

# First-principles Hubbard $U$ and Hund's $J$ corrected approximate density functional theory predicts an accurate fundamental gap in rutile and anatase $\text{TiO}_2$

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Titanium dioxide ( $\text{TiO}_2$ ) presents a long-standing challenge for approximate Kohn-Sham density functional theory (KS-DFT), as well as to its Hubbard-corrected extension,  $\text{DFT}+U$ . We find that a previously proposed extension of first-principles  $\text{DFT}+U$  to incorporate a Hund's  $J$  correction, termed  $\text{DFT}+U+J$ , in combination with parameters calculated using a recently proposed linear-response theory, predicts fundamental band gaps that are accurate to well within the experimental uncertainty in rutile and anatase  $\text{TiO}_2$ . Our approach builds upon established findings that Hubbard correction of both the titanium  $3d$  and oxygen  $2p$  subspaces in  $\text{TiO}_2$ , symbolically giving  $\text{DFT}+U^{d,p}$ , is necessary to achieve acceptable band gaps using  $\text{DFT}+U$ . This requirement remains when the first-principles Hund's  $J$  is included. We also find that the calculated gap depends on the correlated subspace definition even when using subspace-specific first-principles  $U$  and  $J$  parameters. Using the simplest reasonable correlated subspace definition and underlying functional, the local density approximation, we show that high accuracy results from using a relatively uncomplicated form of the  $\text{DFT}+U+J$  functional. For closed-shell systems such as  $\text{TiO}_2$ , we describe how various  $\text{DFT}+U+J$  functionals reduce to  $\text{DFT}+U$  with suitably modified parameters, so that reliable band gaps can be calculated for rutile and anatase with no modifications to a conventional  $\text{DFT}+U$  code.

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## I. INTRODUCTION

Titanium dioxide ( $\text{TiO}_2$ ) has been widely used for several decades in diverse industrial applications such as pigmentation and coating [1–3] due to its nontoxicity, low-cost production, and thermal stability.  $\text{TiO}_2$  came under particularly intense scrutiny beginning with the ground-breaking work of Fujishima and Honda, who demonstrated water splitting in  $\text{TiO}_2$  photochemical cells in the ultraviolet (UV) spectral range in 1972 [4]. Since then,  $\text{TiO}_2$ -based structures have been engineered for diverse optoelectronic applications such as photocatalysts, photovoltaics, sensors, and for energy and environmental applications [5–7]. In nature,  $\text{TiO}_2$  has three common polymorphs: rutile, anatase, and brookite [8].  $\text{TiO}_2$ -rutile and  $\text{TiO}_2$ -anatase are more common in industrial applications, as brookite is less stable and difficult to synthesize in large volumes [9]. The electronic structures of pristine  $\text{TiO}_2$ -rutile and  $\text{TiO}_2$ -anatase have been extensively studied experimentally [10–14], and the most reliable data currently available shows that  $\text{TiO}_2$ -rutile and  $\text{TiO}_2$ -anatase have fundamental (electronic, not optical) band gaps of 3.03 eV [12,13] and 3.47 eV [11], respectively.

First-principles simulations can provide valuable insights into the processes at play in  $\text{TiO}_2$ -based systems, offering clues for the engineering of these systems for desired applications. This requires the accurate description of their electronic structures in the region of their band edges, naturally, and this must necessarily be found by means of computationally feasible and scalable methods if disordered structures and

diverse dopants are to be assessed in any detail. There exist numerous acceptably reliable approaches, such as quantum chemistry methods [15,16], hybrid-functionals [17], and many-body perturbation theory [18–20], but these methods are too computationally demanding for routine application to defective and disordered systems.

Density functional theory (DFT) [21], specifically Kohn-Sham DFT (KS-DFT) [22] using (semi)local density exchange-correlation functionals [22–25] offers a computationally feasible framework to study the electronic structures of spatially complex  $\text{TiO}_2$ -based systems. In the present work, with that challenge in mind, we use a linear-scaling implementation of DFT, the Order-N Electronic Total Energy Package (ONETEP) [26–29]. However, it is well known that (semi)local KS-DFT is unable to capture the approximate magnitude of the band gap of  $\text{TiO}_2$ , a common feature of transition-metal oxides (TMOs) generally [30–32], and so it requires, at the very least, some corrective measures for reliable use.

In this work, we revisit the computationally efficient approach of applying Hubbard-model inspired corrections to approximate KS-DFT, namely  $\text{DFT}+U$  [33–41] which is technically a generalized Kohn-Sham method [42], in terms of its capability of accurately describing the fundamental electronic band gap of  $\text{TiO}_2$  polymorphs. We find that unlike-spin Hund's  $J$  correction, specifically that introduced in the pioneering work of Ref. [43], is the key ingredient that enables the band gaps of  $\text{TiO}_2$  to be accurately described with this method. A corrective functional is only as good as its parameters, and here we use the recently proposed minimum-tracking linear-response formalism of Ref. [44] for calculating them. Encouragingly for practical use, moreover, we find that

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for closed-shell (non-spin-polarized) systems such as pristine  $\text{TiO}_2$  and other TMOs towards the edges of the periodic table  $d$  block, no modification to a standard DFT+ $U$  code is needed to include Hund's  $J$  corrections.

No differently to what has been found in previous works [45–48] and as an inevitable consequence of the O  $2p$  character of the valence-band edge, in order to achieve significantly improved results using DFT+ $U$  we need to apply corrective potentials to oxygen  $2p$  orbitals on the same footing as to titanium  $3d$  orbitals. The addition of Hund's  $J$  does not change this fact, we denote this two-species correction as DFT+ $U^{d,p}$ , short for DFT+ $U^d+U^p$ , following the literature. Unlike prior works on  $\text{TiO}_2$ , in which one or both of  $U^d$  and  $U^p$  was found to require empirical tuning for good results, in this work we only use first-principles calculated  $U$  and  $J$  parameters (specifically, by means of the minimum-tracking linear-response method [44,49]), for both the Ti  $3d$  and O  $2p$  subspaces.

When the unlike-spin Hund's  $J$  term is included (using a particularly simple form of DFT+ $U+J$ , in agreement with the detailed analysis of Ref. [43]) we predict a generalized Kohn-Sham band gap of a better quality than that which hybrid functionals or  $G_0W_0$  give, for both polymorphs, when gauged against reported experimental findings (recent, high-quality ones in the case of anatase, where it seems to be more challenging to measure). The ionic geometries of both polymorphs were found to be very little affected by the force terms due to this functional form. We note in passing that both functional classes, DFT+ $U$  and hybrids, are differentiable in terms of the density matrix and have a nonlocal potential, and so their generalized Kohn-Sham gaps include exchange-correlation derivative discontinuities [50] and are directly comparable to experiment. Promisingly for future  $\text{TiO}_2$  simulation, and as the central conclusion of this work, we find that the same first-principles DFT+ $U^{d,p}+J^{d,p}$  functional predicts the experimental fundamental gap to within the uncertainty of the experiment, for both polymorphs.

## II. METHODOLOGY

Perhaps the most well-known systematic error exhibited by conventional approximate functionals in KS-DFT is the self-interaction error (SIE) [51–55], and its many-body generalization, the delocalization error [56–62]. SIE arises due to spurious self-repulsion of electronic density in the KS-DFT formalism and it also persists, albeit often to a lesser extent, within generalized Kohn-Sham schemes. While the origins of SIE are well understood, it is hard to avoid it in the construction of closed-form approximate functionals. SIE leads to the well-known significant, even drastic underestimation of fundamental band gaps of TMOs in particular [30–32], and  $\text{TiO}_2$ -rutile and  $\text{TiO}_2$ -anatase are no exception in this regard [63]. Less well understood is the generalization of SIE to account for the spin degree of freedom, which is not necessarily less relevant in closed-shell systems where the spin happens to evaluate to zero. In this section, we outline in detail our methodology for computing and incorporating parameters, the Hubbard  $U^{d,p}$  for density-related error and Hund's  $J^{d,p}$  for spin-related error, to correct a very low-cost density functional for the specific case of  $\text{TiO}_2$ .

### A. DFT+ $U+J$ functionals and their simplification for closed-shell systems

DFT+ $U$  is routinely applied to correct for SIE, particularly for the spurious delocalization of electronic states associated with transition-metal  $3d$  orbitals. The DFT+ $U$  total energy is given by

$$E_{\text{DFT}+U} = E_{\text{DFT}} + E_U, \quad (1)$$

where the rotationally invariant form of  $E_U$  for a given SIE-prone subspace [30,38,64], particularly if we take its relatively recent DFT+ $U+J$  form of Ref. [43], is given by

$$E_U[\{n^\sigma\}] = \frac{1}{2} \sum_{\sigma} \sum_{m,m'} \left\{ \underbrace{U[n_{mm'}^\sigma \delta_{m'm} - n_{mm'}^\sigma n_{m'm}^\sigma]}_{\text{I}} - \underbrace{J[n_{mm'}^\sigma \delta_{m'm} - n_{mm'}^\sigma n_{m'm}^\sigma]}_{\text{II}} + \underbrace{J[n_{mm'}^\sigma n_{m'm}^\sigma]}_{\text{III}} - \underbrace{2J[\delta^{\sigma\sigma_{\min}} n_{mm'}^\sigma \delta_{m'm}]}_{\text{IV}} \right\}. \quad (2)$$

Here,  $\sigma$  is a spin index,  $\bar{\sigma}$  is the corresponding opposite spin,  $\sigma_{\min}$  is the index of the minority-population spin channel for the subspace at hand,  $n_{mm'}$  is the subspace-projected KS density matrix. The Hubbard  $U$  is, in this work at least, interpreted as the subspace-and-spin-averaged net Hartree-plus-exchange-correlation interaction. Hund's  $J$  is its spin-splitting counterpart. We will presently detail what, precisely, is meant by spin averaging and spin splitting in this context.

The choice of appropriate form of DFT+ $U(+J)$  energy functional depends on various factors such as the system under consideration, the limitations and robustness of approaches to determine the Hubbard  $U$  and Hund's  $J$  parameters, and the underlying approximate density functional. For instance, it was argued in Ref. [43] that term (IV), which we dub the “minority spin term”, is best not included, as it arises due to the double-counting correction of a type of two-particle density-matrix interaction that is unlikely to be very much present in the underlying density functional. Our numerical results supports this analysis. It was furthermore found to lead to numerical instabilities, and we have also noted this effect in our own calculations. Our tentative explanation of this instability is that, when the net spin of a site is weak, the potential arising due to this term can switch over discretely from one spin channel to the other. The simplest functional form is achieved, of course, by neglecting the explicit correction of exchange and effectively by setting  $J = 0$  eV. If a value for  $J$  is available, then so is the Dudarev functional [33], which includes only like-spin correction terms [the terms (I) and (II)] via an effective parameter,  $U_{\text{eff}} = U - J$ , resulting symbolically in DFT+ $U_{\text{eff}}$ .

Inspired by the Dudarev model, we note and primarily use in this work the fact that the full DFT+ $U+J$  functional of Eq. (2) may be applied to closed-shell systems, without approximation, using an unmodified DFT+ $U$  code with no  $J$  implementation. To see this clearly, we can rearrange Eq. (2) and introduce an additional parameter  $\alpha$ , which is exactly that  $\alpha$  which is available and used to calculate the Hubbard  $U$  in many standard DFT+ $U$  codes [40]. Here, it captures the

inclusion or neglect of the minority spin term (term IV), when rewriting Eq. (2) as

$$E_U = \sum_{\sigma, m, m'} \left\{ \frac{U_{\text{full}}}{2} [n_{mm'}^{\sigma} \delta_{m'm} - n_{mm'}^{\sigma} n_{m'm}^{\sigma}] + \alpha n_{mm'}^{\sigma} \delta_{m'm} \right\}, \quad (3)$$

where  $U_{\text{full}} = U - 2J$ . Three reasonable options for  $\alpha$  are tested in this study, representing different interpretations of the minority spin (term IV).

(1) The most natural treatment of term IV for closed-shell systems, that suggested in Ref. [43], is to interpret  $\sigma_{\text{min}} = \sigma$ , such that  $\delta^{\sigma\sigma_{\text{min}}} = 1$ . This requires us to set  $\alpha = -J/2$ .

(2) A modification of the latter, intended to avoid the implied discontinuity in the total energy at the onset of nonzero spin polarization (it doesn't avoid such a discontinuity in the potential), is to "share" the minority spin term between the two spins, setting  $\delta^{\sigma\sigma_{\text{min}}} = 1/2$  for closed-shell systems. This leads to  $\alpha = 0$  and the resulting Hubbard functional is simply a Dudarev functional with  $U_{\text{full}} = U - 2J$ .

(3) In the last case, the minority spin term is neglected, as it was argued that it is best to do in its originating Ref. [43], by setting  $\delta^{\sigma\sigma_{\text{min}}} = 0$ . For closed-shell systems, DFT+ $U$ + $J$  is then recovered by DFT+ $U$  code with parameters  $U_{\text{full}}$  and  $\alpha = J/2$ .

In this work, we test these different corrective functionals by application to both the Ti 3d and O 2p subspaces of TiO<sub>2</sub>, presenting DFT+ $U^d$  (no O 2p correction) results only for the sake of illustration and completeness. It has previously been comprehensively demonstrated in Ref. [65], that it is not possible to reconcile a reasonable band gap with reasonable lattice constants when applying DFT+ $U$  only to Ti 3d subspaces in TiO<sub>2</sub>. We further motivate the inclusion of O 2p corrections in Appendix A and with reference to Fig. 3. In the Supplemental Material that accompanies this work [66], we illustrate that the favoured DFT+ $U$ + $J$  functional (minority spin term neglected) has only a very small effect on the lattice constants and internal ionic geometries predicted for both polymorphs by the underlying functional. There, we also specify the computational parameters of our study in detail.

### B. The minimum-tracking linear-response approach for first-principles Hubbard $U$ and Hund's $J$ parameters

The results of DFT+ $U^{d,p}$  are only as good as its input Hubbard  $U$  and Hund's  $J$  parameters. Finite-difference linear-response theory provides a practical, widely available first-principles method for calculating these [39,40,43]. It has been found that linear response tends to give Hubbard  $U$  parameters for closed-shell systems that are too high for practical use, and this is usually deemed to be an erroneous overestimation [44,67–69]. The present work provides hints that these values may be correct after all, but that Hund's  $J$  effectively reduces them and so the latter is (counterintuitively, perhaps) more important to include in closed-shell systems. If a system has zero spin polarization, the systematic error in the approximate functional related to the spin *degree of freedom* may still be large. In this work, we employed the recently introduced minimum-tracking variant [49] of linear response as implemented in the ONETEP DFT+ $U$  implementation [26,70],

and in particular, its spin-specific extension introduced in Ref. [44]. The "scaled  $2 \times 2$ " method was used here to evaluate the Hubbard  $U$ , Hund's  $J$ , and effective Hubbard  $U$  parameters ( $U_{\text{eff}} = U - J$  and  $U_{\text{full}} = U - 2J$ ) for the Ti 3d and O 2p subshells of pristine TiO<sub>2</sub>-rutile and TiO<sub>2</sub>-anatase using

$$U = \frac{1}{2} \frac{\lambda_U (f^{\uparrow\uparrow} + f^{\downarrow\downarrow}) + f^{\uparrow\downarrow} + f^{\downarrow\uparrow}}{\lambda_U + 1}, \quad (4)$$

and

$$J = -\frac{1}{2} \frac{\lambda_J (f^{\uparrow\uparrow} - f^{\downarrow\downarrow}) + f^{\uparrow\downarrow} - f^{\downarrow\uparrow}}{\lambda_J - 1}, \quad (5)$$

where

$$\lambda_U = \frac{\chi^{\uparrow\uparrow} + \chi^{\uparrow\downarrow}}{\chi^{\downarrow\uparrow} + \chi^{\downarrow\downarrow}}, \quad \text{and} \quad \lambda_J = \frac{\chi^{\uparrow\uparrow} - \chi^{\uparrow\downarrow}}{\chi^{\downarrow\uparrow} - \chi^{\downarrow\downarrow}}, \quad (6)$$

and where the projected interacting response matrices are given by  $\chi^{\sigma\sigma'} = dn^{\sigma}/dv_{\text{ext}}^{\sigma'}$ . The spin-dependent interaction strengths  $f^{\sigma\sigma'}$  are calculated by solving a  $2 \times 2$  matrix equation given by

$$f = \left[ \left( \frac{\delta v_{\text{KS}}}{\delta v_{\text{ext}}} - 1 \right) \left( \frac{\delta n}{\delta v_{\text{ext}}} \right)^{-1} \right], \quad (7)$$

for which matrix entities are obtained by linear fitting to small changes of the subspace occupancies  $\delta n^{\sigma}$  and subspace-averaged Kohn-Sham potentials  $\delta v_{\text{KS}}^{\sigma}$  with respect to incrementally varying uniform perturbing potentials  $\delta v_{\text{ext}}^{\sigma}$  on the targeted subspaces. These definitions are equivalent to a particular choice of perturbation in the more physically transparent but perturbation-independent expressions,

$$U = \frac{d(v_{\text{Hxc}}^{\uparrow} + v_{\text{Hxc}}^{\downarrow})}{2d(n^{\uparrow} + n^{\downarrow})} \quad \text{and} \quad J = -\frac{d(v_{\text{Hxc}}^{\uparrow} - v_{\text{Hxc}}^{\downarrow})}{2d(n^{\uparrow} - n^{\downarrow})}, \quad (8)$$

where the factor 1/2 signifies averaging (or halving of the splitting between) the subspace averaged Hartree-plus-exchange-correlation potentials  $v_{\text{Hxc}}^{\sigma}$ . Equation (8) can be taken as the definition of minimum-tracking linear response, and if using its two parts separately it is natural to use  $\delta v_{\text{ext}}^{\uparrow} = \delta\alpha = \delta v_{\text{ext}}^{\downarrow}$  for  $U$  and  $\delta v_{\text{ext}}^{\uparrow} = \delta\beta = -\delta v_{\text{ext}}^{\downarrow}$  for  $J$ .

The scaling factors become  $\lambda_U = 1$  and  $\lambda_J = -1$  for spin-unpolarized systems such as the pristine TiO<sub>2</sub>-rutile and TiO<sub>2</sub>-anatase. This reflects the vanishing linear coupling between subspace occupancy and magnetization in such systems. As a result, the "scaled  $2 \times 2$ " method reduces to the "simple  $2 \times 2$ " method [44], which can be summarized as  $U = (f^{\sigma\bar{\sigma}} + f^{\sigma\sigma})/2$ ,  $J = (f^{\sigma\bar{\sigma}} - f^{\sigma\sigma})/2$  (this gives a Dudarev  $U_{\text{eff}} = f^{\sigma\sigma}$ , which is reasonable for a like-spin-only corrective functional). In fact, time-reversal symmetry can be readily exploited for closed-shell systems, where it is sufficient to perturb one spin channel only, filling in half of the matrix elements by symmetry, e.g.,  $\chi^{\uparrow\uparrow} = \chi^{\downarrow\downarrow}$ . This feature of the  $2 \times 2$  approach enabled the *simultaneous* calculation of  $U$  and  $J$  in this work, from a single group of self-consistent calculations perturbing one spin channel only by finite differences. We have verified numerically that Eq. (8) provides the same results under these conditions. The response matrix elements coupling Ti 3d and O 2p subspaces are not projected out, as to include such entries in the response matrices would necessitate corresponding terms in the corrective functional

TABLE I. First-principles LDA-appropriate Hubbard  $U$  and Hund's  $J$  parameters calculated using the minimum-tracking linear-response method [44,49], both for the Ti  $3d$  and O  $2p$  subspaces of TiO<sub>2</sub>-rutile. The Ti  $3d$  parameters depend significantly on the pseudoatomic solver charge configuration used to construct the corresponding DFT+ $U$  subspace, with 3+ providing a significantly more localized subspace and consequently higher parameters. Shown also are the effective Hubbard  $U$  parameter of the Dudarev model ( $U_{\text{eff}}$ ) and that which enables us to reproduce DFT+ $U+J$  for closed-shell systems ( $U_{\text{full}}$ ).

LDA rutile	Ti <sup>0</sup> state		Ti <sup>3+</sup> state	
	Ti	O	Ti	O
$U$	3.56	8.57	5.59	8.57
$J$	0.29	0.92	0.38	0.89
$U_{\text{eff}} = U - J$	3.27	7.66	5.20	7.68
$U_{\text{full}} = U - 2J$	2.98	6.74	4.82	6.80

(these are usually called + $V$ ), which would complicate our analysis focused on Hund's  $J$ .

### III. RESULTS AND DISCUSSION

We first present the calculated Hubbard  $U$  and Hund's  $J$  parameters for pristine, closed-shell TiO<sub>2</sub>-rutile and TiO<sub>2</sub>-anatase. As a preliminary test, LDA-appropriate parameters were calculated for TiO<sub>2</sub>-rutile with two different definitions of the DFT+ $U$  target subspace for Ti  $3d$  orbitals. Specifically, both neutral and 3+ (still non-spin-polarized) atomic DFT calculations were separately performed using functionality available in ONETEP and described in Ref. [71], to generate pseudoatomic orbitals to define the  $3d$  subspace, and also to build the initial density and NGWF guesses. The tensorial representation [72] was used to correctly account for the slight nonorthogonality among the orbitals for a given subspace, which arises due to their sampling in the ONETEP plane-wave-like basis. An OPIUM [73] norm-conserving pseudopotential with a 3+ reference state was used for Ti, while a charge-neutral atomic configuration was used for O (OPIUM pseudopotential generation, DFT+ $U$  definition, and initial density and NGWF guess generation) throughout. The resulting Hubbard  $U$  and Hund's  $J$  parameters are summarized in Table I.

We find that the calculated LDA Hubbard  $U$  value for Ti  $3d$  increases by  $\sim 2$  eV or  $\sim 60\%$  when going from a neutral subspace configuration to a 3+ charge one, due to the pronounced increase in the spatial localization of the subspace, plotted in Fig. 3 of Appendix B. The relatively small calculated  $J$  value also increases somewhat, by a smaller amount in multiplicative terms, 30%.  $U_{\text{full}} = U - 2J$  therefore also increases by  $\sim 60\%$ . We choose the smoother orbitals from the neutral pseudoatomic solver configuration to define DFT+ $U$  in our further calculations, and the reasoning for this will be discussed and demonstrated in Appendix B. There, we will see that, not only does calculating  $U$  and  $J$  from first principles not compensate for the arbitrariness of the DFT+ $U$  projectors in TiO<sub>2</sub>-rutile, it in fact *reinforces* it. We note a small but nonetheless irksome deviation in the O  $2p$   $J$  parameter when moving to a 3+ Ti  $3d$  NGWF initial guess, which results from

TABLE II. First-principles LDA-appropriate Hubbard  $U$  and Hund's  $J$  parameters calculated using the minimum-tracking linear-response method [44,49], both for the Ti  $3d$  and O  $2p$  subspaces of TiO<sub>2</sub>-anatase. Only the neutral pseudoatomic solver configuration Ti<sup>0</sup> is used here. Shown also are the effective Hubbard  $U$  parameter of the Dudarev model ( $U_{\text{eff}}$ ) and that which enables us to reproduce DFT+ $U+J$  for closed-shell systems ( $U_{\text{full}}$ ).

LDA anatase	Ti	O
$U$	3.57	8.56
$J$	0.29	0.91
$U_{\text{eff}} = U - J$	3.28	7.66
$U_{\text{full}} = U - 2J$	3.00	6.75

poorer convergence characteristics when those functions are initialized with excessive localization.

Turning next to the LDA-appropriate Hubbard  $U$  and Hund's  $J$  parameters calculated for TiO<sub>2</sub>-anatase using the same method with a neutral Ti  $3d$  subspace definition, shown in Table II, we note a remarkable degree of similarity with the TiO<sub>2</sub>-rutile values. In fact, the differences are within the noise of the linear-response method, and this reflects the similar LDA charge states (to well within 1% for both the Ti  $3d$  and O  $2p$  DFT+ $U$  subspaces) and coordination chemistry in the two structures.

#### A. The first-principles band gap of pristine TiO<sub>2</sub>-rutile

As a generalized Kohn-Sham theory with an differentiable density-matrix dependence, in the same way that hybrid functionals are [50], the Kohn-Sham gap of DFT+ $U$  (or DFT+ $U+J$ ) includes an explicit derivative discontinuity. The relationship between the Kohn-Sham gap and the fundamental gap is thereby not only assured in principle, but the derivative discontinuity gives, in practice, the opportunity for direct comparability to the experimental insulating gap. Shown in Table III is the band gap of TiO<sub>2</sub>-rutile calculated using LDA and first-principles DFT+ $U$ , DFT+ $U_{\text{eff}}$ , DFT+ $U_{\text{full}}$  with different  $\alpha$  values, and explicit DFT+ $U+J$  [minority spin term (IV) neglected], both when applied only to the Ti  $3d$  subshell and when applied also to the O  $2p$  subshell.

Experimental, first-principles, semiempirical hybrid,  $GW$  results, and several previous DFT+ $U$  results from the literature are also shown in Table III, for comparison.

The experimental direct gap quoted [12,13] is based on absorption, photoluminescence, and resonant-Raman scattering data, and is expected to be very reliable due to the relatively small exciton binding and phonon coupling effects in rutile [11], and moreover in light of its good agreement with available inverse photoemission data [74].

The LDA yields a Kohn-Sham band gap of 1.96 eV, much lower than the experimental band gap of 3.03 eV, as expected given its absence of a derivative discontinuity. Regardless of the Hund's  $J$  incorporation scheme used, and as is generally attested in the literature on calculations with  $J = 0$  eV, first-principles DFT+ $U$  applied to Ti  $3d$  states only performs poorly and here predicts a band gap of 2.17–2.24 eV.

The inadequacy of the conventional DFT+ $U$  subspace definition can be explained by comparing the very different

TABLE III. The fundamental band gap (in eV) of  $\text{TiO}_2$ -rutile calculated within DFT(LDA), DFT+ $U$  with Hund's  $J$  neglected, when treated within the Dudarev model ( $U_{\text{eff}}$ ), and when treated in a manner which fully reproduces DFT+ $U+J$  using only DFT+ $U$  code for closed-shell systems ( $U_{\text{full}}$ ), both when treated with ( $\alpha = -J/2$ ) and without ( $\alpha = J/2$ ) its minority-spin (term IV). DFT+ $U^d$  and DFT+ $U^{d,p}$  results are separately shown, using parameters calculated from first principles using the minimum-tracking linear-response method, using only the neutral pseudo-atomic solver configuration  $\text{Ti}^0$ . Prior experimental, first-principles local, semi-local, meta-generalized-gradient, and semi-empirical hybrid functional; perturbative  $G_0W_0$ ; empirical; first-principles SCF linear-response (Ref. [82]), and ACBN0 (Ref. [83]) DFT+ $U$  values are provided for convenient comparison. Our central results are highlighted in bold.

$\text{TiO}_2$ -rutile $E_{\text{gap}}$		
DFT (LDA)	1.96	
	+ $U^d$	+ $U^{d,p}$
DFT+ $U$	2.24	3.59
DFT+ $U_{\text{eff}} = U - J$	2.21	3.38
DFT+ $U_{\text{full}} = U - 2J, \alpha = -J/2$	2.17	3.32
DFT+ $U_{\text{full}} = U - 2J$	2.18	3.18
DFT+ $U_{\text{full}} = U - 2J, \alpha = J/2$	2.20	<b>3.04</b>
DFT+ $U+J$ (no minority spin term)	2.20	<b>3.04</b>
Experiment [12,13]	<b>3.03</b>	
LDA [48]	1.79	
PBE [19]	1.88	
PBE [75]	1.86	
PBE [76]	1.77	
TB-mBJ [77]	2.60	
SCAN [78]	2.23	
HSE06 [79]	3.3	
HSE06 [19]	3.39	
HSE06 ( $\alpha = 0.2$ ) [76]	3.05	
sX Hybrid [75]	3.1	
LDA+ $G_0W_0$ [18]	3.34	
PBE+ $G_0W_0$ [19]	3.46	
HSE+ $G_0W_0$ [19]	3.73	
DFT+ $U$ ( $U=7.5$ eV) [80]	2.83	
DFT+ $U$ ( $U=10$ eV) [81]	2.97	
DFT+ $U^d$ ( $U = 3.25$ eV) [82]	2.01	
DFT+ $U^{d,p}$ ( $U^d = 3.25$ eV, $U^p = 10.65$ eV) [82]	3.67	
DFT+ $U^{d,p}$ ( $U^d = 3.25$ eV, $U^p = 5.0$ eV) [82]	2.69	
DFT+ $U^{d,p}$ ( $U^d = 0.15$ eV, $U^p = 7.34$ eV) [83]	2.83	

valence and the conduction band edge characters seen in all of the local density of states plots shown in Fig. 1, and additionally motivated by recalling the very similar degree of spatial localization of Ti 3d and O 2p atomic orbitals (see Fig. 3). The valence (conduction) band edge is left almost unaffected by applying the Hubbard correction only to the Ti 3d (O 2p) subshell, for any reasonable Hubbard  $U$  parameter (hence, unreasonable values have been tested in the prior literature). In qualitative agreement with that, we observe that the impact of the method on the band gap increases substantially as soon as correction is also applied to both subshells, within DFT+ $U^{d,p}$  (as we show in detail in Table III and now discuss).

Focusing on our own first-principles DFT+ $U^{d,p}$  results and comparing with experiment, we find that when the

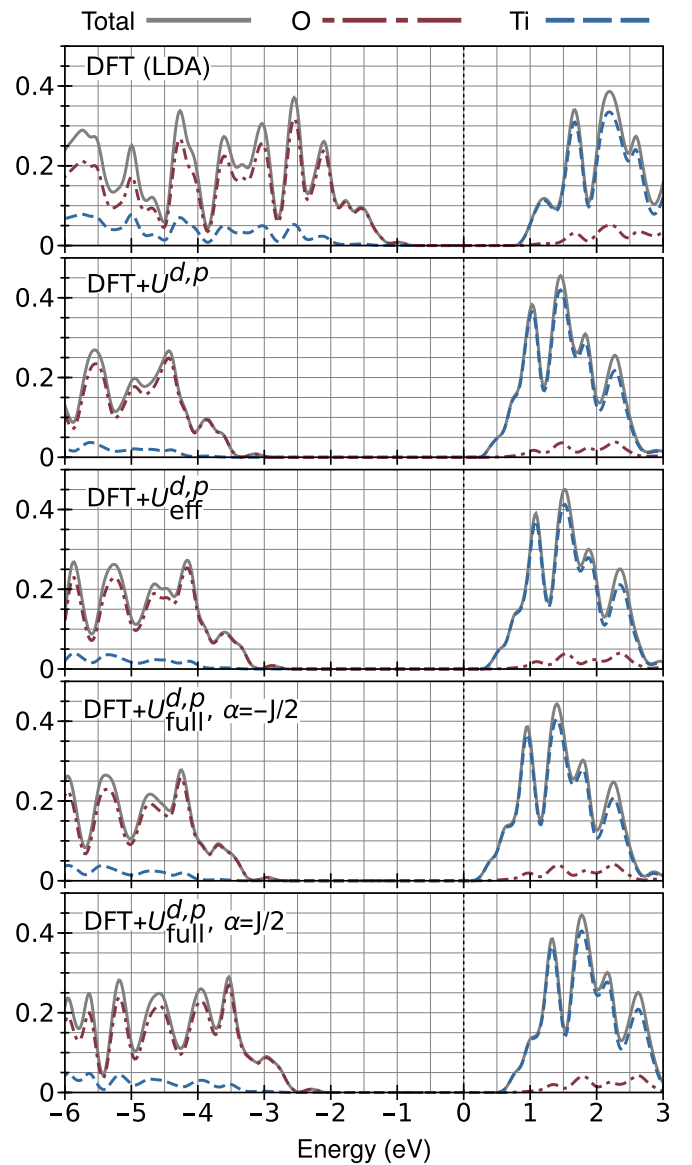


FIG. 1. The total and local generalized Kohn-Sham density of states (LDOS) of pristine  $\text{TiO}_2$ -rutile calculated within DFT(LDA), DFT+ $U$  with Hund's  $J$  neglected, when treated within the Dudarev model ( $U_{\text{eff}}$ ), and when treated in a manner which fully reproduces DFT+ $U+J$  using only DFT+ $U$  code for closed-shell systems ( $U_{\text{full}}$ ), both when treated with ( $\alpha = -J/2$ ) and without ( $\alpha = J/2$ ) its minority spin (term IV). The spectrum is partitioned on a per-species basis using Mulliken analysis based on the variationally optimized NGWFs. DFT+ $U^{d,p}$  results only are shown, using parameters calculated from first principles using the minimum-tracking linear-response method, using only the  $\text{Ti}^0$  pseudoatomic solver configuration, and a Gaussian broadening of 0.1 eV. In order to show the separate effects of the corrective functionals tested on the valence and conduction bands, each panel uses the mid-gap energy of the DFT(LDA) calculation for 0 eV.

correction for energy-magnetization curvature is neglected (letting  $J = 0$  eV), the band gap is overestimated by  $\sim 0.56$  eV with respect to the experimental gap. The important point here is that, even though the system harbors no magnetism in its ground state, this does not imply that the error in the

approximate energy functional related to the magnetic degree of freedom vanishes. When including this effect only in the like-spin term (using Dudarev's  $U_{\text{eff}} = U - J$ ) this overestimation reduces to  $\sim 0.35$  eV, and when also applying the unlike-spin term [using  $U_{\text{full}} = U - 2J$  and  $\alpha = -J/2$ , which is equivalent to DFT+ $U+J$  including its standard minority spin term (IV), for closed-shell systems such as this one], the overestimation reduces further to  $\sim 0.29$  eV.

However, when we apply DFT+ $U+J$  in its simplest form, i.e., neglecting the minority spin term (IV) of Eq. (2) (in practice using  $U_{\text{full}} = U - 2J$  and  $\alpha = J/2$ ), the gap underestimation vanishes to within the expected error in the experiment (using the zero-temperature extrapolation provided in Ref. [12] for the fundamental gap) and the theoretical methodology. We note that the zero-point phonon correction is held to be very small in rutile, unlike in anatase. As shown in Table III, we also carried out DFT+ $U+J$  calculations using explicit + $J$  code, with the same results to a high precision, as expected. We note, in passing, that the deduction in the calculated gap due to the omission of the minority spin term, of  $\sim 0.29$  eV, is very close to  $(J^p - J^d)/2 \sim 0.31$  eV, as might be predicted by considering the different characters of the band edges and the change in the potentials acting upon them.

These fundamental gap changes are reflected in the local density of states (LDOS) plots shown in Fig. 1. Here, we see the successive effects of first turning on + $U^{d,p}$  correction, and then by moderating it using  $J$  per Dudarev's  $U_{\text{eff}} = U - J$  prescription, which mostly brings the valence band back up in energy in this case. Moving ultimately to DFT+ $U_{\text{full}}^{d,p}$ ,  $\alpha = J/2$  (which means  $\alpha^d = J^d/2$ , etc., and which gives identical results to DFT+ $U^{d,p}+J^{d,p}$  by construction), we see a further closing of the gap and upward shift both in the valence and conduction bands. Interestingly, we obtain an extremely similar valence-band LDOS from the Dudarev prescription and DFT+ $U_{\text{full}}^{d,p}$ ,  $\alpha = -J/2$ , i.e., DFT+ $U^{d,p}+J^{d,p}$  with the minority spin term intact. This reflects the almost-complete cancellation of the potentials due to terms (III) and (IV) in Eq. (2), for a subspace near full occupancy.

### B. The first-principles band gap of pristine TiO<sub>2</sub>-anatase

A similar procedure was followed for pristine TiO<sub>2</sub>-anatase as that which we have outlined for TiO<sub>2</sub>-rutile, except that only the neutral atomic configuration of Ti was used in the pseudoatomic solver, in view of our previously discussed findings. As reflected in the calculated  $U$  and  $J$  parameters of Tables I and II, the electronic structures of the two polymorphs are rather similar, and again the valence (conduction) band edge is dominated by O  $2p$  (Ti  $3d$ ) character in TiO<sub>2</sub>-anatase, necessitating DFT+ $U^{d,p}$  for successful gap correction. Shown in Table IV is the fundamental band gap of TiO<sub>2</sub>-anatase calculated using LDA and first-principles DFT+ $U$ , DFT+ $U_{\text{eff}}$ , DFT+ $U_{\text{full}}$ , and DFT+ $U+J$  [minority spin term (IV) included, spin-averaged, and neglected], both when applied only to the Ti  $3d$  subspace and when applied also to the O  $2p$  subspace. The corresponding NGWF-partitioned Mulliken LDOS plots are shown in Fig. 2. We anticipate a slight overestimation in our calculated gap values for TiO<sub>2</sub>-anatase, due to our necessarily finite effective sampling of the Brillouin zone. The band gap of anatase is of indirect character and,

TABLE IV. The band gap (in eV) of TiO<sub>2</sub>-anatase calculated within DFT(LDA), DFT+ $U$  with Hund's  $J$  neglected, when treated within the Dudarev model ( $U_{\text{eff}}$ ), and when treated in a manner which fully reproduces DFT+ $U+J$  using only DFT+ $U$  code for closed-shell systems ( $U_{\text{full}}$ ), both when treated with ( $\alpha = -J/2$ ) and without ( $\alpha = J/2$ ) its minority spin (term IV). DFT+ $U^d$  and DFT+ $U^{d,p}$  results are separately shown, using parameters calculated from first principles using the minimum-tracking linear-response method, using only the neutral pseudoatomic solver configuration Ti<sup>0</sup>. Prior experimental, first-principles local, semilocal, meta-generalized-gradient, and semiempirical hybrid functional; perturbative G<sub>0</sub>W<sub>0</sub>; empirical and first-principles SCF linear-response DFT+ $U$  (Ref. [82]) values from the literature are provided for convenient comparison. Our central results are highlighted in bold.

TiO <sub>2</sub> -anatase $E_{\text{gap}}$		
DFT (LDA)	2.21	
	+ $U^d$	+ $U^{d,p}$
DFT+ $U$	2.51	4.13
DFT+ $U_{\text{eff}} = U - J$	2.48	3.88
DFT+ $U_{\text{full}} = U - 2J, \alpha = -J/2$	2.41	3.81
DFT+ $U_{\text{full}} = U - 2J$	2.45	3.65
DFT+ $U_{\text{full}} = U - 2J, \alpha = J/2$	2.49	<b>3.50</b>
DFT+ $U+J$ (no minority spin term)	2.49	<b>3.50</b>
Experiment [11]		<b>3.47</b>
PBE [19]		1.94
TB-mBJ [77]		3.01
SCAN [78]		2.56
HSE06 [19,79]		3.60
LDA+G <sub>0</sub> W <sub>0</sub> [18]		3.56
PBE+G <sub>0</sub> W <sub>0</sub> [11]		3.61
PBE+G <sub>0</sub> W <sub>0</sub> [19]		3.73
HSE+G <sub>0</sub> W <sub>0</sub> [19]		4.05
DFT+ $U^d$ ( $U=7.5$ eV) [80]		3.27
DFT+ $U^d$ ( $U = 3.23$ eV) [82]		2.43
DFT+ $U^{d,p}$ ( $U^d = 3.23$ eV, $U^p = 10.59$ eV) [82]		4.24
DFT+ $U^{d,p}$ ( $U^d = 3.23$ eV, $U^p = 5.0$ eV) [82]		3.23

while our sampling is chosen to closely sample the LDA band edges, we cannot be guaranteed to precisely sample the valence band maximum (most studies hold the fundamental gap of rutile to be direct at  $\Gamma$ , on the other hand, which we do sample). Again, experimental, first-principles, semiempirical hybrid, many-body perturbation theory, and several previous DFT+ $U$  results from the literature are shown for comparison.

While anatase has been thoroughly studied using optical techniques [84], our focus here is on the fundamental electronic gap. For the latter, very little direct data is available, but fortunately there has recently been reported angle-resolved photoemission spectroscopy with n-type doping (to circumvent the need for inverse photoemission) in Ref. [11], strongly supported by temperature-dependent many-body perturbation theory calculations including electron-phonon coupling. The fundamental gap reported in the latter work is higher than that found elsewhere in older studies, and the reason is that, whereas the commonplace misidentification between the optical and fundamental gap is not very significant for rutile (the exciton binding is  $\sim 4$  meV), it is not at all reasonable

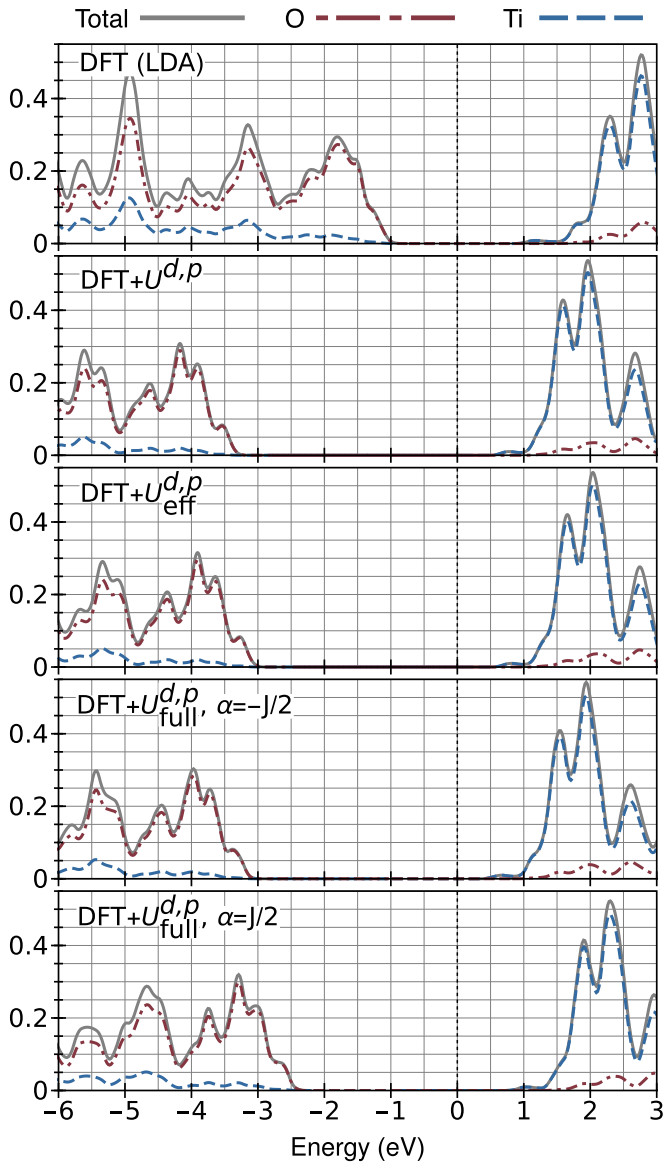


FIG. 2. The total and local generalized Kohn-Sham density of states (LDOS) of pristine  $\text{TiO}_2$ -anatase calculated within DFT(LDA), DFT+ $U$  with Hund's  $J$  neglected, when treated within the Dudarev model ( $U_{\text{eff}}$ ), and when treated in a matter which fully reproduces DFT+ $U$ + $J$  using only DFT+ $U$  code for closed-shell systems ( $U_{\text{full}}$ ), both when treated with ( $\alpha = -J/2$ ) and without ( $\alpha = J/2$ ) its minority-spin (term IV). The spectrum is partitioned on a per-species basis using Mulliken analysis based on the variationally optimized NGWFs. DFT+ $U^{d,p}$  results only are shown, using parameters calculated from first principles using the minimum-tracking linear-response method, using only the  $\text{Ti}^0$  pseudoatomic solver configuration, and a Gaussian broadening of 0.1 eV. In order to show the separate effects of the corrective functionals tested on the valence and conduction bands, each panel uses the mid-gap energy of the DFT(LDA) calculation for 0 eV.

for anatase, which is reported to exhibit relatively very large exciton binding  $\sim 0.18$  eV effects in its low-energy optical spectra [11].

The LDA gives a Kohn-Sham band gap of 2.21 eV, substantially underestimating the experimental electronic gap of

3.47 eV. DFT+ $U^d$  is ineffective at opening the gap as it is in  $\text{TiO}_2$ -rutile, given the LDA-appropriate calculated first-principles  $U$  and  $J$  parameters. DFT+ $U^{d,p}$  opens the gap very efficiently and, closely mirroring what we found for  $\text{TiO}_2$ -rutile, both DFT+ $U$  with  $J$  neglected and Dudarev's DFT+ $U_{\text{eff}}$  cause the gap to be overestimated.

Similarly, again, first-principles DFT+ $U$ + $J$  including O  $2p$  correction gives decent agreement with the experimental gap, overestimating it by 0.03 eV (0.34 eV) when the minority spin term is neglected (included). Interestingly, both the HSE06 and DFT+ $G_0W_0$  approximations seem to recover the anatase gap better than the rutile one, based on the available literature. DFT+ $U_{\text{full}}$ ,  $\alpha = J/2$  [which is to say, technically, first-principles DFT+ $U^{d,p}$ + $J^{d,p}$  with the minority spin term neglected, which doesn't require an explicit Hund's  $J$  implementation for closed-shell systems] seems to be very competitive with respect to both methods as far as both the fundamental gap and computational complexity are concerned. The key ingredient for  $\text{TiO}_2$  in this sort of method, aside from the established message that the O  $2p$  subspace needs to be treated on the same footing as the Ti  $3d$  one, is evidently to correct both for the usual charge-related ( $U$ ) and spin-related ( $J$ ) systematic errors in the approximate functional. Indeed, more generally it has been shown in Ref. [44], by using the  $2 \times 2$  formalism to analyze the linear-response approach for Hubbard  $U$  parameter calculation, that the non-neglect of Hund's  $J$  is advisable even on abstract consistency grounds.

#### IV. CONCLUSIONS

We have shown that the DFT+ $U$ + $J$  functional developed in Ref. [43], in combination with the first-principles procedure for calculating  $U$  and  $J$  parameters developed in Ref. [44], yields fundamental gaps that are in very close agreement with the most sophisticated available zero-temperature-approaching experimental findings for  $\text{TiO}_2$ . The residual errors, 0.01 eV for rutile and 0.03 eV for anatase, are within the anticipated errors due to factors such as neglected zero-point phonon motion and relativistic effects, the pseudopotential approximation, imperfect Brillouin zone sampling (more relevant for anatase), and various sources of experimental uncertainty. Interestingly, the method performs better than both hybrid functionals and perturbative  $G_0W_0$  for the fundamental gap, while retaining a semilocal DFT-like level of computational cost (even linear scaling [70], algorithmically, though we don't exploit that here).

An important and surprising finding of this work, which we go on to discuss in Appendix B, is that, contrary to our expectation, the first-principles calculation of  $U$  and  $J$  for  $\text{TiO}_2$ -rutile acts to reinforce the numerically significant arbitrariness [85] of DFT+ $U$  with respect to the (too often unstated) choice of localized orbitals defining the subspaces targeted for correction. The good news here is that it is the default, neutrally charged, isolated atomic configuration that yields the accurate gaps. In our experience to date, the introduction of chemical intuition when defining atomic solver charge states for DFT+ $U$  projector construction yields worsened results together with worsened convergence behavior.

We judge that our results are, overall, very encouraging for the continued, very widespread use of DFT+ $U$  and its

extensions for studying  $\text{TiO}_2$ , and that they serve as a counterexample to the concept that such methods are fundamentally limited in their applicability to high-spin systems. It remains for a future study to establish whether  $\text{TiO}_2$  is a special case for the combination of the methods introduced in Ref. [43] and Ref. [44], or whether it is as successful for oxides, particularly closed-shell oxides, more generally.

What has hampered closed-shell applications to date, as highlighted in Ref. [86], have been available Hubbard  $U$  values, calculated or otherwise, that are too high for practical use. Our results demonstrate that Hund's  $J$ , which is subtracted from  $U$  once in the Dudarev formalism, and effectively *twice* in DFT+ $U$ + $J$  for closed-shell systems, yielding  $U_{\text{full}} = U - 2J$ , may be the key ingredient in moderating the  $U$ . The first-principles  $U$  values in common circulation for Ti  $3d$  orbitals in  $\text{TiO}_2$ , in the range of approximately 3–4 eV depending on the projector choice, are perhaps fine after all. Meanwhile, our directly calculated, relatively high-seeming-at-first  $U$  values for O  $2p$  orbitals in  $\text{TiO}_2$  (which are more localized than Ti  $3d$  ones; see the plot in Fig. 3) sit among the few previously reported calculated values for  $\text{TiO}_2$  in the literature [83,87].

Our results are consistent with the prescriptions detailed in Refs. [43,44], for the use and calculation of  $U$  and  $J$  parameters, being correct. The contribution of the explicit unlike-spin  $J$  correction [term (III) in Eq. (2)] to the potential subspace matrix elements for spin  $\sigma$ , is given by  $V_{mm'}^{J\sigma} = Jn_{mm'}^{\bar{\sigma}}$ . It seems that this is a very good approximation, given that there are  $J$  parameters involved for two different subspace types and the net result is very accurate as far as the gap is concerned. Our results strongly support the conclusions of Ref. [43] that the minority spin term (IV) of Eq. (2), which arises only due to the double-counting correction of an unlike-spin interaction that is unlikely to be well described in the underlying functional in the first place, should be neglected. Equivalently, they support the conclusion that the fully localized limit double-counting term of Refs. [88,89] is sufficient at this level of theory, at least as far as the potential is concerned.

The DFT+ $U$ + $J$  gap is just one aspect of the potential, of course, and its correctness cannot be used to judge whether the double-counting in the total energy is correct, for example. In previous works, we have pointed out cases where the standard DFT+ $U$  potential fails due to nonsatisfaction of Koopmans' condition [90], or due to inadequate projection onto the states adjacent to the band edges [91], neither of which effects are expected to be alleviated particularly by the incorporation of Hund's  $J$ .

On a similar cautionary note, it is worth emphasizing that our first-principles calculations of  $U$  and  $J$  in  $\text{TiO}_2$  were made simpler by the vanishing occupancy-magnetization coupling in closed-shell systems, by which we mean that  $d(n^\uparrow + n^\downarrow)/d\beta = 0 = d(n^\uparrow - n^\downarrow)/d\alpha$ . In this case, the elegant formulas of Eq. (8) become unambiguous with respect to the spin polarization of the perturbing potential. In our current view, these two formulas are essentially the correct ones for  $U$  and  $J$ , neglecting self-consistency over parameters. As a result, without approximation and very conveniently, we were able to perturb one spin only and obtain  $U$  and  $J$  simultaneously. A disadvantage of this decoupling, however, is that we cannot judge on the basis of the present calculations

between the merits of the “scaled  $2\times 2$ ” and “simple  $2\times 2$ ” procedures of Ref. [44], since they become identical.

Overall, there is without doubt much further work to be done on developing self-contained corrective techniques such as first-principles DFT+ $U$ + $J$  for approximate density functional theory, which sidestep the evolution of increasingly costly closed-form functionals. Meanwhile, our results here may prove to significantly lower the computational barrier to simulating accurate spectral quantities in large, possibly defect-containing or disordered supercells of  $\text{TiO}_2$ .

## ACKNOWLEDGMENTS

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## APPENDIX A: DFT+ $U$ ON $2p$ AND $3d$ ORBITALS: DFT+ $U^{d,p}$

In principle, SIE is harbored by all subshells and cannot be partitioned out between them, however, it is commonly more dominant in  $3d$  subshells due to their spatially localized nature. Hence, in titanium-comprising systems, the Hubbard correction in DFT+ $U$  is conventionally applied to the Ti  $3d$  subshell only. The Hubbard  $U$  parameters used for the  $3d$  orbitals of the Ti atom have ranged over  $\sim 2.5$ – $10$  eV [92], and have most commonly been determined by tuning to some observed quantity [87,93–98]. Even when overlooking our serious concerns regarding the robustness and conceptual validity of  $U$  value calibration to observable quantities, particularly when those are not ground-state observables, a practical problem arises for DFT+ $U$  due to the location of Ti on the extreme left of the transition-metal block.

It is well known that Hubbard  $U$  correction to the  $3d$  orbitals alone is not very effective for opening the band gap of  $\text{TiO}_2$ , which saturates even with unreasonably large  $U$  values, as the dominant  $2p$  states at the valence band edge remain barely affected. Moreover, when actually plotted, as they are in Fig. 3, the  $2p$  pseudoatomic orbitals of O atoms are rather *more* localized than their Ti  $3d$  counterparts, and so it is not at all unreasonable, quite the contrary, to calculate (or at least tune, where calculation is not possible) Hubbard  $U$  and even Hund's  $J$  parameters for O  $2p$ . Indeed, it has been demonstrated in several prior works that applying the Hubbard  $U$  correction simultaneously on the  $3d$  orbitals of Ti and the  $2p$  orbitals of O atoms, symbolically giving DFT+ $U^{d,p}$ , readily addresses the aforementioned gap saturation problem



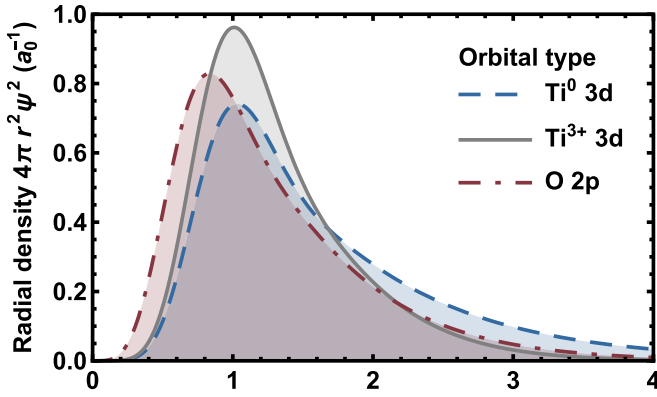


FIG. 3. The radial probability distributions of the three pseudo-orbital types used to define DFT+ $U$ (+ $J$ ) subspaces in this work, as defined in the main text. The oxygen  $2p$  subspaces are more localized than their titanium  $3d$  counterparts, and thus are reasonable candidates for correction testing. The Ti pseudoatomic solver charge state significantly affects the localization of the  $3d$  orbitals, and the resulting gaps.

and provides a more accurate description of the band structure around the Fermi level [45–48].

#### APPENDIX B: THE EFFECTS ON THE DENSITY OF STATES OF THE CHOICE OF PSEUDOATOMIC SOLVER CONFIGURATION FOR GENERATING THE Ti $3d$ DFT+ $U$ SUBSPACE

For the specific case of rutile, we investigate in detail here the effect of varying the charge configuration for Ti used in the pseudoatomic solver [72], which constructs the set of the pseudoatomic orbitals defining the  $3d$  subspace of Ti. The neutral configuration is perhaps a natural choice, giving the relatively smooth, diffuse subspace density shown in Fig. 3. This results in less pressure on the plane-wave convergence and, more importantly, it does not rely on any prior chemical intuition. We also investigated the  $3+$  atomic charge configuration, as a slightly more “informed” spatially localized subspace test case. Given the LDA-appropriate  $U$  and  $J$  parameters calculated for each of the two subspace types and presented in Table I, we performed the matching DFT+ $U$ , DFT+ $U_{\text{eff}}$ , and DFT+ $U+J$  band-gap calculations, both within DFT+ $U^d$  and DFT+ $U^{d,p}$ . We also performed the “cross” calculations in the case of  $\alpha = 0$ , i.e., where we used the  $3+$  subspace parameters for correcting the neutral subspace, and vice versa, in order to illustrate the separate effects of over-localizing the projectors.

The results of these tests are shown in Table V, and a representative set of LDOS plots are shown in Fig. 4. We find that first-principles calculation of the Hubbard  $U$  and Hund’s  $J$  parameters does not compensate for the arbitrariness of the subspace choice, for Ti  $3d$ . Instead, it reinforces this arbitrariness as far as the fundamental gap is concerned in this system. Table V reveals that this trend holds irrespective of whether correction is also applied to O  $2p$  orbitals, denoted DFT+ $U^{d,p}$ , or indeed whether we are using DFT+ $U$ , DFT+ $U_{\text{eff}}$ , or DFT+ $U_{\text{full}}$ .

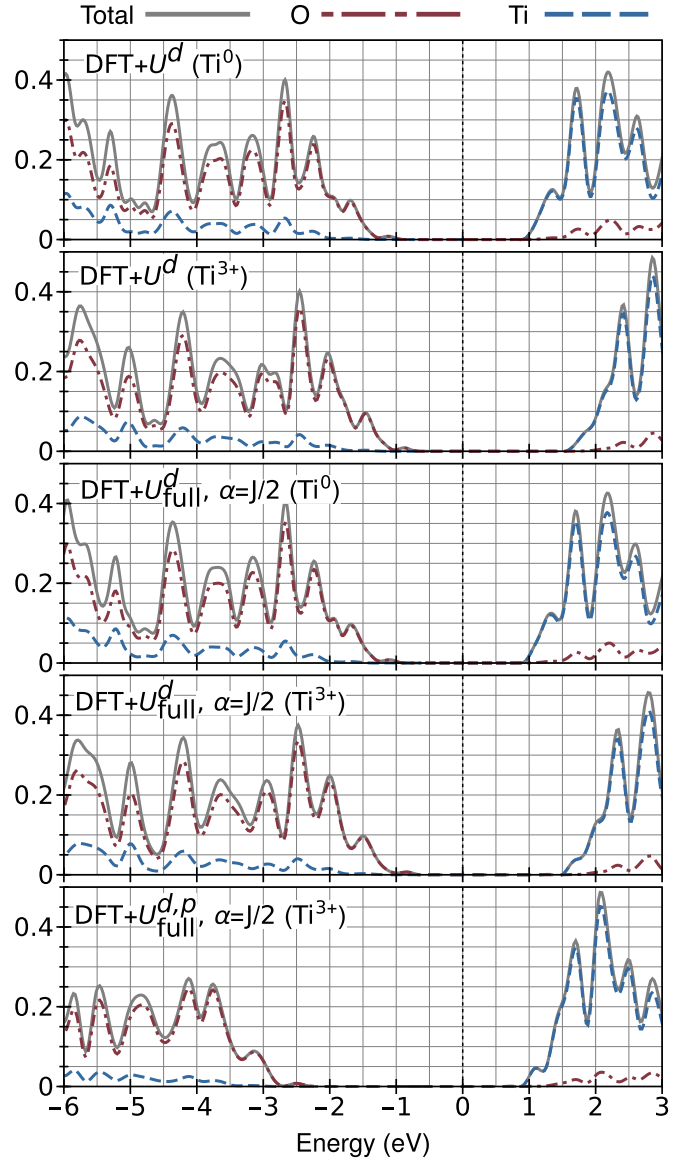


FIG. 4. The total and local generalized Kohn-Sham density of states (LDOS) of pristine TiO<sub>2</sub>-rutile calculated within DFT+ $U^d$ , separately for Ti<sup>0</sup> and Ti<sup>3+</sup> subspace definitions, within DFT+ $U^d+J^d$  for the same two subspace definitions, and finally within DFT+ $U^{d,p}+J^{d,p}$  for the Ti<sup>3+</sup> definition. The spectrum is partitioned on a per-species basis using Mulliken analysis based on the variationally optimized NGWFs. Parameters were calculated from first principles using the minimum-tracking linear-response method, and a Gaussian broadening of 0.1 eV was used. Each panel uses the mid-gap energy of the DFT+ $U^d$  (Ti<sup>0</sup>) calculation for 0 eV.

As previously discussed, the increase in spatial localization of the  $3d$  subspace, when we move from a neutral to a  $3+$  configuration, increases the corresponding calculated  $U$  and  $J$  parameters. This, of course, increases the predicted gap, when those parameters are applied to either subspace type. Moreover, Table V demonstrates that, for either fixed set of parameters, the increase in subspace localization also tends to open the gap, in this system, in fact by roughly the same amount. The net increase in the gap in going from the neutral

TABLE V. This table highlights the arbitrariness of DFT+ $U$  with respect to the targeted subspace choice, which is not compensated for in this system by first-principles calculation of the  $U$  and  $J$  parameters. Shown is the band gap (in eV) of TiO<sub>2</sub>-rutile calculated within DFT(LDA), DFT+ $U$  with Hund's  $J$  neglected, when treated within the Dudarev model ( $U_{\text{eff}}$ ), and when treated in a manner which fully reproduces DFT+ $U+J$  using only DFT+ $U$  code ( $U_{\text{full}}$ ). DFT+ $U^d$  and DFT+ $U^{d,p}$  results are separately shown, and these depend on the pseudoatomic solver configuration (neutral or 3+) used to define the targeted Ti 3d subspace, together with the corresponding subspace-dependent  $U$  and  $J$  parameters. For the intermediate case of  $U_{\text{full}}$  with  $\alpha = 0$ , i.e., DFT+ $U+J$  with its minority term split over the two spins, we show the effect on the gap of separately changing the subspace used to calculate the parameters, and the subspace used to apply the parameters, revealing that these effects combine to reinforce, not to cancel, the subspace dependence in this system. The gaps from "mismatched" calculations, with parameters from the other subspace type, are shown in bold.

Subspace definition	TiO <sub>2</sub> -rutile $E_{\text{gap}}$			
	Ti <sup>0</sup> state		Ti <sup>3+</sup> state	
	1.96		1.96	
DFT(LDA)				
	$+U^d$	$+U^{d,p}$	$+U^d$	$+U^{d,p}$
$U$	2.24	3.59	2.69	4.20
$U_{\text{eff}} = U - J$	2.21	3.38	2.63	3.94
$U_{\text{full}} = U - 2J, \alpha = -J/2$	2.17	3.32	2.52	3.81
$U_{\text{full}} = U - 2J$ from Ti <sup>0</sup>	2.18	3.18	<b>2.31</b>	<b>3.33</b>
$U_{\text{full}} = U - 2J$ from Ti <sup>3+</sup>	<b>2.38</b>	<b>3.46</b>	2.57	3.69
$U_{\text{full}} = U - 2J, \alpha = J/2$	2.20	3.04	2.62	3.58
$U + J$ (no minority spin term)	2.20	3.04	2.64	3.58

to 3+ subspace densities shown in Fig. 3, with corresponding first-principles parameters, is thus approximately due, half-and-half, to the increase in parameters and increase in localization.

On the basis of these results, we can envisage that both the first-principles LDA-appropriate  $U$  and  $J$  parameters, and the fundamental gap for a fixed reasonable set of parameters, will attain maxima for some reasonable (though not generally the same) value of the pseudoatomic configuration charge. A tentative step towards plotting observables as functions of a DFT+ $U$  subspace localization quantifier was presented in Ref. [85]. More recently, the projector dependence of DFT+ $U$  results on rutile TiO<sub>2</sub> has previously been demonstrated at fixed  $U$  values in Ref. [99].

We do not necessarily expect that the projector arbitrariness reinforcement effect will arise transition-metal oxides generally, particularly since projector arbitrariness cancellation has previously been observed in molecular FeO<sup>+</sup>

using a self-consistently evaluated Hubbard  $U$  parameter [100]. This issue in DFT+ $U$  clearly warrants further investigation on diverse systems using various approaches, such as parameter [49,68] or projector [85,101] self-consistency.

Pragmatically, we have found, in our minimum-tracking linear-response calculations to date, that using the simplest, neutral pseudoatomic configuration for constructing the DFT+ $U$  projectors works well relative to more localized charged configurations. This is irrespective of the pseudopotential generator reference state, which is a somewhat different, technical matter related to the transferability in norm-conserving pseudopotentials. We note, in passing, that there is a small discrepancy in the gap from Ti<sup>3+</sup>-only subspace explicit DFT+ $U+J$  and the corresponding unmodified DFT+ $U$  code equivalent form with  $\alpha = J/2$ , reflecting that calculations with excessively localized subspaces are typically less numerically stable, aside from giving less favorable results.

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