96.57–104.41
	Cd @ Fe	2.37/8.43	0.87/0.18	5.4/2.2	33.3–35.9/118.33–127.93
$\text{Bi}_2\text{Fe}_4\text{O}_9$ (FM)	Cd @ Bi	9.99	0.12	–	140.22–151.61
	Cd @ Fe	7.99	0.84	–	112.15–121.25

contradict. The crystal structure of the perovskite BFO offers two different sites for the implanted $^{111\text{m}}\text{Cd}$ probe atom, namely the A-site, substituting the bismuth site and the B-site, substituting iron. To determine the most favorable substitution site as well as to set benchmarks for verification of the PAC results, DFT simulations were done using the Vienna Ab Initio Simulation Package (VASP) program [30]. The employed approach was the projector augmented wave method with a general gradient approximation using the $5d^{10}6s^26p^3$ valence configuration for Bi, $2s^22p^4$ configuration for Fe, $3p^63d^64s^2$ configuration for O, and the $4d^{10}5s^2$ configuration for Cd. For the α -phase ($Z = 6$) calculations, a $2 \times 2 \times 1$ supercell in the hexagonal setting, with chemical formulas $\text{CdBi}_{23}\text{Fe}_{24}\text{O}_{72}$ for Cd substituting Bi and $\text{Bi}_{24}\text{CdFe}_{23}\text{O}_{72}$ for Cd substituting Fe was used. We used a k -point grid of $4 \times 4 \times 3$ for the calculation. The β -phase ($Z = 4$) calculations were also done within a $2 \times 2 \times 1$ supercell with the formulas $\text{CdBi}_{15}\text{Fe}_{16}\text{O}_{48}$ for Cd substituting the Bi-site and $\text{Bi}_{16}\text{CdFe}_{15}\text{O}_{48}$ for Cd substituting the Fe-site with a k -point grid of $5 \times 5 \times 4$. $\text{Bi}_2\text{Fe}_4\text{O}_9$ ($Z = 2$) is a secondary phase detected in the x-ray diffraction measurements with space group $Pb\bar{m}$ and lattice parameters $a = 7.9826 \text{ \AA}$, $b = 8.49636 \text{ \AA}$, and $c = 6.04298 \text{ \AA}$. With this cell the distance between Cd impurities ($c \approx 6 \text{ \AA}$) may be small, so we used a $2 \times 1 \times 2$ supercell where the smallest distance between Cd impurities increases to the lattice parameter $b \approx 8.5 \text{ \AA}$ (k -point grid of $3 \times 2 \times 4$). The k -point densities were similar considering the supercell sizes. For all calculations the energy cutoff was 450 eV. The self-consistent calculations stopped for energy differences between two steps smaller than 10^{-5} eV. Other precision settings were set by the VASP tag `PREC=ACCURATE`. The atomic positions are optimized to minimize the forces below a threshold of 0.01 eV/\AA . The U -parameters in use were a PBE approximation with Coulomb repulsion $U_{\text{eff}} = 3 \text{ eV}$ applied to Fe- d electrons [31]. The electric field gradient V_{zz} , the anisotropy parameter η , and the formation energy ΔH were simulated for Cd substituting the Bi, as well as the Fe sites. The quadrupole interaction frequency ω_0 was then calculated with the use of formulas 2, 3, and 4 with the simulated values of V_{zz} , η , and $Q = 0.641(25) \text{ barn}$ [32]. Even though BFO is paramagnetic at our measurement temperatures, the paramagnetic order is very difficult to simulate. Temperature was not taken into account in the simulations. The paraelectric phase was not simulated with molecular dynamics as the computational cost would have been out of proportion considering the outcome. As a workaround, all simulations were done for ferromagnetic (FM) and G -type antiferromagnetic (AFM) magnetic spin or-

der to determine the EFG dependence on magnetic ordering in general. Simulation results and the calculated values of ω_0 of the α -phase, the β -phase and an additional ternary phase, $\text{Bi}_2\text{Fe}_4\text{O}_9$, which will be explained in Sec. III D, can be found in Table I. As the results of the α - and the β -phases show, the formation energy ΔH is lower in both the FM and the AFM order in the simulated system of Cd substituting the A-site. Thus for the dynamics explained in this work, a Cd substitution on the Bi site is energetically more favorable than a substitution on the Fe site, which is backed by the findings of Gebhardt [33]. We also calculated the local effect of two other structures which have been proposed for the β phase, considering Cd substitutional at Bi. For the $R\bar{3}c$ structure, $V_{zz} = 18.87 \times 10^{21} \text{ V/m}^2$ and $\eta = 0$ were obtained at the Cd site, substitutional at Bi. The corresponding $\omega_0 \approx 250 \text{ Mrad/s}$ is much higher than experimental values, while $\eta = 0$ also does not agree with the high η measured in the β phase. For the $P2_1/m$ structure, a large supercell calculation ($3 \times 2 \times 3$ of the original $P2_1/m$) was used. The obtained EFG at Cd is $V_{zz} = -24.8 \times 10^{21} \text{ V/m}^2$, $\eta = 0.37$. In this case we used a smaller energy cutoff of 400 eV, and the forces are smaller than 4 mRy/(bohr radius), a slightly less accurate calculation, due to the large computational time required. Nevertheless, the result should already have qualitatively converged for V_{zz} , and the corresponding $\omega_0 \approx 350 \text{ Mrad/s}$ is very far from experimental value, allowing us to conclude that this is not the measured configuration.

B. PAC results

Figure 4 shows the experimental $R(t)$ functions as well as fast fourier transformation (FFT) data with fits at measurement temperatures between 800°C and 850°C . The FFT analysis is helpful to visualize the $R(t)$ functions through the transition frequencies ω_n and their ratio. The connection to the asymmetry parameter η (see Fig. 3) of the containing EFGs is directly visible.

As mentioned in Secs. III A and III C, the Cd probe ion substitutes the Bi site within the BFO sample material, leading to sensitivity to changes in its nearest surrounding, which is set up by oxygen atoms forming coordination polyhedra around the Bi(Cd) atom. The change from a sixfold coordination in the α -phase to an eightfold coordination in the β -phase results from a change of the Bi-site symmetry. An ideal, nondistorted perovskite structure with point group $Pm\bar{3}m$ shows a 12-fold coordination on the A-site and a 6-fold coordination on the B-site. In the case of BFO though, the perovskite structure is

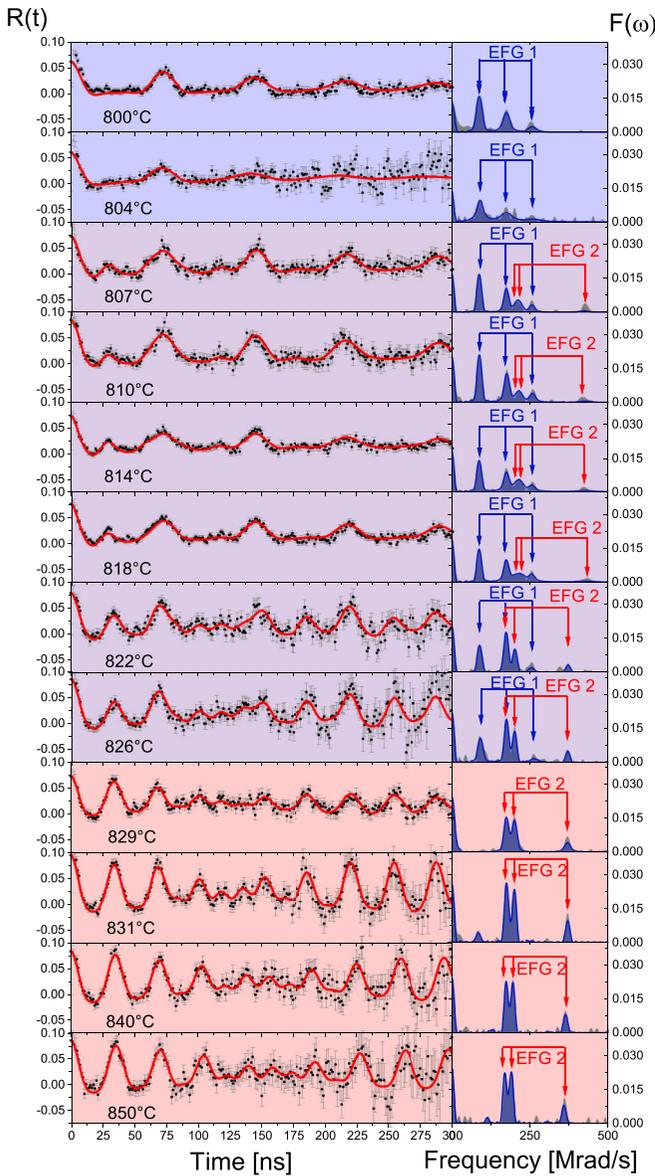


FIG. 4. TDPAC spectra with fitting functions in red and FFT showing the measured EFGs.

distorted, which changes the bond lengths of the coordinating atoms drastically. Regarding that, the coordination numbers mentioned here describe the number of nearest coordinating oxygen atoms which influence the environment of the cation most. The Bi ion in the α -phase is located on a threefold rotation axis so symmetric by 3-fold rotation (site symmetry 3, trigonal point group). On the other hand, the site symmetry of the Bi position in the β -phase is m , which refers to a monoclinic point group, meaning that the Bi ion is located on a mirror reflection plane and so symmetric by mirror reflection. This change in symmetry is accompanied by a change in average coordination lengths from $d_{\text{Bi-O}_\alpha} = 2.42 \text{ \AA}$ [34] to $d_{\text{Bi-O}_\beta} = 2.64 \text{ \AA}$ [9], both visible in the PAC results as the following chapter will illustrate. The visualization of the single EFGs in Fig. 4 already hints towards the changes happening during the phase transition from the $R3c$ to the $Pbnm$ phase as it shows a clear assignment of EFG1 to the α -phase and

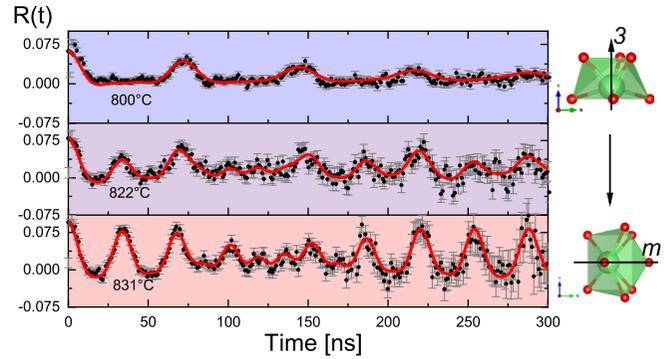


FIG. 5. Three distinctive PAC spectra showing single α -phase on top, the disordered range with a combination of α and β coordination in the middle, and pure β -phase in the bottom spectrum. Specific Bi(Cd)-O polyhedra with symmetry elements to the right.

of EFG2 to the β -phase. The data acquisition as well as the fitting and the calculation of the experimental EFG were done with the GFIT19 software package, a modified version of the NNFIT software package [35,36]. Hyperfine parameters of all fits were calculated and can be found in Table II. The higher number of near neighboring oxygen atoms in the surrounding of the cadmium probe atom in eight-fold coordination (β -phase) leads to an increased quadrupole interaction frequency with respect to the 6-fold coordination in the α -phase which can be seen in Fig. 5. Up to a measurement temperature of 804°C , only the lower frequency EFG ($\omega_0 \sim 86 \text{ Mrad/s}$) is present. Compared with the simulation results listed in Table I, we associate this field gradient (EFG1) to α -BFO in ferromagnetic order with Cd@Bi. The measured asymmetry parameter η also matches the simulated value reasonably well. As is visible on the FFT of EFG1 in Fig. 4, the low value of η close to zero leads to a $\omega_1 : \omega_2 : \omega_3$ ratio of 1 : 2 : 3 (compare to Fig. 3). Starting at 807°C , also the high frequency field gradient ($\omega_0 \sim 100\text{--}120 \text{ Mrad/s}$) appears. This EFG is linked to β -BFO in ferromagnetic order (compare with Table I), also with Cd@Bi. The mismatch of the measured and the simulated value of η is considered less important than the matching values of $\omega_0(V_{zz})$ [37]. Above 826°C , only the high frequency field gradient (EFG2, β -phase) is detected. As seen in Fig. 6, the values of ω_0 assigned to the α -phase are practically constant throughout the temperature increase. On the other hand, the quadrupole frequency of EFG2 which is assigned to the β -phase shows a decrease from $\omega_0 = 123.67 \text{ Mrad/s}$ at 818°C to $\omega_0 = 110.05 \text{ Mrad/s}$ at 822°C . This can also be seen in the FFT functions in Fig. 4 as EFG2 clearly changes between 818°C and 822°C . This feature is the indication for the phase transition. The discordant values of ω_0 of EFG2 right before that decrease indicate the following: With increasing temperature and the approaching of the phase transition temperature, local coordination disordering of the Bi-site, which below 820°C is still in its 6-fold α -phase coordination, appears. This means that a pseudo m -symmetry coordination, which actually belongs to the not yet existing β -phase, starts to appear on the 3-fold-symmetry site of the Bi ion. This disordered coordination distorts the EFG and leads to the high values of ω_0 . The second parameter to distinguish between the α and the β phases is the asymmetry

TABLE II. Experimental PAC values of hyperfine parameters.

Temp. [°C]	ω_0^{PAC} [Mrad/s]	α -phase			β -phase			
		η^{PAC} [1]	δ^{PAC} [%]	%	ω_0^{PAC} [Mrad/s]	η^{PAC} [1]	δ^{PAC} [%]	%
500	106(6)	0.44(7)	30(7)	100				
603	91(3)	0.43(6)	19(2)	100				
650	88.5(3)	0.14(2)	4.6(4)	100				
703	87(1)	0.23(3)	10(1)	100				
776	86.8(5)	0.16(2)	10(1)	100				
800	86.6(3)	0.10(2)	5.3(3)	100				
804	87.3(5)	0.16(4)	10.3(6)	100				
807	86.5(2)	0.08(1)	2.0(3)	78(5)	122(0)	0.92(1)	0.2(7)	22(2)
810	86.8(2)	0.02(2)	2.6(3)	79(9)	120.54(0)	0.99(4)	4(1)	21(3)
814	86.5(2)	0.07(2)	3.3(3)	65(6)	121.14(0)	0.99(2)	5.6(7)	35(2)
818	86.3(2)	0.08(1)	2.2(2)	61(2)	123.67(0)	0.90(3)	10(1)	39(2)
822	85.4(5)	0.16(1)	2.5(1)	50(6)	110.05(0)	0.83(0)	0.7(3)	50(6)
826	88.0(1)	0.15(2)	4.4(9)	49(5)	109.66(0)	0.83(1)	0.03(6)	51(5)
829					109.4(1)	0.8(0)	2.2(2)	100
831					109.2(1)	0.84(0)	0.2(1)	100
840					106.6(2)	0.86(0)	0.6(2)	100
850					105.3(3)	0.86(0)	0.5(2)	100

parameter η . As clearly seen in Fig. 7, the α -phase appears in a rather symmetric setting ($\eta_\alpha \sim 0.1$) compared to the β -phase ($\eta_\beta \sim 0.9$). There is a slight change of the asymmetry parameter in EFG2 between 814° C and 822° C as the value drops from $\eta_{814} = 0.996$ to $\eta_{822} = 0.828$. This dynamic is linked to the coordination disordering described before. The surrounding of the probe atom thus experiences a noticeable structural change in this temperature region. An indication for these dynamics are the slightly increasing values of damping δ in the β -EFG between 807° C and 818° C (Fig. 8), as with the increasing number of disordered Bi-coordination sites the EFG distribution width broadens. The damping δ , shown in Fig. 8, is generally low at all measurements, indicating a rather narrow EFG distribution and thus allows an accurate analysis. It also shows that the change in EFG due to fluctuations is low.

Thus, the crystal structure itself changes likely due to longer range interactions. The disordering of the Bi-sites and the following changes in coordination are an independent effect, which are only visible locally. The visible changes in damping δ , even though they are small, have a significant meaning. EFG1 by comparison shows a peak in damping at 804° C. This feature indicates the emergence of the β -phase EFG, caused by the mentioned pseudo m -symmetry coordinations, which is already present at the next measurement point at 807° C. The increasing δ -value between 807° C and 818° C in EFG2 was already addressed before. At 822° C, the damping again drops to a value of $\delta_{822^\circ\text{C}} = 0.68(27)\%$. This indicates that the phase transition is completed. From now on, the Bi ion sits on the mentioned site with m -symmetry in an eightfold coordination. EFG1 exists up to a temperature of 826° C, where the last

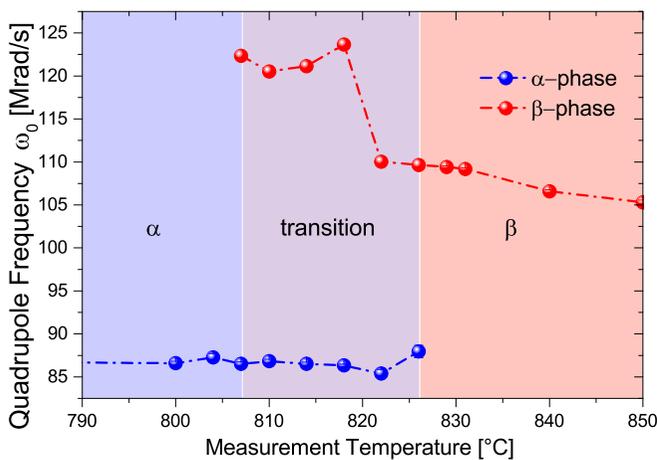


FIG. 6. Temperature dependence of the quadrupole interaction frequency ω_0 . The blue area shows the rhombohedral α -phase region and the red area the orthorhombic β -phase region with the area of disordered coordination in purple in between. If no errorbars are visible, the symmetrical error is smaller than the marker size.

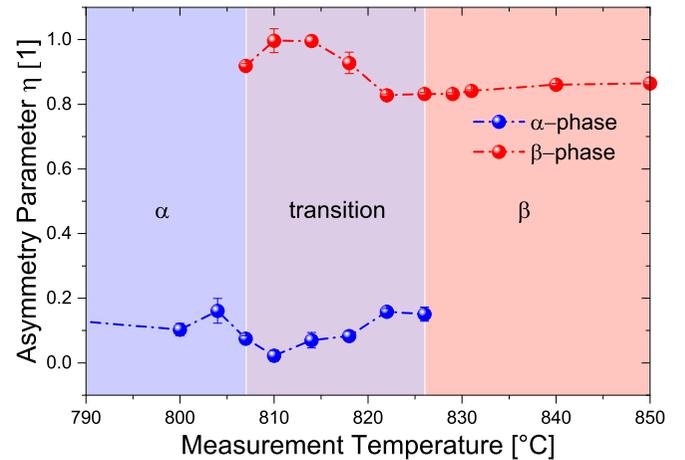


FIG. 7. Temperature dependence of the asymmetry parameter η . The blue area shows the rhombohedral α -phase region and the red area the orthorhombic β -phase region with the area of disordered coordination in purple in between. If no errorbars are visible, the symmetrical error is smaller than the marker size.

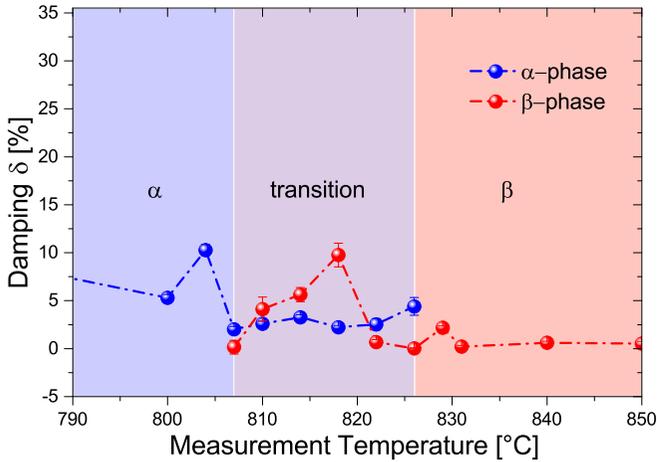


FIG. 8. Temperature dependence of the damping parameter δ . The blue area shows the rhombohedral α -phase region and the red area the orthorhombic β -phase region with the area of disordered coordination in purple in between. If no errorbars are visible, the symmetrical error is smaller than the marker size.

remaining pseudocoordinates with 3-fold-symmetry vanish. As can be seen in Fig. 9, the fraction of EFG2 is increasing with increasing temperature while the fraction of EFG1 decreases. This, once again, underlines the dynamics of the coordination change as the ^{111m}Cd probe ion forecasts the phase transition. An alternative explanation which is not in contradiction with our measurement results was given by Prosandeev *et al.* [38] with a polytype phase that acts as an intermediate structure between the $R3c$ α -phase and the $Pbnm$ β -phase. This putative transition structure is designated “nano-twinned” in the work of Prosandeev *et al.* PAC data, as mentioned, allow a very specific, local picture of the closest environment of the probe atom. Thus, the resulting polytype structure, with a period length of six lattice constants along [001] cannot be measured as a unique, independent EFG but

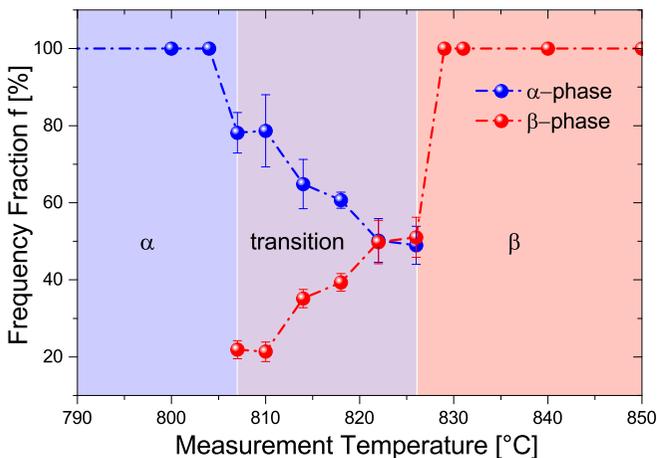


FIG. 9. Temperature dependence of the frequency fraction f . The blue area shows the rhombohedral α -phase region and the red area the orthorhombic β -phase region with the area of disordered coordination in purple in between. If no errorbars are visible, the symmetrical error is smaller than the marker size.

can be interpreted through the coincident appearance of EFG1 and EFG2. In contrast to the predictions of Prosandeev *et al.*, the observed transition structure is stable only in a 19 K as opposed to a 250 K region. Considering this, the phase transition can be explained as a combination of two independent phase transitions following each other. The visible transition range around the α - β phase transition in BFO to our knowledge is a novel result. This range is thus not only a response to the substitutional Cd cations but also to global crystal symmetry change: On the one hand, crystal defects, such as the Cd substitution in BFO, lead to strong distortions in their closest vicinity which also might induce a delayed or accelerated phase transition, at least on a local point of view. The local defect induced broadening of the transition range can only be measured with PAC and not with XRD. On the other hand, the XRD data of pure BFO show a hysteresis around T_c (see below), which is smaller but with a width of around 8 K accounts for roughly 45% of the transition range width measured with PAC. In consideration of the mentioned details, we assume that there is a hysteresis of the long-range order parameter. The increase of the frequency fraction attributed to the β -phase in Fig. 9 can be interpreted as a phase transition of the second order. Such a second-order phase transition could either mean an increase of the fraction of the polytypes predicted by Prosandeev *et al.* or a continuous decrease of the periodicity of the repetition period of the stacking, i.e., an increase in β -BFO like coordination.

C. Probe site assignment

Site assignment for tracer ions is sometimes a difficult task. We here collect arguments for A or B site substitution. DFT simulations, which are only strictly valid at a temperature of 0 K, might to some extent be uncertain at our measurement temperatures. A comparison of the ionic radii of the involved elements strongly suggests a substitution of the Bi site through the Cd ion. Depending on the coordination number (CN), the ionic radii of all ions vary as summarized in Table III. Bi as well as the Fe cations are assumed to be in a 6-fold coordination in the α -phase, with their effective ionic radii of 1.02 Å and 0.55 Å (low spin)/0.645 Å (high spin), respectively. Thus, the value of a 6-fold coordination of Cd^{2+} was taken for comparison [9]. The 6-fold effective ion radius of Cd with 0.95 Å is much closer to the radius of the Bi ion than to the Fe ion, which is the most important indication for the substitution

TABLE III. Effective ion radii of Cd^{2+} , Bi^{3+} , and Fe^{3+} ions in Å for different coordination numbers CN [39].

CN	Ion				
	Cd^{2+}	Bi^{3+}	Fe^{3+}	In^{3+}	Mn^{3+}
IV	0.84	–	0.49	–	–
V	0.87	0.99	–	–	0.58
VI	0.95	1.02	0.55 (LS) 0.645 (HS)	0.79	0.58 (LS) 0.65 (HS)
VII	1	–	–	–	–
VIII	1.07	1.11	–	0.923	–
XII	1.31	–	–	–	–

at the A-site [39]. A substitution of the Fe cation would necessarily lead to a significant distortion of the unit cell, possibly followed by an asymmetry parameter different from zero. A counterargument would be that ferroelectricity in BiFeO₃ is mainly due to the lone pair polarization of the Bi ion. Thus, Cd substituting Bi must experience the local effect of changes in polarization in its field gradient. So, the EFG should change also before the Curie point itself (α to β phase transition) is reached. Landau theory predicts a moderate change of the order parameter just beneath such a strong first-order phase transition [40]. So unlike in second-order phase transitions, where a large change in V_{zz} would have to precede the phase transition, the jump in order parameter at the first-order phase transition temperature allows for only moderate changes just beneath T_c . We observe practically no change in V_{zz} beneath the phase transition temperature and assign this to the very limited temperature range we investigate. In this range no change is within the error bar of analysis with a “moderate” change which would be a few percent only as we know from other data on BiFeO₃. The ^{111m}Cd probe site substitution was also a point of discussion in rare-earth orthochromite ceramics. Comparisons with ¹¹¹In(¹¹¹Cd) PAC measurements done on rare-earth orthoferrites lead to the assumption that in the particular case of SmCrO₃ the ^{111m}Cd probe also occupies the A-site [41]. The ¹¹¹In(¹¹¹Cd) probe used to examine the rare-earth orthoferrites (RFeO₃, REO) occupies both sites, leading to a low frequency EFG describing the Fe-occupancy and a high frequency EFG describing the R-occupancy [42]. As seen in Table III, ¹¹¹In has a significantly smaller ionic radius (0.79 Å in 6-fold coordination) than ^{111m}Cd, which possibly leads to an occupation of both sites. Rearick *et al.* [42] also mentioned that the ¹¹¹In probe primarily substitutes into the rare-earth sites in the heavier REOs. Though bismuth is not a rare-earth element, an A-site occupation is more likely as Bi is heavier than all the rare earth elements. Further literature affirming the A-site substitution are the works of Bellakki [43], which addressed Cd-doped BFO and Ishaq *et al.* [44], which treats Cd-Mn doped BFO. Bellakki states that Cd substitutes only the Bi cation. Ishaq *et al.* described that Cd substitutes the Bi site whereas Mn substitutes the Fe site. Reconsidering the ionic radii of Cd and Mn (Table III) this is expected as the value of Mn is significantly closer to the one of Fe than the ionic radius of Cd which is similar to Bi. Sato *et al.* also describe ^{111m}Cd as substituting the R and A sites in doped perovskite manganese oxides ($R_{1-x}A_xMnO_3$) [45]. As in our work only one EFG is assigned to the α -phase, A-site occupancy is what we conclude from these arguments.

D. XRD results

The powder diffraction measurements were performed in air using a PANalytical X’Pert Pro MPD θ - θ diffractometer equipped with an Anton Paar HTK 1200 high temperature chamber and an X’celerator 128 channel detector. Intensity data were collected in the $2\theta = 20^\circ$ – 70° range with a scan length of 2.122° with 55.2 s exposure per scan length resulting in overall measurement times of about 45 minutes. Data were converted into step size bins of 0.0084° . This measurement routine was used to determine the overall phase composition of the samples at different temperature points around

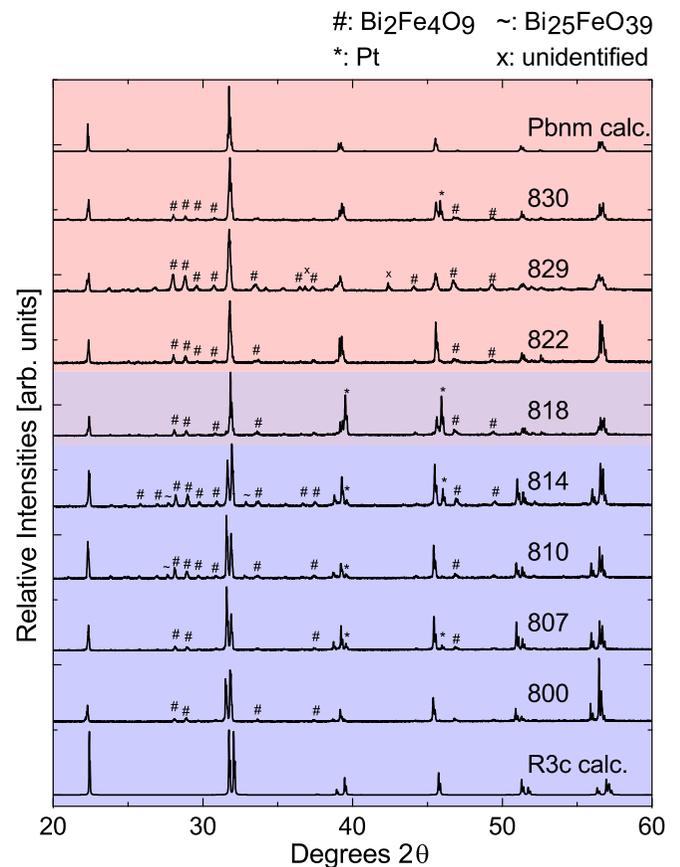


FIG. 10. Temperature dependence of x-ray diffractograms measured between 20° – 70° (graphs are only shown up to 60° as no information gain is achieved from reflections above 60°) at increasing temperatures around the phase transition temperature. The top and bottom graphs show calculated diffractograms of BFOs $R3c$ and $Pbnm$ structure, respectively. The intensities are normalized to the highest intensity.

the presumed Curie temperature as suggested by the PAC measurements. Sweep measurements covering the 2θ -range between 30° and 35° were done to determine the exact phase transition temperature. Samples were ground and the powder was placed on platinum sample holders to avoid reactions with standard sample holders such as Si single crystals, quartz or Macor. Figure 10 shows XRD diffractograms of BFO, measured at temperatures between 800°C and 830°C as well as XRD patterns calculated from crystallographic information files (CIFs) for α -BFO with $R3c$ and β -BFO with $Pbnm$ symmetry. The structural data given in the CIF were also used for the simulations described in Sec. III A. This gives a first impression of the conformance of the results of PAC and XRD measurements. The calculated diffractogram for the α -phase coincides perfectly with the measured data up to 814°C (Fig. 10), thus this temperature region is considered to represent the single α -phase. Rietveld refinement backs this as the β -phase does not appear below 818°C . This temperature can be considered as the phase transition point of a first-order phase transition, as α and β phase are present simultaneously. This diffractogram is mostly consistent with the β -phase. However, minor amounts of the α -phase are still observed for example by the not completely vanished 104 reflection at

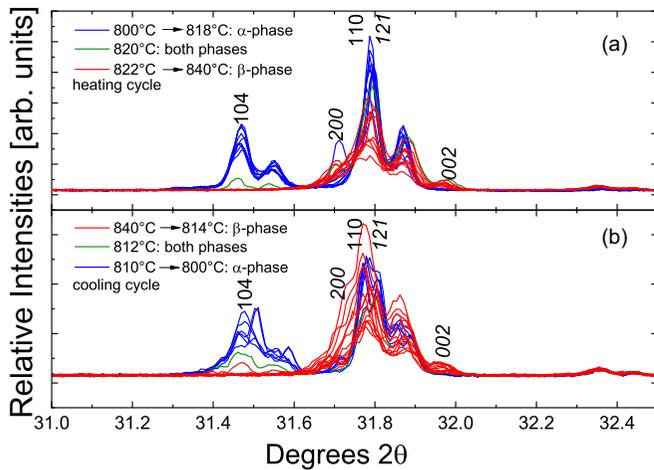


FIG. 11. Reflection behavior at the phase transition. α -phase in blue, mixed phase in green, and β -phase in red. Reflection descriptions of the β -phase are italic. (a) The top diffractogram shows the reflection merging during the heating cycle, (b) the lower diffractogram shows the reflection splitting during the cooling cycle.

$2\theta = 31.5^\circ$, which is most likely caused by a temperature gradient. Rietveld refinement results in a α -BFO fraction of ca. 8%. The additional reflections appearing in the diffractograms are either Pt reflections from the sample holder (marked with *) or of the known decomposition products $\text{Bi}_2\text{Fe}_4\text{O}_9$ (#) and $\text{Bi}_{25}\text{FeO}_{39}$ (~) as BFO is metastable at elevated temperatures [46]. The two reflections labelled with x were not identified but could result from sample contamination during the measurements. As mentioned, the phase transition was also analyzed through sweep measurements in a small angle range, determined to ascertain the exact phase transition temperature as well as the hysteresis around the transition point. A suitable reflection for this examination was found to be the $104\text{-}2\bar{1}0$ doublet of α -BFO at approximately $2\theta = 31.5^\circ$ and $2\theta = 32^\circ$, respectively, which merges to a narrow reflection triplet of 020, 112, and 200 of the orthorhombic β -phase when the structural phase transition occurs (see Fig. 11). The sweep was measured with temperature steps of 2 K between 800°C and 840°C . Up to a temperature of 818°C , the diffractogram of the heating cycle shows clear evidence of only the α -phase being present. In the 820°C dataset, shown in green, the diffractogram shows a distinctly weaker intensity at the 104 reflection and also an attenuated intensity of the $2\bar{1}0$ reflection, leading to the conclusion that the α -phase is vanishing, representing the region of coexistence of both phases. In all datasets acquired above 820°C , shown in red, only the mentioned triplet of the β -phase is present. A hysteresis is visible because the α -phase starts to appear only at 812°C during the cooling cycle. Rietveld refinement of the sweep measurements allows a coarse estimation of the available phase fractions which are shown for the heating cycle in red and the cooling cycle in blue in Fig. 12. For the data line in red in Fig. 11(b), which shows a very first appearance of the 104 reflection (814°C), the Rietveld refinement estimated a zero percent α -phase fraction. Again, through the very narrow angular range of the sweep measurement, an exact refinement is difficult but the results are

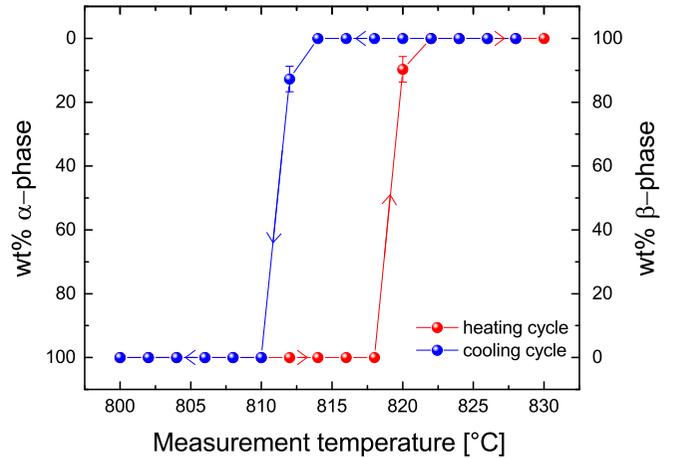


FIG. 12. Temperature development of the Rietveld refinement phase fractions given in weight percent showing a clear hysteresis. The heating cycle is depicted in red and the cooling cycle in blue.

specific enough to describe the appearing hysteresis of around 8 K (Fig. 12). The polytype transition structure described by Prosandeev *et al.* could not be evidenced by XRD data. This is not unusual since such polytypes often cause tiny satellites, which cannot be seen in standard powder diffraction experiments. Sometimes such phase transitions between a basic and a superstructure are reflected in a change of cell parameters. However, we did not observe any anomaly in the thermal expansion of the α -phase close to the phase transition temperature. The temperature difference of 2 K between the phase transition temperature observed in these sweep measurements in the heating cycle (820°C) and the measurements covering the angle between 20° and 70° (818°C) can be explained through uncertainties of the temperature measurement in the high temperature chamber during the different single experiments. Reflections of the ternary mullite $\text{Bi}_2\text{Fe}_4\text{O}_9$ and the selenite $\text{Bi}_{25}\text{FeO}_{39}$ phases are visible in all measured diffractograms (Fig. 10). These decomposition products, which also form under vacuum and inert gas, were not present in the samples before they were subjected to the measurements at high temperatures. In fact, comparisons of diffractograms shown in Fig. 13 recorded at temperatures between 600°C and 800°C show that the impurity phases start to appear around about 650°C with $\text{Bi}_{25}\text{FeO}_{39}$ already being present at about 650°C , whereas $\text{Bi}_2\text{Fe}_4\text{O}_9$ appears slightly later around 700°C – 720°C . Obviously these secondary phases are more stable with respect to pure BFO in this temperature range [47]. BFO decomposes into the mullite and selenite phases following the reaction $49\text{BiFeO}_3 \rightleftharpoons 12\text{Bi}_2\text{Fe}_4\text{O}_9 + \text{Bi}_{25}\text{FeO}_{39}$, leading to significantly higher amounts of the mullite component, which is supported by our XRD results following Rietveld refinement [48]. Besides the measurement temperature and time, also the purity of the samples has a big impact on the formation rate of the ternary phases [18]. As the maximum measured temperature in our research was 840°C , a halting of the decomposition processes as well as the reverse reaction from the secondary phases back to BiFeO_3 as proposed by Selbach *et al.* [47] could not be confirmed. However, Fig. 13 shows that at 840°C the 013 reflection of $\text{Bi}_{25}\text{FeO}_{39}$ is

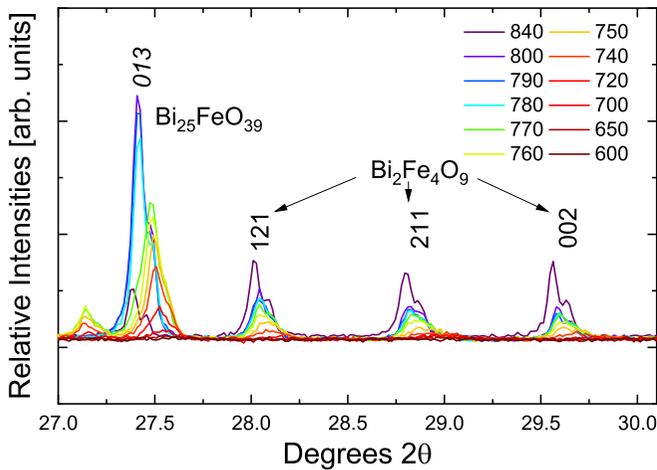


FIG. 13. Peak growth of ternary phases $\text{Bi}_{25}\text{FeO}_{39}$ (italic) and $\text{Bi}_2\text{Fe}_4\text{O}_9$ with temperature. The color gradient corresponds to the temperature change.

significantly weaker compared to the peak intensities at lower temperatures, indicating the beginning of a reduction of the presence of this component. On the other hand, all peaks of $\text{Bi}_2\text{Fe}_4\text{O}_9$ show stronger intensities even at 840° C. Thus, the thermodynamic stabilities of the two ternary phases might not be equal. The simulation results listed in Table I were done to try a fitting of the PAC results including phase fractions of $\text{Bi}_2\text{Fe}_4\text{O}_9$ with the calculated parameters. However, no phase fractions of the mullite phase were present in the PAC spectra. The appearance of the ternary phases in the XRD measurements, which were conducted after the PAC measurements, could be explained through the second heating process at the XRD measurements which leads to an increase of fractions of those phases. Visible ternary phase fractions in samples which were not exposed to a prior PAC measurement might be caused through the vigorously larger surface area of powdered samples in comparison to bulk samples used during the PAC measurements, easing the volatility of Bi and thus the building of the decomposition products.

IV. CONCLUSION

The ferro to paraelectric phase transition of multiferroic BiFeO_3 was investigated through two fundamentally different experimental methods. PAC, which is an extremely sensitive method probing the sample on an atomic scale and XRD. The results of both experimental methods agree with a phase transition from rhombohedral α -BFO in $R3c$ setting to orthorhombic β -BFO with its $Pbnm$ space group. The assumption of the Bi-site substitution by the ^{111m}Cd probe is confirmed by crystal chemical assignment (ion radii) as well as by our DFT and PAC results. Following the PAC results, the first-order phase transition is forecast by the Cd probe

ion as the preceding symmetry change of Bi-coordination is sensed. The phase transition point at 820° C is visible through a drop in quadrupole interaction frequency ω_0 which indicates the site symmetry change of the Bi position. The discordant values of ω_0 and η within the temperature interval of 807° C and 818° C could be explained by the gradual formation of a polytype transition structure or by growing disorder. In the former case, the alpha-beta phase transition would actually be a combination of a second order followed by a first-order phase transition. The first could not be evidenced in XRD. Thus, it is shown that both methods are complementary. The transition range in PAC and thermal hysteresis measured with XRD suggests a change of the long-range order parameter around the phase transition point. A broadening of the transition range at the phase transition as a result of the disturbed local environment around the probe ion itself is visible through the wider range at the PAC measurements compared to the XRD results. The visibility of the mentioned possible dynamics during the phase transition in the results over a temperature range of 19 K again underlines the precision of PAC measurements at high temperatures, which transcends the performances of other nuclear measurement methods like Mössbauer or nuclear magnetic resonance spectroscopy in this temperature range. This is also a very fundamental result as we have proof of the local change of the order parameter caused by defects, which was often assumed in literature before [4]. Following the XRD results, the phase transition happens at a transition temperature of 818° C as this is the only temperature point where both phases appear simultaneously. Indications of a polytype transition structure could not be found in our XRD data but its existence cannot be excluded. The appearance of the ternary phases $\text{Bi}_2\text{Fe}_4\text{O}_9$ and $\text{Bi}_{25}\text{FeO}_{39}$ was observed. They only arise in powdered samples of high surface area as were used for XRD and are not contained in any PAC data.

ACKNOWLEDGMENTS

We acknowledge the financial support received from the Federal Ministry of Education and Research (BMBF) through Grants No. 05K16PGA and No. 05K16SI1. We also acknowledge the support of all the technical teams at ISOLDE for their excellent work in delivering high-quality beams for TDPAC measurements. We acknowledge the support of the European Union's Horizon 2020 Framework research and innovation program under Grant Agreement No. 654002 (ENSAR2) given to the ISOLDE experiment IS647. We thank the Foundation for Science and Technology FCT, Portugal, via Grant No. CERN-FIS- PAR-0005-2017, and Dr. J. G. M. Correia for technical assistance during the beam time. Last but not least, we would like to express special thanks to the insightful thoughts and comments of the reviewers which have lead to an interesting discussion and revision of our paper.

- [1] H. Schmid, *Ferroelectrics* **162**, 317 (1994).
 [2] N. A. Spaldin and M. Fiebig, *Science* **309**, 391 (2005).
 [3] H. Bea, M. Gajek, M. Bibes, and A. Barthélemy, *J. Phys.: Condens. Matter* **20**, 434221 (2008).

- [4] G. Catalan and J. F. Scott, *Adv. Mater.* **21**, 2463 (2009).
 [5] R. Haumont, J. Kreisel, P. Bouvier, and F. Hippert, *Phys. Rev. B* **73**, 132101 (2006).

- [6] R. Palai, R. S. Katiyar, H. Schmid, P. Tissot, S. J. Clark, J. Robertson, S. A. T. Redfern, G. Catalan, and J. F. Scott, *Phys. Rev. B* **77**, 014110 (2008).
- [7] R. Haumont, I. A. Kornev, S. Lisenkov, L. Bellaiche, J. Kreisel, and B. Dkhil, *Phys. Rev. B* **78**, 134108 (2008).
- [8] I. A. Kornev, S. Lisenkov, R. Haumont, B. Dkhil, and L. Bellaiche, *Phys. Rev. Lett.* **99**, 227602 (2007).
- [9] D. C. Arnold, K. S. Knight, F. D. Morrison, and P. Lightfoot, *Phys. Rev. Lett.* **102**, 027602 (2009).
- [10] S. M. Selbach, T. Tybell, M.-A. Einarsrud, and T. Grande, *Adv. Mater.* **20**, 3692 (2008).
- [11] J. Wei, C. Wu, T. Yang, Z. Lv, Z. Xu, D. Wang, R. Haumont, and Z. Cheng, *J. Phys. Chem. C* **123**, 4457 (2019).
- [12] R. Catherall, W. Andreatza, M. Breitenfeldt, A. Dorsival, G. Focker, T. Gharsa, T. Giles, J. Grenard, F. Locci, P. Martins, S. Marzari, J. Schipper, A. Shornikov, and T. Stora, *J. Phys. G* **44**, 094002 (2017).
- [13] K. Johnston, J. Schell, J. Correia, M. Deicher, H. Gunnlaugsson, A. Fenta, E. David-Bosne, A. Costa, and D. C. Lupascu, *J. Phys. G* **44**, 104001 (2017).
- [14] J. Landers, S. Salamon, M. Escobar Castillo, D. Lupascu, and H. Wende, *Nano Lett.* **14**, 6061 (2014).
- [15] T. Karthik, T. D. Rao, A. Srinivas, and S. Asthana, [arXiv:1206.5606](https://arxiv.org/abs/1206.5606).
- [16] G. Arlt and H. Neumann, *Ferroelectrics* **87**, 109 (1988).
- [17] M. E. Lines and A. M. Glass, *Principles and Applications of Ferroelectrics and Related Materials* (Oxford University Press, New York, 2001).
- [18] M. Valant, A.-K. Axelsson, and N. Alford, *Chem. Mater.* **19**, 5431 (2007).
- [19] J. Schell, M. Schmuck, I. Efe, T. Thanh Dang, J. Gonçalves, M. Escobar C., V. V. Shvartsman, A. R. Granadeiro Costa, U. Köster, R. Vianden, D. Noll, and C. Lupascu (unpublished).
- [20] L. Volkova and D. Marinin, *J. Supercond. Novel Magn.* **24**, 2161 (2011).
- [21] J. Ziegler, SRIM-2013 software package, <http://www.srim.org>.
- [22] M. Jäger, K. Iwig, and T. Butz, *Rev. Sci. Instrum.* **82**, 065105 (2011).
- [23] T. Wichert and E. Recknagel, *Microscopic Methods in Metals* (Springer, New York, 1986), pp. 317–364.
- [24] J. Schell, P. Schaaf, and D. C. Lupascu, *AIP Adv.* **7**, 105017 (2017).
- [25] R. Nédélec, Rückstoßimplantation von ^{48}Cr und erste PAC Messungen mit der Sonde $^{48}\text{Cr}/^{48}\text{V}$, Ph.D. thesis, Diplomarbeit, Universität Bonn, 2002.
- [26] E. Gerdau, J. Wolf, H. Winkler, and J. Braunsfurth, *Proc. R. Soc. Lond. A* **311**, 197 (1969).
- [27] P. Friedsam, Anorganische nichtmetallische Festkörper ohne Translationssymmetrie untersucht mit der Methode der gestörten γ - γ -Winkelkorrelation, Ph.D. thesis, Dissertation, Universität Bonn, 2000.
- [28] J. Bartels, Fremdatome in Silizium-Kavitäten untersucht mit der gestörten Winkelkorrelation, Ph.D. thesis, Dissertation, Universität Bonn, 2002.
- [29] J. Schell, D. C. Lupascu, J. G. Martins Correia, A. W. Carbonari, M. Deicher, M. B. Barbosa, R. D. Mansano, K. Johnston, I. S. Ribeiro, and ISOLDE collaboration, *Hyperfine Interact.* **238**, 2 (2016).
- [30] G. Kresse and J. Furthmüller, *Phys. Rev. B* **54**, 11169 (1996).
- [31] J. N. Gonçalves (private communication, 2018).
- [32] H. Haas, S. P. Sauer, L. Hemmingsen, V. Kell, and P. W. Zhao, *Europhys. Lett.* **117**, 62001 (2017).
- [33] J. Gebhardt and A. M. Rappe, *Phys. Rev. B* **98**, 125202 (2018).
- [34] J.-M. Moreau, C. Michel, R. Gerson, and W. J. James, *J. Phys. Chem. Solids* **32**, 1315 (1971).
- [35] J. G. Correia for GFIT19 and N. P. Barradas for NNFIT programs (PAC MANual Lisbon 1992).
- [36] N. P. Barradas, M. Rots, A. A. Melo, and J. C. Soares, *Phys. Rev. B* **47**, 8763 (1993).
- [37] M. Foraker, *Nucl. Instrum. Methods* **106**, 121 (1973).
- [38] S. Prosandeev, D. Wang, W. Ren, J. Íñiguez, and L. Bellaiche, *Adv. Funct. Mater.* **23**, 234 (2013).
- [39] R. T. Shannon and C. T. Prewitt, *Acta Crystallogr. Sect. B* **25**, 925 (1969).
- [40] M. E. Lines and A. M. Glass, *Principles and Applications of Ferroelectrics and Related Materials* (Oxford University Press, New York, 2001).
- [41] G. Oliveira, R. Teixeira, R. Moreira, J. Correia, J. Araújo, and A. Lopes, *Sci. Rep.* **10**, 1 (2020).
- [42] T. M. Rearick, G. L. Catchen, and J. M. Adams, *Phys. Rev. B* **48**, 224 (1993).
- [43] M. Bellakki and V. Manivannan, *J. Mater. Sci.* **45**, 1137 (2010).
- [44] B. Ishaq, G. Murtaza, S. Sharif, M. A. Khan, N. Akhtar, I. Will, M. Saleem, and S. M. Ramay, *Results Phys.* **6**, 675 (2016).
- [45] W. Sato, S. Komatsuda, H. Shimizu, R. Moriichi, S. Abe, S. Watanabe, S. Komatsu, T. Terai, S. Kawata, and Y. Ohkubo, *Phys. Rev. B* **100**, 184111 (2019).
- [46] A. Perejon, P. E. Sanchez-Jimenez, J. M. Criado, and L. A. Perez-Maqueda, *J. Phys. Chem.* **118**, 26387 (2014).
- [47] S. M. Selbach, M.-A. Einarsrud, and T. Grande, *Chem. Mater.* **21**, 169 (2008).
- [48] T. Carvalho and P. Tavares, *Mater. Lett.* **62**, 3984 (2008).