

## Dispersion $C_3$ coefficients for physisorption of heavy ions and atoms with graphene and carbon nanotubes

Harpreet Kaur <sup>1</sup>, Neelam Shukla <sup>2</sup>, Rajesh Srivastava <sup>2</sup> and Bindiya Arora <sup>1,\*</sup>

<sup>1</sup>*Department of Physics, Guru Nanak Dev University, Amritsar, Punjab 143005, India*

<sup>2</sup>*Department of Physics, Indian Institute of Technology Roorkee, Roorkee, Uttarakhand 247667, India*



(Received 6 April 2021; accepted 23 June 2021; published 15 July 2021)

In the present work, the calculations have been performed for dispersion interaction between heavy elements ( $Zn^+$ ,  $Cd^+$ ,  $Hg^+$ ,  $Pb^+$ ,  $Zn$ ,  $Cd$ ,  $Hg$ , and  $Pb$ ) with graphene and carbon nanotubes by evaluating van der Waals  $C_3$  coefficients using the well-known Lifshitz theory. The dispersion coefficients are expressed in terms of reflection coefficients of graphene and carbon nanotubes which are calculated within the framework of the Dirac model. In addition, accurate values of dynamic dipole polarizabilities at imaginary frequencies of considered ions and atoms which are vital have been calculated using relativistic methods. The comparisons of our calculated static dipole polarizabilities for the considered elements with the values reported in the literature are also presented. The dispersion interactions of the considered heavy elements with the graphene and carbon nanotubes of different radii in a wide range of separation distances have been studied. The results have also been analyzed with another subtle variable, i.e., the gap parameter of the graphene wall. The interaction coefficients obtained for both the materials, i.e., for graphene and carbon nanotubes, are mutually compared and it is found that graphene can be said to be a preferable material for adsorption of these toxic heavy elements.

DOI: [10.1103/PhysRevA.104.012806](https://doi.org/10.1103/PhysRevA.104.012806)

### I. INTRODUCTION

Metals occur naturally in the earth's crust, and their contents in the environment influence the ecologies in many habitats [1]. Among various metals, heavy atoms and their ions with high atomic weights and large densities are found to be toxic to the human body even when present in trace amounts in various environmental matrices [2]. This has led to growing public health concerns about heavy metal pollution. Nonbiodegradable characteristics of these elements have the capability of causing detrimental effects to the entire biodiversity [3,4]. The high solubility of heavy ions leads to contamination of natural resources such as water and soil, which as a consequence get accumulated in organisms and enter the food chain leading to a process of biomagnification [5]. Excessive exposure of Zn can lead to brain, respiratory, and gastrointestinal syndrome [6]. Cd species can cause skeletal damage as a secondary response to kidney damage or direct action on the bone cells, whereas Hg species being carcinogenic cause adverse effects on the development of the human brain [7]. Hematopoietic, renal, reproductive, and central nervous systems are vulnerable towards the dangers caused by exposure to the high level of Pb species [8]. The primary sources of these elements are various industrial activities, natural resources, agriculture, and untreated disposal of domestic waste [4,9]. Therefore, accurate and accessible detection of these toxic elements is necessary to ensure environmental quality control and early warning capabilities to avoid public safety adversity.

Detection of these elements with various conventional materials like clay, its minerals, zeolites, activated carbon, fullerenes, biomaterials, etc., has been done previously [10–12]. Further, nanomaterials show great technological advances in a wide range of applications due to extraordinary properties as compared to their bulk counterparts [13]. The rapid growth of nanomaterials for various applications has seen a boost after the discovery of graphene. Many breakthroughs in the research of graphene have been observed in the last decade due to its large surface-to-volume ratio, thin structure, and interface interactions. Graphene and graphene-based nanostructures render unique mechanical, electrical, optical, and thermal properties [14–17] that have significantly made this material one of the most studied two-dimensional (2D) materials in condensed matter physics contributing in various applications like electrochemical devices, solar cells, plasmonic, purifiers, sensors, etc. [18–22]. Besides this one-dimensional (1D) allotrope of carbon, single-walled carbon nanotubes (SWCNTs) with diameter less than 50 nanometers (nm) having different configurations exhibit similar properties as that of single-layer graphene [23].

It has been observed both experimentally and theoretically that adsorption technology can monitor trace amounts of heavy metals. Chemical adsorption of adsorbate on a graphene-based system can modify its properties, providing a nonreversible binding of the atom or molecule to the surface. Therefore, physical adsorption is always preferred due to its reversible nature. Graphene and carbon nanotubes (CNTs) have been extensively explored for physical adsorption of some of the heavy ions, dye molecules, and hydrogen molecules [24–27] for sensor applications. Even now, the interaction studies for physisorption of heavy

\*bindiya.phy@gndu.ac.in

elements with graphene and CNTs have been done theoretically and calculations are performed using density functional theory (DFT) [28–32]. Abdesalam *et al.* [33] and Shtepliuk *et al.* [24] studied the adsorption of toxic heavy elements on the graphene-based system. However, a study by Oyetade *et al.* [34] showed nitrogen-functionalized carbon nanotubes as a good reusable adsorbent for the removal of  $\text{Pb}^{+2}$  and  $\text{Zn}^{+2}$  from waste water.

Other studies for physical adsorption of microparticles with the material given by generalized Lifshitz theory have been conducted using *ab initio* calculations [35–37]. The theory explains the interactions of atoms or molecules with material walls in both retarded and nonretarded regimes giving rise to Casimir-Polder and van der Waals (vdW) forces [38–40]. These forces find diverse applications in circuit technology, adsorption, quantum reflections, and Bose condensation [37,40–43]. Lifshitz theory gave a generalization of both these interaction forces in which the strength of the attractive forces is expressed in terms of the dispersion  $C_3$  coefficient [44]. Dispersion coefficients have been calculated for a number of material walls, including metals, semiconductors, insulators, and dielectrics, by taking the optical properties into account [44–50]. These dispersion coefficients were also measured experimentally using atomic force microscopy (AFM) and spectroscopy techniques [51–54]. Such studies were reported for applications in hydrogen sensing, storage, and designing an up-gradation technology for batteries [39,49,55].

In the present work, we particularly focus on the interaction of heavy elements with carbon-based systems—graphene and CNT, which are considered as two-dimensional free-electron gas. Reflection coefficients of these materials are important contributors to the calculation of the dispersion coefficients. Out of the models proposed in the literature for the evaluation of reflection coefficients, the Dirac model approach is preferred due to its providing results in close agreement with experiment [56]. Previously, studies conducted were based on this approach for the interaction of alkali-metal atoms, alkaline ions, noble gas molecules, hydrogen atom, and hydrogen molecule with graphene and CNT wall [39,57]. Accurate values of the polarizability of microparticles at imaginary frequencies are necessary to compute  $C_3$  coefficients between the microparticle and the material wall given by generalized Lifshitz theory. In this paper, we have calculated the  $C_3$  dispersion coefficients for the interaction of a microparticle with graphene and CNT wall along with the evaluation of static and dynamic polarizabilities of heavy ions and atoms at imaginary frequencies using the sum-over-states approach. There are a few studies that have reported only static polarizabilities. Most of these have used nonrelativistic methods but for such heavy elements it is necessary to adopt a relativistic approach as we have done in the present work for the reliable calculations of atomic properties.

The outline of the paper is as follows. In Sec II, we give a brief overview of the theory. Section III contains the evaluated values of static dipole polarizability of heavy ions and atoms. The dynamic dipole polarizabilities for ions and atoms are also presented in the same section. In addition to this, the dispersion coefficients between considered ions or atoms and materials have been discussed. We have also compared the results of dispersion coefficients for graphene and CNT. The

dependency of the gap parameter on interaction coefficients is also discussed in this section. Atomic units (a.u.) have been used throughout the paper unless stated otherwise.

## II. THEORY

### A. Dispersion coefficient

The generalized Lifshitz formula for the nonretarded vdW interaction energy of atoms or molecules with graphene and CNT wall using proximity force approximation (PFA) can be written in terms of dispersion coefficients  $C_3$  for a separation distance  $a$  in the following form [58]:

$$E(a) = -\frac{C_3(a)}{a^3}. \quad (1)$$

The  $C_3$  coefficient due to interaction between graphene and a microparticle is expressed in the terms of reflection coefficients  $r_{\text{TM}}$  and  $r_{\text{TE}}$  as follows [58,59]:

$$C_3(a) = \frac{1}{16\pi} \int_0^\infty \alpha(i\xi) d\xi \int_{2a\xi\alpha_{fs}}^\infty y^2 e^{-y} dy \times \left[ 2r_{\text{TM}} - (r_{\text{TM}} + r_{\text{TE}}) \frac{4a^2 \xi^2 \alpha_{fs}^2}{y^2} \right], \quad (2)$$

whereas this coefficient for a CNT of radius  $R$  becomes radius dependent and can be expressed as

$$C_3(a, R) = \frac{1}{16\pi} \sqrt{\frac{R}{R+a}} \int_0^\infty \alpha(i\xi) d\xi \int_{2a\xi\alpha_{fs}}^\infty y e^{-y} dy \times \left( y - \frac{a}{2(R+a)} \right) \times \left[ 2r_{\text{TM}} - (r_{\text{TM}} + r_{\text{TE}}) \frac{4a^2 \xi^2 \alpha_{fs}^2}{y^2} \right]. \quad (3)$$

In both the above expressions,  $\alpha$  is the dynamic dipole polarizability of the ion or atom over imaginary frequencies  $i\xi$  and  $\alpha_{fs}$  is the fine structure constant [58].  $y$  is a dimensionless variable given by  $y = 2aq$ , where  $a$  is the separation distance and  $q = \sqrt{k^2 + \xi^2}$ , dependent on wave vector  $k$  [58]. For the evaluation of these reflection coefficients, two models have been proposed in the literature for graphene and CNTs. These two are Dirac [60–62] and hydrodynamic models [39,63]. In the hydrodynamic model, graphene is taken as an infinitesimally thin positively charged sheet with a continuous fluid of mass and negative charge densities. The dispersion relation for quasiparticles in graphene is quadratic with respect to the momentum. However, this model does not take into account some properties of graphene which are important at low energies and due to this reason it overestimates the vdW interactions. In the Dirac model, the quasiparticles in graphene are considered to be Dirac fermions moving with Fermi velocity and following the linear dispersion law. This model has provided results in accord with experimental values [56]. In this work, the Dirac model has been implemented for determination of dispersion coefficients. Under this framework, the explicit

forms of two components of the reflection coefficients are given by [61]

$$r_{TM} = \frac{\alpha_{fs} q \phi(\tilde{q})}{2\tilde{q}^2 + \alpha_{fs} q \phi(\tilde{q})}, \quad (4)$$

$$r_{TE} = \frac{\alpha_{fs} q \phi(\tilde{q})}{2\tilde{q} + \alpha_{fs} q \phi(\tilde{q})}, \quad (5)$$

where  $q = \sqrt{k^2 + \xi^2/c^2}$ ,  $\tilde{q}$  is the function of Fermi velocity  $v_f$  of massless Fermions, and  $\phi$  is the polarization tensor. The expressions of these two parameters can be given as [61]

$$\tilde{q} = \sqrt{\frac{\alpha_{fs}^2 v_f^2 y^2}{4a^2} + (1 - \alpha_{fs}^2 v_f^2) \alpha_{fs}^2 \xi^2}, \quad (6)$$

$$\phi(\tilde{q}) = 4 \left[ \alpha_{fs} \Delta + \frac{\tilde{q}^2 - 4\alpha_{fs}^2 \Delta^2}{2\tilde{q}} \arctan \left( \frac{\tilde{q}}{2\alpha_{fs} \Delta} \right) \right], \quad (7)$$

where  $\Delta$  is the gap parameter [62] whose value lies in the range  $0 < \Delta < 0.1$  eV. Since the value of the gap parameter is still not known, we have taken its value initially as 0.01 throughout the paper unless stated otherwise.

### B. Dipole polarizability

The dipole polarizability for an atomic system in the ground state  $n$  with a closed core and valence electron(s) can be evaluated by calculating two components of the polarizability as follows [64]:

$$\alpha(\omega) = \alpha_{\text{val}}(\omega) + \alpha_c(\omega), \quad (8)$$

where the subscripts val and  $c$  refer respectively to the polarizability contributions due to valence and core orbitals respectively. The dominant contribution to the polarizability is from the valence part which can be further expressed in terms of main and tail contributions. The main term of  $\alpha_{\text{val}}$  contains the contributions due to the low lying allowed transitions from the ground state whereas the tail term has the contributions of the transition from ground to higher states. The main term of the valence contribution can be estimated as follows:

$$\alpha_{\text{val}}^{\text{main}}(\omega) = \frac{2}{3(2J_n + 1)} \times \sum_{m > N_c, m \neq n} \frac{(E_m - E_n) |\langle \psi_n || \mathbf{D} || \psi_m \rangle|^2}{(E_m - E_n)^2 + \omega^2}. \quad (9)$$

In the above equation,  $J_n$  is the total angular momentum quantum number of the ground state of the considered atom or ion; the sum is restricted by including the sum over intermediate  $m$  states after  $N_c$  and up to  $I$ , where  $N_c$  represents the core orbitals and  $I$  refers to the bound states up to which we have determined the reduced matrix elements  $\langle \psi_n || \mathbf{D} || \psi_m \rangle$  in our calculations. We use the relativistic all order method and multiconfigurational Dirac-Fock (MCDF) approximation for ions and atoms respectively to compute the matrix elements used for the main term calculation. In order to do reliable calculations and avoid any uncertainties, we use the experimental excitation energy values  $E_i$  of the corresponding states for the main term taken from the National Institute of Standards and Technology (NIST) database [65].

Similarly, the tail term is evaluated using the following equation:

$$\alpha_{\text{val}}^{\text{tail}}(\omega) = \frac{2}{3(2J_n + 1)} \sum_{m > I} \frac{(\epsilon_m - \epsilon_n) |\langle \psi_n || \mathbf{D} || \psi_m \rangle_{\text{DHF}}|^2}{(\epsilon_m - \epsilon_n)^2 + \omega^2}, \quad (10)$$

where  $\langle \psi_n || \mathbf{D} || \psi_m \rangle_{\text{DHF}}$  are the  $E1$  reduced matrix elements obtained using the Dirac-Hatree-Fock (DHF) method.  $\mathbf{D}$  is the dipole operator defined as  $\mathbf{D} = -e \sum_j \mathbf{r}_j$  with  $\mathbf{r}_j$  being position of a  $j$ th electron, and the sum  $m > I$  corresponds to the excited states whose matrix elements are not accounted for in the main term. The energies calculated using the DHF method are referred to by  $\epsilon_i$ . The calculations of the core polarizabilities of both ions and atoms are carried out in the DHF method using the following expression:

$$\alpha_c(\omega) = \frac{2}{3(2J_n + 1)} \sum_a^{N_c} \sum_m^I \frac{(\epsilon_m - \epsilon_a) |\langle \psi_a || \mathbf{D} || \psi_m \rangle_{\text{DHF}}|^2}{(\epsilon_m - \epsilon_a)^2 + \omega^2}, \quad (11)$$

where  $a$  refers to the core orbitals while  $m$  includes valence or empty orbitals. The evaluation of the core correlation using the above expression does not exclude contributions from excitations from the core to the occupied valence shell which are forbidden by Pauli's exclusion principle. Hence half of this contribution has to be subtracted in the case of ions. Likewise for atoms, twice this contribution has to be excluded from the core polarizability contribution due to fully filled valence  $ns$  for Zn ( $n = 4$ ), Cd ( $n = 4$ ), Hg ( $n = 5$ ) and  $np$  for Pb ( $n = 6$ ) orbitals. These contributions are referred to as the valence core ( $\alpha_{vc}$ ) in our calculations. One can calculate the static values of polarizability by substituting  $\omega = 0$  in Eqs. (9)–(11).

### C. Matrix elements

In order to calculate polarizability of the monovalent ions and divalent atoms, reliable values of the matrix elements have to be calculated. In the present work, wave functions for ions and atoms are calculated using different relativistic methods. For ions, we consider the relativistic all order method confined to the single and double excitation (SD) approximation [66,67]. The exact wave function of the state with the closed core and single valence electron  $v$  is represented as

$$|\psi_v\rangle_{\text{SD}} = \left[ 1 + \sum_{ma} \rho_{ma} a_m^\dagger a_a + \frac{1}{2} \sum_{mlab} \rho_{mlab} a_m^\dagger a_l^\dagger a_b a_a + \sum_{m \neq v} \rho_{mv} a_m^\dagger a_v + \sum_{mla} \rho_{mlva} a_m^\dagger a_l^\dagger a_a a_v \right] |\phi_v\rangle. \quad (12)$$

Here  $|\phi_v\rangle$ , is the mean field wave function constructed as  $|\phi_v\rangle = a_v^\dagger |0_c\rangle$  with  $|0_c\rangle$  representing the DHF wave function of the closed core and  $a^\dagger$ ,  $a$  represents creation and annihilation operators respectively, whereas excitation coefficients are denoted by  $\rho$ .  $\rho_{ma}$ ,  $\rho_{mv}$ ,  $\rho_{mlab}$  and  $\rho_{mlva}$  are the single core, single valence, double core, and double valence excitation coefficients respectively. To obtain the DHF wave functions and matrix elements for each transition, we use a set of 50  $B$  splines of order  $k = 11$  for each angular momentum. The

TABLE I. State polarizability along with contributions from various  $E1$  reduced matrix elements to the static polarizabilities (in a.u.) of ground state of  $Zn^+$ ,  $Cd^+$ ,  $Hg^+$ , and  $Pb^+$ . Main, tail, core, and valence-core contributions are given as well. The numbers in square brackets for contribution from each transition in main term represent powers of 10. The final results are compared with the previously estimated and available experimental values.

$Zn^+$			$Cd^+$			$Hg^+$			$Pb^+$		
Transition	$E1$	$\alpha(0)$	Transition	$E1$	$\alpha(0)$	Transition	$E1$	$\alpha(0)$	Transition	$E1$	$\alpha(0)$
$4S_{1/2} - 4P_{1/2}$	0.189[1]	0.537[1]	$5S_{1/2} - 5P_{1/2}$	0.194[1]	0.623[1]	$6S_{1/2} - 6P_{1/2}$	0.166[1]	0.391[1]	$6P_{1/2} - 7S_{1/2}$	0.101[1]	0.125[1]
$4S_{1/2} - 4P_{3/2}$	0.267[1]	0.106[2]	$5S_{1/2} - 5P_{3/2}$	0.275[1]	0.119[2]	$6S_{1/2} - 6P_{3/2}$	0.235[1]	0.666[1]	$6P_{1/2} - 8S_{1/2}$	0.371[0]	0.113[0]
$4S_{1/2} - 5P_{1/2}$	0.80[-1]	0.46[-2]	$5S_{1/2} - 6P_{1/2}$	0.10[0]	0.77[-2]	$6S_{1/2} - 7P_{1/2}$	0.535[0]	0.19[0]	$6P_{1/2} - 6D_{3/2}$	0.207[1]	0.448[1]
$4S_{1/2} - 5P_{3/2}$	0.85[-1]	0.52[-2]	$5S_{1/2} - 6P_{3/2}$	0.59[-1]	0.27[-2]	$6S_{1/2} - 7P_{3/2}$	0.366[0]	0.88[-1]			
$4S_{1/2} - 6P_{1/2}$	0.86[-1]	0.45[-2]	$5S_{1/2} - 7P_{1/2}$	0.113[0]	0.85[-2]						
$4S_{1/2} - 6P_{3/2}$	0.143[0]	0.13[-1]	$5S_{1/2} - 7P_{3/2}$	0.117[0]	0.91[-2]						
$\alpha_{val}^{Main}$		15.98	$\alpha_{val}^{Main}$		18.14	$\alpha_{val}^{Main}$		10.84	$\alpha_{val}^{Main}$		5.84
$\alpha_{val}^{Tail}$		0.02	$\alpha_{val}^{Tail}$		0.01	$\alpha_{val}^{Tail}$		0.06	$\alpha_{val}^{Tail}$		2.67
$\alpha_{vc}$		0.006	$\alpha_{vc}$		-0.02	$\alpha_{vc}$		-0.04	$\alpha_{vc}$		-2.28
$\alpha_c$		2.05	$\alpha_c$		5.28	$\alpha_c$		8.21	$\alpha_c$		16.30
Total		18.05	Total		23.41	Total		19.07	Total		22.52
Experiment		15.54 [71]									
Others		18.84 [74]	Others		23.68 [74]	Others		19.36 [74]	Others		23.5 [76]
		17.90 [76]			23.1 [76]			17.50 [76]			
					25.21 [75]						

basis set orbitals are constrained to a large spherical cavity of a radius  $R = 220$  a.u.

The required wave functions for divalent systems are obtained from the GRASP2K code which uses the MCDF approach [68]. In MCDF, the atomic state wave function (ASF) in their initial or final state can be written as the linear combination of several configurational state functions (CSFs), having the same parity and total angular momentum, e.g.,

$$|\psi_v\rangle_{MCDF} = \sum_{x=1}^N a_x |\phi_x\rangle, \quad (13)$$

where  $x$  refers to the number of CSFs and  $a_x$  is the mixing coefficient. It is important to mention that the calculation of ASFs is done by including Breit and quantum electrodynamic corrections. In order to increase the accuracy of the ASF, we consider the maximum number of CSFs in the linear contribution and, finally, retain only those which have the value of mixing coefficient greater than  $10^{-3}$ . This method was used for divalent alkaline-earth atoms in Ref. [69].

After obtaining wave functions for the aforementioned ions and atoms, we determine the dipole-allowed ( $E1$ ) matrix element for a transition. The  $E1$  matrix elements between the states  $|\psi_v\rangle$  and  $|\psi_k\rangle$  are evaluated using the following expression [70]:

$$D_{vk} = \frac{\langle \psi_v | D | \psi_k \rangle}{\sqrt{\langle \psi_v | \psi_v \rangle \langle \psi_k | \psi_k \rangle}}. \quad (14)$$

For practical purposes, we calculate the  $E1$  matrix elements of some low lying transitions, which contribute dominantly to the main term of the valence contribution using the above described method. The tail contribution from high lying transitions calculated using DHF method is given for ions only. Due to some computational constraints and the sake of simplicity, the tail contribution in the case of atoms has been neglected.

### III. RESULTS AND DISCUSSION

#### A. Dipole polarizabilities at imaginary frequencies

##### 1. Static dipole polarizability of ions

In Table I, we present the static dipole polarizabilities values of  $Zn^+$ ,  $Cd^+$ ,  $Hg^+$ , and  $Pb^+$  heavy ions. Using Eqs. (9) and (10), the main and tail terms of the valence contribution of polarizability are computed at zero frequency and given explicitly in Table I. We provide the breakdown of polarizability values from every dominant transition required for the calculation of the main term of valence contribution. Values of  $E1$  matrix elements included in the main term of  $Zn^+$ ,  $Cd^+$ , and  $Hg^+$  have been calculated in the present work while  $E1$  matrix elements for  $Pb^+$  ion have been taken from Refs. [72,73] which were calculated by the same method as ours. The core contribution is also tabulated in the same table which has been evaluated using Eq. (11). While the  $\alpha_{vc}$  contribution for  $Zn^+$ ,  $Cd^+$ , and  $Hg^+$  ions is almost negligible, it is notable for  $Pb^+$  and affects the total polarizability value. A similar case is observed for the tail term in which a significant value is observed for  $Pb^+$ .

In the same table, the static polarizability values of the ions are compared with the experimental and other theoretical values to mark the validity of our values using the considered method. The polarizability values of  $Zn^+$ ,  $Cd^+$ , and  $Hg^+$  ions match well with the values calculated by coupled-cluster single double with triple excitations [CCSD(T)] method by Iliáš *et al.* [74]. Our polarizability value of the  $Zn^+$  ion deviates from the experimental value by 16%, but it is in close agreement with other theoretical works. In recent work, Li *et al.* calculated the ground state polarizability for  $Cd^+$  using the DHF approximation, third-order many-body theory, and singles and doubles approximated coupled-cluster method [75]. The only difference in the value calculated by us and Ref. [75] is that they have included the  $4d^9 5s 5p$  configurations. The static polarizability value of  $Pb^+$  in Ref. [76]

TABLE II. State polarizability along with contributions from various  $E1$  reduced matrix elements to the static polarizabilities (in a.u.) of ground state of Zn, Cd, Hg, and Pb. Main, core, and valence-core contributions are given as well. The final results are compared with the previously estimated and available experimental results. The numbers in square brackets for contribution from each transition in main term represent powers of 10. The uncertainty in experimental values are given in the parentheses.

Zn			Cd			Hg			Pb		
Transition	$E1$	$\alpha(0)$	Transition	$E1$	$\alpha(0)$	Transition	$E1$	$\alpha(0)$	Transition	$E1$	$\alpha(0)$
$4^1S_0 - 4^3P_1$	0.17[-3]	0.748[-5]	$5^1S_0 - 5^3P_1$	0.208[-1]	0.278[-3]	$6^1S_0 - 6^3P_1$	0.118[0]	0.437[0]	$6^3P_0 - 7^3P_1$	0.16[-1]	0.692[1]
$4^1S_0 - 4^1P_1$	0.103[2]	0.322[2]	$5^1S_0 - 5^1P_1$	0.949[1]	0.318[2]	$6^1S_0 - 6^1P_1$	0.725[1]	0.196[2]	$6^3P_0 - 7^1P_1$	0.80[-1]	0.213[0]
$4^1S_0 - 5^3P_1$	0.40[-4]	0.839[-6]	$5^1S_0 - 6^3P_1$	0.16[-2]	0.407[-4]	$6^1S_0 - 7^3P_1$	0.93[-3]	0.196[-4]	$6^1S_0 - 7^1P_1$	0.491[1]	0.362[2]
$4^1S_0 - 5^1P_1$	0.466[0]	0.108[1]	$5^1S_0 - 6^1P_1$	0.488[1]	0.119[2]	$6^1S_0 - 7^1P_1$	0.554[-1]	0.114[0]	$6^3P_0 - 6^3D_1$	0.229[1]	0.704[1]
$4^1S_0 - 6^3P_1$	0.40[-4]	0.899[-6]	$5^1S_0 - 7^3P_1$	0.4[-4]	0.865[-6]	$6^1S_0 - 8^3P_1$	0.143[0]	0.273[0]			
$4^1S_0 - 6^1P_1$	0.18[-1]	0.251[-3]	$5^1S_0 - 7^1P_1$	0.282[0]	0.628[0]	$6^1S_0 - 8^1P_1$	0.428[-1]	0.815[-3]			
$\alpha_{val}^{Main}$		33.29	$\alpha_{val}^{Main}$		44.36	$\alpha_{val}^{Main}$		20.51	$\alpha_{val}^{Main}$		50.39
$\alpha_{vc}$		-0.0013	$\alpha_{vc}$		-0.04	$\alpha_{vc}$		-0.08	$\alpha_{vc}$		-4.58
$\alpha_c$		2.05	$\alpha_c$		5.28	$\alpha_c$		8.21	$\alpha_c$		16.30
Total		35.33	Total		49.61	Total		28.65	Total		61.90
Experiment		38.80(0.3)	Experiment		49.65(1.46)	Experiment		33.91(0.34)	Experiment		56.0(18.2)
		[78]			[79]			[80]			[81]
Others		37.6 [82]	Others		46.8 [82]	Others		31.2 [82]	Others		46.96 [83]
		38.4 [76]			46.7 [76]			33.5 [76]			47.9 [76]
		38.12 [84]			44.63 [84]			31.32 [84]			

was calculated using time dependent DFT (TDDFT) without including relativistic effects and fixed core approximation. The incorporation of relativistic effects for heavy elements is required for accurate polarizability values [74], thus the values obtained in the present work using the all-order method are expected to be closer to the actual values. In a number of other studies, the present method has provided accurate values of dipole polarizability for other monovalent atoms and ions [77] hence we can say that the static dipole polarizability value of the  $Pb^+$  ion is also legitimate if calculated by the all-order SD method. Unfortunately, we did not find any experimental measurements for static polarizability values of  $Cd^+$ ,  $Hg^+$ , and  $Pb^+$  ions with which we can compare our calculated results.

## 2. Static dipole polarizability of atoms

In Table II, we give the static polarizability results for Zn, Cd, Hg, and Pb atoms. The breakdown of contribution of the main term from each transition is tabulated in the same table. The  $E1$  matrix elements have been obtained from the GRASP2K code required for the calculation of the main term of considered atoms. The core contribution for the considered atoms is the same as that for the respective ions. The focus is given on the calculation of the valence-core  $\alpha_{vc}$  correlation for atoms. Since the excitations from the core to the occupied valence shell, which is completely filled in the case of considered atoms, are not allowed, exactly twice the  $\alpha_{vc}$  contribution calculated in the case of ions with one valence electron has been excluded in the case of atoms. We have not included the tail term in the case of atoms. However, we anticipate a very small tail value from the considered atoms except for Pb.

In the same table, we also present a comparison of the total value of static polarizability of the atoms calculated by us with experimental and other theoretical works. The static dipole polarizability values for Zn and Hg atoms given by Ye *et al.* [84] using configuration interaction with a semiempir-

ical core-polarization model potential method is found to be slightly larger than those calculated by us. Our polarizability values for Zn and Hg differ from experimental value by  $\sim 9\%$  and  $\sim 15\%$  respectively. The static polarizability values of Cd atom calculated by us agree well with the experimental results. For Pb, the previous theoretical results have given underestimated values as compared to experimental values [76,83]. The recent experimental value of the Pb atom is 56 a.u. with an uncertainty of about  $\pm 18.2$  a.u. which is within uncertainty limits when compared to our result. However, we propose to include more transitions for more accurate polarizability for this atom.

## 3. Dynamic dipole polarizability at imaginary frequency

We determine the dynamic polarizability values at different frequencies using the same method which has been used for evaluation of static polarizability and expect our values to be reliable. Dynamic dipole polarizabilities of ions and atoms at imaginary frequencies are presented in Figs. 1 and 2 respectively. Tabulated values of  $\alpha(\omega)$  for considered ions and atoms are given in the Supplemental Material (SM) [85]. With increase of frequency, polarizability decreases and reaches a small value. This trend is seen for both ions and atoms. From Fig. 1, one notices that at short frequency values, the dipole polarizabilities of  $Cd^+$  and  $Pb^+$  ions are comparable but with an increase in frequency, the polarizability of  $Cd^+$  decreases more rapidly as compared to the  $Pb^+$  ion. Dynamic polarizability of  $Cd^+$  is even lower than polarizability of  $Hg^+$  for  $\omega > 0.25$  a.u. For the  $Zn^+$  ion, the polarizability remains lowest throughout the frequency regime as compared to other ions. Similarly as shown in Fig. 2, for atoms the static polarizability of Zn is larger than Hg at short  $\omega$  but decreases rapidly for Zn as compared to Hg as  $\omega$  increases. For  $\omega > 1$  a.u., the polarizability values of Hg atom are more as compared to value of Zn and Cd and get closer to the polarizability of Pb

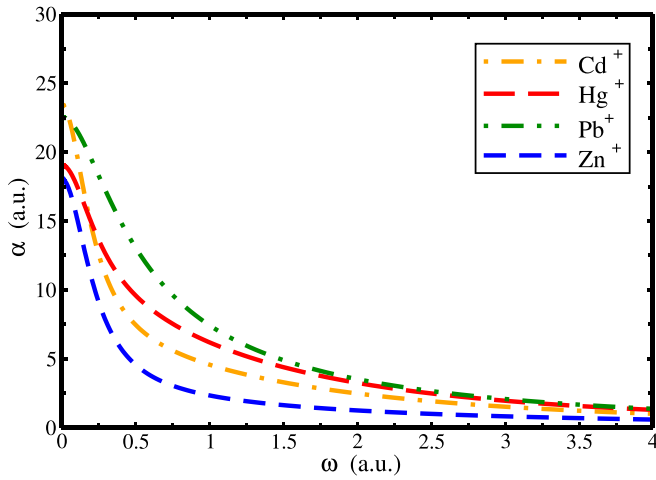


FIG. 1. Dynamic dipole polarizability  $\alpha$  (in a.u.) at imaginary frequencies of  $\text{Zn}^+$  (blue dashed curve),  $\text{Cd}^+$  (orange dotted dashed curve),  $\text{Hg}^+$  (red long dashed curve), and  $\text{Pb}^+$  (green double dotted dashed curve).

atom. If we compare polarizability values among atoms and ions, the lowest value throughout the considered frequency range is observed for the  $\text{Zn}^+$  ion whereas the largest values are for Pb and Cd atoms depending upon the frequency value. These values have been used for computing  $C_3$  dispersion coefficients as a function of separation distance as discussed in the next section.

### B. $C_3$ Coefficients for graphene

In this section, we present the dispersion coefficients between graphene layer with  $\Delta = 0.01$  eV and heavy elements as a function of separation distance. As shown in Figs. 3 and 4,  $C_3$  coefficients exhibit an inverted yield curve, i.e.,  $C_3$  values decrease with increase in distance. This nature of the curve is perceived for every element. Coefficients reach a value less than 0.1 a.u. for distance greater than 30 nm. Among

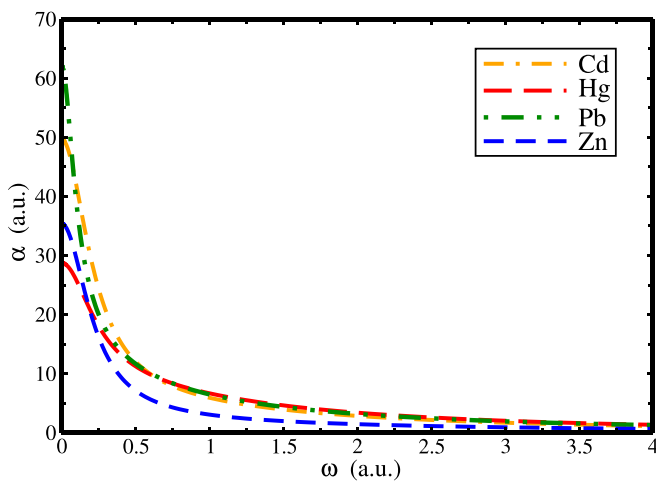


FIG. 2. Dynamic dipole polarizabilities  $\alpha$  (in a.u.) at imaginary frequencies of Zn (blue dashed curve), Cd (orange dotted dashed curve), Hg (red long dashed curve), and Pb (green double dotted dashed curve).

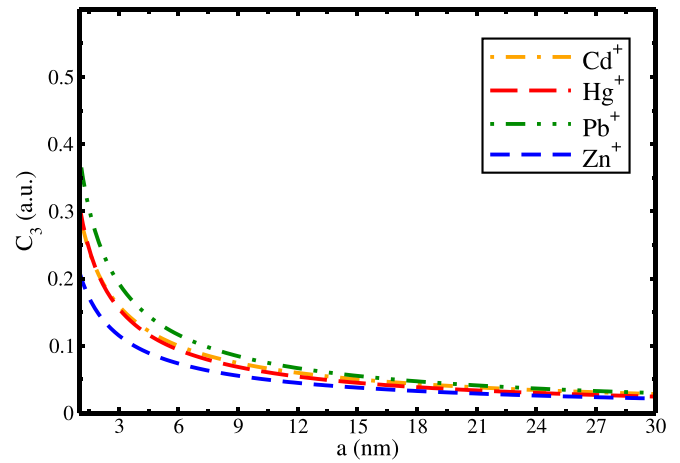


FIG. 3.  $C_3$  dispersion coefficients [in atomic units (a.u.)] for interaction of  $\text{Zn}^+$  (blue dashed curve),  $\text{Cd}^+$  (orange dotted dashed curve),  $\text{Hg}^+$  (red long dashed curve), and  $\text{Pb}^+$  (green double dotted dashed curve) with graphene layer as a function of separation distance  $a$  (in units of nm).

the ions, the largest  $C_3$  value is observed for  $\text{Pb}^+$  indicating stronger interaction with the graphene layer whereas  $\text{Zn}^+$  is least attracted as shown in Fig. 3. The respective  $C_3$  values for  $\text{Cd}^+$  and  $\text{Hg}^+$  ions are approximately the same. In the case of atoms, the large  $C_3$  values have been observed for Pb and Cd. At  $a = 1$  nm,  $C_3$  values are 0.564 and 0.561 for Pb and Cd atoms respectively. However, for separation distance  $a > 7$  nm, the difference in  $C_3$  coefficients for Pb and Cd become appreciable as displayed in Fig. 4. Zn and Hg are least attracted towards graphene. We did not find any literature on  $C_3$  coefficient values for interaction of heavy ions and atoms with carbon-based nanostructures. However, a comparison has been made based on physisorption of heavy atoms with previous DFT study [24]. For atoms, our results are in accordance with the DFT study for physisorption of heavy atoms on graphene layer [24]. In DFT study, vdW

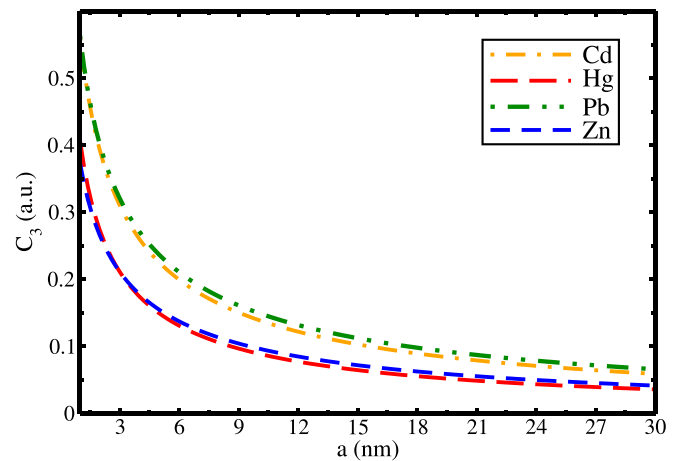


FIG. 4.  $C_3$  dispersion coefficients (in a.u.) for interaction of Zn (blue dashed curve), Cd (orange dotted dashed curve), Hg (red long dashed curve), and Pb (green double dotted dashed curve) atoms with graphene layer as a function of separation distance  $a$  (in units of nm).

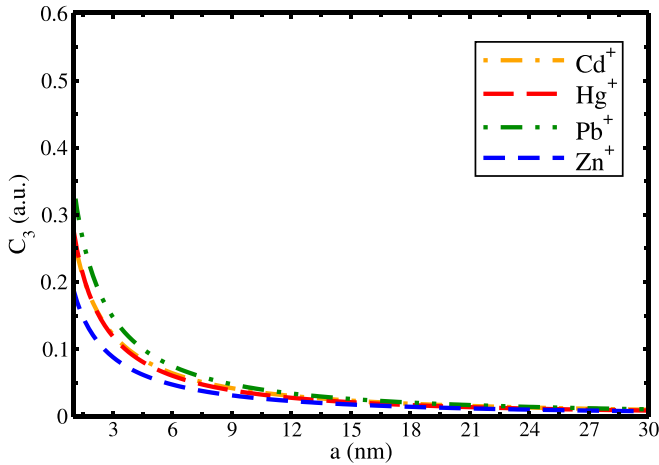


FIG. 5.  $C_3$  dispersion coefficients (in a.u.) for interaction between  $Zn^+$  (blue dashed curve),  $Cd^+$  (orange dotted dashed curve),  $Hg^+$  (red long dashed curve), and  $Pb^+$  (green double dotted dashed curve) with CNT of radius  $R = 6$  nm as a function of separation distance  $a$  (in units of nm).

interactions were described between the adsorbed atom and graphene and the strength of interactions was analyzed as a function of binding energy and charge transfer. The sequence of reducing binding energy for atoms was reported as  $Pb > Cd > Hg$  [24]. A similar trend is observed in our study where we have analyzed the strength of interaction on the basis of  $C_3$  values. Shtepliuk *et al.* [24] also studied the interaction of ions with graphene which resulted in chemisorption. Since our study provides the result for physisorption of microparticles on the material wall, we do not make a similar comparison for ions with Ref. [24]. It is important to note that these ions with large nuclear charge  $Z$ , which we have considered in the present work, are only singly charged, (i.e., having residual unity charge), thus the effect of the latter may not be too significant as compared to the Coulomb potential of heavier  $Z$  atoms in the calculation of matrix elements with which we have evaluated the  $C_3$  coefficients. The overall charge present on the ions leads to stronger Coulomb interactions as compared to weak vdW attractions with both graphene and CNT wall. The considered theory only provides the information regarding the weak vdW forces between the microparticle and considered substrates, which on comparison reveals that the selectivity and sensitivity of graphene and CNT to adsorb heavy atoms is more as compared to ions.

### C. $C_3$ Coefficients for CNT

In addition to graphene, we present the dispersion coefficient for CNT. Figures 5 and 6 represent the influence of separation distance on dispersion coefficients evaluated for a CNT of radius of 6 nm with heavy ions and atoms respectively. The  $C_3$  coefficients and hence interaction are dominant at smaller distances for all the elements. Similar to the case of graphene, the interaction is strongest for  $Pb^+$  and weakest for  $Zn^+$  with CNT. For atoms, a CNT offers a stronger potential to Pb and Cd atoms and weak potential to Zn atom.

The radius of the CNT has been an important parameter in hydrogen storage applications. It has been known that a larger

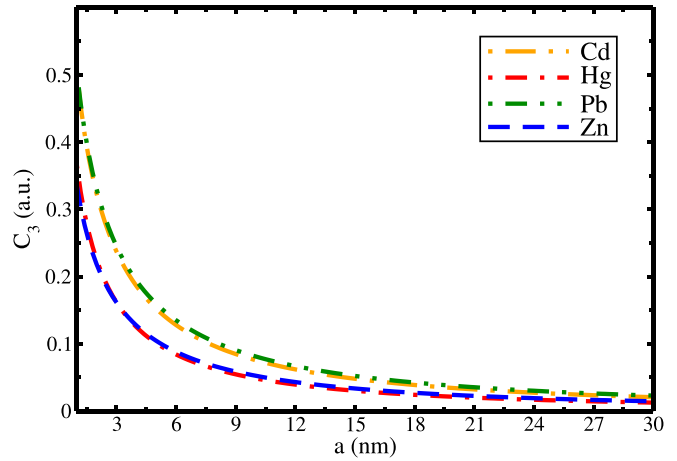


FIG. 6.  $C_3$  dispersion coefficients (in a.u.) for interaction between Zn (blue dashed curve), Cd (orange dotted dashed curve), Hg (red long dashed curve), and Pb (green double dotted dashed curve) atom with CNT of radius  $R = 6$  nm as a function of separation distance  $a$  (in units of nm).

radius of CNT imparts more gravimetric storage amount for hydrogen storage [86]. This motivated us to study the effect of radius of a CNT on  $C_3$  coefficients. Figure 7 demonstrates the effect of radius of a CNT on dispersion coefficients. CNTs with larger radius, i.e., 8 nm, has more potential to adsorb Pb species. This can be attributed due to greater exposure of carbon atoms towards ions and atoms with an increase in radius of CNTs.

### D. Dispersion coefficient for different gap parameter

Most of the previous studies on dispersion coefficients for graphene and CNT have taken the upper bound of the gap parameter  $\Delta$  as 0.1 eV [58,59,87]. The effect of this parameter was shown while studying the interactions between alkali-metal atoms and graphene layer in our previous study [88]. In the present work, we investigate the dependency of a gap

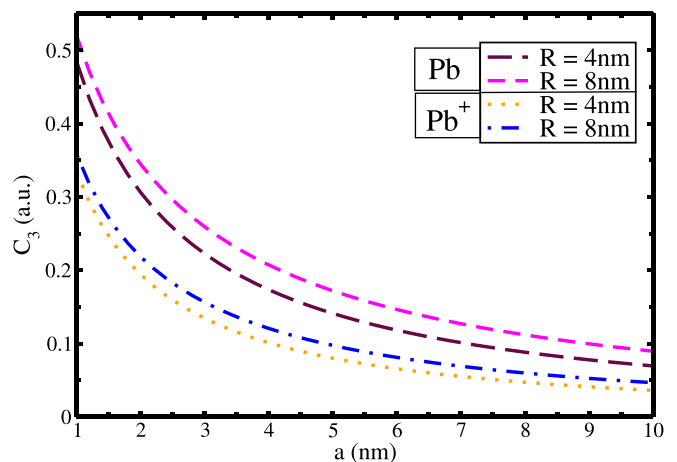


FIG. 7.  $C_3$  dispersion coefficients (in a.u.) for Pb atom (purple long dashed curve for  $R = 4$  nm, pink dashed curve for  $R = 8$  nm) and  $Pb^+$  ion (orange dotted curve for  $R = 4$  nm, blue dotted dashed curve for  $R = 8$  nm) for two different values of radius of CNTs.

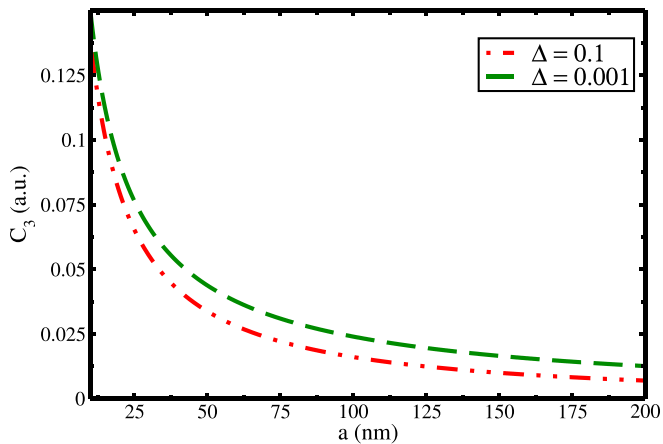


FIG. 8.  $C_3$  dispersion coefficients (in a.u.) for Pb atom with graphene wall as a function of separation distance  $a$  (in units of nm) for two different values of gap parameter  $\Delta$  (green long dashed curve for  $\Delta = 0.1$  and red double dotted dashed curve for  $\Delta = 0.001$ ) in eV.

parameter on  $C_3$  coefficients as well. Since the interaction of the Pb atom is most prominent among all the elements considered in this study, we present the effect of gap parameter on dispersion coefficient between the graphene layer and the Pb atom. To find the influence of  $\Delta$ , we choose two different values of  $\Delta$  as 0.1 and 0.001 eV. The dispersion coefficient is seen to increase by only 0.41% at a separation distance of  $a = 1$  nm with a decrease of  $\Delta$  from 0.1 to 0.001 eV. When investigating the same at a larger separation distance of 100 and 200 nm, the percentage increase in the  $C_3$  coefficient is found to be about 49 and 80 respectively. This is presented in Fig. 8 where the gap between the two curves of different  $\Delta$  increases with an increase in separation.

#### E. Comparison of graphene and CNT

Since we studied the dispersion interactions for two carbon-based materials, it is important to compare these two

and prognosticate a better material for physisorption of these heavy elements. Both the materials show a similar trend for adsorption of ions and atoms and are selective towards Pb and Cd atoms. The weakest interaction is observed for the  $Zn^+$  ion. However, when a comparison is drawn for the interaction of a microparticle with graphene and a CNT it was found that graphene provided a larger  $C_3$  coefficient value and hence stronger interaction as compared to CNT. Contrarily, a number of studies can be found in the literature where the CNT is widely accepted for physisorption applications as compared to graphene. The reason behind this is the ability of experimentalists to tailor the properties and structure of CNTs with ease whereas the bulk preparation of pristine graphene is a major bottleneck that needs a direction [89].

#### IV. CONCLUSION

To conclude, we have probed the dispersion coefficients for  $Zn^+$ ,  $Cd^+$ ,  $Hg^+$ ,  $Pb^+$ , Zn, Cd, Hg, and Pb with graphene and CNT walls. We have provided the dynamic dipole polarizability values for both heavy ions and atoms using the sum-over-states approach. The interactions between heavy elements and material wall as a result of dispersion  $C_3$  coefficients has been found to be maximum for the Pb atom and ion at short separations. The result of interaction studies by our methodology is in agreement with interactions studied by DFT for heavy atoms. The CNT also shows the potential for interaction of heavy elements following a similar trend as that of graphene. We also deduce that graphene is more sensitive for interaction of the considered elements as compared to CNTs. The obtained results could be useful for the formation of highly sensitive and selective sensors for detection of heavy ions and atoms.

#### ACKNOWLEDGMENTS

B.A. is thankful to the SERB-TARE (Research Grant No. TAR/2020/000189), New Delhi, India. R.S. is thankful for Research Grant No. CRG/2020/005597 by SERB-DST, New Delhi, India.

- 
- [1] S. Morais, F. G. E. Costa, and M. D. L. Pereira, *Environmental Health - Emerging Issues and Practice* (InTechOpen, Rijeka, New Croatia, 2012), Vol. 10.
  - [2] M. Jaishankar, T. Tseten, N. Anbalagan, B. B. Mathew, and K. N. Beeregowda, *Interdiscip. Toxicol.* **7**, 60 (2014).
  - [3] E. Tovar-Sánchez, I. Hernández-Plata, M. S. Martínez, L. Valencia-Cuevas, and P. M. Galante, in *Heavy Metals*, edited by H. El-Din M. Saleh and R. F. Aglan (IntechOpen, Rijeka, 2018), Vol. 383, Chap. 21, pp. 383–399.
  - [4] P. C. Nagajyoti, K. D. Lee, and T. Sreekanth, *Environ. Chem. Lett.* **8**, 199 (2010).
  - [5] H. Ali and E. Khan, *Hum. Ecol. Risk Assess.* **25**, 1353 (2019).
  - [6] L. M. Plum, L. Rink, and H. Haase, *Int. J. Environ. Res. Public Health* **7**, 1342 (2010).
  - [7] M. Mahurpawar, *Int. J. Res. Granthaalayah* **3**, 1 (2015).
  - [8] M. A. Assi, M. N. M. Hezme, M. Y. M. Abd Wahid Haron, and M. A. R. Sabri, *Vet. World* **9**, 660 (2016).
  - [9] R. A. Wuana and F. E. Okieimen, *Int. Sch. Res. Notices* **2011**, 402647 (2011).
  - [10] M. K. Uddin, *Chem. Eng. J.* **308**, 438 (2017).
  - [11] M. Hong, L. Yu, Y. Wang, J. Zhang, Z. Chen, L. Dong, Q. Zan, and R. Li, *Chem. Eng. J.* **359**, 363 (2019).
  - [12] A. E. Burakov, E. V. Galunin, I. V. Burakova, A. E. Kucherova, S. Agarwal, A. G. Tkachev, and V. K. Gupta, *Ecotoxicol. Environ. Saf.* **148**, 702 (2018).
  - [13] C. P. Poole, Jr and F. J. Owens, *Introduction to Nanotechnology* (John Wiley & Sons, Hoboken, New Jersey, 2003).
  - [14] D. G. Papageorgiou, I. A. Kinloch, and R. J. Young, *Prog. Mater. Sci.* **90**, 75 (2017).
  - [15] J. Phiri, L. S. Johansson, P. Gane, and T. Maloney, *Composites, Part B* **147**, 104 (2018).



- [16] Z. Fan, D. Z. Y. Tng, C. X. T. Lim, P. Liu, S. T. Nguyen, P. Xiao, A. Marconnet, C. Y. Lim, and H. M. Duong, *Colloids Surf. A* **445**, 48 (2014).
- [17] L. Falkovsky, *J. Phys. Conf. Ser.* **129**, 012004 (2008).
- [18] L. Kavan, J.-H. Yum, and M. Graetzel, *Phys. Status Solidi B* **250**, 2643 (2013).
- [19] B. H. Nguyen and V. H. Nguyen, *Adv. Nat. Sci.: Nanosci. Nanotechnol.* **7**, 023002 (2016).
- [20] A. Grigorenko, M. Polini, and K. Novoselov, *Nat. Photonics* **6**, 749 (2012).
- [21] S. Dervin, D. D. Dionysiou, and S. C. Pillai, *Nanoscale* **8**, 15115 (2016).
- [22] S. Mao, G. Lu, and J. Chen, *J. Mater. Chem. A* **2**, 5573 (2014).
- [23] A. C. Torres-Dias, T. F. Cerqueira, W. Cui, M. A. Marques, S. Botti, D. Machon, M. A. Hartmann, Y. Sun, D. J. Dunstan, and A. San-Miguel, *Carbon* **123**, 145 (2017).
- [24] I. Shteplyuk, N. M. Caffrey, T. Iakimov, V. Khranovskyy, I. A. Abrikosov, and R. Yakimova, *Sci. Rep.* **7**, 3934 (2017).
- [25] M. Yusuf, F. Elfgi, S. A. Zaidi, E. Abdullah, and M. A. Khan, *RSC Adv.* **5**, 50392 (2015).
- [26] M. U. Niemann, S. S. Srinivasan, A. R. Phani, A. Kumar, D. Y. Goswami, and E. K. Stefanakos, *J. Nanomater.* **2008**, 950967 (2008).
- [27] D. Henwood and J. D. Carey, *Phys. Rev. B* **75**, 245413 (2007).
- [28] J. Z. Ou, W. Ge, B. Carey, T. Daeneke, A. Rotbart, W. Shan, Y. Wang, Z. Fu, A. F. Chrimes, W. Wlodarski *et al.*, *ACS nano* **9**, 10313 (2015).
- [29] A. H. Mashhadzadeh, M. Fathalian, M. G. Ahangari, and M. Shahavi, *Mater. Chem. Phys.* **220**, 366 (2018).
- [30] I. Petrushenko and K. Petrushenko, *Surf. Interfaces* **17**, 100355 (2019).
- [31] P. Lazić, Ž. Crljen, R. Brako, and B. Gumhalter, *Phys. Rev. B* **72**, 245407 (2005).
- [32] P. L. Silvestrelli, A. Ambrosetti, S. Grubisić, and F. Ancilotto, *Phys. Rev. B* **85**, 165405 (2012).
- [33] H. Abdelsalam, N. Teleb, I. Yahia, H. Zahran, H. Elhaes, and M. Ibrahim, *J. Phys. Chem. Solids* **130**, 32 (2019).
- [34] O. A. Oyetade, A. A. Skelton, V. O. Nyamori, S. B. Jonnalagadda, and B. S. Martincigh, *Sep. Purif. Technol.* **188**, 174 (2017).
- [35] X.-P. Jiang, F. Toigo, and M. W. Cole, *Surf. Sci.* **145**, 281 (1984).
- [36] S. Rauber, J. R. Klein, M. W. Cole, and L. Bruch, *Surf. Sci.* **123**, 173 (1982).
- [37] E. Zaremba and W. Kohn, *Phys. Rev. B* **13**, 2270 (1976).
- [38] G. L. Klimchitskaya and V. M. Mostepanenko, *Universe* **6**, 150 (2020).
- [39] M. Bordag, B. Geyer, G. L. Klimchitskaya, and V. M. Mostepanenko, *Phys. Rev. B* **74**, 205431 (2006).
- [40] M. Bordag, G. L. Klimchitskaya, U. Mohideen, and V. M. Mostepanenko, *Advances in the Casimir Effect* (Oxford University Press, Oxford, 2009), Vol. 145.
- [41] Y. J. Lin, I. Teper, C. Chin, and V. Vuletić, *Phys. Rev. Lett.* **92**, 050404 (2004).
- [42] V. B. Bezerra, G. L. Klimchitskaya, V. M. Mostepanenko, and C. Romero, *Phys. Rev. A* **78**, 042901 (2008).
- [43] J. Tao and A. M. Rappe, *Phys. Rev. Lett.* **112**, 106101 (2014).
- [44] A. O. Caride, G. L. Klimchitskaya, V. M. Mostepanenko, and S. I. Zanette, *Phys. Rev. A* **71**, 042901 (2005).
- [45] B. Arora and B. K. Sahoo, *Phys. Rev. A* **89**, 022511 (2014).
- [46] A. Derevianko, W. R. Johnson, M. S. Safronova, and J. F. Babb, *Phys. Rev. Lett.* **82**, 3589 (1999).
- [47] G. Łach, M. DeKieviet, and U. D. Jentschura, *Int. J. Mod. Phys. A* **25**, 2337 (2010).
- [48] S. Dutt, S. Singh, A. Mahajan, B. Arora, and B. Sahoo, *Phys. Scr.* **95**, 095506 (2020).
- [49] E. V. Blagov, G. L. Klimchitskaya, and V. M. Mostepanenko, *Phys. Rev. B* **75**, 235413 (2007).
- [50] K. Kaur, J. Kaur, B. K. Sahoo, and B. Arora, *Phys. Lett. A* **380**, 3366 (2016).
- [51] M. Fichet, G. Dutier, A. Yarovitsky, P. Todorov, I. Hamdi, I. Maurin, S. Saltiel, D. Sarkisyan, M.-P. Gorza, D. Bloch *et al.*, *Europhys. Lett.* **77**, 54001 (2007).
- [52] S. Lepoutre, H. Jelassi, V. Lonij, G. Tréneç, M. Büchner, A. D. Cronin, and J. Vigué, *Europhys. Lett.* **88**, 20002 (2009).
- [53] V. P. Lonij, C. E. Klauss, W. F. Holmgren, and A. D. Cronin, *J. Phys. Chem. A* **115**, 7134 (2011).
- [54] P. Schneeweiss, M. Gierling, G. Visanescu, D. Kern, T. Judd, A. Günther, and J. Fortágh, *Nat. Nanotechnol.* **7**, 515 (2012).
- [55] E. V. Blagov, G. L. Klimchitskaya, and V. M. Mostepanenko, *Phys. Rev. B* **71**, 235401 (2005).
- [56] G. L. Klimchitskaya and V. M. Mostepanenko, *Phys. Rev. B* **91**, 045412 (2015).
- [57] K. Kaur, B. Arora, and B. K. Sahoo, *Phys. Rev. A* **92**, 032704 (2015).
- [58] Y. V. Churkin, A. B. Fedortsov, G. L. Klimchitskaya, and V. A. Yurova, *Int. J. Mod. Phys. A* **26**, 3958 (2011).
- [59] B. Arora, H. Kaur, and B. Sahoo, *J. Phys. B: At. Mol. Opt. Phys.* **47**, 155002 (2014).
- [60] A. K. Geim, *Science* **324**, 1530 (2009).
- [61] M. Bordag, I. V. Fialkovsky, D. M. Gitman, and D. V. Vassilevich, *Phys. Rev. B* **80**, 245406 (2009).
- [62] A. H. Castro Neto, F. Guinea, N. M. R. Peres, K. S. Novoselov, and A. K. Geim, *Rev. Mod. Phys.* **81**, 109 (2009).
- [63] M. Bordag, U. Mohideen, and V. M. Mostepanenko, *Phys. Rep.* **353**, 1 (2001).
- [64] B. Arora, D. K. Nandy, and B. K. Sahoo, *Phys. Rev. A* **85**, 012506 (2012).
- [65] A. Kramida, Yu. Ralchenko, J. Reader, and NIST ASD Team, NIST Atomic Spectra Database (ver. 5.8), available at <https://physics.nist.gov/asd> [2021, February 24], National Institute of Standards and Technology, Gaithersburg, MD, 2021.
- [66] M. Safronova and W. Johnson, *Adv. At. Mol. Opt. Phys.* **55**, 191 (2008).
- [67] S. A. Blundell, W. R. Johnson, and J. Sapirstein, *Phys. Rev. A* **43**, 3407 (1991).
- [68] P. Jönsson, G. Gaigalas, J. Bieroń, C. F. Fischer, and I. Grant, *Comput. Phys. Commun.* **184**, 2197 (2013).
- [69] N. Shukla, B. Arora, L. Sharma, and R. Srivastava, *Phys. Rev. A* **102**, 022817 (2020).
- [70] S. A. Blundell, W. R. Johnson, Z. W. Liu, and J. Sapirstein, *Phys. Rev. A* **40**, 2233 (1989).
- [71] M. Kompitsas, C. Baharis, and Z. Pan, *J. Opt. Soc. Am. B* **11**, 697 (1994).
- [72] U. I. Safronova, M. S. Safronova, and W. R. Johnson, *Phys. Rev. A* **71**, 052506 (2005).
- [73] B. K. Sahoo, S. Majumder, H. Merlitz, R. Chaudhuri, B. Das, and D. Mukherjee, *J. Phys. B: At. Mol. Phys.* **39**, 355 (2005).
- [74] M. Iliaš and P. Neogrady, *Chem. Phys. Lett.* **309**, 441 (1999).

- [75] C.-B. Li, Y.-M. Yu, and B. K. Sahoo, *Phys. Rev. A* **97**, 022512 (2018).
- [76] T. Gould and T. Bucko, *J. Chem. Theory Comput.* **12**, 3603 (2016).
- [77] B. Arora, M. S. Safronova, and C. W. Clark, *Phys. Rev. A* **76**, 052509 (2007).
- [78] D. Goebel, U. Hohm, and G. Maroulis, *Phys. Rev. A* **54**, 1973 (1996).
- [79] D. Goebel and U. Hohm, *Phys. Rev. A* **52**, 3691 (1995).
- [80] Y. Singh and B. K. Sahoo, *Phys. Rev. A* **91**, 030501(R) (2015).
- [81] L. Ma, J. Indergaard, B. Zhang, I. Larkin, R. Moro, and W. A. de Heer, *Phys. Rev. A* **91**, 010501(R) (2015).
- [82] V. Kellö and A. J. Sadlej, *Theor. Chim. Acta* **91**, 353 (1995).
- [83] V. Pershina, A. Borschevsky, E. Eliav, and U. Kaldor, *J. Chem. Phys.* **128**, 024707 (2008).
- [84] A. Ye and G. Wang, *Phys. Rev. A* **78**, 014502 (2008).
- [85] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevA.104.012806> for tabulated values of polarizability of considered ions and atoms at imaginary frequencies.
- [86] C.-I. Weng, S.-P. Ju, K.-C. Fang, and F.-P. Chang, *Comput. Mater. Sci.* **40**, 300 (2007).
- [87] M. Chaichian, G. L. Klimchitskaya, V. M. Mostepanenko, and A. Tureanu, *Phys. Rev. A* **86**, 012515 (2012).
- [88] K. Kaur, J. Kaur, B. Arora, and B. K. Sahoo, *Phys. Rev. B* **90**, 245405 (2014).
- [89] P. Petit, C. Mathis, C. Journet, and P. Bernier, *Chem. Phys. Lett.* **305**, 370 (1999).