

Role of correlations in determining the Van Hove strain in Sr₂RuO₄

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Uniaxial pressure applied along a Ru-O-Ru bond direction induces an elliptical distortion of the largest Fermi surface of Sr₂RuO₄, eventually causing a Fermi surface topological transition, also known as a Lifshitz transition, into an open Fermi surface. There are various anomalies in low-temperature properties associated with this transition, including maxima in the superconducting critical temperature and in resistivity. In the present paper, we report refined measurements of the strain at which this transition occurs, employing apparatus in which the stress on the sample is measured, and resonant ultrasound measurement of the low-temperature elastic moduli. The Lifshitz transition is found to occur at a longitudinal strain ϵ_{xx} of $(-0.44 \pm 0.06) \times 10^{-2}$, which corresponds to a B_{1g} strain $\epsilon_{xx} - \epsilon_{yy}$ of $(-0.66 \pm 0.09) \times 10^{-2}$. This is considerably smaller than the strain corresponding to a Lifshitz transition in density functional theory calculations, even if the spin-orbit coupling is taken into account. Using dynamical mean-field theory, we show that electronic correlations reduce the critical strain. It turns out that the orbital anisotropy of the local Coulomb interaction on the Ru site is, furthermore, important to bring this critical strain close to the experimental number and thus well into the experimentally accessible range of strains.

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I. INTRODUCTION

Uniaxial stress has proved to be a valuable tool for investigating the unconventional superconductor Sr₂RuO₄. Stress applied along a Ru-O-Ru bond direction, whether compressive or tensile, causes the critical temperature T_c to increase [1]. Under a stronger stress, T_c passes through a pronounced peak, reaching a value more than twice T_c of unstressed Sr₂RuO₄ [2]. A plethora of other nontrivial renormalizations were subsequently detected at or near the same critical strain. The upper critical field H_{c2} at the same strain is enhanced by a factor of approximately 20 [2]. The low-temperature resistivity is observed to peak there, and the temperature exponent is observed to be reduced from the Fermi liquid value to about 1.5 [3]. Similar enhancement manifests itself in O Knight shift [4], consistent with increased partial density of O states on one O and an overall enhancement of ferromagnetic spin fluctuations on all sites. There is a maximum in the heat capacity, consistent with an increase in the

density of states [5]. While the detailed mechanism of the T_c and H_{c2} enhancement is yet unclear, all observations strongly point to a strain-induced Lifshitz transition in the γ Fermi surface sheet of Sr₂RuO₄, bringing a Van Hove singularity (VHS) in the electronic density of states to the Fermi level. In-plane anisotropic stress causes an elliptical distortion of this Fermi surface sheet, until it eventually transitions into an open Fermi surface. The hypothesized Lifshitz transition was directly observed by angle-resolved photoemission (ARPES) on a sample of Sr₂RuO₄ mounted on a sample stage that utilizes differential thermal contraction to induce anisotropic strain [6].

Throughout the measurements listed above, there is uncertainty in the strain ϵ_{VHS} at which the Lifshitz transition occurs. The range of values reported for ϵ_{VHS} , generally between -5 and -7×10^{-3} , indicates the level of uncertainty. (Negative values of ϵ denote compression.) Except for the ARPES measurement, all measurements were performed with a piezoelectric-based uniaxial stress apparatus in which the applied strain was determined using a displacement sensor placed in parallel with the sample. This means that the sensor measures the sum of displacements arising from sample strain, deformation of the epoxy holding the sample, and deformation of the apparatus overall. The uncertainty in ϵ_{VHS} arises chiefly from difficulties in subtracting off these additional contributions. More accurate determination of ϵ_{VHS} is important as a point of metrology and for understanding

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the electronic structure. With this aim, some of us recently developed a uniaxial stress apparatus that incorporates a force sensor placed in series with the sample [7]. The reading from the force sensor is, except for minor parasitic coupling, independent of displacement applied to the sample and epoxy and therefore is a much more accurate and repeatable measure of the state of the sample. In Ref. [7], the resistivity ρ of Sr_2RuO_4 was found to peak, for temperature $T = 5$ K, at a uniaxial stress of $\sigma_{xx} = -0.7$ GPa.

Here, we report three results. (1) Employing new measurements of the elastic moduli of Sr_2RuO_4 , and measurement of the stress dependence of resistivity and magnetic susceptibility from a second sample, we determine that the Lifshitz transition occurs at a longitudinal strain $\varepsilon_{xx} = (-0.44 \pm 0.06) \times 10^{-2}$; we label this strain value ε_{VHS} . If we take into account Poisson's ratio expansion along the transverse direction, ε_{VHS} corresponds to a B_{1g} strain $\varepsilon_{xx} - \varepsilon_{yy}$ of $\varepsilon_{\text{VHS},B_{1g}} = (-0.66 \pm 0.09) \times 10^{-2}$. (2) Although results for ε_{VHS} from electronic structure calculations based on density functional theory (DFT) are sensitive to the precise calculation method, ε_{VHS} is consistently overestimated in these calculations. (3) Electronic correlations within dynamical mean-field theory (DMFT) substantially reduce ε_{VHS} . Furthermore, the shape of the Fermi surface and thus also the critical strain subtly depend on the anisotropy of the local Coulomb interaction on the Ru site.

We note that in our determination of ε_{VHS} , we identify the peak in low-temperature resistivity versus strain as the Lifshitz transition. As we show below, T_c peaks at a strain a few percent larger than that at which resistivity peaks. The difference is small enough that identifying the peak in T_c as the Lifshitz transition would not alter the conclusions of this paper; however, on general grounds it is more likely the peak in resistivity than in T_c that marks the Lifshitz transition. Scattering between a "cold," i.e. conventional, Fermi surface and high-density-of-states hot spots was analyzed in Refs. [8,9] and found to give at low temperatures a peak in resistivity at the Lifshitz transition. Because the peak in the density of the states at the Lifshitz transition is actually rather modest compared with that of the cold Fermi surfaces of Sr_2RuO_4 [4], it may be that spin-fluctuation scattering, which is directly sensitive to the density of states via the Stoner renormalization, gives a larger contribution to the peak in resistivity than hot spot scattering. Strong enhancement of spin fluctuations, like hot spot scattering [8,9], could also give the observed $T^{1.5}$ temperature dependence of resistivity at the peak [10]. In both of these models, resistivity peaks at the Lifshitz transition. T_c , on the other hand, involves both the density of states and the pairing function, and the latter does not need to be strongest at the Lifshitz transition.

II. EXPERIMENTAL RESULTS

The stress apparatus is illustrated in Fig. 1(a), and full details of its design are given in Ref. [7]. Piezoelectric actuators underneath the device apply a displacement to moving block A, to which one end of the sample is attached. The other end is secured to moving block B, which is secured to the outer frame through thick titanium bars, labeled flexures in Fig. 1(a). Under force applied from the sample, these flexures

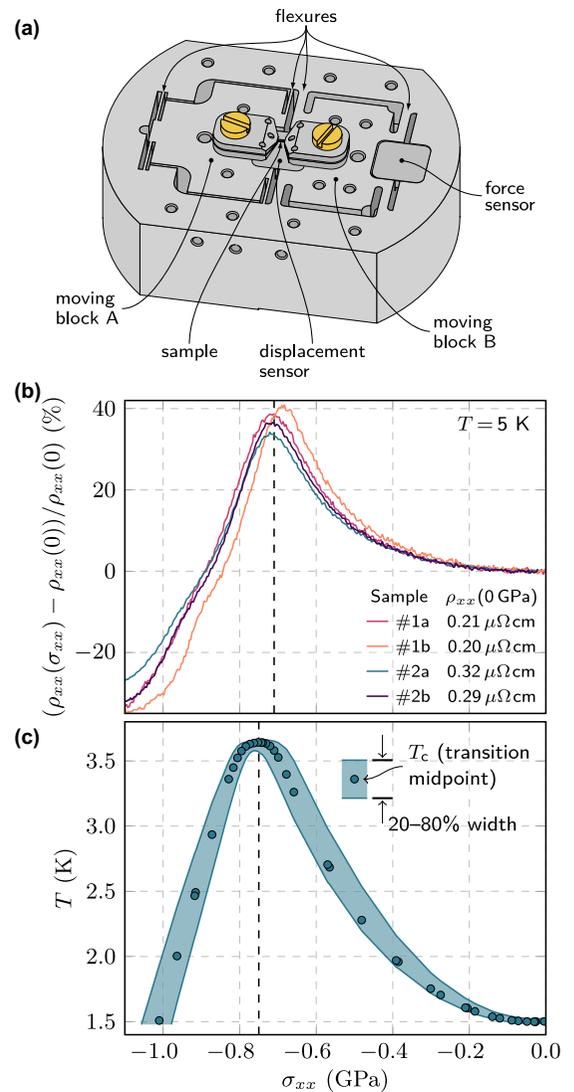


FIG. 1. (a) Schematic of the uniaxial stress cell incorporating both a displacement and force sensor. Both sensors are parallel-plate capacitors, placed underneath protective covers. (b) The resistivity of two different samples as a function of applied stress. For redundancy, two pairs of voltage contacts were attached to each sample; the letters a and b refer to pairs of voltage contacts on opposite sides of the same sample. (c) T_c of the second sample, determined through magnetic susceptibility measurements and measured during the same cooldown as resistivity as a function of applied stress.

bend slightly, allowing block B to move. A parallel-plate capacitive sensor between the block and outer frame measures the displacement of the block from its zero-force position Δx . The force on the sample is then determined as $F = k\Delta x$, where k is the spring constant of the titanium flexures. k was calculated to be $20 \text{ N}/\mu\text{m}$ and was measured at room temperature to be $19 \pm 2 \text{ N}/\mu\text{m}$. The Young's modulus of titanium increases by $(14 \pm 1)\%$ between 300 and 0 K [11], so we take the low-temperature value of k to be $22 \pm 2 \text{ N}/\mu\text{m}$. (All experimental error bars are 2σ , where σ is one standard deviation.)

The relationship between the capacitance of the force sensor C and Δx was measured at both room and cryogenic

temperatures using a fiber-based interferometer. The results are well fitted with a parallel-plate form: $C = \epsilon_0 A / (x_0 + \Delta x) + C_{\text{offset}}$, where $A = 3.76 \text{ mm}^2$ is the area of the capacitor plates and x_0 is the plate spacing at zero force. C_{offset} is an offset due to stray capacitance within the cell and was measured to be $0.29 \pm 0.05 \text{ pF}$. C was generally around 2 pF . To determine Δx it is also necessary to know the zero-force reading of the force sensor C_0 . Because C_0 varies with temperature and drifts over time with repeated thermal cycling, we measured C_0 *in situ* by compressing the sample until it fractured completely and then pulling apart the remnants, such that no force could have been applied through the sample to the force block. Δx is then calculated as $\Delta x = \epsilon_0 A [(C - C_{\text{offset}})^{-1} - (C_0 - C_{\text{offset}})^{-1}]$.

Two samples were measured. The resistivity data for sample 1 are also reported in Ref. [7]. For sample 2, both resistivity and magnetic susceptibility were measured; susceptibility was determined through measurement of the mutual inductance between two concentric coils, the larger with a diameter of $800 \mu\text{m}$, placed on top of the sample. Sample 1 had dimensions $L_{\text{exp}} \times W \times H$ (where L_{exp} is the length of the exposed central portion of the sample, W is the width, and H is the height along the c axis) of $1.0 \times 0.37 \times 0.10 \text{ mm}^3$, and sample 2 had dimensions of $1.0 \times 0.34 \times 0.08 \text{ mm}^3$. Both samples were cut from the same original rod; however, they were found to have slightly different residual resistivities. The resistivity at 5 K for both samples is presented in Fig. 1(b). The peak in resistivity occurs at a stress $\sigma_{xx} = -0.71 \pm 0.08 \text{ GPa}$, where the dominant source of error is the 10% uncertainty on k .

To convert the measured stress to strain, resonant ultrasound spectroscopy (RUS) measurements were performed to determine the low-temperature elastic moduli of Sr_2RuO_4 . A single crystal was polished to a cuboid geometry with faces perpendicular to the $[110]$, $[1\bar{1}0]$, and $[001]$ crystallographic directions. The dimensions are $1.5 \times 1.6 \times 1.4 \text{ mm}^3$, with 1.4 mm along the tetragonal c axis. To perform RUS measurements, the sample is held in weak-coupling contact between two piezoelectric transducers, and the excitation frequency of one transducer is swept while detecting the quadrature response on the second transducer. The lowest 50 mechanical resonance frequencies—which are determined by the dimensions, density, and elastic constants of the sample—were measured in this way and are tabulated in Table I. The elastic constants are then obtained by inverse solving the three-dimensional elastic wave equation following the procedure outlined in Ramshaw *et al.* [12].

The room-temperature Young's modulus for compression along a $\langle 100 \rangle$ direction, $Y_{100} \equiv \sigma_{xx} / \epsilon_{xx}$, was found to be 180 GPa , compared with 176 GPa reported in Ref. [13]. At 4 K , Y_{100} is $160.2 \pm 0.8 \text{ GPa}$, where the error bars were determined by a bootstrap method. The in-plane Poisson's ratio for $\langle 100 \rangle$ compression is $-\epsilon_{yy} / \epsilon_{xx} = 0.508 \pm 0.006$, and the out-of-plane Poisson's ratio $-\epsilon_{zz} / \epsilon_{xx} = 0.163 \pm 0.004$. These parameters yield $\epsilon_{\text{VHS}} = (-0.44 \pm 0.06) \times 10^{-2}$.

T_c of sample 2, shown in Fig. 1(c), peaks at a strain about 5% larger than that at which resistivity peaks. In previous measurements on two further samples, the difference was found to be 11% and 12% [3].

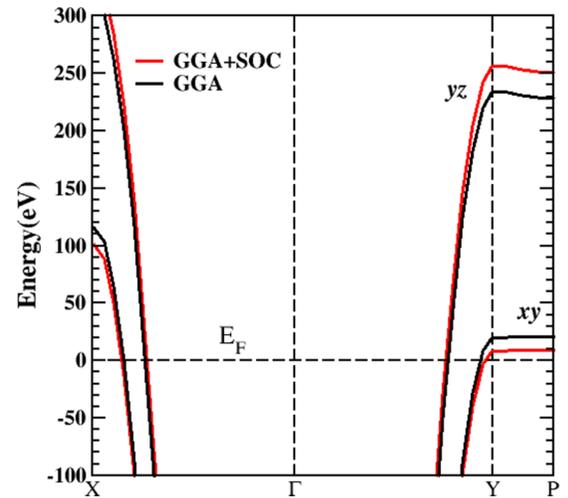


FIG. 2. Band structure calculated within GGA [14] and GGA+SOC for $\epsilon_{xx} = -0.75\%$ using the full-potential WIEN2k code [15].

III. CALCULATION RESULTS

We note first a technical point. The low-temperature elastic moduli were not available at the time the calculations were performed, and therefore, the room-temperature Poisson's ratios were employed. The room-temperature in-plane Poisson's ratio is 0.39 [13], against, as stated above, 0.51 at low temperature. The electronic structure of Sr_2RuO_4 is much more sensitive to the B_{1g} strain $\epsilon_{xx} - \epsilon_{yy}$ than the A_{1g} strain $\epsilon_{xx} + \epsilon_{yy}$ [6]. In the theory portion of this paper we quote strains as calculated, and for comparison with experiment they should be scaled by $1.39/1.51$, which matches the B_{1g} strain. This is not a large enough factor to alter any of our conclusions.

We start with results of straightforward DFT calculations with and without SOC (technical details are described in the Appendix). In Fig. 2 the calculated band structure near the Y point is shown. Three points are worth noting. First, at the Y point there are two unoccupied states: xy and yz . In the simplest tight-binding model, they do not hybridize, and we know that in full electronic structure calculations this hybridization is indeed very weak. However, including SOC triggers interaction between the two states. Indeed, while the matrix element of the L_z operator between the two states is zero, that of the L_{\pm} operators is ± 1 . This pushes the lowest state (xy) down by an amount of the order of $\lambda^2 / (E_{yz} - E_{xy})$, where λ is the SOC constant, thus reducing the critical strain, an effect already noticed in Ref. [2]. Second, a technical but very important point is that ϵ_{VHS} is sensitive to the Fermi energy on a scale of a few meV, and a fine-scale discretization of the Brillouin zone is required to achieve convergence on this level. We increased the number of k points until convergence on this scale was achieved. Third, the exact position of the Van Hove singularity $E_{xy} - E_F$ in the unstrained structure, and, correspondingly, the calculated ϵ_{VHS} , depends on the flavor of the density functional used, with the gradient-corrected functional [the generalized gradient approximation (GGA)] giving a smaller $E_{xy} - E_F$ than the local density approximation (LDA).

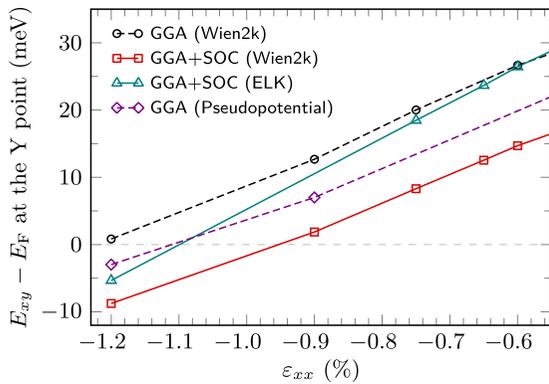


FIG. 3. Energy of the xy band at the Y point as a function of ε_{xx} , as obtained from various codes and approximations. Solid lines: GGA + SOC; dashed lines: GGA [14]. The Van Hove strain ε_{VHS} is the strain where $E_{xy} - E_F$ crosses zero.

In order to further assess the sensitivity of this parameter to computational details, we have compared calculations using two all-electron augmented plane wave methods: WIEN2k [15] and Elk [16]. Calculations were performed using the optimized structures reported in Ref. [2] at each strain. First, we observe that GGA (Perdew-Burke-Ernzerhof functional [14]) indeed gives ε_{VHS} smaller by 10^{-3} than LDA (Perdew-Wang PW [17]). Second, adding SOC substantially reduces, as expected, the calculated Van Hove strain, which without SOC comes to at least -1.1×10^{-2} : More than twice as much as the measurement (see Fig. 3). This is also higher than the value calculated in Ref. [2], which may be due to the fact that calculations employed a different basis set, namely, local orbitals, and/or that it used the Dirac rather than the Pauli equation to describe relativistic effects. In all cases, one may conclude that DFT+SOC does not fully reproduce ε_{VHS} .

This is not surprising *per se* because it is well known that correlation effects may dramatically renormalize effective masses and crystal fields. This is liable to alter the relative occupations of different bands and affect both the $E_{yz} - E_{xy}$ separation and the position of the Fermi level with respect to the xy band and thus the γ sheet of the Fermi surface. Indeed, by using straightforward DFT+DMFT calculations, Zhang *et al.* [18] found that for unstrained Sr_2RuO_4 the Van Hove singularity shifts much closer to the Fermi level, thereby stretching the γ sheet towards the X and Y points. In fact, this deteriorates the agreement of the Fermi surface geometry with experiment. It was shown, however, in the same work that using orbital-dependent (i.e., anisotropic) local Coulomb parameters for the Ru t_{2g} manifold (as suggested by the constrained-random-phase-approximation calculation of these parameters [19]) radically improves the agreement with experiment. We will show below that accounting for this anisotropy is even more important under strain.

With this in mind, we have performed DFT+DMFT calculations using the rotationally invariant slave-boson (RISB) [20,21] scheme as an impurity solver to reveal the effect of the interplay between band structure, spin-orbit coupling, and electron correlations on the critical strain. In line with Ref. [18] as well as other works [22,23], we choose $U = 3.1$ eV and $J_H = 0.7$ eV for the isotropic Coulomb

parameters in the correlated subspace spanned by the three Wannier t_{2g} orbitals of Ru($4d$) character. The SOC constant λ is set to 0.09 eV, consistent with previous DFT examinations [24].

As a first result, we confirm previous findings for the k -resolved spectral function of Sr_2RuO_4 with SOC and local Coulomb interactions: electronic correlations cause substantial band renormalization. Second, we find that correlations enhance an effect of SOC of shifting apart different Fermi-surface sheets, leading also to a slight redistribution of electrons across those sheets. Away from the Fermi surface, the enhanced SOC and a correlation-reduced energy splitting nearly cancel [25], such that the effective SOC-driven avoided crossings are not strongly affected by correlations [23,25,26]. In our calculations, at the Γ point the splitting between the β and γ bands, which at Γ are both well below E_F , decreases from 91 to 78 meV when interactions are introduced.

Crucially, for the present discussion, correlations beyond DFT lead to growth of the γ sheet relative to the α and β sheets. As a result, ε_{VHS} is considerably reduced compared to the DFT value, in line with the previous work by Facio *et al.* [23], in which they found the critical strain to be -0.2% . For our chosen set of local Coulomb parameters and including SOC, the critical strain is shifted essentially to zero. In essence, ε_{VHS} obtained from DFT is *too large*, while from conventional DFT+DMFT it is *too small*.

Stimulated by the findings of the importance of anisotropy in the local Coulomb parameters [18], we introduced orbital differentiation within the Hubbard U matrix. The xy Wannier function is larger in spatial extent compared to $\{xz, yz\}$, making it more sensitive to the Coulomb repulsion among the $O(2p)$ electrons [27,28]. Therefore, a larger $U_{xy,xy}$ matrix element is expected, consistent with Ref. [19]. We treat the intraorbital Coulomb anisotropy $u = U_{xy,xy} - U_{\{xz,yz\},\{xz,yz\}}$ as a parameter. The second anisotropic term, the interorbital Coulomb anisotropy $u' = U_{xy,\{xz,yz\}} - U_{\{xz,yz\},\{yz,xz\}}$, is set to $u' = u/3$, following Ref. [18]. Hence, the different orbital-dependent local Hubbard interactions read

$$\begin{aligned} U_{xy,xy} &= U + \frac{u}{2}, \\ U_{\{xz,yz\},\{xz,yz\}} &= U - \frac{u}{2}, \\ U_{xy,\{xz,yz\}} &= U - 2J_H + \frac{u'}{2}, \\ U_{\{xz,yz\},\{yz,xz\}} &= U - 2J_H - \frac{u'}{2}. \end{aligned} \quad (1)$$

Note, of course, that spin flip and pair hopping still account for a further isotropic J_H contribution in the local Hamiltonian. In essence, the stronger Coulomb interaction for electrons in the γ sheet relative to α and β raises the energy of states in this sheet, which partially unwinds the above-mentioned correlation-driven transfer of particles from the α and β sheets to the γ sheet.

Figure 4 depicts the strain evolution of the Fermi surface for $u = 0.10$ eV and $u = 0.15$ eV. Both values give very good agreement with the experimental Fermi surface at zero strain [29], as indicated by the blue asterisks. It is seen that the strain-induced Lifshitz transition at Y is shifted to larger strain values for larger u . A linear interpolation

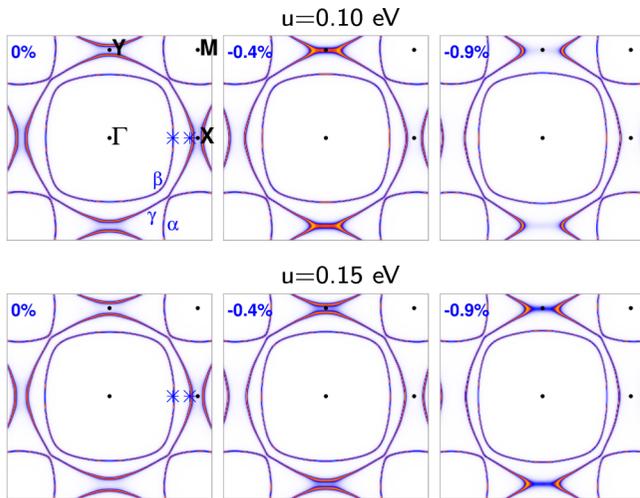


FIG. 4. Evolution of the correlated Fermi surface with α , β , and γ sheets for the local Coulomb anisotropies $u = 0.10$ eV (top) and $u = 0.15$ eV (bottom) from zero strain to $\varepsilon_{xx} = -0.4\%$ and -0.9% . Blue asterisks mark the experimental Fermi crossings along Γ -X for the unstrained compound [29]. The strain-induced Lifshitz transition takes place close to the Y point in the Brillouin zone.

of the calculated $E_{xy} - E_F$ at strains $\varepsilon_{xx} = \{0, -0.4 \times 10^{-2}, -0.9 \times 10^{-2}\}$ yields $\varepsilon_{\text{VHS}} = -0.38 \times 10^{-2}$ for $u = 0.10$ eV and -0.70×10^{-2} for $u = 0.15$ eV. This demonstrates the sensitivity to local Coulomb anisotropy and also shows its potential of driving the theoretical critical strain closer to the experimental value.

In agreement with the strain ARPES data of Ref. [6], the DFT+DMFT calculations here indicate no substantial strain-induced redistribution of carriers between the Fermi surface sheets. For both $u = 0.10$ and 0.15 meV, the β sheet has between 2% and 3% more area at $\varepsilon_{xx} = -0.9 \times 10^{-2}$ than at zero strain. This result should be verified with other impurity solvers; however, it is anyway well below the resolution of the ARPES data in Ref. [6].

We note that a lower isotropic $U = 2.3$ eV and $J_H = 0.4$ eV [25,29], together with the same u, u' as above, also yield a reasonable match to the experimental Fermi surface at zero strain. However, this choice of parameters gives both substantially larger ε_{VHS} than what is observed and smaller averaged band renormalization than the experimental range of $m^*/m \sim 2.5$ – 4.4 [29,30]. Our choice reproduces the lower limit of this range. Note that in the RISB solution to DMFT one typically needs to use slightly larger U values to cope with the correlation strength than those utilized, for instance, in the quantum Monte Carlo impurity solvers. However, in any case, due to the interplay of Hund's metal physics and Van Hove-induced fluctuations acting on different bands, the mass renormalization displays significant band dependence [29].

IV. CONCLUSIONS

First, through our measurements we have refined the experimental determination of ε_{VHS} , the longitudinal strain at which the γ sheet is driven through a Lifshitz transition when Sr_2RuO_4 is uniaxially pressurized along a $\langle 100 \rangle$ lattice direction, to $(-0.44 \pm 0.06) \times 10^{-2}$. This corresponds to

$\varepsilon_{xx} - \varepsilon_{yy} = (-0.66 \pm 0.09) \times 10^{-2}$. Second, although there is some uncertainty in the first-principles calculation, ε_{VHS} is found to be considerably overestimated in DFT calculations, even when SOC is included. We find that Coulomb repulsion reduces the calculated Van Hove strain; however, an isotropic Coulomb interaction strong enough to reproduce the observed band renormalizations gives ε_{VHS} much smaller than observed. Good agreement with experimental data can be achieved only by accounting for orbital anisotropy in the Coulomb interactions on the Ru site. For $U = 3.1$ eV and $J_H = 0.7$ eV, the experimental ε_{VHS} is reproduced for $u \equiv U_{xy,xy} - U_{\{xz,yz\},\{xz,yz\}} \sim 0.10$ – 0.15 eV. This finding highlights the value of high-precision strain measurements in providing additional constraints on many-body theory.

The raw data for this publication may be downloaded at [31].

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APPENDIX

Ab initio density functional calculations were performed using three different codes, two of which are based on the all-electron linearized augmented plane wave method (WIEN2k [15] and Elk [16]), and the third one is a mixed-basis pseudopotential scheme [32,33]. The first two are among the most accurate methods (codes) present on the DFT market. Selected points were also checked against the projector augmented-wave method [34] implemented in the Vienna *Ab initio* Simulation Package (VASP) [35].

The convergence with respect to the size of the basis set as well as to the k -point mesh has been checked (in WIEN2k a mesh up to $50 \times 50 \times 50$ k points and $RK_{\text{max}} = 8.5$ were used, and for Elk $30 \times 30 \times 30$ and $RK_{\text{max}} = 9.0$ were used). Results of the WIEN2k and Elk calculations (together with those of Ref. [2]) set up an error bar for the critical strain as obtained in DFT. It was also used as a hallmark for the pseudopotential code, which was utilized in the DFT+DMFT calculations. The generalized gradient approximation in the Perdew-Burke-Ernzerhof version [14] was used in these calculations for the exchange-correlation potential. The DMFT correlated subspace is defined by the maximally localized Wannier functions [36] for the Ru($4d$) t_{2g} orbitals. A three-orbital Hubbard Hamiltonian of Slater-

TABLE I. Comparison between the first 50 measured and calculated resonant ultrasound frequencies at 4 K.

n	f_{meas} (MHz)	f_{calc} (MHz)	Error %	n	f_{meas} (MHz)	f_{calc} (MHz)	Error %
1	0.888632	0.886403	0.30	26	2.129007	2.124050	0.23
2	1.084435	1.084360	0.01	27	2.151253	2.149868	0.06
3	1.206902	1.212514	0.47	28	2.170007	2.159553	0.48
4	1.291172	1.290493	0.05	29	2.244981	2.253069	0.36
5	1.338181	1.342099	0.29	30	2.363937	2.364255	0.01
6	1.399089	1.403673	0.33	31	2.423232	2.431402	0.34
7	1.446793	1.449544	0.19	32	2.452031	2.442314	0.40
8	1.476043	1.474905	0.08	33	2.484408	2.485476	0.04
9	1.506698	1.510774	0.27	34	2.495670	2.515751	0.80
10	1.659870	1.667086	0.43	35	2.542062	2.543633	0.06
11	1.680165	1.675043	0.30	36	2.549073	2.553515	0.17
12	1.694431	1.700695	0.37	37	2.551501	2.556063	0.18
13	1.721309	1.714025	0.42	38	2.560607	2.558860	0.07
14	1.736905	1.737086	0.01	39	2.572891	2.576978	0.16
15	1.741506	1.745891	0.25	40	2.643618	2.640979	0.10
16	1.793959	1.787555	0.36	41	2.647660	2.644939	0.10
17	1.851191	1.847143	0.22	42	2.658930	2.655164	0.14
18	1.870988	1.877737	0.36	43	2.742533	2.739407	0.11
19	1.915303	1.908402	0.36	44	2.746571	2.743120	0.13
20	1.918634	1.921316	0.14	45	2.765141	2.757169	0.29
21	1.933059	1.928647	0.23	46	2.770066	2.767905	0.08
22	2.043503	2.048846	0.26	47	2.789253	2.770308	0.68
23	2.063880	2.062220	0.08	48	2.816993	2.810757	0.22
24	2.073442	2.067358	0.29	49	2.851807	2.867318	0.54
25	2.076389	2.077160	0.04	50	2.891943	2.902399	0.36

Kanamori form, parametrized by the Hubbard U and the Hund's exchange J_H , is applied in that subspace.

The spin-orbit coupling described by $H_{\text{soc}} = \lambda/2 \sum_{mm'} \sum_{\sigma\sigma'} c_{m\sigma}^\dagger \mathbf{L} \cdot \boldsymbol{\sigma} c_{m'\sigma'}$ was taken into account for correlated orbitals space. Here, m, m' are orbital indices, σ, σ' mark spin projections, \mathbf{L} are t_{2g} angular momentum matrices, and $\boldsymbol{\sigma}$ denotes the Pauli matrices.

The impurity problem was solved by the rotationally invariant slave-boson (RISB) [20,21] framework at the saddle

point. The RISB electronic self-energy consists of a term linear in frequency as well as a static part and thus has a simpler and more restricted form as a possible general local $\Sigma(\omega)$. However, the RISB scheme (here, at formal $T = 0$) is still well suited for many correlated materials problems such as ruthenates [23,26].

For all calculations, crystal structures as optimized in Ref. [2] as a function of strain were used as the structural input.

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