# *Ab initio* **global optimization of the structures of Si***n***H,** *n***= 4 – 10, using parallel genetic algorithms**

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The results of *ab initio* global optimizations of the structures of  $Si<sub>n</sub>H$ ,  $n=4-10$ , atomic clusters using a parallel genetic algorithm are presented. Driving the global search with the parallel implementation of the genetic algorithm are presented and using the density functional theory as implemented in the Carr-Parinello molecular dynamics code to calculate atomic cluster energies and perform the local optimization of their structures, we have been able to demonstrate that it is possible to perform global optimizations of the structure of atomic clusters using *ab initio* methods. The results show that this approach is able to find many structures that were not previously reported in the literature. Moreover, in most cases the new structures have considerable lower energies than those previously known. The results clearly demonstrate that these calculations are now possible and in spite of their larger computational demands provide more reliable results.

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#### **I. INTRODUCTION**

The study of the structure and physical properties of atomic and molecular clusters is an extremely active area of research due to their importance in both fundamental science and applied technology. Existing experimental methods for structural determination seldom can obtain the structure of atomic clusters directly. Therefore the calculation, using theoretical structures and comparison with experimental values of their physical and optical properties, is the most common way to obtain structural information of atomic clusters.

During the last two decades a number of researchers have characterized amorphous hydrogenated silicon  $(\alpha - Si : H)$  by different techniques and determined the physical and chemical properties of amorphous silicon films  $[1,2]$ . The role of defects in the properties of amorphous and microcrystalline silicon ( $\alpha$ -Si and  $\mu$ c-Si, respectively) is very important because these materials are employed in solar cells and optoelectronic devices  $[3]$ . As the hydrogen diffuses and modifies the structure, the energy spectrum and the properties of the devices change. To understand these phenomena it is important to understand how hydrogen affects the local electronic structure and geometry in these systems. Since systems like amorphous and microcrystalline silicon are very difficult to handle computationally, understanding about the effects due to hydrogen addition may be achieved by calculations on small hydrogenated silicon clusters. These calculations critically depend on the available structures of such clusters. Prasad and co-workers [4–10] and Yang *et al.* [11] and references therein constitute a very complete bibliographic record of the state of the art in the determination of the structures of hydrogenated silicon clusters. Due to computational limitations, the authors either limited themselves to calculations using local minimizations when using density

functional theory (DFT) methods for the energy calculations or used approximated methods to calculate the cluster energies when using global search techniques. Unfortunately, this has been the state of the art in the determination of atomic cluster structures [12–14], computational limitations have forced researchers to use either local optimizations or approximate energy calculations. Our recent work in Si*<sup>n</sup>* clusters  $[15]$  shows the need for global optimizations, while the work on Si<sub>n</sub>Cu clusters [16] clearly demonstrates the drawbacks of using approximate energy calculations when the energy hypersurface provided by the approximate method does not closely match the one predicted from first principles. In agreement with other authors  $[12-14]$ , we have concluded that it is of importance to develop the necessary computational capabilities to make possible searches of atomic cluster structures using *ab initio* methods.

In this paper we report the use of a parallel genetic algorithm (MGAC) to predict the structure of medium size hydrogen-silicon clusters, Si*n*H, *n*=4–10, using DFT methods. To our knowledge this is the first calculation of this type reported in the literature.

## **II. METHODS**

Genetic algorithms (GA's) are a family of search techniques rooted on the ideas of Darwinian biological evolution. The introductory sections of Refs.  $[17,18]$  offer a very detailed description of the progress made in the development of GA strategies to perform atomic cluster structure optimizations. The GA methods are based in the principle of survival of the fittest, considering that each string or *genome* represents a trial solution candidate of the problem and that at any generation the *genomes* or "individuals" compete with each other in the population for survival and produce offspring for the next generation by prescribed propagation rules. Operators analogous to crossover, mutation, and natural selection are employed to perform a search able to explore and learn the multidimensional parameter space and determine which

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regions of that space provide good solutions to the problem. One of the advantages of genetic algorithms is that they can provide not only global minima, but also information on other states with energies close to the minimum, an important property when analyzing atomic clusters.

*Genetic coding of the cluster structures*. In this case the genome is quite simple because there is no symmetry or periodicity constrains. In all our work we have followed previous precedence  $[19,20]$  and use a physical space genomic description of the clusters. The genome for a cluster is therefore given by the coordinates of the atoms in the cluster. The genome has a dimension of 3*N*, where *N* is the number of atoms in the cluster.

*GA optimization*. Initially a population of  $N_{\text{pop}}$  random individuals is generated. After that the GA operations of mating (crossover), mutation, and selection are used to evolve one generation into the next. The population replacement is made by employing the steady-state genetic algorithm (SSGA), which typically replaces only a proportion of individuals in each generation  $[21-23]$ . This technique is also known as elitism. Among the population, the best  $N_{\text{pop}}/2$ individuals—that is, 50% of the population—are copied directly into the next generation and the remaining individuals are generated by the GA operators as described below. The criteria for fitness probability, selection of the individuals, and mutation used here are those discussed in detail in Ref. [15]. As with any stochastic minimization procedure the  $GA$ should be run several times to guarantee that the resulting structures are independent of the initial population and statistically significant.

*Techniques used to improve mutations and crossovers*. Connected clusters are arrangements of atoms, which are located at distances such there are significant interatomic interactions among them. Therefore not every possible spatial distribution of atoms makes a valid configuration for physically plausible clusters. This can be used to greatly reduce the effective search space for the GA. In order to take advantage of this feature it is necessary to generate an initial population of compliant structures and to use only genetic operators that guarantee that the new individuals produced have physically plausible structures. In the following we describe the implementation of the genetic operators and initialization techniques used in our current implementation of MGAC.

*Atom connectivity*. We will consider that two atoms are connected if they are at a distance  $[(d_1+d_2)/2]$  where  $d_1$  and  $d_2$  are characteristic interaction distances for each type atom. These distances are selected as proportional to their respective van der Walls radius. In the process of building an atomic cluster we will require that the coordinates of atom *A*<sup>1</sup> satisfy

$$
\|\mathbf{r}_1 - \mathbf{r}_2\|^2 = \left(\frac{d_1 + d_2}{2}\right)^2\tag{1}
$$

for at least one other atom  $A_2$ , where  $\mathbf{r}_1$  and  $\mathbf{r}_2$  are the positions of the atoms. In this way we guarantee that the atoms in the cluster are connected and do not overlap each other. Note that this does not introduce any artificial constrains on the cluster's geometries, because all the structures of the clusters are locally optimized using a quantum mechanical method prior to inclusion in the population.

*Generation of the initial population*. The initial population is generated randomly. Starting with an atom at the center of coordinates, the second atom is placed along the *z* direction at a distance such that complies with the condition given by Eq.  $(1)$ . To continue adding atoms to the cluster the initial fragment is randomly rotated and a new atom is placed along the *z* direction such that its distance to at least one other atom in the cluster follows Eq.  $(1)$ . This procedure is continued until the desired number of atoms is achieved for the cluster. Before including the individual in the population, this construction is followed by a local optimization using the desired quantum mechanical method. This process is used as many times as necessary to generate the desired number of individuals in the initial population.

*Genetic operators*. In the current MGAC implementation we use a real-space representation of the genome defining each individual—i.e., atomic cluster. Therefore, the genetic operators also have to be defined such they operate in real space. The use of this approach for the determination of atomic cluster structures has been previously described [19,20], and many of these GA operators have been proposed. Unfortunately, none of them guarantees that the offspring generated by the operators are compact clusters without disconnected and/or overlapping atoms. Note that the introduction of these unphysical clusters in the population creates a great deal of problems because quite often when using them as starting structures for the local optimizations either take unordinary time to converge to plausible structures or diverge producing abnormal termination of the DFT or alternative code used for the local energy minimizations.

*Recombination operators for homogeneous systems*. Two well-known recombination operators were redefined to produce connected clusters. Those are based in the "cut and splice" and "two cuts and splice" operators defined by Roberts *et al.* [18]. In the case of the "cut and splice" operator the two individuals that are selected for mating—i.e., clusters *A* and *B*—are centered and rotated randomly. The atoms in the two clusters are ordered according to their position in the *z* direction. The index of each atom not only defines the ordering of atoms but also can be used to define an *xy* plane at which the atom belongs. Therefore randomly selecting an index between 1 and *N*, the number of atoms in the cluster, is equivalent to selecting a cut plane for the recombination of the clusters. Following the selection of the cutting plane given by  $m_{\text{pos}}$ , the atoms belonging to cluster *A* with indices in the interval  $[1, m_{pos}]$  are assigned to a fragment  $A_1$  and those with indices  $(m_{\text{pos}}, N]$  are assigned to a fragment  $A_2$ . In a similar manner the individuals belonging to *B* are assigned to the  $B_1$  and  $B_2$  fragments. The mating proceeds by jointing the fragments  $A_1$  with  $B_2$  and  $A_2$  with  $B_1$  to form the two new individuals under the condition that the remaining clusters are connected. This condition is accomplish by moving the two fragments along the direction defined by their center of mass until the positions of all the atoms comply with Eq. (1).

The same procedure follows for "two cuts and splice" operators where in this case is not 1 but 2 the number of cutting planes that are chosen. Therefore for each mating

that the remaining clusters be connected. In both cases, the resulting clusters are locally optimized and incorporated into the new population.

*Recombination operators for heterogeneous systems*. In heterogeneous systems not all the possible planes for cutting are available because the selected fractions do not conserve the number of atoms of each species that define the composition of the cluster. This is because the number of atoms of each species above and below the cutting plane is different. This problem can be resolved by checking for every index if the cutting plane that they define preserves the composition of the subclusters above and below the plane for each cluster. If the composition is preserved, the index is stored and becomes eligible for the random selection process used to determine the cutting plane. Once the cutting planes have been selected, the mating process proceeds as explained above for the homogeneous clusters.

*Mutation operators*. The operators "atom replacement," "twisting," and "permutation" mutation were implemented. For the cases of "atom replacement" and "permutation" the implementation was done following the procedure presented in  $[19]$ . In the case of "twisting," the individual selected for mutation is centered, randomly rotated and its atoms are indexed as described above according to their position along the *z* direction. A randomly chosen index  $m_{\text{pos}}$  is used to assign the atoms with indices  $(m_{\text{pos}}, N]$  to  $A'$  and those to the interval  $[1, m_{\text{pos}}]$  to the fragment *A*<sup>*n*</sup>. The two fragments are independently and randomly rotated and thereafter connected to produce a new structure that after local minimization is incorporated into the population. The use of the "atom replacement" operator can create structures that are slightly not connected, but its use in general is appropriated for keeping a high diversity on the population avoiding the premature convergence of the algorithm.

The new operators introduce here are essentially similar to those that had been previously used in the literature 19,20, but with the incorporation of the notion of connectivity between atoms to restrict the search to connected clusters and therefore speeding up the process of local optimization. We have not tested the effectiveness of the new operators, except for the fact that they were able to produce converged populations in a reasonable number of generations.

*Density functional theory*. The use of DFT to predict the structure of atomic clusters using different exchange correlation functionals is well documented  $[14,24-30]$ , and it is not necessary to discuss its merits. It is universally accepted that DFT is the preferred method for atomic cluster structure calculations and that other less computationally demanding approximations are used due to the lack of the computing capabilities needed for DFT methods  $[14]$ . In this work we used for the local optimizations the DFT methods as they have been implemented in the wave plane code CPMD (Carr Parinello molecular dynamics) [31]. Within the CPMD code we tested different exchange correlation functionals and pseudopotentials, comparing the results with those from

highly correlated calculations (see results section). Based on the tests we decided to use in the local CPMD optimizations the (Becke's exchange correlation and Lee, Yang and Parr exchange functional) [32,33] exchange correlation functional and the Goedecker *et al.* [35] pseudopotential [34,35]. The calculations for the clusters with four to seven Si atoms were done using an energy cutoff of 32 Ry, while the calculations for the larger clusters where done using a cutoff of 16 Ry. In the results section we show that the cutoff of 16 Ry is enough to obtain reliable results while leading to important savings of computational resources with respect to the 32 -Ry cutoff. To avoid the edge effects due to the finite size of the cell introduced by the artificial periodic boundary conditions used by the CPMD code, the cell length was defined as corresponding to the largest dimension of the cluster plus

8 Å on each side. *Computational aspects*. The MGAC package has been written in  $C++$ , MPI, and using GALib [36] for the GA implementation; this makes it very portable as well as easy to maintain and upgrade.

The evaluation of the objective function for our GA—i.e., the local energy minimization using the DFT method requires the largest amount of computer time and it is the limiting factor on the problem size that can be solved using the MGAC method. To alleviate this problem we have used a global parallelization scheme of the genetic algorithm. Such parallelization schemes rely on the simultaneous evaluation of the fitness of the individuals belonging to the same population in every generation using our adaptive parallel genetic algorithm (APGA) [37] strategy. The APGA was designed to perform efficiently on heterogeneous cluster of computers and provides a great degree of adaptability and performance in distributed systems. However, this level of parallelization only allows making the execution time approximately independent of population size. In the ideal case when the number of processors is equal to the number of individuals that are evaluated per generation, the computer time per generation is equal to the evaluation time for one individual. Therefore, the achievable best performance when using this strategy is given by the time needed for the local energy minimization of one cluster using the DFT method. This time rapidly increases with the size of the atomic cluster. To further reduce the total execution time it is necessary to execute the local energy minimization in parallel. This was achieved by using a second level of parallelization, where the evaluation of each individual is done in parallel too. This was accomplished using the parallel capabilities of the CPMD code 31. Table I shows the running and timing parameters from the calculation of the structures of all the isomers studied here with the MGAC-CPMD approach. It is apparent from Table I that our method shows great scalability properties and it can take advantage of modern parallel computer systems.

## **III. RESULTS AND DISCUSSION**

*CPMD validation process*. To validate the suitability of using the different combinations of exchange correlation functionals and pseudopotentials available in the CPMD

TABLE I. Computational and timing parameters used in the calculation of the structures of Si*n*H atomic clusters using the MGAC method for global search and the CPMD method for the local optimization and energy calculations. Note that calculations for  $n=4$ –7 were done using an energy cutoff of 32 Ry, while for *n*  $=8-10$  an energy cutoff of 16 Ry was used.

	Individuals in		Number of computer processors	Elapsed time	Number of	
	the population	Level 1	Level 2	Total	per generation <sup>a</sup>	generations
Si <sub>4</sub> H	10	4	10	40	1.5	10
Si <sub>5</sub> H	10		15	75	$\overline{c}$	15
Si <sub>6</sub> H	15		20	100	2	15
Si <sub>7</sub> H	15		25	175	2	20
Si <sub>8</sub> H	20		25	125	2	30
Si <sub>9</sub> H	30		35	175	1.5	55
Si <sub>10</sub> H	30		43	215	1.5	50

<sup>a</sup>The elapsed time is given in hours. The total execution time of the search is the time per generation times the number of generations needed to achieve convergence.

program we have used a series of high level *ab initio* calculations in H,  $H_2$ , Si, Si<sub>2</sub>, SiH (silylidyne), Si $H_2$  (silicon dihydride),  $SiH_3$  (silyl), and  $SiH_4$  (silane) as benchmarks [38–42]. For each system we calculated the optimized geometry, the binding energy, and the vibrational frequencies for all the valid combinations of exchange correlation functional and pseudo potential available in the CPMD code.

The results for all those systems, when using energy cutoff of 32 Ry and 16 Ry, are summarized in Table II. It is apparent that the quality of the results, as determined by their rms with respect of the experimental values, depends mainly on the accuracy of the pseudopotential (PP) used. Methods

with improved functionals cannot compensate for PP deficiencies. It is observed that the BLYP-GO combination gives one of the best performances for all four geometrical characteristics of the systems. For bond distances BLYP-GO has a rms of 0.025 Å where the best performance was for  $MP2FC/6-311G^*$  with a rms of 0.008 Å. In the cases of bond angles, BLYP-GO has a rms of 0.58° where the best performance was for MP4/6-311G\* with a rms of 0.56°. For binding energy, BLYP-GO has a rms of 0.088 eV where the best performance was for BLYP/6-311G\* with a rms of 0.045 eV. Finally for normal modes frequencies, BLYP-GO has a rms of 59 cm<sup>-1</sup> where the best performance was for

TABLE II. rms between experimental and calculated geometry parameters, binding energy, and vibrational frequencies for selected  $\text{Si}_n\text{H}_m$  systems with  $E_{\text{cut}}$ =32 Ry and 16 Ry.

Method	rms (Re) $(\AA)$	$(\text{deg})$	[eV]	$\text{[cm}^{-1}$	$(\AA)$	$\Delta$ angle rms (Eb) rms (Freq) rms (Re) rms (angle) rms (Eb) rms (Freq) [deg]	[eV]	$\lceil$ cm <sup>-1</sup> $\rceil$
<b>CPMD</b>			Energy cutoff 32 Ry			Energy cutoff 16 Ry		
<b>BLYP/GO</b>	0.025	0.58	0.088	59	0.046	0.53	0.04	102
LDA/GO	0.030	1.23	0.298	92	0.052	1.50	0.22	162
<b>BLYP/MT</b>	0.134	1.46	0.340	270	0.147	1.52	0.54	313
BP/MT	0.122	1.78	0.281	245	0.136	1.78	0.25	294
<b>HCTH/MT</b>	0.129	1.79	0.573	263	0.140	1.72	0.54	283
LDA/MT	0.111	2.34	0.159	248	0.125	2.48	0.31	301
<b>OLYP/MT</b>	0.116	1.70	0.287	223	0.129	1.66	0.34	267
PBE/MT	0.120	1.89	0.338	239	0.133	1.90	3.82	287
revPBE /MT	0.123	1.70	0.363	238	0.135	1.77	3.85	286
All electron calculations								
BLYP/6-311G*	0.038	0.88	0.045	38				
B3LYP/6-311G*	0.020	0.89	0.062	35				
MP2FC/6-311G*	0.008	0.67	0.382	80				
$MP4/6-311G^*$	0.030	0.56	0.319	41				
$QCISD(T)/6-311G^*$	0.032	0.57	0.311	38				
$CCSD(T)/6-311G^*$	0.033	0.57	0.306	41				

TABLE III. Calculated binding energies, average Si-Si bond lengths, Si-H bond lengths, and  $\alpha$ - and  $\beta$ -energy gaps for the Si<sub>n</sub>H clusters found by the MGAC-CPMD method. Their structures are given in Fig. 1. The italicized entries correspond to those found by the MGAC-CPMD method that were not previously reported in the literature  $[4-11]$ .

Bond lengths							
	$(\AA)$						
Isomer	$E_b$ (eV)	$Si-Si$	$Si-H$	$\alpha$ gap (eV)	$\beta$ gap (eV)		
$Si_4H-a^{a,b}$	2.742	2.344	1.511	1.674	1.971		
$Si5H-ab$	2.901	2.389	1.491	1.197	2.126		
$Si5H-ba,b$	2.869	2.425	1.501	0.672	2.701		
$Si5H-ca,b$	2.847	2.497	1.730	1.038	2.186		
$Si6H-ab$	3.006	2.480	1.506	0.953	2.212		
$Si6H-ba$	2.982	2.416	1.502	1.344	1.543		
$Si6H-c$	2.962	2.484	1.498	1.567	1.085		
$Si7H-a$	3.032	2.522	1.425	1.602	1.611		
$\mathrm{Si}_{7}\mathrm{H}$ - $b^{\mathrm{a,b}}$	3.024	2.513	1.507	1.071	2.554		
$Si7H-c$	2.994	2.518	1.519	1.818	1.075		
$Si7H-d$	2.991	2.467	1.499	1.584	1.524		
$Si7H-e$	2.979	2.475	1.503	0.814	1.398		
$Si_8H-a$ <sup>b</sup>	3.056	2.472	1.529	0.672	1.687		
$Si_8H-b$ <sup>a</sup>	3.050	2.525	1.531	1.192	1.171		
$Si_8H-c$	3.044	2.514	1.537	0.832	1.483		
$Si8H-d$	3.042	2.517	1.530	1.346	1.122		
$Si_8H-e$	3.042	2.517	1.530	1.341	1.124		
$Si9H-ab$	3.100	2.488	1.530	0.970	1.360		
$Si9H-b$	3.099	2.496	1.529	0.973	1.367		
$Si9H-c$	3.099	2.508	1.528	1.289	1.308		
$Si9H-d$	3.098	2.495	1.526	1.117	1.155		
$Si9H-e$	3.098	2.501	1.528	1.085	1.446		
$Si10H-aa$	3.173	2.515	1.533	1.408	1.366		
$Si10H-b$	3.172	2.486	1.524	1.165	1.466		
$Si10H-c$	3.171	2.504	1.522	1.257	1.455		
$Si10H-d$	3.166	2.506	1.517	1.397	0.979		

<sup>a</sup>Reported by Prasad and co-workers *et al.* [4–10].

<sup>b</sup>Reported by Yang *et al.* [11].

B3LYP/6-311G<sup>\*</sup> with a rms of 35 cm<sup>-1</sup>. Similar results are obtained when using a smaller cutoff of 16 Ry; again, the best performance for all geometrical characteristics of the systems is the BLYP-GO combination. While the rms values for the smaller cutoff are not as good as those obtained using 32 Ry, they are quite comparable to those obtained using accurate highly correlated methods such as MP2, MP4, QCISD, and CCSD. Following these observations the calculations reported here for the Si*n*H clusters were performed using the BLYP-GO. For clusters with  $n=4-7$  we used a cutoff of 32 Ry, while for the larger ones *n*=8–10 a cutoff of 16 Ry was used.

*Structures of SinH clusters*. In Table III we present the calculated binding energies, average Si-Si bond lengths, Si-H bond lengths, and  $\alpha$ - and  $\beta$ -energy gaps for all the unique Si*n*H clusters found by the MGAC-CPMD method. The corresponding structures of the clusters are given in Fig. 1 and their Cartesian coordinates have been deposited in AIP's EPAPS [43]. Comparing the structures of the clusters in Fig. 1 with the ones reported by Prasad and co-workers  $[4-10]$ and Yang *et al.* [11] it is apparent that the MGAC-CPMD approach is able to find a number of structures that were not found previously. The entries corresponding to these structures have been italicized in Table III. In many cases the new structures correspond to lower-energy structures than those from the literature. The inspection of the calculated vibrational frequencies of all the structures in Fig. 1, deposited as supplementary information in AIP's EPAPS [43], revels that all the calculated frequencies are real, indicating that the structures are true minima in the energy hypersurface and not meta stable configurations.

Most of the structures show one frequency around 2200 cm−1 corresponding to the Si-H stretching vibration mode. The experimental value is  $2042 \text{ cm}^{-1}$  [42] for SiH. This stretching vibrational mode does not appear in  $Si<sub>5</sub>H-c$ 



con atoms.

The  $Si<sub>4</sub>H$  structure corresponds to four silicon atoms forming a flat rhombus and the hydrogen bonded to a silicon atom. From the comparison of the Si-Si bond lengths with those found in Ref. [13] for  $Si<sub>4</sub>$  we see that they are reduced in 0.08 Å due to the addition of the hydrogen. The MGAC-CPMD structure for  $Si<sub>4</sub>H$  is in agreement with all previous reports of the stable structure for this cluster.

FIG. 1. (Continued).

 $(d)$  3.166

 $(c)$  3.171

For  $Si<sub>5</sub>H$  we found three different stable isomers displayed in Fig. 1:  $Si<sub>5</sub>H-a$  matches the new structure reported by Yang *et al.* [11] employing full electron DFT calculations. The four nonhydrogen silicon bonded atoms define a dihedral of 142°. The Si-Si average bond is 2.389 Å and the Si-H bond is 1.491 Å. Si<sub>5</sub>H-*b* and Si<sub>5</sub>H-*c* correspond to those reported by Balamurugan and Prasad 10, but in our case structure  $Si<sub>5</sub>H-c$ , which presents a hydrogen atom bonded to two silicon atoms, is not the most stable in agreement with Yang et al. [11]. The Si-Si distances coincide with those previously reported in  $[13]$  for  $Si<sub>5</sub>$ , but the bond lengths are slightly different, they change in about 0.06 Å and the angles (bonding and dihedral) change by approximately 5°, because of the presence of hydrogen in the cluster.

For  $Si<sub>6</sub>H-a$  the Si-H bond length is 1.506 A, the hydrogen and four silicon atoms are in a plane, and the other two silicon atoms are above and below it, lying in an orthogonal plane formed by four silicon atoms. The Si-Si average bond is 2.480 Å. The corresponding  $Si<sub>6</sub>$  isomer [13] shows an average bond length of 2.434 Å and a dihedral angle of 24° instead of a flat four silicon plane.  $Si<sub>6</sub>H-b$  coincides with the isomer reported by Balamurugan and Prasad [10]. In a near energy range, higher by 0.02 eV, we found that isomer  $Si<sub>6</sub>H-c$  differs from the  $Si<sub>6</sub>H-b$  one in the silicon site to which the hydrogen is bonded.

 $Si<sub>7</sub>H-a$  and  $Si<sub>7</sub>H-c$  are new isomers, with structures very similar to  $Si<sub>7</sub>H-b$ , reported by Balamurugan and Prasad [10]. A five-silicon ring characterizes them, and they differ in the silicon site bonded to the hydrogen atom.  $Si<sub>7</sub>H-d$  and  $Si<sub>7</sub>H$ -e are also new isomers reported here. For isomers (a)-(c) and (e), the silicon structure was previously found in Ref. [13], but the average Si-Si distance have been reduced by 0.03 Å.

We found five stable configurations for  $Si<sub>8</sub>H$ : isomer  $Si_8H-a$  was reported by Yang *et al.* [11], isomer  $Si_8H-b$  was reported by Balamurugan and Prasad [10], and isomers  $Si_8H-c$ ,  $Si_8H-d$ , and  $Si_8H-e$  are new structures produced by MGAC-CPMD. The Si-H bond length is 1.529 Å for isomer (a), 1.531 Å for (b), 1.537 Å for (c), and 1.530 Å for (d) and (e), which are mirror images of the same isomer. The corresponding Si-Si average bond lengths are 2.472 Å, 2.525 Å, 2.514 Å, and 2.517 Å, respectively. Comparing these structures with the  $Si_8$  [13] we observe that the addition of hydrogen increases the symmetry of these isomers.

We found five isomers for  $Si<sub>9</sub>H$ ; structures  $Si<sub>9</sub>H-a$  and Si9H-*b* are mirror structures of the same isomer. Yang *et al.* also reported the ground state  $Si<sub>9</sub>H-a$  [11]. The average Si -Si bond lengths are of the order of 2.50 Å, and Si-H runs between 1.526 Å and 1.529 Å. From the comparison of these structures with those of  $Si<sub>9</sub>$  in Ref. [13] we conclude that the Si-Si bond length is shortened in  $\approx 0.05$  Å and all the Si<sub>9</sub>H isomers might be associated to one of those  $Si<sub>9</sub>$ .

We found four stable isomers for  $Si<sub>10</sub>H$ ; the ground state corresponds to that reported by Balamurugan and Prasad [10]; the Si-Si-average bond length is  $2.515 \text{ Å}$  for isomer  $Si<sub>10</sub>H-a$ , 2.486 Å for isomer  $Si<sub>10</sub>H-b$ , and 2.505 Å for  $Si<sub>10</sub>H-c$  and  $Si<sub>10</sub>H-d$ . The corresponding Si-H bond lengths are 1.533 Å, 1.524 Å, 1.522 Å, and 1.517 Å, respectively. From the comparison of these isomers with those reported in Ref. [13] we observe that for  $Si<sub>10</sub>H-a$ ,  $Si<sub>10</sub>H-c$ , and  $Si<sub>10</sub>H$  $-d$  there is a  $Si<sub>11</sub>$  for which one silicon atom seems replaced by an hydrogen on these  $Si<sub>10</sub>H$  compounds.

Our results show good agreement with previous searches of the configurations of small SiH clusters, but consistently we found increasing numbers of new isomers for the larger ones. This is a clear indication that global searches become increasingly important for larger clusters. On the question of the Si-H binding we found that for the most stable isomer the hydrogen is always bonded to only one Si atom and it is on the outside of the cluster. This result follows the electronegativity argument recently provided by Yang *et al.* [11].

## **IV. CONCLUSIONS**

This paper clearly demonstrates that it is possible, using current computational resources, to perform global searches of the structures of medium size atomic clusters using firstprinciples (DFT) methods. The results clearly show that this procedure is more reliable than previous approximate methods based on either approximate energy calculations and/or more constrained searches. The ability of the MGAC-CPMD method to find better isomers than those previously found is a clear justification for its use in spite of the additional computational resource needed for the calculations. Note the importance of performing global searches greatly increases with the size of the cluster. Moreover, the architecture of the MGAC-CPMD software allows for taking advantage of large computer clusters making these large calculations feasible when using parallel computer systems available at major research centers. The comparison of our DFT results with those for which either experimental data or high-level calculations exists shows that the quality of the results depends mainly on the accuracy of the pseudopotential used. Methods with improved functionals cannot compensate for PP deficiencies. It is observed that the BLYP-GO combination gives one of the best performances for the Si*n*H clusters studied here.

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