Comments and Addenda

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Variation of the Ratio H_{c3}/H_{c2} in the Immediate Vicinity of T_c ⁺

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We have measured the ratio H_{c3}/H_{c2} for tin-rich Sn-In alloys and for indium-rich In-Bi alloys in the immediate vicinity of T_c . For the Sn-In system this ratio decreases as T_c is approached. Similar behavior has been recently reported by Ostenson and Finnemore for pure Nb. The In-Bi alloys, however, show a marked increase in H_{c3}/H_{c2} as the critical temperature is approached.

R ECENTLY, Ostenson and Finnemore¹ have reported a substantial decrease in the ratio of the surface critical field H_{c3} to the bulk critical field of the mixed state H_{c2} for pure Nb in the immediate vicinity of T_c . We report in this paper measurements of H_{c3} and H_{c2} very close to T_c for type-II superconductors of the Sn-rich Sn-In system and the In-rich In-Bi system. A decrease in H_{c3}/H_{c2} , similar to that reported for Nb, is observed in the Sn-In alloys as T_c is approached. The measured alloys of the In-Bi system, however, show

FIG. 1. Typical magnetic transition for $\text{Sn}_{0.95}$ In_{0.05} alloy at a reduced temperature $t=0.943$. Upper and lower curves correspond to H_{c3} and H_{c2} , respectively. The upper curve has been measured for **H** par perpendicular.

† Supported by the Advanced Research Projects Agency and the Army Research Office, Durham, N. C.

¹ J. E. Ostenson and D. K. Finnemore, Phys. Rev. Letters 22,

188 (1969).

an increase in the ratio H_{c3}/H_{c2} , which becomes higher (~ 2.5) than the theoretical value (1.695).²

We have measured Sn-In alloys with 5- and 6-at. $\%$ In, and In-Bi alloys with 2.9- and 2-at. $\%$ Bi. The samples, thin plates 0.1 mm thick, 4 mm wide, and 10 mm long, were prepared by rolling the alloy between stainless-steel rollers. The alloys were prepared by melting and vibrating the components for several hours. The rolled samples were annealed for 24 h before performing the measurements. The critical fields were determined by measuring the mutual inductance of a pair of coils with the sample between them. The ac magnetic field produced by the primary of those coils was kept smaller than 0.1 Oe. The frequency used was 1000 Hz. The surface critical field H_{c3} was determined with an external dc magnetic field H parallel to the large surface of the sample, while H_{c2} was assumed to be the critical field for H perpendicular to that surface. We believe that the surface superconductivity at the sample edges is too small to affect our measurements. This belief is supported by the fact that for $T \ll T_c$ values of H_{c3}/H_{c2} quite close to the theoretical one are obtained by this method. In the course of the measurements, the amplitude of the ac probing field was changed by a factor of 2 and its frequency reduced to 200 Hz. No dependence of H_{c3} and H_{c2} on these parameters was found. The change in mutual inductance was measured with a conventional bridge³ and plotted on an $x-y$ recorder as a function of dc magnetic field. The field H was swept at a constant rate chosen slow enough to give retraceable curves for increasing and decreasing fields.

² D. Saint James and P.-G. de Gennes, Phys. Letters 7, 306

^{(1963).&}lt;br>
⁸ W. L. Pillinger, P. S. Jastram, and J. G. Daunt, Rev. Sci.

Instr. 29, 159 (1958).

FIG. 2. Values of H_{c3} , H_{c2} , and the ratio H_{c3}/H_{c2} for Sn_{0.95} $In_{0.05}$ alloy. The critical temperature chosen for this alloy was $T_c = 3.625$.

Possible errors due to uncertainties in the position of the sample' were minimized by measuring the nominal ratio H_{c3}/H_{c2} for various azimuthal orientations of H. The angle for which a maximum in $H_{\text{e}3}/H_{\text{e}2}$ occurs (**H** parallel to the sample surface) could be determined to better than 1[°]. In a similar manner, we also checked the absence of errors due to the earth's magnetic field.

We show in Fig. 1 the magnetic transition observed for a $Sn_{0.95}$ In_{0.05} alloy at a reduced temperature $t=0.943$. The shape of this transition is typical of all Sn-In samples measured; the transition of the In-Si alloys is somewhat wider. The three most obvious criteria for the definition of the critical field correspond to the points labeled A, 8, and C in Fig. 1, i.e. , the middle point in the vertical axis (A), the linear extrapolation to join the mutual inductance of the normal state (B) , and the rather poorly defined tail point (C) at which all traces of superconductivity seem to have disappeared. We have analyzed our data with these three criteria and found that, to a constant scaling factor, the functional dependence of H_{c2} and H_{c3} on temperature is independent of the criterion used. Therefore, the ratio H_{c3}/H_{c2} does not depend on this criterion, provided the same criterion is used for both H_{c3} and H_{c2} . If different criteria are used for H_{c3} and H_{c2} , the ratio H_{c3}/H_{c2} is changed by a constant factor of the order of 1. The use of the same criterion is necessary in order to obtain a value of H_{c3}/H_{c2} close to the theoretical ratio² for $T \ll T_c$. For the data presented here, we have chosen the points B as the critical fields.

We show in Fig. 2 the values of H_{c3} and H_{c2} obtained for $Sn_{0.95}$ In_{0.05} near T_c . The insert in this figure shows

FIG. 3. Values of H_{c3} , H_{c2} , and the ratio H_{c3}/H_{c2} for In_{0.98} Bi_{0.02}. The critical temperature chosen was $T_c = 3.845$.

the temperature dependence of the ratio H_{c3}/H_{c2} for this alloy, which is rather similar to the corresponding ratio observed by Ostenson and Finnemore' for Nb. The decrease in H_{c3}/H_{c2} starts at the reduced temperature $t=0.92$, while H_{c3} and H_{c2} can be fitted to straight lines up to $t = 0.98$. The decrease in H_{c3}/H_{c2} is associated with the fact that the straight lines do not intercept the horizontal axis at the same point.

Figure 3 shows H_{c3} , H_{c2} , and H_{c3}/H_{c2} for an In-Bi alloy (2-at.% Bi). The ratio H_{c3}/H_{c2} increases for $t>0.91$. For $0.91 \le t \le 0.97$ this increase is due to the fact that the linear variations of H_{c3} and H_{c2} do not extrapolate to the same temperature on the horizontal axis. The observation is similar to that mentioned above for the Sn-In alloy, but in Fig. 3 an increase in H_{c3}/H_{c2} results as T_c is approached due to the fact that H_{c3} extrapolates linearly to a temperature higher than H_{c2} . For $t>0.97$, H_{c3}/H_{c2} keeps increasing, but H_{c3} and H_{c2} do not vary linearly with T. The temperature dependence of H_{c3}/H_{c2} shown in Figs. 2 and 3 seems to be characteristic of the corresponding alloy systems. Samples of $Sn_{0.94}$ In_{0.06} and $In_{0.971}$ $Bi_{0.029}$ gave essentially the same type of dependence of H_{c3}/H_{c2} on temperature as that shown in Figs. 2 and 3, respectively.

A theoretical interpretation of the temperature variation of H_{c3}/H_{c2} near T_c reported here is not available. Further experiments to clarify the concentration dependence of this variation for these and other alloy systems are in progress.

Ãote added in proof: Recently, we have measured H_{c3}/H_{c2} for Sn-In alloys as a function of sample thickness. We observed that the anomalies in H_{c3}/H_{c2} discussed here disappear for thickness ≈ 0.02 mm. The detailed results will be published elsewhere.

⁴ The surface critical field is known to decrease drastically if H is not exactly parallel to the surface. See, for instance, P. Burger, G. Deutscher, E. Guyon, and A. Martinet, Phys. Rev. 137, A853 (1965).