

Fermi-Level-Dependent Charge-to-Spin Conversion of the Two-Dimensional Electron Gas at the γ -Al₂O₃/KTaO₃ Interface

Hui Zhang^{1,*}, Zengtai Zhu,² Yungu Zhu,¹ Xiaobing Chen,^{3,4,5} Qisheng Jiang,² Jinwu Wei,⁶ Chenbo Zhao,^{7,8} Jine Zhang,¹ Furong Han,¹ Huaiwen Yang,¹ Dapeng Zhu,¹ Hao Wu,² Yuansha Chen^{3,4}, Fengxia Hu,^{2,3,4} Baogen Shen,^{2,3,4} Weisheng Zhao,¹ Jing Zhang,^{2,†} Guoqiang Yu,^{2,3,9,‡} and Jirong Sun^{2,3,4,10,§}

¹*School of Integrated Circuit Science and Engineering, Beihang University, Beijing 100191, China*

²*Songshan Lake Materials Laboratory, Dongguan, Guangdong 523808, China*

³*Beijing National Laboratory for Condensed Matter Physics, Institute of Physics, Chinese Academy of Sciences, Beijing 100190, China*

⁴*School of Physical Sciences, University of Chinese Academy of Sciences, Beijing 100049, China*

⁵*Shenzhen Institute for Quantum Science and Engineering (SIQSE) and Department of Physics, Southern University of Science and Technology, Shenzhen 518055, China*


⁶*Key Laboratory for Magnetism and Magnetic Materials of the Ministry of Education, Lanzhou University, Lanzhou 730000, China*

⁷*School of Science and Engineering, The Chinese University of Hong Kong, Shenzhen, Guangdong 518172, China*

⁸*Department of Chemical Physics, University of Science and Technology of China, Hefei, Anhui 230026, China*

⁹*Center of Materials Science and Optoelectronics Engineering, University of Chinese Academy of Sciences, Beijing 100049, China*

¹⁰*Spintronics Institute, Jinan University, Jinan 250022, Shandong Province, China*

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Charge-to-spin conversion is crucial for the application of emerging spintronic devices. A two-dimensional electron gas (2DEG) at a complex oxide interface usually possesses strong Rashba spin-orbit coupling, and spin-momentum locking offers a great possibility for efficient charge-to-spin conversion through the Rashba-Edelstein effect. Here, we report the fabrication of metallic 2DEGs in γ -Al₂O₃/KTaO₃ spinel/perovskite heterostructures and investigate the charge-to-spin conversion for Py/ γ -Al₂O₃/KTaO₃ devices using the technique of spin-torque ferromagnetic resonance. The sizable spin splitting of the band structure results in a large current-induced spin-orbit torque efficiency with values up to around 3.6 at 5 K and about 1.1 at 300 K, which are more than an order of magnitude higher than those of heavy metals (0.07 for Pt at 300 K). Moreover, both theoretical and experimental results show that the charge-to-spin conversion is strongly dependent on the position of the Fermi level. These results demonstrate that optimizing the band filling of a KTaO₃-based 2DEG is crucial for maximizing the conversion efficiency.

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I. INTRODUCTION

Owing to Rashba spin-orbit coupling (SOC), the two-dimensional electron gas (2DEG) at the LaAlO₃/SrTiO₃ (LAO/STO) interface has the characteristic of spin-momentum locking, i.e., both electron momentum and electron spin are constrained to the conductive interface and are locked orthogonally [1–3]. In this case, an electric current flowing along the interface will generate a perpendicular spin current, resulting in a charge-to-spin

conversion [4]. When this spin current is injected into a ferromagnetic layer, the effect of spin-orbit torque (SOT) will be produced, resulting in an electric switching of magnetization [5–7].

Based on SOT, the technique of spin-torque ferromagnetic resonance (ST-FMR) has been developed [8–11], making it possible to get a quantitative description of the efficiency of charge-to-spin conversion. Via ST-FMR, for example, a conversion efficiency of around 6 has been obtained for the LAO/STO 2DEG at room temperature [9]. This efficiency is more than an order of magnitude higher than those of heavy metals such as Pt, W, and Pd [5, 12, 13], etc. High conversion efficiency (about 4) is also found for LaTiO_{3+ δ} /STO 2DEG [10]. It seems to be a general property of the STO-based 2DEGs.

*huizh@buaa.edu.cn

†zhangjing@sslabs.org.cn

‡guoqiangyu@iphy.ac.cn

§jrsun@iphy.ac.cn

As is well established, the efficiency of charge-to-spin conversion is governed by the Rashba SOC that emerges with the formation of the 2DEG in the interfacial layer of STO [3,9,14–16]. Notably, 2DEGs can also be formed at the interfacial layer of KTaO₃ (KTO), exhibiting similar transport behaviors to those of STO-based 2DEGs [17–20]. KTO is an oxide similar to STO in many aspects. However, KTO has a strong atomic SOC that is approximately 20 times as large as that of STO [21,22]. It is therefore reasonable to expect a strong Rashba SOC, thus an efficient charge-to-spin interconversion via KTO-based 2DEGs [23]. There are limited works on spin-to-charge conversion, the inverse process of charge-to-spin conversion, via the 2DEG at the interfacial layer of KTO. By thermally injecting spin current from the ferromagnetic EuO overlayer into the 2DEG at the EuO/KTO interface, Zhang *et al.* observed an electric output [24], i.e., spin current had been converted to charge current. Using the technique of spin pumping to inject spin current into Al/KTO 2DEG, Vicente-Arche *et al.* realized spin-to-charge conversion with an efficiency of $\lambda_{\text{IEE}} \approx -3.5$ nm at 10 K [25]. Despite these works, many aspects of the KTO-based 2DEGs remain unexplored, for example, the process of charge-to-spin conversion. In this work, we perform a systematic investigation of charge-to-spin conversion via the 2DEG at the γ -Al₂O₃/KTO interface, based on the technique of ST-FMR at room temperature and low temperature. A conversion efficiency as high as approximately 3.6 at 5 K and 1.1 at 300 K are detected for the sample with a γ -Al₂O₃ layer of 3 unit cell (u.c.), which are more than an order of magnitude higher than that of Pt (about 0.07 at 300 K). Moreover, a strong dependence of the conversion efficiency on band filling is observed, and the most efficient conversion takes place at the filling level corresponding to Lifshitz transition. Theoretical analysis indicates that enhanced spin splitting of the electronic band structure is responsible for efficient conversion.

II. EXPERIMENT

A. 2DEG preparation and characterization

High-quality 2DEGs are obtained by growing γ -Al₂O₃ films on the (001)-oriented KTO single crystal substrate through pulsed laser deposition. Prior to film growth, we clean the substrate surface with alcohol and acetone successively, and there is no further surface treatment. The growth process is monitored by *in situ* reflection high-energy electron diffraction (RHEED). The target is a single crystal of α -Al₂O₃. The fluence of the laser pulse is 2 J/cm² and the repetition rate is 2 Hz (KrF excimer laser, wavelength 248 nm). During the deposition process, the oxygen pressure is maintained at 1×10^{-4} Pa and the temperature of the substrate is kept at 700 °C. After deposition, the samples are cooled down to room temperature at a rate of 10 °C/min without changing the oxygen atmosphere.

The film thickness is determined by RHEED oscillations. The fabricated samples have the γ -Al₂O₃ layers of 3, 4, 6, and 8 u.c., respectively. The surface morphology is investigated by atomic force microscopy (AFM, SPI 3800N, Seiko). The crystal structure of the films is determined by a Bruker x-ray diffractometer equipped with thin-film accessories (D8 Discover, Cu $K\alpha$ radiation). The x-ray photoelectron spectroscopy (XPS) measurements are performed in a Thermo Scientific ESCALAB 250X instrument using a monochromatic Al $K\alpha$ x-ray source. The electronic transport behavior of the γ -Al₂O₃/KTO heterostructures is measured by a Quantum-designed physical property measurement system employing van der Pauw geometry.

B. ST-FMR device fabrication and measurement

The γ -Al₂O₃/KTO heterostructures are transferred into a chamber for magnetron sputtering, and then the chamber is evacuated up to the background pressure of 1×10^{-6} Pa. Permalloy (Py) films (6 nm) are deposited using magnetron sputtering at room temperature with an Ar pressure of 0.08 Pa. All samples are covered by Ta(3 nm)/MgO(2 nm) capping layers, where the MgO layer is used to protect the Py film from oxidation and the additional Ta layer is added to prevent moisture absorption by the MgO in air, which would lead to a deliquescence of the MgO layer. The Ta/MgO/Py/ γ -Al₂O₃/KTO stacks are then patterned into rectangular strips of 20 μ m in width and 100 μ m in length by the standard photolithography and Ar-ion milling technique. The Au (80 nm)/Cr (5 nm) bilayers are deposited by magnetron sputtering as electrical contact pads. In the ST-FMR measurements, the radio-frequency current (I_{rf}) with a nominal maximum power of 15 dBm is applied along the longitudinal direction of the rectangular bars (x axis). The in-plane external magnetic field (H) with an angle of φ with respect to I_{rf} is swept during measurement. The ST-FMR measurements are performed at room temperature and a low temperature of 5 K, respectively.

III. RESULTS AND DISCUSSION

A. Structural characteristics and transport behavior of γ -Al₂O₃/KTO heterostructures

Al₂O₃ ultrathin film is usually γ -Al₂O₃ with a spinel-type structure, rather than the common α -Al₂O₃ with a corundum structure; the surface energy differences thermodynamically stabilize γ -Al₂O₃ over α -Al₂O₃ [26,27]. A schematic depiction of heteroepitaxial growth of the γ -Al₂O₃ layer on the (001)-oriented KTO substrate is shown in Fig. 1(a). Despite differences in the cation sublattices, the oxygen sublattice of the spinel γ -Al₂O₃ well matches that of the perovskite KTO. The lattice parameter of γ -Al₂O₃ is 7.911 Å, which is nearly twice that of KTO ($a_{\text{KTO}} = 3.989$ Å). The lattice mismatch is as small as

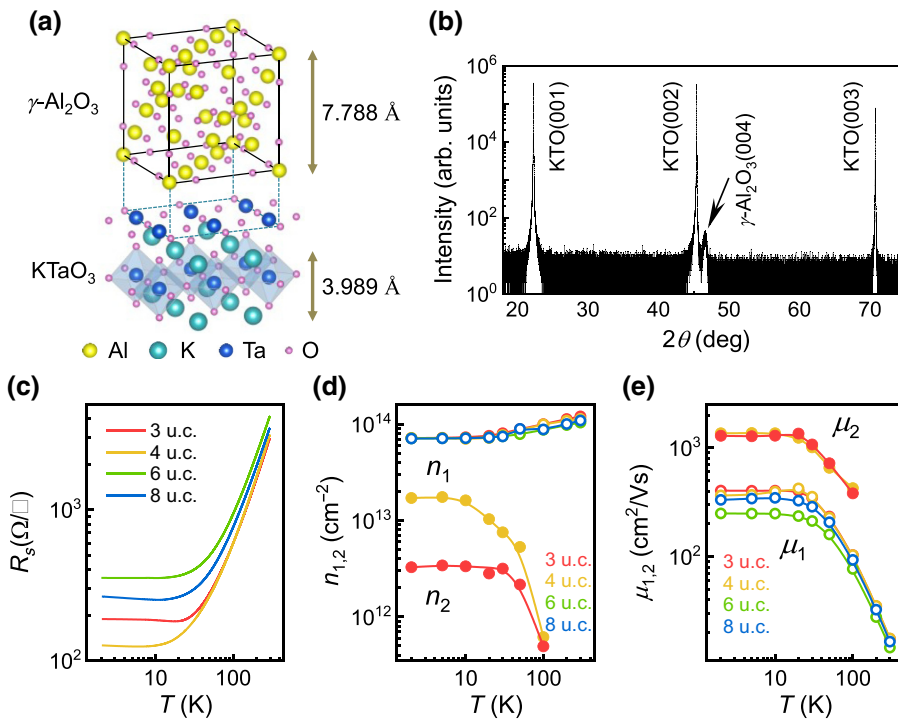


FIG. 1. (a) Sketch of the γ -Al₂O₃/KTO (001) heterostructure. (b) X-ray diffraction spectrum specifying the well-orientated γ -Al₂O₃ film on KTO substrate. (c) Temperature dependence of the sheet resistance for samples with the γ -Al₂O₃ thicknesses of 3, 4, 6, and 8 u.c. (d),(e) Corresponding carrier density and Hall mobility, respectively, as functions of temperature. Solid lines are guides to the eye.

0.84% [27]. Such an excellent lattice match between oxygen sublattices results in an epitaxial layer-by-layer growth of the γ -Al₂O₃/KTO spinel/perovskite heterostructures, as confirmed by the regular oscillations of the RHEED pattern (see Fig. S1 in the Supplemental Material [28]). As shown in Fig. 1(b), the x-ray diffraction (XRD) spectra further reveal the formation of a high-quality single crystalline γ -Al₂O₃ film on the KTO substrate, and the out-of-plane lattice parameter is $c = 7.788 \text{ \AA}$, close to the bulk value 7.911 \AA . Here a γ -Al₂O₃ film with a thickness of 16 nm is adopted for XRD measurements to get a clear signal. The obtained film is very flat, and the root-mean-squared roughness is around 0.21 nm (see Fig. S2 in the Supplemental Material [28]).

Al₂O₃ is widely used as the best insulating material with a band gap normally above 8.0 eV. KTO is also a wide-band-gap insulator with a band gap of 3.6 eV. Fascinatingly, the interface between the two insulators is metallic with electrons as the dominant charge carriers, indicating the formation of 2DEGs at the γ -Al₂O₃/KTO interfaces. Figure 1(c) presents the temperature dependence of the sheet resistance (R_s) for samples with different film thicknesses $t_{\gamma\text{-Al}_2\text{O}_3} = 3, 4, 6,$ and 8 u.c. To evaluate the origin of the interfacial conductivity, XPS measurements are performed (see Fig. S3 in the Supplemental Material [28]). The Ta $4f$ core-level spectrum collected from the bare KTO substrate shows only the Ta⁵⁺ valence state, consistent with its insulating character. However, after depositing a γ -Al₂O₃ layer with a thickness of 3 u.c., peaks associated with Ta⁴⁺ and Ta²⁺ valence states arise, indicating the generation of an electron-rich layer at the γ -Al₂O₃/KTO

heterointerface. This can be ascribed to the formation of oxygen vacancies in KTO during the deposition process of γ -Al₂O₃ rather than polar-discontinuity-induced electronic reconstruction, since each layer of γ -Al₂O₃ is nominally charge neutral [27].

Interfacial electrons are exclusively accumulated on the $5d$ electronic shell of the Ta ions, residing in the t_{2g} conduction bands of KTO. As is well established, quantum confinement lifts the orbital degeneracy of the bulk band structure and leads to the hierarchic band structure of 2DEGs [19]. The d_{xy} band is lower in energy compared with the d_{xz} and/or d_{yz} bands. Therefore, at low carrier densities, all electrons are expected to populate the d_{xy} band, whereas at higher densities, after the Lifshitz transition, the Fermi energy enters the d_{xz} and/or d_{yz} bands. The Hall effect of the 2DEGs at different temperatures is further investigated (see Note 1 and Fig. S4 in the Supplemental Material [28]). Linear R_{xy} - H curves are obtained for $t_{\gamma\text{-Al}_2\text{O}_3} = 6$ and 8 u.c., which means that only one species of carrier exists in the system, residing in the d_{xy} band. The nonlinear Hall effect, which can be satisfactorily described by a two-band model [29], is observed for samples $t_{\gamma\text{-Al}_2\text{O}_3} = 3$ and 4 u.c. This is an indication of the emergence of a second species of charge carriers occupying the d_{xz} and/or d_{yz} orbital states. In Figs. 1(d) and 1(e) we present the deduced carrier density and corresponding Hall mobility, respectively, as functions of temperature. The carrier density of the d_{xy} electrons is $n_1 \sim 7 \times 10^{13} \text{ cm}^{-2}$ at 2 K, almost identical for different samples. The corresponding Hall mobility varies from $\mu_1 \sim 248$ to $404 \text{ cm}^2/\text{Vs}$. As the thickness of the γ -Al₂O₃

layer increases from 3 to 4 u.c., the carrier density of the d_{xz} and/or d_{yz} electrons changes from $n_2 \sim 3.3 \times 10^{12}$ to $1.7 \times 10^{13} \text{ cm}^{-2}$ at 2 K, an increase of nearly 1 order of magnitude, but is still far less than the density of d_{xy} electrons. Such a small carrier density ($n_2 \sim 3.3 \times 10^{12} \text{ cm}^{-2}$) indicates that when $t_{\gamma\text{-Al}_2\text{O}_3} = 3$ u.c., the system undergoes a Lifshitz transition where the d_{xz} and/or d_{yz} bands start to become occupied. As for μ_2 , it takes values of 1288 to 1360 cm^2/Vs at 2 K when $t_{\gamma\text{-Al}_2\text{O}_3}$ increases from 3 to 4 u.c., which is larger than μ_1 by a factor of 3–5. This is because the d_{xz} and/or d_{yz} orbitals extend deeper in KTO, resulting in reduced scattering from interfacial defects and thus high mobility. These transport characteristics are consistent with those reported for other KTO-based heterostructures [19,20].

B. SOT efficiency of Py/ $\gamma\text{-Al}_2\text{O}_3$ /KTO devices

The ST-FMR technique has usually been employed to evaluate the spin current induced by spin Hall effect for heavy metals [8]. Here we apply this technique to quantitatively describe the efficiency of charge-to-spin conversion at the $\gamma\text{-Al}_2\text{O}_3$ /KTO heterointerfaces. For this purpose, Py/ $\gamma\text{-Al}_2\text{O}_3$ /KTO devices are fabricated (see Sec. II for detailed process). The optical image of the ST-FMR device and the schematic diagram of the ST-FMR experimental

setup are shown in Fig. 2(a). The left panel of Fig. 2(a) illustrates the current-induced SOT and the SOT-driven magnetization dynamics in the Py/ $\gamma\text{-Al}_2\text{O}_3$ /KTO device. A radio frequency charge current (I_{rf}) is applied along the x axis through a ground-source-ground electrode. An in-plane magnetic field (H) is applied at an angle φ with respect to the x axis. Owing to the Rashba-Edelstein effect of the 2DEG at the $\gamma\text{-Al}_2\text{O}_3$ /KTO interface, which is induced by the spin-momentum locking determined by the spin splitting of the band structure, nonequilibrium spins accumulate and subsequently generate a spin current in the out-of-plane direction (along the z axis). The spin current is absorbed by the Py layer, producing a dampinglike torque (τ_{DL}), a fieldlike torque (τ_{FL}), or both on the magnetization of Py. In addition, I_{rf} also generates an Oersted field torque (τ_{Oe}). All these current-induced torques cause a precession of the Py magnetization with the same frequency as I_{rf} . Consequently, the oscillatory anisotropic magnetoresistance (AMR) together with I_{rf} result in a rectified voltage (V_{mix}), which is detected by a lock-in amplifier and can be described by [8,30]

$$V_{\text{mix}} = V_S \frac{\Delta H^2}{\Delta H^2 + (H - H_0)^2} + V_A \frac{\Delta H(H - H_0)}{\Delta H^2 + (H - H_0)^2} + V_0, \quad (1)$$

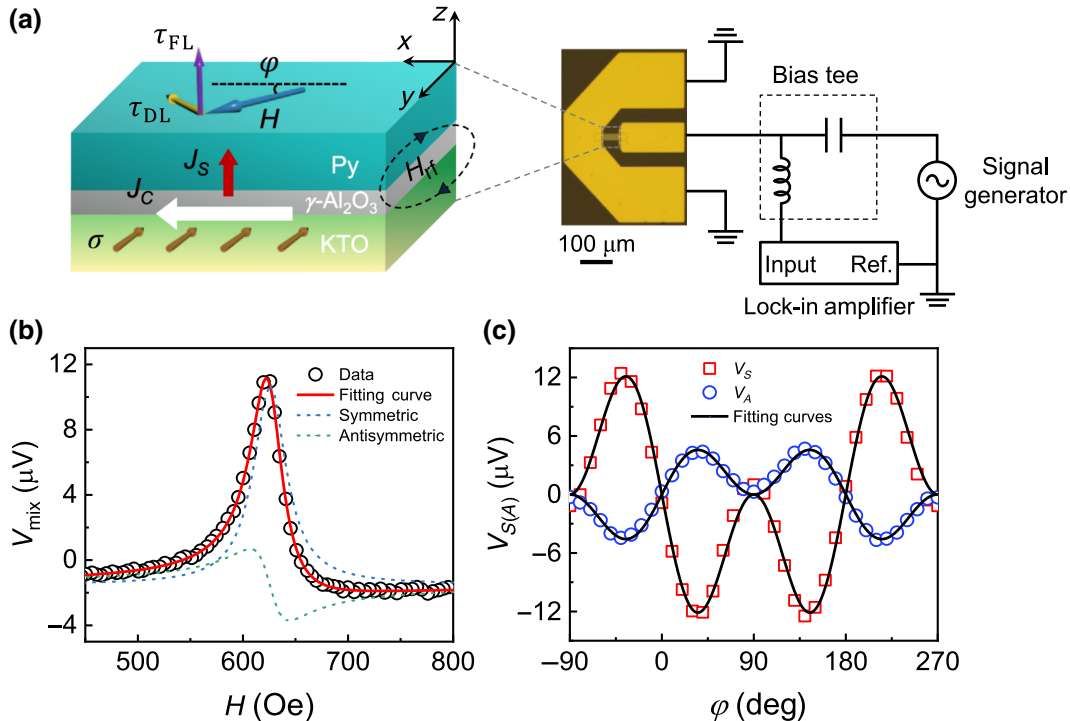


FIG. 2. (a) Schematic diagram of the structure of ST-FMR device with the SOT-induced magnetization dynamics and the ST-FMR experimental setup, where J_C is the uniform charge current density flowing through the $\gamma\text{-Al}_2\text{O}_3$ /KTO 2DEG layer, J_S represents the spin current density injected into the Py layer, and σ denotes the unit vector of spin polarization. (b) Typical ST-FMR signal of Py(6 nm)/ $\gamma\text{-Al}_2\text{O}_3$ (4 u.c.)/KTO device, measured under $f = 7$ GHz and $\varphi = 45^\circ$ at room temperature. Symbols are experimental data and solid line is the result of curve fitting, which can be separated into symmetric (blue dashed) and antisymmetric (green dashed) components. (c) Angular dependence of V_S and V_A and the corresponding fitting curves by using the relationship of $\sin 2\varphi \cos \varphi$.

where ΔH is the linewidth of the resonance peak, H_0 is the resonance magnetic field, and V_0 is the offset value of the measured voltage. The first term represents a symmetric Lorentzian component corresponding to the contribution of the in-plane torque ($\tau_{\parallel} = \tau_{DL}$), and the second term describes an antisymmetric component corresponding to the out-of-plane torque ($\tau_{\perp} = \tau_{FL} + \tau_{Oe}$). V_S and V_A are the symmetric and antisymmetric Lorentzian coefficients, respectively.

Figure 2(b) shows a representative ST-FMR spectrum of the Py(6 nm)/ γ -Al₂O₃(4 u.c.)/KTO device measured by sweeping H at a fixed frequency $f = 7$ GHz at room temperature, where H is applied at a fixed angle $\varphi = 45^\circ$. The ST-FMR spectrum can be quantitatively analyzed by Eq. (1). The symmetric and antisymmetric line shapes can be extracted and are shown in Fig. 2(b). Obviously, a larger symmetric signal compared with the antisymmetric component is observed. By fitting the experimental results, the parameters ΔH , H_0 , V_S , and V_A can be deduced. For comparison, we also measure the ST-FMR signal from the control device with a structure of Py(6 nm)/KTO under the exact same experimental conditions and obtain a negligible signal (see Fig. S5 in the Supplemental Material [28]). This result verifies that the main contribution of the current-induced SOT torque to the Py magnetization dynamics comes from the γ -Al₂O₃/KTO interface.

To get a complete result for SOT, the full angle-dependent ST-FMR measurements are carried out as the magnetic field rotates the magnetization direction of the Py layer in the sample plane at room temperature. ST-FMR spectra corresponding to different φ values are fitted to Eq. (1) to get the angular dependence of V_S and V_A . Notably, the amplitude corresponding to the AMR of Py has a $\sin 2\varphi$ dependence, while the amplitude of the current-induced torque has a $\cos\varphi$ dependence [8,30]. The product of these two contributions then yields the same angular dependence for the symmetric and antisymmetric ST-FMR components: $V_S = S \sin 2\varphi \cos\varphi$ and $V_A = A \sin 2\varphi \cos\varphi$, where S and A are the averaged amplitude of V_S and V_A , respectively. Figure 2(c) shows that both V_S and V_A are well described by this angular dependence, and S and A can be determined by curve fitting. Please see Fig. S6 in the Supplemental Material [28] for the angular dependence of V_S and V_A for $t_{\gamma\text{-Al}_2\text{O}_3} = 3, 6, \text{ and } 8$ u.c. Notably, the angular dependence of V_S and V_A are reversed, i.e., the peaks (valleys) of the V_S - φ curve correspond to the valleys (peaks) of the V_A - φ curve. This result indicates that the sign of S is positive while the sign of A is negative. However, both S and A of Pt are positive according to the angular dependence of V_S and V_A [31], suggesting that the Rashba effect at the γ -Al₂O₃/KTO interface is opposite in sign to that for the heavy metal Pt.

From the ratio of the symmetric and antisymmetric components, the SOT efficiency ξ_{SOT} of the Py/ γ -Al₂O₃/STO device can be expressed as [32],

$$|\xi_{\text{SOT}}| = \left| \frac{S}{A} \right| \frac{e\mu_0 M_S t d}{\hbar} [1 + (4\pi M_{\text{eff}}/H_0)]^{1/2}, \quad (2)$$

where e is the electron charge, \hbar is the reduced Planck constant, μ_0 is the vacuum permeability, M_S is the saturation magnetization of the ferromagnetic Py layer, t is the thickness of the Py layer, d is the thickness of the 2DEG conducting layer (about 10 nm) [9,33], and $4\pi M_{\text{eff}}$ is the effective magnetization of the Py layer, which can be obtained from fitting the frequency-dependent resonance field to the Kittel formula [inset in Fig. 3(a)] [32],

$$f = (\gamma/2\pi)[H_0(H_0 + 4\pi M_{\text{eff}})]^{1/2}, \quad (3)$$

where γ is the gyromagnetic ratio ($\gamma = 1.76 \times 10^{11}$ rad s⁻¹ T⁻¹ for the Py film). The ST-FMR spectra of Py(6 nm)/ γ -Al₂O₃($t_{\gamma\text{-Al}_2\text{O}_3}$)/KTO devices with $t_{\gamma\text{-Al}_2\text{O}_3} = 3, 4, 6, \text{ and } 8$ u.c. at different resonance frequencies are recorded at room temperature, and the corresponding f - H_0 curves are fitted (see Figs. S7 and S8 in the Supplemental Material [28]). Based on the results of curve fitting, $4\pi M_{\text{eff}}$ as a function of $t_{\gamma\text{-Al}_2\text{O}_3}$ is shown in Fig. 3(a). The maximal value of $4\pi M_{\text{eff}}$ is around 9.0 kOe, appearing at $t_{\gamma\text{-Al}_2\text{O}_3} = 4$ u.c., and slightly decreases with the increase of $t_{\gamma\text{-Al}_2\text{O}_3}$. As the $4\pi M_{\text{eff}}$ presents the contribution of interfacial anisotropy, these results indicate that the anisotropy decreases slightly with the increase of the layer thickness of γ -Al₂O₃.

To get the information on magnetic damping, the linewidth ΔH is presented as a function of resonance frequency f when $t_{\gamma\text{-Al}_2\text{O}_3} = 4$ u.c., as shown in the inset of Fig. 3(b). The Gilbert damping coefficient α of the Py layer is estimated from the slope of the linear fitting curve, given by $\Delta H = \Delta H_0 + (2\pi\alpha/\gamma)f$, where ΔH_0 is the inhomogeneous line broadening factor that is usually independent of frequency [32]. Please see Fig. S9 in the Supplemental Material [28] for the fitting processes of α for different samples. As shown in Fig. 3(b), with the increase of the γ -Al₂O₃ layer thickness, α first decreases slightly and then increases significantly, ranging from 0.0057 to 0.0132. This result matches well the result of the Py film of spin valves [10,11]. Notably, the Gilbert damping coefficient depends on disorder and electron-electron interaction [34]. The significant increase of α in the case of $t_{\gamma\text{-Al}_2\text{O}_3} = 8$ u.c. may be a consequence of increased interface roughness of the 8-u.c.- γ -Al₂O₃ film, which would result in stronger disorder and electron-electron scattering.

Based on the results mentioned previously, the interface SOT efficiency $|\xi_{\text{SOT}}|$, i.e., the efficiency for charge-to-spin conversion, can be calculated using Eq. (2) and adopting the deduced parameters. The variation of the $|\xi_{\text{SOT}}|$ with the γ -Al₂O₃ layer thickness at room temperature is presented in Fig. 3(c). $|\xi_{\text{SOT}}|$ ranges from 0.7 to 1.1 as $t_{\gamma\text{-Al}_2\text{O}_3}$ varies, and the highest efficiency appears when $t_{\gamma\text{-Al}_2\text{O}_3} = 3$ u.c. Notably, the SOT efficiency detected here

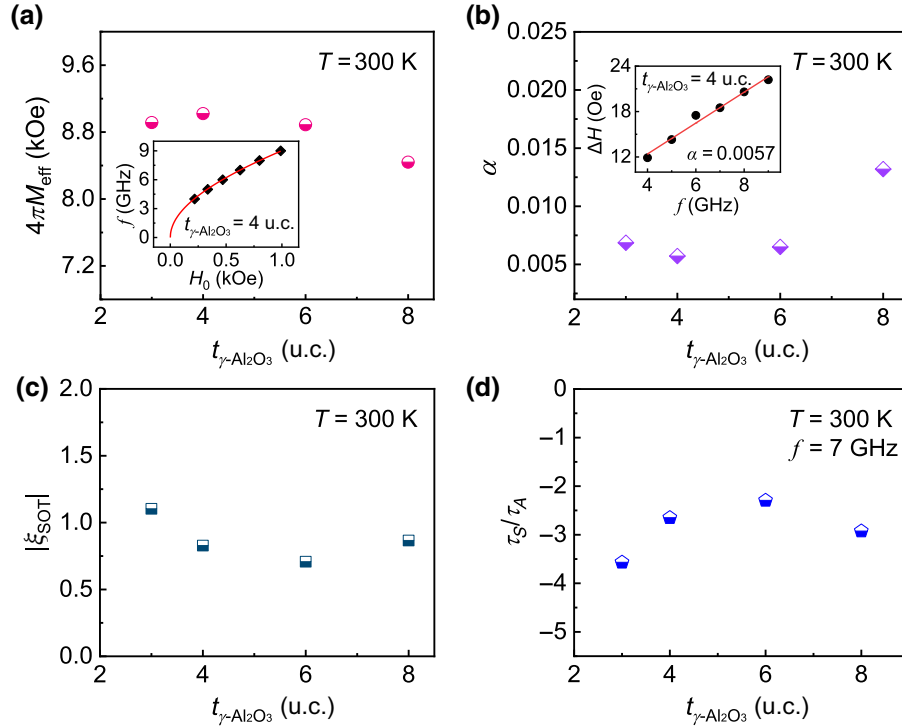


FIG. 3. (a) Effective magnetization as a function of γ -Al₂O₃ layer thickness. The inset is the relationship between the resonant frequency and the resonant field for $t_{\gamma\text{-Al}_2\text{O}_3} = 4$ u.c., and the red line is the result of fitting to the Kittel formula. (b) Dependence of Gilbert damping constant on γ -Al₂O₃ layer thickness. The inset is the linewidth of the resonance peak versus the resonant frequency for $t_{\gamma\text{-Al}_2\text{O}_3} = 4$ u.c., and the red line is the linear fitting. (c) Efficiency of charge-to-spin conversion for different $t_{\gamma\text{-Al}_2\text{O}_3}$ at room temperature. (d) Dependence of the torque ratio τ_S/τ_A on γ -Al₂O₃ thickness for devices at a given frequency ($f = 7$ GHz).

is much larger than that of heavy metals (0.07 in Pt [5] and 0.3 in W [12] at 300 K). Of relevance for magnetization switching based on SOT are the relative signs and strength of the ratio between in-plane and out-of-plane torque [35,36]. The angular dependences for the in-plane torque (τ_{\parallel}) and the out-of-plane torque (τ_{\perp}) have the following forms: $\tau_{\parallel}(\varphi) = \tau_S \cos\varphi$ and $\tau_{\perp} = \tau_A \cos\varphi$, where τ_S and τ_A are independent of φ , and the ratio of in-plane torque τ_S to the out-of-plane torque τ_A is equal to S/A [31]. Figure 3(d) shows the dependence of the torque ratio τ_S/τ_A on γ -Al₂O₃ thickness for devices at a given frequency ($f = 7$ GHz). As $t_{\gamma\text{-Al}_2\text{O}_3}$ varies, the ratio τ_S/τ_A ranges from -2.3 to -3.6 . Notably, the ratio between in-plane and out-of-plane torque for the Py/ γ -Al₂O₃/KTO sample has an opposite sign to that of the Py/Pt sample but is larger in absolute value than that of Py/Pt (τ_S/τ_A is 1.79 in magnitude for Py/Pt) [31].

C. Fermi-level-dependent charge-to-spin conversion at low temperature

Fermi-level-dependent charge-to-spin conversion has been investigated in different materials, such as topological insulator and monolayer transition-metal dichalcogenides [37–39]. In order to reveal the relationships between SOT

efficiency and the position of the Fermi level we further perform the ST-FMR measurement at low temperature. The ST-FMR spectra of devices with different γ -Al₂O₃ layer thicknesses are recorded under a fixed frequency and $\varphi = 45^\circ$ at 5 K (see Fig. S10 in the Supplemental Material [28]). Figure 4(a) presents the relation between SOT efficiency and the Fermi energy (E_F), obtained at 5 K for all devices.

The Fermi energy is determined by $E_F = n_S \hbar^2 / 4\pi m_e$ with \hbar being the Planck constant and m_e being effective electron mass, and the values of n_S at 5 K in Fig. 1(d) are adopted for the calculation of E_F . Considering the difference of the effective mass for d_{xy} and d_{xz} and/or d_{yz} bands, the equation $E_F = n_1 \hbar^2 / 4\pi m_{e1} + n_2 \hbar^2 / 4\pi m_{e2}$ is adopted when two species of carriers coexist, where $m_{e1} = 0.31m_0$ for the d_{xy} band [40] and $m_{e2} = 0.996m_0$ for d_{xz} and/or d_{yz} bands with m_0 being free electron mass (see Note 2 in the Supplemental Material for details [28]). When there is only one species of carrier, the equation $E_F = n_1 \hbar^2 / 4\pi m_{e1}$ is adopted. E_F ranges from about 0.549 to 0.596 eV for 5d 2DEGs at the γ -Al₂O₃/KTO heterointerfaces. $|\xi_{\text{SOT}}|$ displays a strong dependence on the state of band filling. At low Fermi energies, only one species of carrier exists, and $|\xi_{\text{SOT}}| = 1.4$ and 1.8 for $E_F = 0.549$ and 0.552 eV, corresponding to $t_{\gamma\text{-Al}_2\text{O}_3} = 8$ and 6 u.c., respectively. When E_F

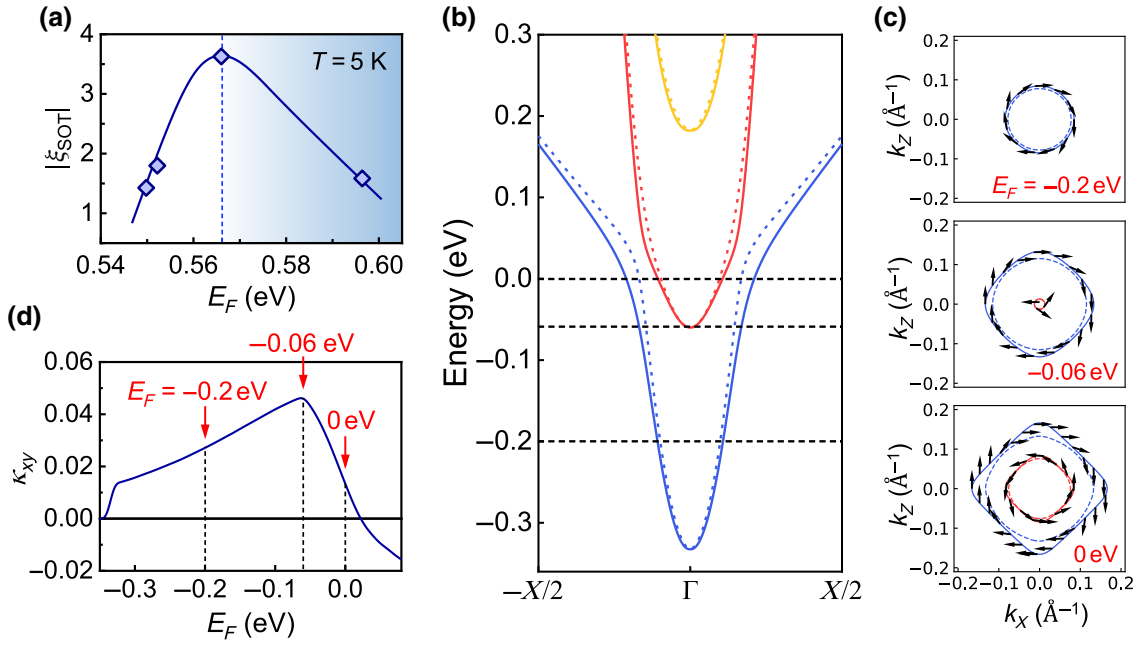


FIG. 4. (a) Charge-to-spin conversion efficiency measured at 5 K as a function of the Fermi energy, where the n_S values at 5 K in Fig. 1(d) are adopted for the calculation of E_F and the dashed line indicates the Lifshitz transition. Solid line is a guide to the eye. (b) Energy dispersion relation of the 2DEG at (001)-oriented KTO interface, calculated by a six-band tight-binding model. Dashed lines mark the positions of typical Fermi levels. (c) Fermi surfaces and spin textures at various energies corresponding to the band filling states marked in (b), arranged in energetically ascending order from top to bottom. For clarity, only clockwise (anticlockwise) spin helicity is shown for the outside (inside) Fermi contours. (d) Dependence of the theoretically calculated Edelstein tensor κ_{xy} on the Fermi energy.

increases to the threshold value of around 0.566 eV, the 2DEG undergoes a crossover from one to two species of charge carriers, i.e., Lifshitz transition, as marked by the dashed line in Fig. 4(a) corresponding to $t_{\gamma-\text{Al}_2\text{O}_3} = 3$ u.c. $|\xi_{\text{SOT}}|$ increases dramatically and the highest value of $|\xi_{\text{SOT}}| = 3.6$ is obtained at the Lifshitz point. A further increase in E_F leads to a rapid decay of $|\xi_{\text{SOT}}|$ to 1.6 corresponding to $t_{\gamma-\text{Al}_2\text{O}_3} = 4$ u.c.

As mentioned previously, the spatial confinement of the 2DEG leads to hierarchic subbands and Rashba spin splitting that is enhanced by orbital hybridization [41–43]. To gain more insight into the relationship between the charge-to-spin conversion and the electronic structure of the 2DEG, we model the band structure of the (001)-oriented KTO interface using a six-band tight-binding Hamiltonian for multiband systems (see Note 3 in the Supplemental Material for details [28]). Without spin-orbit coupling, the t_{2g} bands with sixfold degeneracy at the Γ point are split into double and fourfold degeneracy owing to quantum confinement. Under the effect of spin-orbit interaction, the splitting of the d_{xy} and d_{xz} and/or d_{yz} orbitals occurs, lifting the degeneracy of the electronic states and resulting in six branches of dispersion relations with sizable Rashba spin splitting along the Γ - X direction, as shown in Fig. 4(b). Representative Fermi surfaces with $E_F = -0.2$, -0.06 , and

0 eV are displayed in Fig. 4(c), corresponding to different degrees of band filling at the positions marked with dashed lines in Fig. 4(b). When the energy band filling is low ($E_F = -0.2$ eV), a circular Fermi contour is obtained where Rashba splitting is barely visible. With the increase in band filling, when the Fermi level just crosses the bottom of the orbital with higher energy ($E_F = -0.06$ eV), hump-shaped Fermi arcs are observed above the circular baseline across the k_z and k_x axes shown in the middle panel of Fig. 4(c). Meanwhile, a tiny Fermi ring with the opposite spin helicity appears in the center of the Fermi contour. In this case, the spin-splitting energy is as high as around 8.7 meV, which is attributed to the formation of a hump-shaped Fermi arc. As the band filling further increases to $E_F = 0$ eV, hump Fermi arcs are more pronounced, and the inner Fermi ring expands into a larger Fermi ring [the bottom panel of Fig. 4(c)].

To explore the effect of the structural features of the Fermi surface on charge-to-spin conversion, we characterize the Rashba-Edelstein efficiency by the tensor κ , which relates the spin density \mathbf{s} per surface unit cell to the applied electric field E by the expression $\mathbf{s} = \kappa E$. As depicted in Fig. 4(d), the κ_{xy} calculated using the Boltzmann method [44–46] (see Note 4 in the Supplemental Material for details [28]) exhibits a strong dependence on the Fermi

energy. With the increase of Fermi energy, κ_{xy} grows monotonically and then reaches a peak at $E_F = -0.06$ eV, corresponding to the Fermi level being lifted across the bottom of the d_{xz} and/or d_{yz} bands, where orbital mixing enhances Rashba splitting. The subsequent decrease in κ_{xy} could be ascribed to the reduced spin splitting of the d_{xz} and/or d_{yz} bands as Fermi energy grows. Lastly, for even higher energy, a sign change occurs resulting from the competing contributions from multiple bands.

Based on this theoretical analysis combined with experimental results, we can figure out the correspondence between the charge-to-spin conversion and the band filling for 2DEGs at the KTO interface. When only one species of carrier exists, mobile electrons first occupy the low-lying d_{xy} subband. In this case, the value of κ_{xy} is small, which is consistent with the low efficiency of charge-to-spin conversion at the low Fermi level. As the Fermi energy rises, a portion of the electrons starts to fill into the d_{xz} and/or d_{yz} orbitals, and the enhanced spin splitting results in a peak in κ_{xy} , and thus the highest $|\xi_{\text{SOT}}|$ at the Lifshitz point. Subsequently, κ_{xy} decreases as E_F further increases, which explains the decay of $|\xi_{\text{SOT}}|$ as the Fermi level continues to fill higher.

Recently, the direct visualization of Rashba-split bands, using angle-resolved photoemission spectroscopy, has been reported for the Al/KTO(001) surface, and the theoretical calculation, based on the tight-binding model, of the band-resolved spin and orbital Edelstein effects with respect to the Fermi level position has predicted a highly efficient spin/orbital-charge interconversion [23]. The investigation of the SOT efficiency for the Py/ γ -Al₂O₃/KTaO₃ devices using the ST-FMR technique in our work further verifies the high efficiency of the charge-to-spin conversion in the KTO 2DEGs. Moreover, the effective modulation of SOT efficiency by changing the Fermi-level position is observed experimentally.

IV. CONCLUSION

In summary, we demonstrate that high-quality crystalline γ -Al₂O₃ films can be epitaxially grown on (001)-oriented KTO substrates, benefiting from the excellent lattice match between the oxygen sublattices, and the transport behavior of oxide 2DEGs formed in γ -Al₂O₃/KTO spinel/perovskite heterostructures is investigated. ST-FMR measurements are performed on Py/ γ -Al₂O₃/KTO devices to quantitatively determine the efficiency of charge-to-spin conversion at the γ -Al₂O₃/KTO interface originating from spin-momentum locking. The charge-to-spin conversion efficiency $|\xi_{\text{SOT}}|$ has values of up to 3.6 for the Py(6 nm)/ γ -Al₂O₃(3 u.c.)/KTO device at 5 K. The electronic structure of the 2DEG at the KTO interface calculated using the tight-binding model reveals the relationship between this efficient conversion and band filling.

As evidenced by theoretical calculation, the strong dependence of the charge-to-spin conversion on the Fermi level originates from the Rashba spin splitting of the hierarchic band structure, and the enhanced spin splitting due to orbital hybridization leads to the highest conversion efficiency when the Lifshitz transition occurs. This work provides a path to manipulate the SOT efficiency of KTO-based 2DEGs and will facilitate potential applications in spintronic devices.

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The authors declare no competing financial interest.

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