Magnetic asymmetry induced anomalous spin-orbit torque in IrMn

Jing Zhou⁽⁰⁾,^{1,*} Xinyu Shu,^{1,*} Yaohua Liu,² Xiao Wang,³ Weinan Lin,¹ Shaohai Chen,¹ Liang Liu,¹ Qidong Xie,¹ Tao Hong,²

Ping Yang,^{4,1} Binghai Yan,⁵ Xiufeng Han,³ and Jingsheng Chen^{91,6,†}

¹Materials Science & Engineering, National University of Singapore, 9 Engineering Drive 1, 117575 Singapore

²Neutron Scattering Division, Oak Ridge National Laboratory, Oak Ridge 37831, USA

³Institute of Physics and Beijing National Laboratory for Condensed Matter Physics, Chinese Academy of Sciences, Beijing 100190, China

⁴Singapore Synchrotron Light Source (SSLS), National University of Singapore, 5 Research Link, 117603 Singapore

⁵Department of condensed matter physics, Weizmann Institute of Science, Rehovot 7610001, Israel

⁶Suzhou Research Institute, National University of Singapore, Suzhou 215123, China

) (Received 5 March 2020; revised manuscript received 9 April 2020; accepted 9 April 2020; published 4 May 2020)

We demonstrate an anomalous spin-orbit torque induced by the broken magnetic symmetry in the antiferromagnet IrMn. We study the magnetic structure of three phases of IrMn thin films using neutron diffraction technique. The magnetic mirror symmetry \mathcal{M}' is broken laterally in both L1₀-IrMn and L1₂-IrMn₃ but not γ -IrMn₃. We observe an out-of-plane dampinglike spin-orbit torque in both L1₀-IrMn/permalloy and L1₂-IrMn₃/permalloy bilayers but not in γ -IrMn₃/permalloy. This is consistent with both the symmetry analysis on the effects of a broken \mathcal{M}' on spin-orbit torque and the theoretical predictions of the spin Hall effect and the Rashba-Edelstein effect. In addition, the measured spin-orbit torque efficiencies are 0.61 \pm 0.01, 1.01 \pm 0.03, and 0.80 \pm 0.01 for the L1₀, L1₂, and γ phases, respectively. Our work highlights the critical roles of the magnetic asymmetry in spin-orbit torque generation.

DOI: 10.1103/PhysRevB.101.184403

I. INTRODUCTION

The current induced spin-orbit torque (SOT) has been intensively researched since it can switch magnetization electrically, which is crucial for developing the next-generation magnetic memories [1-5]. After almost a decade since its conception, sizable SOT can be sourced from a large variety of materials, such as the topological insulator [6], heavy metal (HM) [4,7], antiferromagnets (AFM) [8–10], ferromagnetic (FM) semiconductor [11] and FM trilayer [12]. The spin Hall effect (SHE) [4] and the Rashba-Edelstein effect (REE) [5] are the two widely accepted models for the SOT generation. In both scenarios, SOT arises from the charge-to-spin conversion, where a transverse spin polarization $(s_v || y)$ is induced by a longitudinal charge current $(J_c || x)$. This spin polarization exerts an in-plane dampinglike SOT (τ_{DL}) on an FM with in-plane magnetization such as the Ni₈₁Fe₁₉ [also known as permalloy (Py)] [1-3]. In emerging studies on SOT, however, an out-of-plane spin polarization $(s_z||z)$ is explored since it is favorable for switching a perpendicularly magnetized FM [12–16], which is more suitable for high-density applications. In this regard, the 2D material WTe2 has attracted lots of attention since an out-of-plane τ_{DL} , which is equivalent to the effects of an out-of-plane spin polarization, is observed in a WTe₂/Py bilayer [13]. This anomalous torque is attributed to the broken interfacial crystal mirror symmetry (\mathcal{M}) of WTe₂. Ever since then, the crystal-asymmetry-controlled SOT has been found in other hexagonal crystal systems, such as NbSe₂ [17] and MoTe₂ [18,19]. Magnetic moments, when ordered, can lower the symmetry of a material [20,21]. Thus, a natural question is whether the symmetry breaking introduced by the magnetic moments can induce a similar torque.

In our previous report, we found the staggered magnetic moments in L1₀-IrMn (001) films align with the [111] direction [22], as illustrated in Fig. 1(a). We notice that in this magnetic structure there is only one magnetic mirror symmetry (\mathcal{M}') parallel to the (-110) plane and no twofold rotational invariance, as illustrated in Fig. (1b). Since \mathcal{M}' is not found in the (110) plane, the lateral magnetic symmetry is said to be broken. Moreover, L10-IrMn has a high crystal symmetry (P4/mmm) and its (001) plane has a fourfold rotational symmetry. This makes L10-IrMn a good choice for investigating the magnetic asymmetry induced SOT because its crystal symmetry is intact while its magnetic symmetry is broken. A previous study suggests that the SOT induced by broken mirror symmetry would exhibit a strong dependence on the direction of the electric field E [13]. However, it would not be so straightforward to observe such dependence experimentally in L10-IrMn due to the existence of AFM domains. Thin films of AFM, in general, are likely to be multidomain in their ground states [23–26]. In our previous study [22], even we managed to obtain a high-quality epitaxial film, L10-IrMn still appeared as a twin-domain structure, which would have eliminated the differences between $E \parallel \mathcal{M}'$ and $E \perp \mathcal{M}'$. Therefore the SOT induced by the broken \mathcal{M}' has to be evaluated otherwise.

In this work, we exploit the IrMn films with different phases, which allow us to separate the magnetic asymmetry

^{*}These authors contributed equally to this work.

[†]Corresponding author: msecj@nus.edu.sg

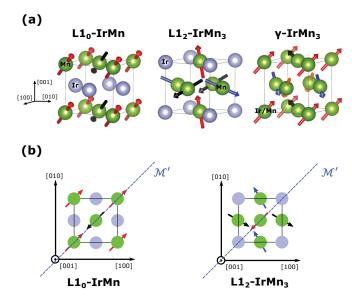


FIG. 1. (a) Unit cells of L1₀-IrMn, L1₂-IrMn₃, and γ -IrMn₃. (b) The magnetic mirror symmetry (\mathcal{M}') in the (001) plane of the L1₀-IrMn and L1₂-IrMn₃. Arrows indicate the magnetic moments with their orientations differentiated by colors.

(broken \mathcal{M}') from specific magnetic structure. We show that a phase change from L10-IrMn to L12-IrMn3 leads to a different magnetic structure but with the same lateral magnetic asymmetry. Using the spin-torque ferromagnetic resonance (ST-FMR) technique, we measure the SOT efficiencies of IrMn thin films, which are 0.61 \pm 0.01, 1.01 \pm 0.03, and 0.80 \pm 0.01 for the L1₀, L1₂, and γ phases, respectively. We demonstrate an anomalous SOT (out-of-plane dampinglike) in both L10-IrMn/Py and L12-IrMn3/Py bilayers. This is compatible with the symmetry analysis about the effects of a broken \mathcal{M}' on SOT. We also show that the microscopic origin of the observed anomalous SOT arises from an out-of-plane spin polarization, which is consistent with theoretical predictions from both the SHE and REE perspectives. In contrast, the anomalous SOT is not observed in the γ -IrMn₃/Py bilayer, which we attribute to the lack of magnetic asymmetry due to disordered magnetic structure in γ -IrMn₃. This paper is organized as the following: Sec. II shows the methods used in this work. In Sec. III, we present the major experimental results including the crystal (Sec. IIIA) and magnetic (Sec. IIIB) structures of IrMn and the observed SOT (Sec. IIIC). In Sec. IV, we discuss the origins of the anomalous SOT, considering the intrinsic effects of \mathcal{M}' on SOT (Sec. IVA-B) and its microscopic origin (Sec. IVC), followed by a conclusion in Sec. V.

II. METHODS

A. Material fabrication

Three types of samples were fabricated in this work, namely, the IrMn(22)/Py(13) bilayer with IrMn in three different phases, the L1₀-IrMn(22)/Cu(t_{Cu})/Py(13) trilayer and the Pt(10)/Py(10) bilayer. Numbers in the parentheses are thicknesses in nanometer. The IrMn (001) layers were deposited epitaxially on KTaO₃ (001) substrate using the DC

magnetron sputtering technique at a base pressure of less than 2×10^{-8} Torr. They were co-sputtered using a Ir₄₀Mn₆₀ target and a Mn target, where the atomic concentration of Ir and Mn were adjusted by the sputtering powers of the two targets. The deposition temperatures for the $L1_0$, $L1_2$, and γ phases were 720 °C, 640 °C, and 320 °C, respectively. The measured atomic concentration of Mn for the $L1_0$, $L1_2$, and γ phases are 46.3%, 67.5%, and 70.9%, respectively. After the samples cooled down to room temperature, the Cu and Py layers were deposited. The sample of Pt/Py was deposited on thermally oxidized silicon substrate at room temperature. All samples were protected by a 2-nm SiO₂ layer. For ST-FMR measurement, the above bilayers and trilayer were patterned into microstrips of 30 μ m \times 50 μ m using a combination of photolithography and ion-beam etching. An electrode of Ti (5)/Cu (100) was deposited using a thermal evaporator.

B. ST-FMR measurement

The SOT generated by IrMn was examined by the established ST-FMR technique [4,8,22]. Figure 2(a) shows the ST-FMR experimental set-up schematically. A microwave (or equivalently the current J_c) sourced by a signal generator was applied along the microstrip. Then a rectifying voltage (V_{mix}) was produced when an in-plane external magnetic field (H) was swept at an angle (ϕ_H) with respect to J_c . The ST-FMR measurement was modulated using a sine function of low frequency. The modulated V_{mix} was collected using a lock-in amplifier and fitted using [4,8,22]

$$V_{\rm mix} = V_S \frac{\Delta H^2}{\Delta H^2 + (H - H_{\rm res})^2} + V_A \frac{\Delta H (H - H_{\rm res})}{\Delta H^2 + (H - H_{\rm res})^2}.$$
(1)

Here, ΔH is the linewidth and $H_{\rm res}$ is the resonant field. In Eq. (1), the first term describe a symmetric component $(V_{\rm sym})$ with an amplitude V_S , which is associated with in-plane torques (τ_{\parallel}) ; the second term describes an antisymmetric component $(V_{\rm asy})$ with an amplitude V_A , which is due to out-of-plane torques (τ_{\perp}) . Fig. 2(b) shows the typical $V_{\rm mix}$ of L1₀-IrMn/Py, L1₂-IrMn₃/Py and γ -IrMn₃/Py bilayers measured at 9 GHz with $\phi_H = -35^\circ$. The data points can be well fitted by Eq. (1).

The SOT efficiency, $\theta_{\parallel,m}$, which represents the magnitude of measured τ_{\parallel} relative to τ_{\perp} , can be extracted using [4,8,22]

$$\theta_{\parallel,m} = \frac{V_S}{V_A} \frac{e\mu_0 M_S d_{\rm IrMn} d_{\rm Py}}{\hbar} \sqrt{1 + \frac{4\pi M_{\rm eff}}{H_{\rm res}}},\tag{2}$$

where e, μ_0 , and \hbar are the electron charge, permeability of free space and reduced Planck constant, respectively; the thickness of the two layers are $d_{\rm IrMn}$ and $d_{\rm Py}$; M_s and $M_{\rm eff}$ are the saturation and effective magnetizations of the Py layer, respectively. M_s is extracted from a *M*-*H* loop measured using a vibrating sampling magnetometer. $M_{\rm eff}$ is extracted by fitting the in-plane Kittel equation $f = \frac{\gamma}{2\pi} \sqrt{(H_{\rm res} + H_k)(H_{\rm res} + H_k + 4\pi M_{\rm eff})}$. Here, f is the microwave frequency and H_k is the effective in-plane magnetic anisotropic field.

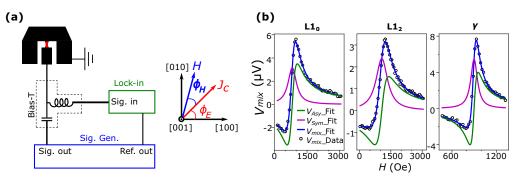


FIG. 2. (a) Experimental set-up of ST-FMR measurement. The red bar indicates the microstrip of IrMn/Py bilayer. On the right shows the coordinates system, which is defined with respect to the crystal lattice of IrMn. (b) The measured V_{mix} for three phases of IrMn and their fittings at 9 GHz as a function of the external magnetic field (*H*).

C. Principles of angle dependent ST-FMR measurement

In order to determine the symmetry of SOT, we examine the dependence of V_S and V_A on ϕ_H . This approach is based on the Landau-Lifshitz-Gilbert-Slonczewski formalism [4,27] that SOT must be one of the two types: a dampinglike torque $\tau_{DL} = \tau_{DL0}(\boldsymbol{m} \times \boldsymbol{s} \times \boldsymbol{m})$ and a fieldlike torque $\tau_{\rm FL} = \tau_{\rm FL0}(\boldsymbol{m} \times \boldsymbol{h})$. In the polar coordinate, $\boldsymbol{m} =$ $m_0[\cos(\theta_H)\cos(\phi_H),\cos(\theta_H)\sin(\phi_H),\sin(\theta_H)]^T$ is the magnetic moment, which is aligned to the external magnetic field direction with a small precession angle. The spin polarization is $s = s[\cos(\theta_s)\cos(\phi_s), \cos(\theta_s)\sin(\phi_s), \sin(\theta_s)]^T$ and the current induced dynamic magnetic field is h = $h[\cos(\theta_h)\cos(\phi_h),\cos(\theta_h)\sin(\phi_h),\sin(\theta_h)]^T$. All the in-plane angles (ϕ_h , ϕ_H , and ϕ_s) are relative to the J_c (defined as x), and the out-of-plane angles $(\theta_h, \theta_H \text{ and } \theta_s)$ are relative to the film plane (x-y plane). The τ_{DL} and the τ_{FL} are differentiated by their definitions, i.e., τ_{DL} (τ_{FL}) is even (odd) in *m*. However, the τ_{\parallel} differs from the τ_{\perp} by their orientations relative to the film normal (defined as z).

In the bilayers dominated by the SHE, such as the Pt/Py, both the current-induced Oersted field and the spin polarization are along the *y* direction when J_c is along the *x* direction [4]. *m* usually has two in-plane components in ST-FMR measurement. Thus $\mathbf{h} = h(0, 1, 0)^T$, $\mathbf{m} = m_0[\cos(\phi_H), \sin(\phi_H), 0]^T$, and $\mathbf{s} = s(0, 1, 0)^T$. Then we have

$$\boldsymbol{\tau}_{\mathrm{FL}} = \tau_{\mathrm{FL0}}(\boldsymbol{m} \times \boldsymbol{h}) = \tau_{\mathrm{FL0}} m_0 h \begin{pmatrix} 0\\0\\\cos(\phi_H) \end{pmatrix}, \quad (3)$$
$$\boldsymbol{\tau}_{\mathrm{DL}} = \tau_{\mathrm{DL0}}(\boldsymbol{m} \times \boldsymbol{s} \times \boldsymbol{m}) = \tau_{\mathrm{DL0}} s m_0^2 \begin{pmatrix} -\sin(\phi_H)\cos(\phi_H)\\\cos^2(\phi_H)\\0 \end{pmatrix}. \quad (4)$$

However, in order to examine the direction of the SOT, the coordinate system must be re-defined with respect to m, i.e. to multiply a rotation matrix

$$R_m = \begin{pmatrix} \cos(\phi_H) & \sin(\phi_H) & 0\\ -\sin(\phi_H) & \cos(\phi_H) & 0\\ 0 & 0 & 1 \end{pmatrix}.$$
 (5)

As a result, τ_{FL} remains the same but

$$\begin{pmatrix} \cos(\phi_H) & \sin(\phi_H) & 0\\ -\sin(\phi_H) & \cos(\phi_H) & 0\\ 0 & 0 & 1 \end{pmatrix} \boldsymbol{\tau}_{\text{DL}} = \boldsymbol{\tau}_{\text{DL}0} s m_0^2 \begin{pmatrix} 0\\ \cos(\phi_H)\\ 0 \end{pmatrix}.$$
(6)

Here, τ_{\parallel} only has the τ_{DL} component and τ_{\perp} only has the τ_{FL} component, both depend only on $\cos(\phi_H)$. Therefore both V_S and V_A depend on $\sin(2\phi_H)\cos(\phi_H)$, where the additional $\sin(2\phi_H)$ is due to the anisotropic magnetoresistance (AMR) effect [4,11,13]. In the unusual event of an out-of-plane spin polarization, we have $\boldsymbol{h} = h(0, 1, 0)^T$, $\boldsymbol{m} = m_0[\cos(\phi_H), \sin(\phi_H), 0]^T$ but $\boldsymbol{s} = \boldsymbol{s}[0, \cos(\theta_s), \sin(\theta_s)]^T$. Following the same procedure in Eqs. (3)–(6), we have

$$\boldsymbol{\tau}_{\mathrm{DL}} = \boldsymbol{\tau}_{\mathrm{DL}0} s m_0^2 \begin{pmatrix} 0\\ \cos(\phi_H)\cos(\theta_s)\\ \sin(\theta_s) \end{pmatrix}, \tag{7}$$

$$\boldsymbol{\tau}_{\rm FL} = \boldsymbol{\tau}_{\rm FL0} m_0 h \begin{pmatrix} 0\\ 0\\ \cos(\phi_H) \end{pmatrix}. \tag{8}$$

In this case, τ_{DL} has both τ_{\parallel} and τ_{\perp} components. Alternatively speaking, τ_{\parallel} still depends only on $\cos(\phi_H)$ but τ_{\perp} depends on $[\cos(\phi_H) + \sin(\theta_s)]$, which after multiplying the $\sin(2\phi_H)$ term due to AMR effect, create a new angular dependence on ϕ_H of V_A .

The results of Eqs. (3), (6)–(8) and other possible factors influencing the angular dependence of V_S and V_A are summarized in Table I. Though these contributions might not be exhaustive, we briefly discuss their possible origins. Case 2 happens if the electric contact between the G-S-G RF probe and the electrode is not homogeneous, leading to a net h_z [28]. A possible cause of case 3 is a Dresselhaus-like field (h_x) . Case 4 describes an unusual spin polarization that is collinear with J_c . Case 6 can arise from an interfacial exchange coupling between an FM and an AFM [29]. The rotation matrix R_m is different for case 6 due to the presence of m_z . In this case, the classification of τ_{DL} and τ_{FL} into τ_{\parallel} and τ_{\perp} is trivial since m is between the in-plane and out-of-plane directions.

Considering all cases in Table I and omitting the $sin(2\phi_H)$ contribution from AMR, both V_S and V_A consist of a $cos(\phi_H)$

TABLE I. Dependence of V_s and V_A on ϕ_H . IP stands for in-plane, OP stands for out-of-plane. The subscripts on *s*, *h* and *m* indicate the present components, e.g. h_{yz} indicates the current-induced field has both *y* and *z* components. Other coefficients before the *sine* and *cosine* terms are omitted to highlight the angular dependence.

		Dependence on ϕ_H		
Case	Configuration	$V_A~(lpha~m{ au}_ot)$	$V_S~(\propto oldsymbol{ au}_\parallel)$	Origin ^a
1	s_v, h_v, m_{xv}	$\cos(\phi_H)$	$\cos(\phi_H)$	Classic SHE
2	s_y, h_{yz}, m_{xy}	$\cos(\phi_H)\cos(\theta_h)$	$\cos(\phi_H) - \sin(\theta_h)$	OP field
3	s_{y}, h_{xy}, m_{xy}	$\sin(\phi_h - \phi_H)$	$\cos(\phi_H)$	IP field
4	s_{xy}, h_y, m_{xy}	$\cos(\phi_H)$	$\sin(\phi_H - \phi_s)$	IP spin polarization
5	s_{yz}, h_y, m_{xy}	$\cos(\phi_H) + \sin(\theta_s)^{\rm b}$	$\cos(\phi_H)\cos(\theta_s)$	OP spin polarization
6	s_y, h_y, m_{xyz}	$\sin(\phi_H)\sin(\theta_m) + \cos(\phi_H)$	$\sin(\phi_H)\sin(\theta_m) + \cos(\phi_H)$	OP magnetic moment

^aIn additional to the configuration of classic SHE.

^bThis is the only ϕ_H independent term for V_A .

term, a $sin(\phi_H)$ term and a term independent of ϕ_H , leading to their general expression

$$V_{S} = \sin(2\phi_{H} - 2\phi_{0})[S_{A}\cos(\phi_{H} - \phi_{0}) + S_{B}\sin(\phi_{H} - \phi_{0}) + S_{C}],$$
(9)

$$V_A = \sin(2\phi_H - 2\phi_0)[A_A\cos(\phi_H - \phi_0) + A_B\sin(\phi_H - \phi_0) + A_C],$$
(10)

where ϕ_0 is a phase correction to ϕ_H . Therefore, based on Table I, Eqs. (9) and (10), the various contributions to SOT can be differentiated by performing the ST-FMR measurement at different magnetic field directions.

III. EXPERIMENTAL RESULTS

A. Crystal structures of IrMn

Since IrMn exists in different crystal phases with different magnetic structures, it is of prime importance to verify the phases of IrMn before investigating its SOT generation. Figure 3(a) shows the analytical rules used in this work to determine the phases of IrMn using x-ray diffraction (XRD) technique. In a θ -2 θ scan along the (001) direction, the presence of the 001 peak indicates the presence of chemically ordered L1₀-IrMn or L1₂-IrMn₃, whereas only the 002 peak is expected in the chemically disordered γ -IrMn₃. The L_{1_2} -IrMn₃ is differentiated from the L_{1_0} -IrMn by the presence of (103) spot on the reciprocal space mapping (RSM). For polycrystalline IrMn (p-IrMn) deposited at room temperature, peaks can be hardly detected from the above XRD tests due to its small grain size and poor crystallinity, and it only produces ringlike features in RSM when the film is sufficiently thick. Figure 3(b) shows the typical results of the θ -2 θ scans. The L1₀ and L1₂ phases have all peaks in the {001} family and no other peaks, indicating high crystallinity and good (001) texture. In addition, both the $L1_2$ and γ phases are slightly strained due to lattice mismatch with the substrate, since $[a(KTaO_3) = 3.989\text{\AA}] > [a(L1_2, \gamma - IrMn_3) = 3.785 \text{\AA}]$ [30]. This is seen from the rightward shift of the 002 peak from the reference bulk value in Fig. 3(b), indicating shorter lattice parameter c, consistent with the usual effects of a tensile strain. The typical RSMs are shown in Fig. 3(c), where

the (113) spot is common for all three phases but only the $L1_2$ phase has the (103) spot.

B. Magnetic structures of IrMn

The three phases of IrMn are further studied by examining their magnetic structure using neutron diffraction technique. The collinear magnetic structure of L10-IrMn shown in Fig. 1(a) is proposed from our previous work [22] based on the combined results from neutron diffraction and ST-FMR. The magnetic structure of L12-IrMn3 is first examined by scanning a large reciprocal space using the time-of-flight quasi-Laue diffractometer CORELLI at the Spallation Neutron Source. Figure 4(a) shows the RSM of (H0L) (left) and two typical line cuts (right). The presence of a lot of diffraction spots of the film indicates good magnetic order. The ringlike features originate from the sample environment such as the contribution of the sample mount. The absence of the (002) and (202) spots on the (H0L) RSM and the absence of the (111) spot on the (HHL) RSM (not shown) indicate a good $L1_2$ order [31]. Given the good magnetic order, we do not observe any superlattice spot, for example, $(\pm 0.5\ 0\ 0)$, $(0.5\ 0\ 0.5)$, and $(1\ 0\ 0.5)$. This is another indicator of the triangular spin structure of the $L1_2$ phase [31]. The magnetic structure of L12-IrMn3 is also studied by measuring the integrated intensities of selected peaks by θ -2 θ scans using the triple-axis spectrometer CTAX at the High Flux Isotope Reactor. Figure 4(b) shows the 100 and 110 peaks and their Gaussian fits. Their integrated intensities are $I_{100} = 450 \pm$ 20 and $I_{110} = 570 \pm 20$, respectively. Among the possible magnetic structures of L1₂-IrMn₃ proposed previously [32], the one shown in Fig. 1(a) is the only viable model. Other models result in either satellite peaks due to doubled unit cell length along c-axis or weaker integrated intensity of the 110 peak than the 100 peak (i.e., $I_{110} < I_{100}$). Both are negative in our results. Therefore, based on the above evidences, we have verified the magnetic structure of the L1₂ phase, which agrees with previous studies [8,30-32]. On the other hand, we have not observed any diffraction spot for the γ -IrMn₃, inconsistent with the magnetic structure proposed previously [30], as shown in Fig. 1(a). This indicates that our γ -IrMn₃ does not have a sufficiently ordered magnetic structure although the observed crystal structure positively identifies a γ phase. Based on the results of neutron diffraction, we illustrate

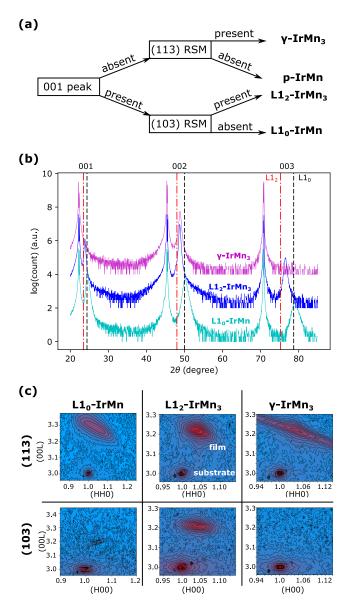


FIG. 3. (a) Guiding rules for determining the phases of IrMn by using the XRD technique. p-IrMn refers to polycrystalline IrMn. (b) XRD θ -2 θ scan of three phases of IrMn along (001) direction. The dashed vertical lines show the reference peak positions [9,30] for bulk IrMn with the *Cu*- $K_{\alpha 1}$ radiation equivalent. (c) Typical RSMs of IrMn in different phases near (103) and (113) spots. The RSMs are indexed based on the reciprocal lattice of KTaO₃.

in Fig. 1(b) that the L1₂-IrMn₃ also has just one magnetic mirror symmetry \mathcal{M}' parallel to the (-110) plane, although its magnetic structure is markedly different from the L1₀-IrMn. Correspondingly, the lateral magnetic symmetry in L1₂-IrMn₃ is broken in the same manner as L1₀-IrMn due to the lack of \mathcal{M}' in other directions. In γ -IrMn₃, \mathcal{M}' is absent and therefore no magnetic asymmetry is identified.

C. Spin-orbit torque in IrMn

The SOT efficiency $\theta_{\parallel,m}$ of three phases of IrMn are summarized in Fig. 5(a), where minimal variation is observed across 8 – 12 GHz. The $\theta_{\parallel,m}$ (averaged over 8 – 12 GHz) are

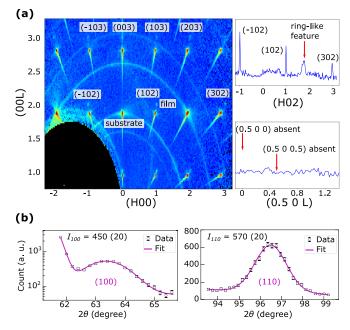


FIG. 4. (a) (Left) large-area neutron diffraction RSM of (H0L) of L1₂-IrMn₃. (Right) line cuts of (H02) and (0.5 0 L) of the RSM on the left. The peaks are indexed in the reciprocal lattice of L1₂-IrMn₃. (b) θ -2 θ scans on L1₂-IrMn₃ along (100) and (110) directions. I_{100} and I_{110} are the integrated intensity adjusted by absorption corrections.

 0.61 ± 0.01 , 1.01 ± 0.03 , and 0.80 ± 0.01 for the L1₀, $L1_2$, and γ phases, respectively. These values are substantially larger than those from previous reports [8,9,33-35], which we attribute to the different crystal and magnetic ordering. Our IrMn thin films have high crystallinity with the ordered magnetic structures in the $L1_0$ and $L1_2$ phases. As a results, the averaging effect [9,22] due to randomly oriented crystallites and magnetic domains is expected to be much smaller, leading to greater values of measured SOT efficiency. It has been reported that $\theta_{\parallel,m}$ is strongly affected by the electrical resistivity (ρ) [7,36]. An effective spin Hall conductivity (σ_s) is therefore calculated using $\sigma_s = \frac{\theta_{\parallel,m}}{\rho} [[36]]$, which is typically $\sim 10^3 \frac{\hbar}{e} \frac{s}{cm}$ in our work. Referring to Fig. 5(b), L1₀-IrMn and L1₂-IrMn₃ have similar ρ . However, the $\theta_{\parallel,m}$ of γ -IrMn₃ is profoundly inflated by its high ρ , and its σ_s is actually close to that of $L1_0$ -IrMn. Interestingly, comparing the $L1_0$ and $L1_2$ phases, the ratio of measured σ_s (3.55/6.12 = 0.58) is very close to the ratio of calculated intrinsic σ_s (102/165 = 0.62) (σ_s values adapted from Refs. [8,22]). This implies that the intrinsic SHE plays an important role in the SOT generated by IrMn.

The SOT efficiency of γ -IrMn₃ has not been reported before to the best of our knowledge. Unlike the magnetic-structure-enhanced SOT efficiency in L1₀-IrMn and L1₂-IrMn₃, the reasons behind the relatively large SOT efficiency and spin Hall conductivity in γ -IrMn₃ are still unclear. However, we find that the mixed experimental evidences from a previous work might support our observation. A giant SOT efficiency of 0.35 has been claimed for L1₂-IrMn₃ (001) [8], but only the 002 peak from the XRD results is observed without the superlattice 001 peak. This makes the phase of the IrMn₃ film under investigation ambiguous since the presence of only the 002 peak corresponds to a γ -IrMn₃.

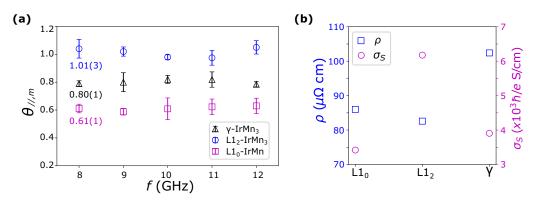


FIG. 5. (a) The measured SOT efficiencies ($\theta_{\parallel,m}$) of the three phases in the frequency range of 8 – 12 GHz with $\phi_H = -35^\circ$ and $\phi_E = 45^\circ$. (b) The electric conductivity (ρ) and spin Hall conductivity (σ_s) of the three phase of IrMn.

In Sec. IIC, we show that both the symmetric voltage amplitude V_S and the antisymmetric voltage amplitude V_A depend on $\sin(2\phi_H)\cos(\phi_H)$ in a bilayer dominated by the in-plane transverse spin polarization (s_v) and Oersted field, such as Pt/Py. Thus V_{mix} is expected to be centrosymmetric with H. However, Fig. 6(a) shows that V_{mix} measured from $L1_0$ -IrMn/Py is not a perfect inversion when H is reversed, where V_{sym} remains the same but V_{asy} is markedly different, indicating the presence of an additional out-of-plane torque. Therefore we perform the ST-FMR measurement with ϕ_H varying from -90° to 270° and fit V_S and V_A using Eq. (9) and (10). Figure 6(b) shows the typical results for L1₀-IrMn/Py. The associated fitting parameters are $S_A = -3.9 \pm 0.3 \,\mu \text{V}$, $S_B = 0.06 \pm 0.7 \,\mu\text{V}, \ S_C = -0.08 \pm 0.4 \,\mu\text{V}, \ A_A = -8.1 \pm$ $0.6 \,\mu\text{V}, A_B = -0.2 \pm 0.7 \,\mu\text{V}$, and $S_C = -0.6 \pm 0.2 \,\mu\text{V}$. All parameters with a larger fitting error than value are considered zero within experimental accuracy. Therefore V_S shows a usual shape with only the S_A contribution. In contrast, V_A has a nonzero A_C term on top of the normal A_A contribution. The fitted components of V_A are displayed separately in Fig. 6(c). We also observe similar dependence of V_{mix} on ϕ_H in L1₂-IrMn₃/Py bilayer and its V_A also has an A_C contribution, as demonstrated in Fig. 6(d). Among the possible components of V_A in Table I, $\sin(\theta_s)$ under Case 5 is the only term that shares the similar angular dependence of A_C , which implies the presence of an out-of-plane spin polarization (s_z). Moreover, according to Eq. (7), the out-of-plane torque associated with $\sin(\theta_s)$ (tagged as τ_{AC}) is dampinglike since it is even in **m**.

We perform two more tests to study τ_{AC} . First, we investigate the dependence of τ_{AC} on the in-plane electric field direction defined by ϕ_E ($\phi_E = 0^\circ || [100]$ of the L1₀-IrMn

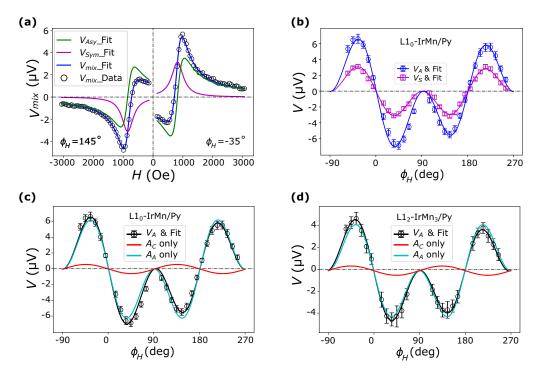


FIG. 6. (a) Raw data of measured voltage (V_{mix}) and its fit for L1₀-IrMn/Py at 9 GHz with $\phi_H = -35^\circ$ and 145°. (b) V_S , V_A , and their fits as a function of ϕ_H for L1₀-IrMn/Py. (c) V_A and its fitted component for L1₀-IrMn/Py and (d) L1₂-IrMn₃/Py. $\phi_E = 45^\circ$ in all plots.

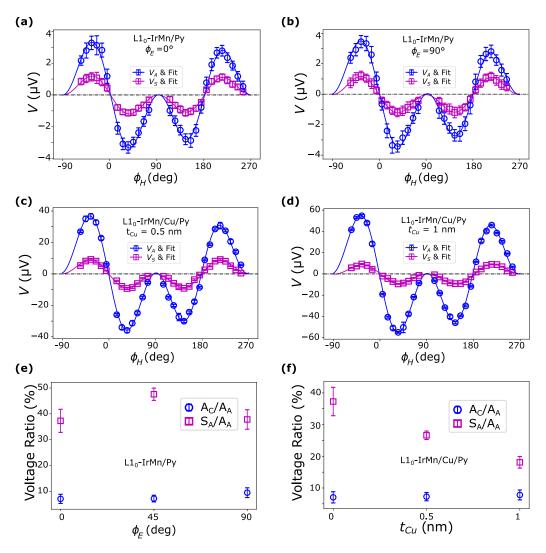


FIG. 7. (a) V_S , V_A , and their fits as a function of ϕ_H for L1₀-IrMn/Py with $\phi_E = 0^\circ$ and (b) 90°. (c) V_S , V_A , and their fits as a function of ϕ_H for L1₀-IrMn/Cu(t_{Cu})/Py with $t_{Cu} = 0.5$ nm and (d) $t_{Cu} = 1$ nm. $\phi_E = 0^\circ$ in (c) and (d). (e) Variation of normalized S_A and A_C with ϕ_E for L1₀-IrMn/Py and (f) t_{Cu} for L1₀-IrMn/Cu(t_{Cu})/Py.

lattice). This is achieved by patterning the microstrip along different in-plane directions on the same sample. Figures 7(a)and 7(b) show the ST-FMR voltages and their fittings for $\phi_E =$ 0° and 90° in L1₀-IrMn/Py, respectively. The results for both $\phi_E = 0^\circ$ and $\phi_E = 90^\circ$ are similar to those for $\phi_E = 45^\circ$ in Fig. 6(b). The magnitudes of A_C for $\phi_E = 0^\circ$, 45°, and 90° are normalized and summarized in Fig. 7(e). The ratio of A_C/A_A for L1₀-IrMn is less than 10% and it is roughly independent of ϕ_E . This is consistent with our prediction that the twin domains of L1₀-IrMn would nullify the directionality of τ_{AC} . Second, we insert a thin Cu layer between L10-IrMn and Py to break their strong exchange coupling. Figures 7(c) and 7(d) show that the measured V_A for $t_{Cu} = 0.5 \text{ nm}$ and $t_{Cu} = 1 \text{ nm}$ have a similar shape to that for $t_{Cu} = 0 \text{ nm}$ in Fig. 7(a), although the magnitude of V_A substantially increases after Cu is inserted due to lower impedance mismatch. The extracted values of A_C/A_A are summarized in Fig. 7(f). Essentially, A_C/A_A is also independent of the Cu spacer thickness in $L1_0$ -IrMn (22)/Cu(t_{Cu})/Py(13). This indicates that the interface between L1₀-IrMn and Py is not likely to account for τ_{AC} , which appears to originate from the AFM layer only. Notice that we have estimated the shunting effect due to the Cu spacer, which is negligible [22]. As a comparison, Figs. 7(e) and 7(f) also display S_A/A_A , which depends on both ϕ_E and t_{Cu} , implying the strong influence of exchange coupling. This further illustrates that the nature of τ_{AC} is different from the usual SOT.

IV. DISCUSSION

A. Symmetry analysis on SOT

The common τ_{AC} in both L1₀-IrMn and L1₂-IrMn₃ can be interpreted from symmetry analysis about the intrinsic effects of \mathcal{M}' on SOT. The magnetic mirror symmetry \mathcal{M}' differs from the crystal mirror symmetry \mathcal{M} . \mathcal{M}' contains a time reversal symmetry T, i.e., $\mathcal{M}' = \mathcal{M} * T$ [20]. An axial vector like the magnetic moment is odd in T whereas a polar vector like the electric field is even in T. As a result, the behaviors of a polar vector and an axial vector are opposite (same) under \mathcal{M} (\mathcal{M}'). This is illustrated in Fig. 8.

Based on these relations and inspired by the approach to analyze crystal mirror symmetry in Ref. [13], we scrutinize

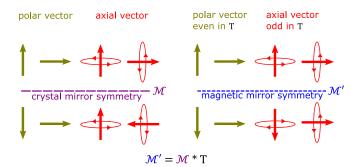


FIG. 8. Effects of symmetry operation \mathcal{M} and \mathcal{M}' on polar and axial vectors. \mathcal{M} and T represents the mirror symmetry and time reversal symmetry.

the effects of the magnetic mirror symmetry \mathcal{M}' on three parameters: (1) the in-plane and out-of-plane SOTs, which are collectively denoted as τ , (2) the electric field *E* and (3) the inplane angle ϕ_H , which defines the direction of magnetization. Notice that ϕ_H is always defined relative to the direction of *E*, as specified in the Fig. 2(a). As a result, the coordinate system (indicted by *x* in Fig. 9) relative to \mathcal{M} and \mathcal{M}' is changing with the direction of *E*. The effects of \mathcal{M}' and \mathcal{M} are compared as the following.

(1) It has been shown that τ is a pseudoscalar [13]. Therefore it remains the same when \mathcal{M}' is applied but it changes sign in the case of \mathcal{M} .

(2) In the cases of both \mathcal{M}' and \mathcal{M} , E is flipped if $E \perp \mathcal{M}'(\mathcal{M})$ and E is unchanged if $E \parallel \mathcal{M}'(\mathcal{M})$.

(3) When $E \perp \mathcal{M}'$, $x \perp \mathcal{M}'$. As a result, $\phi_H \rightarrow \pi - \phi_H$ when **m** is reflected by \mathcal{M}' , as shown in Fig. 9(a). When $E \parallel \mathcal{M}', x \parallel \mathcal{M}'$. As a result, $\phi_H \rightarrow -\phi_H$ when **m** is reflected by \mathcal{M}' , as shown in Fig. 9(b). Since **m** is an axial vector, it has to be flipped again after reflection if \mathcal{M} is applied in the place of \mathcal{M}' . This is because the crystal mirror symmetry \mathcal{M} does not contain the time reversal symmetry. Therefore, as shown in Fig. 9(c) and 9(d), $\phi_H \rightarrow -\phi_H$ for $E \perp \mathcal{M}$ and $\phi_H \rightarrow \pi - \phi_H$ for $E \parallel \mathcal{M}$.

Since SOT is a function of electric field and magnetization direction, i.e., $\tau = \tau (E, \phi_H)$, the effects of \mathcal{M}' on the three parameters pose constraints on τ if τ is Fourier expanded. The unconstrained SOT is $\tau(E, \phi_H) = E[F_0 + F_1 \cos(\phi_H) + F_2 \sin(\phi_H) + F_3 \cos(2\phi_H) + F_4 \sin(2\phi_H) + ...]$ [13]. For $E \perp \mathcal{M}'$ in Fig. 9(a), the net results of $E \rightarrow -E, \tau \rightarrow \tau$ and $\phi_H \rightarrow \pi - \phi_H$ is $\tau(-E, \pi - \phi_H) =$ $\tau(E, \phi_H)$. Since $\tau(E) = -\tau(-E)$ for current induced SOT, we have $\tau(\phi_H) = -\tau(\pi - \phi_H)$, which reduces the unconstrained form of τ to $\tau(E, \phi_H) =$

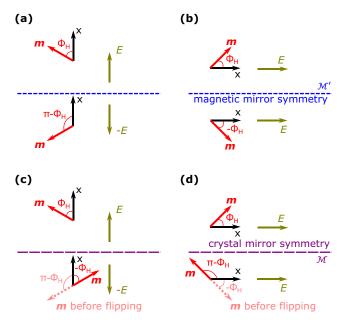


FIG. 9. (a) Effects of \mathcal{M}' on E and ϕ_H for $E \perp \mathcal{M}'$ and (b) $E \parallel \mathcal{M}'$. (c) Effects of \mathcal{M} on E and ϕ_H for $E \perp \mathcal{M}$ and (d) $E \parallel \mathcal{M}$. x is always parallel to E, it shows the coordinate system with which ϕ_H is defined. The dotted arrow of light red color in (c) and (d) indicates the direction of magnetic moment m after reflection by \mathcal{M} but before flipping.

 $E[F_1 \cos(\phi_H) + F_4 \sin(2\phi_H) + F_5 \cos(3\phi_H) + \ldots],$ where F_0 is not allowed. Similarly, it can be shown that for $E \parallel \mathcal{M}'$ in Fig. 9(b), $\tau(\phi_H) = \tau(-\phi_H)$, which means $\tau(E, \phi_H) =$ $E[F_0 + F_1 \cos(\phi_H) + F_3 \cos(2\phi_H) + F_5 \cos(3\phi_H) + \ldots],$ where F_0 is allowed. F_0 carries the similar physical significance as A_C and S_C since it is independent of ϕ_H . Therefore F_0 rationalizes the observed out-of-plane dampinglike torque. These results are summarized in Table II and compared with the effects of \mathcal{M} in WTe₂ [13]. Essentially, \mathcal{M} and \mathcal{M}' are similar that both allow the observed A_C and thus τ_{AC} , but they differ in the direction of E with which this torque is allowed. Note that though F_0 is also allowed for the in-plane SOT, the measured S_C is negligible. We could not identify any material-dependent microscopic origins other than the inhomogeneous contact of probe (Sec. IIC), which we have minimized with the best of our engineering effort. The higher-order components, such as F_3 , F_4 and F_5 , have been demonstrated to be much smaller in magnitude comparing to the lower-order components [13]. We have not observed the higher-order components neither.

TABLE II. Effects of \mathcal{M}' and \mathcal{M} on SOT. The first two columns are adapted from Ref. [13].

	M	\mathcal{M}'		
$E \bot \mathcal{M}$	$E \parallel \mathcal{M}$	$E ot \mathcal{M}'$	$E \parallel \mathcal{M}'$	
$E \rightarrow -E$	$E \rightarrow E$	$E \rightarrow -E$	$E \rightarrow E$	
au ightarrow - au	au ightarrow - au	au ightarrow au	au ightarrow au	
$\phi_H ightarrow - \phi_H$	$\phi_H o \pi - \phi_H$	$\phi_H o \pi - \phi_H$	$\phi_H ightarrow - \phi_H$	
$\tau(-\phi_H) = \tau(\phi_H)$	$ au(\phi_H) = - au(\pi - \phi_H)$	$ au(\phi_H)=- au(\pi-\phi_H)$	$ au(-\phi_H) = au(\phi_H)$	
ϕ_H independent term allowed	ϕ_H independent term forbidden	ϕ_H independent term forbidden	ϕ_H independent term allowed	

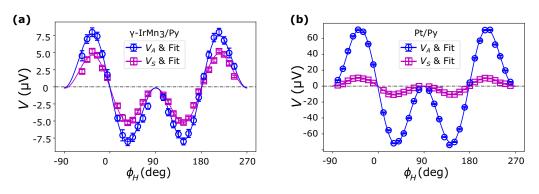


FIG. 10. (a) Absence of the out-of-plane τ_{DL} in samples without magnetic asymmetry, such as γ -IrMn₃ (22)/Py (13) with $\phi_E = 45^\circ$ and (b) Pt (10)/Py (10).

B. Absence of τ_{AC} in control samples without magnetic asymmetry

In order to verify the effects of magnetic asymmetry, we try to observe τ_{AC} in two control samples: γ -IrMn₃(22)/Py(13) and polycrystalline Pt(10)/Py(10). In γ -IrMn₃, the magnetic asymmetry is absent due to disordered magnetic structure, and Pt is known to have negligible magnetic moment let alone any magnetic asymmetry. As expected, τ_{AC} is absent in both cases for any choice of ϕ_E . Figure 10 shows the typical spectra of voltages for the two control samples. Both V_S and V_A shows the usual $\sin(2\phi_H)\cos(\phi_H)$ dependence. This shows that τ_{AC} is strongly correlated with the magnetic asymmetry.

C. Microscopic origin of the out-of-plane τ_{DL}

In Sec. IIIC, the microscopic origin of τ_{AC} is correlated with the out-of-plane spin polarization s_z , which can be rationalized from both the SHE and the REE perspectives. If τ_{AC} is generated via the SHE, at least one of σ^z_{zx} and σ^z_{zy} in the spin Hall conductivity (σ^k_{ij}) tensor should be nonzero, where i, j, and k indicate the orientations of spin current, electric current, and spin polarization, respectively. We find $\sigma^z_{zx} = 68 \frac{\hbar}{e} \frac{S}{cm}$ for L1₀-IrMn [22] and $\sigma^z_{zx} = 95 \frac{\hbar}{e} \frac{S}{cm}$ for L1₂-IrMn₃ [8], both are more than 50% of their respective σ^y_{zx} that accounts for the usual in-plane τ_{DL} .

On the other hand, if τ_{AC} is generated via the REE, the inversion symmetry in the IrMn/Py bilayer should be broken. Figure 11 shows that the nonmagnetic symmetry of the mag-

netic sites is 4mm for both L10-IrMn and L12-IrMn3, which indicates that the inversion symmetry is broken along the caxis. In a 2D Rashba model for the 4mm point group [21,37], the spin polarization s induced by an electric field E can be calculated using the Kubo linear response formalism ($s = \chi E$, where χ is the response tensor) [21]. When magnetic moment is not considered, χ only has the usual x_{21} term (meaning an $E \parallel x$ induces an $s \parallel y$), which accounts for the usual S_A and A_A terms in the ST-FMR measurement. It is the contribution from magnetic moments that generates the nonzero x_{31} and x_{32} [21], which are responsible for the s_7 . Moreover, in the context of REE-based SOT, AFM of different magnetic structures but same symmetry (4mm) can generate SOT of similar symmetry and strength [21]. This is consistent with our observation that τ_{AC} is induced by the common broken \mathcal{M}' in L1₀-IrMn and L1₂-IrMn₃ despite their different magnetic structures. The nonmagnetic symmetry of magnetic sites in the 2D Rashba model, though differed from the magnetic symmetry shown in Fig. 1(b), is useful to analyze the effects of magnetic asymmetry on SOT since it demonstrates the symmetry-breaking by magnetic moments.

V. CONCLUSION

The observed out-of-plane τ_{DL} is different from previously reported incidences that are only phenomenologically similar. In WTe₂, the low crystal symmetry is responsible for a net s_z [13]. However, the crystal symmetries of both L1₀-IrMn

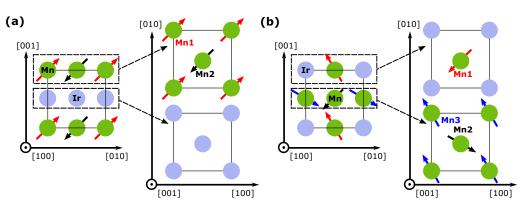


FIG. 11. (a) Interfacial symmetry in L1₀-IrMn and (b) L1₂-IrMn₃.

(*P4/mmm*) and L1₂-IrMn₃ (*Pm* $\bar{3}m$) are high and their (001) planes have fourfold rotational symmetry (Fig. 11). In FM systems, the unusual s_z has been attributed to the spin-orbit precession [12], the spin rotation symmetry [14], the AMR effect and the anomalous Hall effect [15,16,38]. These processes require the net magnetization of FM and s_z is very sensitive to interfaces. In contrast, we have verified that IrMn as an AFM has no net magnetization (not shown) and, as shown in Fig. 7(f), inserting a Cu layer between Py and IrMn does not significantly change the strength of τ_{AC} .

In conclusion, we have measured the SOT efficiencies of three phases of IrMn and demonstrated an out-of-plane τ_{DL} in both L1₀-IrMn/Py and L1₂-IrMn₃/Py. This anomalous SOT is strongly correlated with the magnetic asymmetry in IrMn and is consistent with predictions of the SHE and the REE considering an out-of-plane spin polarization. This out-ofplane spin polarization is favorable for electrical switching of a perpendicularly magnetized ferromagnet, which is compatible with the contemporary high-density magnetic memories. Moreover, unlike FM-based spintronics, AFM materials enjoy many advantages, such as zero stray field, higher stability and faster dynamics [1,21,26]. Therefore the possibility to control SOT via magnetic asymmetry of AFM not only helps to understand the origin of SOT, but also fuels the development of next-generation AFM spintronics.

The Department of Energy will provide public access to these results of federally sponsored research in accordance with the DOE Public Access Plan [39].

- A. Manchon, J. Železný, I. M. Miron, T. Jungwirth, J. Sinova, A. Thiaville, K. Garello, and P. Gambardella, Rev. Mod. Phys. 91, 035004 (2019).
- [2] J. Sinova, S. O. Valenzuela, J. Wunderlich, C. H. Back, and T. Jungwirth, Rev. Mod. Phys. 87, 1213 (2015).
- [3] A. Manchon and S. Zhang, Phys. Rev. B, 78, 212405 (2008).
- [4] L. Liu, T. Moriyama, D. C. Ralph, and R. A. Buhrman, Phys. Rev. Lett. **106**, 036601 (2011).
- [5] I. M. Miron, G. Gaudin, S. Auffret, B. Rodmacq, A. Schuhl, S. Pizzini, J. Vogel, and P. Gambardella, Nat. Mater. 9, 230 (2010).
- [6] Y. Fan, P. Upadhyaya, X. Kou, M. Lang, S. Takei, Z. Wang, J. Tang, L. He, L. Chang, M. Montazeri, G. Yu, W. Jiang, T. Nie, R. N. Schwartz, Y. Tserkovnyak, and K. L. Wang, Nat. Mater. 13, 699 (2014).
- [7] C. Pai, L. Liu, Y. Li, H. W. Tseng, D. C. Ralph, and R. A. Buhrman, Appl. Phys. Lett. 101, 122404 (2012).
- [8] W. Zhang, W. Han, S. Yang, Y. Sun, Y. Zhang, B. Yan, and S. S. P. Parkin, Sci. Adv. 2, e1600759 (2016).
- [9] W. Zhang, M. B. Jungfleisch, W. Jiang, J.E. Pearson, A. Hoffmann, F. Freimuth, and Y. Mokrousov, Phys. Rev. Lett. 113, 196602 (2014).
- [10] P. Wadley, B. Howells, J. Železný, C. Andrews, V. Hills, R. P. Campion, V. Novák, K. Olejník, F. Maccherozzi, S. S. Dhesi, S. Y. Martin, T. Wagner, J. Wunderlich, F. Freimuth, Y. Mokrousov, J. Kuneš, J. S. Chauhan, M. J. Grzybowski, A. W.

ACKNOWLEDGMENTS

The Laboratory is a National Research Infrastructure under the National Research Foundation (NRF) Singapore. J.C. is a member of Singapore Spintronics Consortium (SG-SPIN). This research is supported by the Singapore Ministry of Education MOE2018-T2-2-043, AMEIRG18-0022, A*STAR IAF-ICP 11801E0036, and MOE Tier1 R-284-000-195-114. This project was partially supported by the National Key Research and Development Program of China (Title: Nonvolatile and programmable spin logics based on magnetic heterostructures; Grant No. 2017YFA0206200) for the device micro-fabrications. We would like to acknowledge Singapore Synchrotron Light Source (SSLS) for providing the facility necessary for conducting the research. P.Y. was supported by SSLS via NUS Core Support C-380-003-003-001. A portion of this research used resources at the High Flux Isotope Reactor and Spallation Neutron Source, US-DOE Office of Science User Facilities operated by the Oak Ridge National Laboratory. B.Y. acknowledges the financial support by a research grant from the Benoziyo Endowment Fund for the Advancement of Science. Y.L. and T.H. received support from UT-Batelle, LLC under Contract No. DE-AC05-000R22725 with the U.S. Department of Energy.

The United States Government retains and the publisher, by accepting the article for publication, acknowledges that the United States Government retains a nonexclusive, paid-up, irrevocable, world-wide license to publish or reproduce the published form of this manuscript, or allow others to do so, for United States Government purposes.

Rushforth, K. W. Edmonds, B. L. Gallagher, and T. Jungwirth, Science **351**, 587 (2016).

- [11] D. Fang, H. Kurebayashi, J. Wunderlich, K. Vyborny, L. P. Zarbo, R. P. Campion, A. Casiraghi, B. L. Gallagher, T. Jungwirth, and A. J. Ferguson, Nat. Nanotechnol. 6, 413 (2011).
- [12] S. C. Baek, V. P. Amin, Y. W. Oh, G. Go, S. J. Lee, and G. H. Lee, Nat. Mater. 17, 509 (2018).
- [13] D. MacNeill, G. M. Stiehl, M. H. D. Guimaraes, R. A. Buhrman, J. Park, and D. C. Ralph, Nat. Phys. 13, 300 (2017).
- [14] A. M. Humphries, T. Wang, E. R.J. Edwards, S. R. Allen, J. M. Shaw, H. T. Nembach, J. Q. Xiao, T. J. Silva, and X. Fan, Nat. Comm. 8, 911 (2017).
- [15] A. Bose, D. D. Lam, S. Bhuktare, S. Dutta, H. Singh, Y. Jibiki, M. Goto, S. Miwa, and A. A. Tulapurkar, Phys. Rev. Appl. 9, 064026 (2018).
- [16] Y. Ou, Z. Wang, C. S. Chang, H. P. Nair, H. Paik, N. Reynolds, D. C. RalphDavid, A. MullerDarrell, G. SchlomRobert, and A. Buhrman, Nano Lett. **19**, 3663 (2019).
- [17] M. H. D. Guimarães, G. M. Stiehl, D. MacNeill, N. D. Reynolds, and D. C. Ralph, Nano Lett. 18, 1311 (2018).
- [18] J. Zhou, J. Qiao, A. Bourne, and W. Zhao, Phys. Rev. B 99, 060408(R) (2019).
- [19] G. M. Stiehl, R. Li, V. Gupta, I. E. Baggari, S. Jiang, H. Xie, L. F. Kourkoutis, K. F. Mak, J. Shan, R. A. Buhrman, and D. C. Ralph, Phys. Rev. B **100**, 184402 (2019).

- [20] D. E. Laughlin, M. A. Willard, and M. E. McHenry, *Phase Transformations and Evolution in Materials* (The Minerals, Metals and Materials Society, Warrendale, PA, 2000), pp. 121–137.
- [21] J. Železný, H. Gao, A. Manchon, F. Freimuth, Y. Mokrousov, J. Zemen, J. Mašek, J. Sinova, and T. Jungwirth, Phys. Rev. B 95, 014403 (2017).
- [22] J. Zhou, X. Wang, Y. Liu, J. Yu, H. Fu, L. Liu, S. Chen, J. Deng, W. Lin, X. Shu, H. Y. Yoong, T. Hong, M. Matsuda, P. Yang, S. Adams, B. Yan, X. Han, and J. S. Chen, Sci. Adv. 5, eaau6696 (2019).
- [23] P. Wadley, S. Reimers, M. J. Grzybowski, C. Andrews, M. Wang, J. S. Chauhan, B. L. Gallagher, R. P. Campion, K. W. Edmonds, S. S. Dhesi, F. Maccherozzi, V. Novak, J. Wunderlich, and T. Jungwirth, Nat. Nanotechol 13, 362 (2018).
- [24] O. G. Shpyrko, E. D. Isaacs, J. M. Logan, Yejun Feng, G. Aeppli, R. Jaramillo, H. C. Kim, T. F. Rosenbaum, P. Zschack, M. Sprung, S. Narayanan, and A. R. Sandy, Nature (London) 447, 68 (2007).
- [25] T. Zhao, A. Scholl, F. Zavaliche, K. Lee, M. Barry, A. Doran, M. P. Cruz, Y. H. Chu, C. Ederer, N. A. Spaldin, R. R. Das, D. M. Kim, S. H. Baek, C. B. Eom, and R. Ramesh, Nat. Mater. 5, 823 (2006).
- [26] E. V. Gomonay and V. M. Loktev, Low Temp. Phys. 40, 17 (2014).
- [27] S. Zhang, P. M. Levy, and A. Fert, Phys. Rev. Lett. 88, 236601 (2002).

- [28] M. Harder, Z. X. Cao, Y. S. Gui, X. L. Fan, and C. M. Hu, Phys. Rev. B, 84, 054423 (2011).
- [29] S. Fukami, C. Zhang, S. DuttaGupta, A. Kurenkov, and H. Ohno, Nat. Mater. 15, 535 (2016).
- [30] A. Kohn, A. Kovacs, R. Fan, G. J. McIntyre, R. C. C. Ward, and J. P. Goff, Sci. Rep. 3, 2412 (2013).
- [31] I. Tomeno, H. N. Fuke, H. Iwasaki, M. Sahashi, and Y. Tsunoda, J. Appl. Phys. 86, 3853 (1999).
- [32] E. Kren, G. Kadar, L. Pal, J. Solyom, P. Szabo, and T. Tarnczi, Phys. Rev. 171, 574 (1968).
- [33] W. Zhang, M. B. Jungfleisch, F. Freimuth, W. Jiang, J. Sklenar, J. E. Pearson, J. B. Ketterson, Y. Mokrousov, and A. Hoffmann, Phys. Rev. B 92, 144405 (2015).
- [34] H. Saglam, J. C. Rojas-Sanchez, S. Petit, M. Hehn, W. Zhang, J. E. Pearson, S. Mangin, and A. Hoffmann, Phys. Rev. B 98, 094407 (2018).
- [35] V. Tshitoyan, C. Ciccarelli, A. P. Mihai, M. Ali, A. C. Irvine, T. A. Moore, T. Jungwirth, and A. J. Ferguson, Phys. Rev. B 92, 214406 (2015).
- [36] X. Shu, J. Zhou, J. Deng, W. Lin, J. Yu, L. Liu, C. Zhou, P. Yang, and J. Chen, Phys. Rev. Materials 3, 114410 (2019).
- [37] J. Železný, H. Gao, K. Výborný, J. Zemen, J. Mašek, A. Manchon, and J. Wunderlich, Phys. Rev. Lett. 113, 157201 (2014).
- [38] T. Taniguchi, J. Grollier, and M. D. Stiles, Phys. Rev. Appl. 3, 044001 (2015).
- [39] http://energy.gov/downloads/doe-public-access-plan.